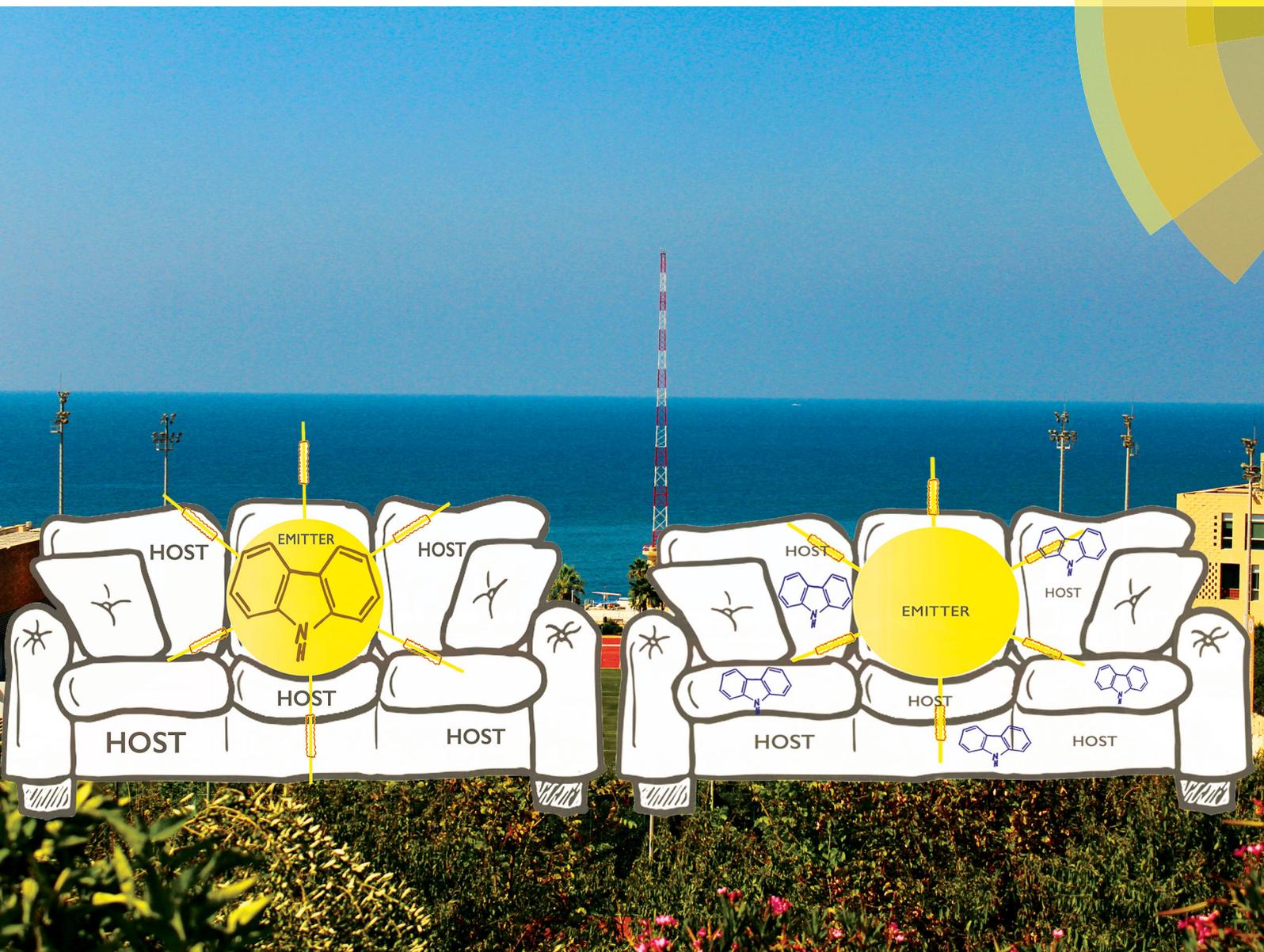


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REVIEW ARTICLE

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Perspective on carbazole-based organic compounds as emitters and hosts in TADF applications

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The field of organic light-emitting devices (OLEDs) has undergone a remarkable journey since its discovery by Tang and VanSlyke with an alternation of utilizing fluorescence and phosphorescence as the emitting vehicle. The latest generation of thermally activated delayed fluorescence (TADF) materials harvest triplet excited states back into the singlet manifold. This booming field has yielded a large array of new compounds as both emitters and hosts. This review is limited to TADF emitters utilizing at least one carbazole unit as a donor and organized according to the various acceptor building blocks such as cyanophenyl, pyridine, biphenyls, anthraquinone, phenyl(pyridine-2-yl)methanone, benzophenone, xanthon, sulfones, triazines, benzils, dicyanopyrazines, diazatriphenylene, and others. A survey of carbazole-containing host materials follows. Density functional theory (DFT) has carved out a significant role in allowing the theoretical prediction of ground state properties for materials applied in OLED technology. Time-dependent DFT extends the reach to model excited state properties important to rationalize the light-output in OLED technology. For TADF, two fundamental factors are of interest: significant separation of frontier molecular orbitals and minimal singlet–triplet energy gap (ΔE_{ST}). In this review, the utilization of DFT calculations to optimize geometries for the visualization of frontier molecular orbital separation was surveyed to find that the B3LYP/6-31G(d) level of theory is the overwhelmingly used approach. In addition, we review the more in-depth approaches to utilizing DFT and time-dependent DFT (TD-DFT) with optimized percentage Hartree–Fock (OHF) and long-range corrected hybrid functionals, tuning procedures and others in an attempt to best quantify the size of ΔE_{ST} as well as the nature of the triplet state as locally excited state (LE) and charge-transfer state (CT).

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OLEDs – an introduction

In an organic light-emitting device (OLED), light is generated by recombination of electrically generated, bound electron-hole pairs, called excitons. Excitons are generally formed *via* one of two operating mechanisms, direct charge carrier recombination of injected charge carriers (electrons from the cathode and holes from the anode) or host-to-dopant energy transfer under the application of an energy transfer mechanism (Förster or Dexter). Spin statistics thereby predicts exciton formation as 25% of the time as singlet and 75% of the time as triplet.¹ Singlet excitons produce electroluminescence by a fast process called fluorescence. Triplet excitons produce electroluminescence by a slow process called phosphorescence. Emission is thereby governed by selection rules known as El-Sayed rules.² Specifically in the case of purely organic materials, slow radiative decay rates are observed, which effectively compete only at very low temperature, generally 77 K. At this temperature, competing non-radiative decay processes are slowed down and phosphorescence is observed. Thus, at ambient temperature, these triplet excitons are generally lost by non-radiative decay processes.

The development of OLED technology has gained tremendous strides. The first generation devices were based on harvesting the emission generated from singlet excitons.³ Hence, the first generation OLED devices were inherently limited to an internal maximum quantum efficiency of 25% since triplet excitons were lost predominantly by non-radiative transitions as outlined above.⁴ The maximum theoretical external quantum efficiency is <5% in fluorescent OLEDs, which can be calculated using eqn (1).^{5,6}

$$\eta_{\text{ext}} = \eta_{\text{int}}\eta_{\text{out}} = \gamma\eta_{\text{ST}}\Phi_{\text{PL}}\eta_{\text{out}} \quad (1)$$

With η_{int} being the internal electroluminescence quantum efficiency, η_{out} the light outcoupling efficiency (0.2–0.3), γ the charge balance factor, η_{ST} the fraction of radiative excitons, *i.e.* 0.25 for fluorescence-based OLEDs, and Φ_{PL} photoluminescence quantum yield of emitter.⁶

The second generation OLED devices were the result of a major breakthrough in improving the external quantum efficiency, achieved through the introduction of transition metal complexes. Therein, phosphorescence-based OLEDs named PHOLEDs based on organometallic phosphors containing noble metals were utilized to harvest the 75% triplet excitons. These PHOLED devices achieved close to 100% internal quantum efficiency due to singlet-triplet mixing through effective spin-orbit coupling.^{7,8} However, transition metal complexes⁹ based on metals such as iridium,¹⁰ platinum,^{11–13} osmium,¹⁴ europium,¹⁵ and ruthenium¹⁶ are costly and scarce resources of low abundance, and are as such not sustainable for mass consumer goods applications. In addition, upon disposal, these materials are potentially harmful for the environment.¹⁷ Significant drawbacks were observed for PHOLEDs due to exciton annihilation among the long-lived triplet states (μs to ms), weakness of the chemical bonds to the metals leading to decomposition, and limited molecular design opportunity due to the restriction set forth by the nature of the geometry of the transition metal complexes.

Both, triplet-triplet annihilation (TTA) and thermally activated delayed fluorescence (TADF) allow dark triplet states to be harnessed by repopulating singlet excitons.¹⁸ TTA is also called P-type delayed fluorescence as it was first observed in pyrene. TADF is also called E-type delayed fluorescence as it was first observed in eosin. The application of TTA for OLEDs allowed the theoretical limit of 25% to be exceeded; however, an inherent limitation for the formation of radiative excitons was shown to be $(25\% + 75\% \times 0.5) = 62.5\%$.¹⁹ The third generation of OLED technology arose from TADF. In TADF materials, the absorption of environmental thermal energy leads to augmentation of the population of the electro-generated emissive singlet excitons from electro-generated, non-radiative triplet excitons^{5,20} by a process called reverse intersystem crossing (RISC) followed by emission. This ensuing light emission is thus, by majority, delayed fluorescence.²¹ RISC occurs effectively if the singlet-triplet energy gap (ΔE_{ST}) is small, eqn (2).²²

$$\Phi_{\text{RISC}} \propto k_{\text{RISC}} = k_{\text{RISC}}' \exp[-\Delta E_{\text{ST}}/(k_{\text{B}}T)] \quad (2)$$

With Φ_{RISC} quantum yield of RISC, k_{RISC} rate constant of RISC; k_{B} Boltzmann constant; T absolute temperature. Thus, materials of this third generation of OLEDs, exhibiting TADF, have augmented fluorescence due to harnessing of triplet excitons into the emissive singlet manifold and this route enables a near 100% internal maximum quantum efficiency to be achieved.³

Notably, triplet to singlet transitions are forbidden when considering the zero order approximation. TADF herein takes advantage of first order mixing that includes spin-orbit and electron spin interactions. The spin-orbit coefficient λ relates spin orbit interaction (H_{SO}) to ΔE_{ST} as described by eqn (3).²³ Spin orbit coupling thereby means an electromagnetic interaction between the electron's spin and the magnetic field created by the electron orbiting around the nucleus, the small ΔE_{ST} maximizes spin-orbit coupling and thus favors the transition.

$$\lambda = \frac{H_{\text{SO}}}{\Delta E_{\text{ST}}} = \frac{\langle T_1 | H_{\text{SO}} | S_1 \rangle}{E_{S_1} - E_{T_1}} \quad (3)$$

Among organic molecules, those with $n-\pi^*$ transitions such as aromatic ketones have small ΔE_{ST} of around 0.1–0.2 eV.²⁴ However, these aromatic ketones do not show fluorescence. Aromatic hydrocarbons show $\pi-\pi^*$ transitions and have large singlet triplet gaps of >1.0 eV and therefore do not directly exhibit TADF behavior. A small exchange integral ($J_{\text{if}} = \frac{1}{2}\Delta E_{\text{ST}}$) for S_1 is proportional to the orbital overlap integral ($S_{\text{if}} = \langle \phi_i | \phi_f \rangle$) and the orbital overlap integral is related to the spatial overlap of orbitals.²⁵

Materials with TADF behavior are designed under application of several design principles. The design principles for TADF materials include the separation of relevant transition orbitals in order to minimize ΔE_{ST} ¹⁸ as outlined above. Often this statement is reduced to the separation of the highest occupied molecular orbital (HOMO) and the lowest unoccupied molecular orbital (LUMO). This, of course, implies the assumption that S_1 is a HOMO-LUMO transition, which, depending on the structure, is not necessarily the case. A separation of overlap



to minimize ΔE_{ST} can be achieved through a twisted intermolecular charge transfer (TICT)^{26,27} framework, which allows only a small relaxation energy due to steric hindrance between donor and acceptor moieties and localization of the triplet exciton or the enlargement of the distance separating electrons ($r_1 - r_2$) of the columbic interaction operator.²⁸ Donor-acceptor (D-A) compounds trigger intermolecular charge transfer (ICT) states, whereby an employed change in solvent polarity modulates the emission properties by specifically stabilizing the charge-transfer (CT) state over the locally excited (LE) state according to eqn (4).²⁹

$$\Psi_{ICT} = c_1|D^*A\rangle_{Loc} + c_2|DA^*\rangle_{Loc} + c_3|D^+A^-\rangle_{CT} \quad (4)$$

It has to be noted here that the molecular configurations involved in the transition ($S_1 \leftarrow T_1$) may not necessarily be the same for the excited singlet and excited triplet state. For example, TADF may occur as a RISC process between a locally excited triplet state (3LE) of a donor and the charge-transfer state 1CT of the TADF emitter.²⁹ In general, molecules with ΔE_{ST} of < 100 meV are particularly suitable for TADF applications.²¹

Moving to the device preparation point of view, the process of determining ΔE_{ST} from solution data by the energy differences of the singlet excited state derived from room temperature fluorescence spectra and triplet excited state from low temperature phosphorescence leads to a significantly different value for ΔE_{ST} in comparison to the data obtained in a host film at room temperature after extracting the barrier to RISC as shown by Lee *et al.*^{30a} and Santos *et al.*^{30b} These ΔE_{ST} differences are particularly important to consider, when choosing experimental data for benchmark studies using computational approaches.

With enough thermal energy present in the system, the thermally activated RISC pathway leads to internal quantum efficiencies (IQEs) of unity considering the singlet state.

The design principle also needs to include the consideration that at the same time, the S_1 state needs to remain radiative. The radiative decay rate (k_r) constant is dependent on oscillator strength, *i.e.* overlap integral. Thus, a minimal orbital overlap²¹ needs to be maintained between the relevant orbital energy levels to ensure that the high k_r of the radiative decay rates is $> 10^6$ s⁻¹.²¹

Highly efficient electroluminescence from TADF emitters is ensured by dispersing TADF emitters at a low concentration into a suitable host matrix in an effort to minimize concentration quenching and TTA.^{20,31} The role of the host matrix is highlighted in detail below.

Both TADF and PHOLEDs can achieve 100% internal quantum efficiency; these devices utilize singlet and triplet excitons as the emitting state, respectively. The external quantum efficiency (EQE) has reached up to 25% for blue, red and green devices in PHOLEDs, while the EQE in TADF lags slightly behind, particularly for blue TADF devices.³² Organic synthesis of molecules towards TADF applications is rich in material diversity. The emitting species in TADF are lower in energy (singlet, triplet) when comparing the emission wavelength of both types of devices. Thus, the driving voltage in TADF devices may be lower since a narrower bandgap host material may be used.³³

Similar to PHOLEDs, blue TADF-based OLED devices exhibit efficiency roll-off at high current density. Roll-off behavior is

characterized by a decrease of device efficiency when high luminance is attempted under high current conditions.^{34,35} PHOLEDs exhibit roll-off behavior attributed to long triplet lifetimes (τ_T) and large ΔE_{ST} . Thus, shortening the excited state lifetime (τ_{TADF}) may suppress TTA, singlet-triplet annihilation (STA) or triplet polaron exciton annihilation³⁶ and minimize ΔE_{ST} and thereby may lead to TADF devices with minimal efficiency roll-off.^{35,37,38}

A large range of building blocks have been used as a foundation for TADF materials including acridines,³⁹⁻⁴¹ phenoxazines,⁴² spirobased hydrocarbons,⁴³⁻⁴⁵ pyridines,⁴⁶ anthraquinones,⁴⁷ oxadiazoles,⁴⁸⁻⁵⁰ phosphine oxides,⁵¹⁻⁵³ dihydrophenazines,³¹ heptaazaphenalenenes,⁵⁴ triazines,^{55,56} dicyanobenzenes (phthalonitriles),²¹ diphenylsulfones,⁵⁷ arylketones such as xanthone,³⁰ benzophenones,⁵⁸ thioxanthenes,⁵⁹ heptazines,⁶⁰ and cuprous complexes in their specific role as donor and acceptor moieties.^{61,62} However, acridines and diphenylsulfones were shown to be chemically unstable under device conditions.⁶³ A series of reviews on TADF materials and devices have been published in foreign languages.⁶⁴⁻⁷¹ Adachi's most recent review written in Japanese on the third generation organic electroluminescence was translated into English;³ the review by Tao *et al.* traces a timeline of development in the field of TADF materials across a wide range of classes of materials.⁷² The review of Bergmann *et al.*⁷³ specifically compares organic and metal-organic TADF materials. Therein, an overview of the origin of the TADF phenomenon is given, the chronologic appearance of materials for TADF as well as the status quo. Leil *et al.* specifically reviewed Cu(I) complexes as applied in TADF applications.⁷⁴ In parallel with our work on this review, Wong and Zysman-Colman⁷⁵ published a seminal review on purely organic TADF materials for OLEDs. Herein, we focus solely on carbazole-containing materials for TADF applications. We trace the development of understanding the TADF process along this journey as emitters and hosts in oligomers, dendrimers and others. A short survey of computational approaches to assess molecules in TADF applications and further their understanding is presented.

Carbazole

Carbazole has been widely used toward optoelectronic device applications in general as a source for host materials and emitters in the form of oligomers, dendrimers and polymers.⁷⁶⁻⁷⁹ Carbazole is an excellent hole-transporter.²⁰ The advantages of carbazole as an organic material are highlighted in four fundamental advantages, (1) inexpensive starting material; (2) ease of functionalization at the nitrogen atom and thus property modification without altering the backbone; (3) several linkage positions on the carbazole backbone; (4) aromatic properties that confer stability under a wide range of conditions.⁷⁸ Before the application of carbazole in TADF is explored, some fundamental spectroscopic and electrochemical properties of carbazole are reviewed.

1878 marks the first reference to carbazole in the scientific literature.⁸⁰ Carbazole was then extracted from the anthracene



fraction of coal tar. In the ground state, carbazole shows hydrogen bonded complexes. Solvents containing OH groups, form complexes with carbazole of $\text{NH} \cdots \text{O}$ type in preference over $\text{OH} \cdots \pi$ type, as well as a bend in the molecular geometry. The dipole moment of carbazole changes from 2 D in the ground state to 3.1 D in the first excited state.⁸¹ The absorption spectrum of carbazole in methanol is characterized by three weak absorption bands at 335 nm, 323 nm and 294 nm with extinction coefficients around $10^3 \text{ M}^{-1} \text{ cm}^{-1}$ and an additional three absorption bands at 252 nm, 244 nm and 233 nm with a 10 fold higher extinction coefficient of $10^4 \text{ M}^{-1} \text{ cm}^{-1}$.⁸² The absorption spectrum in a single-crystal matrix of fluorene at very low temperature (15 K) yields absorption bands at 330 nm, 290 nm, 255 nm, and 230 nm. The absorption as well as the high fluorescence emission properties of carbazole appear to be similar in nature to $\pi \rightarrow \pi^*$ transitions due to the fact that the non-bonding electron pair of the singly bonded nitrogen is perpendicular to the ring plane, thus allowing the effective overlap with the π orbitals of the neighboring carbons.¹⁸ Carbazole shows the two first absorption bands with low lying states of $^1\text{A}_1$ and $^1\text{B}_1$ symmetry,⁸¹ wherein $^1\text{A}_1$ is of lower energy level.⁸³ No significant change in the equilibrium nuclear configuration is observed giving rise to the absence of significant progression in the vibrational mode. A linear relationship and rise in intensity between the $^1\text{L}_b$ band and the polarity of the solvent is observed.⁸⁴ Replacing the N–H hydrogen with alkyl groups significantly only effects the absorption wavelength of the $^1\text{L}_b$ band.⁸⁴

Studies involving the photoselection technique indicate that the lowest lying absorption band and fluorescence band exhibit significant mixed polarization hinting toward a forbidden character for both.⁸⁵ A mirror image symmetry is observed for the absorption and fluorescence emission bands of carbazole. The fluorescence quantum efficiency of carbazole is reported at 0.38.^{86,87} The fluorescence lifetime of carbazole is 15 ns at 77 K in polar EPA (diethyl ether, isopentane and ethanol (5:5:2)) solvent.⁸⁵ Carbazole has a high E_T of 3.02 eV⁸⁸ and the lowest energy triplet state is assigned to $^3\text{L}_a$.⁸⁵ A phosphorescence lifetime of 7.7 s was determined at 77 K in polar EPA.⁸⁵ Hereby, the phosphorescence is influenced by the direct spin-orbit coupling mechanism to the $^1\text{B}_2$ ground state.⁸⁹ It is worth noting that the triplet energy of carbazole is higher in comparison to related biphenyl compounds such as fluorene and biphenyl with 2.90 eV and 2.94 eV, respectively.⁹⁰

The electrochemical behavior of carbazole was thoroughly reviewed by Karon and Lapkowski.⁹¹ Nascent carbazole exhibits one-electron oxidation with a potential of 1.2 eV vs. SCE. The oxidation leads to dicarbazyls and is sensitive to pH. The cation is unstable unless the nitrogen as well as positions 3 and 6 of the ring system are blocked.⁹²

For all of the following materials shown as structures in Fig. 1–12, the material's data in terms of photophysical, electrochemical, and device performance are summarized in Tables 1–3. The absence of standardized reporting of device characteristics along with a variety of approaches to determine and report HOMO/LUMO data is noted. The US Department of Energy

provides the OLED Testing Program⁹³ geared towards the OLED community to accelerate research in this field. Initiatives such as this may serve as an incubator to develop guidelines for reporting to allow cross comparison. Whenever experimental data were only presented in figures, no value was entered in the tables. Quantum yields of TADF materials in thin films were reported unless the data was acquired in NPD or other host materials.

Carbazole-based materials as TADF emitters

Uoyama *et al.* realized design restraints using carbazolyl dicyanobenzene (CDCB) materials, a system of two components, *i.e.* carbazoles acting as donor units and dicyanobenzenes acting as the acceptor unit, both of which are distorted from each other leading to a situation that the HOMO and LUMO are localized on each part separately with an observed small ΔE_{ST} . TADF materials included **2CzPN**, **4CzPN**, **4CzIPN**, **4CzTPN**, **4CzTPN-Me**, and **4CzTPN-Ph**, Fig. 1.²¹ Increasing the power efficiency of a device is achieved through decreasing the drive voltage. Thereby, Seino *et al.* were able to utilize **4CzIPN** to create a device designed with carrier- and exciton confinement combined with energy transfer from an exciplex to create a green OLED with a high power efficiency of over 100 lm W^{-1} . This performance is comparable to PHOLEDs containing iridium-based emitting species.⁹⁴

Masui showed a significant spectral overlap between S_1 and T_1 – T_n absorption in **2CzPN** (Fig. 1), which in the presence of significant singlet exciton density explains an exciton quenching mechanism based on both STA, and TTA mechanisms to be responsible for the significant external quantum efficiency (η_{EQE}) roll-off behavior.⁹⁵

Kretzschmar *et al.* started with the popular TADF fluorophors **4CzIPN** and **4CzTPN** (Fig. 1) and derived mono and dihalogenated derivatives with the results of materials of low singlet–triplet gap of $\sim 0.04 \text{ eV}$ (experimentally determined), and fluorescence lifetimes combined with improved ISC due to heavy-atom effects of the halogens.⁹⁶ No device data was reported.

The reduced operational stability observed in TADF devices may be due to the long-lived triplet energy species leading to unwanted chemical reactions. The introduction of an assistant dopant with large $k_{\text{isc}} \sim 10^6$, **4CzIPN-Me** (Fig. 1), along with an emitting species 2,8-di[*t*-butyl]-5,11-di[4-(*t*-butyl)phenyl]-6,12-diphenylnaphthacene (TBRb) allowed the suppression of TTA due to Förster energy transfer between the singlet excited **4CzIPN-Me** and TBRb at highly optimized concentrations.⁹⁷ Devices that included an assistant dopant exhibited increased operational lifetime (time at which luminance drops to 0.5 of the initial luminance). This lifetime is 5 hours for the traditional TBRb-based OLED device, 1472 hours for the **4CzIPN-Me**-based TADF device and 3775 hours for the TADF-device containing the assistant dopant (termed TAF-device). Device stability may thereby be achieved with assistant fluorophores with even shorter triplet lifetimes.



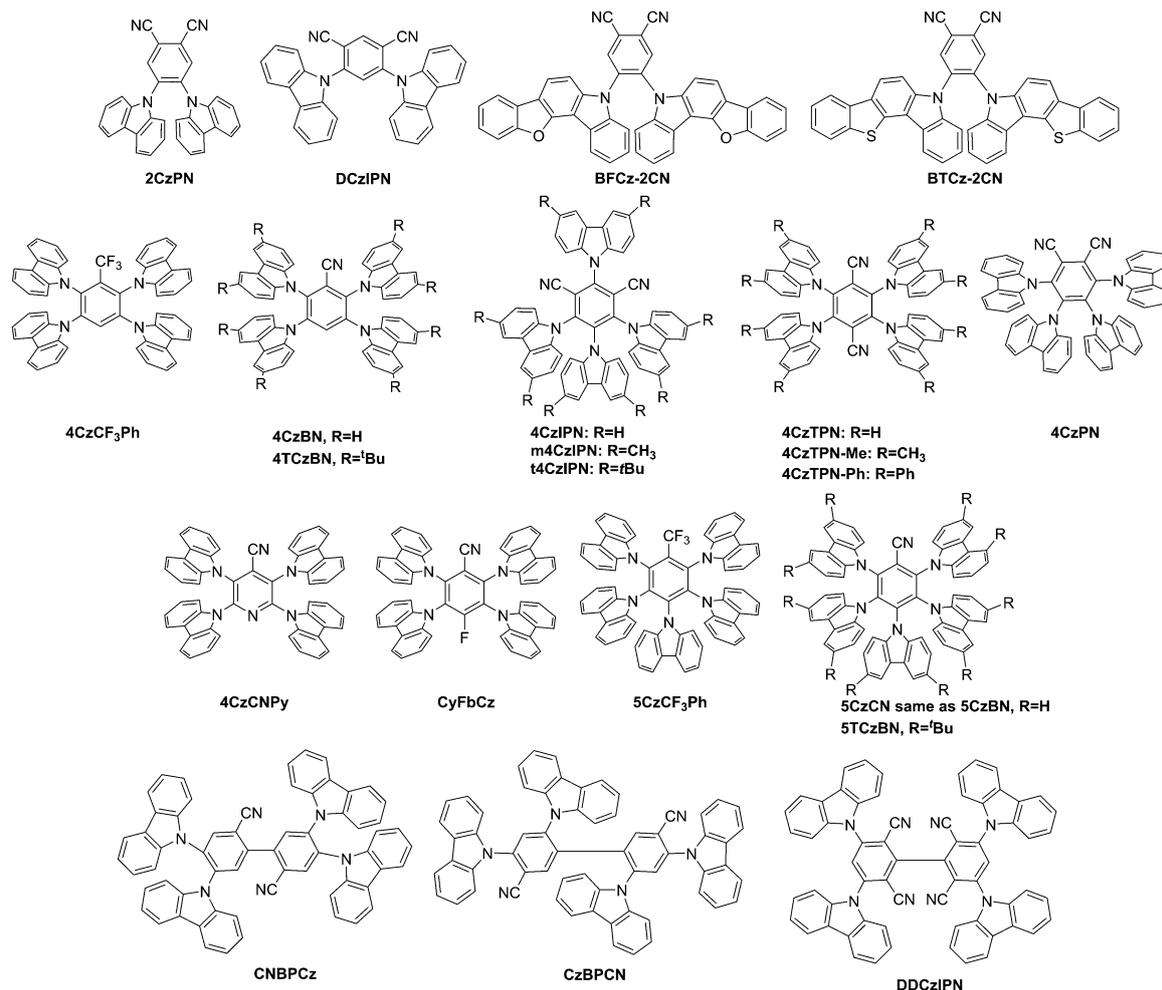


Fig. 1 Structures of carbazoyl cyanobenzene derivatives, tetra and pentacabazoylpyridine and benzenes, and carbazoylated biphenyls.

Sun *et al.* reported a much improved blue-emitting TADF device based on a mixed co-host system of **mCP:PO15** (2,8-bis(diphenylphosphoryl)dibenzothiophene) to improve efficiency roll-off through charge balance. The comparison to previous devices of a similar structure points toward the efficiency roll-off being grounded in an exciton quenching processes due to STA and TTA due to the slow reverse intersystem crossing rate (k_{RISC}) of the emitter **2CzPN**, Fig. 1.⁹⁸

Wang *et al.*⁹⁹ reported **4CzIPN** (Fig. 1) and utilized this material to provide evidence that the mechanism of electroluminescence of devices based on **4CzIPN** is based on the recombination of injected carriers in the (near) absence of energy transfer processes.

On the other hand, Kim *et al.*¹⁰⁰ utilized **4CzIPN**, which has a deep lying HOMO level, to create a solution-based simplified OLED device. The integration of a deep HOMO level buffered material hole injection layer, which due to its self-assembly process has an increased work function as well as an engineered emission layer, avoids exciton quenching.

Cho *et al.*¹⁰¹ prepared materials for solution-processable TADF approaches by inclusion of methyl groups and *t*-butyl groups in the 3,6 positions on each of the four carbazole substituents of the **4CzIPN** emitter leading to new materials

m4CzIPN and **t4CzIPN**. The *t*-butyl groups led to increased solubility and stabilized film morphologies, Fig. 1. Devices were prepared both in solution and by vacuum processing.

White organic light emitting devices (WOLEDs) are used in lighting and display applications. Carbazole-based TADF materials have been utilized for WOLED applications.^{102–107} For example, **DCzIPN** (Fig. 1)^{108,109} was utilized in a hybrid device construct serving as both a blue emitter and a PHOLED host for yellow emitters in WOLED applications by Cho *et al.* where an external quantum efficiency of more than 20% was reached with CIE coordinates of 0.31, 0.33.

Mei *et al.* studied and introduced the non-conjugated negative inductive, *i.e.* electron withdrawing, effect on TADF materials. Therein, trifluoromethyl was included in tetra- and pentacabazoyl substituted TADF materials as an electron acceptor unit leading to **4CzCF₃Ph** and **5CzCF₃Ph** for solution processing, Fig. 1.¹¹⁰ Compound **4CzCF₃Ph** showed blue emission, while **5CzCF₃Ph** showed a lower turn-on voltage of 3.9 eV attributed to the higher HOMO energy level and higher luminance of 2436 cd m⁻² due to the smaller singlet–triplet gap.

The TADF materials **4CzBN** and **5CzBN** (also reported as **5CzCN**) were utilized by Zhang *et al.* to explore the shielding effect of



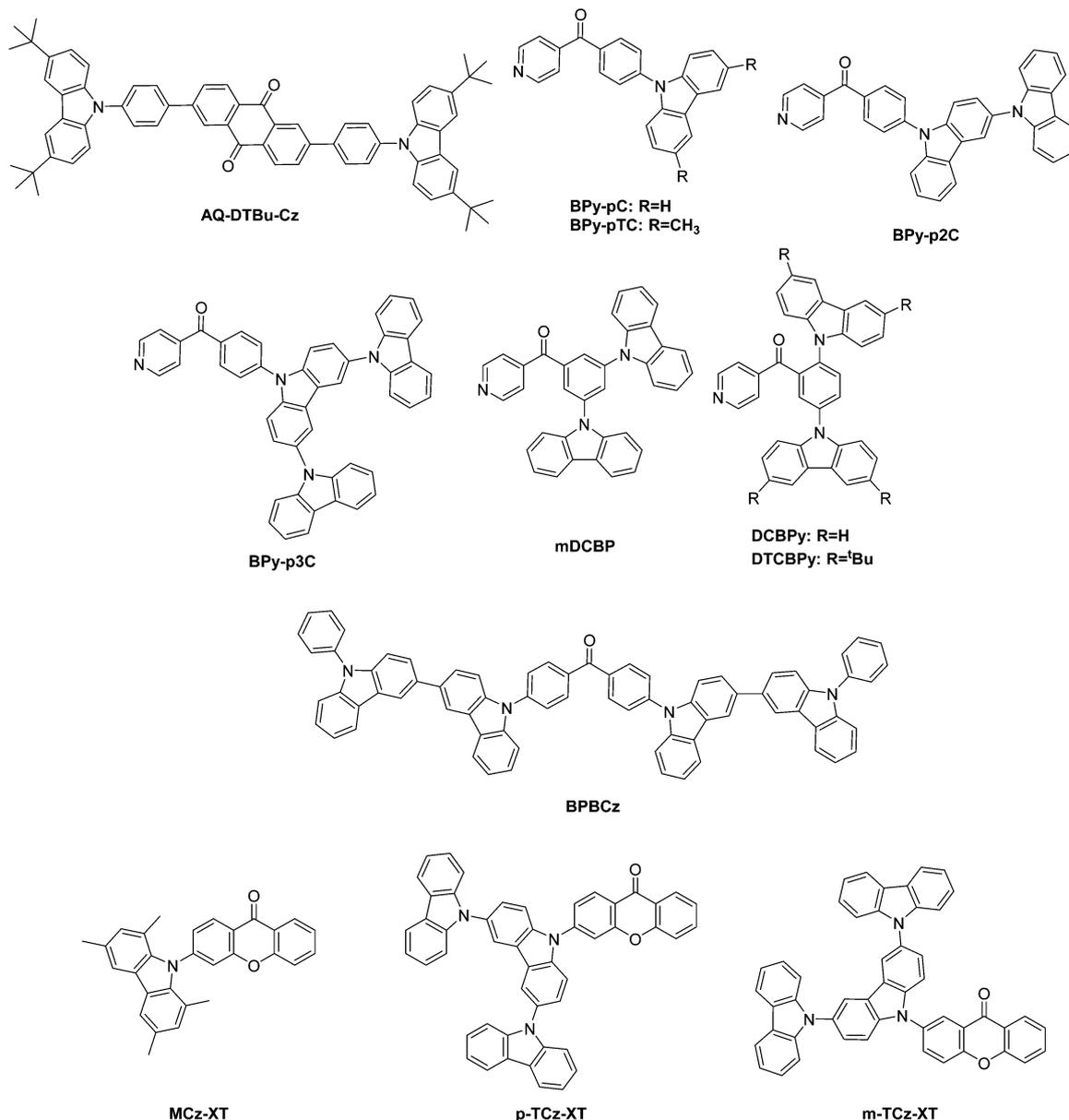


Fig. 2 TADF-materials based on anthraquinone, phenyl(pyridin-2-yl)methanone, benzophenone, and xanthone.

steric crowding by *t*-butyl groups to create **4TCzBN** shielded as **5TCzBN**, see Fig. 1. The modification led to a small reduction of ΔE_{ST} , and an increase in oscillator strength (f). This effected the PL efficiencies, with minimal modification to the CIE coordinates, yet an improvement of the operational lifetime of the devices by 2.7 fold for **4TCzBN** vs. **4CzBN** and 4.6 fold for **5TCzBN** vs. **5CzBN**, respectively.¹¹¹

Two materials wherein the carbazole units were fused to benzofuran, **BFCz-2CN**, and benzothiophene, **BTCz-2CN** (Fig. 1), units were introduced by Lee *et al.*³² in an effort to extend the repertoire of carbazole-based donor units. The TADF materials showed quantum efficiencies of 12% at a doping concentration of 1%.

Tang *et al.*¹¹² modified a previous host material for PHOLEDs, 2,3,5,6-tetracarbazylpyridine (**4CzPy**), by inclusion of a cyanogroup

in position 4 leading to **4CzCNPy** (Fig. 1), a green-emitter. Solution-processing led to a bilayer-device with TADF capability with green emission.

Zhang *et al.* attempted to improve blue-emitting species for TADF applications and included the fluorine atom as a second electron acceptor next to nitrile on **4CzBN** to create **CyFbCz**, (Fig. 1). The modulated bandgap resulted in a low turn-on voltage of 4.1 V and good color stability with the Commission Internationale de L'Eclairage (CIE) coordinates of 0.18, 0.13.¹¹³

Cho *et al.* prepared a blue-emitting TADF material **5CzCN** (Fig. 1) to address the low performance of blue-emitting TADF materials for both vacuum and solution processing. The group chose benzonitrile as the acceptor unit and five carbazole units as donor units and showed a maximum external quantum



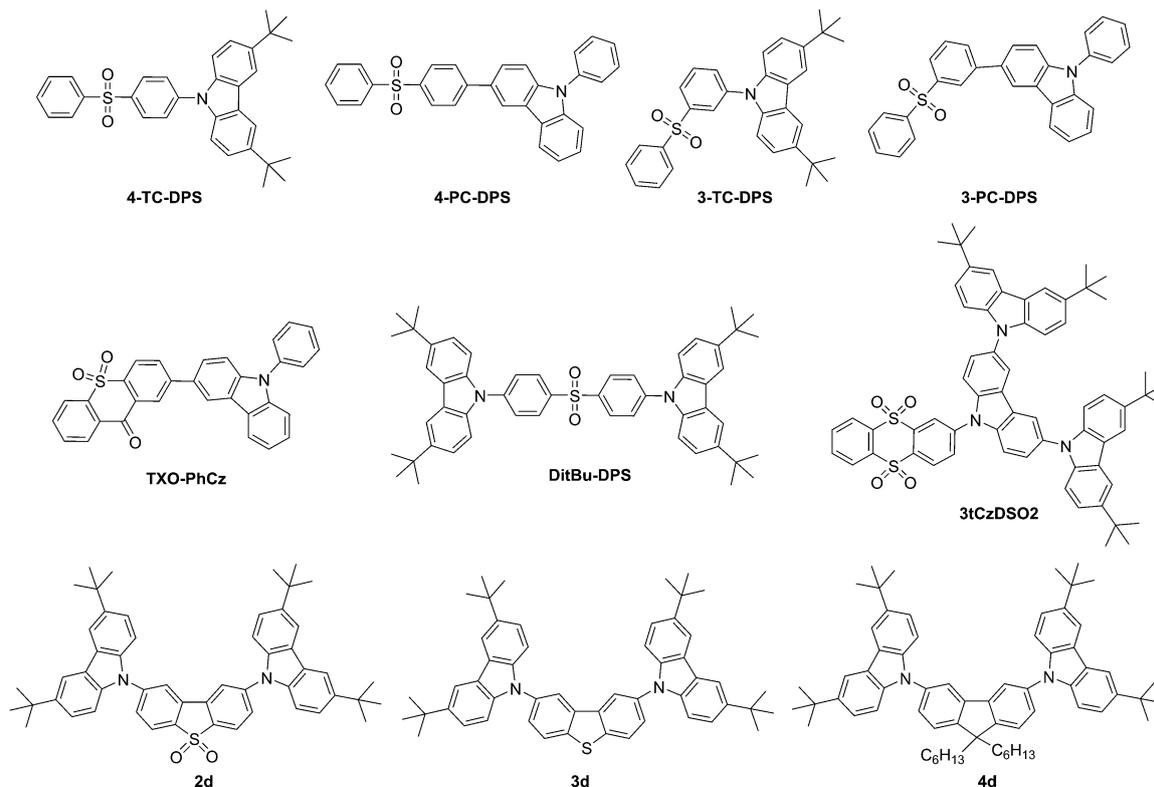


Fig. 3 Sulfoxide-containing TADF materials and others.

efficiency of 19.7% and 18.7% in vacuum deposited and solution processed TADF devices, respectively.¹¹⁴

Starting with the green-emitting **4CzIPN**, Tanimoto *et al.*¹¹⁵ replaced one nitrile group with a carbazole unit to create **5CzBN** (Fig. 1) in an effort to shift the emission to the blue wavelength. Coincidentally, this material was also reported by Zhang *et al.*¹¹¹ and Cho *et al.*¹¹⁴

Cho *et al.* addressed two major challenges of blue-emitting TADF materials, *i.e.* broad emission spectra with a large full-width half maximum (FWHM) of 70–80 nm due to the CT as well as the issue of colour purity. The application of interlocked donor species as the core and organizing the electron donating and withdrawing species along the backbone resulted in new materials **CNBPCz** and **CzBPCN** (Fig. 1) for TADF materials, the latter of which showed a FWHM of only 48 nm and a high EQE of 14%.¹¹⁶

Cho *et al.* designed dual-emitting core **DDCzIPN** based on the **DCzIPN** TADF material in an effort to increase the photoluminescence quantum efficiency of the emitter itself and the emitter in devices. **DDCzIPN** showed improved maximum external quantum efficiency, Fig. 1.¹¹⁷

Zhang *et al.* prepared anthraquinone-based TADF molecules. The group prepared a large series of bipolar molecules composed of a donor- π -acceptor- π -donor-pattern and utilized anthraquinone as the acceptor unit in an attempt to achieve high-efficiency and short lifetime TADF materials. Among others, sterically substituted carbazole units were utilized as donors (**AQ-DTBu-Cz**), Fig. 2. Particularly, the carbazole-derivatives

exhibited high roll-off behavior and undesirable rotational relaxation of the excited state, *i.e.* non-radiative decay. The study concluded that red fluorescent TADF molecules pose still challenges due to an unfavorable energy gap law leading to large non-radiative decay rates.⁴⁷

Rajamalli utilized the benzoylpyridine (BP) building block decorated with carbazole units.^{118,119} Specifically, **mDCBP** (Fig. 2) exhibited mechano- and piezochromism.¹¹⁹ Four devices were prepared with 10–30 wt% **mDCBP** in DPEPO. The blue emitter reached an external quantum efficiency of 18.4% as reported in Table 1, wherein the data reported is for 30 wt%.

The concept of wide dispersion of the HOMO was utilized in a series of donors ranging from carbazole (**BPy-pC**), to *t*-butyl-carbazole (**BPy-pTc**), to 3,9'-bicarbazole (**BPy-p2C**) to 9,3':6',9''-tercarbazole (**BPy-p3C**) combined with a BP building block to prepare improved blue emitters, Fig. 2.¹²⁰ Significantly, **BPy-p3C** showed dispersion of the HOMO over the entire donor building block, which led to narrow singlet-triplet splitting. In addition, an increase in the number of carbazoles increases the EQE to 23.9%.

In addition, Rajamalli *et al.* included two nascent or two *t*-butylcarbazole donor units on the *meta*- and *ortho*-carbons of the benzoylpyridines acceptor, *i.e.* *para* to each other, to create **DCBPy** and **DTCBPy** (Fig. 2). Herein, the group applied the concept of intramolecular space interactions between the donor and acceptor units to create molecules with small ΔE_{ST} of 0.03 and 0.04 eV and EQE of above 24%.¹¹⁸

Kim *et al.*¹²¹ utilized a strong 3,3'-bicarbazole-donor unit to prepare a benzophenone-derivative entitled **BPBCz** (Fig. 2).



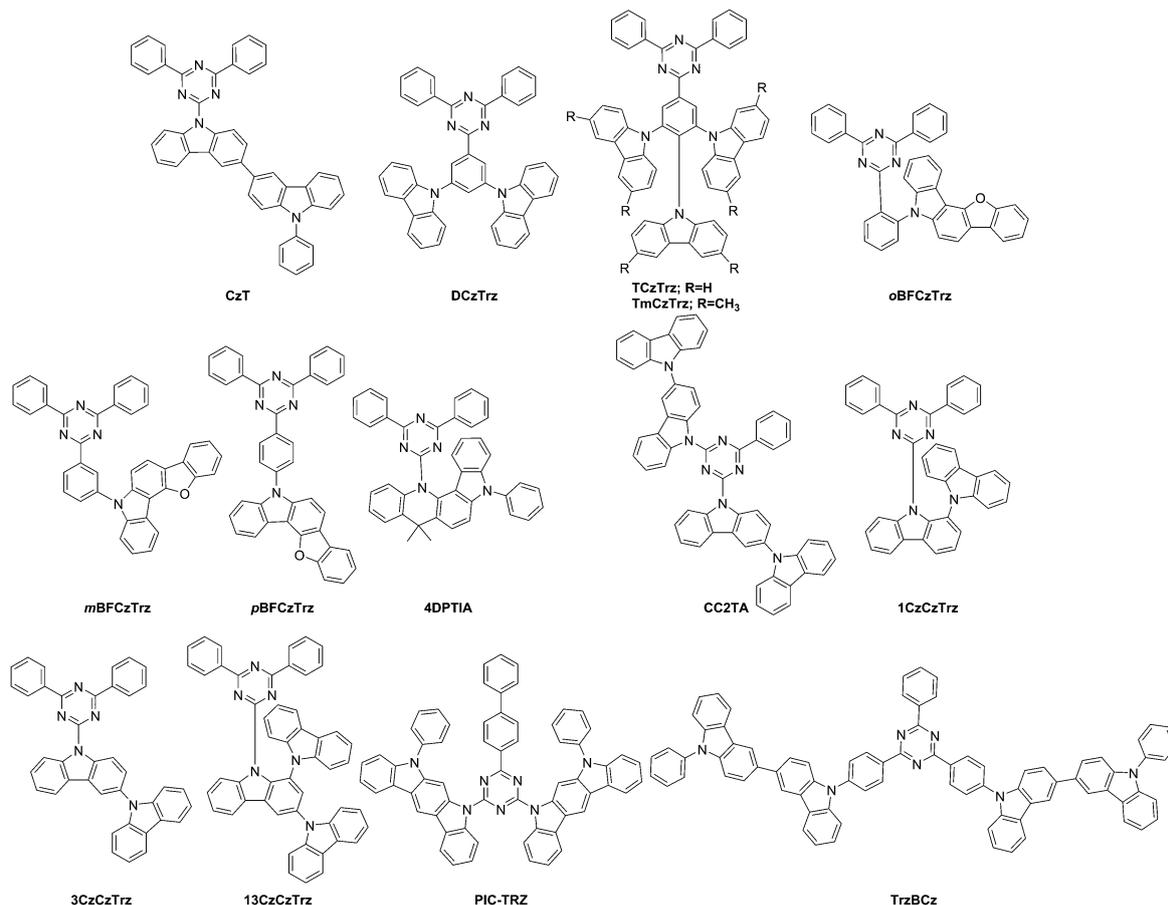


Fig. 4 Carbazoles and fused carbazoles on triazine acceptors.

Carbazole is termed as a weak donor in this study, based on the computational results of the frontier molecular orbitals. Both compounds follow the DAD-type construct. **BPBCz** is a blue emitting species with 23.3% quantum efficiency, wherein the biscarbazole unit extended the lifetime of the device compared to an acridine-donor device.

Lee *et al.*³⁰ prepared a tetramethylated carbazole-unit linked to xanthon entitled **MCz-XT** (Fig. 2) along with a series of xanthon-based TADF materials to address the Dexter energy transfer that is the predominating factor for concentration quenching in TADF materials. In Dexter energy transfer, triplet excitons interact *via* electron-exchange interactions of triplet excitons, whereby a minimal modulation of the molecular geometry in the emitter allows quenching to be suppressed. In a second study, Lee *et al.*¹²² prepared ***p*-TCz-XT** and ***m*-TCz-XT** (Fig. 2) to investigate the effect of regioisomers of *para*-(3-substituted xanthon) *vs.* *meta*-(2-substituted xanthon) linkage in TADF performance, Fig. 2. The compound ***p*-TCz-XT** exhibited higher photoluminescence quantum yield and shorter TADF-lifetime and thus outperformed the other material with a 14.4% EQE.

Zhang *et al.* presented three diphenylsulfone-based TADF materials as pure blue emitters, one of which contained carbazole as a building block with **DitBu-DPS**, Fig. 3.¹²³ The CIE coordinates of the EL for a device based on this materials are (0.15, 0.07), close to the National Television Standards Committee (NTSC)

standard blue with CIE coordinates of (0.14, 0.08). Blue fluorescent OLEDs with pure blue emission of CIEy > 0.1 are still difficult to achieve with transition metal-based PHOLED materials.

A carbazole-containing thioxanthon derivative, **TXO-PhCz** (Fig. 3) was prepared by Wang *et al.*⁵⁹ Doping of emitters in a suitable host is a crucial and challenging process in the formation of TADF devices. Meng *et al.*¹²⁴ utilized **TXO-PhCz** to prepare a multiquantum well structure in the emitting layer to create a nondoped TADF-based OLED with an EQE of 22.6%.

Sun *et al.*¹²⁵ prepared **tCzDPSO₂**, a molecule that showed only aggregation-induced emission. Replacing one single unit of carbazole with triscarbazole resulted in **3tCzDPSO₂** (Fig. 3), a chromophore that in addition to aggregation-induced emission showed TADF behavior.

Huang *et al.* prepared TADF materials utilizing carbazole donors and diphenylsulfone acceptors, **4-TC-DPS**, **4-PC-DPS**, **3-TC-DPS** and **3-PC-DPS** (Fig. 3); however, no device data was acquired.¹²⁶ Only **3-TC-DPS** and **4-TC-DPS** had singlet-triplet splittings at or below 0.24 eV.

Dias *et al.* completed a study of donor-acceptor-donor (DAD) and donor-donor-donor compounds with a range of donor and acceptor units. Carbazole-containing compounds **2d**, **3d**, and **4d** showed TADF behavior, Fig. 3.⁵⁷ The group showed that even with singlet-triplet gaps (¹CT-³ππ*) of more than 0.3 eV a TADF efficiency of unity can be achieved.



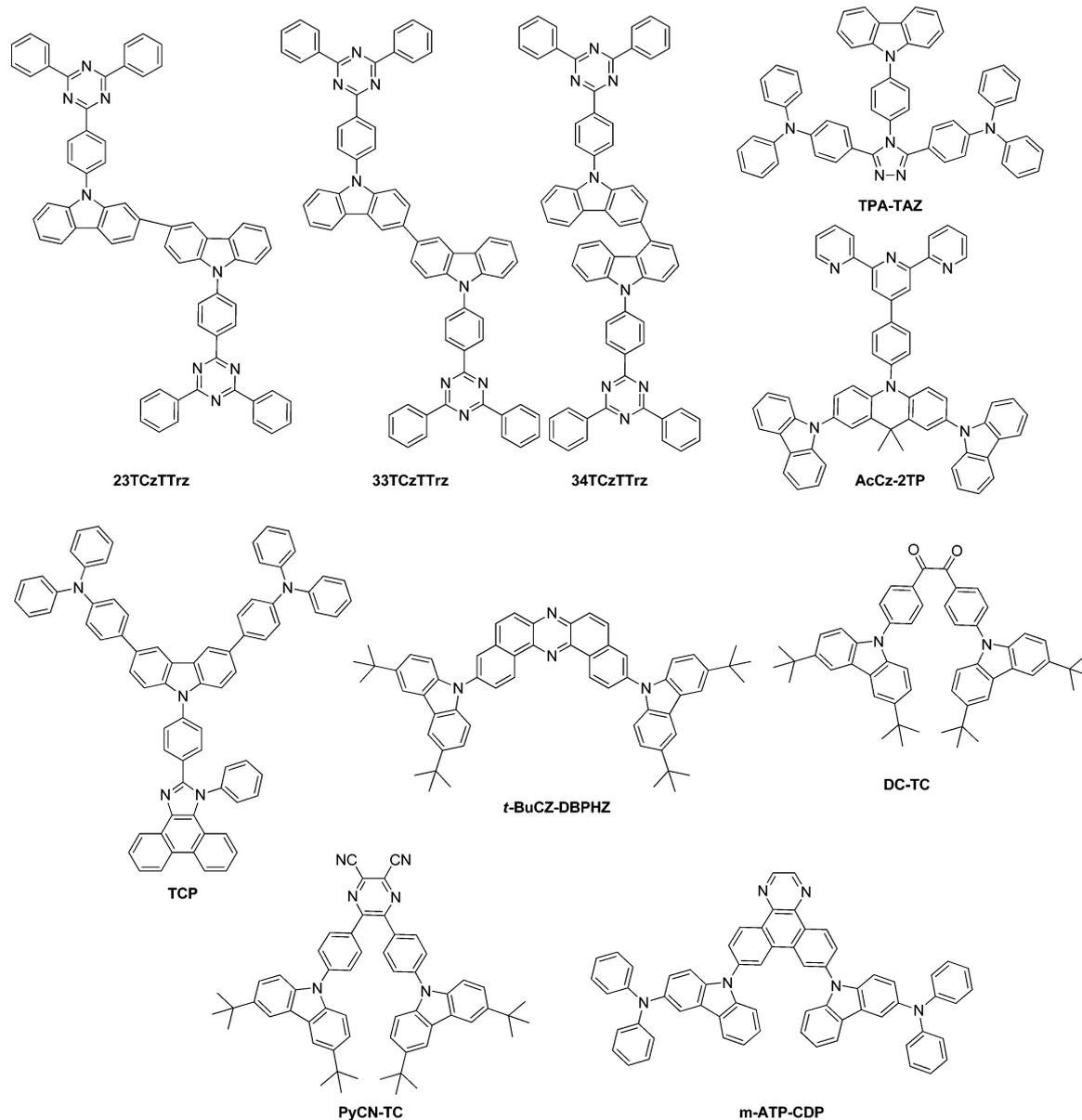


Fig. 5 Twin-emitting cores based on carbazoles; carbazoles on diverse cores, and carbazoles linked to benzil and dicyanopyrazine and 1,4-diazatriphenylene.

Significantly, a linearly disubstituted acceptor unit leads to weak phosphorescence independent of the donor and acceptor units applied. Only compound **2d** containing the dibenzothiophene-*S,S*-dioxide acceptor shows pure TADF behavior in the absence of TTA.

Chang *et al.* developed carbazole-triazine derivative **CzT** (Fig. 4) to serve as a host material for green PHOLEDs.¹²⁷ Due to the presence of special separation of the HOMO-LUMO orbitals as investigated by density functional theory (DFT) calculations, Serevičius *et al.* utilized **CzT** and a derivative **PhCzTAZ** for TADF applications.¹²⁸ Even though both molecules **CzT** and **PhCzTAZ** contain carbazole and triazine units, only **CzT** exhibited TADF behavior while **PhCzTAZ** did not. **PhCzTAZ** has a large singlet-triplet energy gap of 0.48 eV and 0.2 eV in hexane and toluene, respectively explaining the absence

of TADF behavior. The molecule **CzT** exhibits solvent polarity dependent state switching for the singlet states between LE (hexane) and ICT (toluene). The singlet-triplet energy gap is observed as 0.085 eV in hexane and 0.008 eV in toluene.

The second approach utilizing benzofuran-fused carbazoles, *i.e.* a benzofurocarbazole donor with a diphenyltriazine acceptor to prepare blue emitters was presented by Lee *et al.*¹²⁹ The *o*- vs. *m*- vs. *p*-linkages via a phenyl group resulted in compounds **oBFCzTrz**, **mBFCzTrz**, and **pBFCzTrz** (Fig. 4), which showed singlet-triplet energy gaps of 0.05, 0.11, and 0.25 eV, respectively. The *o*-linked material also showed a quantum efficiency of 20% and minimal efficiency roll-off.

Lee *et al.*¹⁴⁵ raised the EQE for green and blue TADF devices up to 25% by evenly dispersing the HOMO of the TADF emitter



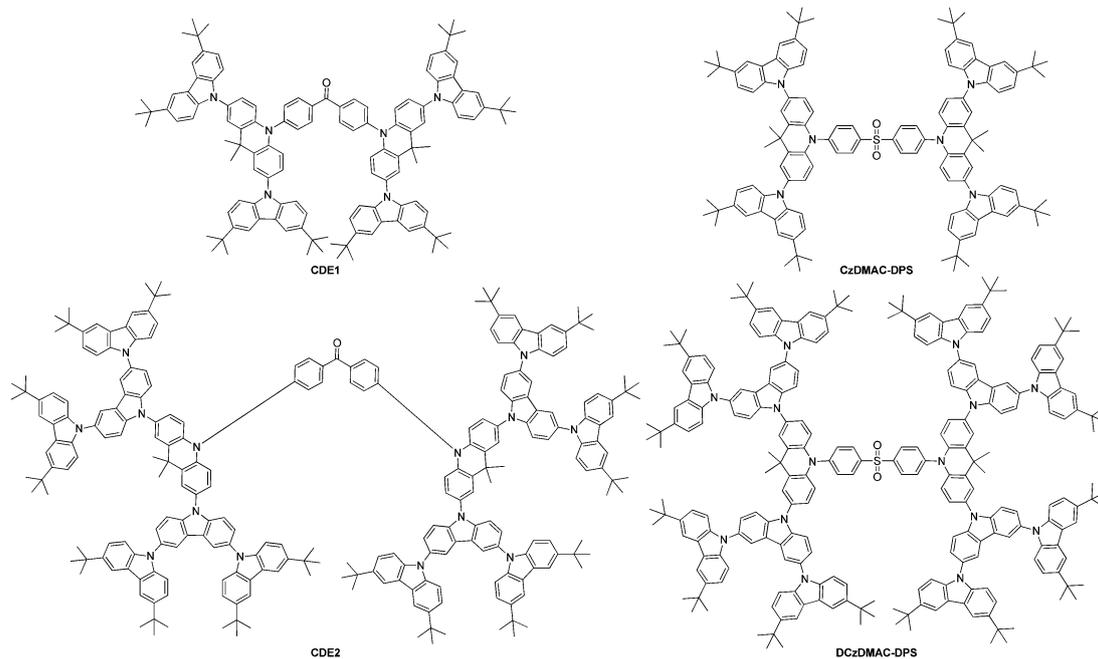


Fig. 8 Benzophenone- and sulfonate-based dendrimers.

not be attributed to TTA, TADF or higher-level RISC, yet a triplet polaron-interaction-induced upconversion involving one-electron transfer mechanism was proposed.¹³⁴

Sasabe *et al.*¹³⁵ introduced a terpyridine unit in combination with an acridine core for TADF applications. When modulated with carbazole units, TADF material **AcCz-2TP** was prepared with a singlet triplet gap of 0.23 eV, Fig. 5.

Data *et al.* utilized dibenzo[*a,j*]phenazine (DBPHZ) as the acceptor unit and prepared several new TADF materials. When flanked with *t*-butylated carbazole donors, the material ***t*-BuCz-DBPHZ** (Fig. 5) was prepared as a green to deep-red/NIR OLED emitter.²⁹ The compound did not show significant CT emission until THF was applied as the solvent medium. Delayed fluorescence was weak and mostly phosphorescence was observed. A significant amount of TTA could not be excluded. In this case, intersystem crossing leading to the observed TADF is based on spin orbit charge transfer between the ¹CT state and the triplet locally excited state on the acceptor (³LE_A) instead of the general case of the donor. A long lived delayed fluorescence lifetime combined with TTA was attributed to the low device efficiencies.

Cai *et al.* attempted to reduce k_R and k_{RISC} by increasing the dihedral angles between the donor and acceptor systems involved and utilizing a molecular design including the TICT. The group combined *t*-butylated carbazoles as donor molecules (TC) and utilized benzil to induce a small k_r on account of $n-\pi^*$ transition and a small ΔE_{ST} to create **DC-TC**, and converted the benzil group to dicyanopyrazine to create a new diazaring for a TADF molecule entitled **PyCN-TC** (Fig. 5). Both compounds follow the donor- π -acceptor- π -donor design. Along with 9,10-dihydroacridine building blocks, the group completed a thorough study of the photophysical, quantum chemical and OLED

characteristics.¹³⁶ Incidentally, **PyCN-TC** showed the lowest energy ³LE state, while **DC-TC** showed the lowest ³CT state. Zhang *et al.* showed that a lower lying ³LE enlarges ΔE_{ST} and hinders the efficiency of RISC processes.¹²³ In the case of **PyCN-TC**, this leads to ΔE_{ST} of 0.46 eV and an increase of triplet exciton population. Taking all of the photophysical data together indicates the value of k_{RISC} as the rate limiting factor in the exciton dynamic process and therefore the key factor for shortening τ_{TADF} . In device geometry, the larger k_{RISC} leads to low efficiency roll-off characteristics.

Takahashi *et al.* chose 1,4-diazatriphenylene with a sufficiently high T₁ energy level of 2.9 eV as the core to develop a TADF material and coupled it with the donor 3-(diphenylamino)-carbazole in a D-A-D-type fashion to yield a sky-blue emitter ***m*-ATP-CDP**, Fig. 5.¹³⁷

Shiu *et al.* introduced a non-carbon based linker to create a TADF material with a minimal orbital overlap between donor and acceptor units. The group applied the rigid, electron-accepting boron atom as a spiro-linker between the pyridyl pyrrolidine and carbazole donor units to create new boron complexes **PrFPCz** and **PrFCzP** for TADF applications, Fig. 6.¹³⁸ TADF behavior was not observed for **PrFPCz** in toluene and related polar solvents, yet it was observed in the solid state.

The compound *o*-carborane is an electron-deficient boron-cluster, which was utilized by Furue *et al.* along with carbazole and triazine to create one D-A-A' triad **PCz-CB-TRZ** as well as along with two phenyl-substituted carbazoles to create a D-A-D triad **2PCz-CB**,¹³⁹ Fig. 6. *o*-Carborane among other compounds is known to induce aggregation-induced emission, *i.e.* the chromophore is not emissive in dilute solutions; however, it is highly emissive in concentrated solutions. Aggregation-induced delayed fluorescence (AIDF) is triggered through the structural design



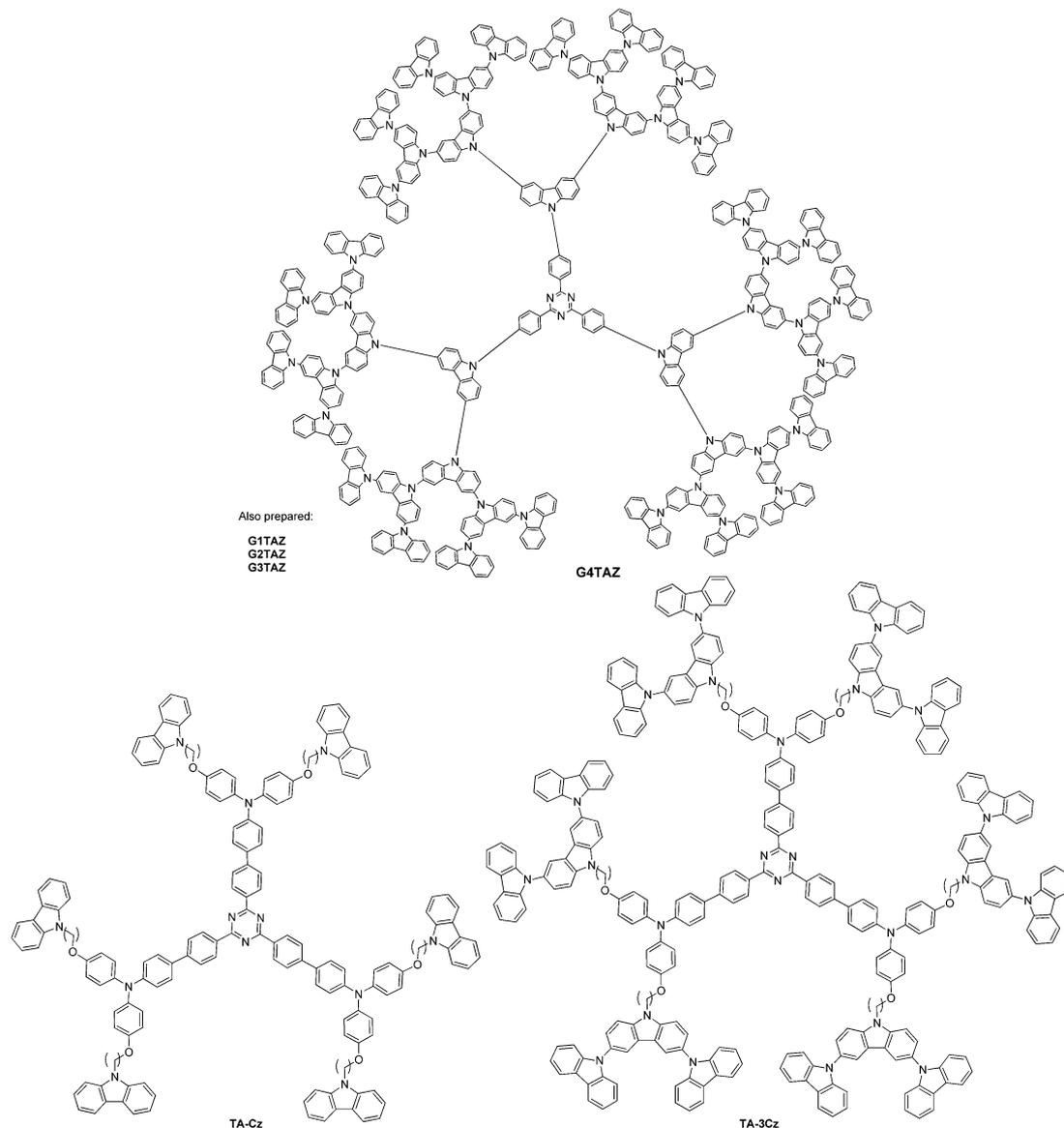


Fig. 9 Self-hosting, solution-processable dendrimeric TADF materials and dendrimers with triazine-cores.

of minimal HOMO–LUMO orbital overlap, which leads to a small singlet–triplet energy gap and thus opens the channel for thermal repopulation of the singlet excited state *via* AIDF behavior. The higher turn-on voltage for **PCz-CB-TRZ** is due to the larger hole-injection barrier, since the HOMO energy level is lower. The high external EL quantum efficiency of $>11\%$ for **PCz-CB-TRZ** was utilized in a dopant-free OLED configuration.

The TADF-approach was also applied as a remedy for blue emitters for PHOLEDs. These emitters are particularly challenging due to the bond dissociation caused by the highly energetic triplet excitons *vide supra*. Zhu *et al.*¹⁴⁰ attempted to utilize all electro-generated excitons by embedding the concept of TADF to PHOLED devices termed metal-assisted delayed fluorescence (MADF). One and two-carbazole-containing molecules, **PdN3O** and **PdN3N** (Fig. 6), were embedded in devices with efficient phosphorescence and delayed fluorescence processes,

respectively. External quantum efficiencies of 20.4% and 20.9% were achieved for **PdN3O** and **PdN3N**, respectively, though a significant roll-off behavior was observed.¹⁴⁰ The large efficiency roll-off was attributed to long triplet lifetimes and poor charge balance.

Particular challenges are still observed for blue TADF-based OLEDs. When screening D–A vs. D–A–D systems, one derivative showed an external quantum efficiency of 19.5% and reduced efficiency roll-off characteristics at high luminance. A methodical study revealed the co-requirement of pre-twisted intramolecular charge-transfer molecules and small singlet–triplet energy gap. The ³LE state needs to necessarily be higher in energy than the ³CT state.¹⁴¹ In addition, the first triplet energy levels of the blue TADF dyes are significantly higher than the PHOLED representatives, *i.e.* 2.9 eV or above.⁵³ This requires that exciton diffusion is suppressed by electron blocking layers and a modification of



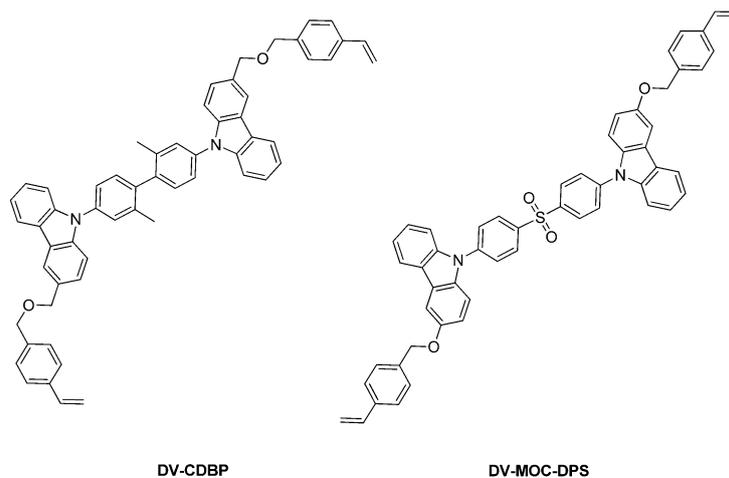


Fig. 10 Cross-linkable host (DV-CDBP) and emitter (DV-MOC-DPS) precursors for TADF applications.

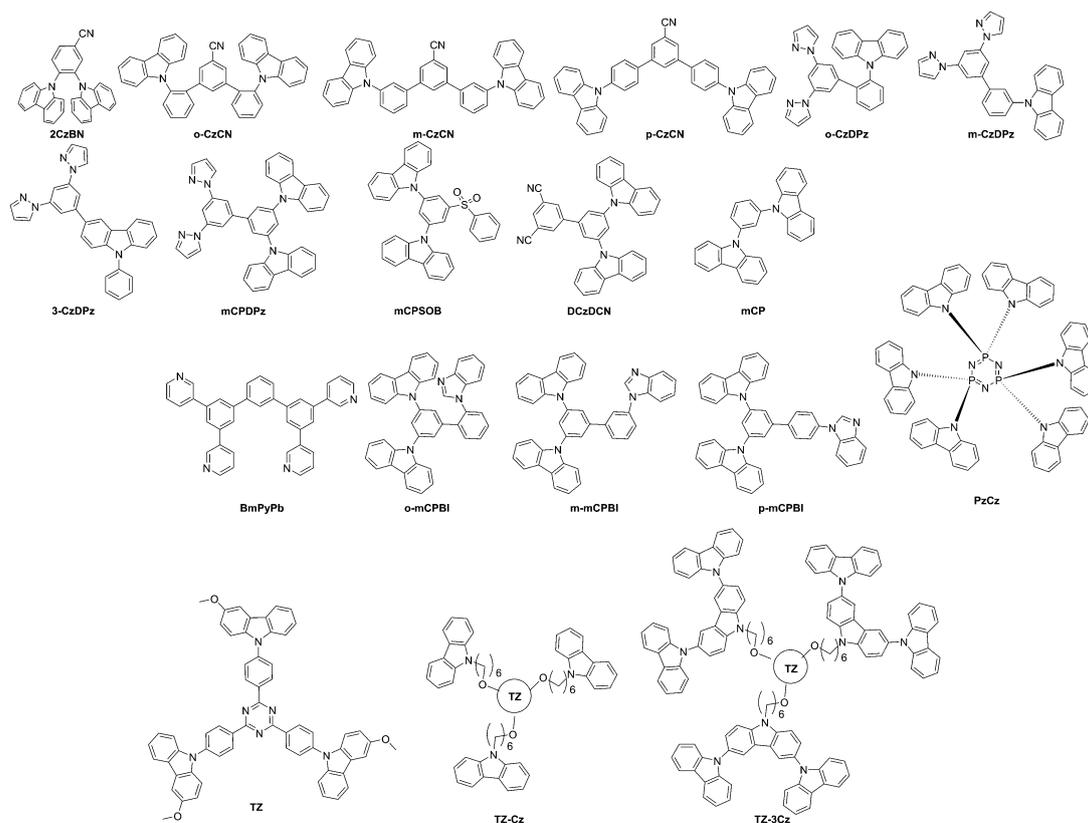


Fig. 11 Small molecule and dendrimeric host materials for TADF applications.

the charge transport layer to ensure a balanced charge transport. These efforts lead to more complex device structures.

Roll-off behavior is a challenge in TADF devices. Roll-off behavior was addressed in the work of Cho *et al.*¹⁴² Considering that both Förster and Dexter mechanisms of energy transfer apply to TADF materials, the optimization of the emissive layer (EML) thickness by widening the trap-free recombination zone and optimizing the doping concentration led to a decrease in the roll-off behavior.

The roll-off behavior of TADF devices was also addressed through device architecture by Tsang and Adachi.¹⁴³ The inclusion of ultrathin layers (2–3 nm) of 8-hydroxyquinolino lithium (liq) between the EML and the hole blocking layer (HBL) as well as between the HBL and the electron transport layer (ETL) led to up to 16-fold extension of device lifetime in which 90% of the initial luminance is reached. Thermally stimulated current measurements allowed Tsang *et al.* to reason that the formation of deep traps leads to decomposition of the



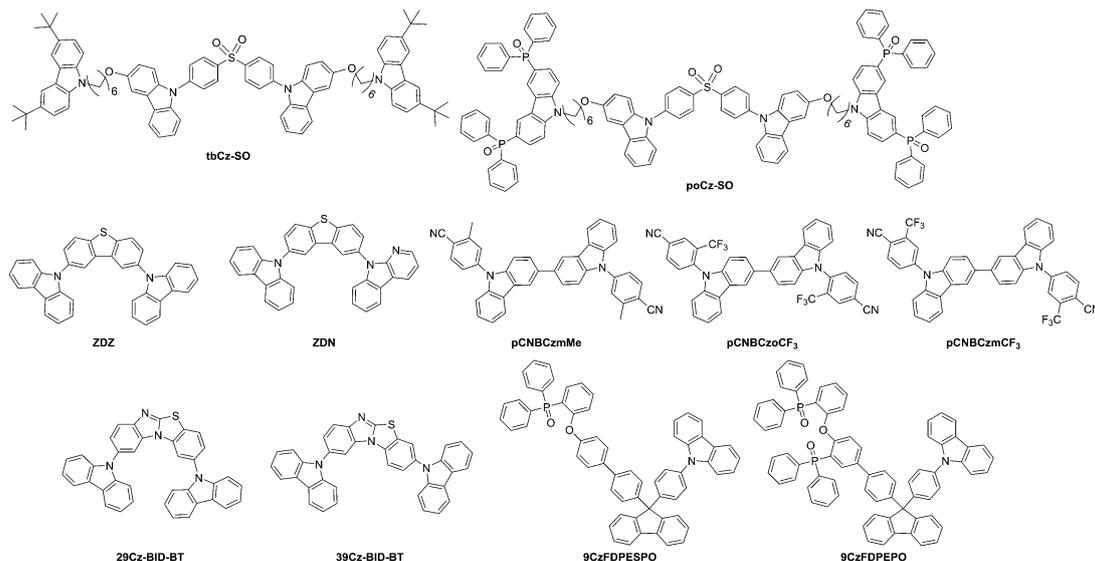


Fig. 12 Carbazole-based host materials containing sulfur, ternary and quaternary phosphine oxide, and fluorenes.

organic emitting material *via* exciton–polaron interactions and thus an increased drive voltage.

Chen *et al.* addressed the high drive voltage of TADF-based OLEDs by the development of a different device architecture and the utilization of a new, carbazole-based p-type material TXFCz. The device architecture is a two-layer active heterojunction.¹⁴⁴

Polymer and dendrimer approaches

During TADF device preparation, careful mixing of the host : dopant ratio is required along with careful scrutiny of the possible phase-separation processes. This hurdle may be avoided by using solution-processed polymers with TADF properties. Dopant-free systems are based on a dendrimer approach based on a carbazole unit due to its (not the least) superior, amorphous film-forming abilities (Table 2).

A D–A type backbone (**pCzBP**) for the polymer was realized by Lee *et al.*, Fig. 7.⁶ Therein, the butterfly-shaped benzophenone is linked *via* the nitrogen atom of carbazole to alkylated carbazoles. The HOMO is localized primarily on the carbazole unit, while the LUMO is localized on the benzophenone units and a small ΔE_{ST} of 0.16 eV is observed. A device with a high external quantum efficiency of 9.3% was realized.

Zhu *et al.* prepared a conjugated D–A type polymer (**PAPCC** and **PAPTC**), wherein the carbazole-containing donors are fixed only in the backbone, while the acceptors are affixed on the side-chains only. This grafting leads to a significantly small ΔE_{ST} due to the physical separation of the HOMO and LUMO, Fig. 7.¹⁴⁶

A dendrimeric TADF emitter was developed by Li *et al.*¹⁴⁷ Dendrimers **CDE1** and **CDE2** (Fig. 8) are characterized by a benzophenone molecule as the anchor, linked to two acridine units, which are substituted in the first and second generation with the nitrogen of the carbazole units. The carbazole units are

substituted themselves in the 3,6 positions with *t*-butyl groups. Solution processed devices utilizing these dendrimers showed an EQE of 13.3% at 1000 cd m⁻² and a low roll-off behavior. Emission in the devices combined the emission from TADF behavior as well as exciplex emission.

Carbazole units were used to encapsulate the DMAC–DPS emitter to synthesize new sulfonate and acridine-containing emitting TADF materials entitled **CzDMAC–DPS** and **DCzDMAC–DPS** (Fig. 8) by Luo *et al.*¹⁴⁸ These new materials were solution-processable and utilized in non-doped devices with a peak EQE of 12.2% and CIE coordinates of 0.22, 0.44.

Dendrimers are oligomers characterized by exact branching and absolute molecular weight. Dendrimers can have the ability to insulate chromophores housed at the core. A dendrimeric approach containing a triphenyl-*s*-triazine core substituted with generations of carbazole units entitled **G2TAZ**, **G3TAZ**, and **G4TAZ** was presented by Albrecht *et al.* as the first, self-hosting TADF material, Fig. 9.¹⁴⁹ TADF devices showed a maximum external quantum efficiency of 3.4% for **G3TAZ**.

Further triazine-containing dendrimers were prepared by Sun *et al.*¹⁵⁰ The dendrimers **TA–Cz** and **TA–3Cz** (Fig. 9) carry peripheral alkylated carbazole and triscarbazoles. The introduction of additional phenyl groups between the core and the dendrons served to physically separate/isolate the core from the periphery and thus induce a small singlet–triplet splitting as well as isolating the core by an encapsulation mechanism. Only 2.4 V was utilized for the drive voltage and an EQE of 11.8% was achieved using **TA–3Cz**.

Sun *et al.*¹⁵¹ presented a solution-processable approach, with a cross-linkable host-precursor **DV–CDBP** and emitter-precursor **DV–MOC–DPS**, Fig. 10. Upon cross-linking at various ratios, the group prepared TADF devices and showed that a mass ratio of 1 : 0.09 of host to emitter yielded a device with the highest photoluminescence quantum yield of 0.71 and a maximum external quantum efficiency of 2%.



Table 1 Overview of photophysical, electronic and device data of carbazole-based TADF materials

Ref.	Compound	Device geometry	Turn-on (V)	L (cd m ⁻²)	CE (cd A ⁻¹)	PE (lm W ⁻¹)	EQE (%)	CIE (x, y)	HOMO (eV)	LUMO (eV)	ΔE_{gap} (eV)	Abs (nm)	PL nm (QY)	PL (TF)	E_{S_1} (eV)	E_{T_1} (eV)	$\Delta E_{\text{S-T}}$ (eV)
Masui <i>et al.</i> ⁹⁵	2CzPN	ITO (100 nm)/ α -NPD (35 nm)/mCP (15 nm)/mCP:60 wt% 2CzPN (40 nm)/PPt (10 nm)/TPBi (40 nm)/LiF (0.8 nm)/Al (100 nm)	7.4	—	—	—	13.6	0.17, 0.30	-5.8 ^a	-3.0 ^b	2.8	—	—	—	—	—	0.09
Cho <i>et al.</i> ¹¹⁷	DCzIPN	ITO (150 nm), PEDOT:PSS (60 nm)/TAPC (20 nm)/mCP (10 nm)/mCP:DCzIPN/TSPO1 (35 nm)/LiF (1 nm)/Al (200 nm)	3.5	—	—	16.4 ^d	0.17, 0.19	—	-6.26 ^e	-3.56 ^e	2.8	—	447, 455 (0.35) ^c	—	2.77	2.72 ^f	0.05
Lee <i>et al.</i> ³²	BFCz-2CN BTCz-2CN	ITO (50 nm)/PEDOT:PSS (60 nm)/TAPC (20 nm)/mCP (10 nm)/mCP:BFCz-2CN or BTCz-2CN (25 nm, 1%)/TSPO1 (35 nm)/LiF (1 nm)/Al (200 nm)	4.0	—	—	—	—	—	-6.19 ^e	-3.58 ^e	—	310, 310–430	(0.946) ^g (0.94) ^g	—	2.59 ^h 2.63 ^h	2.46 ⁱ	0.13 0.17
Mei <i>et al.</i> ¹¹⁰	4CzCF ₃ Ph	ITO/PEDOT:PSS (40 nm)/mCP:4CzCF ₃ Ph (10 wt%, 40 nm)/TmPyPB (60 nm)/LiF (0.8 nm)/Al (100 nm)	4.8	1032 ^d	1.03	—	0.7	0.17, 0.18	-5.6 ^e	-2.62 ^e	2.98	288, 321, 332 ^j	440 ^l	445	3.02	2.78	0.24
Zhang <i>et al.</i> ¹¹¹	4CzBN	ITO/HATCN (5 nm)/NPB (30 nm)/TCTA (10 nm)/mCBP:4CzBN or 4TCzBN (40 wt%) (30 nm)/DpyPA:LiQ (1:1, 30 nm)/LiF (0.5 nm)/Al (150 nm)	3.3	—	—	15.7 ^d	10.6 ^d	0.17, 0.20	-5.73 ^e	-2.87 ^e	—	420 ^e	442 ^c	(0.49) ^k	3.00	2.70	0.30
Cho <i>et al.</i> ¹⁰¹	m4CzIPN	ITO (120 nm)/PEDOT:PSS (60 nm)/PVK (15 nm)/SiCz:m4CzIPN (25 nm, 1 wt%)/TSPO1 (5 nm)/TPBi (30 nm)/LiF (1 nm)/Al (solution processed)	—	—	—	13.2 ^d	8.2 ^d	0.29, 0.57	—	—	—	—	(0.67) ^g	—	—	—	—
Cho <i>et al.</i> ¹⁴²	4CzIPN	ITO/PEDOT:PSS (60 nm)/TAPC (20 nm)/mCP (10 nm)/4CzIPN:4CzTPN-Ph (25 nm)/TmPyPB (30 nm)/LiF (1 nm)/Al (100 nm)	3.0	—	15.8	—	0.1	0.49, 0.49	-5.2 ^e	-3.1 ^e	—	—	—	—	—	—	—
Uoyama <i>et al.</i> ²¹	4CzIPN 4CzTPN-Ph	ITO/ α -NPD (35 nm), 4CzIPN or CzTPN-Ph:CBP (5 wt%, 15 nm)/TPBi (65 nm)/LiF (0.8 nm)/Al (70 nm)	—	—	—	—	19.3 11.2	—	—	—	—	—	507 (93.8) ^g 577 (26.3) ^g	—	—	—	—
Zhang <i>et al.</i> ¹¹³	4CzTPN-Me 4CzTPN 4CzPN CyFBcz	ITO/NPB (40 nm)/TCTA (10 nm)/8 wt% CyFBcz:mCP (25 nm)/TPBi (35 nm)/LiF (1 nm)/Al (100 nm)	4.1	2526.8 ^k	6.64 ^k	2.66 ^k	—	0.18, 0.13	-5.3 ^e	-2.4 ^b	2.94 ^l	384, 364, 336, 323 ^c	426 ^c	437	2.89 ^m	2.8 ^l	0.09
Mei <i>et al.</i> ¹¹⁰	5CzCF ₃ Ph	ITO/PEDOT:PSS (40 nm)/mCP:5CzCF ₃ Ph (10 wt%, 40 nm)/TmPyPB (60 nm)/LiF (0.8 nm)/Al (100 nm)	3.9	2436	11.8	—	5.2	0.21, 0.33	-5.57 ^e	-2.75 ^e	2.82	290, 319, 330 ^o	481 ^l	495	2.82	2.80	0.02



Table 1 (continued)

Ref.	Compound	Device geometry	Tum- on (V V)	L (cd m^{-2} , (V V))	CE (cd A^{-1})	PE (lm W^{-1})	EQE (%)	CIE (x, y)	HOMO (eV)	LUMO (eV)	ΔE_{gap} (eV)	Abs (nm)	PL nm (QY)	PL (TF)	E_{S_1} (eV)	E_{T_1} (eV)	$\Delta E_{\text{S-T}}$ (eV)
Cho <i>et al.</i> ¹⁰¹	t4CzIPN	(ITO) (120 nm)/PEDOT:PSS (60 nm)/PVK (15 nm)/SiCz:t4C- zIPN (25 nm, 1 wt%)/TSPO1 (5 nm)/TPBI (30 nm)/LiF (1 nm)/Al (solution processed)	—	—	—	42.7 ^d	18.3 ^d	0.31, 0.59	—	—	—	—	(0.78) ^{c-g}	—	—	2.39 ^{f,i}	—
Cho <i>et al.</i> ¹¹⁴	5CzCN	ITO/PEDOT:PSS (60 nm)/PVK (15 nm)/SiCz:5CzCN (25 nm, 15%)/TSPO1 (5 nm)/TPBI (30 nm)/ LiF (1 nm)/Al (200 nm) solution processed	—	—	—	43.4 ^d	18.7 ^d	0.17, 0.27	-6.29 ^e	-3.44 ^e	—	—	(0.71) ^g	—	2.95 ^{n,o}	2.79 ^{n,o}	0.19
Zhang <i>et al.</i> ¹¹¹	5CzBN same as 5CzCN	ITO/HATCN (5 nm)/NPB (30 nm)/ TCTA (10 nm)/mCBP:5CzBN or 5TCzBN (40 wt%) (30 nm)/ DpyPA:Liq (1:1, 30 nm)/LiF (0.5 nm)/Al (150 nm)	3.0	—	—	40.0 ^d	16.7 ^d	0.22, 0.40	-5.55 ^e	-2.74 ^e	—	420 ^c	464 ^c	(0.70) ^f	2.90	2.68	0.22
Tang <i>et al.</i> ¹¹²	4CzCNPy	ITO/PEDOT:PSS (40 nm)/ mCP:4CzCNPy (8 wt%, 30 nm)/ TmPyPB (60 nm)/LiF (0.8 nm)/Al (100 nm)	4.7	16 305 ^d	35.4 ^d	17.9 ^d	10.4 ^d	0.35, 0.59	-5.72 ^e	-3.26 ^b	—	285, 336, 456 ^b	536 ^b , (0.549) ^{c-g}	560	2.28 ^m	2.21 ⁱ	0.07
Cho <i>et al.</i> ¹¹⁶	CNBPCz	PEDOT:PSS (60 nm)/TAPC (20 nm)/mCP (10 nm)/DPE- PO:CNBPCz or DPEPO:CzBPCN 5% (25 nm)/TSPO1 (5 nm)/TPBI (30 nm)/cathode	—	—	—	—	4.8	—	—	—	—	—	458 (0.46) ^{c-g}	—	3.10	2.83	0.27
Cho <i>et al.</i> ¹¹⁷	DDCzIPN	ITO/PEDOT:PSS/TAPC/mCP/ mCP:BmPyPb:DDCzIPN/TSPO1/ Li/Al	3.5	—	—	21.5	18.9 ^d	0.22, 0.46	-6.4 ^e	-3.88 ^e	—	230, 287, 329, 400 ^f	(0.91) ^c	—	2.6 ^{m,n}	2.47 ^{f,i,n}	0.13
Zhang <i>et al.</i> ³⁷	AQ-DTBu-Cz	ITO/HATCN (10 nm)/Tris-PCz (30 nm)/10 wt% TADF:CBP (30 nm)/T2T (10 nm)/Bpy-TP2 (40 nm)/LiF (0.8 nm)/Al	3.0	—	—	—	9 ^d	—	—	—	—	—	(0.58) ^c	—	—	—	0.22
Rajamalli <i>et al.</i> ¹¹⁹	mDCBP	ITO/NPB (40 nm)/mCP (10 nm)/ DPEO:mDCBP (30 wt%) (30 nm)/ PPT (5 nm)/TmPyPb (60 nm)/LiF (1 nm)/Al (100 nm)	2.8	8900 @ 18.0 V	42.8 @ 3.5 V	38.2 @ 3.5 V	14.7 @ 3.5 V	0.22, 0.44 @ 8 V	-5.72 ^e	-2.72 ^b	3.0 ^f	334, 372 ^c	467 ^c	—	—	2.94 ^f	0.15
Rajamalli <i>et al.</i> ¹²⁰	BPy-pC	ITO/NPB (30 nm)/TAPC (20 nm)/ mCBP/dopant [7 wt%] (30 nm)/ PPT (10 nm)/TmPyPb (60 nm)/LiF (1 nm)/Al (100 nm)	—	2183 @ 13 V	5.0 @ 4.0 V	3.9 @ 4.0 V	4.2 (4.0)	0.16, 0.13 @ 8 V	-5.68 ^e	-2.54 ^b	3.14 ^d	290, 338, 359 ^c	440 (0.012) ^c	(0.38)	3.14	2.85	0.29
	BPy-pTC	Same device geometry as above	—	8610 @ 14.5 V	16.3 @ 3.5 V	14.6 @ 3.5 V	9.4 (3.5)	0.17, 0.27 @ 8 V	-5.60 ^e	-2.63 ^b	2.97 ^d	295, 327, 373 ^c	467 (0.186) ^c	(0.70)	2.97	2.84	0.13
	BPy-p2C	Same device geometry as above	—	10 800 @ 12.5 V	20.8 @ 4.0 V	16.2 @ 4.0	11.0 (4.0)	0.18, 0.28 @ 8 V	-5.58 ^e	-2.63 ^b	2.95 ^d	292, 342, 362 ^c	480 (0.178) ^c	(0.72)	2.95	2.88	0.07
	BPy-p3C	Same device geometry as above	—	16 700 @ 12.5 V	56.5 @ 3.5 V	50.6 @ 3.5 V	23.9 (3.5)	0.19, 0.32 @ 8 V	-5.53 ^e	-2.58 ^b	2.93 ^d	291, 341, 363 ^c	482 (0.243) ^c	(0.96)	2.93	2.88	0.05



Table 1 (continued)

Ref.	Compound	Device geometry	Tum- on (V V)	L (cd m^{-2} , (V V))	CE (cd A^{-1})	PE (lm W^{-1})	EQE (%)	CIE (x, y)	HOMO (eV)	LUMO (eV)	ΔE_{gap} (eV)	Abs (nm)	PL nm (QY)	PL (TF)	E_{S_1} (eV)	E_{T_1} (eV)	$\Delta E_{\text{S-T}}$ (eV)
Rajamalli <i>et al.</i> ¹¹⁸	DCBPY	ITO/NPB (30 nm)/mCP (20 nm)/ CzPS:DCBPY (5%, 30 nm)/DPEPO (5 nm)/TmPyPb (60 nm)/LiF (1 nm)/Al (100 nm)	2.8	10 300 @ 10.5 V	54.7 @ 3.0 V	57.2 @ 3.0 V	24.0 @ 3.0 V	0.17, 0.36 @ 8 V	-5.75 ^e	-2.88 ^s	2.87 ^d	311, 400 ^c	490 ^c	514	—	2.841	0.03
	DTCPBY	ITO/NPB (30 nm)/TAPC (20 nm)/ CBP:DTCPBY (5%) (30 nm)/PPT (10 nm)/TmPyPb (55 nm)/LiF (1 nm)/Al (100 nm)	—	37 700 @ 14.5 V	94.6 @ 3.5 V	84.5 @ 3.5 V	27.2 @ 3.5 V	0.30, 0.64 @ 8 V	-5.61 ^e	-2.87 ^s	2.74 ^d	320, 418 ^c	508 ^c	518	—	2.71	0.04
Kim <i>et al.</i> ¹²¹	BPBCz	ITO/PEDOT:PSS (60 nm)/TAPC (20 nm)/BPBCz:DPEPO (25 nm)/ TSPO1 (5 nm)/TPBi (30 nm)/LiF (1.5 nm)/Al (200 nm)	8.7	—	51.5 ^g	46.2 ^g	23.3 ^g	0.21, 0.34	-5.78 ^t	-3.4 ^t	—	—	—	—	2.76	2.92	0.16
Lee <i>et al.</i> ³⁰	MCz-XI	ITO (100 nm)/ α -NPD (40 nm)/mCP (10 nm)/MCz-XI (75 wt%) in PPF (30 nm)/PPF (10 nm)/TPBi (20 nm)/LiF (0.8 nm)/Al (80 nm)	3.6	—	—	—	7.6 ^d	—	—	—	—	—	(0.138) ^c	—	—	—	0.011 ^h
Lee <i>et al.</i> ¹²²	p-TCz-XI	ITO, 100 nm/HAT-CN (10 nm)/ α - NPD (40 nm)/CCP (5 nm)/6 wt%- TADF emitter:PPF (20 nm)/PPF (10 nm)/TPBi (30 nm)/LiF (1 nm)/ Al (80 nm)	—	—	—	—	14.4	0.16, 0.24	-5.73 ^u	-2.81 ^{mm}	—	294, 329(sh) ^c , 340, 360(sh) ^c	454 (0.99) ^c	—	2.73 ^{jj}	2.68 ^{jj}	0.05 ^{jj}
	m-TCz-XI	ITO (100 nm)/TPBi (30 nm)/LiF (1 nm)/ Al (80 nm)	—	—	—	—	12.1	0.18, 0.31	-5.73 ^u	-3.12 ^{mm}	—	294, 328, 341, 365(sh) ^c	477 (0.29) ^c	—	2.64 ^{jj}	2.57 ^{jj}	0.07 ^{jj}
Zhang <i>et al.</i> ¹²³	DiFBu-DPS	ITO/NPB (30 nm)/TCTA (20 nm)/ CzSi (10 nm)/dopant:DPEPO (20 nm) DPEPO (10 nm)/TPBi (30 nm)/LiF (0.7 nm)/Al (100 nm)	—	—	—	—	9.9 ^d	0.15, 0.07	-5.89 ^u	-2.52 ^b	—	—	402 (0.57) ^c	(0.6) ^y	—	—	0.32
Wang <i>et al.</i> ⁵⁹	TXO-PhCz	ITO/PEDOT (30 nm)/TAPC (20 nm)/5 wt% TXO-PhCz:mCP (35 nm)/TmPyPb (55 nm)/LiF (0.9 nm)/Al	4.7	21 000 @ 18.3 V	76	70	21.5	0.31, 0.56	-5.78 ^e	-3.58 ^b	2.25 ^{h,i,w}	305, 345, 410 ^h	—	—	—	2.27 ^{i,x}	0.073
Serevičius <i>et al.</i> ¹²⁸	CzT	Glass/ITO/ α -NPD (30 nm)/TCTA (20 nm)/CzSi (10 nm)/3 wt% CzT:DPEPO (20 nm)/DPEPO (10 nm)/TPBi (30 nm)/LiF (0.8 nm)/Al (70 nm)	—	393 ^k	—	9.7 ^k	6.0	0.23, 0.4	—	—	—	—	453, 472 (LE, hex); 512 (ICT, tol)	—	—	—	0.008 ^c
Lee <i>et al.</i> ¹²⁹	oBFCzTtz	ITO/PEDOT:PSS (60.0 nm)/TAPC (20.0 nm)/mCP (10.0 nm)/DPE- PO:TADF emitter (25.0 nm)/ TSPO1 (5.0 nm)/TPBi (30.0 nm)/ LiF (1.5 nm)/Al (200.0 nm)	—	—	—	—	20.4	0.18, 0.31	-6.1	-3.4	—	—	(97.9) ^z	—	2.994 ^q	2.992 ^r	0.002
	mBFCzTtz	Same device geometry as above	—	—	—	—	13.2	0.17, 0.25	-6.1	-3.3	—	—	(31.1) ^z	—	3.204 ^q	3.013 ^r	0.191
	pBFCzTtz	Same device geometry as above	—	—	—	—	16.7	0.15, 0.18	-6.1	-3.2	—	—	(85.3) ^z	—	3.188 ^q	2.885 ^r	0.302
Yoo <i>et al.</i> ¹³¹	4DPIIA	ITO/PEDOT:PSS (60 nm)/TAPC (20 nm)/mCP (10 nm)/DPEPO:e- mitter (20 wt%; 25 nm)/TSPO1 (5 nm)/TPBi (30 nm)/LiF (1 nm)/Al (200 nm)	—	—	28.0	22.0	13.3	—	-6.09 ^e	-3.20 ^e	—	—	410, 530	470	—	—	0.23



Table 1 (continued)

Ref.	Compound	Device geometry	Tum- on (V V)	L (cd m ⁻² , (cd A ⁻¹))	CE (%)	PE (lm W ⁻¹)	EQE (%)	CIE (x, y)	HOMO (eV)	LUMO (eV)	ΔE_{gap} (eV)	Abs (nm)	PL nm (QY)	PL (TF)	E_{S_1} (eV)	E_{T_1} (eV)	$\Delta E_{\text{S-T}}$ (eV)
Mayr <i>et al.</i> ⁵⁶	CC2TA	ITO (110 nm)/ α -NPD (40 nm)/mCP (10 nm)/CC2TA:DPEPO (6 wt%)/DPEPO (10 nm)/TPBi (30 nm)/0.8 nm LiF/Al (80 nm)	—	—	—	—	11.1	—	—	—	—	—	—	—	—	—	0.05
Kim <i>et al.</i> ¹³²	1CzCzTrz	ITO (120 nm)/PEDOT:PSS (60 nm)/TAPC (10 nm)/mCP (10 nm)/DPEPO:TADF emitters (10%, 25 nm)/TSP01 (5 nm)/TPBi (30 nm)/LiF (1 nm)/Al (200 nm)	—	—	27.5 ^d	21.0 ^d	15.7 ^d	0.17, 0.24	-6.08 ^t	-3.32 ^t	3.60 ^{lf}	—	(0.82) ^{fg}	—	3.00 ^g	2.97 ^r	0.03
	3CzCzTrz	ITO (120 nm)/PEDOT:PSS (60 nm)/TAPC (10 nm)/mCP (10 nm)/DPEPO:TADF emitters (10%, 25 nm)/TSP01 (5 nm)/TPBi (30 nm)/LiF (1 nm)/Al (200 nm)	—	—	28.4 ^d	21.8 ^d	12.4 ^d	0.22, 0.36	-6.08 ^t	-3.27 ^t	3.50 ^{lf}	—	(0.62) ^{fg}	—	2.92 ^g	2.80 ^r	0.12
	13CzCzTrz	Same device geometry as above	—	—	27.7 ^d	19.1 ^d	15.7 ^d	0.17, 0.25	-6.09 ^t	-3.32 ^t	3.43 ^{lf}	—	(0.85) ^{fg}	—	2.98 ^g	2.97 ^r	0.01
Endo <i>et al.</i> ⁵	PIC-Trz	ITO/ α -NPD/m-CP/6 wt%-PIC-TRZ:m-CP/BP4mPy/LiF/Al	—	10 000 ^d	—	—	5.3 ^h	—	—	—	—	—	466 (0.35) ^f	—	2.66 ^m	2.55 ⁱ	0.11
Kim <i>et al.</i> ¹²¹	TrzBCz	ITO/PEDOT:PSS (60 nm)/TAPC (20 nm)/TrzBCz:DPEPO (20 wt%, 25 nm)/TSP01 (5 nm)/TPBi (30 nm)/LiF (1.5 nm)/Al (200 nm)	—	—	60.9 ^d	58.4 ^d	23.5 ^d	0.23, 0.44	-5.79 ^e	-3.40 ^e	—	—	—	—	2.90 ^g	2.74 ^r	0.16
Kim <i>et al.</i> ¹³⁰	DCzTrz	ITO (50 nm)/PEDOT:PSS (60 nm)/TAPC (20 nm) mCP (10 nm)/DPE-PO:TADF emitter (20%, 25 nm)/TSP01 (5 nm)/TPBi (30 nm)/LiF (1 nm)/Al (200 nm)	—	—	26.8 ^d	22.4 ^d	17.8 ^d	0.15, 0.16	-5.4 ^z	-2.18 ^z	—	411 ^l	(43) ^{c,fg}	—	2.89 ^g	2.64 ^r	0.25
Lee <i>et al.</i> ¹⁴⁵	TCzTrz	ITO (50 nm)/PEDOT:PSS (60 nm)/TAPC (20 nm) mCP (10 nm)/DPE-PO:TADF emitter (x %, 25 nm)/TSP01 (5 nm)/TPBi (20 nm)/LiF (1 nm)/Al (200 nm)	—	—	—	42.7 ^d	25 ^d	0.18, 0.33	-5.19 ^z	-2.11 ^z	—	414 ^l	100 ^{c,fg}	—	2.96 ^g	2.8 ^r	0.16
	TmCzTrz	ITO (50 nm)/PEDOT:PSS (60 nm)/TAPC (20 nm) mCP (10 nm)/DPE-PO:TADF emitter (x %, 25 nm)/TSP01 (5 nm)/TPBi (20 nm)/LiF (1 nm)/Al (200 nm); doping: TCzTrz (40%); TmCzTrz (30%)	—	—	—	52.1 ^d	25.5 ^d	0.25, 0.50	-5.26 ^z	-2.15 ^z	—	447 ^l	99 ^{c,fg}	—	2.86 ^g	2.79 ^r	0.07
Kim <i>et al.</i> ¹³³	23TCzITrz	ITO (120 nm)/PEDOT:PSS (60 nm)/TAPC (10 nm)/mCP (10 nm)/emitter (20 wt%) in DPEPO/TSP01 (5 nm)/TPBi (30 nm)/LiF (1 nm)/Al (200 nm)	—	—	28.4 ^d	22.3 ^d	10.7 ^d	0.20, 0.26	-5.86 ^t	-3.19 ^t	3.01 ^b	—	(0.82) ^{fg}	—	3.08 ^g	2.75 ^r	0.33
	33TCzITrz	Same device geometry as above	—	—	64.3 ^d	57.7 ^d	25 ^d	0.23, 0.42	-5.74 ^t	-3.21 ^t	2.94 ^b	—	(0.87) ^{fg}	—	3.01 ^g	2.76 ^r	0.25
	34TCzITrz	Same device geometry as above	—	—	19.1 ^d	15.0 ^d	10.3 ^d	0.16, 0.20	-5.98 ^t	-3.19 ^t	3.06 ^b	—	(0.65) ^{fg}	—	3.14 ^g	2.79 ^r	0.35
Obolida <i>et al.</i> ¹³⁴	TPA-TAZ	ITO/MoO ₃ (6 nm)/NPB (30 nm)/TPA-TAZ (20 nm)/TPBi (50 nm)/LiF (0.8 nm)/Al (100 nm)	—	7323 ^d @ 9 V	—	—	6.8 ^d	0.158, 0.043	-5.28 ^e	-2.12 ^e	3.22 ^{aa}	339 ^f	415 ^f	428 ^{bb}	—	—	0.5
Sasabe <i>et al.</i> ¹³⁵	AcCz-2TP	ITO/TPAP:PPBi (20 nm)/TAPC (20 nm)/mCP (5 nm)/10 wt% AcCz-2TP:mCP (10 nm)/B3PyPB (50 nm)/LiF (0.5 nm)/Al (100 nm)	3.06	—	12.6 @ 3.06 V	12.7 @ 3.06 V	9.2	0.16, 0.18 ^{cc}	-5.62 ^u	-2.33 ^b	3.29	310	461	—	3.00	2.75	0.25
Data <i>et al.</i> ²⁹	t-BuCz-DBPHZ	ITO/NPB (40 nm)/t-BuCz-DBPHZ (10 wt%)/CBP (20 nm)/TPBi (20 nm)/BCP (20 nm)/LiF (1 nm)/Al (100 nm)	3.7	17 000	—	—	8.0	—	-5.79 ^e	-3.37 ^e	—	449 ^{dd}	457 ^{dd} (0.61)	509 (0.31)	2.77	2.34	0.43
Cai <i>et al.</i> ¹³⁶	DC-TC	ITO/HATCN (5 nm)/TAPC (20 nm)/TADF emitter:CBP (5 wt%, 35 nm)/TmPyPB (55 nm)/LiF (1 nm)/Al (100 nm)	3.4	—	19.2 ^d	10 ^{cc}	6.2 ^d	0.349, 0.555 ^{cc}	-5.56 ^t	-2.9 ^b	2.58 ^f	389 ^c	553 (0.6) ^{eee}	—	—	—	0.13
	PyCN-TC	ITO/HATCN (5 nm)/TAPC (20 nm)/TADF emitter:CBP (5 wt%, 35 nm)/TmPyPB (55 nm)/LiF (1 nm)/Al (100 nm)	3.4	—	26.7 ^d	3.6 ^{cc}	8.1 ^d	0.396, 0.547 ^{cc}	-5.56 ^t	-3.26 ^b	2.3 ^f	429 ^c	532 (53.8) ^{eee}	—	—	—	0.43

Table 1 (continued)

Ref.	Compound	Device geometry	L Turn-on (V) (cd m ⁻² , on (V) V)	CE (cd A ⁻¹)	PE (lm W ⁻¹)	EQE (%)	CIE (x, y)	HOMO (eV)	LUMO (eV)	ΔE_{gap} (eV)	Abs (nm)	PL nm (QY)	PL (TF)	E_{T_1} (eV)	E_{S_1} (eV)	$\Delta E_{\text{S-T}}$ (eV)
Takahashi <i>et al.</i> ¹³⁷	m-ATP-CDP ITO (100 nm)/ α -NPD (35 nm)/mCP (10 nm)/6 wt%-emitter:mCP (15 nm)/PPT (10 nm)/TPBi (40 nm)/LiF (0.8 nm)/Al (70 nm)	Glass substrate/ITO/PEDOT:PSS (30 nm)/TCTA: 8 wt% TADF emitter (40 nm)/3TPYMB (50 nm)/LiF (0.6 nm)/Al (100 nm)	4.8 ^{ff} 3290 ^d	13.4 ^{cc}	6.4 ^{cc}	7.5 ^d	—	-5.7 ^h	-3.1 ^b	2.5	303, 370 ^{cc,gg}	532 (0.40) ^{c,gg}	—	3.02 ^h	2.76 ^h	0.26
Shiu <i>et al.</i> ¹³⁸	PtFPCz	ITO/ α -NPD (35 nm)/mCP (10 nm)/LiF (0.8 nm)/Al (80 nm)	2–2.5	22.2 ^d	24.9 ^d	7.6 ^d	—	-5.39 ^e	-2.81 ^e	2.58 ^e	350 ^c	390–410, 485 ^c	522 (0.4)	—	—	-0.057 ^{fr}
Furue <i>et al.</i> ¹³⁹	PtFCzP	ITO/ α -NPD (35 nm)/mCP (10 nm)/LiF (0.8 nm)/Al (80 nm)	2–2.5	11.8 ^d	13.3 ^d	4.8 ^d	—	-5.44 ^e	-2.72 ^e	2.72 ^e	350 ^c	390–410, 485 ^c	518 (0.38)	—	—	-0.065 ^{fr}
	PCZ-CB-TRZ	ITO/ α -NPD (35 nm)/mCP (10 nm)/EML (20 nm)/PPT (40 nm)/LiF (0.8 nm)/Al (80 nm)	6.3	16.7 ^d	7.6 ^d	11.0 ^d	—	-6.4 ^e	-3.05 ^b	3.35	(278, 322, ^{gg} 336 ^{gg}) ^c	346, 467 (0.03) ^{hh}	557 (0.97) ⁱⁱ	2.406 ^{jj}	2.403 ^{jj}	0.003 ^{jj}
	2PCZ-CB	Same device geometry as above	4.4	19.9 ^d	11.2 ^d	9.2 ^d	—	-6.33 ^a	-2.80 ^b	3.53	(292, 323, ^{gg} 337 ^{gg}) ^c	350 (0.01) ^{hh}	571 (0.94) ⁱⁱ	2.482 ^{jj}	2.464 ^{jj}	0.018 ^{jj}
Zhu <i>et al.</i> ¹⁴⁰	PdN3N	ITO/HATCN (10 nm)/NPD (40 nm)/TAPC (10 nm)/6% emitter: 26 mCPy (25 nm)/DPPS (10 nm)/BmPyPb (40 nm)/LiF/Al	—	—	—	20.9	0.30, 0.61	kk	kk	—	<400	522 (0.79) ^o	0.72	2.47 ^{ll}	2.32 ^{ll}	0.15
	PdN3O	Same device geometry as above	—	—	—	20.4	0.30, 0.62	kk	kk	—	<400	(0.76) ^o	0.76	2.54 ^{ll}	2.36 ^{ll}	0.18

L: luminance; CE: current efficiency; PE: power efficiency; EQE: external quantum efficiency; α -NPD: 4,4-bis[N-(1-naphthyl)-N-phenylamino]biphenyl; mCP: 1,3-bis(N-carbazolyl)benzene; PPT: 2,8-bis(diphenylphosphoryl)dibenzol[*b,d*]thiophene; TPBi: 1,3,5-tris(N-phenylbenzimidazole-2-yl)benzene; TSP01: diphenylphosphine oxide-4-(triphenylsilyl)phenyl; PPF: 2,8-bis(diphenylphosphoryl)dibenzol[*b,d*]furan; HAT-CN: 2,3,6,7,10,11-hexacyano-1,4,5,8,9,12-hexaazatriphenylene; CCP: 9-phenyl-3,9-bicarbazole; TmPyPb: 3,3'-[5'-(3-pyridinyl)-phenyl][1,1':3',1'-terphenyl]-3,3'-diyl]bis-pyridine; HATCN: dipyrzino[2,3-f:2',3'-h]quinoline-2,3,6,7,10,11-hexacarboxylic diimide; NPB: N,N'-bis(naphthalene-1-yl)-N,N'-bis(phenyl)-benzidine; T2T: 2,4,6-tris(biphenyl-3-yl)-1,3,5-triazine; TCTA: 4,4',4''-tris(N-carbazolyl)triphenylamine; DPVPA-Liq: 9,10-bis(3-(pyridin-3-yl)phenyl)anthracene-(8-hydroxyquinolinolato)-lithium; PVK: poly(9-vinylcarbazole); SiCz: diphenylid(4-(9-carbazolyl)phenyl)silane; TPAC: 1,1-bis(4-di-*p*-tolylaminophenyl)ether oxide; Tris-PCz: 9,9'-diphenyl-6-(9-phenyl-9H-carbazol-3-yl)-9H,9'H-3,3'-bicyclopentadiene-bis[N,N-bis(4-methylphenyl)aniline]; DPEPO: bis[2-(diphenylphosphino)phenyl] ether oxide; Tris-PCz: 9,9'-diphenyl-6-(9-phenyl-9H-carbazol-3-yl)-9H,9'H-3,3'-bicyclopentadiene-bis[N,N-bis(4-methylphenyl)aniline]; DPAP: triphenylamine-containing polymer; PPBi: 4-isopropyl-4'-methylidiphenyl-iodonium tetrakis(pentafluorophenyl)borate; 3TPYMB: tris[3-(3-pyridyl)mesityl]borane. LE: locally excited state; ICT: intermolecular charge transfer state; hex: hexane; tol: toluene. ^a Photoelectron spectroscopy. ^b HOMO-absorption edge. ^c Measured in toluene. ^d Maximum value. ^e CV experiment. ^f Measured in THF. ^g N₂ atmosphere. ^h Measured in thin film. ⁱ First phosphorescence peak zero-zero energy (E_{0-0}) measured at low temperature. ^j Measured in DCM. ^k Maximum value. ^l Estimated from UV-vis absorption edge. ^m Zero-zero energy (E_{0-0}) of the fluorescence spectrum at room temperature. ⁿ 1% in PS film. ^o Measured at low temperature. ^p Measured in CHCl₃. ^q Calculated from onset of fluorescence. ^r Calculated from onset of phosphorescence. ^s HOMO – E_{g} . ^t Calculated from onset of phosphorescence. ^u Photoelectron yield spectroscopy of thin film. ^v DPEPO film (10 wt%). ^w 1240/ λ_{onset} . ^x Measured in 2-MeTHF. ^y Low current density. ^z Calculated by TD-DFT at the B3LYP/6-31G(d) level of theory. ^{aa} Solution, 3.16 V for film. ^{bb} FWHM 55 nm. ^{cc} Value recorded at 100 cd m⁻². ^{dd} Measured in cyclohexane. ^{ee} At atmosphere. ^{ff} Value recorded at 1 cd m⁻²; oxygen-free solution. ^{gg} Shoulder. ^{hh} THF/water mixture. ⁱⁱ Neat film deposited on a quartz substrate. ^{jj} Calculated by TD-DFT at the PBE1PBE/6-31G(d) level. ^{kk} Oxidation and reduction potentials noted in ESI. ^{ll} Measured at room temperature; mm $E_{\text{S}}-E_{\text{T}}$ as determined from the Arrhenius plot of rate constant for RISC. ^{mm} LUMO = HOMO + E_{g} (optical energy gap).





Table 2 Polymeric and dendrimeric carbazole-based TADF emitters

Ref.	Emitter	Device geometry	Turn-on (V)	L cd m ⁻²	CE (cd A ⁻¹)	PE (lm W ⁻¹)	EQE (%)	CIE (x, y)	HOMO (eV)	LUMO (eV)	Abs (nm)	Em (nm) (QY)	Em (nm) (TF)	E _s (eV)	E _r (eV)	ΔE _{s-r} (eV)		
Polymeric TADF emitters																		
Zhu <i>et al.</i> ¹⁴⁶	PAPCC PAPTC	ITO/PEDOT:PSS (50 nm)/	3.0 ^a	554 ^b	3.6	3.67	1.34	0.25, 0.47	-5.38 ^c	-2.57 ^d	—	472 ^e (9 ^f)	487 (8)	—	—	—		
		PAPTC or PAPCC (40 nm)/	2.6 ^a	10251 ^b	41.8	37.1	12.63	0.30, 0.59	-5.33 ^c	-2.77 ^d	—	510 ^e (22 ^f)	510 (28)	—	—	—		
		TmPyPB (50 nm)/LiF (1 nm)/Al (100 nm)	—	—	—	—	—	—	—	—	—	—	—	—	—	—		
Lee <i>et al.</i> ⁶	pCzBP	ITO/PEDOT:PSS (40 nm)/10 wt% pCzBP:TCTA:TAPC blend (40 nm)/TmPyPB (50 nm)/LiF (0.8 nm)/Al (80 nm)	6	5100 ^b	24.9	9	8.10	0.28, 0.43	-5.41 ^g	-2.76 ^d	307, 361	472 (28)	508 (23)	—	2.69	0.18		
		ITO/PEDOT:PSS (30 nm)/GnTAZ (35 nm)/TPBI (40 nm)/Ca (10 nm)/Al	3.3	—	—	—	2.40	0.251, 0.493	-5.76 ^g	-3.01 ^d	—	—	—	—	2.77	2.74	0.03	
Albrecht <i>et al.</i> ¹⁴⁹	G2TAZ G3TAZ G4TAZ	Same device geometry as above	3.5	—	—	—	3.40	0.266, 0.485	-5.72	-2.97	—	—	—	2.79	2.74	0.06		
		ITO/PEDOT:PSS (30 nm)/GnTAZ (35 nm)/TPBI (40 nm)/Ca (10 nm)/Al	3.5	—	—	—	1.50	0.232, 0.368	-5.68	-2.8	—	—	—	—	2.86	2.79	0.06	
		ITO/PEDOT:PSS (30 nm)/CDE1 (devices A1-A4) or CDE2 (devices B1-B4) (70 nm)/TPBI (40 nm)/LiQ (2 nm)/Al	4.8	>10 000	—	—	—	12 ^b	0.38, 0.56	-5.12 ^h	-2.54 ^d	289, 299, 349 ⁱ	520 (77)	—	—	—	—	0.11
		ITO/PEDOT:PSS (40 nm)/EML (40 nm)/TPBI (40 nm)/LiQ (1.6 nm)/Al (100 nm)	7.7	>10 000	—	—	—	5.2 ^b	0.32, 0.51	-5.25 ^h	-2.69 ^d	289, 298, 348 ⁱ	499 (75)	—	—	—	—	0.15
Luo <i>et al.</i> ¹⁴⁸	CzDMAC-DPS DCzDMAC-DPS	ITO/PEDOT:PSS (40 nm)/EML (40 nm)/TPBI (40 nm)/LiQ (1.6 nm)/Al (100 nm)	4.0	—	30.6 ^j	24 ^b	12.2 ^b	0.22, 0.44	-5.24 ⁱ	-2.31 ^c	240, 299, 498	492	492	2.95	2.86	0.09		
		ITO/PEDOT:PSS/TA-Cz or TA-3Cz/TPBI/Cs ₂ CO ₃ /Al	2.6 ^a	25 085 ^b	18.2 ^b	14.3 ^b	2 ^b	2.2 ^b	0.18, 0.27	-5.18	-2.09 ^b	240, 299, 484	464	464	3.07	2.87	0.2	
Sun <i>et al.</i> ¹⁵⁰	TA-Cz TA-3Cz	Same device geometry as above	2.4 ^c	23 145 ^b	39.0 ^b	40.8 ^b	11.8 ^b	0.39, 0.56	-4.92 ^c	-2.28 ^c	238, 263, 295, 341, 408 ^k	56	56	2.49 ^e	2.33 ^e	0.17		
		ITO/PEDOT:PSS (30 nm)/P9 DV-CDBP and DV-MOC-DPS in mass ratios of 1:0.09	5.3	899 ^l	1.6	0.9 ^b	2.00 ^b	0.12, 0.13	-5.29	-	-	300, 351 ^e	367	367	3.63	2.95	—	
Thermally cross-linkable TADF emitters																		
Sun <i>et al.</i> ¹⁵¹	DV-CDBP (host) DV-MOC-DPS (emitter)	ITO/PEDOT:PSS (30 nm)/P9 DV-CDBP and DV-MOC-DPS in mass ratios of 1:0.09	5.3	899 ^l	1.6	0.9 ^b	2.00 ^b	0.12, 0.13	-5.29	-	356 ^e	430 ^e	—	—	—	—		
		Cs ₂ CO ₃ (2 nm)/Al (100 nm)	—	—	—	—	—	—	—	—	—	—	—	—	—	—	—	

L: luminance; CE: current efficiency; PE: power efficiency; EQE: external quantum efficiency; TmPyPB: 3,3'-[5'-[3-(3-pyridinyl)-phenyl][1,1':3',1''-terphenyl]-3,3''-diyl]bispyridine; TPBI: 1,3,5-tris(phenyl)benzimidazole-2-yl)benzene; Ca calcium; Cs₂CO₃ cesium carbonate; Liq 8-hydroxyquinolinolato-lithium. ^a At a luminance of 1 cd m⁻². ^b Maximum value. ^c CV experiment. ^d Estimated from the HOMO level and the absorption edge. ^e Measured in toluene. ^f Measured in air. ^g PES of the film in air. ^h Calculated from onset voltages of oxidation peak with the equation of $-(V_{\text{onset}} + 4.78)$ eV. ⁱ Thin film. ^j Measured in DCM.

Table 3 Carbazole-based host materials for TADF

Ref.	Compound	TGA (°C)	T _g (°C)	HOMO (eV)	LUMO (eV)	λ _{abs} (nm)	λ _{fl} (nm)	E _T (eV)	E _g (eV)	Emitter	Device geometry	Turn-on (V)	CE (cd A ⁻¹ V ⁻¹)	EQE (%)	CIE (x, y)	L (cd m ⁻²)
Li <i>et al.</i> ¹⁷	<i>o</i> -CzCN	350–440	94	-5.74 ^a	-2.16 ^c	295, 328, 341 ^b	403	3.01 ^c	3.58 ^d	2CzPN	ITO/PEDOT:PSS (40 nm)/TAPC (20 nm)/host:2CzPN (4 wt%)/TmPyPB (40 nm)/LiF (1 nm)/Al (200 nm)	5.1	29.23 ^e	18.36 ^e	0.17, 0.23	2746 ^e
	<i>m</i> -CzCN	350–440	121	-5.62 ^a	-2.14 ^c	296, 329, 344 ^b	403	2.81 ^c	3.48 ^d	—	ITO/PEDOT:PSS (40 nm)/TAPC (20 nm)/host:5 wt% 4CzIPN (20 nm)/TmPyPB (40 nm)/LiF (1 nm)/Al (200 nm)	4.8	26.37 ^e	16.56 ^e	0.17, 0.26	4992 ^e
	<i>p</i> -CzCN	350–440	140	-5.59 ^a	-2.16 ^c	296, 332, 343 ^b	406	2.77 ^c	3.43 ^d	—	ITO/PEDOT:PSS (40 nm)/TAPC (20 nm)/host:5 wt% 4CzIPN (20 nm)/TmPyPB (40 nm)/LiF (1 nm)/Al (200 nm)	3.8	14.41 ^e	11.12 ^e	0.17, 0.28	7856 ^e
Li <i>et al.</i> ¹⁷¹	<i>o</i> -CzDPz	333	70	-5.69 ^f	-2.16 ^c	293, 327, 340	397	3.02	3.53 ^d	4CzIPN	ITO/PEDOT:PSS (40 nm)/TAPC (20 nm)/host:5 wt% 4CzIPN (20 nm)/TmPyPB (40 nm)/LiF (1 nm)/Al (200 nm)	4.8	39.6 ^e	23.7 ^e	0.26, 0.54	12780 ^e
	<i>m</i> -CzDPz	350	83	-5.63 ^f	-2.13 ^c	293, 326, 341	403	2.83	3.5 ^d	4CzIPN	ITO/PEDOT:PSS (40 nm)/TAPC (20 nm)/host:5 wt% 4CzIPN (20 nm)/TmPyPB (40 nm)/LiF (1 nm)/Al (200 nm)	4.5	37.5 ^e	22.0 ^e	0.30, 0.57	24050 ^e
	<i>p</i> -CzDPz	378	89	-5.61 ^f	-2.21 ^c	288, 348	380	2.78	3.4 ^d	4CzIPN	ITO/PEDOT:PSS (40 nm)/TAPC (20 nm)/TAPC (20 nm)/mCP (10 nm)/TPBI (60 nm)/LiF (1 nm)/Al (200 nm)	3.8	41.1 ^e	32.2 ^e	0.31, 0.58	19890 ^e
Nishimoto <i>et al.</i> ²⁰	<i>m</i> CPDPz	434	140	-5.67 ^f	-2.15 ^c	292, 326, 340	407	2.76	3.52 ^d	4CzIPN	Same device geometry as above	4.9	37.9 ^e	19.8 ^e	0.22, 0.49	12130 ^e
	PzCz	474	—	-6.4 ^h	-2.5 ^g	287, 315	332	3.00 ⁱ	3.9	CzTPN	ITO/α-NPD (35 nm)/mCP (10 nm)/3 wt%-CzTPN:host (20 nm)/PPT (40 nm)/LiF (0.8 nm)/Al (80 nm)	4.20	37.2	24.4	0.18, 0.45	—
Cho <i>et al.</i> ¹⁷²	DCzDCN	—	—	-6.14 ^f	-3.26 ^f	236, 251, 280, 291	427	2.71	2.88	4CzIPN	ITO (50 nm)/PEDOT:PSS (60 nm)/TAPC (20 nm)/mCP (10 nm)/DCzDCN:4CzIPN (3%, 25 nm)/TSPO1 (35 nm)/LiF (1 nm)/Al (200 nm)	3	—	—	—	—
Gaj <i>et al.</i> ¹⁷³	mCPSOB	—	110	-5.8	-2.5	—	—	3.02 ^c	3.3	4CzIPN	ITO/MoO ₃ (15 nm)/Poly-TriCz (50 nm)/mCPSOB:4CzIPN (25 nm)/TPBI (60 nm)/LiF (1 nm)/Al (100 nm)	4.8 ⁱ	81	79	21.5 ^j	—
Kim and Lee ³³	mCP	312	50–60	-6.1	-2.4	—	—	2.9 ^k	3.7	4CzIPN	ITO (50 nm)/PEDOT:PSS (60 nm)/TAPC (20 nm)/mCP (10 nm)/(mCP: BmPyPb 1:1):4CzIPN (25 nm)/TSPO1 (35 nm)/LiF (1 nm)/Al (200 nm)	—	—	—	—	—
	BmPyPb	—	—	-6.4	-2.7	—	—	2.78 ^k	3.7	—	ITO (50 nm)/PEDOT:PSS (60 nm)/TAPC (20 nm)/mCP (10 nm)/(mCP: BmPyPb 1:1):4CzIPN (25 nm)/TSPO1 (35 nm)/LiF (1 nm)/Al (200 nm)	—	—	—	—	—
Zhao <i>et al.</i> ¹⁷⁶	<i>o</i> -mCPBI	400	130	-5.44 ^f	-1.86 ^g	324, 338	—	3.00	3.58	4CzIPN	ITO/MoO ₃ (5 nm)/TAPC (65 nm)/host:guest (15 nm)/TmPyPb (35 nm)/LiF (1 nm)/Al (100 nm)	3.4 ⁱ	60.4 ^e	42.0 ^e	0.27, 0.58	—
	<i>m</i> -mCPBI	394	124	-5.4 ^f	-1.84 ^g	324, 338	—	2.80	3.56	—	ITO/MoO ₃ (5 nm)/TAPC (65 nm)/host:guest (15 nm)/TmPyPb (35 nm)/LiF (1 nm)/Al (100 nm)	—	—	—	—	—
	<i>p</i> -mCPBI	429	141	-5.44 ^f	-1.89 ^g	324, 338	—	2.71	3.55	—	ITO/MoO ₃ (5 nm)/TAPC (65 nm)/host:guest (15 nm)/TmPyPb (35 nm)/LiF (1 nm)/Al (100 nm)	—	—	—	—	—
Ban <i>et al.</i> ¹⁷⁷	TZ-Cz	382	96	-5.2 ^f	-2.31 ^g	236, 265, 295, 379 ^m	490 ^b	2.8 ⁱ	2.89 ^d	Self-host emitter	ITO/PEDOT:PSS (25 nm)/TZ-Cz or TZ-3Cz (35 nm)/Cs ₂ CO ₃ (2 nm)/Al (100 nm)	40.00	20.0 ^e	6.5 ^e	0.24, 0.51	18200
	TZ-3Cz	407	128	-5 ^f	-2.11 ^g	238, 263, 293, 379 ^m	487 ^b	2.8 ⁱ	2.89 ^d	Self-host emitter	ITO/PEDOT:PSS (25 nm)/TZ-Cz or TZ-3Cz (35 nm)/Cs ₂ CO ₃ (2 nm)/Al (100 nm)	3.60	30.5 ^e	10.1 ^e	0.24, 0.51	22950
Ban <i>et al.</i> ¹⁵⁶	tbCz-SO	380	80	-5.51 ^f	-2.32	235, 265, 298, 350 ^m	440 ^b	2.91	3.19 ^d	Self-host emitter	ITO/PEDOT:PSS (tbCz-SO or 5.1 ^l poCz-SO)/TmPyPb/Cs ₂ CO ₃ /Al	—	—	—	0.16, 0.19	—
	poCz-SO	410	113	-5.6 ^f	-2.41	231, 278, 291, 348 ^m	458 ^b	2.90	3.19 ^d	Self-host emitter	ITO/PEDOT:PSS (tbCz-SO or 5.1 ^l poCz-SO)/TmPyPb/Cs ₂ CO ₃ /Al	6.1 ^l	10.5 ^e	6.2 ^e	0.18, 0.27	—





Table 3 (continued)

Ref.	Compound	TGA (°C)	T _g (°C)	HOMO (eV)	LUMO (eV)	λ _{abs} (nm)	λ _{em} (nm)	QY	E _{T1} (eV)	E _g (eV)	Emitter	Device geometry	Turn-on (V)	CE (cd A ⁻¹)	PE (lm W ⁻¹)	EQE (%)	CIE (x, y)	L (cd m ⁻²)
Kang <i>et al.</i> ¹⁷⁹	ZDZ	—	—	-5.71 ^f	-2.19	294, 327, 341 ⁿ	378 ⁿ	378 ⁿ	2.94 ^c	3.51 ^d	2CzPN	ITO (50 nm)/HATCN (7 nm)/TAPC (75 nm)/ZDZ or 2CzPN 6 wt% host (ZDZ or ZDZ) (20 nm)/TmPyPB (50 nm)/LiF (1.5 nm)/Al (100 nm)	5.00	10.72 ^j	—	18.5 ^j	0.17, 0.34	3231
	ZDN	—	—	-5.72 ^f	-2.27	294, 341 ⁿ	378 ⁿ	378 ⁿ	2.92 ^c	3.45 ^d	2CzPN	ITO (50 nm)/HATCN (7 nm)/TAPC (75 nm)/ZDZ or 2CzPN 6 wt% host (ZDZ or ZDZ) (20 nm)/TmPyPB (50 nm)/LiF (1.5 nm)/Al (100 nm)	4.70	14.29 ^j	—	25.7 ^j	0.17, 0.34	6366
Cao <i>et al.</i> ¹⁵⁷	pCnBCzmmMe	447	141	-5.31 ^f	-2.18 ^g	298, 339 ^m	463 ^m (51/63 ^o)	2.69	3.13	4CzCNPy	ITO/PEDOT:PSS (40 nm)/(pCnBzmMe or pCNBCzoCF ₃ or pCNBCzmCF ₃ :4CzCNPy (40 nm)/TmPyPB (60 nm)/LiF (0.8 nm)/Al (100 nm)	3.2 ⁱ	27.8 ⁱ	16.8 ⁱ	8.8 ⁱ	0.31, 0.60	16100 ^e	
	pCNBCzoCF ₃	400	165	-5.39 ^f	-2.47 ^g	289, 327 ^m	534 ^m (13/20 ^o)	2.64	2.92	—	ITO/HAT-CN (10 nm)/TAPC (35 nm)/host:10 wt% DPAC-TRZ (20 nm)/TSPO1 (10 nm)/TPBi (40 nm)/LiF (0.8 nm)/Al (120 nm)	3.7 ⁱ	26.5 ⁱ	13.4 ⁱ	8.0 ⁱ	0.32, 0.61	14370 ^e	
	pCNBCzmCF ₃	402	134	-5.41 ^f	-2.45 ^g	285, 358 ^m	543 ^m (43/55 ^o)	2.64	2.96	—	ITO/HAT-CN (10 nm)/TAPC (35 nm)/host:10 wt% DPAC-TRZ (20 nm)/TSPO1 (10 nm)/TPBi (40 nm)/LiF (0.8 nm)/Al (120 nm)	3.3 ⁱ	26.4 ⁱ	13.5 ⁱ	8.0 ⁱ	0.33, 0.60	19200 ^e	
Cui <i>et al.</i> ¹⁵⁴	29Cz-BID-BT	—	—	-6.01 ^p	-2.55 ^q	270–300, 325, 360	360	360	3.02	3.46	DPAC-TRZ	ITO/HAT-CN (10 nm)/TAPC (35 nm)/host:10 wt% DPAC-TRZ (20 nm)/TSPO1 (10 nm)/TPBi (40 nm)/LiF (0.8 nm)/Al (120 nm)	—	—	—	20.8	0.16, 0.34	—
	39Cz-BID-BT	—	—	-6.07 ^p	-2.62 ^q	270–300, 325, 360	360	360	3.04	3.45	—	ITO/HAT-CN (10 nm)/TAPC (35 nm)/host:10 wt% DPAC-TRZ (20 nm)/TSPO1 (10 nm)/TPBi (40 nm)/LiF (0.8 nm)/Al (120 nm)	—	—	—	20.4	0.16, 0.34	—
Ding <i>et al.</i> ¹⁸⁰	9CzFDPEPO	511	191	-6.07 ^f	-2.39 ^f	341, 327, 283, 349, 366, 263, 229 ^m	383 ^m (49%) ^q	3.0 ⁱ	3.68	DMAC-DPS	ITO/MoO ₃ (6 nm)/NPB (70 nm)/mCP (5 nm)/9CzFXPO:DMAC-DPS (10%, 3.50 20 nm)/DPEPO (5 nm)/BPhen (40 nm)/LiF (1 nm)/Al	3.50	31.3	28.1	16.7	0.15, 0.30	—	
	9CzFDPEPO	474	211	-6.07 ^f	-2.52 ^f	341, 329, 281, 349, 366, 228 ^m	383 ^m (58%) ^q	3.0 ⁱ	3.55	DMAC-DPS	ITO/MoO ₃ (6 nm)/NPB (70 nm)/mCP (5 nm)/9CzFXPO:DMAC-DPS (10%, 3.50 20 nm)/DPEPO (5 nm)/BPhen (40 nm)/LiF (1 nm)/Al	3.50	25.1	22.4	13.2	0.15, 0.30	—	

CE: current efficiency; PE: power efficiency; EQE: external quantum efficiency; L: luminance; TAPC: 4,4'-(cyclohexane-1,1-diyl)bis(N-phenyl-N-p-tolylamine); TCTA: N,N,N'-tris(4-(9-carbazolyl)phenyl)amine; TmPyPB: 1,3,5-tri(*m*-pyrid-3-ylphenyl)benzene; α-NPD: 4,4'-bis[N-(1-naphthyl)-N-phenyl-aminobiphenyl]; TSP01: diphenylphosphine oxide-4-(triphenylsilyl)phenyl; TPBi: 2,2',2'-(1,3,5-benzenetriyl)tris[1-phenyl-1*H*-benzimidazole]; DPAC-TRZ: 10-(4-(4,6-diphenyl-1,3,5-triazin-2-yl)phenyl)-9,9-diphenyl-9,10-dihydroacridine; MoO₃: molybdenum trioxide; Poly-TiCl₃: tris-carbazole polymer; mCPSOB: 3,5-di(carbazol-9-yl)-1-phenylsulfonobenzene; HATCN: dipyrzino[2,3-f:2',3'-h]quinoxaline-2,3,6,7,10,11-hexacarbonitrile; NPB: N,N'-bis(1-naphthalenyl)-N,N'-diphenyl-1,1'-biphenyl-4,4'-diamine; DPEPO: bis(2-(diphenylphosphino)phenyl)ether oxide; BPhen: 4,7-diphenyl-1,10-phenanthroline. 2CzPN: 4,5-di(9*H*-carbazol-9-yl)phthalonitrile; 4CzPN: 1,2,3,5-tetrakis(carbazol-9-yl)-4,6-dicyanobenzene; CzIPN: 2,5-bis(carbazol-9-yl)-1,4-dicyanobenzene; 4CzCNPy: 4,7-diphenyl-1,10-phenanthroline. 2CzPN: 4,5-di(9*H*-carbazol-9-yl)phthalonitrile; 4CzPN: triazin-2-yl)phenyl)-9,9-diphenyl-9,10-dihydroacridine; DMAC-DPS: bis[4-(9,9-dimethyl-9,10-dihydroacridine)phenyl]sulfone. ^a vs. SCE (4.4 eV). ^b Thin film. ^c 2-MeTHF at 77 K. ^d From absorption edge. ^e Maximum value. ^f From cyclic voltammetry. ^g Calculated from HOMO-optical gap. ^h Photoelectron yield spectroscopy in thin film. ⁱ From 0–0 transition in phosphorescence. ^j At 1000 cd m⁻². ^k PL at liquid nitrogen temperature. ^l Applied voltage for 1 cd m⁻². ^m In DGM. ⁿ In CHCl₃. ^o Under N₂ atmosphere. ^p Photoelectron spectroscopy. ^q vs. 9,10-diphenylanthracene.

Carbazole-based host materials for TADF applications

Organic host materials for PHOLEDs including carbazole-based materials were thoroughly reviewed by Tao *et al.*¹⁵² The article highlights hole-transporting, electron-transporting and hybrid bipolar transport materials. A thorough representation of the application of these materials is presented in a tabulated format for host properties and device characteristics. Particular challenges remain in the creation of bipolar hosts for blue phosphors, including improvement of efficiency roll-off and the overall simplification of devices in general. Solution processability thereby is presented by way of polymers. Fundamentally, however, small oligomers are key to the development of efficient, stable PHOLED devices. In general, host materials are conductive environments that serve to avoid aggregation-induced quenching of the long-lived triplet excitons and an effective system for energy transfer between host and guest emitting species. Host materials applied to TADF applications are cited herein along with physical, spectroscopic and device characteristics, Table 3. The host material for TADF applications is an important component in TADF devices as it influences the electroluminescence, lifetime and fluorescence quantum yield.¹⁵³ Design principles are common for materials toward first and second generation OLED applications such as high physical and morphological stability, good film-forming abilities, bipolar charge transport properties for balanced hole and electron densities in the emitting layer, as well as appropriate alignment of the HOMO and LUMO with adjacent layers to facilitate charge injection.¹⁷ The criteria specific to function as a host material for TADF applications require efficient spectral overlap with the dopant (TADF emitter), a higher triplet energy level (E_T) compared to the dopant to avoid reverse energy transfer and a widened HOMO–LUMO energy gap (E_g) compared to the dopant so that charges get trapped significantly only on the dopant.²⁰ Herein, the widened HOMO–LUMO gap is a significant drawback as this leads to increased drive voltage of the devices. The molecular design for high triplet energy includes building blocks with inherent high triplet energies (carbazole) and a handle on the π -conjugation length through modulation of the linker units and incorporation of distortions in the molecule.¹⁷ For bipolar charge transport properties, a strong intramolecular charge transfer between donor and acceptor units is needed, which may lower the triplet energy level. Bipolar charge transport is also needed for high emission efficiency and ensuring low efficiency roll-off. Thus, a disruption of the π -conjugation-based connectivity between the donor and acceptor units by insulating, saturated atoms (sp^3 -hybridized C or Si) or a twisted π -conjugated spacer is favorable¹⁵⁴ for maintaining a triplet energy level and blocking of intramolecular electronic coupling, yet negatively effects carrier transport properties.¹⁵⁵ The role of the host specifically serves to separate the emitting species to reduce the concentration of high energy excited state species to reduce loss processes through photophysical processes, notably triplet–triplet annihilation,^{19,136} and device efficiency, *i.e.* roll-off behavior.³⁵

The host serves as a matrix for the emitting species and thus phase-separation and crystallization are of concern to ensure device stability particularly upon heating as well as during device operation.¹⁵⁶ These concerns were addressed by the development of dendrimeric self-host materials comprising a combination of host and emitting species as outlined below. The donor component in a host material is often based on carbazole or derivatives thereof due to its inherent high triplet energy (E_T) and good hole transport ability.¹⁵⁷ Of interest is the acid/base chemistry of the nitrogen at position 9, as well as the electrochemically active positions 3 and 6 of nascent carbazole *vide supra*.^{91,92} A range of non-carbazole-based hosts were developed specifically for TADF applications including twisted and spirocyclic phosphine oxides¹⁵⁸ and acridine-based sulfones.¹⁵⁹ In addition, Sandanayaka *et al.* highlight the significant role the host – as opposed to the emitter – plays in device stability – or the lack thereof.¹⁶⁰

Known carbazole-based host materials include TCTA, CzSi, and CBP, which were utilized by Zhang *et al.* in TADF applications.¹⁴¹ TCTA^{161–163} is characterized by a high glass transition temperature of 151 °C.¹⁶⁴ CzSi is known for a high glass-transition temperature of > 131 °C and hole mobilities of $5 \times 10^{-5} \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$.^{88,165} CBP is a widely used material^{10,166} which tends to crystallize, suffers from a low glass transition temperature of merely 62 °C,¹⁶⁷ and has a known decomposition profile as investigated by Kondakov.¹⁶⁸ In addition, the low triplet energy of 2.56 eV results in inefficient energy transfer from the host to the guest and thus to poor device characteristics.¹⁶⁹

Tanaka *et al.* presented a wide-energy gap material of 3.26 eV and high T_1 energy (2.92 eV) entitled 3,4-di(9H-carbazol-9-yl)benzotrile (2CzBN¹⁷⁰) for use in TADF applications originally as an emitter. Since no TADF activity was found, the research group utilized 2CzBN as a host in various devices.

Li *et al.* designed three host materials *o*-, *m*-, and *p*-CzCN¹⁷ and provided evidence that among the three constitutional isomers of *o*-, *m*-, and *p*-CzCN a balanced charge-carrier mobility as well as sufficiently high singlet and triplet energy level are additional factors leading to better TADF device performance for both *m*-, and *o*-CzCN compared to *p*-CzCN, Fig. 11.

Li *et al.*¹⁷¹ prepared bipolar host materials, several containing carbazole units for the p-type component and pyrazole for the n-type component in a ratio of 2 : 1 for *o*-CzDPz, *m*-CzDPz, and 3-CzDPz with a control compound of 1 : 1 ratio *m*CPDPz, Table 3. All materials exhibit high enough triplet energy to be hosts for all types of (RGB) emitters, Fig. 11.

Cho *et al.* introduced a universal host material entitled DCzDCN for both PHOLED and TADF applications and created a material with a similar HOMO energy level, however, a deeper LUMO level compared to other carbazole-based hosts. A device containing DCzDCN compared to CBP showed a decrease in turn-on voltage by 0.5 V, the absence of roll-off behavior, and a 20-fold improved lifetime coupled, however, with limited luminance since the device utilizing 4CzIPN (1,2,3,5-tetrakis(carbazol-9-yl)-4,6-dicyanobenzene) did not reach 1000 cd m^{-2} , Fig. 11.¹⁷²



Gaj *et al.*¹⁷³ developed **mCPSOB** as a new host material for TADF applications characterized by ambipolar charge properties. Under the application of 4CzIPN as a TADF emitter, an EQE of 21.5% at the luminance value of 1000 cd m⁻² was recorded. At the same time, a low roll-off at high current densities was recorded. This was attributed to the high energy level of the triplet state of **mCPSOB** (3.02 eV), which suppresses triplet exciton quenching, Fig. 11.

Host materials can be tuned by mixing as presented by Kim and Lee.³³ The known host material **mCP** shows only a glass transition temperature of 65 °C, a HOMO/LUMO energy level of -5.68 eV/-2.17 eV and a triplet energy level of 2.9 eV.^{174,175} A mixed host system, however, with a deep HOMO level combined with a high singlet energy and triplet energy level was achieved by mixing **mCP:BmPyPb**. This allows for the suppression of exciton quenching. The system performed as an exciplex-free host system with the green emitting species 4CzIPN, Fig. 11 and Table 3.

Zhao *et al.* studied *o*-**mCPBI**, *m*-**mCPBI**, and *p*-**mCPBI** as host materials for TADF applications.¹⁷⁶ In comparison to **mCP**, these new materials showed higher T_g and thus the presence of benzimidazole significantly improved the morphological stability. The geometric parameters of *o*- vs. *m*- vs. *p*-linkages modulated the photophysical properties and specifically the triplet state around the values of 3.00, 2.80, and 2.71 eV, respectively, Fig. 11.

Nishimoto *et al.* developed an organic-inorganic hybrid material hexakis(9*H*-carbazol-9-yl)cyclotriphosphazene (**PzCz**) (Fig. 11) to serve as a host material in TADF devices and 2,5-bis(carbazol-9-yl)-1,4-dicyanobenzene (CzTPN) as a blue-green TADF emitter.²⁰ **PzCz** is characterized by high triplet energy ($E_T = 3.0$ eV) and showed external electroluminescence quantum efficiencies of 15–18% when combined with CzTPN and 4CzIPN, Fig. 11.

Ban *et al.* constructed dendrimeric structures as self-host TADF materials. The combination of host and emitting species on the dendrimeric molecule avoids phase separation.^{156,177} **Tz-Cz** and **Tz-3Cz**¹⁷⁷ molecules included alkyl chains to allow a non-conjugated linkage between the host and the emitting species, Fig. 11. This concept led to increased solubility, avoided crystallization, and improved charge transport and the insulation of the emissive core from the surrounding host material as was previously applied to PHOLED-hosts.¹⁷⁸ For dendrimeric materials **tbCz-SO** and **poCz-SO**, Ban *et al.* combined *t*-butyl-carbazoles and phosphine oxide carbazoles,¹⁵⁶ wherein the phosphine oxide carbazoles afforded high thermal stability, balanced charge transport, and stable color purity, Fig. 12.

Kang *et al.*¹⁷⁹ synthesized new, bipolar host materials containing carbazole, pyridindole, and dibenzothiophene entitled **ZDZ** and **ZDN** for blue-emitting TADF applications. Utilizing the emitter **2CzPN**, the host containing all aforementioned units, **ZDN**, exhibited an EQE of 25.7%. Among these, **ZDN** has very low current density attributed to a large k_{RISC} along with charge balance. The improved performance of the **ZDN** host material is attributed to the smaller non-radiative decay rate in comparison to other host materials, Fig. 12.

Solution-processed host materials **pCNBCzmMe**, **pCNBCzoCF₃**, and **pCNBCzmCF₃** (Fig. 12) were introduced by Cao *et al.*¹⁵⁷ These new materials are based on bicarbazoles and Cao *et al.*¹⁵⁷ were able to create new materials with improved solubility, electron-transport properties and electrochemical stability compared to nascent biscarbazole.

The application of benzimidazobenzothiazole in combination with carbazoles led to another type of bipolar host entitled **29Cz-BID-BT** and **39Cz-BID-BT** (Fig. 12) as developed by Cui *et al.*¹⁵⁴ Therein, a disruption of the donor-acceptor π -conjugated system was achieved utilizing the sp³-hybridization of the nitrogen atom of carbazole linked on the benzimidazobenzothiazole acceptor units. Cui *et al.*¹⁵⁴ utilized DFT and triplet-spin density models to localize electron density firmly on the carbazole unit. Both hosts have high triplet energy levels of 3.02 eV and 3.04 eV, respectively. The group was able to prepare blue TADF devices with a high EQE of 20.8 and 20.4% with low efficiency roll-off, respectively.

Hosting blue-emitting TADF materials places specific requirements on the host materials such as high triplet energy that allows for exothermic triplet host-dopant energy transfer. Ding *et al.*¹⁸⁰ investigated the special effects governing hosts by preparing new quaternary phosphine oxide hosts **9CzFDPEPO**, and **9CzFDPEPO**, which included π -conjugated extender systems for the former and for the latter, a doubling of the phosphine oxide units compared to 9CzFSPO,¹⁵⁵ in which the electron mobility was found to be proportional to the number of phosphine oxide units present, Fig. 12. The high triplet energy of 3.0 eV was preserved and HOMO/LUMO energy levels of -6.1 eV and -2.5 eV were observed for effective energy transfer and carrier injection, which allowed for the observation of high external quantum efficiencies of 22.5% for **9CzFDPEPO**.

Overall, newly developed host materials pivoted around high triplet energy, bipolar behavior and solution processability. New materials were designed by a variation of substitution patterns, and host property variation by mere mixing up to dendrimerization of emitters. Acceptor units included nitrile, pyrazole and its derivatives, sulfoxide, phosphine oxide, benzodithiophene, and benzimidazobenzothiazole.

Computational analysis as applied on materials for TADF applications

DFT¹⁸¹ computations have been utilized to model materials for TADF applications and along with the extension of time-dependent DFT to fundamentally understand the electronic and photophysical properties leading to TADF behavior. The observed charge-transfer nature of TADF materials along with the small singlet-triplet splitting of ≤ 0.1 eV requires a careful choice of the level of theory in order to achieve balance between efficiency (qualitative) and accuracy (quantitative). Standard exchange correlation functionals (SVWN, BLYP, or B3LYP) are known to underestimate excitation energies (HOMO-LUMO gap)¹⁸² due to a mismatched exchange correlation, electron self-interaction or delocalization error, derivative discontinuity



or electron–electron potential at large distances. The incorrect asymptotic behavior observed in the orbital energies of the CT excited states is the result of a self-interaction problem. Thereby, the term electron-transfer self-interaction is used because the orbital that accepts the electron exhibits coulombic repulsion between both the accepting and donating orbitals, which is not present in the CT state, yet not canceled by the time-dependent density functional theory (TD-DFT) calculations except if exact exchange is present.

A survey of literature presenting quantum-chemical calculations in the context of the synthesis of TADF materials and their characterization in TADF devices revealed that the B3LYP functional combined with the 6-31G(d) basis set predominates for geometry optimizations of the ground state and computations involving excited states. Other utilized functionals include CAM-B3LYP, M06, M06-2X, PBE0 (25% HF) and HF. The basis sets range from STO-3G, to 6-31+G(d), D98(d,p), 6-311+G(d,p), and cc-pVDZ, LANL2DZ, Table 4.

Tapping into the biological toolbox, Shu and Levine presented a computational, simulated evolution approach for the identification of fluorophores for OLED application under the caveat of TADF functionality namely small S_1 - T_1 gap and large S_1 transition dipole moment.¹⁹⁴ The group applied a tree-based genetic algorithm with eight genetic optimizations utilizing a molecular space of candidate fluorophores utilizing parole, thiophene, furan, pyridine, pyridine, and benzene acceptor

groups with cyan-, aldehyde, carboxyl-, trifluoromethyl and 2,2-dicyanoethenyl EWG and carbazolyl, phenothiazinyl, carbolinyl, thieno(3,2-*b'*-indolyl), and indolyl EDG. Initial optimizations were run at the B3LYP/STO-3G level of theory followed by optimization of S_0 at the CAM-B3LYP/6-31G(d,p) level of theory and the T_1 state computed at the RO-CAM-B3LYP/6-31G(d,p) level of theory and the S_1 state at the CAM-B3LYP/6-31G(d,p) level of theory to compute the S_1 - T_1 gap along with the S_0 - S_1 transition dipole moment as utilized in the optimization. The limitation of using the CAM-B3LYP functional did not allow the assessment of the target compounds for emission wavelength and the S_1 and T_1 computed at S_0 geometry rather than the Franck–Cotton point – a notion leading to an expected error of <0.1 eV. A range of possible molecules is suggested with a need to include synthetic accessibility into the algorithm. On the selected 7518 molecules, Shu and Levine¹⁹⁴ presented the known example of carbazolyl and dicyanobenzene examples as exploited by Uoyama *et al.*²¹ and pyridines bound to various electron donors (without withdrawing group), 3-cyano- or 3,5-dicyanopyridines bound to various electron donors, carbazolyl pyridines with carboxyl or aldehyde withdrawing groups as already explored by Tang *et al.*,¹⁹⁹ carbazolyl cyanopyrazines, and phenothiazinyl 3,4-di(2,2-dicyanoethenyl)-furans.

Chen *et al.*²⁰⁰ tackled the experimental observation that DAD systems under special circumstances show singlet excitons exceeding the theoretical limit of 25% and deviate while having

Table 4 Survey of applied levels of theory in DFT calculations for assessment of the ground state and excited states in TADF materials

Ref.	Ground state geometry optimization		TD-DFT for excited states	
	Functional	Basis set	Functional	Basis set
Albrecht <i>et al.</i> ¹⁴⁹	HF, PM6 ^a	6-31G	<i>b</i>	<i>b</i>
Ding <i>et al.</i> ¹⁸⁰	B3LYP	6-31G(d)	B3LYP	6-31G(d)
Huang <i>et al.</i> ¹²⁶	B3LYP	6-31G(d)	B3LYP	6-31G(d)
Ishimatsu <i>et al.</i> ¹⁸³	B3LYP	6-31+G(d) ^c	<i>b</i>	<i>b</i>
Ishimatsu <i>et al.</i> ¹⁸⁴	M06-2X	6-31G(d)	M06-2X	6-31G(d)
Kitamoto <i>et al.</i> ¹⁸⁵	M06	6-31G(d)	M06	6-31G(d)
Komino <i>et al.</i> ³⁵	B3LYP	6-31G(d)	B3LYP	6-31G(d) ^d
Lee <i>et al.</i> ¹⁸⁶	B3LYP	6-31G(d)	BLYP, MPW1B95, BMK, M062X, and M06HF ^e	6-31G(d)
Lee <i>et al.</i> ³¹	CAM-B3LYP	cc-pVDZ	CAM-B3LYP	cc-pVDZ
Lee <i>et al.</i> ⁴⁹	CAM-B3LYP	cc-pVDZ	CAM-B3LYP	cc-pVDZ
Li <i>et al.</i> ¹⁸⁷	B3LYP	6-31G(d)	<i>b</i>	<i>b</i>
Linfoot <i>et al.</i> ¹⁸⁸	B3LYP	LANL2DZ	B3LYP	LANL2DZ
Liu <i>et al.</i> ¹⁸⁹	B3LYP	6-31G(d)	<i>b</i>	<i>b</i>
Lu <i>et al.</i> ¹⁹⁰	PBE0 (25% HF)	cc-pVDZ	PBE0, B3LYP, CAM-B3LYP	cc-pVDZ
Mayr <i>et al.</i> ⁵⁶	B3LYP	6-31G(d,p)	B3LYP	6-31+G(d,p)
Park <i>et al.</i> ¹⁹¹	B3LYP, PBE0	6-31G(d)	PBE0	6-31G(d)
Sagara <i>et al.</i> ¹⁹²	PBE0	6-31G(d)	PBE0	6-31G(d)
Shizu <i>et al.</i> ⁷⁶	M06-2X	cc-pVDZ	<i>b</i>	<i>b</i>
Shizu <i>et al.</i> ¹⁹³	M06-2X	cc-pVDZ	M06-2X	cc-pVDZ
Shu <i>et al.</i> ¹⁹⁴	B3LYP, then CAM-B3LYP	STO-3G then 6-31G(d,p)	CAM-B3LYP (TD, TDA)	6-31G(d,p)
Tanaka <i>et al.</i> ¹⁹⁵	CAM-B3LYP	cc-pVDZ	CAM-B3LYP	cc-pVDZ
Tanaka <i>et al.</i> ³⁷	PBE0	6-31G(d)	PBE0	6-31G(d)
Tsai <i>et al.</i> ⁴¹	B3LYP	6-311G(d)	<i>b</i>	<i>b</i>
Wang <i>et al.</i> ¹⁹⁶	B3LYP	6-31G(d)	B3LYP	6-311+G(d)
Wu <i>et al.</i> ¹⁹⁷	B3LYP	6-31G(d)	<i>b</i>	<i>b</i>
Xie <i>et al.</i> ¹⁹⁸	B3LYP	6-31G(d,p)	M06-2X	6-31G(d)
Zhang <i>et al.</i> ⁴⁷	B3LYP	6-31G(d)	B3LYP; optimal HF exchange method	6-31G(d)
Zhang <i>et al.</i> ¹²³	B3LYP	6-31G(d)	B3LYP	6-31G(d)
Zhao <i>et al.</i> ¹⁷⁶	B3LYP	6-31G(d)	<i>b</i>	<i>b</i>

^a Semiempirical level (PM6) ^b No data related to these calculations is reported; 6-31G(d). ^c The reported basis set 6-31+(d) was corrected to 6-31+G(d). ^d The reported basis set 6-31(d) was corrected to 6-31G(d) as confirmed by the authors. ^e Tabulated data does not specify which particular functional was applied for the reported results of each compound.



Table 5 Comparative and benchmarking computational studies of a range of functionals and basis sets for the assessment of TADF materials in ground state and excited states as well as singlet–triplet gap

Ref.	Functional(s)	Basis set(s)
Uoyama <i>et al.</i> ²¹ Moral <i>et al.</i> ²⁰¹	PBE0 (25% HF), B3LYP, CAM-B3LYP, ω B97X-D, M06-2X PBE0, B2-PLYP, B2GP-PLYP	6-31G(d) 6-31G(d), def2-TZVP, def2-TZVP/JK and def2-TZVP/C def2-TZVP (def2-SVP for H)
Penfold <i>et al.</i> ²⁰² Liang <i>et al.</i> ²²	PBE with Grimme's D3 dispersion scheme and Becke–Johnson damping scheme B3LYP (20% HF) compared to BLYP (0% HF), MPWLYP1M (5% HF), B3LYP* (15% HF), PBE0, MPW1B95 (31% HF), BMK (42% HF), M06-2X (56% HF), M06-2X (56% HF), M06-HF (100% HF); excited states at (CAM-B3LYP, ω B97X-D); OHF-method	6-31G(d)
Huang <i>et al.</i> ²⁰³	BLYP (0% HF), MPWLYP1M (5% HF), TPSSH (10% HF), B3LYP (15% HF), B3LYP (20% HF), PBE0, MPW1B95 (31% HF), BMK (42% HF), M06-2X (56% HF), M06-HF (100% HF); Cam-B3LYP, LC- ω PBE, ω B97X-D, LC-BLYP; OHF method	6-31G(d)
Zhang <i>et al.</i> ¹⁴¹ Sun <i>et al.</i> ²⁰⁴	B3LYP, PBE0, MPW1B95, BMK, M06-2X and M06-HF PBE (0% HF), B3LYP (20% HF), M062X (56% HF), M06-HF (100% HF) as well as CAM-B3LYP, LC- ω PBE, ω B97X-D	6-31G(d) 6-31G(d) to 6-31+G(d), 6-311G(d), 6-311+G(d), and 6-311++G(d,p)

larger than recommended energy gaps. Under application of potential energy scans along donor–acceptor angles utilizing TD-DFT using the BMK functional combined with the 6-31G(d) basis set, the group identified conical intersections when the acceptors are at 90° where degenerate molecular geometries exist. At points close to 90°, the conical intersection is accessible and thermal energy is enough to couple T₁ geometry with the conical intersection and thus allows non-adiabatic coupling to promote T₁ → S₁ RISC rendering the butterfly type chromophore important for TADF applications.

The following is an overview of comparative and benchmarking computational studies, which tackle the electronic and photophysical properties of carbazole-containing TADF materials, Table 5. Uoyama utilized DFT calculations to optimized geometries of the singlet ground (S₀) and first excited state (S₁) along with the first excited triplet (T₁) state and thus to predict CDCB properties. The cyanogroup inhibits non-radiative deactivation and minimizes modifications to the geometries of S₁ and T₁.²¹ Chromatic tunability is integrated through the number and substituent pattern on the dangling carbazole units. Among the functionals PBE0, B3LYP, CAM-B3LYP, ω B97X-D and M06-2X that were utilized, the latter parameters performed best for absorption wavelength, singlet–triplet energy gap, charge transfer character and geometric features of dihedral angles between geometries, Table 5.

Moral *et al.* specifically explored the singlet–triplet energy gap in OLED materials with particular care on studying the charge-transfer character of the materials employed by identifying and validating a theoretical model, bracketing the influence on calculations and analyzing which excitations led to the singlet and triplet states obtained.²⁰¹ The ground state geometry is best modeled using a large basis set and dispersion correction, *i.e.* the PBE0-D3(BJ)/def2-TZVP level of theory. The $E_{VA}(S_1)$ and $E_{VA}(T_1)$ were modeled utilizing a Tamm–Dancoff approximation (TDA) approach, which showed a negligible effect on $E_{VA}(S_1)$ up to 0.11 eV and a significant effect on $E_{VA}(T_1)$ up to 0.4 eV independent on functional choice (TDA-PBE0/6-31G(d) *vs.* TDA-B3LYP/6-31G(d)). The impact, thus, transpires to ΔE_{ST} with an improved maximum deviation from 0.58 eV to 0.23 eV. Intramolecular dispersion effects did not affect the ground state geometry with a maximum deviation of 0.01 eV,

however, the larger basis sets (def2-TZVP and def2-QZVP) modulate $E(S_1)$ by ± 0.1 eV yet to a smaller degree for $E(T_1)$ with ± 0.05 eV, where a convergence with the smaller, less costly def2-TZVP basis set was shown.

Comparison with experimental values based on the onset of absorption and emission spectra necessitates the comparison with $E_{00}(S_1)$ and $E_{00}(T_1)$, respectively where the application of PBE0-D2(BJ)/def2-TZVP showed a close agreement between experimental and theoretical values with MAD and RMSD of 0.1 and 0.06 eV. Obtaining $E_{00}(S_1)$ showed convergence problems. Immersed in a solvent environment, molecules with CT component require the non-equilibrium polarizable continuum model (PCM) leading to MAD and RMSD of 0.11 and 0.13. A completed natural transition orbital (NTO) analysis keeps the excitation energies the same, however, does compact all the transitions associated with an excitation to one single electron–hole pair. Using this approach, an electron–hole distance Δr was computed and a cutoff-value of > 1.5 – 2 Å used to determine the presence of a CT excitation and an inverse relationship between distance and ΔE_{ST} values.^{205,206}

Penfold²⁰² surveyed 31 molecules for overlap between the HOMO and LUMO to develop a qualitative approach for predicting the singlet–triplet energy gap for effective computational screening of TADF materials. In a second approach, optimally tuned range-separated LC-BLYP functionals were utilized to quantitatively assess ΔE_{ST} and oscillator strength.

Noteworthy in this context is the work of Liang *et al.*,²² who focused on using DFT and TD-DFT to calculate the ΔE_{ST} for carbazole-based TADF emitters. Carbazole-based TADF-emitters undergo significant structural relaxation upon excitation to the singlet and triplet excited states with relaxational energies ranging from 0.4 to 0.5 eV for the singlet excited state and 0.1 to 0.2 eV for the triplet excited state. Using the adiabatic excitation method, Liang *et al.*²² were able to show that the functional B3LYP and basis set 6-31G(d) was sufficient to model the ΔE_{ST} for a series of carbazole-based TADF emitters deviating by only 0.00 eV to 0.02 eV from the experimental data since the substantial relaxation effects are accounted for, values that are significantly smaller in comparison to the optimal Hartree–Fock (OHF) method with deviation from 0.3 to 0.4 eV. A comparison of the adiabatic method along a broad range of functionals demonstrated that functionals with more than 25% HF are not



suitable (PBE0, MPW1B95, BMK), long-range corrected functionals overestimate adiabatic ΔE_{ST} since the intramolecular CT transitions are shorter-ranged and the results of BLYP, MWLYP1M, and B3LYP* (<20% HF exchange) are close to the experimental values, however, these functionals underestimate $E_{00}(S_1)$ and $E_{00}(T_1)$. B3LYP* indicates 15% HF exchange. The OHF method localizes the lowest energy excited state as LE, while the lowest excited triplet state is of CT nature, which can only correctly be modeled using the adiabatic method. Thus, B3LYP (20% HF exchange) along with the 6-31G(d) basis set under application of the adiabatic method results in theoretically calculated values for ΔE_{ST} closest to experimental values for carbazole-based TADF emitters, Table 5.

Huang *et al.*²⁰³ explored the computational prediction of singlet- and triplet-transition energies for charge-transfer compounds. Pure TD-DFT functionals do not consider electrostatic interactions between the separated charges in the CT states. Thus, a detailed exploration to determine the optimal percentage of Hartree-Fock (OHF) in the TD-DFT calculation of $E_{VA}(S_1)$ was carried out. The influence of the HF exchange percentage (%HF) in XC functionals on the calculated $E_{VA}(S_1)$ was observed by screening 10 functionals ranging from 0% to 100% HF in correlation to the charge-transfer amount (q). The value of " q " was determined using orbital composition analysis.²⁰⁷ A series of 17 compounds were screened. Based on experimental data, the best %HF was assigned. Overall, the OHF equals $42q$ and the relative error of vertical transition energy for each functional is dependent on the CT amount. Since the CT state is always more polar than the ground state, calculation of E_{VE} is carried out in the aromatic solvent toluene. Overall, reproduction of experimental data is achieved with an error of ± 0.06 eV.

The prediction of T_1 is difficult as this triplet state may be a CT or LE state founded in either the donor or acceptor. Evaluation of CT vs. LE states was achieved by studying the change in $E_{VA}(S_1) - E_{VA}(T_1)$ as a function of %HF and an independent calculation of $E_{0-0}({}^3CT)$ and (3LE). The assumption that 3CT and 1CT have the same orbital transitions and the same CT amount allows an error correction to be developed using $E_{0-0}({}^3CT) = E_{0-0}(S_1) - [E_{VA}(S_1, OHF) - C \times E_{VA}(T_1, BLYP)]$. In this formula, $C = E_{VA}(S_1, OHF)/E_{VA}(S_1, BLYP)$. The correction factor C was also applied in the correction for the LE state using the formula $E_{0-0}({}^3LE) = [E_{VA}(T_1/C)] - \Delta E_{Stokes}$. This procedure allowed for an effective identification of LE vs. CT states in good agreement with the experimental data. The group reported that range-separated hybrid functionals (CAM-B3LYP, LC- ω PBE, wB97XD and LC-BLYP) dramatically overestimate the $E_{VA}(S_1)$ in the studied molecular group.

Hybrid functionals are used in DFT as approximations to the exchange and correlation energy terms. Therein, DFT and exact Hartree-Fock (HF) exchange energies are utilized. CT states in molecules are generally more sensitive to the percent of Hartree-Fock than LE states. Zhang *et al.*¹⁴¹ optimized structures with the B3LYP functional and the 6-31G(d) basis set. Thereby, a percent of Hartree-Fock close to OHF was utilized in order to calculate the change in energy during non-adiabatic (vertical) excitation from S_0 to S_1 . The OHF ratio equal to $42q$ ²⁰³

was utilized. Only a minimal difference in the CT amount q is observed for the S_0 and S_1 geometries in ICT molecules and therefore the group applied the geometry optimization of S_1 utilizing a functional with a HF% close to OHF for calculations involving vertical absorption energy from S_0 to S_1 ($E_{VA}(S_1)$) to minimize error.

Sun *et al.*²⁰⁴ have explored the use of tuned range-separated functionals to calculate the singlet-triplet gap in a wide range of organic emitters including emitters for TADF applications to overcome the limitations of standard exchange functionals *vide supra*. The observed charge-transfer in TADF systems was studied using range-separated exchange density functionals for TD-DFT calculations under the application of an empirical tuning process for the range separation parameter ω (ω).²⁰⁸ Ground state geometries were optimized using the B3LYP/6-31G(d) level of theory and the range separation parameter ω was optimally tuned for the LC- ω PBE functional with the 6-31+G(d) basis set. Considering the vertical excitation processes, the lowest singlet $E_{VA}(S_1)$ and triplet $E_{VA}(T_1)$ excited states and vertical single-triplet gaps ΔE_{ST} . In addition, adiabatic singlet-triplet gaps ΔE_{ST^*} were determined for three groups of model compounds showing singlet-triplet gaps categorized as large gaps such as **PhCz** and **CBP** [0.55–0.8 eV], moderate gaps such as **PIC-TRZ**, **DTC-DPS**, **CC2TA**, **2CzPN**, and **4CzPN** [0.15–0.5 eV] and very small gaps such as **4CzIPN**, **4CzTPN**, and **4CzTPN-Me** [0.00–0.10 eV]. In the above mentioned range of compounds, only carbazole-containing materials were chosen. The group investigated functionals including pure GGA, PBE (0% eX), hybrid GGA: B3LYP (20% eX), meta GGA functionals M062X (56% eX) and M06HF (100% eX) as well as three range-separated functionals LC- ω PBE and ω B97XD and CAM-B3LYP with short-range \sim long range eX% ranging from 19–65%, 0–100%, and 22–100%, respectively. A range of basis sets from 6-31G(d) to 6-31+G(d), 6-311G(d), 6-311+G(d), and 6-311++G(d,p) were employed. Fundamentally, the 6-31+G(d) basis set exhibited good accuracy associated with a moderate computational cost. The range separation parameter ω (ω) is unaffected by basis set extension from 6-31G(d) to 6-31+G(d). For comparison with experimental values derived from delayed fluorescence or phosphorescence spectra taken in solution (usually in toluene), CAM-B3LYP and UCAM-B3LYP along with the PCM model (toluene) were applied for calculation of adiabatic singlet-triplet gaps ΔE_{ST^*} . When comparing TD-DFT vs. TDA-DFT calculations for tuned LC- ω PBE* functionals, the TDA approach shows improved description of ΔE_{ST} due to the better description of the triplet excitation energies along with lower computational costs and increased stability²⁰⁹ during computation of ΔE_{ST} . A direct comparison of the results of TD-DFT calculations using the B3LYP/6-31G(d) level of theory optimized in the ground state followed by the LC- ω PBE/6-31G(d) level of theory for TD/TDA-DFT computations with LC- ω PBE functional showed negligible differences.²¹⁰

A thorough analysis of mean absolute deviation (MAD), relative error (RE), and linear correlation coefficients (R^2) and a contrast to high level calculations using coupled cluster CC2 theory and double-hybrid functionals such as that completed



by Moral *et al.*²⁰¹ concludes that optimally tuned RC functionals such as the tuned LC- ω PBE functional deliver accurate results at reasonable computational costs as exemplified by the MAD of only 0.15, 0.07, and 0.09 for the calculated $E_{\text{VA}}(S_1)$, ΔE_{ST} , and ΔE_{ST^*} , respectively, when compared to the experimental values in toluene. Tian *et al.*²¹¹ reviewed several approaches contrasting the works of Huang *et al.*²⁰³ and Sun *et al.*²⁰⁴

Recently published work from the Brédas²¹² research group computationally addressed k_{RISC} , spin-orbit coupling (SOC), and ΔE_{ST} in a series of known and experimentally well-characterized TADF materials, *i.e.* 2CzPN, CC2TA, PIC-TRZ, PXZ-TRZ, ACRFLCN, spiro-CN, 4CzIPN, 4CzIPN-Me, 4CzPN, 4CzTPN, and 4CzTPN-Me along with two non-TADF materials CBP, and α -NPD. The group utilized the B3LYP/6-31G(d) level of theory to optimize the ground state, CAM-B3LYP, as well as TD-DFT and uSCF for singlet and triplet excited state modeling, respectively under application of PCM for toluene. Optimal ω values were tuned and applied using the best performing range-separated LC- ω PBE/6-31+G(d) level of theory under application of TDA to compute excitation energies and LE *vs.* CT contributions. In addition, the SOC matrix was evaluated under application of the COSMOS continuum solvation model and LC- ω PBE functional as well as reorganization energies assessed under application of the DUSHIN program. Among other findings, the study concluded that the spatial separation of HOMO and LUMO are not the only factors contributing to a small ΔE_{ST} . Improvements are predicted to come from stabilizing the lowest energy triplet state as a charge-transfer state and strengthening the SOC so as to approach El-Sayed rule character in the system all the while ensuring a high PLQY. An outlook for experimental approaches and theoretical approaches to contribute further to the field is provided.

Conclusions and future outlook

While TADF-based OLED devices reach near 100% internal quantum efficiency, and an external quantum efficiency of 20–30%, current challenges in TADF-based OLED devices remain staggering. While the field is gaining momentum with the number of publications rising tremendously, there is still a tremendous need to address the challenge to achieve a low drive voltage at high quantum efficiency in order to optimize the device power efficiency, color purity and FWHM for blue emitters, as well as roll-off behavior. Part of this challenge is rooted in the fact that carbazole-containing TADF-emitting materials carry interrupted π -conjugation, which depresses charge mobility. Increasing conjugation, however, may compromise triplet energy. What is staggeringly absent is a broad, diverse range of materials with high stability and reliable luminescence. The presence of wide bandgap materials causes additional interface barriers and challenges in operating voltage of OLEDs, which should be close to the energy of the emitted photons. For a future outlook, the synthetic preparation of highly soluble, versatile, tunable organic materials will enable solution processing and ultimately the development of cheap, consumer-good devices; however, this solution processability is still in its infancy. A breakthrough, ultimately, in clever and deployable material

design strategies will be enabled by a thorough understanding of the photophysics that drives the repopulation of the singlet excited state from the triplet state and a stabilization of the material in its triplet state during device operation, which is fundamentally rooted in smart device construction. This review shows the tremendous contribution carbazole-based materials have made to the field, both as emitters, and as hosts, wherein carbazole-containing host materials substituted particularly with benzimidazolbenzothiazole and phosphine oxides as acceptor units were shown to be successfully modulated for high triplet energy, solution processability, and devices exhibiting low efficiency roll-off. The localized triplet state of carbazole hosts, however, poses a significant challenge for hosting blue emitters. Exploring suitable doping levels and the inevitable risk of phase separation remain as challenges to be addressed for self-hosts. The interplay of experiments (synthesis and property determination), device construction and DFT-based studies tremendously aid in deepening the understanding and potentially the prediction of TADF properties for new materials. Although, currently, computational studies explore gas and condensed phases of single molecules, the effect of the bulk material should not be underestimated. Fundamental approaches such as the optimal Hartree-Fock Method and the application of optimally tuned range-separated functionals are viable tools to significantly improve the modeling of crucial properties such as the energy gap between the singlet and triplet states and separation of the frontier orbitals involved, and extend the understanding of the impact of the nature and energy level of excited states and the rates of photophysical properties, as well as the impact of spin-orbit coupling. With the tremendous global ongoing research efforts, the materialization of these efforts seems to be inching ever closer to soon becoming tangible.

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