



Cite this: *J. Mater. Chem. C*, 2017, 5, 518

Received 28th October 2016,
Accepted 14th December 2016

DOI: 10.1039/c6tc04679g

www.rsc.org/MaterialsC

Temperature-insensitive electro-optic response of polymer-stabilized blue phases

Gaby Nordendorf,^a Jürgen Schmidtke,^a David Wilkes^b and Heinz Kitzerow^{*a}

Polymer-stabilized blue phases (PSBP) – promising candidates for developing a new generation of advanced liquid crystal displays – still suffer from the temperature dependence of the operation voltage. Here, two non-mesogenic cross-linkers were found to induce different signs of the temperature coefficient. Combining these cross-linkers results in a nearly temperature-independent operation voltage, which makes application feasible.

The development of liquid crystals (LCs) and liquid crystal displays (LCDs) over the last few decades is a remarkable chain of scientific and economic successes. The application started with small monochrome displays for pocket calculators and digital watches in the early 1970s and yielded full-colour large area flat panel displays at the turn of the century. The discovery of the twisted nematic (TN) mode^{1,2} in 1970 paved the way for today's widespread application. In TN cells, a thin layer of a nematic liquid crystal is sandwiched between two parallel glass slides, which are covered by transparent electrodes. The preferred local alignment direction of the rod-like liquid crystal molecules shows a continuous twist by 90° from one substrate to the other owing to appropriate alignment layers. Consequently, the plane of linearly polarized light is rotated by 90° and the cell placed between two crossed linear polarizers shows maximum transmission. For positive dielectric anisotropy, the molecules align along the electric field direction under the influence of an applied voltage and the display appears dark. Based on similar principles, further LC switching modes, such as the super twisted nematic (STN) mode and other modifications have been developed. In spite of limited viewing angle and contrast ratio these normally bright LCDs were widely used and lead to a 14 inch full colour display in 1988.³ Shortly after the TN mode was discovered, vertical alignment (VA) mode⁴ (utilizing a nematic liquid crystal with negative dielectric anisotropy) and in-plane switching (IPS) mode⁵ (based on fields applied parallel

to the substrate) were suggested. However, their commercial application became possible only in the 1990s,^{6,7} when liquid crystal mixtures and cells with adequate properties became available. Today, displays based on TN, VA and IPS mode are ubiquitous. However, there is a continuous demand for further improvements, which drives topical research on optimized cell designs and LC materials or new electro-optic effects.

Polymer-stabilized blue phases are candidates for a new generation of liquid crystal displays, which promise large areas, simplified manufacturing, large contrast ratio and high switching speed. Liquid crystalline blue phases (BPs) have a cubic superstructure with lattice constants typically in the sub-wavelength range. Therefore, they are optically isotropic. Placed between crossed polarizers, they appear dark in transmission without needing any alignment layer. At the same time, they show a very large field-induced birefringence owing to the parallel alignment and collective motion of neighbouring LC molecules in volumes much smaller than the cubic unit cell. For moderate electric field strength E , this field-induced birefringence Δn can be formally described as an electro-optic Kerr effect

$$\Delta n = \lambda K E^2, \quad (1)$$

where λ is the wavelength of the incident light and K is the Kerr coefficient. For larger electric fields, $E \approx E_{\text{sat}}$, the birefringence approaches a saturation value Δn_{sat} and can be described by the equation

$$\Delta n = \Delta n_{\text{sat}} \left\{ 1 - \exp \left[- \left(\frac{E}{E_{\text{sat}}} \right)^2 \right] \right\} \quad (2)$$

(extended Kerr model).⁸ The BP Kerr effect and its benefits have been recognized many years ago,⁹ but the narrow temperature range of BP appearance seemed to impede applications, until Kikuchi and coworkers¹⁰ succeeded in 2002 to prepare polymer stabilized BPs, which show an enhanced BP temperature interval of more than 60 K while maintaining the electro-optic switching capabilities. Since then, the switching performance has been

^a Faculty of Science, University of Paderborn, Warburger Str. 100, 33098 Paderborn, Germany. E-mail: Heinz.Kitzerow@upb.de

^b Merck KGaA Division Performance Materials BU Liquid Crystals – Research & Development, Frankfurter Str. 250, 64293 Darmstadt, Germany



considerably improved by developing LCs with large dielectric anisotropy and large birefringence, optimizing cell geometries, and lowering the viscosity in order to reduce the addressing voltage and the switching times, respectively. Ways of reducing the operation voltage V_{op} ,^{11–13} hysteresis^{14–17} and residual birefringence^{16,18} have been addressed in many publications. Today, sub-ms switching at voltages below 10 V is feasible.¹⁹ However, commercial application also requires a temperature-independent operation voltage in order to achieve reproducible switching under different ambient conditions. This temperature-independence remained a challenge.^{20,21} Here, we report a method to enhance the temperature-stability of the operation voltage.

The investigated samples are based on the nematic liquid crystal mixture MDA-PB-3.^{16,22} Chirality was induced by adding 3.8% (by weight) of the chiral dopant R5011.^{16,22} For stabilization, 12% reactive monomers and 0.2% of the photo-initiator IRG 651 (Ciba-Geigy) were added.^{16,22} Except for the photo-initiator, the compounds were provided by Merck KGaA, Darmstadt (Germany) and have been described in previous publications.^{16,22} In this study, the mono-functional monomer BPRM-1 and the di-functional monomers hexanediol diacrylate (HDDA) and diethylene glycol dimethacrylate (DEGDMA) have been used. The exact content of the monomers in the different samples is given in Table 1. To distinguish the samples, they are named after the cross-linkers (HDDA or/and DEGDMA) used in combination with a number, which increases with increasing cross-linker concentration. Polymer-stabilization results from in-situ photo-polymerization (3 min, $I_{UV-A} = 3 \text{ mW cm}^{-2}$) of the samples at a temperature 0.5 K above the cholesteric to blue phase transition.

To measure the effective birefringence Δn_{eff} of the resulting PSBP samples, an in-plane switching (IPS) test cell (*cf.* inset in Fig. 1) was placed between crossed polarizers so that the electrode strips were aligned at an azimuthal angle of 45° with respect to both polarizers. By reorientation of the molecules in an electric field (1 kHz square wave) a bright state is

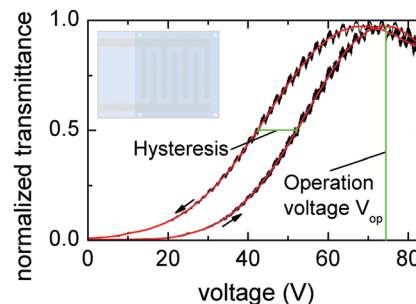


Fig. 1 A typical voltage-transmittance curve obtained with an in-plane switching cell (see inset) between crossed polarizers. The operation voltage V_{op} corresponds to the voltage at the first maximum of transmittance and the hysteresis corresponds to the difference of voltages at half height of the maximum.

induced (Fig. 1). The intensity I changes with increasing birefringence according to the relation

$$I = I_{\max} \sin^2 \left(\frac{\pi \Delta n_{eff} d}{\lambda} \right), \quad (3)$$

where I_{\max} is the maximum intensity and d is the cell thickness. The effective birefringence Δn_{eff} depends on the voltage V . The voltage, where the intensity reaches its first maximum, *i.e.* where the relation $\Delta n_{eff} d \lambda^{-1} = 0.5$ is fulfilled, is defined as the operation voltage V_{op} . The value V_{op} is related to the Kerr coefficient K in a moderate electric field E according to eqn (1). The larger the Kerr coefficient, the lower is the operation voltage.

In this work, in-plane switching cells (cell thickness d , electrode width w and electrode spacing s : $d = w = s = 10 \mu\text{m}$) were used and the Kerr coefficient at the wavelength $\lambda = 542 \text{ nm}$ (using a white light source and a colour filter) was determined assuming a homogeneous field distribution. Of course, the field distribution in IPS cells is not homogeneous so that this effective Kerr coefficient may deviate from the Kerr coefficient of the bulk liquid crystal material. However, the effective Kerr coefficient is of practical use for applications.

In general, samples containing the cross-linker HDDA [Fig. 2(a)] have a lower operation voltage than samples containing DEGDMA [Fig. 2(b)]. The operation voltage depends on the monomer mass fraction of the respective cross-linker. It increases with increasing amount of DEGDMA, while it decreases with increasing amount of HDDA. However, the absolute value of the slope increases for both kinds of samples with increasing amount of the cross-linker (Table 1). A low operation voltage corresponds to a larger Kerr coefficient and vice versa. A decreasing Kerr coefficient for HDDA [Fig. 2(c)] and an increasing Kerr coefficient for DEGDMA [Fig. 2(d)] are observed for increasing temperature. Here, the Kerr coefficient increases with increasing amount of HDDA and decreases with increasing amount of DEGDMA. Again, the absolute value of the slope increases with increasing amount of the cross-linker regardless of the cross-linker used (Table 1).

The different signs of the temperature dependence of samples containing HDDA and DEGDMA support the idea to combine these cross-linkers in order to create samples with reduced temperature dependence. To investigate whether compensation

Table 1 Monomer content (% by weight) for different samples and respective temperature dependences of the Kerr coefficient dK/dT and of the operation voltage dV_{op}/dT

Sample	BPRM-1 (%)	HDDA (%)	DEGDMA (%)	dK/dT ($\text{nm V}^{-2} \text{ K}^{-1}$)	dV_{op}/dT (V K^{-1})
■ HDDA-1	7.2	5.0	—	−0.0127	0.4145
▲ DEGDMA-1	7.2	—	4.8	0.0048	−0.2156
● HDDA + DEGDMA-1	7.2	2.4	2.4	−0.0001	−0.0057
□ HDDA-2	5.8	5.7	—	−0.0144	0.4548
△ DEGDMA-2	6.0	—	6.0	0.0076	−0.4246
⦿ HDDA + DEGDMA-2	5.8	2.9	3.0	−0.0031	0.1573
□ HDDA-3	4.8	7.2	—	−0.0172	0.4417
△ DEGDMA-3	4.8	—	7.2	0.0098	−0.9597
○ HDDA + DEGDMA-3	4.8	3.6	3.6	−0.0030	0.1119



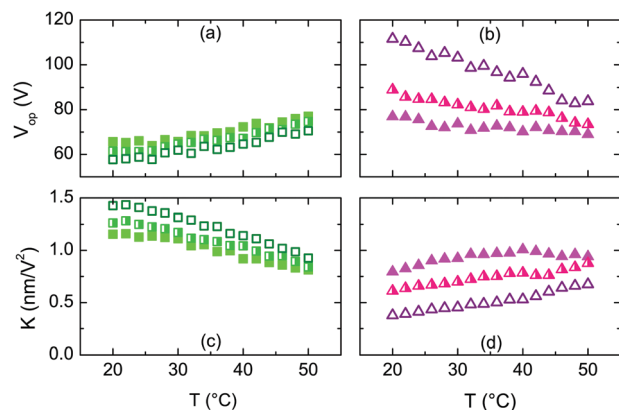


Fig. 2 Temperature dependence of (a and b) the operation voltage V_{op} and (c and d) the Kerr coefficient K for the samples (a and c) HDDA-1 (■), HDDA-2 (■) and HDDA-3 (□) and (b and d) DEGDEMA-1 (▲), DEGDEMA-2 (▲) and DEGDEMA-3 (△).

of the temperature dependences occurs, samples containing HDDA and DEGDEMA in a 1:1 ratio were prepared for three monomer mass fractions, called HDDA + DEGDEMA-1, -2 and -3, with the total monomer concentration (including the BPRM-1) kept constant at 12% (cf. Table 1). The results for the first and the last mixture (with weight ratios of mono-/difunctional monomers 3:2 and 2:3, respectively) are shown in Fig. 3, where the Kerr coefficients [Fig. 3(a) and (b)] and the operation voltages [Fig. 3(c) and (d)] of HDDA-, DEGDEMA- and HDDA + DEGDEMA-samples with identical total monomer concentrations are directly compared. The temperature dependence of the combined samples (HDDA + DEGDEMA) is significantly reduced compared to the HDDA and DEGDEMA samples and the absolute values of the investigated parameters are between those of the samples with only one cross-linker. Because of the inverse trends of Kerr coefficient and operation voltage in combination with increasing temperature dependence for an increasing amount of HDDA and DEGDEMA, the absolute values of Kerr coefficient and the operation voltage for all mixed samples are similar, $K \approx 1 \text{ nm V}^{-2}$ and $V_{op} \approx 70 \text{ V}$, respectively.

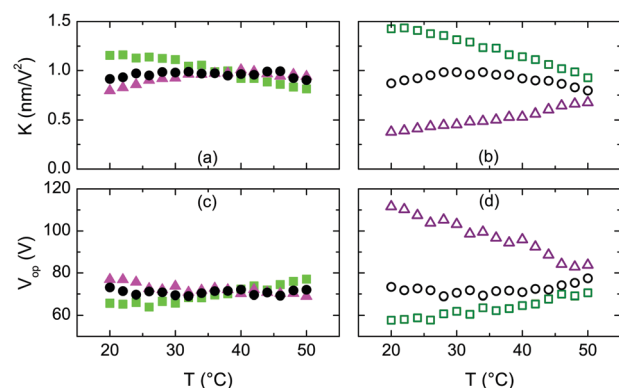


Fig. 3 Temperature dependence of (a and b) the Kerr coefficient K and (c and d) the operation voltage V_{op} for (a and c) HDDA-1 (■), DEGDEMA-1 (▲), HDDA + DEGDEMA-1 (●) and (b and d) HDDA-3 (□), DEGDEMA-3 (△), HDDA + DEGDEMA-3 (○).

Surprisingly, for all investigated temperatures the operating voltage of the combined samples (HDDA + DEGDEMA-3) is lower than the interpolated value (derived from the results for the pure HDDA- and DEGDEMA-mixtures). The temperature dependence of the investigated effects is most strongly reduced in sample HDDA + DEGDEMA-1, where the operating voltage is, within the experimental error, virtually constant over a temperature range of at least 30 K [cf. Fig. 3(c)].

Conclusions

In summary, combination of two non-mesogenic cross-linkers, which yield different temperature coefficients of the Kerr coefficient, is found to result in significantly improved temperature-stability of the operation voltage. The Kerr coefficient K and its temperature dependence are known to depend on the birefringence Δn , the dielectric anisotropy $\Delta\epsilon$ and the effective elastic coefficient k of the nematic liquid crystal component, and the induced helix pitch p ,^{23–25} $K \propto \Delta n \Delta\epsilon k^{-1} p^2$. In PSBP, also the polymer components influence these properties.^{10,22,26} The product $\Delta n \Delta\epsilon k^{-1}$ is expected to decrease with increasing temperature. Thus, the unusual positive temperature coefficient dK/dT in PSBP systems containing DEGDEMA can most likely be attributed to an increase of the pitch p with increasing temperature in these systems. This interpretation may help to identify further monomers, which can be used for the temperature compensation method presented here.

The temperature insensitivity is very important, because displays are typically controlled with temperature-independent addressing voltages. The independence on temperature will not only help to achieve reproducible grey scale, but also help to reduce a disturbing hysteresis: high contrast requires utilizing the full range of addressing voltages between $V = 0$ and $V = V_{op}$. However, applying voltages of $V > V_{op}$ is known to cause hysteresis. Both the largest possible contrast and avoidance of hysteresis can be achieved owing to the reduced temperature dependence of V_{op} .

Acknowledgements

Financial support of this study by Merck KGaA (Darmstadt) is gratefully acknowledged.

References

- 1 M. Schadt and W. Helfrich, *Appl. Phys. Lett.*, 1971, **18**, 127.
- 2 J. L. Fergason, T. R. Taylor and T. B. Harsch, *Electro-Technology*, 1970, **85**, 41.
- 3 T. Nagayasu, T. Oketani, T. Hirobe, H. Kato, S. Mizushima, H. Take, K. Yano, M. Hijikigawa and I. Washizuka, A 14-in-diagonal full color a-Si TFT LCD, in *Proc. Int. Display Research Conf.*, San Diego, CA, 1988, pp. 56–58.
- 4 F. J. Kahn, *Appl. Phys. Lett.*, 1972, **20**, 199.
- 5 R. A. Soref, *Appl. Phys. Lett.*, 1973, **22**, 165.
- 6 G. D. Baur, W. Fehrenbach, R. D. Kiefer, S. B. Weber and F. Windscheid, *Pat.* DE4000451 B4 2004.
- 7 D. Pauluth and K. Tarumi, *J. Mater. Chem.*, 2004, **14**, 1219.



- 8 J. Yan, H.-C. Cheng, S. Gauza, Y. Li, M. Jiao, L. Rao and S.-T. Wu, *Appl. Phys. Lett.*, 2010, **96**, 071105.
- 9 G. Heppke, H.-S. Kitzerow and M. Krumrey, *Mol. Cryst. Liq. Cryst. Lett.*, 1985, 2(1–2), 59–65.
- 10 H. Kikuchi, M. Yokota, Y. Hisakado, H. Yang and T. Kajiyama, *Nat. Mater.*, 2002, **1**, 64.
- 11 J. Yan, Z. Luo, S.-T. Wu, J.-W. Shiu, Y.-C. Lai, K.-L. Cheng, S.-H. Liu, P.-J. Hsieh and Y.-C. Tsai, *Appl. Phys. Lett.*, 2013, **102**, 011113.
- 12 L. Rao, Z. Ge, S.-T. Wu and S. H. Lee, *Appl. Phys. Lett.*, 2009, **95**, 231101.
- 13 Y. Chen, Y. Sun and G. Yang, *Liq. Cryst.*, 2011, **38**, 555.
- 14 L. Rao, J. Yan, S.-T. Wu, Y.-C. Lai, Y.-H. Chiu, H.-Y. Chen, C.-C. Liang, C.-M. Wu, P.-J. Hsieh, S.-H. Liu and K.-L. Cheng, *J. Disp. Technol.*, 2011, **7**, 627.
- 15 P. Nayek, H. Jeong, S.-W. Kang, S. H. Lee, H.-S. Park, H. J. Lee, H. S. Kim and G.-D. Lee, *J. Soc. Inf. Disp.*, 2012, **20**, 318.
- 16 G. Nordendorf, A. Lorenz, A. Hoischen, J. Schmidtke, H. Kitzerow, D. Wilkes and M. Wittek, *J. Appl. Phys.*, 2013, **114**, 173104.
- 17 Y. Liu, S. Xu, D. Xu, J. Yan, Y. Gao and S.-T. Wu, *Liq. Cryst.*, 2014, **41**, 1339.
- 18 Y. Chen, D. Xu, S.-T. Wu, S. Yamamoto and Y. Haseba, *Appl. Phys. Lett.*, 2013, **102**, 141116.
- 19 C.-Y. Fan, H.-C. Jau, T.-H. Lin, F. C. Yu, T.-H. Huang, C. Liu and N. Sugiura, *J. Disp. Technol.*, 2011, **7**, 615.
- 20 L. Rao, J. Yan, S.-T. Wu, S.-I. Yamamoto and Y. Haseba, *Appl. Phys. Lett.*, 2011, **98**, 081109.
- 21 F. Peng, Y. Chen, J. Yuan, H. Chen, S.-T. Wu and Y. Haseba, *J. Mater. Chem. C*, 2014, **2**, 3597.
- 22 G. Nordendorf, A. Hoischen, J. Schmidtke, D. Wilkes and H.-S. Kitzerow, *Polym. Adv. Technol.*, 2014, **25**, 1195.
- 23 P. R. Gerber, *Mol. Cryst. Liq. Cryst.*, 1985, **116**, 197.
- 24 L. Rao, J. Yan and S.-T. Wu, *J. Soc. Inf. Disp.*, 2010, **18**(11), 954.
- 25 H. Choi, H. Higuchi and H. Kikuchi, *Soft Matter*, 2011, **7**(9), 4252.
- 26 J. Yan and S.-T. Wu, *J. Disp. Technol.*, 2011, **7**(9), 490.

