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Self-regulating reversible photocatalytic-driven chromism of a cavity enhanced optical field TiO₂/CuO nanocomposite†

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Over the years, the exploration of inorganic chromogenic materials commonly interfaced with expensive noble metals has been limited, while the study of organic photochromic materials has proliferated. The key challenge lies in the optimization of inorganic materials' constituents and structural design to achieve enhanced light-matter interaction chromism with performance commensurate with that of their organic counterparts. Here, we demonstrate a tailored inorganic transition metal cavity that boosts optically controlled reversible and repeatable chromism driven by a photocatalytic reaction. The solution-processable TiO₂/CuO nanocomposite is endowed with a mesoporous cavity that is highly adept at performing self-regulating reversible photochromism under solar irradiation. The improved photoreactive chromism stems from the tailored critical structural parameters of the highly accessible mesoporous shell, reduced charge carrier diffusion length thin shell, and cavity enhanced optical field for photon-matter interactions. Consequently, the nanocomposite exhibits dual-functional photoregulated effects, i.e. photochromism mediated light transmittance modulation and rewritable printing/patterning. The nanocomposite offers high sensitivity, resulting in a short response time due to the efficient charge transfer, and such chromism effects are stable in the ambient environment. Furthermore, controlled switching of the chromism effects may be obtained simply by low-temperature heating. The concerted combination of inexpensive materials/production, low toxicity and a highly transparent noble metal-free inorganic nanocomposite renders chromogenic properties that will trigger a renewed interest in smart light-stimulus integrated technology.

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Introduction

Materials that self-regulate their color in response to electromagnetic radiation stimuli are important in emerging smart dynamic technologies. The sensitivity of the materials to light radiation makes them useful for a wide variety of applications, *i.e.* photolithography, self-developing photography, dosimetry and camouflage, *etc.*¹⁻³ Materials with good photochromic reversibility have potential applications in reusable optical memories, information storage media, data displays, sensors and smart windows.¹⁻⁴ Thus far, most photochromic materials are organic-based and are well known to undergo reversible

colour switching based on photochemical and/or thermal conversion.⁵ The significant colour change under stimuli makes them promising materials for rewritable imaging/printing/ display media. This provides a vital solution for reducing the usage of traditional paper for temporary information storage/ display, thus helping to minimise wastage of both paper and ink, as well as the associated environmental issues.^{6,7} However, there has not been a substantial amount of research done on photochromic materials in this area.^{7,8} This is mainly due to a few limitations, i.e. generally slower colour change in solid media than in solution form due to the restricted molecular mobility;9 progressive degradation with prolonged exposure to light, which limits its rewritability and outdoor applications;8,10 and short colour retention from a few minutes to several hours under ambient conditions due to visible-light stimulated decolourization processes. 11-13 Moreover, the toxicity of organic dyes may give rise to potential safety and environmental issues.6,10

On the other hand, the exploration of inorganic materials' photochromism has been limited, though these materials are known to possess apparent advantages, such as superior thermal stability, strength and chemical resistance, over their

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organic counterparts.3 The challenge lies in the optimization of inorganic composite materials for light manipulation and efficient charge interactions at the molecular level.3 Among the inorganic materials, TiO2-based photochromogenic materials are widely interfaced with a noble metal, ¹⁴ predominantly a Ag compound, to capture the photoexcited electrons from TiO2 and reduce silver ion to the metallic state based on a photocatalytic reduction reaction. 15-22 The heterogeneous TiO2-metal is widely pursued in photocatalytic reactions to counteract the inherent shortcomings of rapid charge recombination in TiO2. 23-25 Thus, the tuning of photochromic properties to achieve desirable performance in a rewritable imaging/printing medium can be done by controlling the behaviour of these charge carriers through photocatalytic reactions. Moreover, the other exceptional properties of TiO2, such as its biological and chemical inertness, availability, environmental friendliness, low cost, and long-term stability against photo- and chemical corrosion, make it one of the best candidates for realizing light-mediated photochromism. Although there are various ways to fabricate heterogeneous TiO2 materials, due to its facile and economical route and advantages such as the dispersivity of nanocomposites and the ease of coating on non-conformal substrates, a solution-based approach is often used.

Here we developed the first ever solution processable TiO₂/ CuO optical cavity that ingeniously manipulates light-matter interactions for enhanced chromogenic capabilities for rewritable imaging/printing/display purposes. It is noteworthy that this photocatalytic chromism phenomenon is highly dependent not only on the chemical constituents, but also on the structural design. Hence, the crucial aspects, such as light-matter interactions, accessibility and contact interfaces, must be taken into consideration in the construction of light stimulated phenomena. Accordingly, a hollow porous shell nanocomposite is designed to render high specific surface areas with a large number of active sites, a thin shell with reduced charge carrier diffusion length and improved accessibility through its porous framework. The nanocomposite not only possesses the aforementioned attributes, but its thin porous shell and cavity also favourably allow incident light to penetrate and display extensive local field enhanced areas.26 In regard to chemical constituents, CuO nanoparticles are specifically chosen as the counterpart of TiO2, owing to their fine-dispersivity, antimicrobial activity and cost effectiveness compared to other noble metals, and efficient photon capture and charge separation properties.27,28 The specific tailoring of TiO2/CuO in the nanoscale composition-structural aspects, leading to the above favourable properties, is expected to aid in enhancing photocatalytic activity, thereby boosting the light-mediated chromism performance.

Finally, a proof-of-concept of all-light induced reactions of the nanocomposite towards both applications in transmittance modulation and high-resolution ink-free rewritable patterning/printing is validated. The nanocomposite possesses relatively high sensitivity, which results in a reasonable/short response time and high-resolution printing for fast and convenient visualization. The tunable intensity enables the construction of 3D images that exhibit persistency in image retention in the

ambient environment, and warrants enough time for reading or temporary information storage. It also displays sufficiently fast switching under low-temperature heating for simple erasing, and good stability with multiple cycles for prolonged and repeated usage. The combination of a solution-processable, highly transparent noble metal-free inorganic nanocomposite with self-regulating light modulated properties can potentially meet society's increasing demand for integrated smart and energy-saving sustainable technology.

Results and discussion

Fig. 1a shows the process flow for synthesizing TiO_2/CuO hollow nanocomposites. In the process, two different TiO_2 nanostructures, namely $SiO_2@TiO_2$ core-shell and hollow TiO_2 structures, were synthesized by a sol-gel coating of a TiO_2 shell on a SiO_2 template, followed by annealing to obtain crystalline TiO_2 shells. Subsequently, the SiO_2 template was removed completely to obtain the hollow TiO_2 structures. The presynthesized CuO nanoparticles were loaded onto hollow TiO_2 spheres with high uniformity and dispersivity. The finely dispersed CuO/TiO_2 system is highly favoured in photocatalysis, such as in photocatalytic hydrogen generation, degradation, reduction of CO_2 or oxidation of volatile organic compounds,

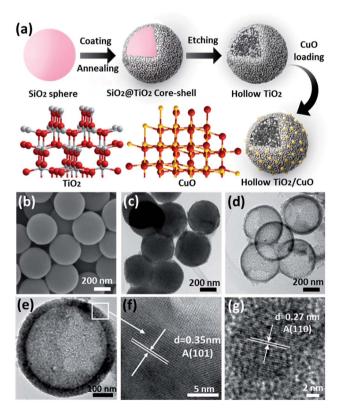


Fig. 1 (a) Schematic synthesis process for $SiO_2@TiO_2$ core—shell, hollow TiO_2 and hollow TiO_2/CuO structures. (b) SEM image of SiO_2 nanospheres. TEM images of (c) $SiO_2@TiO_2$ core—shell and (d) hollow TiO_2 nanospheres. (e) TEM image of the hollow TiO_2/CuO nanocomposite. High-resolution TEM images of (f) hollow TiO_2 and (g) TiO_2 0 nanoparticles in the TiO_2/CuO 0 nanocomposite.

due to the efficient interparticle charge transfer process between the bonding of CuO and TiO2, for enhanced charge separation of photogenerated electron-hole pairs.29-32

The morphologies of the as-prepared samples were characterized by SEM (Fig. 1b and S1†) and TEM (Fig. 1c-e). Fig. 1b shows the spherical SiO₂ template of a uniform size of 400 nm. Fig. 1c shows the SiO₂@TiO₂ core-shell nanosphere with a continuous TiO₂ coating of \sim 50 nm. Fig. S2† shows a bright field image of the SiO2@TiO2 core-shell nanosphere and its respective elemental mappings. It can be seen that the O signal is found throughout the nanosphere, while the Si signal is located in the center portion with the Ti signal surrounding it. The hollow TiO₂ structure was prepared by etching the SiO₂@-TiO₂ core-shell nanosphere in weak NaOH solution (0.1 M) for 36 h to completely remove the SiO₂ template (Fig. 1d). The XRD spectra of the SiO2@TiO2 core-shell and hollow TiO2 nanospheres exhibit highly crystalline TiO2 anatase diffraction peaks, while no peak is observed for the non-crystalline SiO₂ nanospheres (Fig. S3†). The TEM image (Fig. 1e) shows the TiO2/CuO hollow nanocomposite with a hollow cavity and a porous shell consisting of TiO2 nanoparticles. CuO nanoparticles were successfully loaded onto hollow TiO2 and evidently detected by EDX (Fig. S4a†). Moreover, CuO nanoparticles were finely dispersed on the TiO2 shell with an average size of ~2 nm (Fig. S4b†). High-resolution TEM was then carried out at the marked locations. The selected area show a lattice fringe spacing of 0.35 nm, attributed to the (101) plane of anatase TiO₂ nanoparticles (Fig. 1f) and the (110) plane of CuO nanoparticles (Fig. 1g). These many small size heterojunctions formed between TiO2 and CuO play a crucial role in the efficient photocatalytic reduction of CuO.33-35

As the photochromic phenomenon is photocatalytically initiated by TiO₂ nanocrystals generating electron-hole pairs under UV irradiation, the photogenerated electrons from TiO2 are transferred to CuO for reduction to Cu, while the holes are scavenged by the sacrificial reagent. In order to achieve better photochromic performance in terms of response time, contrast and reversibility, a few factors need to be considered, including the light-harvesting capability for electron-hole pair generation, the optimal interface between TiO₂ and CuO for interparticle charge transfer and accessibility of the reactant for efficiently removing holes. These factors associated with photo-stimulated reactions can be examined through photocatalysis, using the hydrogen generation rate as an indicator. The two different CuO-loaded TiO2 nanostructures, namely the SiO2@TiO2/CuO core-shell and hollow TiO2/CuO structures, were evaluated for photocatalytic properties and compared against P25 TiO2/CuO nanoparticles. The same loading method has been carried out for all three samples with various loading amounts of CuO ranging from 3 wt% to 10 wt%. Fig. S5† shows that both the SiO₂@TiO₂ core-shell nanocomposites and hollow TiO₂ nanospheres consistently outperformed P25 nanoparticles, even when CuO was absent, with \sim 4 and \sim 10 times enhancement of the photocatalytic H2 production rate, respectively. The photocatalytic performance of these materials is largely associated with structural features, such as surface area, cavity size, pore diameter and architecture, which can directly affect the reactant

diffusion and light absorption capabilities that play important roles in photocatalysis. On the basis of heterogeneous catalytic reactivity being kinetically controlled by ion adsorption-diffusion in the pores of catalysts, the surface areas and pore sizes of these materials were investigated using BET. The N2 adsorption-desorption isotherm and the corresponding Barrett-Joyner-Halenda (BJH) pore size distributions (insets) are shown in Fig. S6.† Both the as-prepared samples have type IV isotherms with hysteresis loops that suggest the presence of mesopores of pore distribution \sim 4-8 nm (Fig. S6†). This is consistent with TEM analysis, whereby the TiO₂ shells are observed to be made up of aggregated highly dispersed TiO2 nanoparticles, with the interparticle voids forming the mesopores. Noticeably, the TiO₂ hollow structure shows distinctively larger hysteresis loops, implying a higher absorption volume and hence larger BET surface area. This observation agrees with the fact that the hollow TiO₂ structure has the largest surface area (72 m² g⁻¹), followed by the P25 nanoparticles (43 m² g⁻¹), while the coreshell structure has the lowest surface area of 30 m² g⁻¹. It is apparent that the porous hollow TiO₂ structure has the highest surface area and an ideal mesoporous shell that renders numerous active sites, a highly accessible mass diffusion pathway, short charge migration distance and high viability to achieve more dispersed CuO loading that facilitates photoreactivity. 21,22,26 When various amounts of CuO nanoparticles were loaded, the H₂ production rate increased with the amount of CuO nanoparticles up to an optimal amount, where significant H₂ production rates of 3475 μmol g⁻¹ h⁻¹, 1148 μmol g⁻¹ h^{-1} and 930 μ mol g^{-1} h^{-1} were observed for the SiO_2 @ TiO_2 core-shell nanocomposites, hollow TiO2 nanospheres and P25 nanoparticles, with CuO loading amounts of 5 wt%, 5 wt% and 3 wt%, respectively (Fig. S5†). The remarkable enhancement is ascribed to the evenly-dispersed CuO co-catalyst nanoparticles well-interfaced with the TiO2 photocatalyst for efficient charge separation, which reduces the recombination rate.27 However, higher loading of CuO nanoparticles is seen to decrease the rate of H₂ generation, possibly due to excessive loading, which leads to particle agglomeration, and consequently shielding of the photosensitive TiO2 surfaces. It is also suggested that overloading of the CuO co-catalyst may absorb photons, hence reducing the photoexcitation capacity of TiO2.37 Not only does the TiO₂/CuO hollow nanocomposite exhibit a remarkable photocatalytic response, it also shows superior stability under UV light. The cycling of photocatalysis was conducted for the 5 wt% CuO-loaded TiO2 nanocomposite (Fig. S7†), which exhibits a stable hydrogen generation rate of 3387 µmol g⁻¹ h⁻¹ with a variance of 2%. This ensures prolonged usability of the sample even under strong outdoor light.

As the enhancement in photocatalytic hydrogen production is disproportional to the increase of surface area, this suggests that surface area is not the only factor that affects the photocatalytic hydrogen production rate. The optical properties of these two structures, namely the SiO₂@TiO₂ core-shell and hollow TiO2 structures, were studied. The SiO2-based core-shell structure has been demonstrated as a successful candidate to improve various optical properties, such as imaging, focusing, and optical nonlinearity.38,39 However, for various applications

that require high light-matter interactions, it is possible to design novel structures with better performances than that of the SiO₂-based core-shell structure. The key mechanism for the SiO₂-based core-shell structure is the focusing effect induced by the refraction of the SiO₂ particle lens. In this paper, the proposed TiO₂ hollow structure has a similar focusing effect and better light-matter interactions. This hollow shell structure with the high refractive index TiO2 material induces light refraction, leading to the focusing effect (Fig. 2a). To better understand the optical performance of the hollow TiO2 structure, we conducted Finite-Difference Time-Domain (FDTD) simulations with Lumerical software (the details for the simulation parameters can be found in the experimental section). As shown in Fig. 2b and c, both the SiO2@TiO2 core-shell and hollow TiO₂ structures induce a focusing effect due to refraction by the spherical shape. However, the focusing effects induced by these two systems are not identical. For the SiO₂@TiO₂ coreshell structure, the silica core serves as a particle lens. This particle lens focuses the light into a relatively small volume to create a high local field intensity. For the hollow TiO₂ structure, due to the lack of particle lens, the light is not restricted to a small area. Instead, the local field enhanced area extends across a large region. Internally, it is found in the simulation that the hollow TiO2 system has a lower transmittance than the SiO₂@TiO₂ core-shell system (simulation method in ESI†). From the distribution of the light energy (Fig. 2b), it is obvious that the SiO₂@TiO₂ core-shell system focuses the light better.

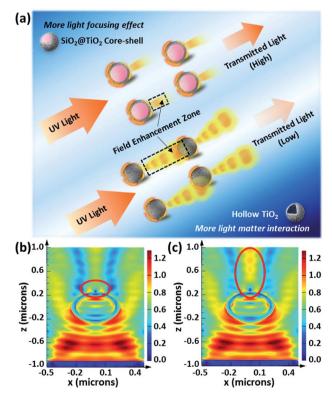


Fig. 2 (a) Schematic to demonstrate the differences between the $SiO_2@TiO_2$ core—shell and hollow TiO_2 nanostructures. The light Efield distribution of the (b) $SiO_2@TiO_2$ core—shell and (c) hollow TiO_2 nanostructures.

In other words, most of the light transmits through the hybrid system in the central region, as shown in Fig. 2b. In the hollow ${\rm TiO_2}$ system, the lack of focusing leads to better light–matter interactions on the ${\rm TiO_2}$ shell. More local field enhanced areas can be found near the ${\rm TiO_2}$ shell (red circles in Fig. 2b and c). As ${\rm TiO_2}$ is the material that absorbs the light, less light can transmit through the hollow ${\rm TiO_2}$ system compared to the ${\rm SiO_2@TiO_2}$ core–shell system. Essentially, these external and internal effects make the hollow ${\rm TiO_2}$ structure a more favorable platform for many photo-chemical reactions, as confirmed by the following experiments.

From the UV-vis spectra presented in Fig. 3a, TiO₂/CuO hollow nanospheres show distinct light absorption enhancement in the UV region compared to SiO2@TiO2/CuO core-shell nanocomposites and P25/CuO nanoparticles. This is most likely caused by the stronger light-matter interactions of the hollow TiO₂ nanostructure, based on the simulation results that have been discussed earlier. The light absorption of this light focusing scheme is comparatively lower than that of the hollow structural configurations, as seen from the simulation results (additional simulation results, Sections 1 & 2 in ESI†). Moreover, the presence of the SiO₂ core directly in contact with the TiO₂ shell greatly reduces the exposed surface area, resulting in the lowest accessible active surface sites. The P25/CuO nanoparticles have the lowest absorption, due to light passing through the particles directly without any light focusing effect.²⁶ Hence, based on the aforementioned reasons, the hollow TiO₂/ CuO nanospheres outperform both the SiO₂@TiO₂/CuO coreshell structure and P25/CuO nanoparticles for both light absorption capability and reactant diffusion properties, which can be translated into better photochromic performance.

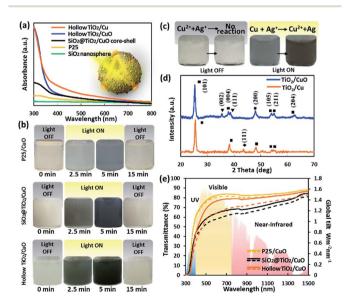


Fig. 3 (a) Absorption spectra of SiO₂, P25/CuO nanoparticle, SiO₂@-TiO₂/CuO core—shell, hollow TiO₂/CuO and hollow TiO₂/Cu structures. (b) Digital images of the photochromic effect of P25/CuO nanoparticles, SiO₂@TiO₂/CuO core—shell and TiO₂/CuO hollow nanocomposites in solution. (c) Visual validation of Cu formation with AgNO₃ addition. (d) XRD spectra before and after light irradiation. (e) Transmittance spectra of various nanocomposites coated on glass.

The samples with the optimal CuO loading amount were tested for their sensitivity to UV light (365 nm) to measure the rate of the photochromic response. After a few minutes' exposure, the white coloured solution turned light grey, and subsequent illumination resulted in a darker grey solution. Among the three samples, the hollow TiO2/CuO and SiO2@TiO2/CuO core-shell structures had significantly fast response times with only 2.5 min irradiation, while the colour change of the P25/ CuO nanoparticles was not obvious with the same irradiation time, as shown in Fig. 3b. The reversal of the colour back to its initial state can be observed 10 min after turning off the light. The photochromic reversibility is derived from two processes, i.e. the photo-reduction of CuO to Cu metal, 14,40 which turns the colour of solution dark grey due to the presence of the Cu metal, and the oxidation of Cu metal back to CuO in the presence of oxygen.41 The presence of Cu after irradiation can be confirmed by the redox reaction with the addition of AgNO₃ (Fig. 3c). AgNO₃ acts as an oxidising agent, while Cu acts as the reducing agent. Hence, the Ag⁺ ions are reduced to Ag particles (eqn (1)), while Cu is oxidised to Cu²⁺ (eqn (2)). 42 The end product is the formation of Cu²⁺ ions and Ag metal precipitate (eqn (3)).

$$2Ag(s) \rightarrow 2Ag^{+}(aq) + 2e^{-}[oxidation]$$
 (1)

$$Cu^{2+} + 2e^{-} \rightarrow Cu$$
 (s) [reduction] (2)

$$Cu(s) + 2Ag^{+}(aq) \rightarrow Cu^{2+}(aq) + 2Ag(s)$$
 (3)

0.1 M AgNO₃ solution was added separately to TiO₂/CuO hollow nanocomposite solutions with and without UV light exposure. No colour change is observed in the absence of UV, since CuO is not photoreduced to Cu metal and does not react with AgNO₃ (Fig. 3c). In contrast, a colour change from white to grey is observed with light irradiation, and upon addition of AgNO₃ solution, a further colour change from grey to brownishgrey is noted (Fig. 3c). This indicates the precipitation of Ag particles and hence confirms the formation of Cu metal with UV light exposure. XRD was carried out to verify the reduction of CuO to Cu metal. Both samples exhibit a strong anatase TiO2 peak before and after exposure to UV light (Fig. 3d), which affirms that TiO₂ remains unchanged and the observed colour change can be solely attributed to the transformation of CuO. Two additional peaks at 35.4° and 38.8° are observed for the TiO2/CuO sample without UV light exposure, which correspond to the (002) and (111) planes of CuO, respectively. Conversely, a peak at 43.5°, corresponding to the (111) plane of Cu metal, is observed for the sample that was subjected to UV exposure. This evidently reveals the complete transformation of CuO to Cu metal under UV light irradiation. The light absorption properties of TiO2/CuO and TiO2/ Cu were investigated by UV-vis spectroscopy. From the UV-vis spectra presented in Fig. 3a, the TiO2/Cu hollow nanocomposite shows light absorption enhancement compared to the TiO2/CuO hollow nanocomposite. The excited electrons can migrate to the valence band of the CuO nanoparticles, thus enhancing the absorption centered at \sim 450 nm. The electrons are also able to reduce the Cu²⁺ ions, thus creating the absorption tail in the 500-600 nm range due to the plasmonic absorption of the Cu metal. 14,43

With the success of the solution-based photochromism demonstration, attempts were made to formulate a coating of TiO2/CuO hollow nanocomposites on various rigid/flexible substrates. The photochromic properties of the SiO₂@TiO₂/ CuO core-shell, hollow TiO₂/CuO and P25/CuO nanoparticle structures loaded with an optimal amount of CuO nanoparticles coated on a glass substrate were studied in the UV-vis-NIR range by measuring their respective transmittance before (solid line) and after (dotted line) UV light illumination (Fig. 3e). The transmitted efficiency (TE) and shielding efficiency (SE) of different wavelengths (UV: 300-400 nm, vis: 400-750 nm and NIR: 750-1500 nm) can be derived from the transmittance spectra and standard solar spectrum (ESI Table 2†).44 Comparing the samples of CuO-loaded P25, SiO2@TiO2 coreshell and hollow TiO2 after UV illumination, the SE in UV, visible and NIR light region increases from 63.5, 22.8, 18.8% (P25/CuO) and 76.8, 44.7, 32.9% (SiO₂@TiO₂/CuO core-shell) to 83.3, 40.2, 26.3% (hollow TiO₂/CuO), respectively. This evidently shows that the hollow TiO2 structure has the highest solar shielding across the UV-vis-NIR range. This shielding effect, along with sufficient visible light transmittance, can be of use in variable optical transmission materials, such as photochromic camera filters or photochromic lenses,45 and also in energyefficient building applications, such as smart windows or glass facades.46 Hollow TiO2/CuO nanocomposites with a fast photochromic response time, high solar shielding and excellent stability under UV light were demonstrated for chromogenic erasable patterning/printing on both rigid and flexible substrates with tunable intensity, excellent legibility and high resolution. A schematic diagram of the patterning setup is shown in Fig. 4a. Photomasks with different printing patterns were used with a substrate coated with a thin layer of TiO2/CuO hollow nanocomposites. Fig. S10† shows the nanocomposite coating of \sim 1 µm thickness on glass that exhibits a reasonably high optical clarity. The hollow TiO₂/CuO-coated glass with its respective SEM cross-section and top views are shown in Fig. S10† as well. The uniform coating on light-weight and flexible PET (Fig. 4b) offers a contour shape useful for conformable and portable displays and photodetectors/sensors, etc.45 The light illumination intensity can be tuned for fine tailoring of chromogenic contrasts, as shown by the printed colors with transmittance measurements (Fig. 4c). Meanwhile, selective control of chromogenic contrast without direct contact with the surface of the coating can be demonstrated, making it remotely controllable. Accordingly, various projected pattern masks were used to realize selective photochromogenic transparent displays (Fig. 4d). Furthermore, the coating is extended to a rewritable display capable of projecting 3D patterns to establish the capacity of the dynamic display. The write-erase repetitions as well as assorted switchable contrasts of different faces of the same 3D box pattern are validated (Fig. 4e). The reversibility of the chromogenic coating makes it an ideal erasable-writable medium. Finally, the high-resolution pattern is demonstrated on a rewritable display (Fig. 4f and g). Notably, the pattern of \sim 100–200 µm can be resolved and conspicuously observed under high magnification. Micro- to nano-scale patterning of Merlion, an iconic landmark of Singapore, was

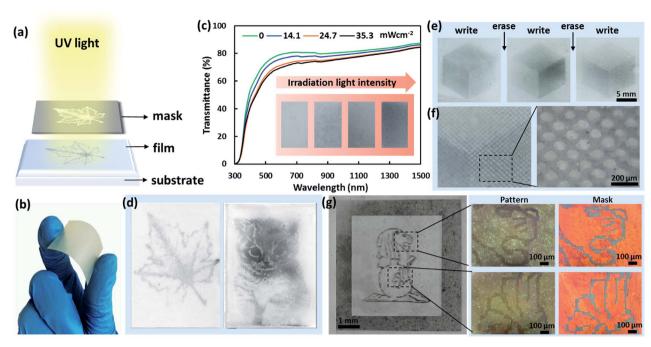


Fig. 4 (a) Schematic representation of the multi-functional photoregulated effects of the hollow TiO_2/CuO nanocomposite. (b) Digital image of the TiO_2/CuO coating on flexible PET. (c) Transmittance spectra and digital photos of hollow TiO_2/CuO on the glass substrate with different light irradiation intensities. Demonstrations of (d) a variety of chromogenic patterns, (e) write-erase repetitions with assorted switchable contrasts and (f and g) high-resolution patterning.

also demonstrated, with the highest resolution of ${\sim}20~\mu m$ observed under the microscope as shown in Fig. 4g and S11,† with both the printed images and the mask used. The important attributes of a photochromic display are that the pattern and information are exhibited by reflecting the surrounding light (of a non-emissive nature), that it is easy on the eye, ink-free, rewritable and can be adopted even in bright outdoor space. In this regard, the outdoor photochromic projection can make use of natural sunlight for dynamic displays that change in accordance with the projected pattern/information. Moreover, such a display that effectively responds to a light stimulus, unlike other active solar devices, is easy and inexpensive to implement and maintain since it does not involve any mechanical and electrical components.

The sensitivity, reversibility and repeatability of the TiO₂/ CuO hollow nanocomposite coating were studied. Firstly, the sensitivity of the as-prepared materials was tested using glass as the substrate. Light illumination duration and intensity can be varied and controlled to have multiple chromogenic contrasts alongside selective patterning on the substrate. The transmittance measurements with different light intensities were demonstrated earlier, in Fig. 4c. Here, the duration of illumination under UV light was examined and the results are shown in Fig. S12.† The response time was \sim 1 min in order to obtain a legible print with a font size of 10, and prolonged illumination enhanced both the intensity and the lifetime of the print. In addition, the writing process under common solar irradiation was also demonstrated with a longer printing time of 4.5 min to obtain similar contrast to that obtained with 1 min UV irradiation (Fig. S13†). The information printed under 2 min UV illumination can be retained for 20 days under ambient

conditions and standard office brightness (500 lx), as shown in Fig. S14,† which allows its practical use for fabricating rewritable media with long-lasting legibility. The time taken to revert the colour change is far longer than what has been previously reported.17,22 Secondly, the photoreversible chromogenic function is validated. The color is very slow to revert under ambient conditions, however it can be sufficiently fast when lowtemperature heating is applied. The printed information can be erased within 10 min, 5 min and 4 min under different heating conditions of 85, 95 and 105 °C, respectively, with less than 1% variance in transmittance compared with an initial stage at 600 nm (Fig. S15†). The transmittance at 600 nm with writing and erasing (at 95 °C) time is presented in Fig. 5a, and the respective images are shown in Fig. 5b and c. Thirdly, the stability and repeatability of the material were examined. The transmittance of the TiO2/CuO hollow nanocomposites was measured for ten repetitive reduction-oxidation cycles, as shown in Fig. 5d. The photochromic effect is highly reversible as the transmittance reverts to its pristine state across the UV-vis-NIR range for all the cycles. The specific transmittances at wavelengths of 600 and 1000 nm were measured in relation to the transparency and photothermal characteristics (Fig. 5e). The average initial transmittances at 600 and 1000 nm were measured to be \sim 78 and \sim 80%, respectively. After irradiation with UV light, the average transmittances dropped to ~71 and \sim 75%, which indicate a \sim 9 and \sim 6% drop in the visible and NIR wavelengths, respectively. The demonstration of repeated cycles establishes the reversibility of the photochromic effect and the stability of the hollow TiO₂/CuO nanocomposite. It was shown that the hollow TiO2 nanocomposite can exhibit selfregulating reversible photochromism for at least 25 cycles

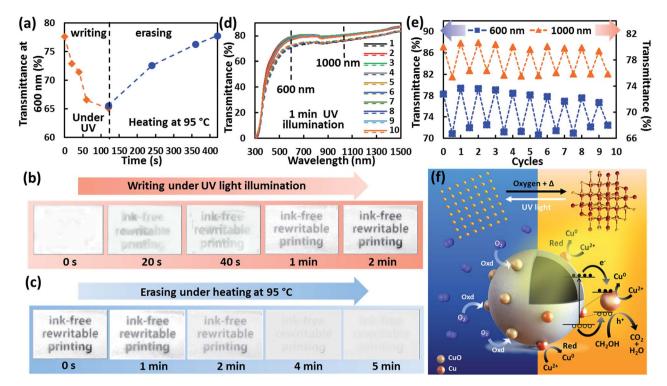


Fig. 5 (a) Transmittance at 600 nm of the hollow TiO₂/CuO coating on glass with writing and erasing processes. Digital photos of the hollow TiO₂/CuO coating on glass with (b) writing and (c) erasing processes over time. (d) Transmittance modulation cycles of oxidation (solid lines) and reduction (dotted lines). (e) The switchable photochromogenic transmittance at 600 nm and 1000 nm. (f) Mechanism of reversible photochromic coupled photocatalysis phenomena.

(Fig. S16†). Fig. 5f shows the proposed mechanism for the light stimulus-responsive reaction of the hollow TiO2/CuO nanocomposite. When light illuminates on the hollow TiO2/CuO nanocomposite, electron-hole pairs are generated within TiO2. Since CuO is in direct contact with the TiO₂ shell, and the energy potential of CuO lies slightly below that of TiO2, migration of the photogenerated electrons from the conduction band of TiO₂ to CuO can effectively take place, while the holes are scavenged by the sacrificial reagent (which is menthol in this case) to prevent electron-hole pair recombination.47 This results in efficient charge transfer of photogenerated electrons from TiO2 to CuO, 48,49 and meanwhile these transferred electrons react with CuO, which undergoes a reduction process to produce Cu.50 This reduction process, which is initiated by the photocatalytic reaction in TiO2, causes the colour change of the substrate from white (TiO₂/CuO) to dark grey (TiO₂/Cu) under light irradiation. To revert the colour to white, the substrate is left in ambient air and Cu can easily react with O2 to form CuO again, while heating the substrate will accelerate the oxidation process. Generally, nanoparticles are metastable owing to their relatively high surface area, causing them to be readily oxidized in oxygen. 41,51 This mechanism provides the viability for ink-free rewritable printing with a high resolution and long lifetime, as well as reasonably fast erasing upon the simple low-temperature heating process.

Conclusions

We have designed and demonstrated a solution-based hollow TiO₂/CuO nanocomposite which is capable of displaying both

photochromism and catalysis effects under solar irradiation. The TiO₂/CuO hollow nanocomposite outperforms the SiO₂@-TiO₂ yolk-shell and core-shell structures. The enhanced light harvesting and reactant diffusion properties of the TiO2/CuO hollow nanocomposite are attributed to the designed structural parameters of highly accessible numerous active sites in the mesoporous shell, reduced charge carrier diffusion length in the thin shell, homogeneous heterostructures for enhanced molecular interfacial interactions and optical fields for photon-matter interactions. The nanocomposite is proven to effectively utilize light irradiation for photochromism, which is manifested in variable phototransmittance and ink-free high-resolution rewritable printing/displays. Hence, such an innovative noble metal-free inorganic nanocomposite with multifunctional light modulated chromogenic coupled catalysis properties can deliver promising light-stimulus smart technology solutions.

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References

- 1 B. W. Faughnan, D. L. Staebler and Z. J. Kiss, Applied Solid State Science, New York: Academic Press, New York, 1971.
- 2 K. V. Wong and R. Chan, J. Energy Resour. Technol., 2013, 136, 012002.

- 3 T. He and J. Yao, Prog. Mater. Sci., 2006, 51, 810-879.
- 4 G. Wirnsberger, B. J. Scott, B. F. Chmelka and G. D. Stucky, *Adv. Mater.*, 2000, **12**, 1450–1454.
- 5 W. Wang, N. Xie, L. He and Y. Yin, *Nat. Commun.*, 2014, 5, 5459.
- 6 X. Du, T. Li, L. Li, Z. Zhang and T. Wu, *J. Mater. Chem. C*, 2015, 3, 3542–3546.
- 7 R. Klajn, P. J. Wesson, K. J. M. Bishop and B. A. Grzybowski, *Angew. Chem., Int. Ed.*, 2009, **48**, 7035–7039.
- 8 I. Kawashima, H. Takahashi, S. Hirano and R. Matsushima, *J. Soc. Inf. Disp.*, 2004, **12**, 81–85.
- 9 R. A. Evans, T. L. Hanley, M. A. Skidmore, T. P. Davis, G. K. Such, L. H. Yee, G. E. Ball and D. A. Lewis, *Nat. Mater.*, 2005, 4, 249–253.
- 10 L. Sheng, M. Li, S. Zhu, H. Li, G. Xi, Y.-G. Li, Y. Wang, Q. Li, S. Liang, K. Zhong and S. X.-A. Zhang, *Nat. Commun.*, 2014, 5, 3044.
- 11 B. Garai, A. Mallick and R. Banerjee, *Chem. Sci.*, 2016, 7, 2195–2200.
- 12 J. Su, T. Fukaminato, J.-P. Placial, T. Onodera, R. Suzuki, H. Oikawa, A. Brosseau, F. Brisset, R. Pansu, K. Nakatani and R. Métivier, *Angew. Chem.*, 2016, **128**, 3726–3730.
- 13 S.-J. Lim, B.-K. An, S. D. Jung, M.-A. Chung and S. Y. Park, *Angew. Chem.*, 2004, **116**, 6506–6510.
- 14 D. M. Tobaldi, N. Rozman, M. Leoni, M. P. Seabra, A. S. Škapin, R. C. Pullar and J. A. Labrincha, *J. Phys. Chem. C*, 2015, 119, 23658–23668.
- N. Crespo-Monteiro, N. Destouches, L. Bois, F. Chassagneux,
 S. Reynaud and T. Fournel, *Adv. Mater.*, 2010, 22, 3166–3170.
- 16 Q. Qiao, X. Zhang, Z. Lu, L. Wang, Y. Liu, X. Zhu and J. Li, *Appl. Phys. Lett.*, 2009, **94**, 074104.
- 17 K. Naoi, Y. Ohko and T. Tatsuma, *Chem. Commun.*, 2005, 1288–1290, DOI: 10.1039/b416139d.
- 18 E. Kazuma and T. Tatsuma, *Chem. Commun.*, 2012, **48**, 1733–1735.
- 19 K. Kawahara, K. Suzuki, Y. Ohko and T. Tatsuma, *Phys. Chem. Chem. Phys.*, 2005, 7, 3851–3855.
- 20 R. Han, X. Zhang, L. Wang, R. Dai and Y. Liu, Appl. Phys. Lett., 2011, 98, 221905.
- 21 N. Crespo-Monteiro, N. Destouches, L. Nadar, S. Reynaud, F. Vocanson and J. Y. Michalon, *Appl. Phys. Lett.*, 2011, **99**, 173106.
- 22 Y. Ohko, T. Tatsuma, T. Fujii, K. Naoi, C. Niwa, Y. Kubota and A. Fujishima, *Nat. Mater.*, 2003, 2, 29–31.
- 23 Z. W. Seh, S. Liu, M. Low, S.-Y. Zhang, Z. Liu, A. Mlayah and M.-Y. Han, *Adv. Mater.*, 2012, 24, 2310–2314.
- 24 M. Gao, C. K. N. Peh, W. L. Ong and G. W. Ho, *RSC Adv.*, 2013, 3, 13169–13177.
- 25 M. Gao, C. K. N. Peh, Y. Pan, Q.-H. Xu and G. W. Ho, *Nanoscale*, 2014, **6**, 12655–12664.
- 26 M. Gao, L. Zhu, W. L. Ong, J. Wang and G. W. Ho, *Catal. Sci. Technol.*, 2015, 5, 4703–4726.

- 27 C. K. N. Peh, M. Gao and G. W. Ho, J. Mater. Chem. A, 2015, 3, 19360–19367.
- 28 L. Zhu, M. Hong and G. W. Ho, Nano Energy, 2015, 11, 28-37.
- 29 K. Lalitha, G. Sadanandam, V. D. Kumari, M. Subrahmanyam, B. Sreedhar and N. Y. Hebalkar, J. Phys. Chem. C, 2010, 114, 22181–22189.
- 30 J. Bandara, C. P. K. Udawatta and C. S. K. Rajapakse, *Photochem. Photobiol. Sci.*, 2005, 4, 857–861.
- 31 D. Barreca, G. Carraro, V. Gombac, A. Gasparotto, C. Maccato, P. Fornasiero and E. Tondello, Adv. Funct. Mater., 2011, 21, 2611–2623.
- 32 L. Li, W. Li, A. Ji, Z. Wang, C. Zhu, L. Zhang, J. Yang and L.-F. Mao, *Phys. Chem. Chem. Phys.*, 2015, 17, 17880–17886.
- 33 I. Mondal and U. Pal, *Phys. Chem. Chem. Phys.*, 2016, **18**, 4780–4788.
- 34 G. D. Moon, J. B. Joo, I. Lee and Y. Yin, *Nanoscale*, 2014, 6, 12002–12008.
- 35 X.-J. Lv, S.-X. Zhou, C. Zhang, H.-X. Chang, Y. Chen and W.-F. Fu, *J. Mater. Chem.*, 2012, 22, 18542–18549.
- 36 S. Storck, H. Bretinger and W. F. Maier, *Appl. Catal.*, *A*, 1998, **174**, 137–146.
- 37 G. Li, N. M. Dimitrijevic, L. Chen, T. Rajh and K. A. Gray, *J. Phys. Chem. C*, 2008, **112**, 19040–19044.
- 38 M. Wu, R. Chen, J. Soh, Y. Shen, L. Jiao, J. Wu, X. Chen, R. Ji and M. Hong, *Sci. Rep.*, 2016, **6**, 31637.
- 39 Z. Du, L. Chen, T.-S. Kao, M. Wu and M. Hong, *Beilstein J. Nanotechnol.*, 2015, **6**, 1199–1204.
- 40 S. Xu, J. Ng, A. J. Du, J. Liu and D. D. Sun, *Int. J. Hydrogen Energy*, 2011, 36, 6538–6545.
- 41 S. Xu and D. D. Sun, Int. J. Hydrogen Energy, 2009, 34, 6096-6104.
- 42 R. Glicksman, H. Mouquin and C. V. King, *J. Electrochem. Soc.*, 1953, **100**, 580–585.
- 43 H. Irie, S. Miura, K. Kamiya and K. Hashimoto, *Chem. Phys. Lett.*, 2008, **457**, 202–205.
- 44 W. Zhong, N. Yu, L. Zhang, Z. Liu, Z. Wang, J. Hu and Z. Chen, *RSC Adv.*, 2016, **6**, 18881–18890.
- 45 R. Pardo, M. Zayat and D. Levy, *Chem. Soc. Rev.*, 2011, **40**, 672–687.
- 46 S. K. Deb, S.-H. Lee, C. Edwin Tracy, J. Roland Pitts, B. A. Gregg and H. M. Branz, *Electrochim. Acta*, 2001, 46, 2125–2130.
- 47 M. Shen and M. A. Henderson, J. Phys. Chem. Lett., 2011, 2, 2707–2710.
- 48 D. Praveen Kumar, M. V. Shankar, M. Mamatha Kumari, G. Sadanandam, B. Srinivas and V. Durgakumari, *Chem. Commun.*, 2013, **49**, 9443–9445.
- 49 S. S. Lee, H. Bai, Z. Liu and D. D. Sun, *Appl. Catal.*, *B*, 2013, **140–141**, 68–81.
- 50 G. Lee, M.-K. Yeo, M.-H. Um and M. Kang, *Int. J. Photoenergy*, 2012, **2012**, 10.
- 51 L. Youngil, C. Jun-rak, L. Kwi Jong, E. S. Nathan and K. Donghoon, *Nanotechnology*, 2008, **19**, 415604.