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## Greigite Fe<sub>3</sub>S<sub>4</sub> as a new anode material for high-performance sodium-ion batteries†

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Transition metal dichalcogenide materials have been considered as promising anode materials for rechargeable sodium-ion batteries because of their high specific capacity and low cost. Here, we demonstrate an iron sulfide Fe<sub>3</sub>S<sub>4</sub> as a new anode material for a rechargeable sodium-ion battery. The involved conversion mechanism has been proved when the as-prepared Fe<sub>3</sub>S<sub>4</sub> was used as the host material for sodium storage. Remarkably, a compound FeS<sub>x</sub> with quantum size generated by conversion reaction overcame the kinetic and thermodynamic constraints of chemical conversion to achieve superior cycling and rate capability. As a result, the as-prepared Fe<sub>3</sub>S<sub>4</sub> electrode delivers a high reversible specific capacity of 548 mA h g<sup>-1</sup> at 0.2 A g<sup>-1</sup>, together with an excellent cycling stability of 275 mA h g<sup>-1</sup> after 3500 cycles at 20 A g<sup>-1</sup>.

Received 21st June 2016

Accepted 31st July 2016

DOI: 10.1039/c6sc02716d

www.rsc.org/chemicalscience

## Introduction

Owing to the increasing demand for sustainable and renewable power sources, much effort has been devoted to energy storage innovation over the past decades. Due to their high energy density and fast recharge capability, lithium-ion batteries (LIBs) have been successfully applied in many aspects of our daily life.<sup>1,2</sup> In particular, the application of LIBs in electric vehicles (EVs) and hybrid EVs has reduced significantly our energy dependence on 'one-off' resources. Nevertheless, concerns about LIBs have arisen both in terms of their cost and the supply limits of lithium resources in recent years.<sup>3</sup> Alternatively, sodium-ion batteries (SIBs) have recently attracted considerable interest owing to the low cost, wide distribution and abundant resource of sodium.<sup>4,5</sup> However, compared to Li<sup>+</sup> ion, the larger ionic radius and molar mass of Na<sup>+</sup> ion often lead to inferior cyclability and lower specific capacity.<sup>6,7</sup> There are still many challenges to exploit host materials for sodium with high capacity, fast charge-discharge, and long cycle life, especially for anode materials.<sup>6</sup>

The emerging transition metal dichalcogenides (TMD) materials which have been researched in electrochemistry<sup>8–12</sup> for many years have drawn extensive attention for SIBs in recent years.<sup>13–17</sup> These TMD materials often involve a multi-step reaction mechanism (intercalation and conversion, such as

MoS<sub>2</sub>) which contributes a high specific capacity but with poor cycling life.<sup>18</sup> Among these TMD materials, iron sulfides (FeS,<sup>13</sup> FeS<sub>2</sub> (ref. 15, 19 and 20)) have been researched in LIBs and SIBs numerous times owing to their high capacity, low cost and environmental friendliness. Unfortunately, the limited cycling life of iron sulfides severely restricts their real application in energy storage.<sup>21,22</sup> Wang *et al.* constructed the multi-functional yolk-shell FeS@C structure to improve the cycling stability, but which could only prolong the cycling life to 300 cycles.<sup>13</sup> Through controlling the cut-off voltage to avoid the conversion reaction, Hu *et al.* have improved the cycling life of iron sulfides to a quite high level (20 000 cycles) but with inferior capacity.<sup>15</sup> The key point to achieve high capacity and stability simultaneously is to sustain the high reversibility of the conversion reactions. Ultrafine nanoparticles have proved to be advantageous in this respect, which is attributed to nanoparticles having a size comparable to the diffusion length of the cation in the host-materials, leading to highly reversible and efficient conversion reaction.<sup>14</sup> However, it still remains a challenge to make common materials to reach the quantum size.<sup>23</sup>

Greigite Fe<sub>3</sub>S<sub>4</sub>, an important semi-metallic magnetic material, has been widely used in paleomagnetism, electrochemistry, biomedicine and environmental magnetic studies.<sup>24,25</sup> However, to the best of our knowledge, there is no report on Fe<sub>3</sub>S<sub>4</sub> as the anode of SIBs. Herein, we demonstrate Fe<sub>3</sub>S<sub>4</sub> as a promising host-material for sodium storage. The involved conversion reaction pulverizes the Fe<sub>3</sub>S<sub>4</sub> particles to quantum size during the sodiation/desodiation processes, resulting in a high capacity and superior stability. The synthesized Fe<sub>3</sub>S<sub>4</sub> particles display a discharge capacity of 548 mA h g<sup>-1</sup> in a wide operating voltage between 0.5 and 3 V. Meanwhile, the remarkable long-term cyclic stability (275 mA h g<sup>-1</sup> after 3500 cycles at 20 A g<sup>-1</sup>)

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† Electronic supplementary information (ESI) available. See DOI: 10.1039/c6sc02716d

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and excellent rate capability ( $233 \text{ mA h g}^{-1}$  at  $40 \text{ A g}^{-1}$ ) assure its great potential for practical utilization. This high reversible conversion mechanism presents a new method to enable SIBs possessing both high capacity and long-cycle stability.

## Results and discussion

As the counterpart of the oxide magnetite  $\text{Fe}_3\text{O}_4$ , greigite  $\text{Fe}_3\text{S}_4$  contains 32 atoms of sulfur and 24 atoms of iron per unit cell. There are two sublattices of iron atoms where the  $\text{Fe}^{3+}$  ions occupy tetrahedral A-sites and both  $\text{Fe}^{2+}$  and  $\text{Fe}^{3+}$  ions occupy octahedral B-sites (Fig. 1a).<sup>25</sup> Fig. 1b shows the X-ray diffraction (XRD) pattern of the as-prepared  $\text{Fe}_3\text{S}_4$ . All diffraction peaks are fully consistent with JCPDS no. 89-1998, showing a cubic  $Fd\bar{3}m$  space group. The morphology of the as-prepared  $\text{Fe}_3\text{S}_4$  is characterized by scanning electron microscopy (SEM) (Fig. 1c and d). The particles present octahedral features and the particle size is 100–200 nm. Fig. 1f clearly shows two lattices: (111) and (1–1–1), which are parallel to the surface of the octahedron. The intersection angle of these two lattice plane is measured to be  $109.5^\circ$ , which is consistent with the theoretical value. Combining the crystal structure of  $\text{Fe}_3\text{S}_4$  (Fig. 1a), it is speculated that the exposed faces are the  $\{111\}$  family of crystal planes.

To test the electrochemical performances of the as-prepared  $\text{Fe}_3\text{S}_4$  particles, CR2016 coin-type cells were fabricated. Fig. 2a shows the galvanostatic discharge/charge processes for the as-prepared  $\text{Fe}_3\text{S}_4$  electrode at a low current rate of  $0.2 \text{ A g}^{-1}$  in a range of 0.5–3.0 V. The  $\text{Fe}_3\text{S}_4$  delivers a high initial discharge capacity of  $571 \text{ mA h g}^{-1}$ , and the first charge capacity is 548

$\text{mA h g}^{-1}$ , which shows an impressive initial coulombic efficiency of 96%. After 100 cycles, it still delivers a reversible discharge capacity of  $536 \text{ mA h g}^{-1}$  (Fig. S1†). The cyclic voltammogram (CV) curves of the as-prepared  $\text{Fe}_3\text{S}_4$  electrode at a scan rate of  $0.2 \text{ mV s}^{-1}$  show that the charge and discharge processes maintain stable curves after the initial cycle (Fig. S2a†). The long-term cycling is tested under a relatively high specific current ( $5$  and  $20 \text{ A g}^{-1}$ ) as shown in Fig. 2b and S3.† After 50 cycles, the  $\text{Fe}_3\text{S}_4$  anode delivers a stable discharge capacity of  $435 \text{ mA h g}^{-1}$  at  $5 \text{ A g}^{-1}$ . After 1000 cycles, a capacity of  $401 \text{ mA h g}^{-1}$  is still obtained, showing an impressive cycling stability. Moreover, the shape of the capacity–voltage curves shows little change during cycling especially after 200 cycles, which confirms the stable and reversible discharge/charge processes (Fig. S2b†). The coulombic efficiency is kept at nearly 100% from beginning to end at such a high specific current. More attractively, a relatively high capacity of  $275 \text{ mA h g}^{-1}$  is also obtained after 3500 cycles even at a high specific current of  $20 \text{ A g}^{-1}$  (Fig. 2b). It should be pointed out that the superior cycling performance benefits from both the ether based electrolyte and the cut-off voltage, as shown in (Fig. S4†). Carbonate-based electrolytes ( $\text{NaClO}_4/\text{EC-DMC}$ ) suffer from rapid capacity fading ( $584 \text{ mA h g}^{-1}$  at the first cycle and  $15 \text{ mA h g}^{-1}$  at the 200th cycle respectively, Fig. S4a and b†). Moreover, when we extend the operating voltage to 0.01–3 V, the capacity seriously fades from 748 to  $132 \text{ mA h g}^{-1}$  within 100 cycles at  $2 \text{ A g}^{-1}$  (Fig. S4c and d†).

The rate capabilities of the as-prepared  $\text{Fe}_3\text{S}_4$  electrode are further investigated at various specific currents ranging from 0.2 to  $40 \text{ A g}^{-1}$  (Fig. 2c). Fig. 2d exhibits the corresponding charge and discharge curves of the  $\text{Fe}_3\text{S}_4$  electrode at different specific currents. The capacities show slight decline as the specific current gradually increases. When the specific currents reach 0.2, 0.5, 1, 2, 5, 10, 20, and  $25 \text{ A g}^{-1}$ , the discharge capacities are 548, 508, 476, 446, 407, 365, 308, and  $283 \text{ mA h g}^{-1}$ , respectively. It is noteworthy that even at an extremely high specific current of  $40 \text{ A g}^{-1}$ , a capacity of  $233 \text{ mA h g}^{-1}$  is still achieved, corresponding to 43% capacity utilization within 21 s. Its corresponding areal current densities and areal capacities are shown in Table S1.† Comparing the excellent rate performance with the state of the art conversion type anode materials, the  $\text{Fe}_3\text{S}_4$  anode has a distinct advantage at high specific currents (Fig. 2e).<sup>13,15,16</sup>

### Sodium-storage mechanism

$^{57}\text{Fe}$  Mössbauer spectra and TEM were used to study intensively the sodium-storage mechanism of the as-prepared  $\text{Fe}_3\text{S}_4$  particles (Fig. 3). Three states have been chosen to investigate the mechanism. The first charge and discharge curves (I (fresh state)  $\rightarrow$  II (discharge to 0.5 V)  $\rightarrow$  III (charge to 3 V)) of the  $\text{Fe}_3\text{S}_4$ -based battery at a specific current of  $0.2 \text{ A g}^{-1}$  are shown in Fig. 3a. The sodium-storage mechanism is first revealed *via*  $^{57}\text{Fe}$  Mössbauer spectra. The representative  $^{57}\text{Fe}$  Mössbauer spectra recorded at room temperature for all samples are shown in Fig. 3c. The pristine  $\text{Fe}_3\text{S}_4$  sample consists of two magnetic sextets and one central doublet, shown in Table S1.† Two sextets

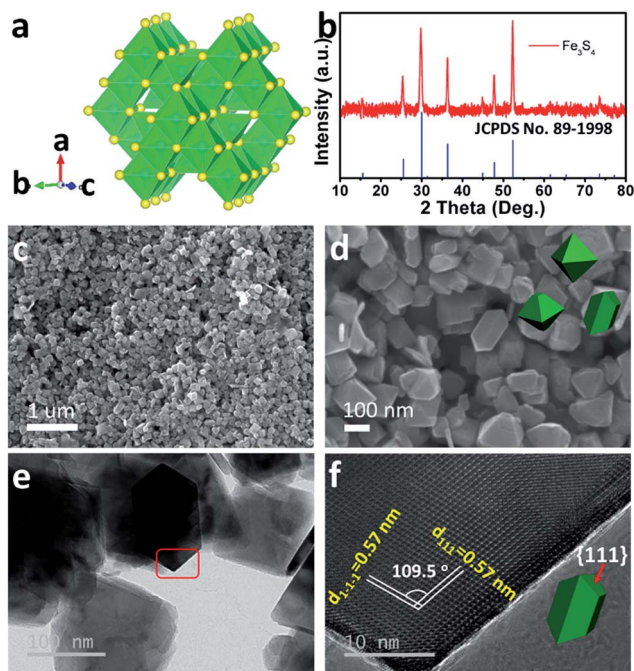


Fig. 1 Structure characterization of the as-prepared  $\text{Fe}_3\text{S}_4$ . (a) Crystal structure and (b) XRD pattern of the as-prepared  $\text{Fe}_3\text{S}_4$ . (c and d) SEM images, (e) TEM and (f) HRTEM images for the as-prepared  $\text{Fe}_3\text{S}_4$ .



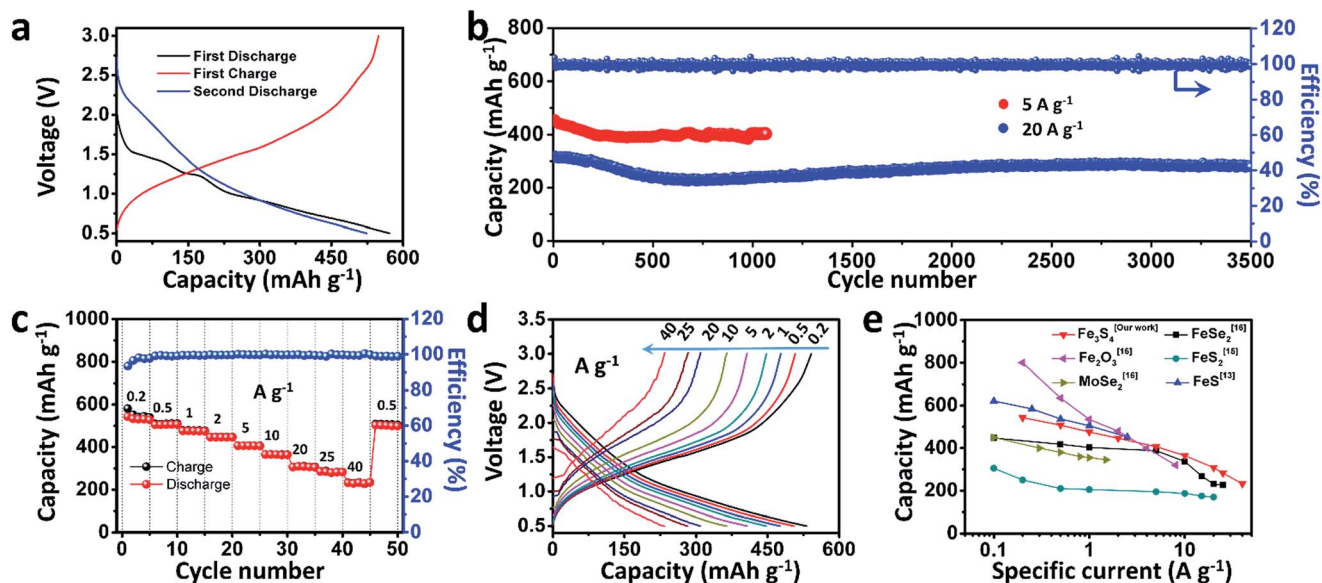


Fig. 2 Electrochemical performance of the as-prepared  $\text{Fe}_3\text{S}_4$ . (a) Discharge–charge curves of the first two cycles at a specific current of  $0.2 \text{ A g}^{-1}$ ; (b) cycling performances at 5 and  $20 \text{ A g}^{-1}$ ; (c) rate capability and (d) discharge–charge curves at various current rates; (e) comparison of the as-prepared  $\text{Fe}_3\text{S}_4$  with other conversion anode materials of SIBs.

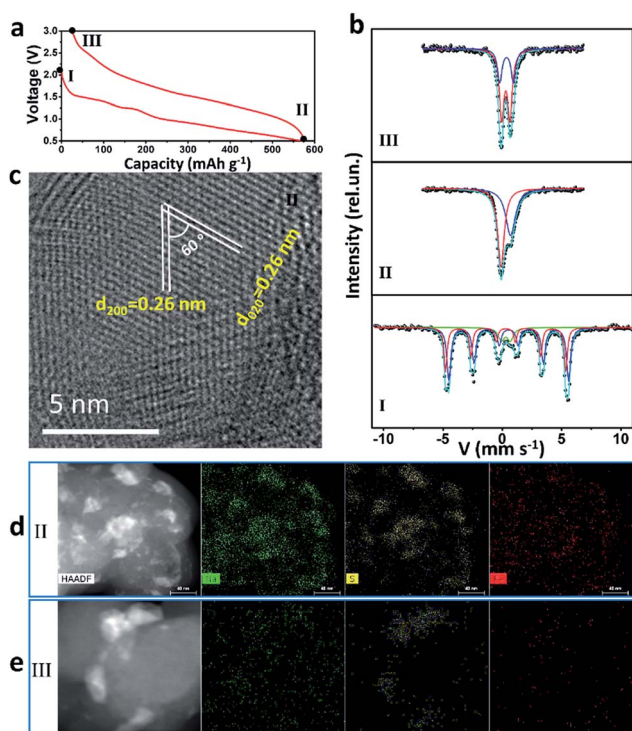


Fig. 3 (a) Discharge–charge curves of the first cycle at the specific current of  $0.2 \text{ A g}^{-1}$ ; (b)  $^{57}\text{Fe}$  Mössbauer spectra, (c) TEM image, (d) STEM and (e) EDS mappings of the as-prepared  $\text{Fe}_3\text{S}_4$  at different states.

represent hyperfine interactions of Fe ions in octahedral and tetrahedral sites. The quadrupole doublet is probably associated with thermally relaxed fine particles present in the sample but already not visible by XRD.<sup>26</sup> The obtained data is in good

agreement with the previous findings for natural and synthetic greigite.<sup>27</sup> The spectra of discharged and charged samples represent no magnetic sextets due to superparamagnetic behavior of small particles. The center shift (CS), quadrupole splitting (QS), and area ratio (A) obtained from analysis of the spectra are listed in Table S1.† For the discharged sample, the observed CS value of the bigger singlet (singlet 1) is  $-0.08 \text{ mm s}^{-1}$ , which clearly indicates that the iron is present in the nanostructured metallic state ( $\alpha\text{-Fe}$ ), the smaller singlet (singlet 2) can be attributed to nanoparticles of hexagonal  $\text{FeS}$ .<sup>28</sup> Moreover, the HRTEM image is collected at the state II (Fig. 3c), which displays two sets of parallel fringes with the same  $d$ -spacing of  $0.26 \text{ nm}$  and an included angle of  $60^\circ$  between them, corresponding to the (200) and (020) planes of  $\text{FeS}$  (JCPDF no. 37-0477), which is consistent with the results obtained by Mössbauer techniques. To verify the state of sulphur, scanning TEM (STEM) and energy dispersive X-ray spectrometer (EDS) mappings were collected at state II (Fig. 3d) and III (Fig. 3e). When discharged to  $0.5 \text{ V}$ , the distribution of Na is well consistent with that of S, verifying the formation of  $\text{Na}_2\text{S}$ . Fe shows a uniform distribution. Therefore, the initial discharge reaction can be expressed as eqn (S1).†

According to this equation, the theoretical capacity of  $\text{Fe}_3\text{S}_4$  when discharging to  $0.5 \text{ V}$  is calculated to be  $543 \text{ mA h g}^{-1}$ , which is very consistent with the reversible capacity of  $548 \text{ mA h g}^{-1}$  obtained at  $0.2 \text{ A g}^{-1}$ . Furthermore, the  $\text{Fe}^0$  produced during the conversion reaction improves the conductivity of the electrode significantly, which is confirmed by the electrochemical impedance spectrum (EIS) (Fig. S5†). The EIS spectrum shows two compressed semicircles from the high to medium frequency range of each spectrum at the state II, for which the second semicircle describes the charge transfer resistance ( $R_{ct}$ ) of the electrode. After simulation, the values of



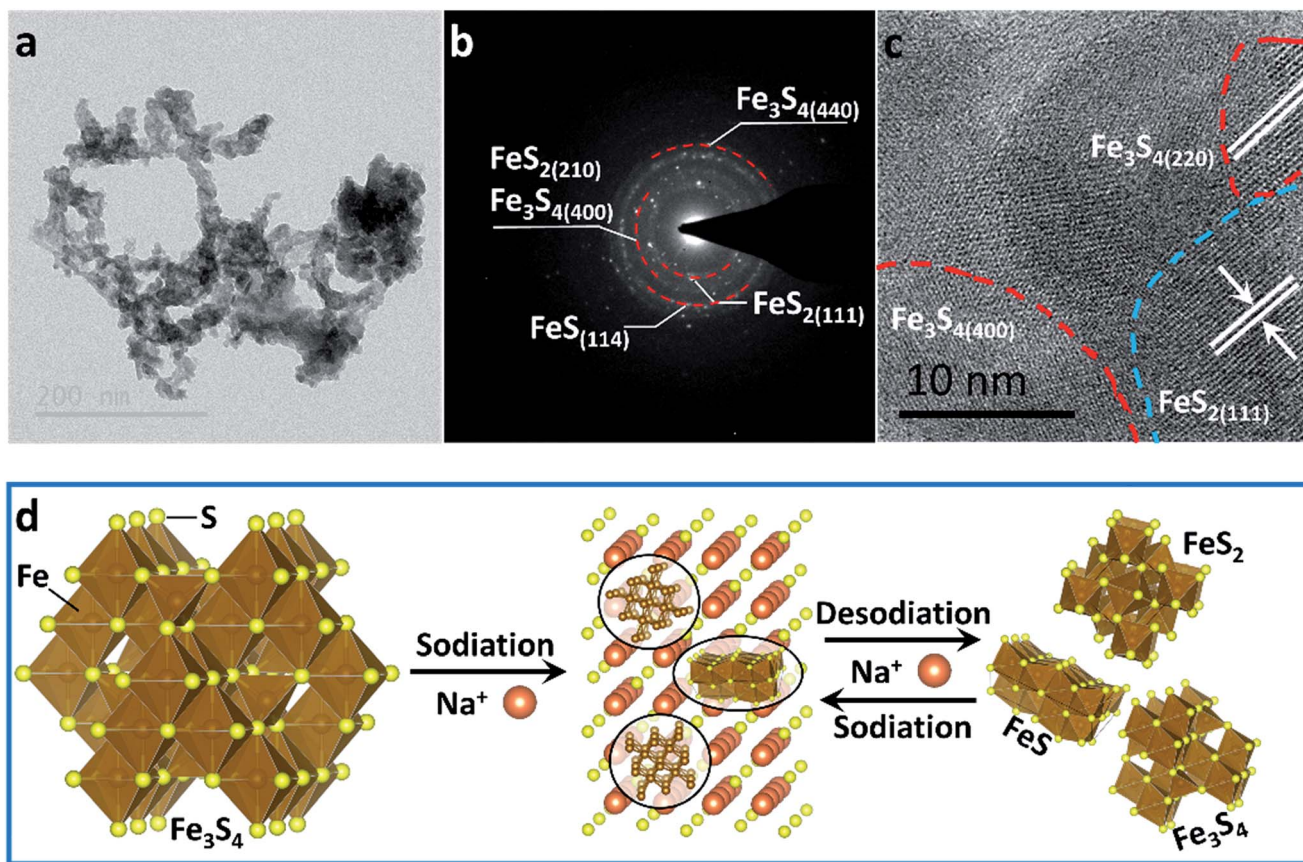


Fig. 4 (a) TEM image, (b) SAED pattern and (c) HRTEM image of the as-prepared  $\text{Fe}_3\text{S}_4$  at a full charge state after 200 cycles. (d) Schematic illustration of the sodium-storage mechanism in the  $\text{Fe}_3\text{S}_4$  electrode.

$R_{ct}$  for the state I and II electrodes are calculated to be 98.8 and 10.7  $\Omega$ , respectively. The improved conductivity after the formation of Fe has great benefits for high-rate performance.<sup>29</sup>

For charging process, the Mössbauer spectra for the charged sample can be fitted with two doublets with similar CS values but different QS values of 0.69 and 1.15  $\text{mm s}^{-1}$ . The first more-intense doublet can be attributed to the ferrous low spin  $\text{Fe}^{2+}$  state, probably in tetrahedral  $\text{FeS}_x$ .<sup>30</sup> The nature of the second doublet is unclear, and the ferric ion  $\text{Fe}^{3+}$  state can be caused by some other iron sulfide species. These results indicate that, after first charging to 3 V, the active materials exist in the form of compounds, which can be defined as  $\text{FeS}_x$ . The  $\text{FeS}_x$  is also confirmed by the HRTEM image at state III (Fig. S6†), which is composed of  $\text{FeS}_2$ ,  $\text{Fe}_3\text{S}_4$  and  $\text{FeS}$ . Notably, due to the conversion reaction, the  $\text{FeS}_x$  compounds are pulverized to nanocrystals with the size of  $\sim 1\text{--}10$  nm (Fig. S6b†). This quantum size is of great significance for iron sulfide to achieve improved cycling and better rate capability because of the shorter diffusion lengths of Fe in iron sulfide ( $L_D = 10^{-17}$   $\text{cm}^2 \text{s}^{-1}$  at 100  $^\circ\text{C}$  or  $\sim 10^{-18}$   $\text{cm}^2 \text{s}^{-1}$  at room temperature in  $\text{FeS}_2$ ).<sup>14,31,32</sup> The compound  $\text{FeS}_x$  with quantum size, which is comparable or smaller than the Fe diffusion distance during cation exchange, overcomes the significant kinetic and thermodynamic constraints of chemical conversion to achieve an excellent cycling and rate capability.<sup>14</sup> After charging back to 3 V, the

elemental mapping images display the uniform distribution of Na and Fe, but the S is still concentrated in some areas (Fig. 3e). These results demonstrate that the  $\text{S}^{2-}$  is at least in part transformed into  $\text{S}^0$ . Therefore, the possible reaction during the reversible charging processes is summarized as eqn (S2).†

To confirm the active material in the subsequent cycles, TEM images of the electrode at full charge state after 200 cycles were collected (Fig. 4). The Selected Area Electron Diffraction (SAED) patterns confirm the coexistence of  $\text{Fe}_3\text{S}_4$ ,  $\text{FeS}_2$  and  $\text{FeS}$  (Fig. 4b). The HRTEM image shows the nanocrystal of  $\text{Fe}_3\text{S}_4$  and  $\text{FeS}_2$  (Fig. 4c). The whole sodium-storage mechanism is illustrated in Fig. 4d. In the sodiation process,  $\text{Na}^+$  exchanges with  $\text{Fe}^{x+}$  to form  $\text{Na}_2\text{S}$ , and the exchanged  $\text{Fe}^{x+}$  obtain electrons to form  $\text{Fe}^0$ . A portion of  $\text{Fe}^{x+}$  still occupies octahedral sites to form  $\text{FeS}$  due to the controlled cut-off voltage. In the sodiation process,  $\text{Fe}^0$  exchanges with  $\text{Na}^+$  to form the Fe–S tetrahedron or octahedron, which further assembles to form the quantum-sized  $\text{FeS}_2$ ,  $\text{Fe}_3\text{S}_4$  and  $\text{FeS}$ . The quantum-sized  $\text{FeS}_x$  insure a synergistic and highly reversible conversion reaction which results in the superior cyclability and rate capability.<sup>14,33</sup>

## Conclusions

In summary,  $\text{Fe}_3\text{S}_4$  particles have been prepared successfully and used as an anode material for SIBs for the first time. A



conversion mechanism with 6 Na<sup>+</sup> per formula has been proved between a wide operating window of 0.5–3 V. Due to the conversion reaction, Fe<sub>3</sub>S<sub>4</sub> particles are pulverized to the quantum-sized compound FeS<sub>x</sub> which is composed of FeS, FeS<sub>2</sub> and Fe<sub>3</sub>S<sub>4</sub> quantum dots. The quantum-sized FeS<sub>x</sub> insure a synergistic and high reversible conversion reaction to provide the electrode with excellent cyclability and rate capability. As a result, Fe<sub>3</sub>S<sub>4</sub> delivers a stable discharge capacity of 275 mA h g<sup>-1</sup> after 3500 cycles at 20 A g<sup>-1</sup>. Even at 40 A g<sup>-1</sup>, a high discharge capacity of 233 mA h g<sup>-1</sup> is obtained. This remarkable performance makes Fe<sub>3</sub>S<sub>4</sub> a promising application candidate for the development of SIBs with high-rate capability and long-term cyclability. We believe that this high reversible conversion mechanism provides a new direction to improve the electrochemical performance of TMD materials for SIBs. Moreover, the involved electrochemical pulverization process provides a new route to synthesise quantum-sized materials.

## Acknowledgements

This work was supported by the National Basic Research Program of China (2013CB934103), the International Science & Technology Cooperation Program of China (2013DFA50840), the National Natural Science Foundation of China (51521001, 51272197, 51302203), the National Natural Science Fund for Distinguished Young Scholars (51425204), the Hubei Provincial Natural Science Fund for Distinguished Young Scholars (2014CFA035), and the Fundamental Research Funds for the Central Universities (WUT: 2016III001, 2016III002, 2016III003, 2016III004, 2016III006).

## Notes and references

- 1 B. Kang and G. Ceder, *Nature*, 2009, **458**, 190–193.
- 2 L. Li, R. Jacobs, P. Gao, L. Gan, F. Wang, D. Morgan and S. Jin, *J. Am. Chem. Soc.*, 2016, **138**, 2838–2848.
- 3 J.-M. Tarascon and M. Armand, *Nature*, 2001, **414**, 359–367.
- 4 S. W. Kim, D. H. Seo, X. Ma, G. Ceder and K. Kang, *Adv. Energy Mater.*, 2012, **2**, 710–721.
- 5 V. Palomares, M. Casas-Cabanas, E. Castillo-Martínez, M. H. Han and T. Rojo, *Energy Environ. Sci.*, 2013, **6**, 2312–2337.
- 6 V. Palomares, P. Serras, I. Villaluenga, K. B. Hueso, J. Carretero-González and T. Rojo, *Energy Environ. Sci.*, 2012, **5**, 5884–5901.
- 7 M. S. Islam and C. A. Fisher, *Chem. Soc. Rev.*, 2014, **43**, 185–204.
- 8 Y. Sun, C. Liu, D. C. Grauer, J. Yano, J. R. Long, P. Yang and C. J. Chang, *J. Am. Chem. Soc.*, 2013, **135**, 17699–17702.
- 9 N. Kornienko, J. Resasco, N. Becknell, C.-M. Jiang, Y.-S. Liu, K. Nie, X. Sun, J. Guo, S. R. Leone and P. Yang, *J. Am. Chem. Soc.*, 2015, **137**, 7448–7455.
- 10 A. B. Wong, S. Brittman, Y. Yu, N. P. Dasgupta and P. Yang, *Nano Lett.*, 2015, **15**, 4096–4101.
- 11 X. Xu, Z. Fan, X. Yu, S. Ding, D. Yu and X. W. D. Lou, *Adv. Energy Mater.*, 2014, **4**, 1400902.
- 12 X. Cao, C. Tan, X. Zhang, W. Zhao and H. Zhang, *Adv. Mater.*, 2016, **28**, 6167–6196.
- 13 Y.-X. Wang, J. Yang, S.-L. Chou, H. K. Liu, W.-x. Zhang, D. Zhao and S. X. Dou, *Nat. Commun.*, 2015, **6**, 8689.
- 14 A. Douglas, R. Carter, L. Oakes, K. Share, A. P. Cohn and C. L. Pint, *ACS Nano*, 2015, **9**, 11156–11165.
- 15 Z. Hu, Z. Zhu, F. Cheng, K. Zhang, J. Wang, C. Chen and J. Chen, *Energy Environ. Sci.*, 2015, **8**, 1309–1316.
- 16 K. Zhang, Z. Hu, X. Liu, Z. Tao and J. Chen, *Adv. Mater.*, 2015, **27**, 3305–3309.
- 17 W. J. Li, C. Han, S. L. Chou, J. Z. Wang, Z. Li, Y. M. Kang, H. K. Liu and S. X. Dou, *Chem.–Eur. J.*, 2016, **22**, 590–597.
- 18 L. David, R. Bhandavat and G. Singh, *ACS Nano*, 2014, **8**, 1759–1770.
- 19 T. Kim, J. Choi, H. Ryu, G. Cho, K. Kim, J. Ahn, K. Cho and H. Ahn, *J. Power Sources*, 2007, **174**, 1275–1278.
- 20 T. Kim, W. Jung, H. Ryu, K. Kim, J. Ahn, K. Cho, G. Cho, T. Nam, I. Ahn and H. Ahn, *J. Alloys Compd.*, 2008, **449**, 304–307.
- 21 R. Fong, J. Dahn and C. Jones, *J. Electrochem. Soc.*, 1989, **136**, 3206–3210.
- 22 D. Golodnitsky and E. Peled, *Electrochim. Acta*, 1999, **45**, 335–350.
- 23 H. Gu, R. Zheng, X. Zhang and B. Xu, *J. Am. Chem. Soc.*, 2004, **126**, 5664–5665.
- 24 Z. He, S. H. Yu, X. Zhou, X. Li and J. Qu, *Adv. Funct. Mater.*, 2006, **16**, 1105–1111.
- 25 I. Lyubutin, S. Starchikov, C.-R. Lin, S.-Z. Lu, M. O. Shaikh, K. Funtov, T. Dmitrieva, S. Ovchinnikov, I. Edelman and R. Ivantsov, *J. Nanopart. Res.*, 2013, **15**, 1–13.
- 26 L. Chang, A. P. Roberts, Y. Tang, B. D. Rainford, A. R. Muxworthy and Q. Chen, *J. Geophys. Res.*, 2008, **113**, B06104.
- 27 R. Vandenberghe, E. De Grave, P. De Bakker, M. Krs and J. Hus, *Hyperfine Interact.*, 1992, **68**, 319–322.
- 28 W. Kim, I. J. Park and C. S. Kim, *J. Appl. Phys.*, 2009, **105**, 07D535.
- 29 Q. Xie, Y. Ma, X. Wang, D. Zeng, L. Wang, L. Mai and D.-L. Peng, *ACS Nano*, 2016, **10**, 1283–1291.
- 30 I. Dékány, L. Turi, Z. Homonnay, A. Vértés and K. Burger, *Colloids Surf., A*, 1996, **119**, 195–203.
- 31 M. T. McDowell, Z. Lu, K. J. Koski, J. H. Yu, G. Zheng and Y. Cui, *Nano Lett.*, 2015, **15**, 1264–1271.
- 32 J. Chen and W. Harvey, *Metall. Trans. B*, 1975, **6**, 331–339.
- 33 J. Qian, Y. Xiong, Y. Cao, X. Ai and H. Yang, *Nano Lett.*, 2014, **14**, 1865–1869.

