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Temperature and rhodamine B sensing based on fluorescence intensity ratio of Er^{3+} upconversion emissions†

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In this paper, we report the temperature and rhodamine B (RhB)-concentration sensing behavior of $\text{Ag}/\text{ZnO}/\text{Er}^{3+}:\text{YbMoO}_4$ composite films based on the fluorescence intensity ratio (FIR) of two green upconversion (UC) emissions which are ascribed to the $^2\text{H}_{11/2}/^4\text{S}_{3/2} \rightarrow ^4\text{I}_{15/2}$ transitions of Er^{3+} . Through the strong and non-overlapping green UC emissions, the FIR of the two green emissions is closely related to temperature in the range of 300–650 K, which shows a high sensing accuracy and the maximum sensitivity of 0.01574 K^{-1} . Due to the wavelength-dependent absorption of dye molecules, the FIR of the two green UC emissions exhibits an excellent exponential relationship with the RhB concentration in the range of 0–1000 ppm, which is ascribed to the radiative energy transfer (RET) from the composite film to RhB molecules. It is anticipated that the FIR technique based on the UC luminescence of rare-earth ions is a potential method for multifunctional application both in thermometers and biosensors.

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Introduction

Over the past few decades, rare-earth (RE) ion-doped luminescence materials have drawn increasing attention for the efficient lighting, visualization, optical communication and health applications because of their various outstanding features, such as sharp emissions, large Stokes shifts and long luminescence lifetime.^{1–4} RE ion-doped upconversion (UC) materials, where higher energy photons can be emitted after absorbing lower energy photons through multi-photon processes, have been getting more and more interest due to their wide applications in biomedicine, photocatalysis, solar cells, sensing, etc.^{5–8} Among these applications, UC luminescence-based sensors, such as optical temperature sensors and biosensors, have attracted growing attention because they are non-contact and non-invasive, capable of fast response, high accurate and spatial resolution characteristics, and suitable to be applied in the biological and medical fields due to the large penetration depth into tissues, less photodamage and photobleaching, and low background noise.^{9,10}

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In general, UC luminescence-based optical temperature sensors are based on the variation of the UC emission intensity of RE ions with increasing temperature due to the temperature-dependent nonradiative rate at a particular energy level of interest.^{11,12} Also, the UC luminescence-based biosensors are usually based on the variation of the emission intensity of chromophores resulting from the concentration-dependent energy transfer (ET) from RE ions to target molecules.^{5,13} However, the fluorescence emission depends on many factors including the host material, the energy level of interest, the dimension of the material doped with RE ion, the excitation condition, etc. The above-mentioned temperature-dependent nonradiative rate and concentration-dependent energy transfer would often be incorrectly interpreted and susceptible to errors due to changes in those factors. One way of avoiding these obstacles is to use the fluorescence intensity ratio (FIR) method, which consists of measuring the intensity of fluorescence from two different emissions that have different temperature dependence or concentration dependence, whose ratio provides a measurand that is essentially independent of other disturbance factors, such as the reduction of the variability induced by the measuring conditions, and helps to improve the sensitivity, accuracy and resolution.¹¹

The FIR measurement technique has already been successfully used in optical temperature sensors based on the ratio of fluorescence intensity originating from two closely spaced energy levels with separations of the order of the thermal energy, which is independent of fluorescence loss and fluctuations in the excitation intensity.^{11,14,15} Although a variety of RE ions and host materials are adopted for temperature sensing



based on the FIR technique, two factors that mainly affect the FIR-based optical temperature sensing behavior are the overlap of the two fluorescence wavelengths from the coupled levels, and the sufficient fluorescence intensity from two coupled energy levels.¹⁴ Currently, biological detection using RE-doped UC materials as a biosensor mainly based on the fluorescence of chromophores activated by the ET process from RE ions resulting from the spectral overlap between RE ions (donor) and chromophores (acceptor).^{5,13,16,17} It is established that ET can be realized by either fluorescence resonance energy transfer (FRET) or radiative energy transfer (RET). FRET is a non-radiative dipole-dipole coupling process with an efficiency inversely proportional to the sixth power of the distance between donor and acceptor.¹⁸ FRET is extremely sensitive to small changes in the distance of the donor and acceptor, which are usually separated by a few nanometers.¹⁹ Thus, relatively complex methods must be used to combine the RE-doped UC materials with other chromophores. These methods are usually carried out in a solution which is usually unstable, which makes them irreproducible, with poor processability and low UC efficiency.^{13,20-22} Compared with FRET, RET is a process where the emission from the donor is reabsorbed by the acceptor. The propagation of radiation from donor to acceptor can be remote, which permits the development of solid state UC biosensor with a high UC efficiency, an easy and stable operation, and an excellent recyclability and tailorability. However, the low efficiency of the RET process usually leads to weak fluorescence of chromophores, which results in lower concentration detection sensitivity and a larger error.

The Ag/ZnO/Er³⁺:YbMoO₄ composite film under infrared excitation produces strong and non-overlapping green UC emissions of Er³⁺ ions. Based on such performance, the temperature sensing behavior of the composite film based on the FIR of two green UC emissions was greatly improved in this work. For instance, the highest temperature sensitivity of 0.01574 K⁻¹ achieved in the temperature range of 300–650 K is better than that of other reported Er³⁺ based UC sensing materials and most of the other RE based sensing materials. In addition, different from the biodetection methods based on the fluorescence of chromophores,^{5,13,16} the rhodamine B (RhB) dye concentration sensing behavior of the composite film was investigated based on the FIR of the two green UC emissions of the Er³⁺ ions. A good exponential relationship was found between the FIR and RhB concentration in the large concentration range of 0–1000 ppm, indicating the excellent biosensing behavior based on the FIR technique of the RE UC emissions.

Experimental

The Ag/ZnO/Er³⁺:YbMoO₄ composite film, consisting of three sublayers of Ag, ZnO nanowires and Er³⁺-doped YbMoO₄, was prepared as reported elsewhere.²³ First, Ag sublayer with a thickness of 28 nm was deposited on a glass substrate by direct-current magnetron sputtering. Second, radio-frequency magnetron sputtering was used to deposit ZnO seeds on the Ag sublayer and the hydrothermal method was utilized to grow a sublayer of ZnO nanowires (900 nm in length), based on the

ZnO seeds. Third, an Er³⁺-doped YbMoO₄ (molar ratio of Er : Yb = 1 : 10) sublayer with a thickness of 500 nm was spin-coated on the ZnO nanowires with a stoichiometric solution of Er(NO₃)₃·5H₂O, Yb(NO₃)₃·5H₂O and (NH₄)₆Mo₇O₂₄·4H₂O. Finally, the whole composite film was annealed at the temperature of 773 K for 2 h in a normal atmosphere.

The luminescence emission spectra of the composite film were detected by a Jobin Yvon iHR550 monochromator coupled to a CR131 photomultiplier tube under a 980 nm laser diode (LD) excitation. The spectral resolution of the experimental set-up was 0.1 nm. For the temperature-dependent luminescence measurement, a temperature controlling system was used to adjust the temperature from room temperature to 673 K, where the measuring and controlling accuracy were about ± 0.5 K. For the dye concentration-dependent luminescence measurement, the aqueous solution with a certain RhB concentration was adopted and the measurement was conducted at room temperature.

Results and discussion

Y-Type fiber optic reflection/backscatter probe bundles were used to introduce the 980 nm excitation light and simultaneously collect the temperature-dependent UC emissions, and a schematic diagram of the luminescence measurement is presented in Fig. 1. The advantages of this measuring system are that the flexible fiber makes the measurement process easier to operate in practical application, the composite film can be easily integrated into the fiber head, and the measuring light path is very simple.

The UC emission spectra of the Ag/ZnO/Er³⁺:YbMoO₄ composite film at different temperatures recorded by using the above measuring system are shown in Fig. 2. Intense green and faint red UC emissions are observed from the $^2\text{H}_{11/2}/^4\text{S}_{3/2} \rightarrow ^4\text{I}_{15/2}$ and $^4\text{F}_{9/2} \rightarrow ^4\text{I}_{15/2}$ transitions of Er³⁺ in the wavelength range of 500–700 nm for all temperatures, separately. The two green emission bands originating from the $^2\text{H}_{11/2}/^4\text{S}_{3/2} \rightarrow ^4\text{I}_{15/2}$ transitions show nearly no overlap, which boosts the accuracy of the measurement of the integral intensity. In addition, the variation of temperature does not change the position of the UC emission peaks, whereas the emission intensity clearly changes. The inset in Fig. 2 shows the integral intensity of UC emissions as a function of temperature. The intensity of the green UC emissions of the $^2\text{H}_{11/2} \rightarrow ^4\text{I}_{15/2}$ transition (I_{H}) and $^4\text{S}_{3/2} \rightarrow ^4\text{I}_{15/2}$ transition (I_{S}) undergo complex changes with temperature, whereas the red UC emission intensity of the $^4\text{F}_{9/2} \rightarrow ^4\text{I}_{15/2}$ transition (I_{Red}) is nearly independent of temperature.

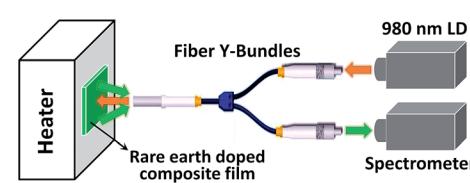


Fig. 1 Schematic diagram of the measurement of the temperature-dependent UC emissions.



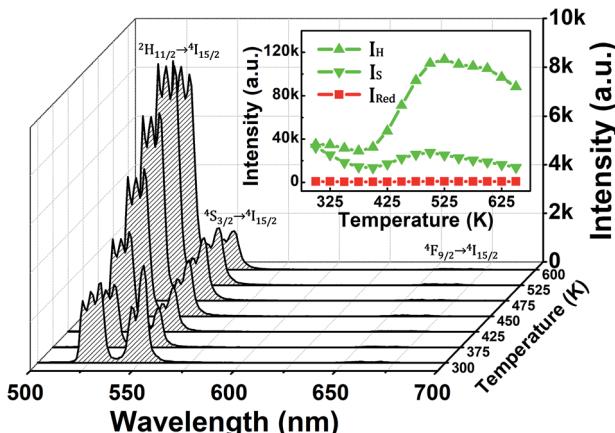


Fig. 2 UC emission spectra of the Ag/ZnO/Er³⁺:YbMoO₄ composite film at different temperatures. The inset shows the green and red emissions intensity versus the temperature.

It is well-known that the coupled energy levels of $^2\text{H}_{11/2}$ and $^4\text{S}_{3/2}$ are thermalized, whose population follows the Boltzmann's distribution. The FIR of the two green UC emissions from $^2\text{H}_{11/2} \rightarrow ^4\text{I}_{15/2}$ and $^4\text{S}_{3/2} \rightarrow ^4\text{I}_{15/2}$ transitions can be expressed as:¹¹

$$\text{FIR} = R = \frac{I_{\text{H}}}{I_{\text{S}}} = \frac{N(^2\text{H}_{11/2})}{N(^4\text{S}_{3/2})} = C \exp\left(-\frac{\Delta E}{kT}\right) \quad (1)$$

where, T is the absolute temperature, k is the Boltzmann constant, ΔE is the energy gap between the $^2\text{H}_{11/2}$ and $^4\text{S}_{3/2}$ levels. The pre-exponential factor C is a constant relative to the degeneracy, emission cross-section, and angular frequency of the corresponding transitions. A monolog plot of the FIR of the two green UC emissions as a function of the inverse absolute temperature is shown in Fig. 3a. The excellent linear fitting of the plot reveals that FIR is dependent on $1/T$ with an intercept of 3.36 ± 0.01 and a slope of about 988.10 ± 5.01 . The FIR of the two green UC emissions relative to the temperature is presented in Fig. 3b. The FIR of the two green UC emissions increases from 1.09 to 6.31 as the temperature increases from 300 K to 650 K, indicating an obvious variation of the FIR for the two green UC emissions. The value of the pre-exponential factor C is around 28.74 according to the fit of experimental data. The large determination coefficients (r^2) in Fig. 3a and b indicate the high

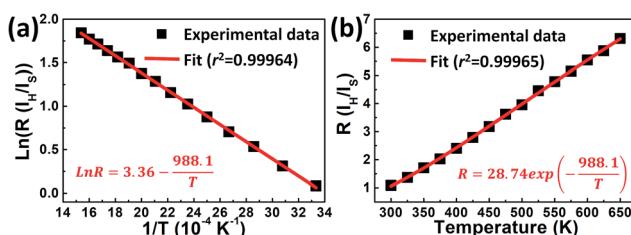


Fig. 3 (a) Monolog plot of the FIR of the green UC emissions for the $^2\text{H}_{11/2}/^4\text{S}_{3/2} \rightarrow ^4\text{I}_{15/2}$ transitions as a function of the inverse absolute temperature. (b) FIR of the green UC emissions for the $^2\text{H}_{11/2}/^4\text{S}_{3/2} \rightarrow ^4\text{I}_{15/2}$ transitions relative to the temperature.

accuracy of the measurement by the FIR technique using the composite film based temperature sensor.

The temperature sensitivity is an important characteristic for evaluating the performance of a thermometer. The relative sensitivity S_r , which allows the comparison between the sensitivities obtained from different thermally coupled energy levels, can be expressed as:¹¹

$$S_r = \frac{1}{R} \frac{dR}{dT} = \frac{\Delta E}{kT^2} \quad (2)$$

S_r is only dependent on the energy gap ΔE of the pair energy levels with larger ΔE in support of S_r . The S_r of the composite film-based sensor is $988.1/T^2$ from the fitting in Fig. 3. For comparison, Table 1 presents the relative sensitivities of other temperature-sensing materials based on the FIR technique. Although the energy gap ΔE for the $^2\text{H}_{11/2}$ and $^4\text{S}_{3/2}$ levels is different in those sensing materials, maybe as a result of the different environments of Er³⁺, the small difference of ΔE leads to the weak variation of the relative sensitivity S_r in Er³⁺-doped sensors (this work and ref. 24–32 in Table 1). Other RE ions (such as Nd³⁺, Eu³⁺, Dy³⁺ and Ho³⁺ in Table 1) based temperature sensing materials have higher S_r than that of this work because of the larger ΔE in those RE ions.

However, the absolute sensitivity S_a is a more important parameter for the practical application of optical thermal sensing, and is given by:¹¹

$$S_a = \frac{dR}{dT} = R \frac{\Delta E}{kT^2} \quad (3)$$

The absolute sensitivity S_a as a function of temperature for the composite film based sensor is depicted in Fig. 4. In the temperature range of our measurement, S_a initially increases and then decreases with elevated temperature, with the maximum value of about 0.01574 K^{-1} at 494 K. From our previous study,³⁸ the maximum absolute sensitivity S_{max} is expressed by:

$$S_{\text{max}} = \frac{0.54C}{\Delta E/k} \quad (4)$$

which depends not only on the energy gap ΔE , but also on the pre-exponential factor C . The comparison of S_{max} between this work and other sensing materials is also presented in Table 1. The S_{max} in this work is larger than those of other sensing materials based Er³⁺ and most of the other RE ions. To obtain higher S_{max} , both smaller energy gap ΔE and larger pre-exponential factor C are required, like the Tm³⁺-doped sensing materials (ref. 37 in Table 1). In addition, the optimal operating temperature range is also vital for temperature sensors. The parameter T_{max} , at the temperature where the sensor has maximum S_{max} , is given by

$$T_{\text{max}} = \frac{\Delta E}{2k} = \frac{S_r T^2}{2} \quad (5)$$

which is only relative to the energy gap ΔE . Like the relative sensitivity S_r , there is also no obvious difference in the T_{max} among Er³⁺-doped sensing materials (around 500 K) due to the similar energy gap ΔE of $^2\text{H}_{11/2}$ and $^4\text{S}_{3/2}$ coupled energy levels,



Table 1 Temperature sensing properties based on the luminescent materials with RE ions as the activator

Materials	Transitions	ΔE (cm ⁻¹)	S_f (K ⁻¹)	C	S_{max} (10 ⁻⁴ K ⁻¹)	T_{max} (K)	Ref.
Ag/ZnO/Er ³⁺ :YbMoO ₄ composite film	Er ³⁺ : $^2H_{11/2}, ^4S_{3/2} \rightarrow ^4I_{15/2}$	687	988.1/T ²	28.74	157.4	494	This work
Er ³⁺ /Yb ³⁺ :CaWO ₄	Er ³⁺ : $^2H_{11/2}, ^4S_{3/2} \rightarrow ^4I_{15/2}$	745	1072/T ²	13.87	72.1	535	24
Er ³⁺ /Yb ³⁺ /Li ⁺ :NaYF ₄	Er ³⁺ : $^2H_{11/2}, ^4S_{3/2} \rightarrow ^4I_{15/2}$	804	1198.2/T ²	18.54	59.2	453	25
Er ³⁺ :KYb ₂ F ₇	Er ³⁺ : $^2H_{11/2}, ^4S_{3/2} \rightarrow ^4I_{15/2}$	844	1224/T ²	10.29	45.4	590	26
Er ³⁺ :Yb ₂ Ti ₂ O ₇	Er ³⁺ : $^2H_{11/2}, ^4S_{3/2} \rightarrow ^4I_{15/2}$	473	679.2/T ²	9.3	74	340	27
Er ³⁺ :PLZT ceramics	Er ³⁺ : $^2H_{11/2}, ^4S_{3/2} \rightarrow ^4I_{15/2}$	773	1108.3/T ²	3.88	18.9	554	28
Er ³⁺ :PKAZLF glass	Er ³⁺ : $^2H_{11/2}, ^4S_{3/2} \rightarrow ^4I_{15/2}$	850	1218.6/T ²	21.5	79	630	29
Er ³⁺ /Yb ³⁺ :Y ₂ O ₃	Er ³⁺ : $^2H_{11/2}, ^4S_{3/2} \rightarrow ^4I_{15/2}$	827	1185.5/T ²	19.89	91	593	30
Er ³⁺ /Yb ³⁺ :SrWO ₄	Er ³⁺ : $^2H_{11/2}, ^4S_{3/2} \rightarrow ^4I_{15/2}$	602	866.1/T ²	21.12	149.8	403	31
Er ³⁺ :BaMoO ₄	Er ³⁺ : $^2H_{11/2}, ^4S_{3/2} \rightarrow ^4I_{15/2}$	655	939/T ²	—	227	463	32
Nd ³⁺ :P-K-Ba-Al glass	Nd ³⁺ : $^4F_{5/2}, ^4F_{3/2} \rightarrow ^4I_{9/2}$	908	1301.8/T ²	5.36	22.2	651	33
Eu ³⁺ : $(Y_{0.75}Gd_{0.25})_2O_3$	Eu ³⁺ : $^5D_1, ^5D_0 \rightarrow ^7F_1$	1633	2350/T ²	24.84	49	823	34
Dy ³⁺ :Y ₄ Al ₂ O ₉	Dy ³⁺ : $^4I_{5/2}, ^4F_{9/2} \rightarrow ^6H_{15/2}$	1347	1937.6/T ²	11.0	30	873	35
Ho ³⁺ /Yb ³⁺ :CaWO ₄	Ho ³⁺ : $^5F_{2,3} \rightarrow ^3K_8, ^5G_6 \rightarrow ^5F_1 \rightarrow ^5I_8$	1304	1890/T ²	16.67	50	923	36
Tm ³⁺ /Yb ³⁺ :Y ₂ O ₃	Tm ³⁺ : $^1D_2 \rightarrow ^3F_4, ^3H_4 \rightarrow ^3H_6$	394	564.9/T ²	82.76	780	270	37

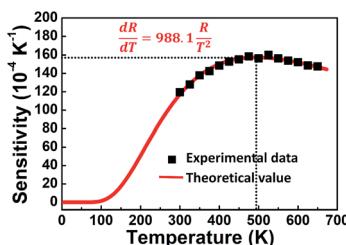


Fig. 4 Dependence of absolute sensitivity on the temperature for the composite film based sensor.

as shown in Table 1. Using an RE ion with larger energy gap ΔE contributes to higher T_{max} , such as Nd³⁺, Eu³⁺, Dy³⁺ and Ho³⁺, shown in Table 1.

A schematic diagram of the luminescence measurement of the RhB concentration-dependent UC emission is presented in Fig. 5. The Ag/ZnO/Er³⁺:YbMoO₄ composite film was attached to the outer wall of a cuvette containing the aqueous solution with different RhB concentration. The composite film was excited by the 980 nm LD and the corresponding luminescent signal was collected at the other side of the cuvette. The advantage of this measuring system is that the amount of RhB solution can be precisely controlled for the measurement of different RhB

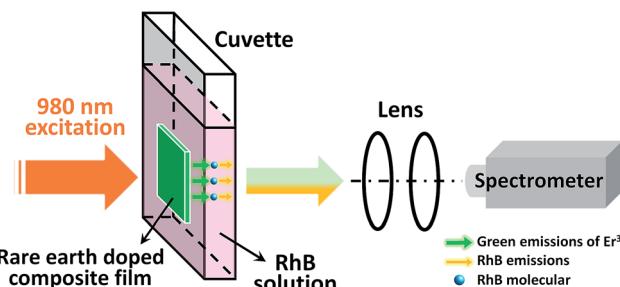
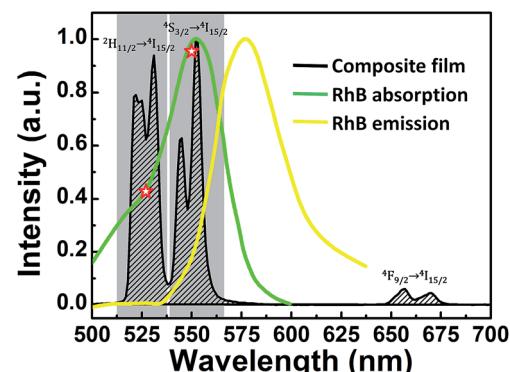


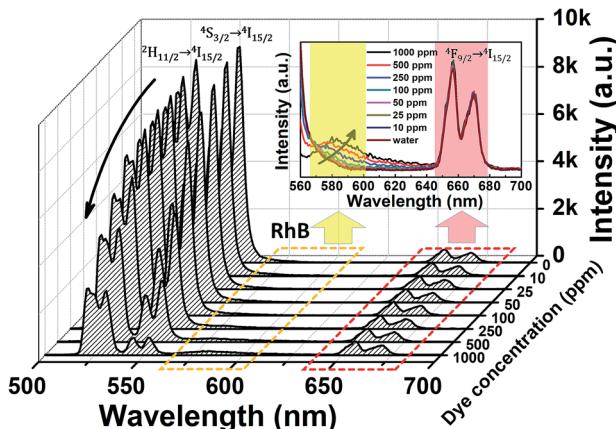
Fig. 5 Schematic diagram of the luminescence measurement of the RhB concentration-dependent UC emissions.

concentration, and the measuring system can be easily, stably and repeatedly reused in the practical application.

Under the above measuring system, the Ag/ZnO/Er³⁺:YbMoO₄ composite film still exhibits strong green UC emissions and weak red UC emissions, and the UC emission spectrum for the RhB concentration of $C_{RhB} = 0$ (pure water) is shown in Fig. 6. The normalized RhB relative absorption and emission spectra are also presented for comparison.³⁹ The RhB absorption band has a large degree of overlap with the green UC emission band of the composite film, which suggests that ET may be taking place between the composite film (as the donor) and the RhB molecules (as the acceptor).

The UC emission spectra of the composite film at different C_{RhB} is shown in Fig. 7. The spectra reveal that with increasing C_{RhB} , the intensity of the green UC emissions gradually decreases and the FIR of two green UC emissions also varies likewise, while a new emission band from RhB appears in the wavelength range of 560–640 nm. The inset of Fig. 7 reveals that

Fig. 6 UC emission spectrum of the Ag/ZnO/Er³⁺:YbMoO₄ composite film (black line) under the measurement condition in Fig. 5. The normalized RhB absorption (green line) and emission (yellow line) spectra are presented for comparison. The asterisks calculated by eqn (6) denote the relative RhB absorption coefficients of the two green UC emissions for the $^2H_{11/2}/^4S_{3/2} \rightarrow ^4I_{15/2}$ transitions.

Fig. 7 UC emission spectra of the composite film at different C_{RhB} .

the intensity of the RhB emissions increases with RhB concentration, and the RhB emission peak gradually redshifts with increasing C_{RhB} , which could be explained by the reabsorption effect of RhB.⁴⁰ The red UC emission intensity from the composite film remains almost unchanged with the variation of the RhB concentration, indicating that the red UC emission has no interaction with the RhB dye. Note that the RhB emission at low concentration (<100 ppm) is difficult to observe because of the relatively low intensity.

To clearly detect the influence of the RhB concentration on the fluorescence emission, the integrated intensity of the green UC emissions and RhB emissions as a function of the RhB concentration is given in Fig. 8. The monotonical decrease of the green UC emissions and increase of RhB emissions can be detected. Due to the identical measuring condition, the decrease of the green UC emissions is uniquely caused by the increased absorption resulting from the increasing RhB concentration. Meanwhile, the intensity change of the RhB emissions can be ascribed to the ET process between the composite film and RhB molecules because of no absorption of RhB in the wavelength around 980 nm.

As mentioned above, FRET only occurs in a nanoscale spacing between the donor and the acceptor, thus, due to the large gap in the cuvette wall between the composite film and the RhB solution, only RET process exists in this work. A plausible schematic diagram is presented in Fig. 9 to illustrate the emission and ET process. Under the 980 nm LD excitation, the

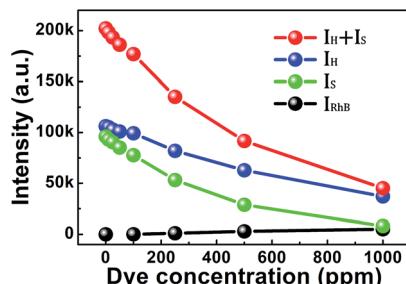


Fig. 8 Integral intensity of the green UC emissions and RhB emissions as a function of the RhB concentration.

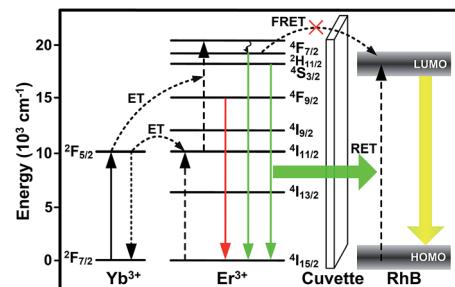


Fig. 9 Schematic diagram of a plausible mechanism for the emission and ET processes in the measuring system.

well-known processes of absorption and subsequent ETs are ascribed to the green and red UC emissions of the composite film. Then, part of the green UC emissions from the $^2\text{H}_{11/2}/^4\text{S}_{3/2} \rightarrow ^4\text{I}_{15/2}$ transitions could be reabsorbed by the RhB dye molecules through the RET process, exciting the RhB molecules from the highest occupied molecular orbital (HOMO) to the lowest unoccupied molecular orbital (LUMO). Next, the radiative transition of RhB molecules from LUMO to HOMO produces the RhB emissions as shown in Fig. 7.

As the decrease of the green UC emission intensity is caused only by the reabsorption of RhB molecules, the relationship between the green UC emissions and RhB concentration satisfies the Beer–Lambert law,

$$I_{\lambda}(C) = I_{\lambda}(0) \times 10^{-K_{\lambda} l C} = I_{\lambda}(0) \exp(-2.303 K_{\lambda} l C) \quad (6)$$

where, C is the concentration of the absorbent substance (RhB molecules in this work), $I_{\lambda}(0)$ and $I_{\lambda}(C)$ are the intensity of the transmitting light with a wavelength of λ at the concentration of 0 and C , respectively. K_{λ} and l are the absorption coefficient at wavelength of λ and the optical length, respectively. Since the absorption property of RhB is relative to the wavelength (green line in Fig. 6), we study the intensity evolution of the two green UC emissions for the $^2\text{H}_{11/2} \rightarrow ^4\text{I}_{15/2}$ and $^4\text{S}_{3/2} \rightarrow ^4\text{I}_{15/2}$ transitions with elevated RhB concentration. According to eqn (6), the FIR of the two green UC emissions can be expressed as:

$$\text{FIR} = R = \frac{I_{\text{H}}}{I_{\text{S}}} = \frac{I_{\text{H}0} \exp(-2.303 K_{\text{H}} l C_{\text{RhB}})}{I_{\text{S}0} \exp(-2.303 K_{\text{S}} l C_{\text{RhB}})} = \frac{I_{\text{H}0}}{I_{\text{S}0}} \exp(2.303 \Delta K_{\text{S-H}} l C_{\text{RhB}}) \quad (7)$$

where, $I_{\text{H}0}$ and I_{H} are the UC intensity of the $^2\text{H}_{11/2} \rightarrow ^4\text{I}_{15/2}$ transition at the RhB concentration of 0 and C_{RhB} , respectively, and $I_{\text{S}0}$ and I_{S} are the UC intensity of the $^4\text{S}_{3/2} \rightarrow ^4\text{I}_{15/2}$ transition at the RhB concentration of 0 and C_{RhB} , respectively. K_{H} and K_{S} are the absorption coefficient of RhB at the center wavelength of the $^2\text{H}_{11/2} \rightarrow ^4\text{I}_{15/2}$ and $^4\text{S}_{3/2} \rightarrow ^4\text{I}_{15/2}$ transitions, respectively, and $\Delta K_{\text{S-H}} = K_{\text{S}} - K_{\text{H}}$. Eqn (7) shows that $\ln R$ is linearly relative to C_{RhB} , and the corresponding plot of FIR as a function of C_{RhB} is displayed in Fig. 10. Within the concentration range from 0 to 1000 ppm, there is a good linear relationship between $\ln R$ and C_{RhB} with an intercept of 0.096 ± 0.006 and a slope of 0.0014 ± 0.000017 . The large determination coefficients (r^2) in Fig. 10a

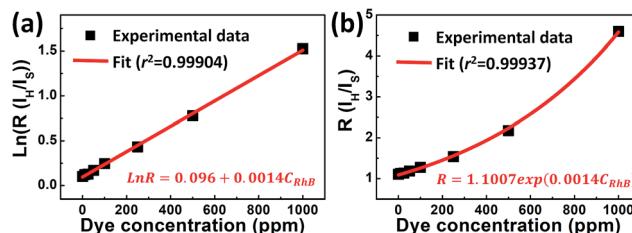


Fig. 10 (a) Monolog plot of the FIR of the green UC emissions for the $^2\text{H}_{11/2}/^4\text{S}_{3/2} \rightarrow ^4\text{I}_{15/2}$ transitions as a function of the RhB concentration. (b) FIR of the green UC emissions for the $^2\text{H}_{11/2}/^4\text{S}_{3/2} \rightarrow ^4\text{I}_{15/2}$ transitions relative to the RhB concentration.

and b indicate that the FIR technique based on the two green UC emissions from the composite film is suitable for the detection of the RhB concentration.

Eqn (7) also indicates that the FIR is relative to the difference of absorption coefficient for the two green UC emissions ($\Delta K_{\text{S}-\text{H}}$) and the initial intensity ratio of the two green UC emissions ($I_{\text{H}0}/I_{\text{S}0}$). Since the two green UC emissions of Er^{3+} have different absorption coefficients (the asterisks shown in Fig. 6), the FIR of the two green UC emissions is an efficient measurand that is essentially independent of the fluorescence loss or fluctuations in excitation intensity. Another advantage of using RE UC emissions for RhB sensing is that the UC emissions by infrared excitation are very suitable for biodetection. In addition, there are plenty of monochromatic RE UC emissions which can be used for the FIR technique. For instance, the green UC emissions of the composite film can be divided into five different monochromatic Stark UC emissions (Figs. S1 and S2†), where the FIR of two of them can be suitably used for RhB concentration sensing, as shown in Fig. S3(e–j).†

Like the relative sensitivity of temperature sensing in eqn (2), the relative sensitivity S_r of the RhB concentration sensing can be given by:

$$S_r = \frac{1}{R} \frac{dR}{dC_{\text{RhB}}} = 2.303 \Delta K I \quad (8)$$

where, ΔK is the difference of absorption coefficients for the two UC emissions. The S_r is the slope of the $\ln R - C_{\text{RhB}}$ plot (Fig. 10a), which is strongly relative to ΔK , and large ΔK helps to improve the relative sensitivity S_r . The relative absorption coefficients of the two green UC emissions for the $^2\text{H}_{11/2}/^4\text{S}_{3/2} \rightarrow ^4\text{I}_{15/2}$ transitions (the asterisks calculated by eqn (6) which are very consistent with the absorption spectrum of RhB, as shown in Fig. 6) have a large difference of about 55%, which is suitable for the FIR technique. It is also found that the two Stark UC emissions with a ΔK larger than 37% (Fig. S3(i)†) can be used for the FIR technique with high accuracy (Fig. S3(e–j)†), while the small ΔK of the two Stark UC emissions will cause a large error, which deviates from the linear relationship (Fig. S3(a–d)†).

Note that the RhB emissions at the wavelength range of 560–620 nm are also relative to the RhB concentration (Fig. 7). Accordingly, we also study the relationship between the RhB emission intensity (I_{RhB}) and C_{RhB} . It is found that I_{RhB} is exponentially related to the C_{RhB} but has a very large fitting error

(Fig. S4†). Due to the low RhB fluorescence efficiency (only about 2.82% calculated from Fig. S4†), the RhB fluorescence emissions was very weak and even undetectable at lower concentration (Fig. 7 and S2†), indicating that it is unsuitable for RhB sensing based on the RhB emissions in this work. Thus, compared with the RhB sensing based on the RhB emissions, the FIR technique based on the two green UC emissions of Er^{3+} is proved to be an effective and excellent approach for RhB sensing with high detection accuracy. Moreover, the FIR technique is simpler than other methods, such as the lifetime measurement, because of the easy fluorescence detection and the low cost of measurement. Compared with conventional absorption measurements, the use of FIR technique would be beneficial in microfluidic devices where short optical paths prevent the use of the conventional attenuated transmission technique for optical measurements.

Conclusions

In summary, both the temperature and RhB-concentration sensing properties of the $\text{Ag}/\text{ZnO}/\text{Er}^{3+}:\text{YbMoO}_4$ composite film were investigated using the FIR technique. The FIR of the two green UC emissions of Er^{3+} was studied as a function of temperature. The maximum sensitivity was approximately 0.01574 K^{-1} , which is higher than that of other Er^{3+} -based sensing materials and most of the other RE ions-based sensing materials. The FIR of the two green UC emissions of Er^{3+} was also found to be exponentially dependent on the RhB concentration with high measurement accuracy. It is suggested that the FIR technique is an effective method in the multifunctional temperature and dye concentration sensors.

Conflicts of interest

There are no conflicts to declare.

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Notes and references

- 1 H. Yu, D. Kim, J. Lee, S. Baek, J. Lee, R. Singh and F. So, *Nat. Photonics*, 2016, **10**, 129–134.
- 2 S. L. Gai, C. X. Li, P. P. Yang and J. Lin, *Chem. Rev.*, 2014, **114**, 2343–2389.
- 3 H. Lin, G. Meredith, S. B. Jiang, X. Peng, T. Luo, N. Peyghambarian and E. Y. B. Pun, *J. Appl. Phys.*, 2003, **93**, 186–191.
- 4 F. Auzel, *Chem. Rev.*, 2004, **104**, 139–173.
- 5 S. Xu, S. H. Xu, Y. S. Zhu, W. Xu, P. W. Zhou, C. Y. Zhou, B. Dong and H. W. Song, *Nanoscale*, 2014, **6**, 12573–12579.



6 Y. N. Tang, W. H. Di, X. S. Zhai, R. Y. Yang and W. P. Qin, *ACS Catal.*, 2013, **3**, 405–412.

7 J. Wang, Y. Niu, M. Hojamberdiev, F. M. Alamgir, Y. Cai and K. Jacob, *Sol. Energy Mater. Sol. Cells*, 2017, **160**, 361–371.

8 B. Dong, B. S. Cao, Y. Y. He, Z. Liu, Z. P. Li and Z. Q. Feng, *Adv. Mater.*, 2012, **24**, 1987–1993.

9 F. Wang and X. G. Liu, *Chem. Soc. Rev.*, 2009, **38**, 976–989.

10 J. Zhou, Z. Liu and F. Y. Li, *Chem. Soc. Rev.*, 2012, **41**, 1323–1349.

11 S. A. Wade, S. F. Collins and G. W. Baxter, *J. Appl. Phys.*, 2003, **94**, 4743–4756.

12 S. W. Allison and G. T. Gillies, *Rev. Sci. Instrum.*, 1997, **68**, 2615–2650.

13 S. Xu, W. Xu, Y. F. Wang, S. Zhang, Y. S. Zhu, L. Tao, L. Xia, P. W. Zhou and H. W. Song, *Nanoscale*, 2014, **6**, 5859–5870.

14 X. F. Wang, Q. Liu, Y. Y. Bu, C. S. Liu, T. Liu and X. H. Yan, *RSC Adv.*, 2015, **5**, 86219–86236.

15 M. D. Dramicanin, *Methods Appl. Fluoresc.*, 2016, **4**, 042001.

16 K. C. Liu, Z. Y. Zhang, C. X. Shan, Z. Q. Feng, J. S. Li, C. L. Song, Y. N. Bao, X. H. Qi and B. Dong, *Light: Sci. Appl.*, 2016, **5**, e16136.

17 R. Chen, V. D. Ta, F. Xiao, Q. Y. Zhang and H. D. Sun, *Small*, 2013, **9**, 1052–1057.

18 H. Edelhoch, L. Brand and M. Wilchek, *Biochemistry*, 1967, **6**, 547–559.

19 R. Gill, M. Zayats and I. Willner, *Angew. Chem., Int. Ed.*, 2008, **47**, 7602–7625.

20 J. L. Liu, Y. Liu, Q. Liu, C. Y. Li, L. N. Sun and F. Y. Li, *J. Am. Chem. Soc.*, 2011, **133**, 15276–15279.

21 D. T. Tu, L. Q. Liu, Q. Ju, Y. S. Liu, H. M. Zhu, R. F. Li and X. Y. Chen, *Angew. Chem., Int. Ed.*, 2011, **50**, 6306–6310.

22 K. Kuningas, T. Rantanen, T. Ukonaho, T. Lovgren and T. Soukka, *Anal. Chem.*, 2005, **77**, 7348–7355.

23 J. L. Wu, B. S. Cao, L. Rino, Y. R. Fang, L. Hu, Z. Y. Zhang, Y. Z. Huang and B. Dong, *Nanoscale*, 2017, submitted.

24 S. Sinha, M. K. Mahata, K. Kumar, S. P. Tiwari and V. K. Rai, *Spectrochim. Acta, Part A*, 2017, **173**, 369–375.

25 A. Dubey, A. K. Soni, A. Kumari, R. Dey and V. K. Rai, *J. Alloys Compd.*, 2017, **693**, 194–200.

26 J. K. Cao, F. F. Hu, L. P. Chen, H. Guo, C. K. Duan and M. Yin, *J. Alloys Compd.*, 2017, **693**, 326–331.

27 B. S. Cao, Y. Y. He, Z. Q. Feng, Y. S. Li and B. Dong, *Sens. Actuators, B*, 2011, **159**, 8–11.

28 A. S. S. de Camargo, J. F. Possatto, L. A. d. O. Nunes, É. R. Botero, É. R. M. Andreatta, D. Garcia and J. A. Eiras, *Solid State Commun.*, 2006, **137**, 1–5.

29 N. Vijaya, P. Babu, V. Venkatramu, C. K. Jayasankar, S. F. León-Luis, U. R. Rodríguez-Mendoza, I. R. Martín and V. Lavín, *Sens. Actuators, B*, 2013, **186**, 156–164.

30 V. Lojpur, G. Nikolić and M. D. Dramicanin, *J. Appl. Phys.*, 2014, **115**, 203106.

31 A. Pandey, V. K. Rai, V. Kumar, V. Kumar and H. C. Swart, *Sens. Actuators, B*, 2015, **209**, 352–358.

32 A. K. Soni, A. Kumari and V. K. Rai, *Sens. Actuators, B*, 2015, **216**, 64–71.

33 C. Perez-Rodriguez, L. L. Martin, S. F. Leon-Luis, I. R. Martin, K. K. Kumar and C. K. Jayasankar, *Sens. Actuators, B*, 2014, **195**, 324–331.

34 M. G. Nikolic, V. Lojpur, Z. Antic and M. D. Dramicanin, *Phys. Scr.*, 2013, **87**, 055703.

35 Z. Boruc, M. Kaczkan, B. Fetlinski, S. Turczynski and M. Malinowski, *Opt. Lett.*, 2012, **37**, 5214–5216.

36 W. Xu, H. Zhao, Y. X. Li, L. J. Zheng, Z. G. Zhang and W. W. Cao, *Sens. Actuators, B*, 2013, **188**, 1096–1100.

37 V. Lojpur, M. Nikolic, L. Mancic, O. Milosevic and M. D. Dramicanin, *Ceram. Int.*, 2013, **39**, 1129–1134.

38 B. S. Cao, J. L. Wu, X. H. Wang, Y. Y. He, Z. Q. Feng, B. Dong and L. Rino, *AIP Adv.*, 2015, **5**, 087136.

39 J. A. Sutton, B. T. Fisher and J. W. Fleming, *Exp. Fluids*, 2008, **45**, 869–881.

40 T. C. He, R. Chen, W. W. Lin, F. Huang and H. D. Sun, *Appl. Phys. Lett.*, 2011, **99**, 081902.

