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Novel nanofiber yarns synchronously endued with tri-functional performance of superparamagnetism, electrical conductivity and enhanced fluorescence prepared by conjugate electrospinning

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Novel one-dimensional [Fe₃O₄/PANI/PAN]//[Eu(BA)₃phen/PAN] (PANI = polyaniline, PAN = polyacrylonitrile, BA = benzoic acid, and phen = 1,10-phenanthroline) heterogeneous nanofiber yarns synchronously endued with excellent tri-functionality of fluorescence, superparamagnetism and electrical conductivity were successfully fabricated for the first time using conjugate electrospinning. The heterogeneous nanofiber yarns are composed of [Fe₃O₄/PANI/PAN] electrical-superparamagnetic nanofibers and [Eu(BA)₃phen/PAN] fluorescent nanofibers, realizing effectively the isolation of dark-colored Fe₃O₄ nanoparticles (NPs) and PANI from Eu(BA)₃phen complexes, and thus enhanced fluorescence intensity is acquired. The morphologies and properties of the samples were studied in detail using X-ray diffraction, scanning electron microscopy, energy-dispersive spectrometry, four-probe testing, a vibrating sample magnetometer and an F-7000 fluorescence spectrophotometer. The results reveal that the prepared heterogeneous nanofiber yarns have a large aspect ratio and uniform diameter, and that the nanofibers in the yarns demonstrate high orientation. It is found that the superparamagnetism and electrical conductivity of the heterogeneous nanofiber yarns can be respectively adjusted by modulating the amounts of Fe₃O₄ NPs and PANI in the yarns, and the highest electrical conductivity reaches up to the order of 10⁻³ S cm⁻¹. More importantly, the fluorescence intensity of the heterogeneous nanofiber yarns is much stronger than that of the counterpart [Fe₃O₄/PANI/Eu(BA)₃phen/PAN] homogeneous nanofiber yarns with the same compositions and contents owing to the fact that the Fe₃O₄ NPs and PANI were not directly mixed with the Eu(BA)₃phen complexes. Furthermore, the design concept and preparation method provide an effective route for manufacturing other multi-functional nanofiber yarns.

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Introduction

Nanofibers have been widely applied in the fields of the national defensive industry, transparent filters, optical devices, biomaterials, *etc.* owing to their large surface area (higher than that of ordinary microfibrils), high aspect ratio (length to diameter ratio) and the ability to penetrate other substances.¹⁻⁴ Nevertheless, most nanofibers are collected in the form of random nano-weaves with poor alignment.^{5,6} Randomly oriented fibrous structures limit the application of nanofibers in certain areas, such as sensors, cell directed migration and growth in which orientation of the nanofibers is strictly demanded.⁷⁻⁹ Therefore,

some researchers have shifted their focus onto preparing oriented nanofiber bundles or yarns in order to enhance the actual usage of the nanofibers for innovative applications.¹⁰ Up to now, several studies on the preparation and application of nanofiber yarns have been reported. Wu, *et al.*¹¹ fabricated an aligned conductive nanofiber yarn network within a hydrogel shell using an enhanced wet-dry electrospinning method to construct a 3D hybrid scaffold, and illustrated the great potential application in engineering anisotropic 3D cardiac scaffolds. Wu, *et al.*⁷ manufactured uniaxially aligned nanofiber yarns (UANYs) successfully by using a novel electrospinning system, and the UANYs possessed high cell survivability and displayed the ability to improve the cell adhesion and proliferation. Wang, *et al.*¹² prepared poly(caprolactone) (PCL)/silk fibroin (SF)/polyaniline (PANI) aligned nanofiber yarns *via* a developed dry-wet electrospinning method. The aligned nanofiber yarns have good biocompatibility and the ability to

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induce cellular alignment, elongation and differentiation. Xie, *et al.*¹³ demonstrated that a carbon nanofiber precursor can be prepared by electrospinning PAN into nanofiber yarns for the first time. The yarns show outstanding drawing performance, and can be drawn uniformly up to 6 times their original length without breaking.

With the development of nanotechnology, the demand for nanomaterials has been increased dramatically.^{14,15} Single functional nanomaterials are gradually unable to meet the needs of practical applications, therefore, many researchers from different countries and regions have focused their efforts on turning single functional nanomaterials into multifunctional nanomaterials in recent years.¹⁶ Considerable research effort has been devoted to nanomaterials possessing magnetic–fluorescence bifunctionality, which benefits many technical fields such as biological imaging,^{17,18} drug delivery^{19,20} and magnetic resonance.²¹ Thus, more and more researchers pay attention to fluorescent–electrical–magnetic bi- or tri-functional nanomaterials. Generally, rare earth (RE) complexes have excellent fluorescence properties due to the f–f electron transition and thus have applications in sensors, laser materials, solar energy conversion materials, *etc.*^{22,23} Fe₃O₄ NPs as magnetic materials show great potential for various fields such as magnetic-controlled switches, electronics and biological processes because of their unique superparamagnetism, good biocompatibility and high permeability.^{24,25} PANI, as a very popular conductive polymer, has been widely applied in supercapacitor electrodes, electromagnetic shielding materials, and gas sensors on account of the advantages of easy synthesis, low cost, adjustable performance, good environmental stability and tunable high conductivity.^{26–28} There is no doubt that the fluorescence intensities of RE complexes will greatly abate if Fe₃O₄ NPs, Fe₂O₃, PANI or any other dark-colored materials are directly mixed with the RE complexes for preparing multifunctional nanomaterials.^{29–31} In order to get strong fluorescence properties and to overcome the above drawback, the dark-colored materials must be effectively separated from the white-colored RE complexes. So far, some types of fluorescent–electrical–magnetic trifunctional nanomaterials have been reported, such as a [Eu(TTA)₃(TPPO)₂/polyvinyl pyrrolidone (PVP)]/[PANI/Fe₃O₄/(PAN)] nanofibrous membrane,³² [Tb(BA)₃-phen/Fe₃O₄/PANI/PVP] hollow nanofibers³³ and [polyfluorene (PF-Na)/Fe₃O₄/polyvinyl alcohol (PVA)] composite nanofibers,³⁴ which have potential applications in cell separation, magnetic resonance, biological imaging, drug targeting, *etc.*

Electrospinning is a relatively straightforward and efficient method for fabricating continuous nanofibers from polymer solutions or melts, and has become a common method used by many researchers in recent years.^{35–41} PAN is one of the most important highly polymeric materials in electrospinning owing to its low price, good thermal and chemical stability, anti-oxidation and high mechanical properties, as well as the prepared nanofibers having stable physical and chemical properties,^{42–44} and thus it was selected for this work. Herein, we designed and constructed novel [Fe₃O₄/PANI/PAN]/[Eu(BA)₃-phen/PAN] fluorescent–electrical–superparamagnetic trifunctional nanofiber yarns by using a conjugate electrospinning

method. The [Fe₃O₄/PANI/PAN]/[Eu(BA)₃-phen/PAN] nanofiber yarns were made up of two different nanofibers, namely [Fe₃O₄/PANI/PAN] nanofibers and [Eu(BA)₃-phen/PAN] nanofibers. We have named this kind of nanofiber yarn as a heterogeneous nanofiber yarn. In order to highlight the excellent performance of heterogeneous nanofiber yarns, [Fe₃O₄/PANI/Eu(BA)₃-phen/PAN] nanofiber yarns were also prepared *via* the same method, which were named as homogeneous nanofiber yarns. The homogeneous nanofiber yarns consisted of only one kind of nanofiber, namely [Fe₃O₄/PANI/Eu(BA)₃-phen/PAN] nanofibers. Finally, the prepared nanofiber yarns were systematically characterized using different modern determination techniques, and some valuable outcomes were gained. To our knowledge, there are no reports on fabricating nanofiber yarns which possess the tri-functionality of fluorescence, electrical conductivity and superparamagnetism. On account of their multifunctionality, this novel type of heterogeneous nanofiber yarn will have potential applications in nano-devices, drug targeting and imaging, electromagnetic interference shielding, *etc.* More importantly, the synthetic strategy applied in this work can be extended to prepare nanofiber yarns with other functions.

Experimental section

Chemical reagents

Eu₂O₃ (99.99%), concentrated nitric acid (HNO₃) (A.R.), 1,10-phenanthroline (phen) (A.R.), benzoic acid (BA) (A.R.), FeSO₄·7H₂O (A.R.), FeCl₃·6H₂O (A.R.), polyethylene glycol (PEG, *M_w* ≈ 20 000) (A.R.), NH₄NO₃ (A.R.), ammonia (NH₃·H₂O) (A.R.), oleic acid (OA) (A.R.), aniline (ANI) (A.R.), (1S)-(+)-camphor-10 sulfonic acid (CSA) (A.R.), ammonium persulfate (APS) (A.R.), polyacrylonitrile (PAN) (A.R.), *N,N*-dimethylformamide (DMF) (A.R.), and anhydrous ethanol (A.R.) were all bought from Aladdin Chemistry Co., Ltd.

Syntheses of OA-modified Fe₃O₄ NPs

In this work, we used a co-precipitation method to prepare Fe₃O₄ NPs with particle sizes of 8–10 nm. The detailed preparation procedures are described in the literature.⁴⁵

Syntheses of europium complexes

Eu(BA)₃-phen complexes were prepared according to the method described in the literature.⁴⁶

Fabrication of the [Fe₃O₄/PANI/PAN]/[Eu(BA)₃-phen/PAN] fluorescent–electrical–superparamagnetic trifunctional heterogeneous nanofiber yarns by conjugate electrospinning

An electrospinning solution A for fabricating [Eu(BA)₃-phen/PAN] nanofibers was prepared as follows: Eu(BA)₃-phen complexes were added into DMF (5.0000 g) and then PAN (0.5000 g) was introduced under magnetic stirring at 60 °C for 3 h to form a uniform solution. The actual components of spinning solution A are shown in Table 1. An electrospinning solution B for fabricating [Fe₃O₄/PANI/PAN] nanofibers was prepared as follows: Fe₃O₄ NPs were dispersed into DMF (7.5000 g) under ultrasonic dispersion for 20 min and then PAN



(0.9000 g) was added into the above suspension with mechanical agitation at 60 °C for 3 h. Then, ANI and CSA were dispersed into the suspension at room temperature for 3 h, which was named as solution 1. APS was dissolved into the DMF (1.5000 g) as an oxidant and stirred at room temperature for 3 h, which was denoted as solution 2. The solutions 1 and 2 were placed in a refrigerator at 0 °C for 1 h. Then, solution 1 was poured into solution 2 to mix together, and the mixture was mechanically agitated in an ice-water bath for 2–3 h. After that, the reaction system was maintained for one day at 0 °C to obtain solution B (Table 2).

The electrospinning system for fabricating the nanofiber yarns is shown in Fig. 1. Solutions A and B were loaded into two plastic syringes equipped with spinnerets, which were respectively connected to the positive and negative high-voltage direct current (DC) power supplies using copper wires. The two syringes were fixed symmetrically on both sides of the grounded copper funnel. An applied voltage was tuned to about ± 15 kV, and the distance from the top of the spinneret to the side of the copper funnel was 10 cm. The collecting device was placed below the copper funnel, and consisted of a speed-adjustable motor and a metal rod. The distance between the lower edge of the funnel and the collecting device was 18 cm. PAN-based continuous nanofiber yarns were prepared at a room temperature of 20 ± 5 °C and the relative humidity was $20 \pm 5\%$.

After the power supplies were turned on, an electrostatic field was formed between the positive and negative spinnerets. The spinning solutions A and B were ejected under the action of the electrostatic field, and then nanofibers were formed from pulling under the action of the electrostatic attraction and solidifying as the solvent was volatilized. The nanofibers produced from solutions A and B were bonded and charge neutralized after they encountered one another. A piece of grounded metal wire was placed near the central area under the copper funnel, and then a stable conical assemblage consisting of two kinds of nanofibers was gradually formed between the funnel edges and the endpoint of the metal wire. The twisted nanofiber yarns were successfully prepared by rotating the funnel and the yarns were drawn to the collecting device by the guidance of the metal wire. The revolving speed of the copper funnel and the collection rate were 100 rpm and 0.03 m s^{-1} , respectively. For convenience of comparison, the volume of spinning solution in each syringe was fixed to 2 mL during every electrospinning process, and the experiment was not stopped until the solutions were synchronously and completely consumed.

Table 1 Compositions of spinning solution A

Solution	Eu(BA) ₃ phen : PAN (wt%)	Eu(BA) ₃ phen (g)	PAN (g)	DMF (g)
S _{A1}	5	0.0250	0.5000	5.0000
S _{A2}	10	0.0500	0.5000	5.0000
S _{A3}	15	0.0750	0.5000	5.0000
S _{A4}	20	0.1000	0.5000	5.0000
S _{A5}	25	0.1250	0.5000	5.0000

Fabrication of the [Fe₃O₄/PANI/Eu(BA)₃phen/PAN] fluorescent–electrical–superparamagnetic trifunctional homogeneous nanofiber yarns by conjugate electrospinning

To highlight the advantages of the heterogeneous nanofiber yarns, [Fe₃O₄/PANI/Eu(BA)₃phen/PAN] fluorescent–electrical–superparamagnetic trifunctional homogeneous nanofiber yarns were also fabricated as a contrast sample. The preparation method was as follows: spinning solutions A3 and B3 were mixed together at a volume ratio of 1 : 1 in a conical flask under mechanical agitation for 2 h until the mixture was evenly mixed, and then electrospinning was executed under the same conditions as depicted in the above process.

Characterization methods

The as-prepared Fe₃O₄ NPs, [Fe₃O₄/PANI/PAN]/[Eu(BA)₃phen/PAN] heterogeneous nanofiber yarns and [Fe₃O₄/PANI/Eu(BA)₃phen/PAN] homogeneous nanofiber yarns were analyzed using X-ray power diffraction (XRD). The morphology of the nanofiber yarns was observed using a scanning electron microscope (SEM), which was equipped with an energy-dispersive X-ray spectrometer (EDS). A four-probe tester (RTS-4) was used to detect the electrical conduction properties of the samples. The fluorescence properties were studied using an F-7000 fluorescence spectrophotometer made by Hitachi. Then, the superparamagnetic performance of the Fe₃O₄ NPs and nanofiber yarns was investigated using a vibrating sample magnetometer (VSM), purchased from Quantum Design Inc., of MPMS SQUID XL type. All of the tests were conducted at room temperature.

Results and discussion

Phase analyses

Fig. 2 shows the XRD results of the as-prepared Fe₃O₄ NPs, [Fe₃O₄/PANI/PAN]/[Eu(BA)₃phen/PAN] heterogeneous nanofiber yarns (fabricated from S_{B3}/S_{A3}) and [Fe₃O₄/PANI/Eu(BA)₃phen/PAN] homogeneous nanofiber yarns. Compared with the PDF standard card of Fe₃O₄ (PDF#88-0866), the XRD pattern of the as-prepared Fe₃O₄ NPs corresponds with a cubic structure, and there are no characteristic peaks of other impurities such as ferric oxide (Fe₂O₃) and basic ferrous oxide (FeO(OH)). The diffraction peaks of the Fe₃O₄ NPs are found in the [Fe₃O₄/PANI/PAN]/[Eu(BA)₃phen/PAN] heterogeneous nanofiber yarns and [Fe₃O₄/PANI/Eu(BA)₃phen/PAN] homogeneous nanofiber yarns, which means that the Fe₃O₄ NPs have been introduced into the two kinds of nanofiber yarn successfully. A wide diffraction peak between 15° and 25° in the yarns manifests the presence of amorphous PANI, Eu(BA)₃phen complexes and PAN.

Morphology and structure

The SEM images, EDS spectra and diameter distribution histograms of nanofibers revealed in Fig. 3 demonstrate the detailed structure, element compositions and the diameter of the nanofibers in the [Fe₃O₄/PANI/PAN]/[Eu(BA)₃phen/PAN] heterogeneous nanofiber yarns and [Fe₃O₄/PANI/Eu(BA)₃phen/PAN] homogeneous nanofiber yarns. As revealed in Fig. 3a and e, the obvious and twisted structure of the heterogeneous



Table 2 Compositions of spinning solution B

Solution B	Fe ₃ O ₄ : PAN (mass ratio)	PANI : PAN (wt%)	Fe ₃ O ₄ (g)	PANI (g)	CSA (g)	APS (g)	DMF (g)	PAN (g)
S _{B1}	1 : 1	40	0.9000	0.3600	0.4489	0.8834	9.0000	0.9000
S _{B2}	1 : 1	50	0.9000	0.4500	0.5620	1.1036	9.0000	0.9000
S _{B3}	1 : 1	60	0.9000	0.5400	0.6740	1.3233	9.0000	0.9000
S _{B4}	1 : 1	70	0.9000	0.6300	0.7862	1.5439	9.0000	0.9000
S _{B5}	0.5 : 1	60	0.4500	0.5400	0.6740	1.3233	9.0000	0.9000
S _{B6}	2 : 1	60	1.8000	0.5400	0.6740	1.3233	9.0000	0.9000

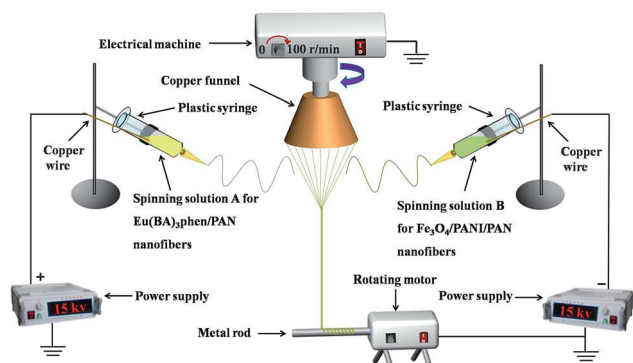
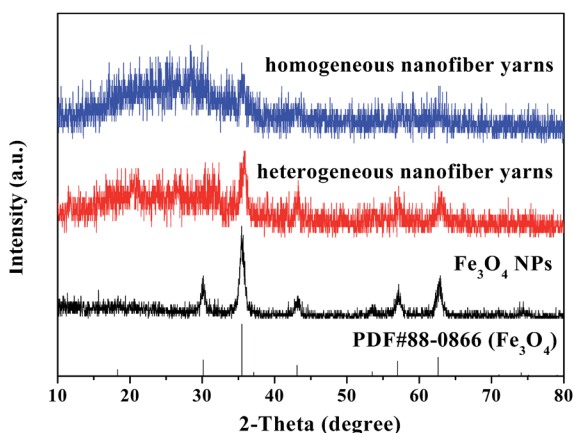


Fig. 1 Schematic diagram of the conjugate electrospinning setup with a copper funnel.

Fig. 2 XRD patterns of the Fe₃O₄ NPs, [Fe₃O₄/PANI/PAN]/[Eu(BA)₃phen/PAN] heterogeneous nanofiber yarns and [Fe₃O₄/PANI/Eu(BA)₃phen/PAN] homogeneous nanofiber yarns with the PDF standard card of Fe₃O₄.

nanofiber yarns and homogeneous nanofiber yarns can be clearly seen from the SEM images at low magnification, the nanofibers in the yarns are tightly arranged together in a certain direction. The corresponding EDS spectra shown in Fig. 3b and f reveal that the heterogeneous nanofiber yarns and homogeneous nanofiber yarns are comprised of the elements C, N, O, Fe, Eu, Pt and S. The peaks for Pt in the spectra are from the conductive film coated on the surface of the samples for SEM observation. The nanofibers in the yarns exhibit high degrees of

alignment at high magnification, as indicated in Fig. 3c and g. As seen from Fig. 3c, some aggregations of Fe₃O₄ NPs can be seen on the surface of some parts of the nanofibers, and on the contrary the other nanofibers have smooth surfaces, which can be attributed to the fact that the Fe₃O₄ NPs are only distributed in the [Fe₃O₄/PANI/PAN] nanofibers but not in the [Eu(BA)₃phen/PAN] nanofibers in the heterogeneous nanofiber yarns. As for the homogeneous nanofiber yarns, the Fe₃O₄ NPs are dispersed in every single [Fe₃O₄/PANI/Eu(BA)₃phen/PAN] nanofiber, as revealed in Fig. 3g. Fig. 3d and h demonstrate the histograms of the diameter distribution of nanofibers in the heterogeneous nanofiber yarns and homogeneous nanofiber yarns, respectively. The average diameters of the heterogeneous nanofiber yarns and their internal nanofibers are 201 μm and 508 ± 8 nm, and the mean diameters of the homogeneous nanofiber yarns and their internal nanofibers are 148 μm and 329 ± 2 nm, respectively.

Fig. 4a shows a digital photo of the conical nanofiber assemblage and the [Fe₃O₄/PANI/PAN]/[Eu(BA)₃phen/PAN] heterogeneous nanofiber yarns coiled around the metal rod. It can be seen from Fig. 4b that the heterogeneous nanofiber yarns have a large aspect ratio, and Fig. 4c displays the red light emitted from the heterogeneous nanofiber yarns under 278 nm UV illumination in darkness, which corresponds to Fig. 4b.

Fluorescence performance

A series of [Eu(BA)₃phen/PAN] nanofibers (fabricated by spinning solutions A1–5) were prepared using a traditional single-spinneret electrospinning method to obtain the optimum mass percentage of Eu(BA)₃phen complexes to PAN in this work, and the mass percentages of Eu(BA)₃phen complexes to PAN were 5%, 10%, 15%, 20% and 25%, respectively. The left side of Fig. 5 exhibits the excitation spectra of the [Eu(BA)₃phen/PAN] nanofibers at a monitoring wavelength of 616 nm. A wide excitation band range from 200 to 400 nm is observed in all samples. The peak at 278 nm ascribed to the π → π* electronic transition of ligands can be identified. As illustrated in the emission spectra on the right side of Fig. 5, under 278 nm excitation the strong characteristic peaks of Eu³⁺ appear at 580, 593, and 616 nm, which are attributed to the ⁵D₀ → ⁷F₀, ⁵D₀ → ⁷F₁, and ⁵D₀ → ⁷F₂ energy level transitions, and the ⁵D₀ → ⁷F₂ hypersensitive transition at 616 nm (red light) is the primary emission peak. As revealed in Fig. 5, with the addition of more Eu(BA)₃phen complexes, the fluorescence intensity of the nanofibers increases at first and then decreases. When the



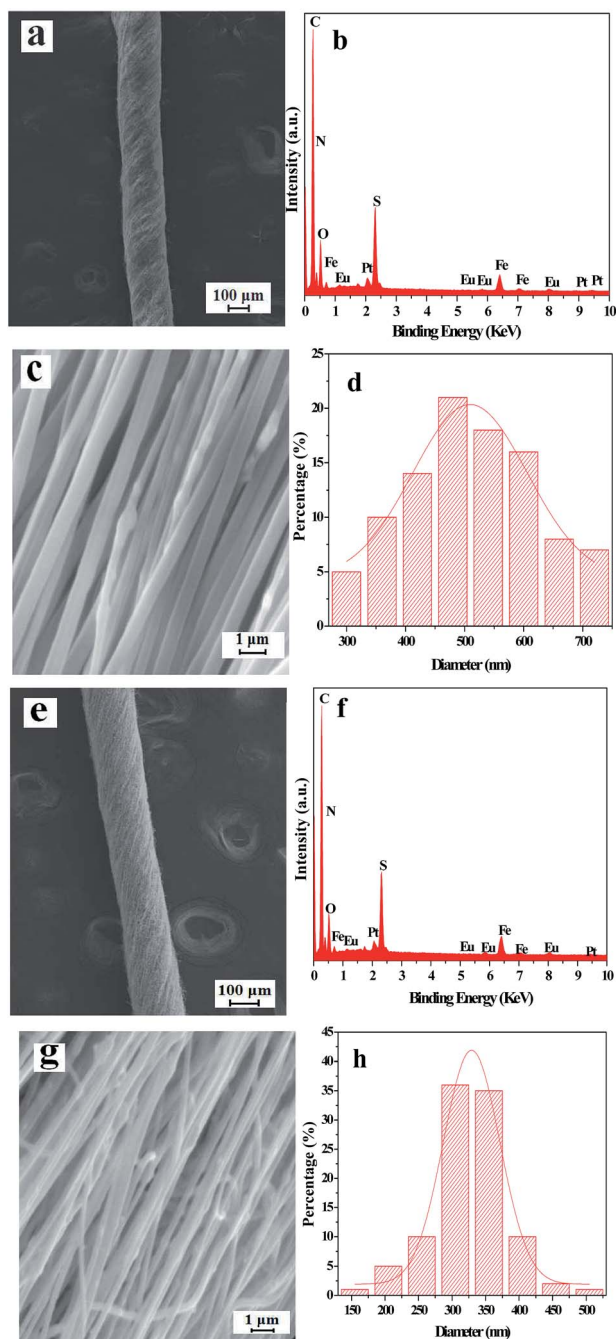


Fig. 3 SEM images at low magnification (a and e), EDS spectra (b and f), SEM images at high magnification (c and g), and diameter distribution histograms (d and h) of the nanofibers in $[\text{Fe}_3\text{O}_4/\text{PANI}/\text{PAN}]/[\text{Eu}(\text{BA})_3\text{phen}/\text{PAN}]$ heterogeneous nanofiber yarns (a–d) and $[\text{Fe}_3\text{O}_4/\text{PANI}/\text{Eu}(\text{BA})_3\text{phen}/\text{PAN}]$ homogeneous nanofiber yarns (e–h).

fluorescence intensity reaches the maximum value, the percentage of $\text{Eu}(\text{BA})_3\text{phen}$ complexes to PAN is 15%. However, as the content of $\text{Eu}(\text{BA})_3\text{phen}$ complexes continues to increase, the excess $\text{Eu}(\text{BA})_3\text{phen}$ complexes will be agglomerated and then excitonic transitions will lead to much stronger non-radiative transitions among the Eu^{3+} ions. Therefore, the fluorescence intensity of the $[\text{Eu}(\text{BA})_3\text{phen}/\text{PAN}]$ nanofibers is

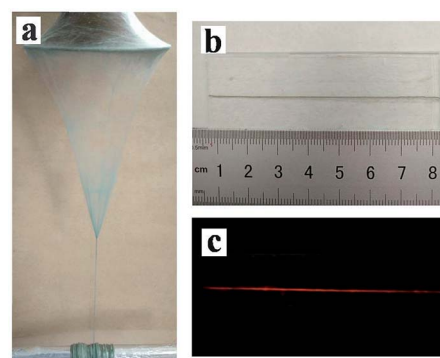


Fig. 4 Digital photos of the conical nanofiber assemblage and the $[\text{Fe}_3\text{O}_4/\text{PANI}/\text{PAN}]/[\text{Eu}(\text{BA})_3\text{phen}/\text{PAN}]$ heterogeneous nanofiber yarns coiled around the metal rod (a), heterogeneous nanofiber yarns (b) and the emission light emitted from the heterogeneous nanofiber yarns under 278 nm excitation in darkness (c).

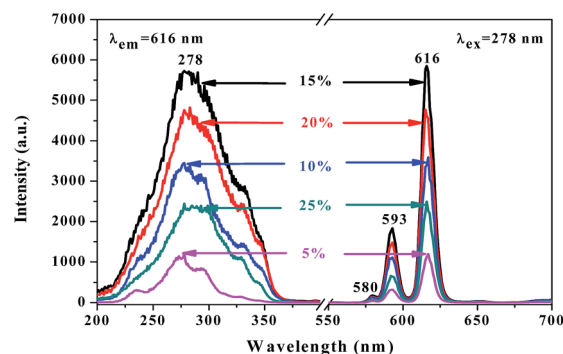


Fig. 5 Excitation spectra (left) and emission spectra (right) of $[\text{Eu}(\text{BA})_3\text{phen}/\text{PAN}]$ nanofibers doped with different amounts of $\text{Eu}(\text{BA})_3\text{phen}$ complexes to PAN.

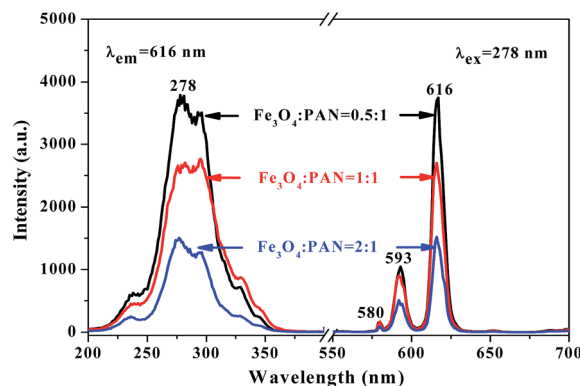


Fig. 6 Excitation spectra (left) and emission spectra (right) of $[\text{Fe}_3\text{O}_4/\text{PANI}/\text{PAN}]/[\text{Eu}(\text{BA})_3\text{phen}/\text{PAN}]$ heterogeneous nanofiber yarns containing different mass ratios of Fe_3O_4 NPs when the mass percentage of $\text{Eu}(\text{BA})_3\text{phen}$ complexes to PAN was fixed at 15% and the mass percentage of PANI to PAN was fixed at 60%.

weakened. So, the content of $\text{Eu}(\text{BA})_3\text{phen}$ complexes to PAN was fixed at 15% to prepare fluorescent–electrical–magnetic trifunctional nanofiber yarns in this study.



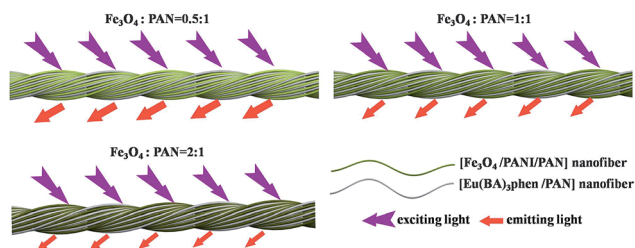


Fig. 7 Schematic diagrams of the excitation light and emission light in heterogeneous nanofiber yarns containing different mass ratios of Fe_3O_4 NPs.

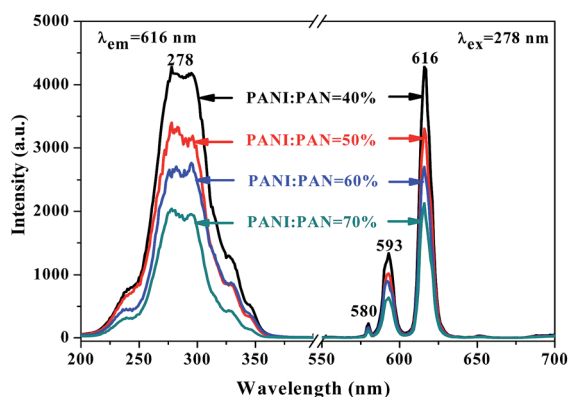


Fig. 8 Excitation spectra (left) and emission spectra (right) of $[\text{Fe}_3\text{O}_4/\text{PANI}/\text{PAN}]/[\text{Eu}(\text{BA})_3\text{phen}/\text{PAN}]$ heterogeneous nanofiber yarns containing different mass percentages of PANI when the mass percentage of $\text{Eu}(\text{BA})_3\text{phen}$ complexes to PAN was fixed at 15% and the mass ratio of Fe_3O_4 to PAN was fixed at 1 : 1.

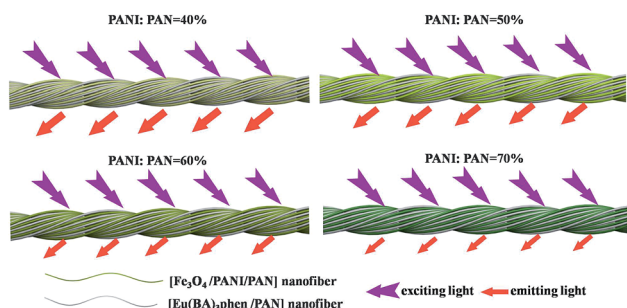


Fig. 9 Schematic diagrams of the excitation light and emission light in heterogeneous nanofibers yarns containing different mass percentages of PANI.

Meanwhile, $[\text{Fe}_3\text{O}_4/\text{PANI}/\text{PAN}]/[\text{Eu}(\text{BA})_3\text{phen}/\text{PAN}]$ heterogeneous nanofiber yarns containing different amounts of Fe_3O_4 NPs and PANI were also fabricated to study the effect of different amounts of Fe_3O_4 NPs (fabricated from $\text{S}_{\text{B}3}/\text{S}_{\text{A}3}$, $\text{S}_{\text{B}5}/\text{S}_{\text{A}3}$ and $\text{S}_{\text{B}6}/\text{S}_{\text{A}3}$, the excitation and emission spectra are illustrated in Fig. 6) and PANI (fabricated from $\text{S}_{\text{B}1}/\text{S}_{\text{A}3}$, $\text{S}_{\text{B}2}/\text{S}_{\text{A}3}$, $\text{S}_{\text{B}3}/\text{S}_{\text{A}3}$ and $\text{S}_{\text{B}4}/\text{S}_{\text{A}3}$, the excitation and emission spectra are given in Fig. 8) on the fluorescence properties. It can be seen from Fig. 6 and 8 that the excitation and emission intensities of

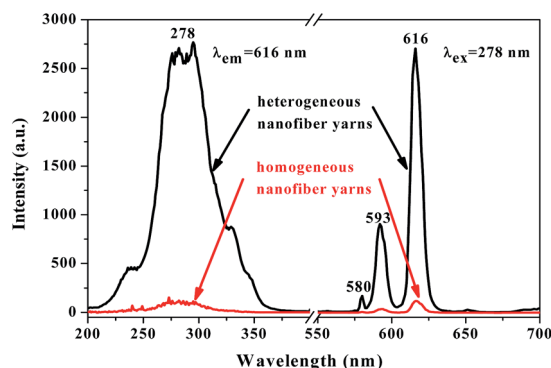


Fig. 10 Excitation spectra (left) and emission spectra (right) of $[\text{Fe}_3\text{O}_4/\text{PANI}/\text{PAN}]/[\text{Eu}(\text{BA})_3\text{phen}/\text{PAN}]$ heterogeneous nanofiber yarns and $[\text{Fe}_3\text{O}_4/\text{PANI}/\text{Eu}(\text{BA})_3\text{phen}/\text{PAN}]$ homogeneous nanofiber yarns.

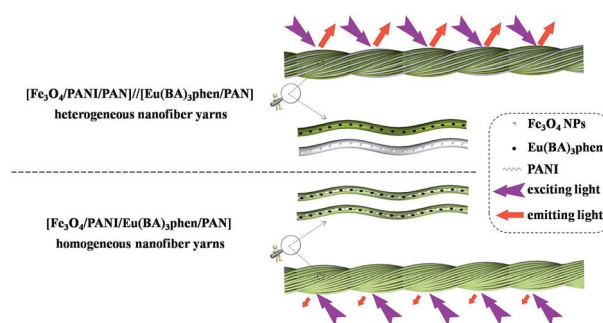


Fig. 11 Schematic diagrams of the excitation light and emission light in the $[\text{Fe}_3\text{O}_4/\text{PANI}/\text{PAN}]/[\text{Eu}(\text{BA})_3\text{phen}/\text{PAN}]$ heterogeneous nanofiber yarns and $[\text{Fe}_3\text{O}_4/\text{PANI}/\text{Eu}(\text{BA})_3\text{phen}/\text{PAN}]$ homogeneous nanofiber yarns.

the heterogeneous nanofiber yarns decrease gradually with the addition of more Fe_3O_4 NPs and PANI. This phenomenon may be caused by light absorption by the dark-colored Fe_3O_4 NPs and PANI. In order to explain the effect of different amounts of Fe_3O_4 NPs and PANI on the fluorescence properties of the heterogeneous nanofiber yarns more clearly and intuitively, schematic diagrams of the exciting light and emitting light in heterogeneous nanofiber yarns are demonstrated in Fig. 7 and Fig. 9. It is already known that absorption by Fe_3O_4 NPs in ultraviolet light (where the wavelength is less than 400 nm) is much easier than in visible light (where the wavelength ranges from 400 to 760 nm), and that PANI strongly absorbs light in the region of 400–760 nm.⁴⁷ Thus, Fe_3O_4 NPs and PANI can absorb the excitation light (281 nm) and emission light (580, 593 and 616 nm) of the heterogeneous nanofiber yarns. Furthermore, the more Fe_3O_4 NPs and PANI introduced into the yarns, the stronger the light absorption by the Fe_3O_4 NPs and PANI is, and the darker the color of the samples and the weaker the emitting light from the samples are, which can also be seen in Fig. 7 and 9.

Another interesting finding is that the fluorescence property of $[\text{Fe}_3\text{O}_4/\text{PANI}/\text{PAN}]/[\text{Eu}(\text{BA})_3\text{phen}/\text{PAN}]$ heterogeneous nanofiber yarns is much stronger (about 23 times) than that of



Table 3 Electrical conductivities of the heterogeneous nanofiber yarns with different mass ratios of PANI and Fe₃O₄ NPs, and of the [Fe₃O₄/PANI/Eu(BA)₃phen/PAN] homogeneous nanofiber yarns (contrast sample)

Samples	Variables	Conductivity (S cm ⁻¹)	
		Measurement values	Average values
S _{B1} //S _{A3}	Fe ₃ O ₄ : PAN = 1 : 1 PANI : PAN = 40%	2.65 × 10 ⁻⁴	3.65 × 10 ⁻⁴
		4.60 × 10 ⁻⁴	
		3.71 × 10 ⁻⁴	
		9.09 × 10 ⁻⁴	
S _{B2} //S _{A3}	Fe ₃ O ₄ : PAN = 1 : 1 PANI : PAN = 50%	7.31 × 10 ⁻⁴	8.17 × 10 ⁻⁴
		8.10 × 10 ⁻⁴	
		1.29 × 10 ⁻³	
		1.81 × 10 ⁻³	
S _{B3} //S _{A3}	Fe ₃ O ₄ : PAN = 1 : 1 PANI : PAN = 60%	1.40 × 10 ⁻³	1.50 × 10 ⁻³
		4.44 × 10 ⁻³	
		6.29 × 10 ⁻³	
		4.18 × 10 ⁻³	
S _{B4} //S _{A3}	Fe ₃ O ₄ : PAN = 1 : 1 PANI : PAN = 70%	2.14 × 10 ⁻³	4.97 × 10 ⁻³
		1.15 × 10 ⁻³	
		1.27 × 10 ⁻³	
		1.17 × 10 ⁻³	
S _{B5} //S _{A3}	Fe ₃ O ₄ : PAN = 0.5 : 1 PANI : PAN = 60%	1.31 × 10 ⁻³	1.52 × 10 ⁻³
		2.12 × 10 ⁻³	
		1.26 × 10 ⁻³	
		1.23 × 10 ⁻³	
Contrast sample	Fe ₃ O ₄ : PAN = 1 : 1 PANI : PAN = 60%	1.23 × 10 ⁻³	1.21 × 10 ⁻³
		1.13 × 10 ⁻³	

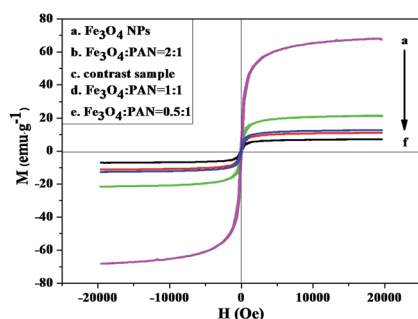


Fig. 12 Hysteresis loops of the Fe₃O₄ NPs, the heterogeneous nanofiber yarns with different mass ratios of Fe₃O₄ NPs and of the homogeneous nanofiber yarns (contrast sample).

[Fe₃O₄/PANI/Eu(BA)₃phen/PAN] homogeneous nanofiber yarns under the same compositions and content, as revealed in Fig. 10. The schematic diagrams of [Fe₃O₄/PANI/PAN]//[Eu(BA)₃phen/PAN] heterogeneous nanofiber yarns and [Fe₃O₄/PANI/Eu(BA)₃phen/PAN] homogeneous nanofiber yarns in

Fig. 11 exhibit the detailed structure of the yarns. As illustrated in the upper part of Fig. 11, heterogeneous nanofiber yarns are made up of two different nanofibers, namely the light-colored [Eu(BA)₃phen/PAN] fluorescent nanofibers and dark-colored [Fe₃O₄/PANI/PAN] electrical-superparamagnetic nanofibers. This special structure helps to isolate the Eu(BA)₃phen complexes from the Fe₃O₄ NPs and PANI effectively so that the excitation light and emission light in the [Eu(BA)₃phen/PAN] nanofibers will be hardly affected by the dark-colored Fe₃O₄ NPs and PANI. The homogeneous nanofiber yarns, as demonstrated in the lower part of Fig. 11, only contain one kind of composite nanofiber. Eu(BA)₃phen complexes, Fe₃O₄ NPs and PANI are randomly dispersed in the [Fe₃O₄/PANI/Eu(BA)₃phen/PAN] nanofibers in the homogeneous nanofiber yarns. Hence, the excitation light must pass through many Fe₃O₄ NPs and PANI to approach and excite the Eu(BA)₃phen complexes, as such part of the excitation light is absorbed by Fe₃O₄ NPs and PANI in the process, so the excitation light is weakened to a great extent before reaching the Eu(BA)₃phen complexes. Likewise, the emission light emitted by the Eu(BA)₃phen complexes also must pass through the Fe₃O₄ NPs and PANI, and

Table 4 Saturation magnetizations of the Fe₃O₄ NPs, the heterogeneous nanofiber yarns with different mass ratios of Fe₃O₄ NPs and of the homogeneous nanofiber yarns (contrast sample)

Samples	Saturation magnetization (M _s)/(emu g ⁻¹)	
Heterogeneous nanofiber yarns	Fe ₃ O ₄ NPs	67.37
	Fe ₃ O ₄ : PAN = 0.5 : 1	7.11
	Fe ₃ O ₄ : PAN = 1 : 1	11.08
	Fe ₃ O ₄ : PAN = 2 : 1	21.28
Homogeneous nanofiber yarns	Contrast sample	12.60



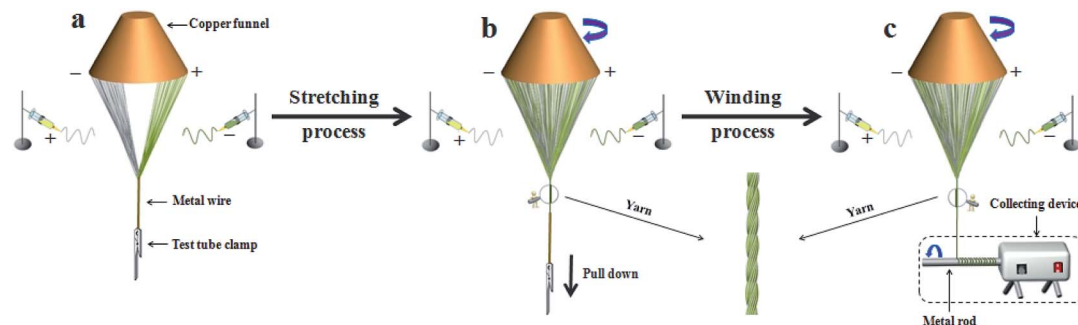


Fig. 13 Schematic diagram for the formation mechanism of heterogeneous nanofiber yarns.

is absorbed by them. As a result, both the excitation light and emission light are severely weakened. According to the above analysis, heterogeneous nanofiber yarns possess superior fluorescence performance compared to the homogeneous nanofiber yarns.

Electrical conductivity analyses

In order to study the electrical properties of $[\text{Fe}_3\text{O}_4/\text{PANI}/\text{PAN}]/[\text{Eu}(\text{BA})_3\text{phen}/\text{PAN}]$ heterogeneous nanofiber yarns and $[\text{Fe}_3\text{O}_4/\text{PANI}/\text{Eu}(\text{BA})_3\text{phen}/\text{PAN}]$ homogeneous nanofiber yarns, a series of nanofiber yarns with lengths of 2.0 cm were prepared. All samples were tested three times and the average values were used as final results, the testing results are summarized in Table 3. As can be seen from Table 3, the average conductivity values of the heterogeneous nanofiber yarns increase greatly from 3.65×10^{-4} to $4.97 \times 10^{-3} \text{ S cm}^{-1}$ when the mass percentages of PANI to PAN are changed from 40% to 70% under the condition that the mass percentage of $\text{Eu}(\text{BA})_3\text{phen}$ to PAN and mass ratio of Fe_3O_4 NPs to PAN are set to 15% and 1 : 1, respectively. This result can be explained by the fact that the charge transport ability provided by the connected network of PANI is improved by adding more PANI. It also means that the electrical conductivity properties of the heterogeneous nanofiber yarns can be adjusted by changing the content of PANI. In addition, to investigate the influence of Fe_3O_4 NPs on the electrical conductivity, the conductivity values of the heterogeneous nanofiber yarns with different quantities of Fe_3O_4 NPs are also measured when the other parameters are fixed (the mass percentages of $\text{Eu}(\text{BA})_3\text{phen}$ to PAN and PANI to PAN were set as 15% and 60%, respectively). The results in Table 3 indicate that the presence of Fe_3O_4 NPs has little effect on the electrical conductivity of the samples. The conductivity of the heterogeneous nanofiber yarns ($S_{\text{B3}}/S_{\text{A3}}$) is slightly higher than that of the counterpart homogeneous nanofiber yarns for the reason that the insulative $\text{Eu}(\text{BA})_3\text{phen}$ complexes are dispersed in all of the homogeneous nanofiber yarns, which hinders the formation of the continuous conductive network to a certain extent.

Magnetic properties

The typical hysteresis loops of as-prepared Fe_3O_4 NPs, $[\text{Fe}_3\text{O}_4/\text{PANI}/\text{PAN}]/[\text{Eu}(\text{BA})_3\text{phen}/\text{PAN}]$ heterogeneous nanofiber yarns

with different mass ratios of Fe_3O_4 NPs and $[\text{Fe}_3\text{O}_4/\text{PANI}/\text{Eu}(\text{BA})_3\text{phen}/\text{PAN}]$ homogeneous nanofiber yarns are indicated in Fig. 12, and their saturation magnetizations are revealed in Table 4. It is known that the saturation magnetization of a magnetic composite material is determined by the content of the magnetic substance doped in them. It can be seen from Table 4 that the saturation magnetization of the as-prepared Fe_3O_4 NPs reaches 67.37 emu g^{-1} , which is relatively lower than its bulk value of 92 emu g^{-1} . This phenomenon comes from the nanosized effect, where the surface layer has a number of defects for the fine magnetic particles, and thus does not contribute to the magnetization.⁴⁸ Additionally, with the increasing content of Fe_3O_4 NPs in the nanofibers, the saturation magnetization of the yarns is also increased, and is enhanced from 7.11 to 21.28 emu g^{-1} , implying that the heterogeneous nanofiber yarns possess tunable superparamagnetism. The heterogeneous nanofiber yarns have a superparamagnetic property close to that of the homogeneous nanofiber yarns, which is because they have the same theoretical Fe_3O_4 NP mass. Compared with Fe_3O_4 NPs, the loss of saturation magnetization of the yarns is due to the existence of the non-magnetic materials such as PAN, PANI and $\text{Eu}(\text{BA})_3\text{phen}$ complexes.

Formation mechanism for the nanofiber yarns

A formation mechanism for the heterogeneous nanofiber yarns is presented in Fig. 13. After the positive and negative high voltage are applied, the spinning solutions in the positive and negative spinnerets are synchronously ejected towards their opposites under the action of the electrostatic field, in which process the jets swing due to the instability of electrospinning. Meanwhile, an electrostatic induction phenomenon occurs in the copper funnel under the electrostatic field. The positively charged spinneret induces a negative charge on the edge of the funnel near the spinneret, whereas the negatively charged spinneret induces a positive charge on the edge of funnel near this spinneret,^{7,49} as revealed in Fig. 13a. In this case, the jets tend to be attracted to the edges of the copper funnel and the grounded metal wire so that the electric charges in the jets can be released. Thus at the beginning of the experiment, ordered nanofibers are formed between the edges of the copper funnel and the endpoint of the metal wire by electrostatic attraction of



the charged copper funnel and the grounded metal wire as well as the swing of the jets, as displayed in Fig. 13a. It can be seen from Fig. 13b that a hollow-cone shaped nanofiber bundle is formed by rotating the funnel. Meanwhile, a twisted nanofiber yarn is generated between the tip of the nanofiber bundle and the endpoint of the metal wire. On slowly pulling down the metal wire, the length of the nanofiber yarn becomes longer. Once the nanofiber yarn has sufficient length, the nanofiber yarn is wrapped around the metal rod of the collecting device to realize automatic and continuous collection of nanofiber yarns, as depicted in Fig. 13c. Electrostatic induction always exists due to the existence of the electrostatic field throughout the experiment process. Therefore, the edge of the funnel opposite to the positively charged spinneret is always negatively charged and the edge of the funnel opposite to the negatively charged spinneret is always positively charged, as can be seen in Fig. 13.

Conclusions

In summary, $[\text{Fe}_3\text{O}_4/\text{PANI}/\text{PAN}]/[\text{Eu}(\text{BA})_3\text{phen}/\text{PAN}]$ fluorescent–electrical–superparamagnetic trifunctional heterogeneous nanofiber yarns were successfully prepared for the first time by using a conjugate electrospinning method. The average diameters of the heterogeneous nanofiber yarns and their internal nanofibers are $201\ \mu\text{m}$ and $508 \pm 8\ \text{nm}$, respectively. The heterogeneous nanofiber yarns have an obviously twisted structure and the nanofibers in the yarns exhibit high alignment and orientation. The heterogeneous nanofiber yarns contain two different types of nanofiber, namely $[\text{Fe}_3\text{O}_4/\text{PANI}/\text{PAN}]$ electrical–superparamagnetic nanofibers and $[\text{Eu}(\text{BA})_3\text{phen}/\text{PAN}]$ fluorescent nanofibers. Compared with $[\text{Fe}_3\text{O}_4/\text{PANI}/\text{Eu}(\text{BA})_3\text{phen}/\text{PAN}]$ homogeneous nanofiber yarns, the heterogeneous nanofiber yarns have enhanced fluorescence owing to their peculiar structure, achieving the efficient separation of white-colored $\text{Eu}(\text{BA})_3\text{phen}$ complexes from dark-colored Fe_3O_4 NPs and PANI. Besides, the superparamagnetism and electrical conductivity of the heterogeneous nanofiber yarns can be respectively tuned by changing the content of Fe_3O_4 NPs and PANI. The heterogeneous nanofiber yarns with unique properties are expected to be applied in the fields of electromagnetic shielding, gas sensors, drug targeting, flexible nano-devices, etc.

Conflicts of interest

There are no conflicts of interest to declare.

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References

- 1 A. Balaji, M. V. Vellayappan, A. A. John, A. P. Subramanian, S. K. Jaganathan, E. Supriyanto and S. I. A. Razak, *RSC Adv.*, 2015, **5**, 57984–58004.
- 2 B. Khalid, X. Bai, H. Wei, Y. Huang, H. Wu and Y. Cui, *Nano Lett.*, 2017, **17**, 1140–1148.
- 3 A. Edwards, D. Jarvis, T. Hopkins, S. Pixley and N. Bhattarai, *J. Biomed. Mater. Res., Part B*, 2015, **103**, 21–30.
- 4 J. H. Lee, D. W. Shin, K. B. Nam, Y. H. Gim, H. S. Ko, D. K. Seo, G. Hui lee, Y. H. Kim, S. W. Kim, T. S. Oh and J. B. Yoo, *Polymer*, 2016, **84**, 52–58.
- 5 F. Ko, Y. Gogotsi, A. Ali, N. Naguib, H. Ye, G. L. Yang, C. Li and P. Willis, *Adv. Mater.*, 2003, **15**, 1161–1165.
- 6 H. Okuzaki, T. Takahashi, N. Miyajima, Y. Suzuki and T. Kuwabara, *Macromolecules*, 2006, **39**, 4276–4278.
- 7 S. H. Wu, B. Duan, P. H. Liu, C. D. Zhang, X. H. Qin and J. T. Butcher, *ACS Appl. Mater. Interfaces*, 2016, **8**, 16950–16960.
- 8 M. Abbasipour and R. Khajavi, *Adv. Polym. Technol.*, 2013, **32**, 21363.
- 9 P. H. Liu, S. H. Wu, Y. Zhang, H. N. Zhang and X. H. Qin, *Nanomaterials*, 2016, **6**, 121.
- 10 X. R. Li, M. Y. Li, J. Sun, Y. Zhuang, J. J. Shi, D. W. Guan, Y. Y. Chen and J. W. Dai, *Small*, 2016, **12**, 5009–5018.
- 11 Y. B. Wu, L. Wang, B. L. Guo and P. X. Ma, *ACS Nano*, 2017, **11**, 5646–5659.
- 12 L. Wang, Y. B. Wu, B. L. Guo and P. X. Ma, *ACS Nano*, 2015, **9**, 9167–9179.
- 13 Z. G. Xie, H. T. Niu and T. Lin, *RSC Adv.*, 2015, **5**, 15147–15153.
- 14 Y. Lu and S. Ozcan, *Nano Today*, 2015, **10**, 417–420.
- 15 C. M. Park, K. H. Chu, J. Heo, N. Her, M. Jang, A. Son and Y. Yoon, *J. Hazard. Mater.*, 2016, **309**, 133–150.
- 16 A. M. El-Toni, M. A. Habila, J. P. Labis, Z. A. Allothman, M. Alhoshan, A. A. Elzatahry and F. Zhang, *Nanoscale*, 2016, **8**, 2510–2531.
- 17 H. Y. Chen, D. C. Colvin, B. Qi, T. Moore, J. He, O. T. Mefford, F. Alexis, J. C. Gore and J. N. Anker, *J. Mater. Chem.*, 2012, **22**, 12802–12809.
- 18 F. Chen, M. Chen, C. Yang, J. Liu, N. Q. Luo, G. W. Yang, D. H. Chen and L. Li, *Phys. Chem. Chem. Phys.*, 2015, **17**, 1189–1196.
- 19 L. L. Li, C. Dong, Y. Q. Zhang, Z. T. Deng, X. L. Ren, X. W. Meng, F. Q. Tang, J. Ren and L. Zhang, *Nanotechnology*, 2007, **18**, 405102.
- 20 L. Li, C. Liu, L. Y. Zhang, T. T. Wang, H. Yu, C. G. Wang and Z. M. Su, *Nanoscale*, 2013, **5**, 2249–2253.
- 21 J. W. Lan, J. Y. Chen, N. X. Li, X. H. Ji, M. X. Yu and Z. K. He, *Talanta*, 2016, **151**, 126–131.
- 22 H. L. Gao, L. Jiang, W. M. Wang, S. Y. Wang, H. X. Zhang and J. Z. Cui, *Inorg. Chem.*, 2016, **55**, 8898–8904.
- 23 B. Q. Shao, Q. Zhao, W. Z. Lv, M. M. Jiao, W. Lu and H. P. You, *J. Mater. Chem. C*, 2015, **3**, 1091–1098.



- 24 T. T. Li, H. W. Liu, Z. N. Wu, Y. Liu, Z. X. Guo and H. Zhang, *Nanoscale*, 2016, **8**, 11792–11796.
- 25 B. Shen, W. T. Zhai, M. M. Tao, J. Q. Ling and W. G. Zheng, *ACS Appl. Mater. Interfaces*, 2013, **5**, 11383–11391.
- 26 H. G. Wei, X. R. Yan, S. J. Wu, Z. P. Luo, S. Y. Wei and Z. H. Guo, *J. Phys. Chem. C*, 2012, **116**, 25052–25064.
- 27 N. German, A. Popov, A. Ramanaviciene and A. Ramanavicius, *Polymer*, 2017, **115**, 211–216.
- 28 S. K. Simotwo, C. DelRe and V. Kalra, *ACS Appl. Mater. Interfaces*, 2016, **8**, 21261–21269.
- 29 X. Y. Xu and B. Yan, *ACS Appl. Mater. Interfaces*, 2015, **7**, 721–729.
- 30 H. Shao, Q. L. Ma, X. T. Dong, W. S. Yu, M. Yang, Y. Yang, J. X. Wang and G. X. Liu, *Phys. Chem. Chem. Phys.*, 2015, **17**, 21845–21855.
- 31 X. Xi, J. X. Wang, X. T. Dong, Q. L. Ma, W. S. Yu and G. X. Liu, *Chem. Eng. J.*, 2014, **254**, 259–267.
- 32 Z. J. Wang, Q. L. Ma, X. T. Dong, D. Li, X. Xi, W. S. Yu, J. X. Wang and G. X. Liu, *ACS Appl. Mater. Interfaces*, 2016, **8**, 26226–26234.
- 33 Y. W. Liu, Q. L. Ma, M. Yang, X. T. Dong, Y. Yang, J. X. Wang, W. S. Yu and G. X. Liu, *Chem. Eng. J.*, 2016, **284**, 831–840.
- 34 D. C. Guo, Z. Y. Sun, L. D. Xu, Y. Gao, M. M. Dai, S. H. Wang, Q. Chang, C. Wang and D. G. Ma, *Mater. Lett.*, 2015, **159**, 159–162.
- 35 F. Bi, X. T. Dong, J. X. Wang and G. X. Liu, *New J. Chem.*, 2015, **39**, 3444–3451.
- 36 N. Lv, J. L. Zhang, G. M. Li, X. Wang and J. Z. Ni, *J. Phys. Chem. C*, 2017, **121**, 11926–11931.
- 37 T. L. Chen and Y. A. Elabd, *Electrochim. Acta*, 2017, **229**, 65–72.
- 38 F. Bi, X. T. Dong, J. X. Wang and G. X. Liu, *ChemPlusChem*, 2014, **79**, 1713–1719.
- 39 D. D. Yin, Q. L. Ma, X. T. Dong, N. Lv, J. X. Wang, W. S. Yu and G. X. Liu, *ChemPlusChem*, 2015, **80**, 568–575.
- 40 X. Xi, Q. L. Ma, X. T. Dong, J. X. Wang, W. S. Yu and G. X. Liu, *IEEE Trans. Nanotechnol.*, 2015, **14**, 243–249.
- 41 L. Han, Q. L. Ma and X. T. Dong, *RSC Adv.*, 2015, **5**, 95674–95681.
- 42 F. J. Miao, C. L. Shao, X. H. Li, N. Lu, K. X. Wang, X. Zhang and Y. C. Liu, *Electrochim. Acta*, 2015, **176**, 293–300.
- 43 R. Zhao, X. Li, B. Sun, H. Ji and C. Wang, *J. Colloid Interface Sci.*, 2017, **487**, 297–309.
- 44 Q. J. Niu, J. X. Guo, B. L. Chen, J. Nie, X. D. Guo and G. P. Ma, *Carbon*, 2017, **114**, 250–260.
- 45 Q. L. Ma, J. X. Wang, X. T. Dong, W. S. Yu and G. X. Liu, *Chem. Eng. J.*, 2015, **260**, 222–230.
- 46 Q. L. Ma, W. S. Yu, X. T. Dong, J. X. Wang, G. X. Liu and J. Xu, *J. Nanopart. Res.*, 2012, **14**, 1203.
- 47 J. Tian, Q. L. Ma, X. T. Dong, W. S. Yu, M. Yang, Y. Yang, J. X. Wang and G. X. Liu, *RSC Adv.*, 2016, **6**, 36180–36191.
- 48 D. H. Han, J. P. Wang and H. L. Luo, *J. Magn. Magn. Mater.*, 1994, **136**, 176–182.
- 49 J. X. He, K. Qi, Y. M. Zhou and S. Z. Cui, *J. Appl. Polym. Sci.*, 2014, **131**, 40137.

