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# How side-chain substituents and substrates influence mechanochromic luminescence: case study with pyrene†

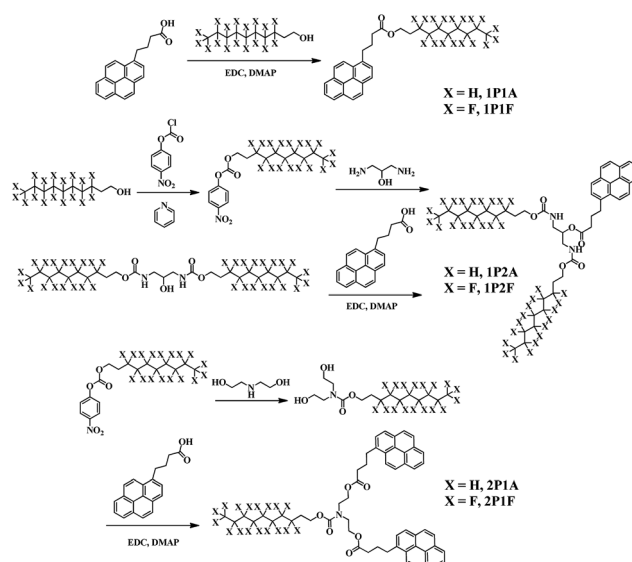
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Many organic molecules exhibit reversible, force-induced emission change known as mechanochromic luminescence (ML) and can potentially be used as mechanosensors. Here we provide an example on how ML behaviours can be modulated by side-chain substituents which serve as the directing group, as well as substrates which may disrupt the function of the directing group.

The photoluminescence (PL) properties of molecular solids, such as emission wavelength, lifetime, quantum yield and coherence, show critical reliance on intermolecular arrangements.<sup>1–4</sup> For a crystalline solid with a given fashion of molecular packing, a corresponding, dominant solid-state PL arises owing to delocalized excitation unique to the specific arrangement.<sup>5–7</sup> After mechanical stimuli, various excitonic splittings which result in new emission bands and excited-state energy transfers leading to amplification of low-energy emission can conspicuously alter the PL properties of the solids.<sup>8–10</sup> The phenomenon is usually referred to as mechanochromic luminescence (ML).<sup>11–14</sup> In the past decade, numerous examples of ML dyes have been reported and mechanisms carefully studied.<sup>15–22</sup> Despite the success in identifying increasingly diverse molecules with such properties in the solid state, a pressing issue remains as how to put these ML dyes into practical application. The challenge facing ML application has two major aspects: (1) the ML behaviours of molecular solids, usually dictated by the substituent groups, are unpredictable, indicating that there is a lack of basis in design rationale; (2) the substrate effects, which can significantly alter the ML properties, are largely unknown while the real application requires that sensing be done on a specific substrate. As a result, ML is still in the theoretical phase and confined in the laboratory.

In this communication, we present the first case study in the ML field, employing one of the most studied dyes, pyrene (Fig. 1), to provide some preliminary clues for the design principles in terms of the effect of substituents and substrates. The rationale for the molecules used in the study is as follows: (1)

the conditions for producing monomer and excimer emissions, respectively, are very well understood,<sup>23–27</sup> which allows one to easily correlate fluorescence with intermolecular arrangements of the pyrene derivatives; (2) three types pyrene dyes (linear, **1P1F**, wedge-shaped, **1P2F** and tweezers-shaped, **2P1F**) are synthesized and we predict that only the linear structure has a high probability of exhibiting ML because the tweezers are bound to form intramolecular excimers while the pyrenes are likely to be pushed together in the wedge-shaped molecules to form intermolecular excimers; (3) non-fluorinated and fluorinated pyrene derivatives were synthesized to interact with different substrates and we predict that the ML recovery is likely to be hindered on substrates with strong interactions with the dye molecules.



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Fig. 1 Syntheses and chemical structures of alkyl- and fluoro-substituted pyrene derivatives.



The syntheses for these pyrene derivatives are shown in Fig. 1 and the products are verified with  $^1\text{H-NMR}$  and mass spectroscopy, respectively (ESI $^\dagger$ ). The physical properties of the non-fluorinated and fluorinated pyrene dyes were compared: the most dramatic difference is that all three alkyl derivatives (**1P1A**, **2P1A** and **1P2A**) are transparent liquid at room temperature while the fluorinated counterparts (**1P1F**, **2P1F** and **1P2F**) are crystals with **2P1F** exhibiting a crystal-liquid borderline situation at room temperature. This is not surprising since the interactions among alkyl chains and/or pyrenes are rather weak in the ground state. On the other hand, the attraction forces between fluorinated carbon chains are considerably stronger.

When dissolved in solutions, all six pyrene derivatives exhibit identical absorption spectra characteristic of pyrene (Fig. S1 $^\dagger$ ). The steady-state emission spectra, however, are different depending on their molecular architectures (Fig. 2). The four linear and wedge-shaped molecules with a single pyrene show strong monomer fluorescence ( $\lambda_{\text{em}} = 376$  nm) with slight variation in the vibrational progression known to be affected by polarity difference. The tweezers-like **2P1A** and **2P1F** show predominant excimer fluorescence ( $\lambda_{\text{em}} = 483$  nm) due to the close proximity of the two pyrene moieties. In the solid state, the only dye that exhibits monomer-like fluorescence ( $\lambda_{\text{em}} = 404$  nm,  $\tau_{\text{FL}} = 0.14$  ns and  $\Phi_{\text{FL}} = 0.039$ ) is the linear, fluorinated **1P1F** molecule, while the other five are dominated by excimer fluorescence ( $\lambda_{\text{em}} = 471$ – $486$  nm,  $\tau_{\text{FL}} = 18.69$ – $46.46$  ns, Table 1). At this stage, it is rather safe to presume that **1P1F** is the only molecule with ML properties. We have previously elucidated that the main mechanism for ML involves two steps, *i.e.*, generation of low-energy emitting centres and exciton migration from higher-energy emitting centres.<sup>9</sup> In the case of pyrene derivatives, the lowest emitting trap is the excimeric ( $\lambda_{\text{em}} \sim 488$  nm) configurations where the two pyrene molecules stack on top of each other.<sup>28</sup> Consequently, **1P1F** is the only molecule that does not adopt the excimer geometry in the solid state. For the alkyl-substituted pyrenes, their liquid nature at room temperature lends mobility and thus the key ingredient for

Table 1 Fluorescence properties of all six pyrene derivatives in the solid state

	$\lambda_{\text{FL}}^a$ (nm)	$\tau_{\text{FL}}^b$ (ns)/%
<b>1P1A</b>	488	0.18/36.47
		39.5/63.53
<b>1P1F</b>	404	0.14/100
<b>1P2A</b>	484	0.21/44.18
		33.31/55.82
<b>1P2F</b>	474	39.53/42.24
		73.07/53.37
		5.91/4.37
<b>2P1A</b>	484	4.86/10.55
<b>2P1F</b>	486	42.21/89.45
		13.67/11.02
		50.52/88.98

<sup>a</sup> Steady-state fluorescence emission maximum ( $\lambda_{\text{ex}} = 370$  nm).

<sup>b</sup> Fluorescence emission lifetime components and weights.

excimer formation; while the two fluorinated counterparts (**1P2F**,  $\lambda_{\text{em}} = 474$  nm,  $\Phi_{\text{FL}} = 0.071$  and **2P1F**,  $\lambda_{\text{em}} = 486$  nm) form pre-arranged excimers.<sup>29,30</sup>

Fig. 3a shows the single-crystal structure and molecular packing for **1P1F**. As expected, no excimeric configurations are found in the crystal; instead, the pyrene rings loosely stack and are offset from the ring centres by the equivalent of more than one benzene ring. Consequently, the overlapping area of adjacent rings is less than 50% the size of pyrene. Incidentally, from the space-fill model in Fig. 3a inset, it is clear that the main directing force for packing stems from the strong interactions among fluorinated alkyl chains,<sup>31,32</sup> which are in short contacts, instead of pyrene stacking. Fig. 3b shows that the deep-blue fluorescence of **1P1F** crystalline solids can be altered to emit in the green region typical of pyrene excimer when smeared by a cotton swab on a piece of weighing paper under UV light. The corresponding steady-state emission spectra are provided in Fig. 3c, where the emission maximum of the crystal is red-shifted from 404 to 484 nm after smearing. The measured fluorescence lifetime is also increased from 0.18 ns to 51 ns, indicating a typical excimer emission after the application of mechanical force. The phenomenon has been well-investigated in many other systems,<sup>1–22,33</sup> in which force induces a crystal-line-to-amorphous phase transition and the increased molecular rotational freedom allows for excimer/dimer formation. The observed ML is likely to be a competitive effect between pyrene packing and F–F interactions in fluorinated carbon chains, where a similar mechanism was previously reported from the Weder group.<sup>34,35</sup>

Fig. 3d–f manifests the reversible ML properties of **1P1F** solid on weighing paper under UV light ( $\lambda_{\text{ex}} = 365$  nm). After thoroughly distributing **1P1F** on weighing paper (1.5 mg on  $5 \times 5$  cm<sup>2</sup>) *via* smearing with a latex glove, the paper is briefly thermally annealed ( $\sim 10$  s) in an oven pre-heated at 80 °C. A light touch of a cotton swab can induce the fluorescence switch from blue ( $\lambda_{\text{em}} = 462$  nm) to green-blue ( $\lambda_{\text{em}} = 484$  nm), visually. The green-blue emission slowly reverts (15–20 min) back to a more blue-shifted emission overtime at room temperature or

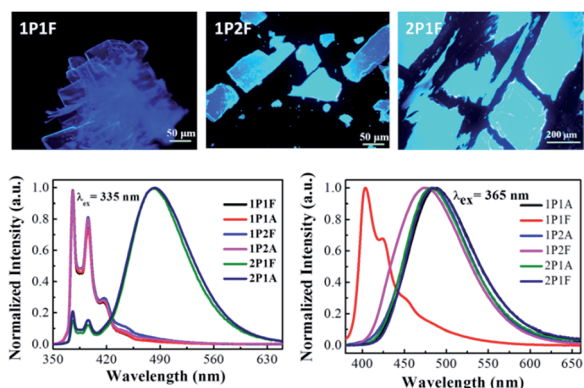


Fig. 2 Top: fluorescence microscopy images showing solid-state morphologies of fluorinated pyrene derivatives **1P1F**, **1P2F** and **2P1F**. Bottom: steady-state fluorescence emission spectra of all six pyrene dyes in dichloromethane solution (left) and in the solid-state (right) at room temperature.



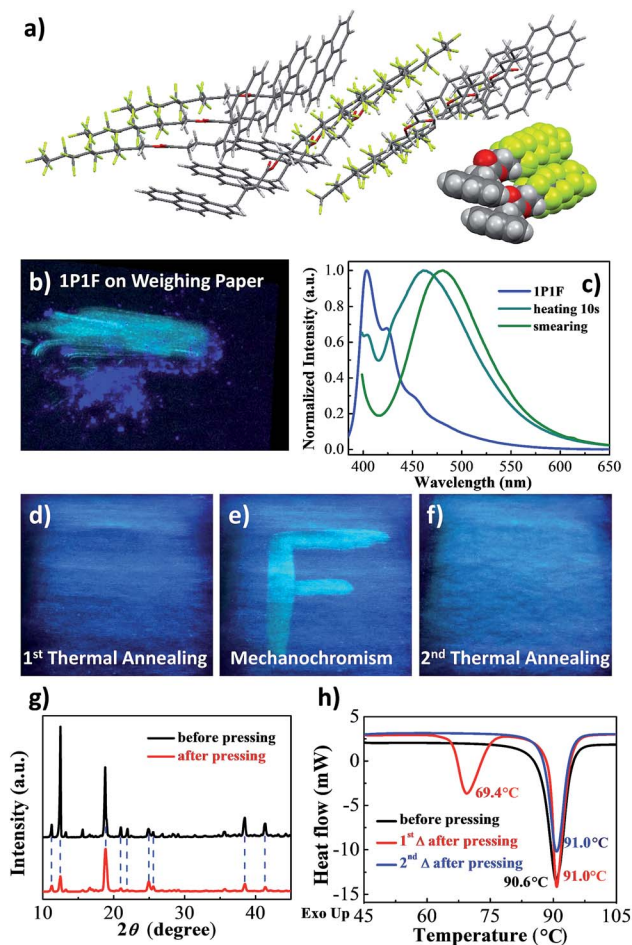


Fig. 3 (a) Structure and molecular packing of **1P1F** determined by single-crystal X-ray diffraction, inset: space-fill model showing two adjacent **1P1F** molecules with strong interactions between fluorinated alkyl chains. (b) Photo showing mechanochromic luminescence of **1P1F** on weighing paper and (c) their corresponding fluorescence emission spectra. The reversible ML behaviour of **1P1F** solid film is illustrated in (d–f). (g) X-ray diffraction patterns of **1P1F** crystalline powders before (black line) and after (red line) pressing with a cotton swab. (h) Differential scanning calorimetry (DSC,  $20\text{ }^{\circ}\text{C min}^{-1}$ ) heating traces for **1P1F** crystals before pressing (black line) and after pressing (the red line represents the 1<sup>st</sup> heating trace and the blue line represent the 2<sup>nd</sup> heating trace after cooling).

immediately when heated at  $80\text{ }^{\circ}\text{C}$ . These observations are consistent with previously reported pyrene<sup>36–38</sup> and  $\text{BF}_2\text{dbm}$ <sup>39,40</sup> systems, that smearing-caused amorphous phase generates low-energy emitting traps, through which high-energy excitons migrate and emit.<sup>9</sup> Thermal annealing leads to molecular rearrangement partially to the original packing. However, since certain exciton traps cannot be entirely removed *via* simple thermal treatment, exciton migration is still prevalent and efficient, which explains the difference in fluorescence between the crystalline solids and recovered solid film. Examination of the smeared or recovered solid films at low temperature can suppress such migration and reveal the main emission species.<sup>9</sup>

To examine the proposed mechanism that has been seen in the studies from Nakano<sup>41</sup> and Fraser.<sup>42</sup> We performed X-ray

diffraction (XRD) measurements for **1P1F** crystalline solids before and after mechanical pressing. The results are presented in Fig. 3g and Table 2. Although the fluorescence emission has been largely altered from monomer-like to excimer-like, the peaks in the XRD pattern do not disappear for the pressed samples. Instead, we observed substantial intensity decrease for peaks at a smaller  $2\theta$  angle ( $11.2^{\circ}$ ) and broadening for all the peaks (Table 2). This observation suggests that while the majority of the solid remains in the crystalline phase, an observable portion of the ordered molecular packing have been destroyed to become amorphous, which is again consistent with our previous conclusion that energy migration from ordered crystalline region to amorphous, low-energy emitting traps can be responsible for the ML phenomenon at room temperature.<sup>9</sup> The DSC traces from Fig. 3g also corroborate the assertion, where one additional exothermic peak at  $69.4\text{ }^{\circ}\text{C}$  could be recorded for the freshly pressed sample of **1P1F**, a sign of existence of force-induced defects in substantial quantities, apart from the melting peak at  $91\text{ }^{\circ}\text{C}$ . Thus far, we have been able to demonstrate that linear substitution has a higher probability for the pyrene dye to adopt non-excimer configurations.

Finally, we show that the choice of substrate can have a tremendous effect in the ML behaviours of the dye. Fig. 4 shows the ML properties of **1P1F** on polytetrafluoroethylene (PTFE) and polyethylene (PE) substrates, respectively. Since **1P1F** is heavily fluorinated, it is expected that the interaction between **1P1F** and PTFE can disrupt the ordering or directing function of the fluorinated alkyl chains so that the randomly distributed pyrene moieties has a strong proclivity to the formation of excimer. As can be seen from the left panel of Fig. 4, indeed, no ML has been observed on PTFE. That cooling the solid film in liquid nitrogen does not alter the fluorescence behaviour indicates lack of ordered region, or intermolecular interactions, in the solid state. On the other hand, when a PE substrate was used, thermally reversible ML could once again be noted (Fig. 4, right panel). Similar to the observation on weighing paper, the fluorescence emission maxima of **1P1F** switch in between  $462\text{ nm}$  and  $484\text{ nm}$ . For the smeared region, lowering temperature reveals that most excitons formed prior to migration are in fact in the blue region, which suggests that the fluorinated alkyl chains still function as the directing group for ML.

Table 2 Half width of peaks for X-ray diffraction patterns from crystalline **1P1F**

$2\theta_{\text{calc}}$ (degree)	Pristine sample	Pressed sample
11.26	0.24	0.29
12.46	0.23	0.30
18.84	0.28	0.39
20.99	0.25	0.38
21.92	0.24	0.30
24.90	0.38	0.41
25.56	0.32	0.37
38.44	0.30	0.34
41.34	0.35	0.40



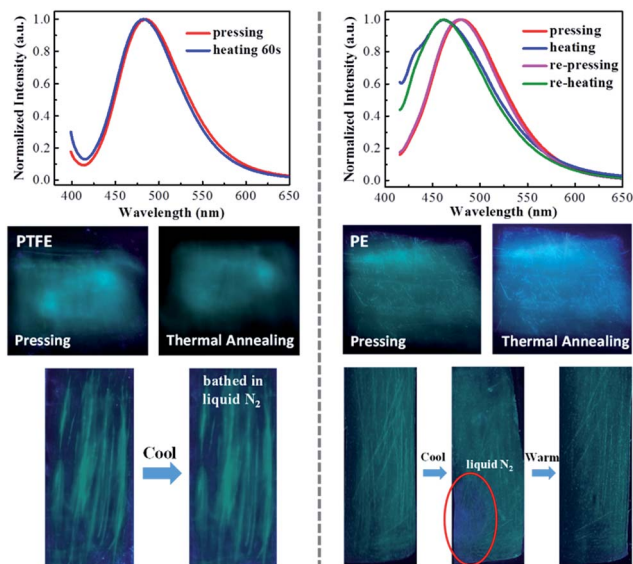


Fig. 4 Comparison of mechanochromic luminescence properties of 1P1F on polytetrafluoroethylene (PTFE, left panel) and polyethylene (PE, right panel).

In conclusion, we designed and successfully synthesized three types of fluorinated and non-fluorinated carbon-chain derivatives of pyrene dyes and examined their mechanochromic luminescence properties in the solid state, with an emphasis on the role of substituents and substrates. Based on our design rationale, we predict that only one of the six molecules has a high probability to exhibit ML behaviour. Experimental results agree with the prediction and show that the fluorinated linear structure is more favourable for ML. The substrate effects were also explored and it can be concluded that strong interactions between the dye and the substrate can cost ML properties by randomizing dye distribution on a molecular level. Weak dye-substrate interactions and strong dye-dye interactions are perhaps the best scenario to design a functional ML system. These preliminary results can be employed as a general strategy to guide rational ML design in future works.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

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