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# Tailored fabrication of TiO<sub>2</sub>@carbon nanofibers composites *via* foaming agent migration

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Based on the merits and demerits of conventional methods for fabricating TiO<sub>2</sub>@carbon nanofibers (TiO<sub>2</sub>@CNFs) composites, the composites were prepared *via* a foaming-assisted electrospinning strategy and subsequent thermal treatment. Betaine was used as the foaming agent and assembled with TiO<sub>2</sub> before electrospinning. It was found that the foaming agent of betaine introduced within the solutions played a crucial role on the migration of TiO<sub>2</sub> nanoparticles to the surface of CNFs, enabling their regular movement when the carbonization temperature increased from 400 to 1000 °C. The TiO<sub>2</sub> loading amount on the surface of CNFs increased gradually with an enhanced temperature, resulting in an increase in specific surface area from 30.09 to 419.8 m<sup>2</sup> g<sup>-1</sup>. In addition, the obtained TiO<sub>2</sub>@CNFs composites exhibited excellent hydrophilicity and high degree of graphitization. Accordingly, when the carbonization temperature increased, the composites presented an enhanced photocatalytic activity toward the degradation of rhodamine B. Hence, a novel method for preparing TiO<sub>2</sub>@CNFs composites with tailored TiO<sub>2</sub> loading amount could be envisioned.

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## 1. Introduction

Civilization dictates the progress of society, which results in more production from textile industry with increasing quantity of discharge of wastewater and organic dyes.<sup>1</sup> As an effective approach, photocatalytic degradation has been employed for effluent treatment, where most of the photocatalysts were clean, economical and environment-friendly. Oxide semiconductor materials have been gradually developed into a promising approach to solve environmental protection issues, such as air purification, hazardous waste remediation and water purification.<sup>2–4</sup> Among the various semiconductor photocatalysts, TiO<sub>2</sub> has been extensively studied for the removal of toxic dyes due to its high chemical stability, controllable structure, nontoxicity and low cost.<sup>5</sup> However, the large band gap prevents it from practical applications, *i.e.*, the issue of recombination of photogenerated charge carriers has not been fully overcome.<sup>6</sup>

To overcome such limitations, various approaches have been studied to improve the photocatalytic activity by extending the photoresponsive range and increasing the electron–hole extending pair separation efficiency.<sup>7–10</sup> Lee *et al.* demonstrated that electrostatic layer-by-layer deposition effectively immobilized and stabilized TiO<sub>2</sub> nanoparticles on polymer nanofibers for degradation of bisphenol A.<sup>11</sup> One-dimensional carbon nanofibers (CNFs) have been used for the development of load-bearing nanoparticles due to their good

conductive property, high adsorptive capacities and mechanical strength as well as strong fatigue resistance.<sup>12–15</sup> Electrospinning was reported as a straightforward technique that could produce continuous fibers with fine flexibility, low density, high porosity, and ease of scalable synthesis.<sup>16–18</sup> Therefore, electrospinning technique combined with thermal treatment provided a promising and cost-effective approach to prepare continuous CNFs loaded with TiO<sub>2</sub> nanoparticles (TiO<sub>2</sub>@CNFs composites).<sup>19–21</sup> In recent years, there have been two main approaches to prepare TiO<sub>2</sub>@CNFs composites *via* electrospinning technique: blended spinning<sup>22</sup> and post-assembly pathways.<sup>23</sup> It is well known that most of the TiO<sub>2</sub> nanoparticles were embedded inside the CNFs in blended spinning method and part of TiO<sub>2</sub> nanoparticles could fall off from CNFs in post-assembly pathways. Therefore, the preparation and optimization of TiO<sub>2</sub>@CNFs composites still remains a challenge.

Based on these considerations, the objective of this study is to demonstrate a novel TiO<sub>2</sub>@CNFs composite formed by the combination of foaming-assisted electrospinning strategy and thermal treatment involving pre-oxidation and carbonization. Betaine was used as the foaming agent and assembled with TiO<sub>2</sub> before electrospinning. Further, on calcination, the abundant vapors released from betaine could drive TiO<sub>2</sub> nanoparticles from within the surface of the CNFs, leading to formation of tailored TiO<sub>2</sub> loading amount composites. The morphology, structure and photocatalytic activity of TiO<sub>2</sub>@CNFs composites with different heat treatment temperatures were then investigated.

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## 2. Experimental section

### 2.1 Materials

Polyacrylonitrile (PAN,  $M_w = 150\,000$ ) was provided by Shaoxing Gimel Advanced Materials Technology Co., Ltd, China.  $\text{TiO}_2$  nanoparticles (20–50 nm) were purchased from Shanghai Titanium Dioxide Chemical Products Co., Ltd. Betaine was provided by Tianjin Tianzhi Fine Chemical Co., Ltd.  $N,N$ -Dimethylformamide was used as a solvent, purchased from Tianjin Fuyu Chemical Co., Ltd. Rhodamine B (RhB) was provided by Tianjin Fengchuan Chemical Reagent Science And Technology Co., Ltd. All chemicals were used as received.

### 2.2 Preparation of $\text{TiO}_2$ @CNFs composites

A certain amount of  $\text{TiO}_2$  nanoparticles were dispersed in betaine solution and sonicated for 1 h to obtain a homogeneous solution (10 wt%). Then, the resulting mixtures were stirred for 24 h and dried at 80 °C overnight.

A homogeneous solution of PAN and betaine- $\text{TiO}_2$  mixtures (12 wt%) dissolved in  $N,N$ -dimethylformamide was constantly stirred at room temperature for 24 h. Then, the nanofibers were electrospun onto the aluminium foil. The voltage, feed rate and collection distance were 25 kV, 1 mL  $\text{h}^{-1}$  and 15 cm, respectively. After electrospinning, the blend fibers were dried at 60 °C under vacuum overnight.

The dried blend fibers were stabilized in an air atmosphere at 270 °C for 4 h with a ramp rate of 2 °C  $\text{min}^{-1}$  in order to induce dimensional stability of the carbon nanofibers (CNFs). Then, the above stabilized fibers were carbonized in a tube furnace at 400, 600, 800 and 1000 °C for 2 h under  $\text{N}_2$  atmosphere with a ramp rate of 5 °C  $\text{min}^{-1}$ , respectively. The preparation process of  $\text{TiO}_2$ @CNFs composites is shown in Fig. 1.

### 2.3 Characterization

The morphology of the as-prepared composites was observed using scanning electron microscope (SEM, Hitachi S-4800) and transmission electron microscope (TEM, JEM-2100). The Brunauer-Emmett-Teller (BET) specific surface areas were examined using  $\text{N}_2$  adsorption-desorption isotherms with Micromeritics Analytical Services (TriStar 3000, GA). X-Ray Diffraction (XRD) patterns of composites were recorded at a scanning speed of 5°  $\text{min}^{-1}$  from 3 to 80° with Cu K $\alpha$  radiation ( $\lambda = 1.54059$  Å). Raman spectrum was used to investigate the

crystallinity of carbon nanofibers with an excitation wavelength of 514 nm. The water contact angles were tested using an image processing program based on the sessile-drop method (JYSP-180 Contact Angle Analyzer). At least five contact angle values of different spots were collected to obtain the final contact angle value.<sup>24,25</sup>

### 2.4 Photocatalytic tests

The photocatalytic activity of the as-prepared composites was measured by degrading RhB under the irradiation of a 250 W ultraviolet lamp. The obtained composites (2 cm  $\times$  2 cm) were mixed with a solution of 10 mg  $\text{L}^{-1}$  RhB using a mechanical shaker in the dark for 60 min to ensure a good dispersion and establish adsorption-desorption equilibrium between the organic molecules and the catalyst surfaces.<sup>26</sup> Then, the mixtures were exposed to UV light. After UV irradiation, 3 mL dye solution was withdrawn at given intervals of illumination (10 min). Finally, the concentrations of RhB aqueous solution were recorded by UV spectrophotometer (Shimadzu UV-2450, Japan) at the maximum adsorption wavelength ( $\lambda = 554$  nm) according to Lambert-Beer's law.<sup>27</sup> In the entire photo-degradation process, the temperature of the solution was kept at  $20 \pm 2$  °C.

## 3. Results and discussion

### 3.1 Morphology and structure of $\text{TiO}_2$ @CNFs composites

Fig. 2 shows the SEM images of  $\text{TiO}_2$ @CNFs composites prepared at different carbonization temperatures. Clearly, the content of  $\text{TiO}_2$  nanoparticles on the surface of CNFs could be viewed, indicating that the  $\text{TiO}_2$  loading amount on the surface of CNFs was enhanced when the carbonization temperature increased. During the carbonization process, betaine would be decomposed and rapidly volatilized. The outward diffusion of some gas generated a force to migrate the  $\text{TiO}_2$  nanoparticles outside. In this case, it could be assumed that the interface layer between the  $\text{TiO}_2$  and carbon nanofibers would change due to the increased temperature. Therefore, the composites with tailored  $\text{TiO}_2$  loading amount could be obtained. It is interesting to note that when the carbonization temperature was 1000 °C, the  $\text{TiO}_2$  nanoparticles were agglomerated and became larger (red circles in Fig. 2d), indicating that a small amount of the  $\text{TiO}_2$  nanoparticles could fall off. The SEM images not only presented a uniform distribution of  $\text{TiO}_2$  nanoparticles on the

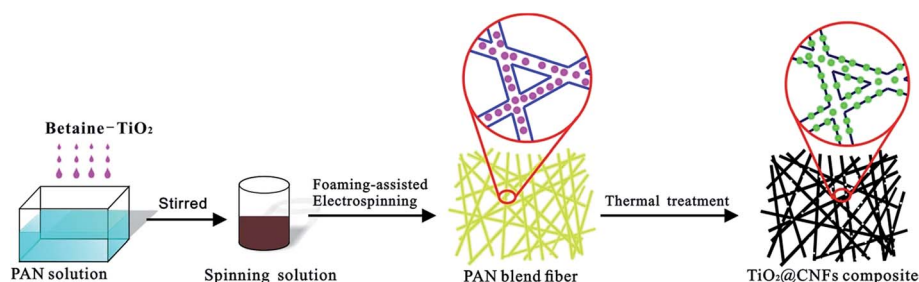


Fig. 1 Schematic illustrations of the synthetic pathways of  $\text{TiO}_2$ @CNFs composites.



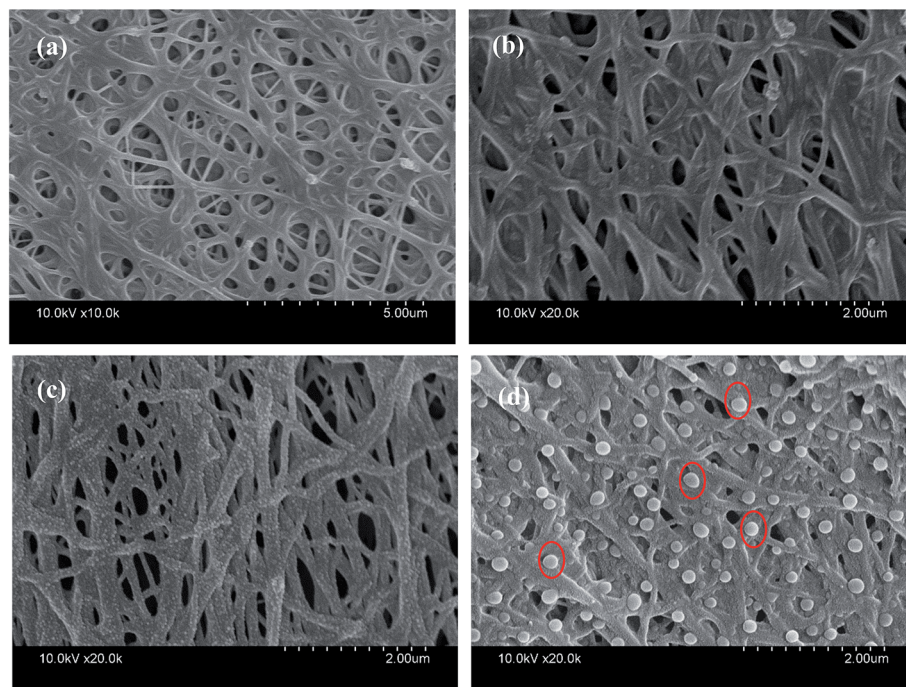


Fig. 2 SEM images of  $\text{TiO}_2\text{@CNFs}$  composites at (a) 400 °C, (b) 600 °C, (c) 800 °C, and (d) 1000 °C.

surface of CNFs with further increasing the carbonization temperature, but also showed that most of the carbon nanofibers were bonded to each other, which could affect the results of photocatalytic tests.

The relevant  $\text{N}_2$  adsorption–desorption isotherm curves of the  $\text{TiO}_2\text{@CNFs}$  composites are shown in Fig. 3, exhibiting a type IV isotherms with a variety of typical adsorptive behaviours, such a monolayer adsorption. This indicated that the mesoporous structure could exist within the composites,<sup>28</sup> which were formed by the betaine volatilizing with the increasing carbonization temperature. Table 1 shows the calculated BET specific surface areas, presenting a dramatic

increasing trend. Notably, the specific surface areas of composites increased from 30.09 to 419.8  $\text{m}^2 \text{g}^{-1}$  with the increase in carbonization temperature. Such large changes could be explained by the gradual migration of  $\text{TiO}_2$  nanoparticles to the surface of CNFs, which formed the rough structures, indicating that a small amount of photocatalytic activity would be expected.

The TEM images in Fig. 4 reveal clear lattice fringes of the nanocrystal with an interplanar space of 0.35 nm (ref. 29) when the carbonization temperature changed from 400 to 800 °C. The fringes correspond to the (101) crystal plane of anatase  $\text{TiO}_2$  nanoparticles. When the carbonization temperature was increased to 1000 °C, the interplanar space changed to 0.32 nm, indicating that part of the crystal type has transformed to rutile  $\text{TiO}_2$ .<sup>30</sup> In particular, for the composite prepared at high carbonization temperature, it could be seen that the ultrasonic process during the sample preparation for TEM examination hardly had an effect on the  $\text{TiO}_2$  nanoparticles; hence, they did not fall off the CNFs, implying that the interface between  $\text{TiO}_2$  nanoparticles and CNFs was strong. In addition, for the composite prepared at high carbonization temperature, the

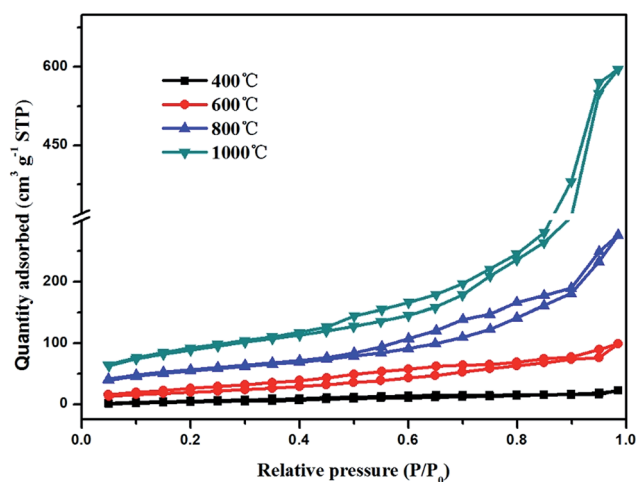


Fig. 3  $\text{N}_2$  adsorption–desorption isotherms of  $\text{TiO}_2\text{@CNFs}$  composites.

Table 1 Structure parameters of  $\text{TiO}_2\text{@CNFs}$  composites at different carbonization temperatures

	$2\theta$ (°)	$d_{002}$ (nm)	$I_D/I_G$	Specific surface area ( $\text{m}^2 \text{g}^{-1}$ )
400 °C	25.18	0.3496	1.339	30.09
600 °C	25.30	0.3479	1.1913	113.2
800 °C	25.33	0.3475	1.035	287.4
1000 °C	25.38	0.3469	0.9833	419.8





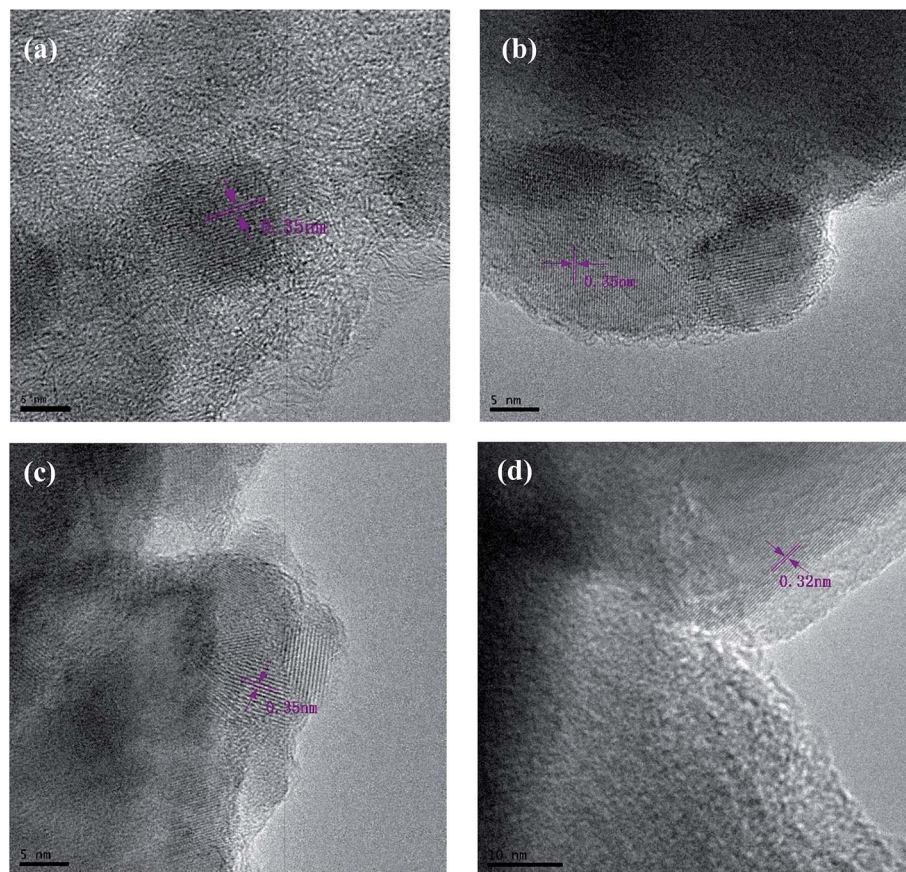


Fig. 4 TEM showing the  $\text{TiO}_2$  nanocrystals in (101) orientation at (a) 400 °C, (b) 600 °C, (c) 800 °C, and (d) 1000 °C.

migration process of  $\text{TiO}_2$  nanoparticles could be viewed visually from the TEM pictures, which was consistent with the results of SEM tests.

The crystalline structure of composites could be analysed by the XRD method. Fig. 5a shows the XRD patterns of  $\text{TiO}_2$ @CNFs composites at different carbonization temperatures, which were consistent with the  $\text{TiO}_2$  anatase phase (JCPDS no. 21-1272), indicating that the crystalline structure of  $\text{TiO}_2$  nanoparticles remained unchanged with the increasing carbonization

temperature. Nevertheless, when the carbonization temperature was increased up to 1000 °C, the crystal type of  $\text{TiO}_2$  nanoparticles was transformed into rutile phase (JCPDS no. 21-1276), which was consistent with the results of TEM analyses. It is well known that anatase  $\text{TiO}_2$  is in favour of photocatalytic degradation, indicating that a slight decrease in photocatalytic activity of the composites would occur when the carbonization temperature was 1000 °C. The interlayer distance  $d_{002}$  of carbon fibers, which was calculated from XRD spectra using Bragg

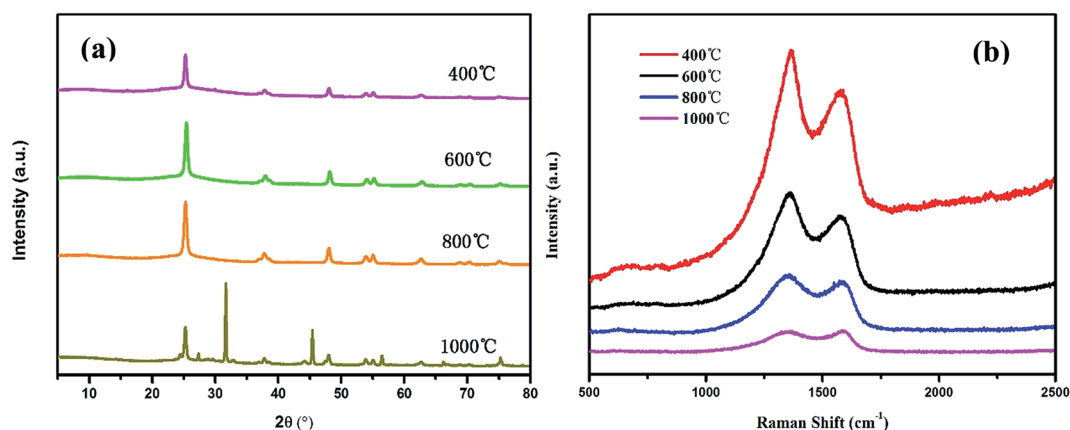


Fig. 5 XRD patterns (a) and Raman spectra (b) of  $\text{TiO}_2$ @CNFs composites.



equations,<sup>31</sup> is shown in Table 1. As it is well known, the carbon fiber has a broad peak at  $2\theta$  of  $25.6^\circ$ , which is attributed to the lattice plane of graphitized carbon (002).<sup>32,33</sup> It was evident that the diffraction peaks of anatase  $\text{TiO}_2$  (101) and carbon fiber (002) were coincident. In addition, it was reported that the changes in diffraction peak at about  $25.6^\circ$  in  $\text{TiO}_2$ @CNFs composites mainly resulted from the enhancement in degree of graphitization of CNFs.<sup>34</sup> It could be observed from Table 1 that the  $d_{002}$  interlayer spacing, indicative of the degree of graphitization,<sup>35</sup> decreased from 0.3496 to 0.3469 when the carbonization temperature increased from 400 to 1000  $^\circ\text{C}$ , indicating the improvement of graphitization degree. Hence, the improved photocatalytic activity of composites would be expected.

Raman spectroscopy of  $\text{TiO}_2$ @CNFs composites, which were prepared at different carbonization temperatures, is demonstrated in Fig. 5b. As shown in Fig. 5b and Table 1, the degree of graphitization of composites increased gradually due to the lower  $I_D/I_G$  value when the carbonized temperature ranged from 400 to 1000  $^\circ\text{C}$ , where the  $I_D$  and  $I_G$  represent the intensity of defect graphite structure and  $\text{sp}^2$  hybridized carbon,<sup>36</sup> respectively. Notably, the  $I_D/I_G$  values of composites decreased, confirming the transformation of disordered carbon into graphitized carbon with an enhanced carbonization temperature,<sup>37</sup> which matched well with the abovementioned XRD results.

### 3.2 Hydrophilicity of $\text{TiO}_2$ @CNFs composites

The water contact angle was an important parameter for testing the surface hydrophilicity of the as-prepared composites. The contact angle results are shown in Fig. 6, from which a clear trend could be drawn that the contact angle values of  $\text{TiO}_2$ @CNFs composites decreased gradually when the carbonization temperature increased. Notably, when the carbonization temperature increased above 800  $^\circ\text{C}$ , the contact angle values were almost  $0^\circ$ . It should be ascribed to the hydrophilic  $\text{TiO}_2$  nanoparticles migrated to the surface of CNFs to reduce the interface energy with the increase in carbonization

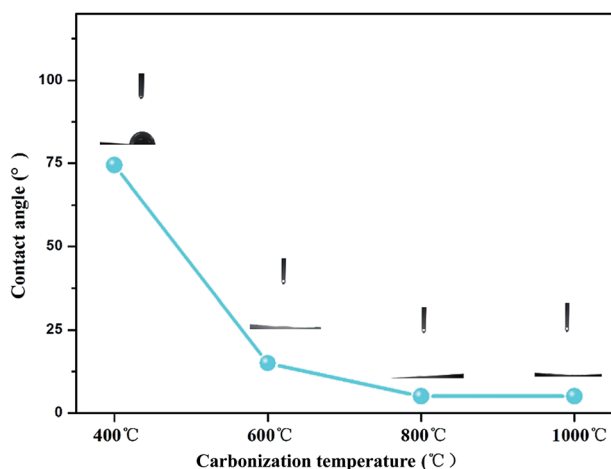


Fig. 6 Contact angles of  $\text{TiO}_2$ @CNFs composites at different carbonization temperatures.

temperature, making the composite surface hydrophilic, which was consistent with the abovementioned SEM and TEM results. The results also presented that  $\text{TiO}_2$  nanoparticles were anchored firmly on the surface of the CNFs, confirming that the interface between  $\text{TiO}_2$  nanoparticles and CNFs was strong, indicating that the hydrophilic property of  $\text{TiO}_2$ @CNFs composites could play a favourable role in elevating the photocatalytic activity.

### 3.3 Photocatalytic activity of $\text{TiO}_2$ @CNFs composites

The photocatalytic activities of the  $\text{TiO}_2$ @CNFs composites were evaluated by the degradation of RhB. Fig. 7a shows the plots of the time-dependent degradation efficiencies of RhB over the four photocatalysts, which clearly show the degradation trends. For the first 60 min, the degradation rate changed slightly, indicating that the absorption-desorption equilibrium between the RhB solution and the composite surfaces was achieved. The photocatalytic activities of  $\text{TiO}_2$ @CNFs composites gradually increased with an increase in temperature, which was attributed to the gradual increase in degree of graphitization and  $\text{TiO}_2$  loading amount on the surface of CNFs. Fantastically, in spite of the transformation of  $\text{TiO}_2$  crystal structure, the photocatalytic activities of the composites improved, indicating that  $\text{TiO}_2$  loading amount on the surface of CNFs was vital for enhancing photocatalytic activities. However, when the carbonization temperature was 1000  $^\circ\text{C}$ , the degradation rate was only 87.9%, which could be owing to the fact that part of  $\text{TiO}_2$  nanoparticles had fallen off and most of the carbon nanofibers were bonded to each other. Hence, the parameters of thermal treatment and the addition of betaine- $\text{TiO}_2$  mixture would be further investigated in our later study.

From the kinetic studies on the photocatalytic reactions over the four samples, their photocatalytic performances could be investigated by the Langmuir-Hinshelwood pseudo-first-order kinetics model<sup>38</sup> using the equation  $\ln(C_0/C) = kt$  (where  $C_0$  and  $C$  are the concentrations of the initial and real-time RhB, respectively, and  $k$  is the apparent rate constant). From Fig. 7b, it can be observed that the corresponding reaction rate constants were calculated as 0.019, 0.025, 0.023 and 0.029  $\text{min}^{-1}$  with correlation coefficients ( $R^2$ ) of 0.996, 0.994, 0.991 and 0.988, respectively. These results implied that the photocatalytic performances of  $\text{TiO}_2$ @CNFs composites improved as the carbonization temperature increased, which matched well with the photocatalytic activities illustrated in Fig. 5a.

Fig. 8 shows the schematic illustration of the mechanism of photocatalytic activity of  $\text{TiO}_2$ @CNFs composites for RhB solution. First, the RhB organics were adsorbed to the surface of CNFs, which was ascribed to the high specific area of composites. Second, when the  $\text{TiO}_2$  nanoparticles were illuminated by UV light with higher photon energy than the band gap of  $\text{TiO}_2$  nanoparticles, electrons in the valence band (VB) could be excited to the conduction band (CB), resulting in improvement in the photocatalytic activity of  $\text{TiO}_2$  nanoparticles.<sup>39</sup> In addition, there have been some reports that CNFs could introduce UV-generated electrons ( $e^-$ ) into the conduction band (CB) of



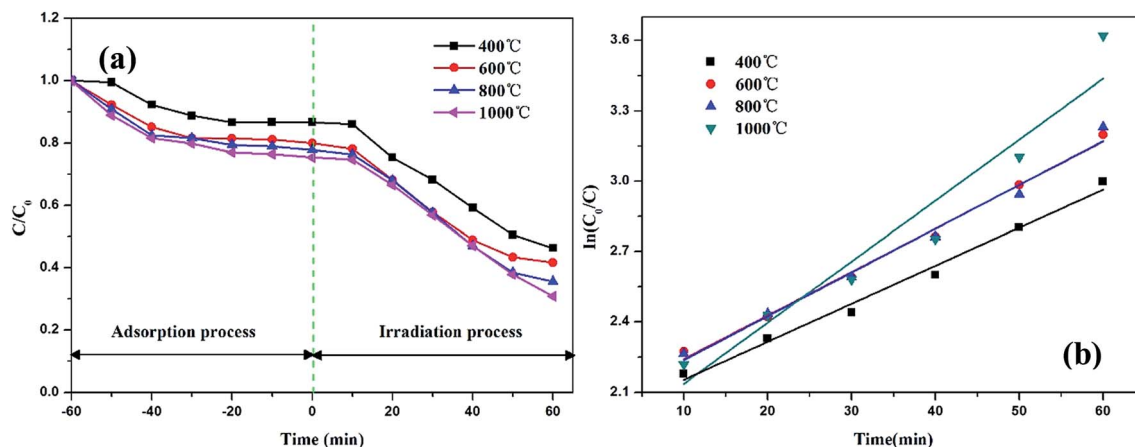


Fig. 7 Photocatalytic degradation profiles (a) and kinetic linear fitting curves (b) of RhB over  $\text{TiO}_2$ @CNFs composites.

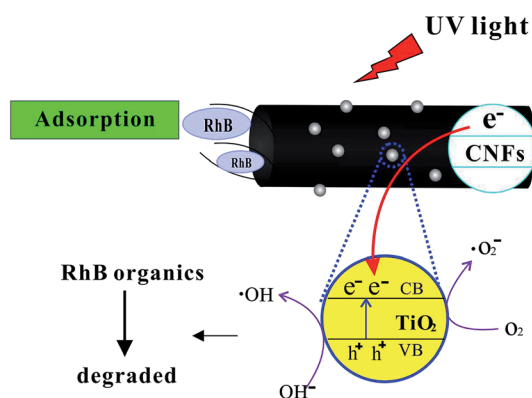


Fig. 8 Proposed mechanism of photocatalytic degradation of RhB.

$\text{TiO}_2$  nanoparticles,<sup>40,41</sup> indicating that the electron–hole pair separation efficiency increased, which could improve the photocatalytic activity of composites. These electrons transferred between CNFs and  $\text{TiO}_2$  nanoparticles could react with  $\text{O}_2$ , producing the highly reactive superoxide radical ions ( $\text{O}_2^{\bullet-}$ ). Simultaneously, the positively charged holes ( $\text{h}^+$ ) could be generated due to the transfer of electrons from the valence band of  $\text{TiO}_2$  nanoparticles to the conduction band of CNFs. These positively charged holes ( $\text{h}^+$ ) might then react with the  $\text{OH}^-$  produced by the dissociation of  $\text{H}_2\text{O}$ , which could lead to the formation of the hydroxyl radical ( $\text{•OH}$ ). In conclusion, CNFs could play an important role in forming highly reactive superoxide radical ions ( $\text{O}_2^{\bullet-}$  and  $\text{•OH}$ ).

## 4. Conclusions

We have reported a simple method to fabricate  $\text{TiO}_2$ @CNFs composites as photocatalysts *via* the foaming-assisted electrospinning technique and a subsequent thermal treatment process. The introduced betaine used as the foaming agent in the initial solutions played an important role in the formation of the composites with tailored  $\text{TiO}_2$  loading amount, enabling effective migration of  $\text{TiO}_2$  to the surface of CNFs. When the

carbonization temperature ranged from 400 to 1000 °C, the  $\text{TiO}_2$  loading amount on the surface of CNFs increased, resulting in an increase in specific surface area from 30.09 to 419.8  $\text{m}^2 \text{g}^{-1}$ . In addition, the obtained  $\text{TiO}_2$ @CNFs composites exhibited excellent hydrophilicity and high degree of graphitization. The investigation of the photocatalytic degradation of RhB disclosed that the composites presented an enhanced photocatalytic activity when the carbonization temperature increased. In addition, at a temperature of 1000 °C, the composites showed the best photocatalytic activity. It is promising that the present study could provide some useful insight for the exploration of high loading amount of photocatalysts with excellent performances.

## Conflicts of interest

The authors declare that they have no conflict of interest to this study.

## References

- 1 Z. Xu, X. Li, K. Teng, B. Zhou, M. Ma, M. Shan, K. Jiao, X. Qian and J. Fan, *J. Membr. Sci.*, 2017, **535**, 94–102.
- 2 J. Li, W. Ma, C. Chen, J. Zhao, H. Zhu and X. Gao, *J. Mol. Catal. A: Chem.*, 2007, **261**, 131–138.
- 3 B. Cheng, Y. Le and J. Yu, *J. Hazard. Mater.*, 2010, **177**, 971–977.
- 4 D. Wang, X. Li, J. Chen and X. Tao, *Chem. Eng. J.*, 2012, **198–199**, 547–554.
- 5 X. Li, B. Zhou, W. Wang, Z. Xu, N. Li, L. Kuang, C. Li, W. Mai, H. Fu and H. Lv, *J. Alloys Compd.*, 2017, **706**, 103–109.
- 6 R. Sedghi, H. R. Moazzami, S. S. Hosseiny Davarani, M. R. Nabid and A. R. Keshtkar, *J. Alloys Compd.*, 2017, **695**, 1073–1079.
- 7 G. Zhou, X. Liu, C. Nan, Y. Liu, D. Wang and X. Chen, *New J. Chem.*, 2013, **37**, 2582–2588.
- 8 H. Liang, D. Yu, J. Bai, C. Li and T. Ma, *Compos. Interfaces*, 2015, **22**, 663–671.





- 9 S. Dutta, R. Sahoo, C. Ray, S. Sarkar, J. Jana, Y. Negishi and T. Pal, *Dalton Trans.*, 2015, **44**, 193–201.
- 10 A. Morawski, M. Janus, B. Tryba, M. Toyoda, T. Tsumura and M. Inagaki, *Pol. J. Chem. Technol.*, 2009, **11**, 46–50.
- 11 J. A. Lee, Y. S. Nam, G. C. Rutledge and P. T. Hammond, *Adv. Funct. Mater.*, 2010, **20**, 2424–2429.
- 12 Y. Xiong, J. Qian, Y. Cao, X. Ai and H. Yang, *ACS Appl. Mater. Interfaces*, 2016, **8**, 16684–16689.
- 13 P. Zhang, C. Shao, Z. Zhang, M. Zhang, J. Mu, Z. Guo and Y. Liu, *Nanoscale*, 2011, **3**, 2943–2949.
- 14 S. A. Patil, S. Chigome, C. Hagerhall, N. Torto and L. Gorton, *Bioresour. Technol.*, 2013, **132**, 121–126.
- 15 M. Inagaki, Y. Yang and F. Kang, *Adv. Mater.*, 2012, **24**, 2547–2566.
- 16 X. Li, K. Teng, J. Shi, W. Wang, Z. Xu, H. Deng, H. Lv and F. Li, *J. Taiwan Inst. Chem. Eng.*, 2016, **60**, 636–642.
- 17 J. Liu, J. Li, A. Sedhain, J. Lin and H. Jiang, *J. Phys. Chem. C*, 2008, **112**, 17127–17132.
- 18 K. I. Choi, W. Lee, S. H. Lee and C. Lim, *Mater. Lett.*, 2015, **158**, 36–39.
- 19 X. Li, H. Lin, X. Chen, H. Niu, T. Zhang, J. Liu and F. Qu, *New J. Chem.*, 2015, **39**, 7863–7872.
- 20 C. H. Kim, B.-H. Kim and K. S. Yang, *Carbon*, 2012, **50**, 2472–2481.
- 21 T. J. Athauda, J. G. Neff, L. Sutherlin, U. Butt and R. R. Ozer, *ACS Appl. Mater. Interfaces*, 2012, **4**, 6917–6926.
- 22 F. Mehrpouya, H. Tavanai, M. Morshed and M. Ghiaci, *J. Nanopart. Res.*, 2012, **14**, 1074–1084.
- 23 C. Su, X. Ran, J. Hu and C. Shao, *Environ. Sci. Technol.*, 2013, **47**, 11562–11568.
- 24 J. Zhang, Z. Xu, W. Mai, C. Min, B. Zhou, M. Shan, Y. Li, C. Yang, Z. Wang and X. Qian, *J. Mater. Chem. A*, 2013, **1**, 3101.
- 25 T. Wu, B. Zhou, T. Zhu, J. Shi, Z. Xu, C. Hu and J. Wang, *RSC Adv.*, 2015, **5**, 7880–7889.
- 26 X. Wang, J. Choi, D. R. G. Mitchell, Y. B. Truong, I. L. Kyratzis and R. A. Caruso, *ChemCatChem*, 2013, **5**, 2646–2654.
- 27 R. Zhang, X. Wang, J. Song, Y. Si, X. Zhuang, J. Yu and B. Ding, *J. Mater. Chem. A*, 2015, **3**, 22136–22144.
- 28 Y. Si, X. Wang, Y. Li, K. Chen, J. Wang, J. Yu, H. Wang and B. Ding, *J. Mater. Chem. A*, 2014, **2**, 645–652.
- 29 C. Zhao, L. Liu, Q. Zhang, J. Rogers, H. Zhao and Y. Li, *Electrochim. Acta*, 2015, **155**, 288–296.
- 30 H. Hou, M. Shang, L. Wang, W. Li, B. Tang and W. Yang, *Sci. Rep.*, 2015, **5**, 15228.
- 31 J. Jin, Z. Q. Shi and C. Y. Wang, *Electrochim. Acta*, 2014, **141**, 302–310.
- 32 X. Sui, Z. Xu, C. Hu, L. Chen, L. Liu, L. Kuang, M. Ma, L. Zhao, J. Li and H. Deng, *Compos. Sci. Technol.*, 2016, **130**, 46–52.
- 33 M. B. Vázquez-Santos, E. Geissler, K. László, J. N. Rouzaud, A. Martínez-Alonso and J. M. D. Tascón, *J. Phys. Chem. C*, 2012, **116**, 257–268.
- 34 K. Tang, Y. Li, H. Cao, C. Su, Z. Zhang and Y. Zhang, *Electrochim. Acta*, 2016, **190**, 678–688.
- 35 H. Xiao, Y. Lu, M. Wang, X. Qin, W. Zhao and J. Luan, *Carbon*, 2013, **52**, 427–439.
- 36 L. Liu, F. Wu, H. Yao, J. Shi, L. Chen, Z. Xu and H. Deng, *Appl. Surf. Sci.*, 2015, **337**, 241–248.
- 37 J. J. Wang, Y. Y. Shao, Z. B. Wang and Y. Z. Gao, *J. Phys. Chem. C*, 2008, **112**, 5784–5789.
- 38 Z. Guo, C. Shao, J. Mu, M. Zhang, Z. Zhang, P. Zhang, B. Chen and Y. Liu, *Catal. Commun.*, 2011, **12**, 880–885.
- 39 J. Yun, H. I. Kim and Y. S. Lee, *J. Mater. Sci.*, 2013, **48**, 8320–8328.
- 40 H. R. P. Bishweshwar Panta, H. Y. K. Yuri Choic and N. A. M. B. Mira Parkc, *Ceram. Int.*, 2013, **39**, 7029–7035.
- 41 H. E. Unalan, D. Wei, K. Suzuki, S. Dalal, P. Hiralal, H. Matsumoto, S. Imaizumi, M. Minagawa, A. Tanioka, A. J. Flewitt, W. I. Milne and G. A. J. Amaratunga, *Appl. Phys. Lett.*, 2008, **93**, 133116–133119.

