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## Superior photocatalytic performance of $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$ heterojunction nanocomposites under visible light irradiation

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New types of  $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  heterostructures were successfully prepared and the enhanced photocatalytic hydrogen evolution and degradation activities under visible light irradiation were determined. They possessed the features of a Z-scheme photocatalysis system. The photoexcited electrons on the conduction band of  $\text{LaFeO}_3$  were transferred to the valence band of  $\text{g-C}_3\text{N}_4$  by the solid–solid intimately contacted interfaces, where the electrons and holes were recombined, and thus improved the separation of photogenerated electrons and holes of  $\text{g-C}_3\text{N}_4$ . The  $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  heterostructures showed a higher hydrogen evolution rate and a higher amount of  $\cdot\text{OH}$  than pure  $\text{LaFeO}_3$  and  $\text{g-C}_3\text{N}_4$ . The construction of the  $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  heterostructures was used to demonstrate an effective strategy for improving the photocatalytic property. The 5%- $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  exhibited the highest photodegradation and water splitting rate. More than 95% of methylene blue (MB) was degraded after 15 min in the presence of 25 mg EDTA-2Na in 100 mL MB solution which was irradiated using a 3 W light-emitting diode. The 5%- $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  heterojunction nanocomposite had a maximum hydrogen evolution rate of  $158 \mu\text{mol g}^{-1} \text{h}^{-1}$ .

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### 1. Introduction

The rapid development of modern industry using the consumption of non-renewable fossil fuels has caused a serious global energy shortage and environmental pollution.<sup>1</sup> As the largest renewable energy source, the exploitation of solar energy could mean less use of fossil fuels and thereby protect the environment. The utilization of solar energy using semiconductor-based photocatalysis has the capability of hydrogen ( $\text{H}_2$ ) evolution and pollutant removal, and is considered as an economical, renewable, clean and safe technology. A variety of semiconductor photocatalysts have been developed over the past few decades.<sup>2–5</sup> Because visible light energy accounts for more than 40% of the sunlight energy, people are committed to developing visible light driven photocatalysts. So far, various visible light responsive semiconductor photocatalysts have been developed,<sup>6–9</sup> but the fast charge recombination and insufficient stability still make it impractical for general use.

Very recently, graphitic carbon nitride ( $\text{g-C}_3\text{N}_4$ ) was found to be a promising metal-free material with a large surface area, high thermal and chemical stability, fast charge transfer rate,

and a suitable bandgap of approximately 2.7 eV.<sup>10–13</sup> It is consequently regarded as the next generation photocatalyst for water splitting and degradation of pollutants under visible light irradiation. However, the low photocatalytic efficiency of  $\text{g-C}_3\text{N}_4$  caused by its narrow response range to visible light, fast recombination of photogenerated electron–hole pairs and low quantum efficiency limits its further applications. Therefore, researchers have focussed on different modification methods to improve its photocatalytic activity, including copolymerization,<sup>14</sup> morphology control<sup>15</sup> and doping.<sup>16</sup> The construction of the  $\text{g-C}_3\text{N}_4$ –semiconductor heterostructures is an effective strategy to promote the separation of a photogenerated electron and hole, and thus improve the photocatalytic property.

A variety of nanocomposites have been reported in the literature for dye degradation and  $\text{H}_2$  evolution. Halogenides,<sup>17,18</sup> sulfides<sup>19,20</sup> and oxides<sup>21,22</sup> have been developed to construct the heterojunction with  $\text{g-C}_3\text{N}_4$ . These  $\text{g-C}_3\text{N}_4$ –semiconductor heterostructures possess excellent photocatalytic performance in degradation of organic pollutants. Fu *et al.*<sup>17</sup> reported novel heterojunctions of bismuth oxybromide–carbon nitride ( $\text{BiOBr–C}_3\text{N}_4$ ) fabricated by depositing  $\text{BiOBr}$  nanoflakes onto the surface of  $\text{C}_3\text{N}_4$ . The optimum photocatalytic activity of the 0.5 $\text{BiOBr}$ –0.5 $\text{C}_3\text{N}_4$  heterojunction was 4.9 and 17.2 times as high as those of individual  $\text{BiOBr}$  and  $\text{C}_3\text{N}_4$ , respectively, under visible light irradiation. Jiang *et al.*<sup>20</sup> synthesized calcium indium sulfide ( $\text{CaIn}_2\text{S}_4$ )/ $\text{g-C}_3\text{N}_4$  heterojunction nanocomposites using a two-step method. It exhibited a  $\text{H}_2$  evolution rate of  $102 \mu\text{mol g}^{-1} \text{h}^{-1}$ , which was more than 3 times that of

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pristine  $\text{CaIn}_2\text{S}_4$ . The highest activity was obtained for a 30%  $\text{CaIn}_2\text{S}_4/\text{g-C}_3\text{N}_4$  heterojunction nanocomposite, over which 90% of methyl orange was degraded after 120 min. Zhao *et al.*<sup>21</sup> prepared novel  $\text{g-C}_3\text{N}_4/\text{tz-Bi}_{0.92}\text{Gd}_{0.08}\text{VO}_4$  heterojunctions using a microwave hydrothermal method. The degradation rate of Rhodamine B was 94% after 20 min, which was 39.97% higher than that of  $\text{g-C}_3\text{N}_4$ . All the  $\text{g-C}_3\text{N}_4/\text{tz-Bi}_{0.92}\text{Gd}_{0.08}\text{VO}_4$  heterojunctions exhibited a higher degradation rate than that of the pure one. These  $\text{g-C}_3\text{N}_4$ -semiconductor heterostructures possess excellent photocatalytic performance in degradation of organic pollutants. However, there are only a few reports in the literature involving  $\text{g-C}_3\text{N}_4$ -based heterostructures with simultaneously enhanced photocatalytic degradation and  $\text{H}_2$  evolution performance.

Perovskite lanthanum ferric oxide ( $\text{LaFeO}_3$ ) has been applied to catalysts, sensors, environmental monitoring and membranes in syngas production.<sup>23,24</sup>  $\text{LaFeO}_3$  exhibits an interesting catalytic activity in the degradation of chlorinated volatile organic compounds.<sup>25</sup>  $\text{LaFeO}_3$  has a band gap of about 2.0 eV, which is advantageous for absorbing visible light, for good thermal stability and for high catalytic activity.<sup>26</sup> In this work,  $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  Z-scheme photocatalysis system was constructed. The photoexcited electrons on the conduction band (CB) of  $\text{LaFeO}_3$  were transferred to the valence band (VB) of  $\text{g-C}_3\text{N}_4$  by the solid-solid intimately contacted interfaces, where the electrons and holes were recombined, and thus improved the separation of the photogenerated electron and hole of  $\text{g-C}_3\text{N}_4$ . Combining this with a sacrificial agent [triethanolamine (TEOA) or ethylenediaminetetraacetic acid disodium salt (EDTA-2Na)] which reacted with the holes on the VB of  $\text{LaFeO}_3$ , the photocatalytic efficiency was improved significantly. The  $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  Z-scheme photocatalysis system showed a higher  $\text{H}_2$  evolution rate and a higher amount of  $\cdot\text{OH}$  than pure  $\text{LaFeO}_3$  and  $\text{g-C}_3\text{N}_4$ . More than 95% of methylene blue (MB) was degraded after 15 min in the presence of 25 mg EDTA-2Na in 100 mL MB solution irradiated using a 3 W light-emitting diode (LED). The 5%- $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  heterojunction nanocomposite had a maximum  $\text{H}_2$  evolution rate of 158  $\mu\text{mol g}^{-1} \text{h}^{-1}$ .

## 2. Experimental

### 2.1 Chemicals

Iron(III) nitrate nonahydrate ( $\text{Fe}(\text{NO}_3)_3$ ), lanthanum(III) nitrate hexahydrate ( $\text{La}(\text{NO}_3)_3$ ) hexachloroplatinic acid ( $\text{H}_2\text{PtCl}_6$ ), terephthalic acid, MB and melamine were purchased from Shanghai Aladdin Biochemical Technology Co., Ltd. All the chemicals were used without further purification.

### 2.2 Preparation of $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$

Graphitic carbon nitride was prepared by the polycondensation of melamine (6 g) at 500 °C for 4 h in a muffle furnace. The ramping rate was 2 °C min<sup>-1</sup>. For a typical synthetic reaction of the  $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  nanocomposites, 2 g of  $\text{g-C}_3\text{N}_4$ , 0.1700 g of  $\text{Fe}(\text{NO}_3)_3$ , and 0.1800 g of  $\text{La}(\text{NO}_3)_3$  were dissolved in 50 mL of deionized water ( $\text{H}_2\text{O}$ ) with stirring for 1 h. Then, the solution

was evaporated to generate a solid product which was then ground into powder. The powder was placed in a crucible with a cover and then heated for another 2 h at 450 °C in a muffle furnace. The  $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  nanocomposites were obtained after cooling down the powder to room temperature. The as-prepared product was 5%- $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$ . Changing the amounts of  $\text{Fe}(\text{NO}_3)_3$  and  $\text{La}(\text{NO}_3)_3$  added, 10%, 20%, 40% and 60% nanocomposites were synthesized. The as-prepared  $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  containing 5, 10, 20, 40, 60 wt% of  $\text{LaFeO}_3$  were denoted as LF1-5 or 5%- $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  and so on.

### 2.3 Characterization

Transmission electron microscopy (TEM) and high-resolution-TEM (HRTEM) images were performed on a FEI Tecnai G2 F20 transmission electron microscope. X-ray diffraction patterns (XRD) patterns were carried out on a Rigaku D/Max 2500 X-ray diffractometer with Cu K $\alpha$  radiation ( $\lambda = 1.5406 \text{ \AA}$ ). X-ray photoelectron spectroscopy (XPS) was carried out using an ESCA PHI500 spectrometer. Infrared spectra were obtained using a Nicolet NEXUS 470 Fourier-transform infrared spectroscopy (FTIR) instrument in the range of 4000–500  $\text{cm}^{-1}$ . Ultraviolet-visible (UV-vis) spectra were recorded on a Varian Cary 50 spectrophotometer. UV-vis diffuse reflectance spectra (DRS) were measured on Shimadzu UV-2401 spectrophotometer equipped with a spherical diffuse reflectance accessory.

### 2.4 Photoelectrochemical measurement

A portion of photocatalyst (8 mg;  $\text{g-C}_3\text{N}_4$ ,  $\text{LaFeO}_3$  or  $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$ ), 2 mL of ethanol and 40  $\mu\text{L}$  of 5 wt% Nafion were mixed to form a slurry. Then the suspension was dropped onto fluorine doped tin oxide (FTO) glass with an area of 1  $\text{cm}^2$  which performed as the working electrode in a three-electrode system. Platinum (Pt) foil was selected as auxiliary electrode, and silver/silver chloride (Ag/AgCl) as reference electrode. A 3 W LED lamp was the light source. The photocurrent was measured in 0.2 M aqueous solution of sodium sulfate.

### 2.5 Photocatalytic degradation of MB

The photocatalytic activities of photocatalysts ( $\text{g-C}_3\text{N}_4$ ,  $\text{LaFeO}_3$  or  $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$ ) were determined in a 5  $\text{mg L}^{-1}$  aqueous solution of MB. For a typical photodegradation process, 100 mg of photocatalyst was dispersed in 100 mL of MB solution with magnetic stirring in dark for 30 min to reach an adsorption-desorption equilibrium. Then the photodegradation was tested under 3 W LED irradiation. A portion (5 mL) of reaction mixture was collected at the certain time intervals and centrifuged to remove the photocatalyst. The absorption peak at 664 nm was chosen to determine the concentration of MB using a Varian Cary 50 spectrophotometer.

### 2.6 Measurement of hydroxyl radicals

A 0.5 mM solution of terephthalic acid (TA) was obtained by dissolving TA in 2.0 mM sodium hydroxide. Then 100 mg photocatalyst was suspended in 100 mL of TA solution. The mixture was then irradiated under 3 W LED irradiation. A



portion (5 mL) of reaction mixture was collected at the certain time intervals and then centrifuged. The fluorescence intensity of 2-hydroxyterephthalic acid (HTA) was measured on a PerkinElmer LS 55 luminescence spectrometer.

## 2.7 Photocatalytic hydrogen evolution

The photocatalytic  $H_2$  evolution reactions were implemented in an irradiation reaction vessel with a Pyrex top connected to a glass closed gas circulation system. Photocatalyst (100 mg) ( $g\text{-C}_3\text{N}_4$ ,  $\text{LaFeO}_3$  or  $\text{LaFeO}_3/g\text{-C}_3\text{N}_4$ ) was dispersed in 200 mL of an aqueous solution containing 10 vol% TEOA. Pt (3 wt%) as a cocatalyst was deposited on the photocatalysts using an *in situ* photodeposition approach using  $\text{H}_2\text{PtCl}_6$ . The reaction temperature was kept at 20 °C. The  $H_2$  concentration was monitored using a Unisense microsensor monometer equipped with a Clark-type electrochemical  $H_2$  microsensor. The light source was a 300 W xenon (Xe) lamp with a 400 nm longpass cutoff filter.

## 3 Results and discussion

The XRD patterns of  $g\text{-C}_3\text{N}_4$ ,  $\text{LaFeO}_3$  and  $\text{LaFeO}_3/g\text{-C}_3\text{N}_4$  heterojunction nanocomposites are presented in Fig. 1. The  $g\text{-C}_3\text{N}_4$  sample exhibited two diffraction peaks at 13.2° and 27.5°, corresponding to (100) and (002) diffraction planes, respectively. All the diffraction peaks of  $\text{LaFeO}_3$  were indexed to an orthorhombic phase (PDF#37-1493), indicating the purity and high crystallinity of  $\text{LaFeO}_3$ . After forming the  $\text{LaFeO}_3/g\text{-C}_3\text{N}_4$  heterojunction nanocomposites with  $g\text{-C}_3\text{N}_4$ , only the XRD pattern of 5%- $\text{LaFeO}_3/g\text{-C}_3\text{N}_4$  was similar to that of  $g\text{-C}_3\text{N}_4$ . Other  $\text{LaFeO}_3/g\text{-C}_3\text{N}_4$  nanocomposites simultaneously maintained the features of pure  $g\text{-C}_3\text{N}_4$  and  $\text{LaFeO}_3$ . With the increase of  $\text{LaFeO}_3$  content in the nanocomposites, diffraction peaks originating from  $g\text{-C}_3\text{N}_4$  were gradually decreased. These XRD results proved the formation of the  $\text{LaFeO}_3/g\text{-C}_3\text{N}_4$  composites.

TEM images of the 5%- $\text{LaFeO}_3/g\text{-C}_3\text{N}_4$  nanocomposites are shown in Fig. 2. The  $g\text{-C}_3\text{N}_4$  exhibited a thin and irregular nanosheet structure with wrinkles (Fig. 2a).<sup>20</sup> The  $\text{LaFeO}_3$

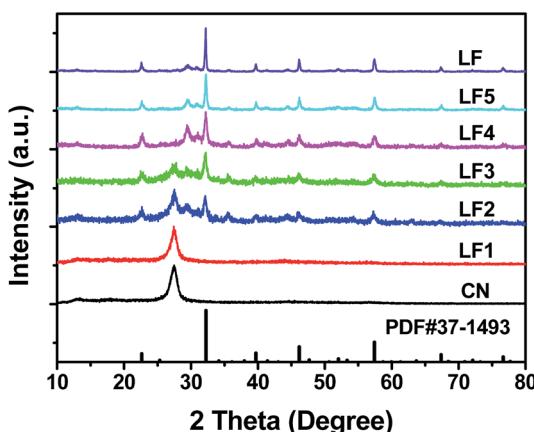


Fig. 1 XRD patterns of  $g\text{-C}_3\text{N}_4$  (CN),  $\text{LaFeO}_3$  (LF) and  $\text{LaFeO}_3/g\text{-C}_3\text{N}_4$  (LF1–LF5).

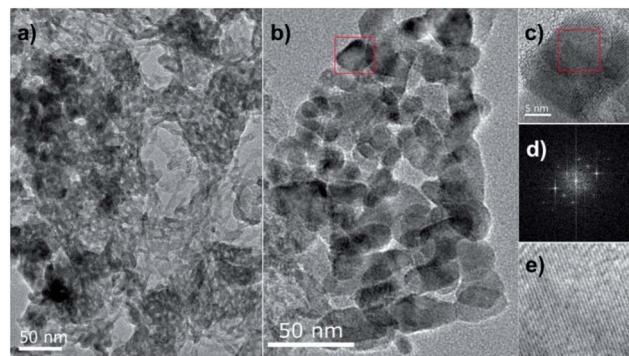


Fig. 2 TEM images (a, b) and HRTEM micrograph (c) of the selected area image of 5%- $\text{LaFeO}_3/g\text{-C}_3\text{N}_4$ , with the fast Fourier transform (FFT) pattern (d) and the corresponding inverse FFT pattern (e).

nanoparticles with a diameter of about 30 nm were distributed in  $g\text{-C}_3\text{N}_4$  to form a heterostructure with interfacial contact. This heterostructure could effectively improve the photocatalytic activity by supplying plenty of reactive sites to adsorb reactant and capture charge carriers. The representative particle in the selected region displayed continuous lattice fringes with a *d*-spacing of 0.2480 nm, corresponding to the (201) planes of the orthorhombic phase of  $\text{LaFeO}_3$  (Fig. 2c–e).

Fig. 3a displays the FTIR spectra of  $g\text{-C}_3\text{N}_4$ ,  $\text{LaFeO}_3$  and  $\text{LaFeO}_3/g\text{-C}_3\text{N}_4$ . The broad absorption peak centered at 3200  $\text{cm}^{-1}$  and the peak at 1640  $\text{cm}^{-1}$  in the spectra of  $g\text{-C}_3\text{N}_4$  and  $\text{LaFeO}_3/g\text{-C}_3\text{N}_4$  were assigned to the stretching and bending vibrations of N–H. With the increase of  $\text{LaFeO}_3$  content in the  $\text{LaFeO}_3/g\text{-C}_3\text{N}_4$  nanocomposites, the intensities of these two absorption peaks were gradually decreased and ultimately disappeared. Fig. 3b shows the corresponding magnification of FTIR spectra in the range of 650  $\text{cm}^{-1}$  to 1800  $\text{cm}^{-1}$ . The absorption at 810  $\text{cm}^{-1}$  was derived from the breathing vibration of *s*-triazine. The absorption peaks at 1575, 1413, 1326 and 1249  $\text{cm}^{-1}$  were ascribed to characteristic C–N stretching of the  $g\text{-C}_3\text{N}_4$  heterocycles.<sup>20,27</sup> The decrease of the intensities revealed the reduced content of  $g\text{-C}_3\text{N}_4$ . However, for 40%- $\text{LaFeO}_3/g\text{-C}_3\text{N}_4$  and 60%- $\text{LaFeO}_3/g\text{-C}_3\text{N}_4$ , these characteristic absorption peaks had disappeared in the same way as for  $\text{LaFeO}_3$ . Combined with the XRD results, these results suggested that the 5%- $\text{LaFeO}_3/g\text{-C}_3\text{N}_4$ , 10%- $\text{LaFeO}_3/g\text{-C}_3\text{N}_4$  and 20%- $\text{LaFeO}_3/g\text{-C}_3\text{N}_4$  samples contained two components of  $\text{LaFeO}_3$  and  $g\text{-C}_3\text{N}_4$ .

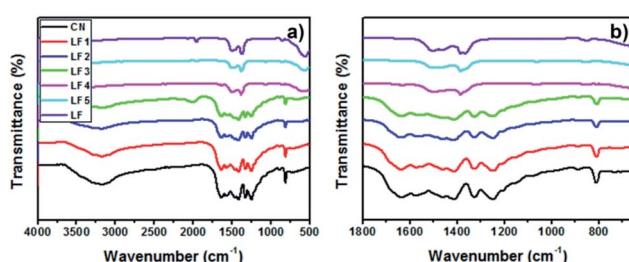


Fig. 3 (a) FTIR spectra of  $g\text{-C}_3\text{N}_4$ ,  $\text{LaFeO}_3$  and  $\text{LaFeO}_3/g\text{-C}_3\text{N}_4$  and (b) the corresponding magnification from 650  $\text{cm}^{-1}$  to 1800  $\text{cm}^{-1}$ .



Fig. 4a shows the XPS survey spectra of 5%-LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>, which proves the presence of carbon (C), iron (Fe), lanthanum (La), nitrogen (N), and oxygen (O), in the nanocomposite. Fig. 4b-f shows the corresponding high resolution spectra of the C 1s, N 1s, Fe 2p, La 3d and O 1s orbitals. The N 1s peak at 398.43 eV was a characteristic peak allocated to sp<sup>2</sup> hybridized nitrogen (C=N-C).<sup>17,22</sup> The peak at 399.96 eV was attributed to the tertiary nitrogen and amino groups with an H atom. In the C 1s spectra, two peaks at 284.82 and 287.98 eV were derived from contaminated carbon on the surface of LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> sample and sp<sup>2</sup> C atoms bonded to amino groups in the triazine cycles. The N 1s and C 1s spectra confirmed the presence of the g-C<sub>3</sub>N<sub>4</sub>. The peak of Fe 2p3/2 and Fe 2p1/2 at 711.1 and 724.9 eV, respectively, were assigned to the core level spectra of Fe<sup>3+</sup> in its oxide form.<sup>28,29</sup> Two La 3d peaks at 836.2 and 852.3 eV were identified as spin-orbit splitting of 3d5/2 and 3d3/2 of La<sup>3+</sup>, respectively, in the oxide.<sup>30</sup> For O 1s spectra, only one peak at 531.6 eV was observed and was ascribed to the oxygen network in the perovskite lattice of LaFeO<sub>3</sub>. The binding energy of O 1s, Fe 2p and La 3d for 5%-LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> shifted to higher values. However, the peaks of N 1s and C 1s shifted to lower binding energies. These shifts of binding energies demonstrated that g-C<sub>3</sub>N<sub>4</sub> was covered closely on the surface of LaFeO<sub>3</sub>, and showed a strong chemical interaction between g-C<sub>3</sub>N<sub>4</sub> and LaFeO<sub>3</sub>.<sup>22</sup>

The UV-vis DRS of g-C<sub>3</sub>N<sub>4</sub>, LaFeO<sub>3</sub> and LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> with different LaFeO<sub>3</sub> contents are shown in Fig. 5a. Pure g-C<sub>3</sub>N<sub>4</sub> displayed an absorption edge at 455 nm, corresponding to a band gap energy ( $E_g$ ) of 2.73 eV (Fig. 5b), which was consistent

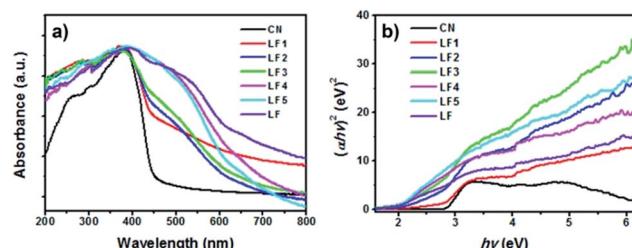


Fig. 5 (a) UV-vis DRS spectra and (b) band gap energies of pure g-C<sub>3</sub>N<sub>4</sub>, LaFeO<sub>3</sub> and LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> heterojunctions.

with previous results found in the literature.<sup>31</sup> LaFeO<sub>3</sub> showed a strong light absorption in the range of 200 to 800 nm. LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> samples had broader absorptions and longer absorption edge wavelengths than that of g-C<sub>3</sub>N<sub>4</sub>, which was attributed to the strong interaction between LaFeO<sub>3</sub> and g-C<sub>3</sub>N<sub>4</sub> in the heterostructures.

The band gap energies of g-C<sub>3</sub>N<sub>4</sub>, LaFeO<sub>3</sub> and LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> were calculated using the Tauc relationship. Fig. 5b shows the Tauc plots of  $(ahv)^2$  versus  $h\nu$ .<sup>32</sup> The band gap energies of g-C<sub>3</sub>N<sub>4</sub> and LaFeO<sub>3</sub> were 2.73 and 1.92 eV, respectively. The  $E_g$  of LaFeO<sub>3</sub> was slightly smaller than the reported values found in the literature (2.0 eV).<sup>33</sup> The  $E_g$  values of LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> with different LaFeO<sub>3</sub> content from 5% to 60% were 2.69, 2.55, 2.45, 1.92 and 1.92 eV. These results indicated that the  $E_g$  reduced as the LaFeO<sub>3</sub> content increased from 5% to 20%. However, the 40%-LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> and 60%-LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> had the same  $E_g$  values as pure LaFeO<sub>3</sub>. It was suggested that LaFeO<sub>3</sub> and g-C<sub>3</sub>N<sub>4</sub> had not formed the heterostructures in 40%-LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> and 60%-LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> samples efficiently, which was also demonstrated by the XRD and FTIR results. The most probable causes proposed are given next. In this work, 60% or 40% of g-C<sub>3</sub>N<sub>4</sub> was heated with iron nitrate and lanthanum nitrate for 2 h at 450 °C to prepare the samples. This heat treatment procedure could give rise to volatilization of g-C<sub>3</sub>N<sub>4</sub>, which decreased the actual amounts of g-C<sub>3</sub>N<sub>4</sub> in the samples. The relative higher LaFeO<sub>3</sub> content and lack of g-C<sub>3</sub>N<sub>4</sub> lead to self-agglomeration of LaFeO<sub>3</sub>, and consequently the heterostructures could not be formed efficiently.

The CB and VB edge positions of the g-C<sub>3</sub>N<sub>4</sub> and LaFeO<sub>3</sub> were calculated using the following two equations:

$$E_{VB} = X - E_e + 0.5E_g$$

$$E_{CB} = E_{VB} - E_g$$

where  $E_{VB}$ ,  $X$ ,  $E_e$ ,  $E_{CB}$  represented the VB edge potential, the absolute electronegativity, the energy of free electrons on the H<sub>2</sub> scale (4.5 eV) and the CB edge potential, respectively. The  $E_g$  was obtained from the UV-vis DRS. The  $E_{VB}$  and  $E_{CB}$  were 1.54 and -1.28 eV for g-C<sub>3</sub>N<sub>4</sub>, and 2.03 and 0.11 eV for LaFeO<sub>3</sub>, which were similar to the reported  $E_{VB}$  and  $E_{CB}$  of +1.57 and -1.12 eV for g-C<sub>3</sub>N<sub>4</sub> (ref. 32) and 2.115 and 0.025 eV for LaFeO<sub>3</sub>.<sup>23</sup>

To test the photocatalytic activity of the LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> heterojunctions, photocatalytic degradation of MB under 3 W LED irradiation was implemented. The absorption peak at 664 nm

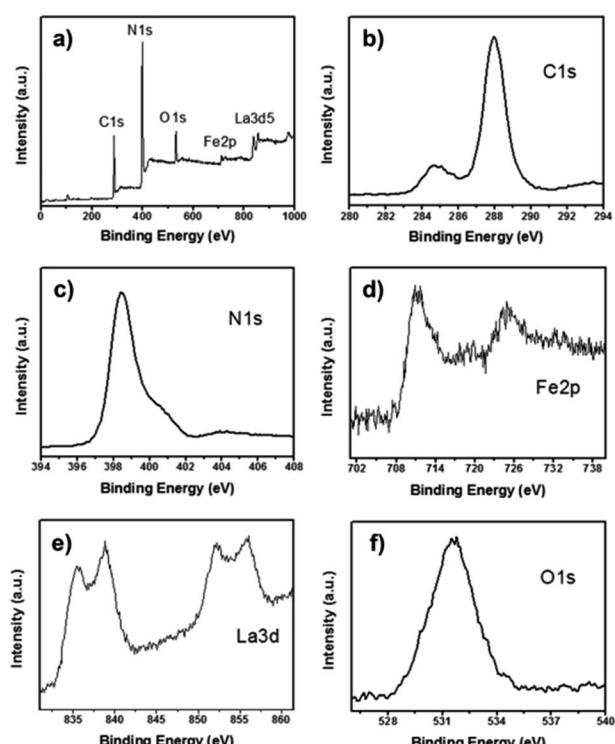


Fig. 4 (a) Survey XPS spectra and high resolution spectra of C (b), N (c), Fe (d), La (e) and O (f) of 5%-LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>.



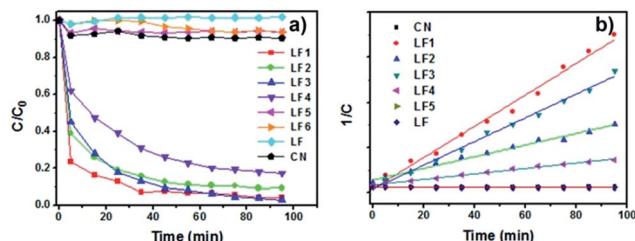


Fig. 6 (a) Photocatalytic degradation of MB over  $\text{g-C}_3\text{N}_4$ ,  $\text{LaFeO}_3$  and  $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  under 3 W LED irradiation and (b) corresponding second-order kinetics plots.

was selected to monitor the degradation of MB. As shown in Fig. 6a, only 3–10% of MB was degraded over the  $\text{g-C}_3\text{N}_4$ ,  $\text{LaFeO}_3$  and 60% $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  under 3 W LED irradiation for 120 min. By comparison, the degradation rate of MB on the  $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  samples with 5%, 10% and 20%  $\text{LaFeO}_3$  was significantly enhanced. The 5% $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  heterojunction exhibited the highest photocatalytic activity, over which more than 95% of MB was degraded after irradiation for 120 min.

The second-order kinetics model was used to fit the photocatalytic degradation data. The results are shown in Fig. 6b. The 5% $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  heterojunction nanocomposite exhibited the highest photocatalytic activity. Its second-order reaction rate constant was around 49.7 and 86.9 times more than pure  $\text{g-C}_3\text{N}_4$  and  $\text{LaFeO}_3$  (Fig. 7), respectively. On the basis of the previous XRD, FTIR and DRS measurements, this photocatalytic degradation result demonstrated that  $\text{LaFeO}_3$  and  $\text{g-C}_3\text{N}_4$  had effectively formed the heterojunction composites. The strong interfacial interaction between  $\text{LaFeO}_3$  and  $\text{g-C}_3\text{N}_4$  in the composites could improve the transfer of the photoproduced charge and enhance their photocatalytic performances.<sup>17</sup> This strong interfacial interaction between  $\text{LaFeO}_3$  and  $\text{g-C}_3\text{N}_4$  in the composites was investigated using simple mechanical mixing of  $\text{LaFeO}_3$  and  $\text{g-C}_3\text{N}_4$  in the same ratio as the composites and then using it to degrade MB under 3 W LED irradiation (curve LF6 in Fig. 6a). The photocatalytic activity of the 5% $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  heterojunction nanocomposite was much higher than that of a simple mechanical mixture. This result indicated the formation of intimately contacted interfaces in the heterojunction

nanocomposite, which is also displayed in the TEM images (Fig. 2).

The stability of the 5% $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  heterojunction nanocomposite was studied by recycling 5% $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  under the same photocatalytic degradation conditions for degrading MB. As shown in Fig. 8a, after four recycles, no significant changes were observed. This result proved that the  $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  nanocomposites possessed a relatively high stability for photocatalytic degradation of MB.

These photocatalytic degradation results revealed that the intimately contacted interfaces between  $\text{LaFeO}_3$  and  $\text{g-C}_3\text{N}_4$  in the heterojunction nanocomposite resulted in the high transfer and separation of photoproduced charges at the interface of the heterostructures.<sup>17</sup> This assumption was further confirmed by the transient photocurrent responses of the  $\text{g-C}_3\text{N}_4$ ,  $\text{LaFeO}_3$  and 5% $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$ . The photocurrent-time ( $I-t$ ) curves were obtained by intermittently cutting off the visible light irradiation (Fig. 8b). The photocurrent value of 5% $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  was about two times higher than pure  $\text{LaFeO}_3$  and  $\text{g-C}_3\text{N}_4$ . This was attributed to the efficient separation of the photoproduced charges in space and consequently reduced the carrier recombination.

To investigate which type of reactive species, including hydrogen peroxide ( $\text{H}_2\text{O}_2$ ),  $\text{h}^+$ ,  $\text{e}^-$ ,  $\cdot\text{OH}$  and  $\cdot\text{O}_2^-$ , were involved in the photocatalytic degradation of MB over the 5% $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$ , ammonium oxalate (AO), EDTA-2Na, isopropanol (IPA),  $\text{NO}_3^-$ , catalase (CAT) and benzoquinone (BQ) were added to the degradation solutions.<sup>34</sup> AO and EDTA-2Na were the scavenger of  $\text{h}^+$ , IPA was the scavenger of  $\cdot\text{OH}$  radicals,  $\text{NO}_3^-$  acted as the  $\text{e}^-$  scavenger, CAT was the  $\text{H}_2\text{O}_2$  scavenger, and BQ acted as the  $\cdot\text{O}_2^-$  radical scavenger. As shown in Fig. 9a, no obvious changes were found when  $\text{NO}_3^-$  and CAT were added to the solutions, indicating that  $\text{e}^-$  and  $\text{H}_2\text{O}_2$  were not the active species in the photocatalytic degradation of MB. The addition of IPA significantly decreased the photodegradation rate. It was revealed that  $\cdot\text{OH}$  radicals were the dominant species for MB photodegradation. As shown in Fig. 9a, BQ also decreased the photodegradation rate slightly, demonstrating the minor role of  $\cdot\text{O}_2^-$ . However, in the presence of AO and EDTA-2Na, the photocatalytic degradation rate was remarkably increased. Only a few reports in the literature have reported the increase of photodegradation rate by the addition of EDTA-2Na in to the solution.<sup>35,36</sup> It was considered that the addition of EDTA-2Na

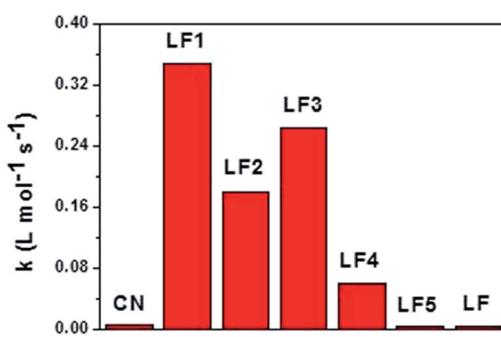


Fig. 7 The second-order reaction rate constants of photocatalytic degradation of MB on  $\text{g-C}_3\text{N}_4$ ,  $\text{LaFeO}_3$  and  $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  under 3 W LED irradiation.

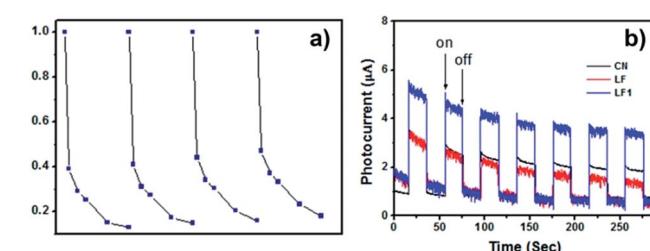


Fig. 8 (a) Stability study of photocatalytic degradation of MB over the 5% $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  heterojunction under 3 W LED irradiation, and (b) transient photocurrent responses of the  $\text{g-C}_3\text{N}_4$ ,  $\text{LaFeO}_3$  and 5% $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  heterojunction.

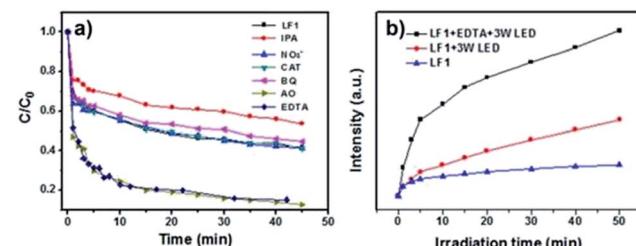


Fig. 9 (a)  $C/C_0$  variation of the photocatalytic degradation of MB over the 5%-LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> in the presence of different scavengers under 3 W LED irradiation, and (b) HTA fluorescence intensities over the 5%-LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> monitored at the peak of 424 nm under 3 W LED irradiation or under indoor light irradiation.

trapped the  $\text{h}^+$  and promoted the separation rate of the  $\text{e}^-/\text{h}^+$  pairs, and consequently more 'OH radicals were achieved. Therefore the photodegradation activity was improved.

To confirm the increase of 'OH radicals in the presence of EDTA-2Na, TA was used to react with 'OH to form 2-hydroxy-terephthalic acid (HTA), which possessed strong fluorescence.<sup>17,37</sup> The results of the HTA experiment are shown in Fig. 9b. It shows that the peak intensities of HTA were enhanced significantly when EDTA-2Na was added in to the degradation solution. Fig. 9b also revealed that more 'OH radicals were generated under 3 W LED irradiation than were produced under indoor light irradiation. Different dosages of EDTA-2Na were added to the degradation solution to test the influence of EDTA-2Na concentration on the photodegradation of MB over the 5%-LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>. The results are shown in Fig. 10. It indicated that by increasing the amount of EDTA-2Na, the photocatalytic degradation rate was enhanced dramatically. More than 95% of MB was degraded by adding 25 mg EDTA-2Na to 100 mL MB solution and then irradiated with a 3 W LED for 15 min.

The photocatalytic H<sub>2</sub> evolution activity of g-C<sub>3</sub>N<sub>4</sub>, LaFeO<sub>3</sub> and LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> heterojunctions in aqueous solution containing 10 vol% TEOA under 300 W Xe lamp irradiation were studied. As shown in Fig. 11, the H<sub>2</sub> evolution rate was negligible over the pure LaFeO<sub>3</sub>, g-C<sub>3</sub>N<sub>4</sub>, 40%-LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> and 60%-LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>. The photocatalytic H<sub>2</sub> evolution activity of

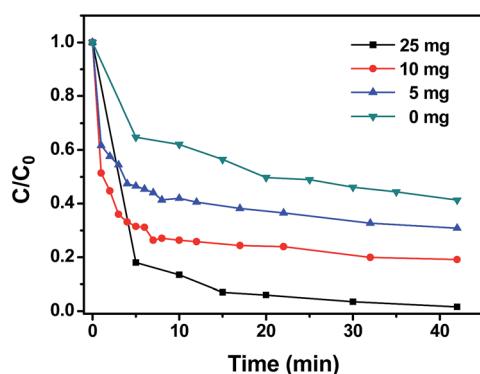


Fig. 10 The influence of different dosages of EDTA-2Na on the photodegradation of MB over the 5%-LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> with 3 W LED irradiation.

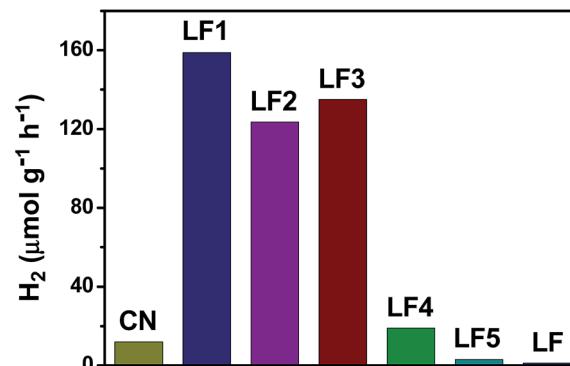


Fig. 11 The photocatalytic hydrogen evolution activity under 300 W Xe lamp irradiation over g-C<sub>3</sub>N<sub>4</sub>, LaFeO<sub>3</sub> and LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> nanocomposites.

LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> containing 5, 10 and 20 wt% of LaFeO<sub>3</sub> was remarkably enhanced. The 5%-LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> heterojunction nanocomposite in particular had a maximum H<sub>2</sub> evolution rate of 158  $\mu\text{mol g}^{-1} \text{h}^{-1}$ .

The recyclability of 5%-LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> nanocomposites was tested using five cycles of photocatalytic H<sub>2</sub> evolution under identical reaction conditions. As shown in Fig. 12, 450  $\mu\text{mol}$  of H<sub>2</sub> was generated after irradiation for 3 h under visible light in the first run. The amount of H<sub>2</sub> was finally reduced to 395  $\mu\text{mol}$  in the fifth run. Only 12.2% H<sub>2</sub> production was lost after the fifth photocatalytic experiment. The results indicated that 5%-LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> was quite stable for photocatalytic H<sub>2</sub> production under visible light irradiation.

On the basis of the results of the 'OH trapping experiments, the potential route of chemical reactions responsible for the generation of 'OH was determined. The photocatalytic procedure over LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> was in accordance with the features of a Z-scheme photocatalysis system. Under visible light irradiation, both LaFeO<sub>3</sub> and g-C<sub>3</sub>N<sub>4</sub> were excited to obtain photoinduced electron and hole pairs. The photoexcited electrons of LaFeO<sub>3</sub> and g-C<sub>3</sub>N<sub>4</sub> transited from the VB to the CB and the holes were left on the VB. The electrons from the CB of LaFeO<sub>3</sub>

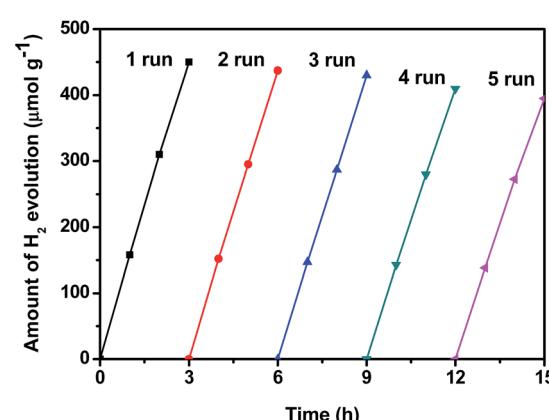


Fig. 12 Recyclability of the H<sub>2</sub> evolution test over 5%-LaFeO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> nanocomposites under 300 W Xe lamp irradiation.



transferred to the VB of  $\text{g-C}_3\text{N}_4$  using the solid–solid intimately contacted interfaces, where the electrons and holes were recombined and thus improved the separation of the photo-generated electrons and holes of  $\text{g-C}_3\text{N}_4$ . Then the excited electrons reduced  $\text{O}_2$  adsorbed on the surface of  $\text{g-C}_3\text{N}_4$  to produce  $\cdot\text{O}_2^-$ . The  $\cdot\text{O}_2^-$  reacted with  $\text{H}_2\text{O}$  to achieve  $\cdot\text{OH}$ . Finally, MB was degraded by  $\cdot\text{OH}$  and  $\cdot\text{O}_2^-$ . As shown in Fig. 13, the CB edge potential of  $\text{LaFeO}_3$  was more positive than the standard redox potential of  $\text{H}^+/\text{H}_2$ , which cannot reduce  $\text{H}^+$  to form  $\text{H}_2$ .  $\text{H}^+$  was subsequently reduced by the CB electrons of  $\text{g-C}_3\text{N}_4$  to generate  $\text{H}_2$ . The holes on the VB of  $\text{LaFeO}_3$  reacted with TEOA or EDTA-2Na (as the sacrificial agent) to promote the separation of the photogenerated electrons and holes, which resulted in the improvement of the photocatalytic performance.

The photocatalytic  $\text{H}_2$  evolution activity was consistent with the photodegradation rate of MB. The excess  $\text{LaFeO}_3$  (above 20 wt% in the nanocomposite) decreased the  $\text{H}_2$  evolution rate. Fig. 13 clearly shows the correlation between the higher  $\text{H}_2$  evolution rate and higher amount of  $\cdot\text{OH}$  generated over the  $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  heterojunction nanocomposites. It possessed the features of a Z-scheme system. Under visible light irradiation, both  $\text{LaFeO}_3$  and  $\text{g-C}_3\text{N}_4$  were excited to obtain photoinduced electron and hole pairs. The photoexcited electrons of  $\text{LaFeO}_3$  and  $\text{g-C}_3\text{N}_4$  transited from the VB to the CB and the holes were left on the VB. The electrons from the CB of  $\text{LaFeO}_3$  were transferred to the VB of  $\text{g-C}_3\text{N}_4$  by the solid–solid intimately contacted interfaces, where the electrons and holes were recombined and thus improved the separation of the photo-generated electrons and holes of  $\text{g-C}_3\text{N}_4$ . Then the excited electrons on the CB of  $\text{g-C}_3\text{N}_4$  efficiently generated the  $\cdot\text{OH}$ . Meanwhile, the improved separation of the photo-generated electron and hole of  $\text{g-C}_3\text{N}_4$  of this Z-scheme system also efficiently enhanced the  $\text{H}_2$  evolution rate. The holes on the VB of  $\text{LaFeO}_3$  reacted with TEOA or EDTA-2Na (as the sacrificial agent) to promote the separation of photo-generated electrons and holes, which resulted in the improvement of the photocatalytic performance.

The results revealed that the optimum concentration of  $\text{LaFeO}_3$  was 5 wt% for the maximum photocatalytic  $\text{H}_2$  evolution and photodegradation rate. Too high a content of  $\text{LaFeO}_3$  was

adverse for the efficacious formation of heterojunction nanocomposites and also gave reduced photocatalytic  $\text{H}_2$  evolution and photodegradation rate.  $\text{LaFeO}_3$  and  $\text{g-C}_3\text{N}_4$  did not form efficiently the heterostructures in 40%- $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  and 60%- $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  samples, which was demonstrated by the XRD and FTIR results. This led to the 40%- $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  and 60%- $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  samples having a much lower degradation rate.

5%- $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$ , 10%- $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  and 20%- $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  formed intimately contacted interfaces in the nanocomposites, which had higher degradation rate. Among these nanocomposites, 5%- $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  gave the optimum results for MB degradation. The absorption edges of  $\text{g-C}_3\text{N}_4$ ,  $\text{LaFeO}_3$  and  $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  are compared in Fig. 5a. This revealed that  $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  samples had broader absorptions and longer absorption edge wavelengths than that of  $\text{g-C}_3\text{N}_4$ . The absorption intensity of 5%- $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  in the visible light region was much higher than that of 10%- $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  and 20%- $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$ , which indicated that the 5%- $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  produced a greater amount of photogenerated electron–hole pairs.<sup>34</sup> Based on the UV-vis DRS spectra, XRD and FTIR results, it could be deduced that the greater amount of photogenerated electron–hole pairs and the efficient charge separation of the Z-scheme photocatalytic system gave the 5%- $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  heterojunction, the highest MB degradation and  $\text{H}_2$  evolution rate.

## 4. Conclusion

The new type of  $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  heterostructures were successfully prepared using a two-step method. It possessed the features of a Z-scheme photocatalysis system. The photoexcited electrons on the CB of  $\text{LaFeO}_3$  were transferred to the VB of  $\text{g-C}_3\text{N}_4$  by the solid–solid intimately contacted interfaces, where the electrons and holes were recombined and thus improved the separation of photo-generated electrons and holes of  $\text{g-C}_3\text{N}_4$ .  $\text{LaFeO}_3$  and  $\text{g-C}_3\text{N}_4$  also effectively formed intimately contacted interfaces. This strong interfacial interaction could improve the transfer photoproduced charge and enhance their photocatalytic performances. Thus,  $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  heterostructures had a higher  $\text{H}_2$  evolution rate and a higher amount of  $\cdot\text{OH}$  than pure  $\text{LaFeO}_3$  and  $\text{g-C}_3\text{N}_4$ . The construction of  $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  heterostructures was demonstrated to be the effective strategy to improve the photocatalytic property. The 5%- $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  exhibited the highest photodegradation and water splitting rate. More than 95% of MB was degraded after 15 min in the presence of 25 mg EDTA-2Na in 100 mL of MB solution with irradiation using a 3 W LED. The photogenerated  $\cdot\text{OH}$  radicals were the main oxidative species for the degradation of MB. Its second-order reaction rate constant was about 49.7 and it was 86.9 times more than that of pure  $\text{g-C}_3\text{N}_4$  and  $\text{LaFeO}_3$ . The 5%- $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  heterojunction nanocomposite had a maximum  $\text{H}_2$  evolution rate of  $158 \mu\text{mol g}^{-1} \text{h}^{-1}$ . This research shows an easy approach to manufacturing  $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  heterojunction nanocomposites to promote the exploitation of solar energy.

## Conflicts of interest

There are no conflicts to declare.

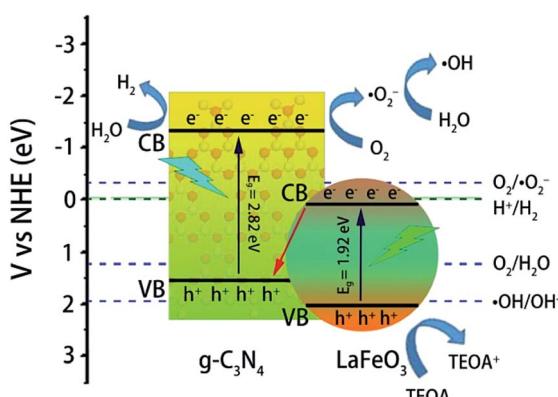


Fig. 13 Possible photocatalytic degradation of MB and  $\text{H}_2$  evolution mechanism over  $\text{LaFeO}_3/\text{g-C}_3\text{N}_4$  heterojunctions under visible light irradiation.



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## Notes and references

- 1 S. Cao, J. Low, J. Yu and M. Jaroniec, *Adv. Mater.*, 2015, **27**, 2150.
- 2 A. Fujishima and K. Honda, *Nature*, 1972, **238**, 37.
- 3 J. Yu, J. Low, W. Xiao, P. Zhou and M. Jaroniec, *J. Am. Chem. Soc.*, 2014, **136**, 8839.
- 4 X. Wang, M. Liao, Y. Zhong, J. Y. Zheng, W. Tian, T. Zhai, C. Zhi, Y. Ma, J. Yao, Y. Bando and D. Golberg, *Adv. Mater.*, 2012, **24**, 3421.
- 5 W. C. Huang, L. M. Lyu, Y. C. Yang and M. H. Huang, *J. Am. Chem. Soc.*, 2012, **134**, 1261.
- 6 Z. Zou, J. Ye, K. Sayama and H. Arakawa, *Nature*, 2001, **414**, 625.
- 7 M. Liu, W. You, Z. Lei, G. Zhou, J. Yang, G. Wu, G. Ma, G. Luan, T. Takata, M. Hara, K. Domen and C. Li, *Chem. Commun.*, 2004, 2192.
- 8 Y. Hu, X. Gao, L. Yu, Y. Wang, J. Ning, S. Xu and X. W. Lou, *Angew. Chem., Int. Ed.*, 2013, **52**, 5636.
- 9 X. Zhou, Q. Xu, W. Lei, T. Zhang, X. Qi, G. Liu, K. Deng and J. Yu, *Small*, 2014, **10**, 674.
- 10 Y. Wang, X. Wang and M. Antonietti, *Angew. Chem., Int. Ed.*, 2012, **51**, 68.
- 11 E. Kroke, *Angew. Chem., Int. Ed.*, 2014, **53**, 11134.
- 12 A. Thomas, A. Fischer, F. Goettmann, M. Antonietti, J. Müller, R. Schlögl and J. M. Carlsson, *J. Mater. Chem.*, 2008, **18**, 4893.
- 13 Y. Zheng, J. Liu, J. Liang, M. Jaroniec and S. Z. Qiao, *Energy Environ. Sci.*, 2012, **5**, 6717.
- 14 J. Zhang, X. Chen, K. Takanabe, K. Maeda, K. Domen, J. D. Epping, X. Fu, M. Antonietti and X. Wang, *Angew. Chem., Int. Ed.*, 2010, **49**, 441.
- 15 Y. Zheng, L. Lin, X. Ye, F. Guo and X. Wang, *Angew. Chem., Int. Ed.*, 2014, **53**, 11926.
- 16 G. Liu, P. Niu, C. Sun, S. C. Smith, Z. Chen, G. Q. Lu and H. Cheng, *J. Am. Chem. Soc.*, 2010, **132**, 11642.
- 17 J. Fu, Y. Tian, B. Chang, F. Xi and X. Dong, *J. Mater. Chem.*, 2012, **22**, 21159.
- 18 S. Yang, W. Zhou, C. Ge, X. Liu, Y. Fang and Z. Li, *RSC Adv.*, 2013, **3**, 5631.
- 19 W. Tian, Q. Shen, N. Li and J. Zhou, *RSC Adv.*, 2016, **6**, 25568.
- 20 D. Jiang, J. Li, C. Xing, Z. Zhang, S. Meng and M. Chen, *ACS Appl. Mater. Interfaces*, 2015, **7**, 19234.
- 21 C. Zhao, G. Tan, J. Huang, W. Yang, H. Ren and A. Xia, *ACS Appl. Mater. Interfaces*, 2015, **7**, 23949.
- 22 F. Li, S. Liu, Y. Xue, X. Wang, Y. Hao, J. Zhao, R. Liu and D. Zhao, *Chem.-Eur. J.*, 2015, **21**, 10149.
- 23 J. Yang, R. Hu, W. Meng and Y. Du, *Chem. Commun.*, 2016, **52**, 2620.
- 24 P. Li, X. Hu, L. Zhang, H. Dai and L. Zhang, *Nanoscale*, 2011, **3**, 974.
- 25 X. Ren, H. Yang, S. Gen, J. Zhou, T. Yang, X. Zhang, Z. Cheng and S. Sun, *Nanoscale*, 2016, **8**, 752.
- 26 Q. Yu, X. Meng, T. Wang, P. Li, L. Liu, K. Chang, G. Liu and J. Ye, *Chem. Commun.*, 2015, **51**, 3630.
- 27 L. M. Sun, Y. Qi, C. J. Jia, Z. Jin and W. L. Fan, *Nanoscale*, 2014, **6**, 2649.
- 28 M. Markova-Velichkova, T. Lazarova, V. Tumbalev, G. Ivanov, D. Kovacheva, P. Stefanov and A. Naydenov, *Chem. Eng. J.*, 2013, **231**, 236.
- 29 H. Wu, R. Hu, T. Zhou, C. Li, W. Meng and J. Yang, *CrystEngComm*, 2015, **17**, 3859.
- 30 S. Thirumalairajan, K. Girija, N. Y. Hebalkar, D. Mangalaraj, C. Viswanathan and N. Ponpandian, *RSC Adv.*, 2013, **3**, 7549.
- 31 H. Zhang, L. Guo, L. Zhao, B. Wan and Y. Yang, *J. Phys. Chem. Lett.*, 2015, **6**, 958.
- 32 H. Liu, Z. Jin, Z. Xu, Z. Zhang and D. Ao, *RSC Adv.*, 2015, **5**, 97951.
- 33 L. Jing, Y. Qu, H. Su, C. Yao and H. Fu, *J. Phys. Chem. C*, 2011, **115**, 12375.
- 34 H. Li, Y. Liu, X. Gao, C. Fu and X. Wang, *ChemSusChem*, 2015, **8**, 1189.
- 35 S. Hu, L. Ma, J. You, F. Li, Z. Fan, F. Wang, D. Liu and J. Gui, *RSC Adv.*, 2014, **4**, 21657.
- 36 J. Zhang, S. Hu and Y. Wang, *RSC Adv.*, 2014, **4**, 62912.
- 37 R. Marschall, A. Mukherji, A. Tanksale, C. Sun, S. C. Smith, L. Wang and G. Q. Lu, *J. Mater. Chem.*, 2011, **21**, 8871.

