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A 2 micron passively Q-switched bulk state pulsed laser based on WS₂

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We report a 2 micron passively Q-switched bulk state pulsed laser with a pulse of 430 ns based on graphene-like 2D WS₂ nanosheets. To the best of our knowledge, it is the shortest pulse compared with other bulk state lasers based on WS₂ saturable absorbers at this waveband. The WS₂ nanosheets are prepared through a facile hydrothermal reaction. Its max output power and pulse energy are 488 mW and 5.422 μJ, respectively. The wavelength is distributed from 1934 to 1942 nm. These results reveal that WS₂ is a promising nanomaterial in the fields of photonics and electronics.

1. Introduction

In recent years, owing to the strong absorption of water molecules, 2 μm waveband pulsed lasers have attracted increasing attention because of their vital application in the areas of laser radar systems, medical surgical operations, and environmental protection. Passively Q-switching technology based on saturable absorbers (SAs) has been demonstrated to be a facile and efficient method to realize nanosecond pulsed lasers.^{1–10} It is quite natural that the SA plays a significant role in this process.

Transition element-doped host materials^{1–3} and semiconductor saturable absorber mirrors (SESAMs)^{4,5} are the important traditional SA materials. However, the development of traditional SA materials is hampered by their narrow absorption band, complex manufacturing processes and high cost. Fortunately, as a novel SA material, the appearance of graphene brings a new idea to explore more valuable SA materials.¹¹ This 2D-structured layered carbon material possesses a Dirac-like gapless electronic band structure, resulting in broadband light absorption.¹¹ Besides, due to its simple manufacturing processes and low cost, graphene has drawn much more attention to realize pulsed lasers at broadband.^{6,12–16} Inspired by the 2D structure of graphene, a new family of graphene-like 2D materials, including transition metal dichalcogenides (TMDs),^{7,17,18} topological insulators (TIs)⁸ and black phosphorus (BP),⁹ has been developed to realize pulsed lasers rapidly. As a representative of the family of TMDs, each WS₂

layer consists of a single plane of W atoms held between two planes of S atoms by strong covalent bonds, and these single layers are stacked together by weak Van der Waals forces.¹⁹ It is interesting that its bandgap shifts from indirect (1.3 eV) for bulk state to direct (2.1 eV) for monolayer flakes.²⁰ Hence, its bandgap could be engineered by manipulating the thickness or atomic defects, resulting in broadband saturable absorption directly.^{18,21–24} In 2015, Mao²¹ demonstrated a stable mode-locked fiber laser operation by depositing WS₂ nanosheets on an Er-doped fiber for the first time. Later on, Tan²² realized a WS₂-based Q-switched waveguide laser in a Nd:YAG crystal. In addition, Luan²³ first reported its nonlinear saturable absorption property at around 2 μm and realized 660 ns bulk state laser pulses. In order to have a wide variety of practical applications, producing even shorter pulses with higher laser power is still urgently needed and worthy of further investigation.

In this paper, we demonstrate a Q-switched bulk laser of 430 ns with a WS₂ SA material at about 2 μm. As far as we know, it has the narrowest pulse width compared with those of other 2 μm bulk lasers based on WS₂. The WS₂ was prepared through a facile hydrothermal reaction. A better quality of WS₂ SA film was prepared by a spin coating technique. Its phase, morphology and absorbance at this waveband were investigated. Our results indicated the WS₂ is a type of promising nanomaterial in the fields of photonics and electronics.

2. Experimental section

The few layered WS₂ was synthesized by a facile one-step hydrothermal reaction method. All of the chemicals used in our experiments were of analytical purity and were used without further purification. In a typical experiment, 0.5116 g WCl₆ and 0.4846 g C₂H₅NS were dissolved in 16 mL deionized water. After violent stirring for 1 h using a magnetic stirrer at room

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Fig. 1 Schematic of the experimental setup.

temperature, the solution was transferred into a 20 mL Teflon-lined stainless steel autoclave and then heated at 538 K for 24 h. After cooling naturally, the products were washed with absolute

ethyl alcohol three times. Finally, the products were dried in air at 60 °C for 6 h.

To compress the volume of the laser setup, a compact plane-plane straight cavity was chosen in our experiment. Fig. 1 shows the schematic experimental setup. The gain medium was a $3 \times 3 \times 5 \text{ mm}^3$ a-cut Tm:YAP crystal. It was wrapped in indium foil and mounted in a bronze holder with water-cooling to 293 K. It was end pumped by a fiber-coupled continuous wave diode laser at 798 nm with a core diameter of 100 μm . The pump light was then focused into the gain medium by a focusing system. The input mirror (IM) was a plane mirror coated for

Fig. 2 (a) XRD diffraction pattern, (b and c) SEM images, (d) TEM image, (e) absorption spectrum from 1900 to 2000 nm, and (f) nonlinear transmittance of the WS_2 nanosheets.

antireflection (AR) at around 798 nm and high-reflection (HR) at around 1980 nm. The output coupler (OC) was also a plane mirror with 3.5% transmittance at 1980 nm. To reduce the insert loss, the WS₂ dispersion was directly spin-coated on the OC.

3. Results and discussion

The phase of the products was tested by X-ray powder diffraction (XRD). As indicated in Fig. 2(a), the products display a pure phase of WS₂ with a space group *P6₃/mmc* (194), in good agreement with the standard card JCPDF #08-0273. To obtain uniform WS₂ nanosheets, a certain amount of the products was dispersed in absolute ethyl alcohol and centrifuged at 1500 rpm for 20 min. The morphology of the upper supernatant was studied by scanning electron microscopy (SEM). Fig. 2(b) shows that an abundance of WS₂ nanosheets were obtained. Fig. 2(c) was also obtained by SEM with higher resolution. From this sub-micro sized 0D aggregate, we can clearly ascertain the 2D nature of WS₂ and its thickness was measured to be about 5 nm. Fig. 2(d) shows the typical transmission electron microscope image (TEM). It reveals the stacking of the WS₂ (002) layers with an interplanar spacing of 0.62 nm. The absorbance characteristics were explored through the infrared spectra. As shown in Fig. 2(e), WS₂ possessed broadband absorption characteristics in the wavelength ranging from 1900 to 2000 nm. The absorbance at this waveband was around 0.2, indicating that the WS₂ may be applied as a SA for this waveband. We measured its SA behavior through a pump-probe measurement. To obtain the nonlinear transmission curve, we employed a homemade acousto-optic Q-switched bulk laser using Tm:YAP as the gain medium. The laser power before and after the WS₂ was recorded. The corresponding result is shown in Fig. 2(f). By fitting the curve with the equation

$$T(I) = 1 - \Delta T \times \exp(-I/I_{\text{sat}}) - T_{\text{ns}} \quad (1)$$

where $T(I)$ is the transmittance, ΔT is the modulation depth, I is the input intensity, I_{sat} is the saturation intensity, T_{ns} is the

Fig. 4 (a) The Q-switching pulse train and (b) a typical single pulse profile at an absorbed pump power of 5.3 W.

nonsaturable loss, the saturation intensity I_{sat} and the modulation depth ΔT are extracted as 3.66 MW cm^{-2} and 20.56%, respectively. The nonsaturable loss is 15.70%, and considering that the transmittance of the quartz substrate is about 93.50% at $1.9 \mu\text{m}$, the loss of the WS₂ is about 9.20%.

Initially, we investigated the performance of the continuous-wave Tm:YAP laser without WS₂. When the absorbed pump power was tuned from the threshold to 6 W, no self Q-switching was observed during this course. Then the WS₂ dispersion was directly spin-coated onto the OC and it was placed as shown in Fig. 1. Once the absorbed pump power exceeded the threshold of 3.54 W, the stable pulsed laser train began to be realized. Fig. 3(a) shows the dependence of the pulse width and repetition rate on the absorbed pump power. When the absorbed pump power was at the threshold of 3.54 W, the pulse width was at the maximum of 1350 ns. Once the absorbed pump power

Fig. 3 (a) Evolutions of repetition rate and pulse width with absorbed pump power. (b) Evolutions of output power and pulse energy with absorbed pump power.

Fig. 5 The laser spectra at an absorbed pump power of 5.3 W.

was increased, the pulse width of the pulse train became continuously narrower. When the absorbed pump power exceeded 5.3 W, the pulse train began to become unstable. The stable pulse train appeared again once the absorbed pump power was below 5.3 W. As a consequence, the WS₂ was undamaged at this irradiation intensity. The pulse width reached a minimum of 430 ns when the absorbed pump power was 5.3 W. As shown in Fig. 3, the repetition rate increased almost linearly from 64 to 90 kHz. We also recorded the output power. The output power increased from a minimum of 248 mW to a maximum of 488 mW. The corresponding result is shown in Fig. 3(b). According to these data, the pulse energy was obtained as shown in Fig. 3(b). A maximum pulse energy of 5.422 μJ could be deduced.

Fig. 4(a) shows the typical pulse train when the absorbed pump power was 5.3 W. To investigate the stability of Q-switching, we further evaluated the quality of pulse-amplitude equalization. It could be characterized by clock amplitude jitter (CAJ). The CAJ is defined as the ratio of the standard deviation σ to the mean value (M) of the intensity histogram at the pulse peak intensity, as described below:

$$\text{CAJ} = \sigma/M \times 100\% \quad (2)$$

The CAJ of the pulse train in Fig. 4(a) was calculated to be 9.54%, revealing good intensity stability. Fig. 4(b) is a typical

Table 1 Comparison of this 2 μm passively Q-switched bulk laser performance with other SAs

Type of SA	Min pulse width/ns	Max pulse energy/μJ	Repetition rate/kHz	Ref.
Graphene	285	1.6	190	25
Gold nanorod	796	4.9	77	26
Black phosphorus	1780	7.84	19.25	27
MoS ₂	800	2.08	48.09	28
WS ₂	660	17.1	63	23
WS ₂	430	5.42	90	This work

single pulse profile. Simultaneously, the laser spectrum was recorded. As shown in Fig. 5, the wavelength was distributed from 1934 to 1942 nm.

In the end, we compared our results with those of 2 μm bulk state lasers with other SAs. As shown in Table 1, we listed the min pulse width, max pulse energy and repetition rate. Based on these data the laser results, especially the min pulse width, could be considered as superior or at least comparable to the others. Even so, our present results may be improved by optimizing the laser cavity and the quality of the SA film. It would be interesting for this to be further developed.

4. Conclusion

2 μm passively Q-switched bulk laser operations have been demonstrated using WS₂ SA. The min pulse width could be as low as 430 ns. This is the shortest pulse compared with other equivalent WS₂ materials in this waveband. Its corresponding output power and pulse energy were 488 mW and 5.422 μJ, respectively. Our results reveal that this WS₂ SA could be a promising SA material to realize 2 μm passively Q-switched bulk pulse lasers and that it possesses a potential application in the fields of photonics and electronic.

Conflicts of interest

There are no conflicts to declare.

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