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Simultaneously enhanced J_{sc} and FF by employing two solution-processed interfacial layers for inverted planar perovskite solar cells†

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Interface engineering has been proved to be a practical strategy to enhance power conversion efficiency (PCE) and stability of perovskite solar cells. Recently approaches involving ultra-thin layer deposition and vacuum-processing, while improving PCE, increase the processing complexity, and thus the overall cost. In this paper, we demonstrate a high-efficiency inverted planar perovskite solar cell, obtained with a simple wet-chemistry based, room-temperature, and cost-effective process, which synthesizes two interfacial layers: solution processed molybdenum oxide (named s-MoO_x) for anode and titanium(IV) oxide bis(2,4-pentanedionate) (TOPD) for cathode buffer layers. Steady-state photoluminescence (PL), time resolved photoluminescence (TRPL) and electrochemical impedance spectroscopy (EIS) are conducted to characterize the charge transport properties, confirming an enhanced charge extraction efficiency and a suppressed charge recombination. In addition, the introduction of the interfacial layers facilitates formation of an ohmic contact with the metal electrode, reducing the charge transfer resistance and increasing the FF to as high as 80.7%. The optimized solar cells achieve the best PCE of 16.04% which far exceeds the PCE of devices without these interfacial layers (PCE = 11.2%).

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1. Introduction

Since the first report of organometallic halide sensitized solar cells by Miyasaka *et al.* in 2009,¹ perovskite solar cells (Per-SCs) have experienced a rapid development and shown great potential as the next-generation photovoltaic platform, owing to their excellent optical and electronic properties, including high absorption coefficient,² low exciton binding energy,³ ambipolar charge transport,⁴ long electron-hole diffusion length and the solution processability.⁵ Until now, accompanied by the optimization in perovskite films and relevant device architectures, Per-SCs have achieved a PCE of ~22%.⁶ Among them, the inverted Per-SCs (p-i-n structure), such as ITO/PEDOT:PSS/perovskite/

PCBM/Ag architecture cells, are considered to be superior to the conventional n-i-p structured cells due to the simple fabrication processing, cost-effectiveness, low temperature and suppressed hysteresis.^{7,8} Recently, an efficient inverted perovskite solar cell with a PCE of about 20% was demonstrated.^{9,10}

Engineering of the interfacial layers, including hole or electron transport layer (ETL and HTL), blocking layer, as well as the electrode buffer layers, is critical in achieving high-efficiency inverted planar Per-SCs.¹¹⁻¹³ Firstly, these layers can decrease the band energy offset between the active layer and the electrode, leading to the enhancement of the open circuit voltage (V_{oc}). Secondly, the interfacial materials can also reduce the contact potential, forming the ohmic contact with electrodes, reducing the series resistance and improving the charge extraction and transport efficiency, thus lead to a higher short circuit current density (J_{sc}) and fill factor (FF). LiF is widely used as a cathode buffer layer between the PCBM (as an ETL) and back electrode to form a ohmic contact. However, its thickness is strictly limited to be less than 1 nm, or else significantly increases the series resistance of the device.^{14,15} Recently, thermally evaporated double C₆₀ and BCP layers were investigated as interfacial layers between PCBM and the silver contact, leading to FF as high as 80.1%.¹⁶ However, expensive vacuum techniques are incompatible with the concept of low-cost fabrication of perovskite solar cells. Polymers such as PFN, PN₄N *etc.* were used to improve the cathode interface, and resulting remarkable enhancement in PCE from 12.4%

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to 15.0%.¹⁷ TIPD (titanium(diisopropoxide) bis(2,4-pentanedionate)) as a cathode buffer layer has been used in Per-SCs,¹⁸ however the process of thermal annealing at 150 °C to change TIPD to TOPD,¹⁹ increases the risk of the perovskite films decomposition. Thus, a direct-processing of the TOPD in Per-SCs could be potentially a more effective way for interface improvement, which was demonstrated in organic solar cells.²⁰

In inverted Per-SCs, PEDOT:PSS is the most commonly used HTL with a good hole transporting and collecting efficiency. However, the large band energy offset (~ 0.3 eV) between PEDOT:PSS and ITO leaves the room for further improvement of V_{oc} and the charge transfer efficiency for solar cells. Moreover, the natural acidity of PEDOT:PSS can corrode ITO, resulting in device long-term stability issue.²¹ Many robust materials, such as NiO_x,^{22,23} CuO_x,²⁴ MoO_x,^{25,26} V₂O₅,²⁷ and PbS²⁸ have been selected to replace PEDOT:PSS, while the efficiencies are lower than those of the state-of-the-art cells based on PEDOT:PSS. Recently, the MoO₃/PEDOT:PSS bilayer structure was used as the HTL for inverted Per-SCs, fabricated by a vacuum deposition, showing enhanced PCE and stability.²⁹

In this work, we used two interfacial layers: s-MoO_x for anode and TOPD for cathode buffer layers in inverted CH₃NH₃PbI₃ Per-SCs, with a simple wet-chemistry based, room-temperature, and cost-effective procession. This strategy improves the charge extraction efficiency and suppresses the charge recombination, as well as prevents ITO corrosion and forms a reduced contact potential between active layer and electrodes. The PCEs of these Per-SCs are strongly enhanced, from 11.4% (control cells) to 16.05%, mostly due to the increase of FF (from 69.2% to 80.7%) and J_{sc} (from 18.3% to 21.75 mA cm⁻²). Moreover, the PCEs of over 85% of all solar cells are higher than 14%, demonstrating good process reproducibility.

2. Experimental section

2.1. Materials and synthesis

2.1.1. Materials. An aqueous dispersion of PEDOT:PSS (AI 4083) was obtained from Heraeus Co. Fullerene derivatives (PC₆₁BM (99.8%)) were purchased from Nano-C Co. TOPD, bis(2,4-pentanedionato)molybdenum(vi) dioxide (MoO₂(acac)₂) and PbI₂ (99.999%) were purchased from Alfa Aesar. All these commercially available materials were used as received without further purification.

2.1.2. Synthesis of CH₃NH₃I. CH₃NH₃I was synthesized through the reaction of 28.7 mL methylamine (40 wt% in methanol, Aladdin) and 29.8 mL hydroiodic acid (57 wt% in water, Aladdin) under nitrogen atmosphere in 250 mL round-bottom flask in an ice bath for 2 h with stirring. The crystals of methylammonium iodide (CH₃NH₃I) were collected using a rotary evaporator at 50 °C for 2 h to remove the solvent. The product was dissolved in ethanol, followed by recrystallization by diethyl ether. The crystals were filtered and washed three times with diethyl ether. At last, the solid was dried at 60 °C in vacuum oven overnight. The detailed process is shown in ref. 30.

2.2. Solar cells fabrication

ITO glasses (AGC, 11-8, 7 Ω sq⁻¹) were patterned by laser cutting and ultrasonically cleaned with detergent, deionized water, acetone and isopropanol for 20 min, sequentially, followed by plasma cleaning for 20 min prior to use. 5 mg mL⁻¹ of MoO₂(acac)₂ isopropanol solution was spin-coated at 4000 rpm for 30 s on the precleaned ITO glass, and then baked in air at 150 °C for 10 min to prepared s-MoO_x film, followed by plasma cleaning for 1 min. Subsequently, PEDOT:PSS aqueous solution filtered through a 0.22 μm filter was spin-coated at 4000 rpm for 30 s on the s-MoO_x film, and then baked at 120 °C in air for 15 min. Then the ITO/s-MoO_x/PEDOT:PSS substrate was transferred to a nitrogen-filled glove-box for the perovskite film deposition. 461 mg of PbI₂, 159 mg of CH₃NH₃I was dissolved in a mixed solvent of DMF and DMSO (7 : 3, v/v) at 60 °C with stirring for 12 h to prepare perovskite precursor. 40 μL completely dissolved perovskite precursor solution was spin-coated on the PEDOT:PSS layer at 500 rpm for 3 s, followed by 4000 rpm for 30 s. 240 μL of chlorobenzene was quickly dripped on the rotating substrate at the beginning of 8–12 s in the second spin coating step. The substrate was immediately dried on a hot plate at 100 °C for 10 min and obtained a dense CH₃NH₃PbI₃ film. After that, 20 mg mL⁻¹ of PCBM solution in chlorobenzene was spin-coated at 1000 rpm for 20 s on the perovskite absorber layer. The TOPD buffer layer was prepared by spin-coating a 3 mg mL⁻¹ TOPD isopropanol solution on the PCBM at 4000 rpm for 30 s and then stays in glove box for 24 hours for solvent annealing. The thickness of the TOPD layer was about 10 nm. Finally, Ag electrode was deposited by using thermal evaporator at a constant evaporation rate of 1 Å s⁻¹.

2.3. Characterizations

Scanning electron microscope (SEM) images were obtained by using FE-SEM (ZEISSUltra55). Transmittance spectra were recorded with an integrating sphere system (Ocean Optics, USA) in the 400–900 nm range. X-ray diffraction (XRD) analysis was performed on a PANalytical X'Pert PRO diffractometer with the Cu-K radiation at a scan rate of 4° min⁻¹. Steady-state PL spectra were measured using an excitation wavelength of 467 nm in an HORIBAfluorolog3. TRPL measured by Time Correlated Single Photon Counting (TCSPC, picoharp300) with a femto second laser source. J - V curves measurements were carried out using Keithley 2400 at room temperature under AM1.5G illuminations (1000 W m⁻²) from a solar simulator (Newport, 91160), which was calibrated using a standard silicon solar cell device by the NREL. The incident photon to converted current efficiency (IPCE) spectra was measured from 300–850 nm using a Xe source (Newport, 66902). The light intensity at each wavelength was calibrated with a standard single-crystal Si photovoltaic cell. The IPCE measurement was performed under ambient atmosphere at room temperature. EIS was measured with a CHI6601 Electrochemical Workstation with an AC signal of 200 mV in the frequency range of 0.1 Hz to 1 MHz.



3. Result and discussion

Fig. 1a shows the schematic structure of the optimized inverted planar $\text{CH}_3\text{NH}_3\text{PbI}_3$ Per-SC, with two interfacial layers (denoted as device C). In the structure, $s\text{-MoO}_x$ film is the anode interfacial layer above ITO electrode, and TOPD (the molecular structure in Fig. 1b) is the cathode interfacial layer between ETL and back metal contact. For comparison, two control devices, without TOPD and $s\text{-MoO}_x$ (denoted as device A) and with only TOPD (denoted as device B) have also been fabricated.

The continuous and uniform $s\text{-MoO}_x$ layer, as an anode interfacial material, was prepared according to ref. 31. Isopropanol solution of $\text{MoO}_2(\text{acac})_2$ (the molecular structure shown in Fig. 1b) was spin-coated on the surface of pre-cleaned ITO glass, then baking in air at 150°C for 10 min to completely transferring into $s\text{-MoO}_x$. The obtained $s\text{-MoO}_x$ layer is highly transparent in the visible region (in Fig. S1†). The cathode interfacial layer TOPD was prepared by room-temperature spin-coating without afterwards thermal annealing. The preparation of two interfacial materials is both based on the low-temperature solution processing, which is fully compatible with flexible

plastic substrate and low-cost R2R production mode. The details of solar cell fabrication are shown in Experimental section.

The band structures of the materials involve in this study, $s\text{-MoO}_x$,³¹ PEDOT:PSS,¹² $\text{CH}_3\text{NH}_3\text{PbI}_3$,³² PCBM,¹² and TOPD²⁰ are depicted schematically in Fig. 1c. The lowest unoccupied molecular orbital (LUMO) energy level of TOPD (-3.86 eV) is very close to that of PCBM (-3.9 eV), which is favorable for electrons generated in the active layer of perovskite, and transporting from the active layer to metal electrode through TOPD. This also enhances the interfacial quality between ETL and the back contact electrode. Meanwhile its highest occupied molecular orbital (HOMO) energy level (-6.8 eV) is much lower than that of $\text{CH}_3\text{NH}_3\text{PbI}_3$ (-5.4 eV), showing a strongly hole blocking capability, with bare PCBM (HOMO -5.9 eV) as an ETL. Similarly, the bottom of conduction band of $s\text{-MoO}_x$ (-4.9 eV) is close to the HOMO energy level of PEDOT:PSS (-5.0 eV), reducing the contact potential between ITO electrode and PEDOT:PSS, potentially benefiting to the hole transfer and the charge collection efficiency. In addition, both $s\text{-MoO}_x$ and TOPD interfacial materials present almost zero energy level offset comparing with other layers in devices, theoretically favoring the increase of V_{oc} of device.

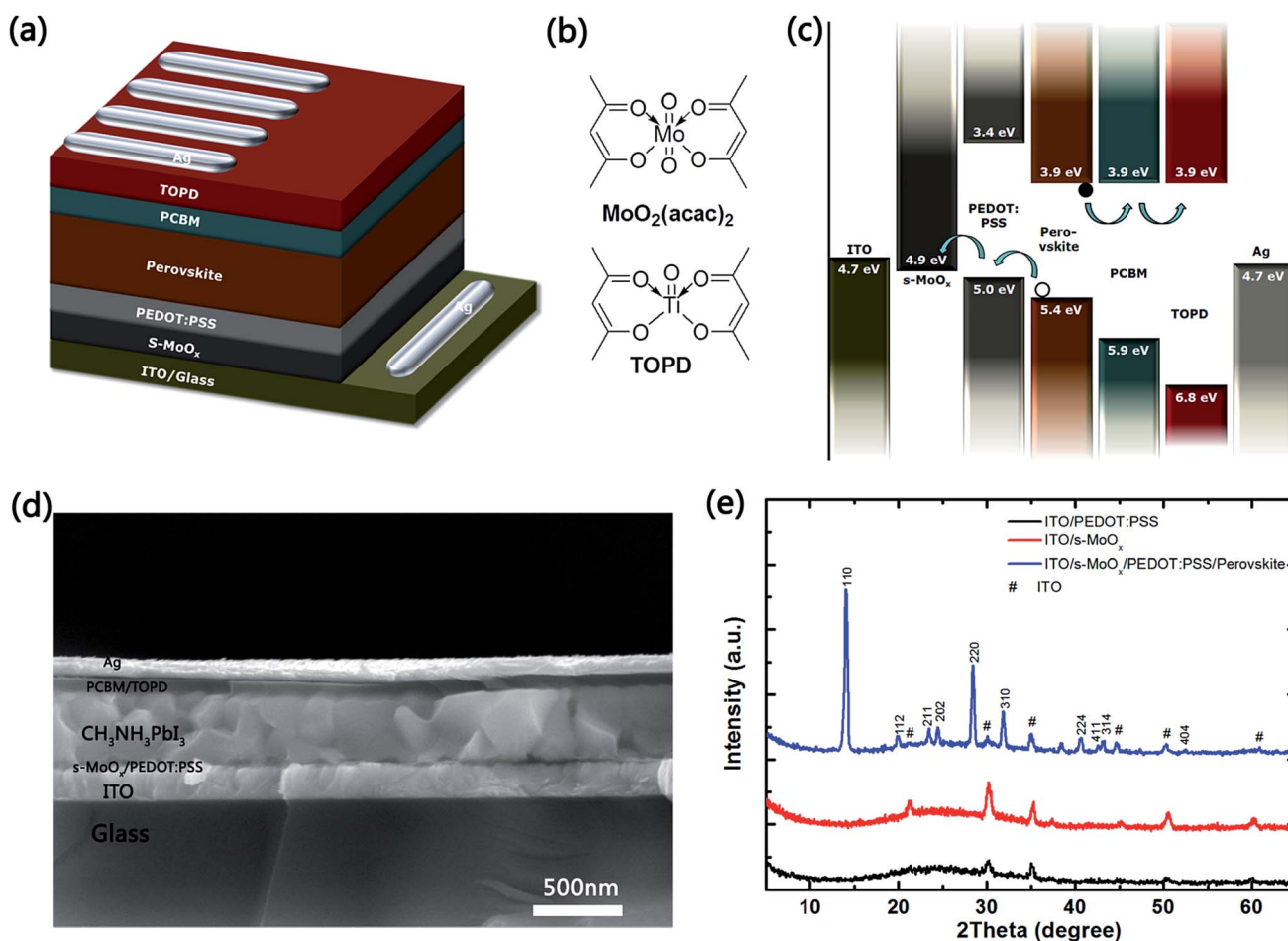


Fig. 1 (a) Schematic architecture of optimized device C. (b) Molecular structures of $\text{MoO}_2(\text{acac})_2$ and TOPD. (c) Schematic energy band structure of the device C. (d) Cross-section of the SEM image of the device C. (e) XRD patterns of the ITO/PEDOT:PSS, ITO/ $s\text{-MoO}_x$, and ITO/ $s\text{-MoO}_x$ /PEDOT:PSS/ $\text{CH}_3\text{NH}_3\text{PbI}_3$.



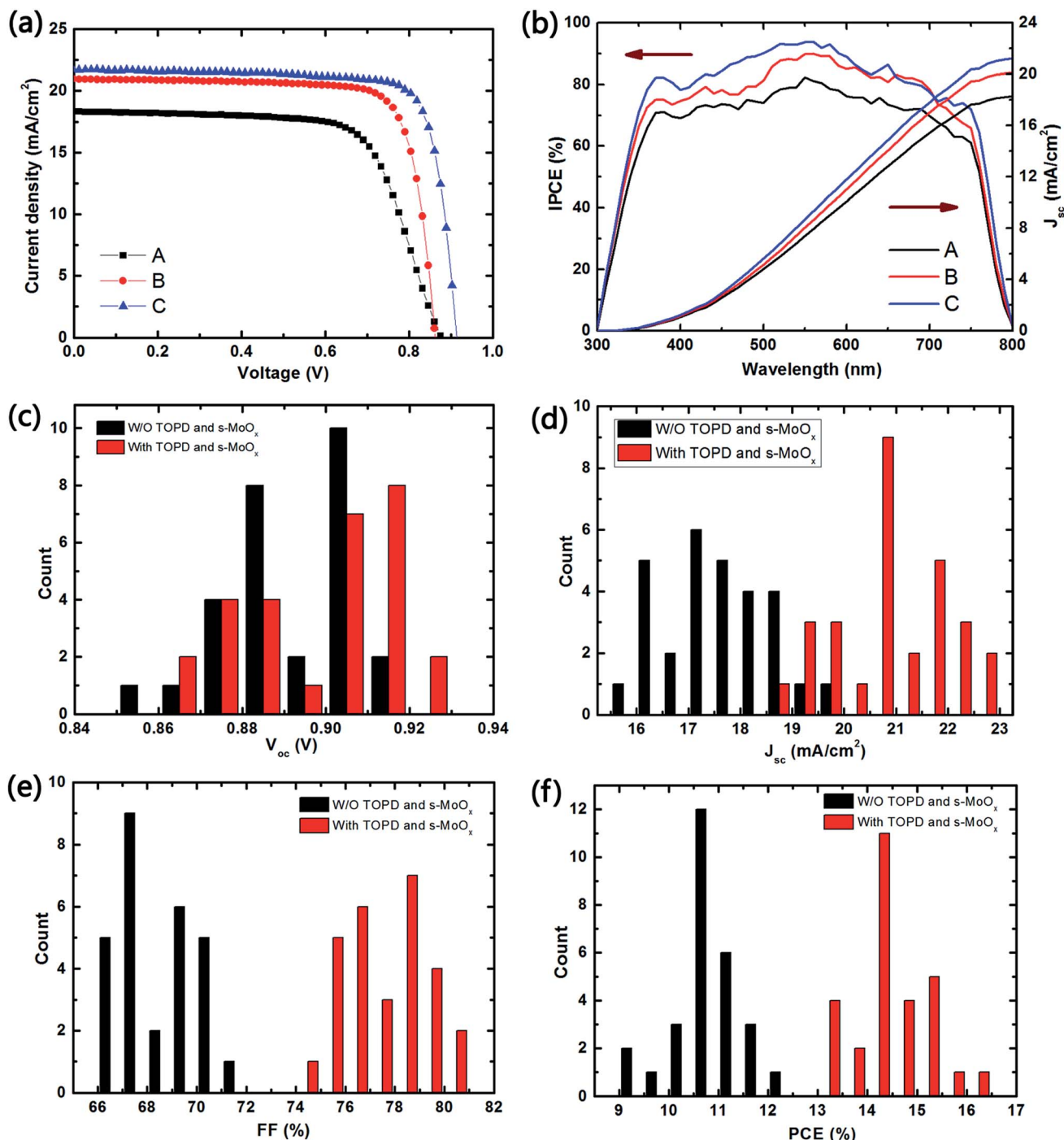


Fig. 2 Photovoltaic performance of devices. (a) J - V curves for devices (A–C) at reversed scan, under one sun illumination of AM1.5G, 100 mW cm^{-2} (b) IPCE spectra (left) with integrated J_{sc} (right) based on IPCE measurement. Distribution histograms of V_{oc} (c), J_{sc} (d), FF (e), and PCE (f) of the 28 independently fabricated devices.

Fig. 1d shows the cross-section scanning electron microscopy (SEM) of optimized device C, with double interfacial layers. A well-defined, layer-by-layer structure with clear interfaces can be observed. The thicknesses of s-MoO_x, PEDOT:PSS, CH₃NH₃PbI₃, PCBM, and TOPD are about 10, 30, 430, 50, and 10 nm, respectively. The CH₃NH₃PbI₃ film was uniform and the grain size in the vertical direction is comparable to or even larger than the whole film size, remarkably reducing the defects

in the grain boundaries. This indicates that the photo generated charge carriers can effectively transport across a single crystal of the CH₃NH₃PbI₃ films and reach the corresponding charge-carrier extraction interfaces before recombination. Fig. 1e shows the X-ray Diffraction (XRD) patterns of s-MoO_x, PEDOT:PSS, and s-MoO_x/PEDOT:PSS/CH₃NH₃PbI₃ films deposited on ITO glass. The patterns of ITO/PEDOT:PSS and ITO/s-MoO_x only present a series of ITO characteristic peaks,



Table 1 Photovoltaic parameters of devices of A, B and C

Device#	V_{oc} (V)	J_{sc} (mA cm^{-2})	FF (%)	PCE (%)
(A) W/O s-MoO _x and TOPD	0.88	18.30	69.2	11.4
(B) With TOPD	0.86	20.97	79.4	14.35
(C) With s-MoO _x and TOPD	0.91	21.75	80.7	16.04

illustrating an amorphous phase of PEDOT:PSS and s-MoO_x, which agrees with the ref. 29 and 31. The main diffraction peaks of 2θ at 14.02° , 28.4° , and 31.84° correspond to the (110), (220) and (310) planes of CH₃NH₃PbI₃ respectively, indicating a pure standard tetragonal crystal structure.^{33,34}

The current density–voltage (J – V) characteristics curves (under one sun illumination of AM1.5G, 100 mW cm^{-2}) of device C, as well as control cells of A, B, are shown in Fig. 2a, and the detailed photovoltaic parameters are summarized in Table 1. The control device A shows a PCE of 11.4%, with a J_{sc} of 18.3 mA cm^{-2} , a V_{oc} of 0.88 V and a FF of 69.2%. Device B, exhibits an improved PCE of 14.35%, with a J_{sc} 20.97 mA cm^{-2} , a V_{oc} of 0.86 V and a FF of 79.4%. Comparing to A, the PCE of device B has been increased more than $\sim 25\%$. This could be ascribed to the enhanced FF ($\sim 15\%$) due to the introduction of TOPD interfacial layer and lowering the series resistance. Device C, optimized with two interfacial layers (TOPD in cathode and s-MoO_x in anode) leads to a further remarkable enhancement in J_{sc} from 20.97 to 21.75 mA cm^{-2} , V_{oc} from 0.86 to 0.914 V, and FF from 79.4% to 80.7%, achieving a highest PCE of 16.04%. Due to the well-matched energy levels, as well as effective electrons and holes extraction from CH₃NH₃PbI₃ layer, device C shows significant enhancement in J_{sc} ($\sim 18\%$) and FF ($\sim 16\%$) comparing to device A. Moreover, a higher V_{oc} was obtained in device C due to the smallest bandgap offset. The incident photon to converted current efficiency (IPCE) spectra of the corresponding devices (A, B, and C) is shown in Fig. 2b. Device C, presenting a broad plateau of $\sim 80\%$ in the region from 350 nm to 750 nm, shows a higher intensity than those of device B ($\sim 75\%$) and A ($\sim 70\%$). The integrated J_{sc} from the IPCE spectra are 18.28 mA cm^{-2} , 20.13 mA cm^{-2} and 21.25 mA cm^{-2} for device A, B, and C, respectively, in agreement with the results obtained from J – V measurement mentioned above.

The statistics of the J_{sc} , V_{oc} , FF and PCE for 28 independently devices A and C are shown in Fig. 2c–f. The detailed parameters for all cells are shown in Table S1.† Statistical distribution of V_{oc} (0.90 V) for device C is slightly higher than that of device A (0.89 V), whereas its J_{sc} and FF (20.89 mA cm^{-2} and 68.6%) are remarkably larger than that of device A (17.54 mA cm^{-2} and 68.6%). Most importantly, the PCEs of over 85% of device C are higher than 14%, demonstrating the good reproducibility of the high-performance Per-SCs with s-MoO_x and TOPD as interfacial layers. The hysteresis in the J – V curves of device C has been studied at different scan rate (0.14 – 1.4 V s^{-1}) and opposite scan directions (Fig. S2†). Those curves, both in different scan rates and opposite directions present almost the same, demonstrating that there is no obvious photocurrent hysteresis for the optimized device C.

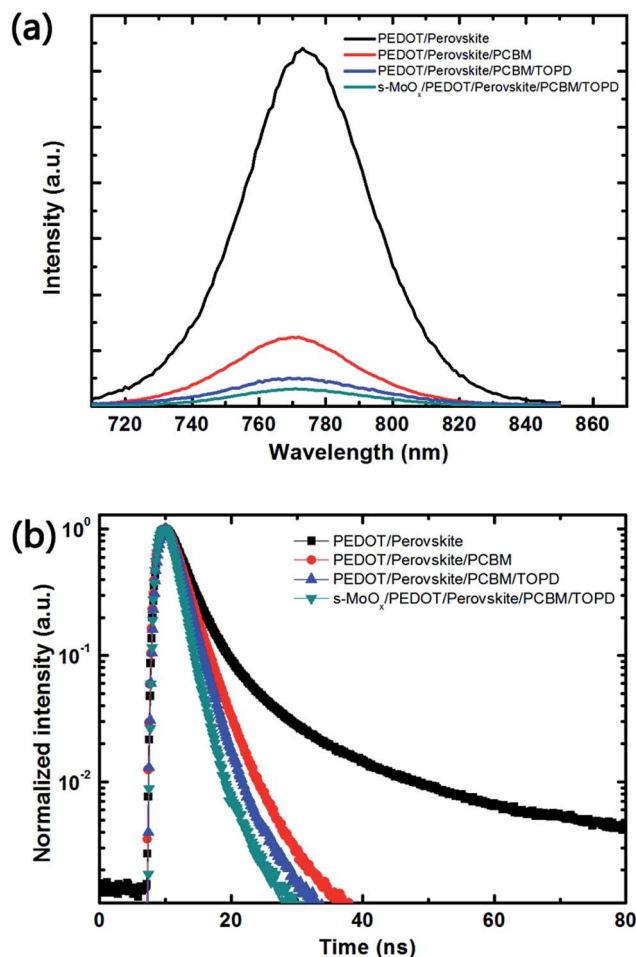


Fig. 3 Steady state PL spectra (a) and TRPL (b) of PEDOT:PSS/perovskite, PEDOT:PSS/perovskite/PCBM, PEDOT:PSS/perovskite/PCBM/TOPD, s-MoO_x/PEDOT:PSS/perovskite/PCBM/TOPD deposited on the ITO substrates.

To investigate the effect of s-MoO_x and TOPD interfacial layers on the charge transport properties, steady-state PL spectra and TRPL were conducted and the thickness of the CH₃NH₃PbI₃ in all films is almost fixed at an identical level. As shown in Fig. 3a, the steady-state PL intensity of the CH₃NH₃PbI₃ film shows a strong emission peak near 770 nm, and significantly quenched after PCBM deposition. The PL intensity is completely quenched after two interfacial layers of s-MoO_x and TOPD are deposited. This confirms that photo-induced carriers are more easily extracted from perovskite layer to electrode layers with the reducing charge recombination. Fig. 3b shows the TRPL for PEDOT:PSS/perovskite, PEDOT:PSS/perovskite/PCBM, PEDOT:PSS/perovskite/PCBM/TOPD and s-MoO_x/PEDOT:PSS/perovskite/PCBM/TOPD deposited on the ITO substrates. The decay times were calculated to be 37.53 ns, 2.94 ns, 2.52 ns, and 1.98 ns for above four films respectively, by fitting the measured TRPL data with a bi-exponential decay function $I(t) = A_1 e^{-t/\tau_1} + A_2 e^{-t/\tau_2}$ (the dominating lifetime is used for comparison). For the perovskite layer without a interfacial layer, the PL lifetime is about 37.53 ns. On topped by a PCBM layer, the PL lifetime quickly decreases to 2.94, which ascribes to the charge-carrier transfer from



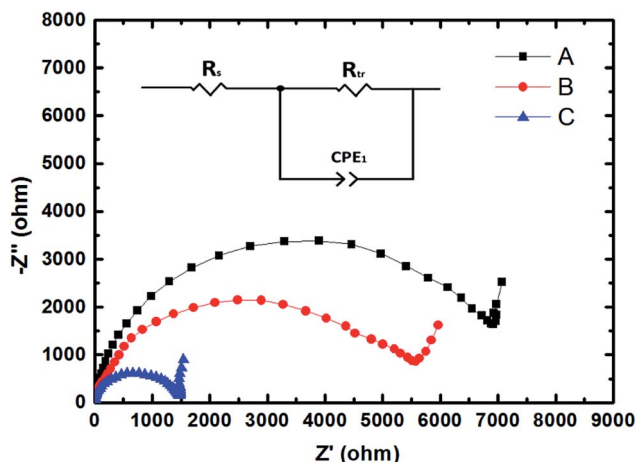


Fig. 4 Nyquist plots of devices A–C measured under dark ambient conditions at 200 mV applied bias. The inset shows the equivalent circuit.

perovskite to ETL and leads to a PL intensity quenched. After imbedding the TOPD in perovskite film, the PL lifetime decreases as low as 2.52 ns. The PL life time is further reduced (1.98 ns) with introducing the s-MoO_x layer on another side of the perovskite layer. This confirms that introducing of TOPD and s-MoO_x layers contributes to effective PL quenching, charge-carrier transferring, and suppressing the charge recombination.

To further gain insight into the effect of s-MoO_x and TOPD interfacial layers on the charge transfer resistance (R_{tr}) or recombination resistance (R_{rec}), the EIS for devices of A, B and C was carried out at 200 mV applied bias in dark condition. Fig. 4 shows the Nyquist plots of three devices (frequency range from 0.1 Hz to 100 KHz). There are two main characteristic arcs (a high frequency one and a low frequency one). We mainly study the impedance spectra at high frequency region, which related to the charge transfer resistance between perovskite film and electrodes.³⁵ The obtained spectra were fitted to a simplified circuit model as inset shown. The transfer resistance (R_{tr}) is proportional to the arc size which is ranked in the sequence of A > B > C. This indicates that the insertion of both TOPD and s-MoO_x interfacial layers can significantly improve the contact between active layer and electrodes and decrease the resistance of charge transfer process.

4. Conclusions

In conclusion, we demonstrate a high-efficiency inverted planar Per-SC by introducing two interfacial layers, s-MoO_x for anode and TOPD for cathode layer. The characterizations of EIS, PL and TRPL reveal that this optimized architecture significantly reduces charge transfer resistance, suppresses charge recombination and enhances charge extraction efficiency. Those optimized perovskite solar cells show the highest PCE of 16.04%, with a remarkably high FF of 80.7%, J_{sc} of 21.75 mA cm⁻², and V_{oc} of 0.91 V. In addition, the photocurrent hysteresis is negligible and the process is in good reproducibility. These results suggest the strategy with two interfacial materials (s-MoO_x and

TOPD), fabricated with a low-temperature and solution processing is quite promising in obtaining an efficient and low-cost perovskite solar cell.

Author contributions

W. H. and J. W. conceived the idea. W. H., Y. J. and Y. D. conducted material and device fabrications and tests. W. H., J. W., K. K, Y. J. co-wrote the paper. All other authors discussed the results and commented on the manuscript. J. W. directed the researches.

Conflicts of interest

There are no conflicts to declare.

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Notes and references

- 1 A. Kojima, K. Teshima, Y. Shirai and T. Miyasaka, *J. Am. Chem. Soc.*, 2009, **131**, 6050–6051.
- 2 S. Kazim, M. K. Nazeeruddin, M. Grätzel and S. Ahmad, *Angew. Chem., Int. Ed.*, 2014, **53**, 2812–2824.
- 3 J. S. Manser and P. V. Kamat, *Nat. Photonics*, 2014, **8**, 737–743.
- 4 W. A. Laban and L. Etgar, *Energy Environ. Sci.*, 2013, **6**, 3249.
- 5 G. Xing, N. Mathews, S. Sun, S. S. Lim, Y. M. Lam, M. Gratzel, S. Mhaisalkar and T. C. Sum, *Science*, 2013, **342**, 344–347.
- 6 F. Bella, G. Griffini, J.-P. Correa-Baena, G. Saracco, M. Gratzel, A. Hagfeldt, S. Turri and C. Gerbaldi, *Science*, 2016, **354**, 203–206.
- 7 J. H. Heo, H. J. Han, D. Kim, T. K. Ahn and S. H. Im, *Energy Environ. Sci.*, 2015, **8**, 1602–1608.
- 8 E. L. Unger, E. T. Hoke, C. D. Bailie, W. H. Nguyen, A. R. Bowring, T. Heumüller, M. G. Christoforo and M. D. McGehee, *Energy Environ. Sci.*, 2014, **7**, 3690–3698.
- 9 C.-H. Chiang, M. K. Nazeeruddin, M. Grätzel and C.-G. Wu, *Energy Environ. Sci.*, 2017, **10**, 808–817.
- 10 D. Luo, L. Zhao, J. Wu, Q. Hu, Y. Zhang, Z. Xu, Y. Liu, T. Liu, K. Chen, W. Yang, W. Zhang, R. Zhu and Q. Gong, *Adv. Mater.*, 2017, **201**, 1–9.
- 11 H. Zhou, Q. Chen, G. Li, S. Luo, T.-B. Song, H.-S. Duan, Z. Hong, J. You, Y. Liu and Y. Yang, *Science*, 2014, **345**, 542–546.



- 12 O. Malinkiewicz, A. Yella, Y. H. Lee, G. M. Espallargas, M. Graetzel, M. K. Nazeeruddin and H. J. Bolink, *Nat. Photonics*, 2013, **8**, 128–132.
- 13 T. Liu, K. Chen, Q. Hu, R. Zhu and Q. Gong, *Adv. Energy Mater.*, 2016, **6**, 1–17.
- 14 J. Seo, S. Park, Y. Chan Kim, N. J. Jeon, J. H. Noh, S. C. Yoon and S. Il Seok, *Energy Environ. Sci.*, 2014, **7**, 2642.
- 15 K. Sun, J. Chang, F. H. Isikgor, P. Li and J. Ouyang, *Nanoscale*, 2015, **7**, 896–900.
- 16 Q. Wang, Y. Shao, Q. Dong, Z. Xiao, Y. Yuan and J. Huang, *Energy Environ. Sci.*, 2014, **7**, 2359–2365.
- 17 Q. Xue, Z. Hu, J. Liu, J. Lin, C. Sun, Z. Chen, C. Duan, J. Wang, C. Liao, W. M. Lau, F. Huang, H.-L. Yip and Y. Cao, *J. Mater. Chem. A*, 2014, **2**, 19598–19603.
- 18 C. Li, F. Wang, J. Xu, J. Yao, B. Zhang, C. Zhang, M. Xiao, S. Dai, Y. Li and Z. Tan, *Nanoscale*, 2015, **7**, 9771–9778.
- 19 Z. Tan, W. Zhang, Z. Zhang, D. Qian, Y. Huang, J. Hou and Y. Li, *Adv. Mater.*, 2012, **24**, 1476–1481.
- 20 F. Wang, L. Li, Q. Xu, D. Qian, S. Li and Z. Tan, *Org. Electron.*, 2013, **14**, 845–851.
- 21 X. Liu, P. Cai, D. Chen, J. Chen, S. Su and Y. Cao, *Sci. China: Chem.*, 2014, **57**, 973–981.
- 22 J. H. Kim, P. W. Liang, S. T. Williams, N. Cho, C. C. Chueh, M. S. Glaz, D. S. Ginger and A. K. Y. Jen, *Adv. Mater.*, 2015, **27**, 695–701.
- 23 X. Yin, Z. Yao, Q. Luo, X. Dai, Y. Zhou, Y. Zhang, Y. Zhou, S. Luo, J. Li, N. Wang and H. Lin, *ACS Appl. Mater. Interfaces*, 2017, **9**, 2439–2448.
- 24 Z. K. Yu, W. F. Fu, W. Q. Liu, Z. Q. Zhang, Y. J. Liu, J. L. Yan, T. Ye, W. T. Yang, H. Y. Li and H. Z. Chen, *Chin. Chem. Lett.*, 2017, **28**, 13–18.
- 25 B. S. Kim, T. M. Kim, M. S. Choi, H. S. Shim and J. J. Kim, *Org. Electron.*, 2015, **17**, 102–106.
- 26 Y. Zhao, A. M. Nardes and K. Zhu, *Appl. Phys. Lett.*, 2014, **104**, 1–7.
- 27 G. Li, C. W. Chu, V. Shrotriya, J. Huang and Y. Yang, *Appl. Phys. Lett.*, 2006, **88**, 3–6.
- 28 L. Hu, W. Wang, H. Liu, J. Peng, H. Cao, G. Shao, Z. Xia, W. Ma and J. Tang, *J. Mater. Chem. A*, 2015, **3**, 515–518.
- 29 F. Hou, Z. Su, F. Jin, X. Yan, L. Wang, H. Zhao, J. Zhu, B. Chu and W. Li, *Nanoscale*, 2015, **7**, 9427–9432.
- 30 J.-H. Im, I.-H. Jang, N. Pellet, M. Grätzel and N.-G. Park, *Nat. Nanotechnol.*, 2014, **9**, 927–932.
- 31 Z. Tan, D. Qian, W. Zhang, L. Li, Y. Ding, Q. Xu, F. Wang and Y. Li, *J. Mater. Chem. A*, 2013, **1**, 657–664.
- 32 D. Zhao, M. Sexton, H. Y. Park, G. Baure, J. C. Nino and F. So, *Adv. Energy Mater.*, 2015, **5**, 1–5.
- 33 N. J. Jeon, J. H. Noh, Y. C. Kim, W. S. Yang, S. Ryu and S. Il Seok, *Nat. Mater.*, 2014, **13**, 897–903.
- 34 N. Ahn, D. Y. Son, I. H. Jang, S. M. Kang, M. Choi and N. G. Park, *J. Am. Chem. Soc.*, 2015, **137**, 8696–8699.
- 35 J. A. Christians, R. C. M. Fung and P. V. Kamat, *J. Am. Chem. Soc.*, 2014, **136**, 758–764.

