

Cite this: *RSC Adv.*, 2017, 7, 41136

Encapsulation of transition metal dichalcogenides crystals with room temperature plasma deposited carbonaceous films†

Rakesh D. Mahyavanshi,^{ID} *^a Golap Kalita,^{*a} Rupesh Singh,^c Masaharu Kondo,^b Takehisa Dewa,^b Toshio Kawahara,^c Masayoshi Umeno^c and Masaki Tanemura^a

Encapsulation and its effect on two-dimensional (2D) dichalcogenides layered materials are of significant importance for electronic device fabrication. Here, we developed a dry and rapid encapsulation method for the 2D dichalcogenides by coating a carbonaceous film at room temperature using a plasma technique. Thin encapsulation film is coated on dichalcogenide crystals by microwave assisted surface wave plasma (MW-SWP) chemical vapor deposition (CVD) within a very short duration. Photoluminescence (PL) analysis shows the influence on excitation as well as the emission light with the coating of a carbonaceous film. The encapsulated dichalcogenide samples remain intact after heating at 200 °C as observed by PL and microscopic analysis. We also demonstrate obtaining stable photoconductivity for the dichalcogenide layers with encapsulation of a carbonaceous film. The rapid and dry coating technique at room temperature can be extended to various other 2D materials as the MW-SWP process allows damage-free encapsulation.

Received 19th June 2017
Accepted 17th August 2017

DOI: 10.1039/c7ra06816f

rsc.li/rsc-advances

1. Introduction

In recent years, transition metal dichalcogenides (TMDCs) and other two-dimensional (2D) layered materials have attracted significant interest for nanoelectronic and optoelectronic device applications.^{1–5} The TMDCs are a large family of semiconductors (e.g. MoS₂, WS₂ etc.), metals (e.g. TiS₂, VSe₂), superconductors (e.g. TaS₂, NbSe₂), semimetals (e.g. WTe₂ and TiSe₂) and insulators (e.g. HfS₂).^{6–16} The thickness dependent electronic and optoelectronic properties of group VI dichalcogenides (MX₂, M = Mo, W; X = S, Se, etc.) semiconductors are quite promising for novel optoelectronic devices.^{17,18} Transition from indirect band gap to direct band gap for the semiconducting trigonal prismatic MX₂ has been revealed as the thickness reduces to monolayer.^{19–22} The MX₂ semiconductor with excellent optical absorption in visible range and large exciton binding energy has attracted wide attention for high-performance photodetectors, efficient solar cells, and multi-functional nanoelectronics.^{23–26} TMDCs layered materials have been derived by exfoliation process, hydrothermal synthesis

and chemical vapor deposition (CVD) techniques.^{27–30} The CVD technique has a significant advantage for the synthesis of large-area polycrystalline films, single crystal domains, vertical or lateral heterostructures of two different dichalcogenides.^{29–32}

Recent studies show structural transformation and degradation of MX₂ layered materials with oxygen exposure, UV irradiation, ozone exposure and heating under ambient conditions.^{33–36} Significant structural degradation of MX₂ layers has been also observed with long-term exposure to ambient conditions.³⁶ The other 2D layered semiconductors, such as black phosphorus are more unstable and degradable under ambient conditions.³⁷ In this context, encapsulation and device engineering are the key aspects that need to be addressed in the coming years for practical application of these layered semiconductors. Gao *et al.*, has demonstrated coating of poly(methyl methacrylate) (PMMA) and Parylene C on MoS₂ and WS₂ layers to protect from ageing.³⁶ Ahn *et al.* has demonstrated the transferring process of PMMA/h-BN layer on WSe₂ layer as degradation protective layer.³⁸ Recently, Woodhead *et al.* have developed an optical solution to encapsulate monolayer MX₂ from physical damage and degradation.³⁹ It has been also reported that depending on the layered materials a compressive or other lattice strain can be observed with the encapsulation layer.^{39,40} In contrast to the previous reports, we developed a rapid and dry approach to fully encapsulate MX₂ layered materials preventing degradation in ambient conditions. As follows, we demonstrate the encapsulation and PL emission properties with the plasma CVD deposited carbonaceous film on MX₂ layers.

^aDepartment of Physical Science and Engineering, Nagoya Institute of Technology, Gokiso-cho, Showa-ku, Nagoya 466-8555, Japan. E-mail: rmahyavanshi@gmail.com; kalita.golap@nitech.ac.jp; Fax: +81-527355216; Tel: +81-527355216

^bDepartment of Life Science and Applied Chemistry, Nagoya Institute of Technology, Gokiso-cho, Showa-ku, Nagoya 466-8555, Japan

^cDepartment of Electronics and Information Engineering, Chubu University, 1200 Matsumoto-cho, Kasugai, 487-8501, Japan

† Electronic supplementary information (ESI) available. See DOI: 10.1039/c7ra06816f



2. Experimental section

2.1 Synthesis of MoS₂ and WS₂ crystals

MoS₂ and WS₂ crystals were grown on SiO₂/Si substrate by thermal chemical vapor deposition (CVD) method by controlling the growth parameters. The synthesis was performed in a quartz tube CVD system with a mixture of argon (Ar) and H₂ gas flow at atmospheric pressure. Prior to the thermal CVD growth, the substrate was cleaned using acetone and following isopropyl alcohol (IPA) for 15 min each by sonication. Sulfur powder (1 g) in a ceramic boat was inserted into the quartz tube of length 90 cm and diameter of 4.5 cm. It was placed in a low-temperature furnace (LTF) heated to 180 °C. The substrate and MoO₃ and WO₃ powder (10 mg) were kept on a ceramic boat for the growth of MoS₂ and WS₂ crystals, respectively in two separate experiments. MoS₂ and WS₂ crystals were synthesized at 750 and 850 °C, respectively in the high-temperature furnace (HTF). Both the experiments were carried out using 100 standard cubic centimeters per minute (sccm) of argon and H₂ gas mixture (97 : 3).

2.2 Deposition of a carbonaceous film by plasma CVD

Carbonaceous film was deposited by microwave assisted surface wave plasma (MW-SWP) chemical vapor deposition (CVD) technique on the MoS₂ and WS₂ crystals synthesized on SiO₂/Si substrates. A metal grid was placed above the MoS₂ and WS₂ crystal samples (Fig. S1†) to avoid plasma induced deformation of the TMDCs layers. The MW-SWP CVD chamber was

evacuated to 5×10^{-4} Pa prior to the deposition. The thin film deposition was performed at 10–20 Pa gas composition pressure with a flow rate of 100–200 sccm of Ar and 10–50 sccm of CH₄. The microwave power was fixed as 800 W with negligible amount of fluctuation. Deposition of the thin films was carried out for a duration of 60–240 s at room temperature (~24 °C).

2.3 Characterization of prepared samples

Prepared samples were characterized by optical microscopy, Raman spectroscopy, field-emission scanning electron microscopy (FE-SEM), photoluminescence (PL) spectroscopy and X-ray photoelectron spectroscopy (XPS). Optical microscopy analysis was performed using the digital optical microscope VHX-500 in reflectance mode with a Moticam 2000 2.0 M pixel camera. Raman analysis was carried out using the NRS 3300 laser Raman spectrometer with a laser excitation wavelength of 532.08 nm. FE-SEM study was performed with JEOL JSM-7800F with an accelerating voltage of 5 kV. Infrared (IR) spectroscopy analysis was carried out with grazing incident using JASCO IRT-5000 FT-IR microscope. Absorbance and transmittance of the carbonaceous film was measured by JASCO V-670K spectrophotometer. XPS analysis was performed by VersaProbe using monochromated Al K α excitation source (1486.6 eV). PL analysis was carried with a confocal laser scanning microscope (NX-3DFLIM-N03, Tokyo Instruments, Japan) equipped with a Nd:YVO₄ diode laser. For the device fabrication, Au electrodes were deposited by the

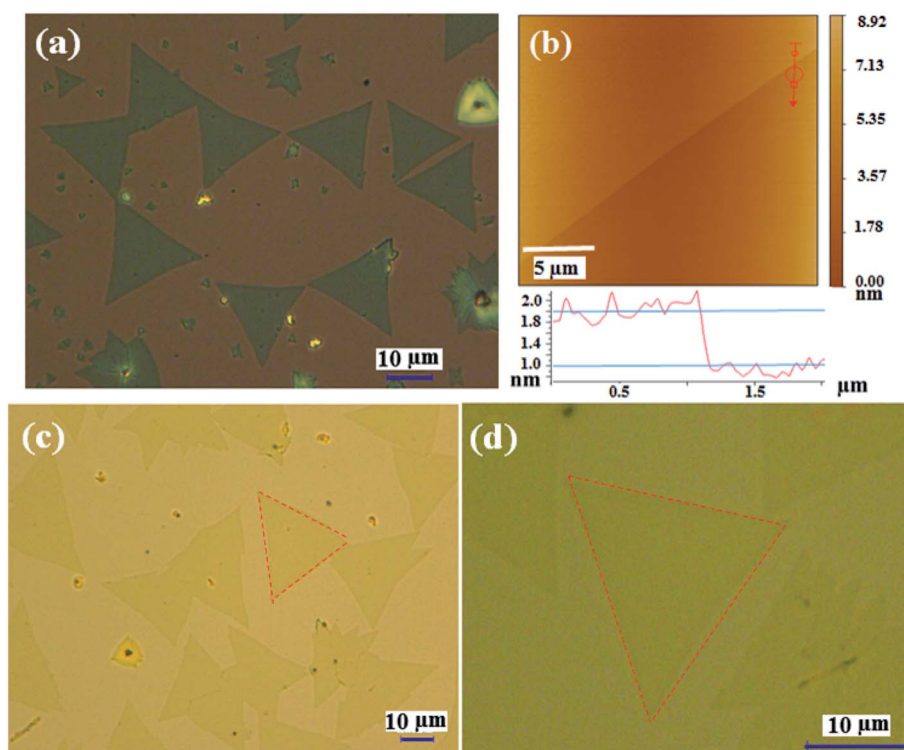


Fig. 1 (a) Optical microscope images of as-synthesized MoS₂ crystals on the SiO₂/Si substrate. (b) AFM image of MoS₂ crystal and line profile presenting the thickness. Optical microscope images of carbonaceous film coated (c) MoS₂ crystals and (d) higher resolution view of a triangular MoS₂ crystal.



thermal evaporation technique using ULVAC VPC-260F. The current–voltage (I – V) characteristics were measured using a two probe system and a Keithley 2401 SourceMeter.

3. Results and discussion

Fig. 1a shows optical microscope image of MoS₂ crystal synthesized on the SiO₂/Si substrate by thermal CVD technique. Triangular shaped MoS₂ crystals with curved edges and size of around 20 μm were obtained in the particular growth condition. Fig. 1b shows an AFM image of the as-synthesized MoS₂ crystal. The line profile shows a thickness of nearly ~ 0.9 nm for the individual MoS₂ crystals. The synthesized samples were transferred to MW-SWP CVD chambers for coating without any other treatment. Microwave excited plasma sources have been used for effective deposition of an insulating encapsulation layer for the conventional semiconductors.^{41,42} We explore the possibility of 2D layered materials encapsulation at room temperature by SW-MWP technique. The MW-SWP can produce an over-dense plasma even at a low gas pressure, with a density higher than the cut-off density ($7.4 \times 10^{16} \text{ m}^{-3}$ for 2.45 GHz). The deposition process was rapid and substrates were placed at a longer distance from plasma shower which allowed to maintain the substrate temperature at ~ 24 °C. Fig. 1c shows an optical microscope

image of the coated MoS₂ crystals. The triangular MoS₂ crystal remain intact without any visible structural damage. Fig. 1d shows a higher resolution optical microscopy image of the MoS₂ triangular crystals, where the morphology is unaffected by deposition of the encapsulation film. Unlike other plasma technique, the MW-SWP is a promising tool for large-area carbonaceous film deposition and useful to avoid plasma induced damages of the substrate surface.⁴³ The details of the deposition process and photograph of the deposited film are included in ESI (Fig. S1 and S2†).

Fig. 2a shows an IR absorption spectra of the carbonaceous film coated on the MoS₂/SiO₂/Si sample (~ 15 nm). Few weak peaks are observed in the frequency range from 1300 to 1600 cm^{-1} , where the peak centered at $\sim 1530 \text{ cm}^{-1}$ can be assigned to C=C stretching vibrations. At a higher frequency range, three sharper peaks at 2870, 2925 and 2961 cm^{-1} were observed corresponding to the C–H vibrations. These absorption peaks can be assigned to $\text{sp}^3 \text{CH}_x$ symmetric and asymmetric bands. Fig. 2b shows a typical Raman spectra of the carbonaceous coating. There was no significant Raman features for the plasma deposited carbonaceous film, other than a broad hump and reduction in intensity of Si Raman peak. A carbon film coated around ~ 150 °C with similar plasma parameters shows broad D and G peaks, clearly indicating the difference in

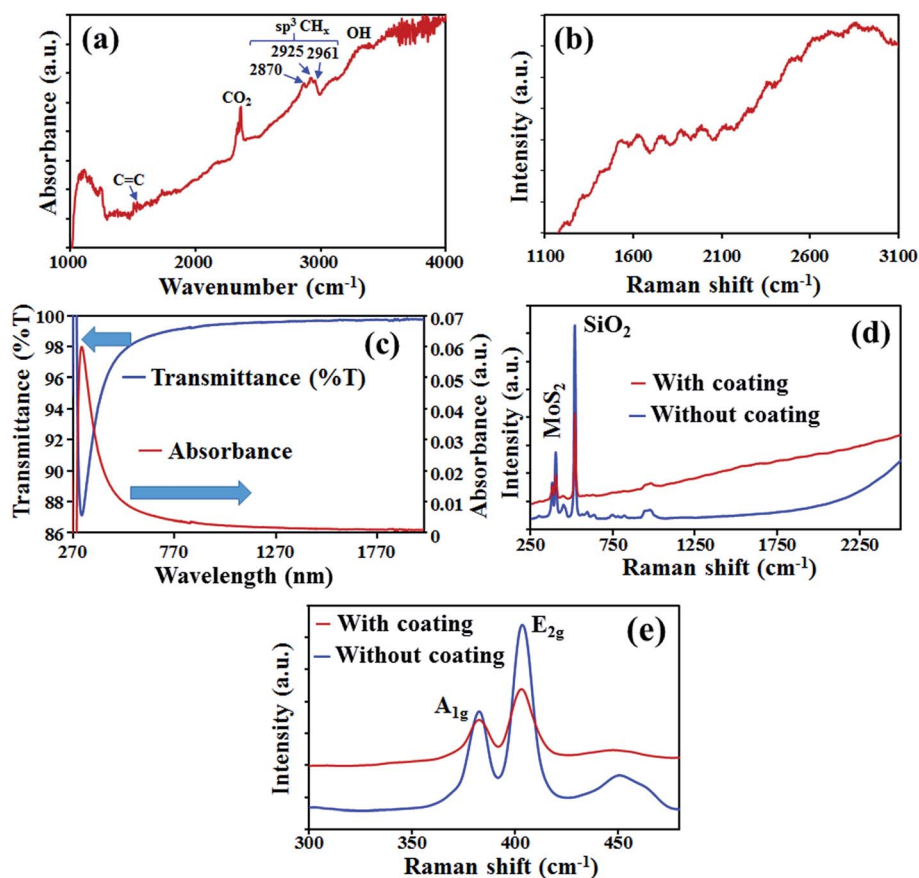


Fig. 2 (a) IR (b) Raman (c) transmittance and absorbance spectra of the carbonaceous film on SiO₂/Si and glass substrates, deposited along with the coating of TMDCs crystals samples. Raman spectra of the (d) MoS₂ crystal on SiO₂/Si substrate (e) A_{1g} and E_{2g} peaks with and without encapsulation film.



carbon structures for room temperature deposited film (Fig. S3†). The presence of sp^3 CH_x bonds and no clear Raman peaks (D and G bands) signifies formation of insulating carbonaceous film with polymeric nature.^{43,44} The room temperature deposited carbon film is significantly different than that of normal crystalline and amorphous carbon forms. Fig. 2c shows transmittance and absorbance of the carbonaceous film deposited on glass substrate. The coated film shows excellent transparency in the wavelength range of 0.3–2 μ m. More than 95% transparency is obtained in the visible light region, which can be tuned with appropriate thickness of the deposited film. The coated film on SiO_2/Si substrate was characterized by Raman spectroscopy with and without the TMDCs crystals. Fig. 2d and e shows Raman spectra of the MoS_2 crystals with and without encapsulated film. The Raman vibration modes E_{2g}^1 and A_{1g} for MoS_2 are observed at 383.2 and 403.3 cm^{-1} , (frequency difference of ~ 20 cm^{-1}) corresponding to in-plane vibration of molybdenum and sulfur and out-of-plane vibration of sulfur atoms, respectively. We observed a reduction in Raman peak intensity for the encapsulated sample, while no significant shift in peak position. Influence of the coated layer was further analyzed by PL study to investigate the possible effect on PL excitation and light emission.

Fig. 3a shows FE-SEM image of the as-synthesized MoS_2 crystals on SiO_2/Si substrate by the CVD technique. The triangular shaped MoS_2 crystals can be confirmed by the FE-SEM analysis. Fig. 3b shows the FE-SEM image of the plasma deposited carbonaceous film on MoS_2 crystals. The surface morphology of the encapsulated MoS_2 sample was observed by FE-SEM analysis using the upper electron detector (UED) and 2 kV accelerating voltage. The triangular MoS_2 crystals can be

confirmed below the encapsulation layer, although the non-conducting encapsulation film significantly affects the micrograph. Previously, plasma functionalization of TMDCs layer has been also demonstrated for modification of electronic properties. PL properties of the encapsulated MoS_2 layer were investigated considering the surface functionalization by the plasma process and possible effect on PL emission.⁴⁵ Fig. 3c shows the PL spectra of coated and non-coated MoS_2 crystal. Quenching and red shift of the PL peak for the encapsulated MoS_2 sample was observed (Fig. 3c). The PL quenching and shift confirm influence of PL excitation and emission of light for the MoS_2 crystal with the encapsulation. Woodhead *et al.* have also reported significant suppression of the PL peak of MoS_2 monolayer by solution coated epoxy due to the compressive strain.³⁹ Compressive strain can modulate the direct to indirect band gap of MoS_2 crystals, which reduces the emission intensity. The required values of strain to induce a shift from direct to indirect bandgaps for MoS_2 and WS_2 are 0.5% and 1% respectively.²² However, the reduction and red shift of the PL peak with the carbonaceous encapsulated layer cannot solely confirm as compressive strain, which was further investigated for the triangular WS_2 crystals.

Fig. 4a shows optical microscope image of CVD synthesized triangular WS_2 crystals on SiO_2/Si substrate. Fig. 4b shows optical microscope image of WS_2 crystals on the SiO_2/Si substrate after coating a thin carbonaceous film (~ 15 nm). Similarly, Fig. 4c shows an optical microscope image of WS_2 sample with a thicker coated layer (~ 30 nm). The difference in coating thickness can be also identified with variation in contrast of the optical images. The WS_2 crystals remain unaffected after the room temperature deposited encapsulation

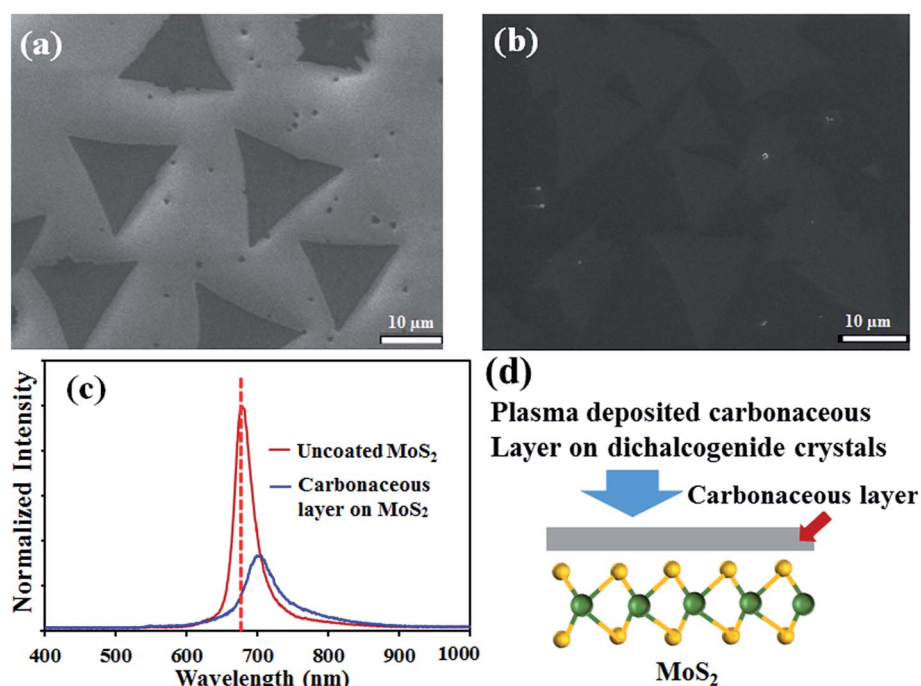


Fig. 3 FE-SEM image of (a) as-synthesized and (b) carbonaceous film coated MoS_2 crystals. (c) Photoluminescence (PL) spectra of the uncoated and coated MoS_2 crystal. (d) Schematic of plasma deposited carbonaceous film encapsulation on MoS_2 crystal.



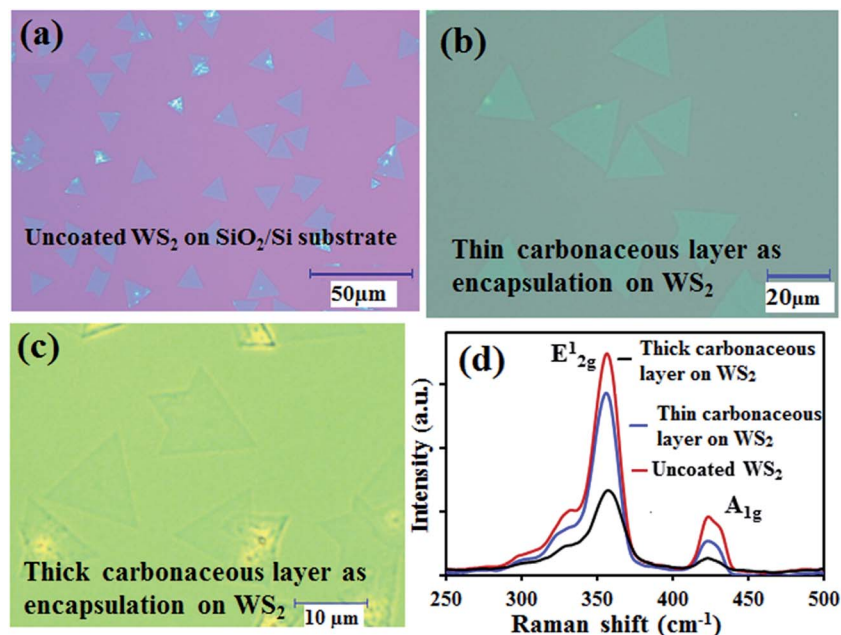


Fig. 4 Optical microscope image of (a) CVD synthesized triangular WS_2 crystals on SiO_2/Si substrate, (b) and (c) MW-SWP deposited carbonaceous film on the CVD synthesized WS_2 crystals with different thickness. (d) Raman spectra of the WS_2 crystal before and after encapsulation.

layer as observed from the optical microscopy images. Further, Raman and X-ray photoelectron spectroscopy (XPS) analysis were carried out to investigate the structural properties. XPS analysis of the as-grown dichalcogenide layers and encapsulated samples are included in ESI (Fig. S4 and S5†). Fig. 4d shows the Raman spectra of the WS_2 crystals before and after coating the encapsulation layer. The as-synthesized WS_2 crystals on SiO_2/Si substrate show characteristic Raman shift at ~ 356 and $\sim 423 \text{ cm}^{-1}$ corresponding to E_{2g} mode for in-plane vibration of tungsten and sulfur atoms and A_{1g} mode for out-of-plane vibration of sulfur atoms, respectively. Although, the Raman shift does not show any considerable change from the original sample, the intensity of Raman peak shows reduction with increasing thickness of encapsulation layer. The structure and the chemical nature of the WS_2 crystals is virtually unaffected by carbonaceous coating as the Raman vibration modes are at the same frequency for as-synthesized as well as coated samples. Raman analysis also does not show substantial evidence of strain, while significant reduction and shift of PL peak were evident.^{36,39,46,47} Fangfei Li *et al.* has reported only small shift in Raman peaks with applying a significantly high pressure (0.84 GPa) to induce a compressive strain.⁴⁶

The encapsulation behavior of WS_2 crystals with the plasma-deposited carbonaceous film was investigated by accelerated heating test in atmospheric condition. Fig. 5a shows FE-SEM image of a pristine triangular WS_2 crystal after heating in atmosphere at 200°C for 1 hour. White traces and their branched structures appear throughout the crystals, signifying the heat-induced degradation of the MX_2 layer under atmospheric conditions. Fig. 5b shows FE-SEM image of the encapsulated WS_2 crystals after the same accelerated heating test. The triangular WS_2 crystals remain intact, however, it is difficult to

identify any kind of traces as observed for the uncoated sample due to the presence of the encapsulation layer. PL analysis was carried out to investigate the encapsulation effect of the WS_2 crystals before and after the heating of the samples. Fig. 5c shows PL spectra for the uncoated and encapsulated WS_2 samples. The intensities were normalized for maximum value to obtain a clear idea of the relative changes in PL. A sharp PL peak at 655 nm (1.89 eV) was obtained for the pristine WS_2 crystals. Significant difference in PL peak was observed with coating a carbonaceous layer on the WS_2 crystals. Coating a thin encapsulation layer ($\sim 15 \text{ nm}$) on the WS_2 crystals, decrease in PL intensity and red shift (shifted to 662 nm corresponding to 1.87 eV) was observed (Fig. 5c). By increasing the thickness of the encapsulation layer ($\sim 30 \text{ nm}$), we observed a much reduced PL peak with the similar red shift. It confirms the influence of the carbonaceous film thickness on the PL excitation as well as the emitted light. Then, PL analysis was performed for the coated and uncoated WS_2 samples after heating at 200°C in atmospheric condition. Fig. 5d shows the normalized PL spectra for the heating-tested samples. After the heating process, both the encapsulated WS_2 samples show similar intensity PL peaks with a peak position at 650 nm (1.90 eV). The PL peak of the uncoated WS_2 crystals was reduced due to the degradation under atmospheric heating as observed from the SEM analysis. Interestingly, the PL peak position of the coated sample slightly blue shifted ($\sim 5 \text{ nm}$) and significantly differed prior to the heating ($\sim 12 \text{ nm}$). The effect is complicated to understand, as different processes can occur, such as reduction in internal absorption of emitted light and change in induced strain.⁴⁸ Then, the coated samples were further tested by heating at a higher temperature (250°C) in atmospheric condition. Fig. 5e shows the optical image of the carbonaceous-coated WS_2



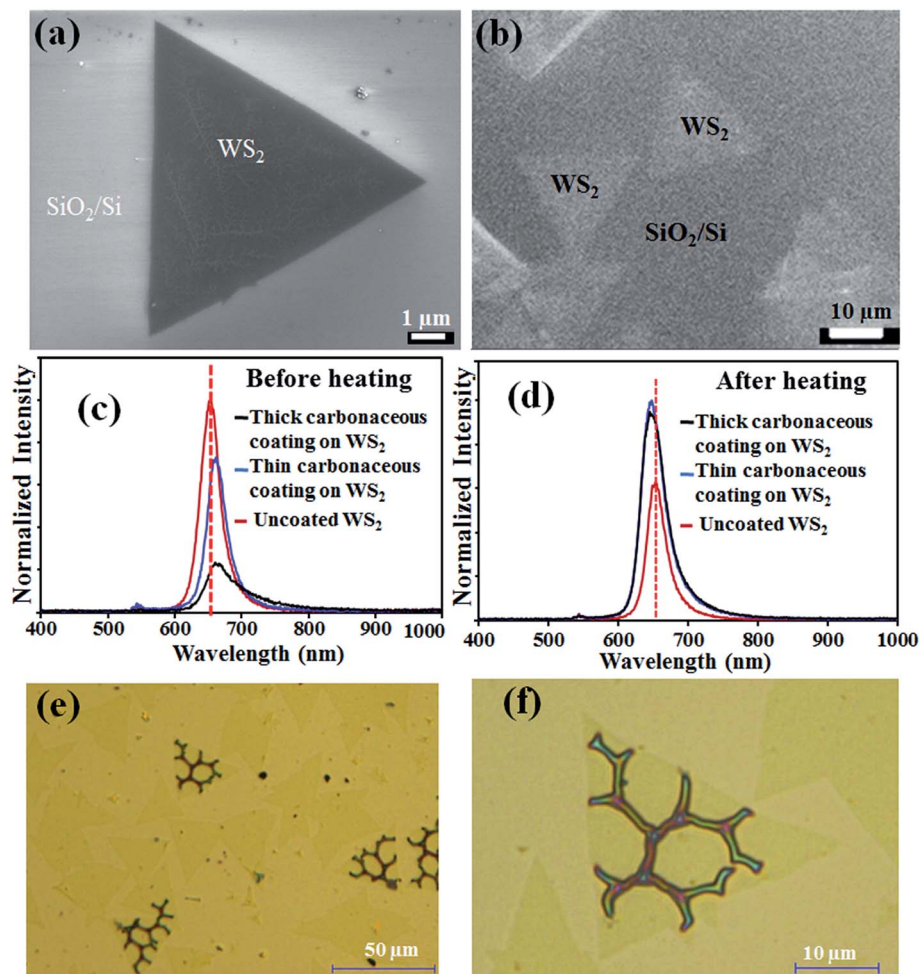


Fig. 5 FE-SEM images of triangular WS_2 crystals (a) without and (b) with coating a carbonaceous film after heating in atmosphere at 200°C for 1 hour. PL spectra for (c) before and (d) after heating of the WS_2 crystals with different thickness of encapsulation layers in comparison with uncoated sample. Optical microscope images of encapsulated (e) WS_2 sample and (f) cracks formation in a WS_2 crystals after heating in atmosphere at 250°C .

sample after the heating experiment. We observed crack formation in some of the WS_2 crystals after the heating process. Fig. 6d shows a higher magnified view of the coated WS_2 crystals with clearly visible cracks. The cracks appeared in the WS_2 crystals, which propagate in hexagonal patterns, with nearly 120° angles. This signifies localized heat induced defect enhancement in the individual encapsulated WS_2 crystals.³⁶ Significantly, we observed the appearance of the cracks only in few WS_2 crystals, where all other crystals remain unaffected. The reason behind this anomaly can be attributed to the difference in defect level of individual crystals and variation in induced strain with thermal expansion differences. It can be expected that better thermal conductivity of the TMDCs crystals will lead to relatively higher local temperature of WS_2 than the other areas of carbonaceous encapsulated sample. Previously, Yan *et al.* has reported the local heat induced thermal expansion properties of the TMDCs layer.⁴⁰

The effect of encapsulation layer on electrical and optical properties of MoS_2 was studied by Schottky junction fabrication. Fig. 6a shows the schematic diagram of the Au/ MoS_2 /Au

device with the carbonaceous layer. The distance between two Au electrodes was approximately $200\ \mu\text{m}$ covering MoS_2 layer in between as shown in the optical microscope image (Fig. 6b). I - V characteristics of the device was measured with (white light, $\sim 100\ \text{mW cm}^{-2}$) and without light illumination conditions. Fig. 6c shows I - V characteristic under dark and light conditions of the as-fabricated device with plasma deposited encapsulated layer. Nonlinear Schottky junction behavior was obtained for the semiconducting MoS_2 layer and Au metal electrode, where the encapsulated layer did not show much influence on junction behavior. With illumination of light, we observed significant photo-response for Schottky junction with photo-exciton in the MoS_2 layer. Then, the device was annealed at 200°C under atmospheric condition for 1 h as discussed above. Fig. 6d shows I - V characteristic of the same device after annealing. The Schottky junction behavior and photo-responsivity was unchanged and no hysteresis was observed with the annealing in presence of the encapsulating layer. Thus, the effectiveness of carbonaceous encapsulating layer on electrical and photo-responsive properties of MoS_2 layer was confirmed.



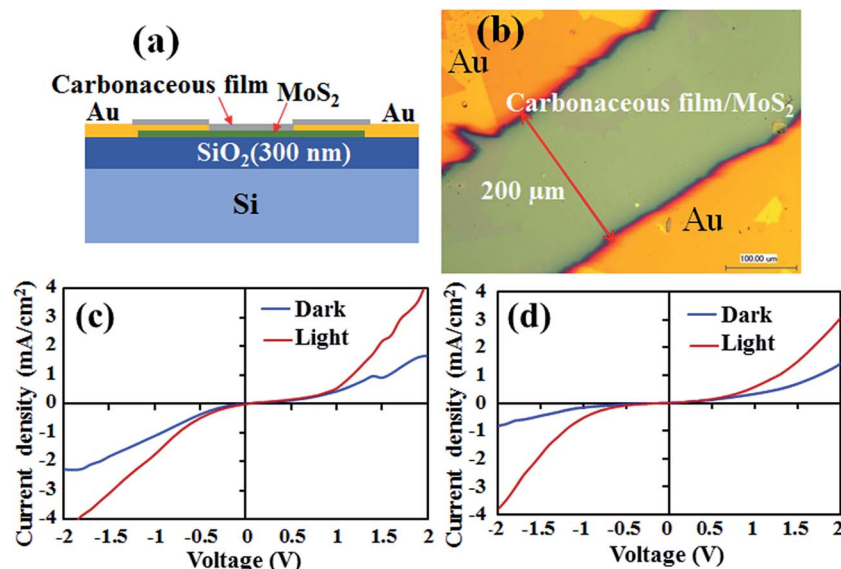


Fig. 6 (a) Schematic diagram and (b) optical microscope image of Au/MoS₂/Au device with encapsulating carbonaceous layer. Current–voltage (*I*–*V*) characteristics of (c) as-fabricated and (d) after annealing at 200 °C for 1 h Au/MoS₂/Au device under dark and light illumination.

4. Conclusion

In conclusion, we have demonstrated deposition of a carbonaceous film at room temperature on MoS₂ and WS₂ crystals by the SW-MWP CVD technique as an encapsulation layer. The encapsulation process is dry, rapid and nondestructive, which can be carried out at room temperature for large area substrate. PL analysis showed the influence on excitation as well as the emission light with the coating of a carbonaceous film. The encapsulated samples remained intact as observed by the optical microscopy, SEM and PL analysis in an accelerated degradation test by heating in ambient condition (humidity ~65%). We also observed change in PL emission for the encapsulated WS₂ layer after heating process, attributing to reduction in internal absorption of emitted light and change in induced strain. We have also demonstrated obtaining stable photoconductivity for the TMDCs dichalcogenide layers with encapsulation of a carbonaceous film. The rapid and dry coating technique at room temperature can be extended to various other 2D materials as MW-SWP process allows damage free encapsulation.

Conflicts of interest

There are no conflicts to declare.

References

- 1 K. F. Mak, C. Lee, J. Hone, J. Shan and T. F. Heinz, *Phys. Rev. Lett.*, 2010, **105**, 136805.
- 2 Q. H. Wang, K. Kalantar-Zadeh, A. Kis, J. N. Coleman and M. S. Strano, *Nat. Nanotechnol.*, 2012, **7**, 699–712.
- 3 K. S. Novoselov, A. K. Geim, S. V. Morozov, D. Jiang, Y. Zhang, S. V. Dubonos, I. V. Grigorieva and A. A. Firsov, *Science*, 2004, **306**, 666–669.
- 4 P. Vogt, P. De Padova, C. Quaresima, J. Avila, E. Frantzeskakis, M. C. Asensio, A. Resta and B. G. Le Lay, *Phys. Rev. Lett.*, 2012, **108**, 155501.
- 5 L. Li, Y. Yu, G. Jun Ye, Q. Ge, X. Ou, H. Wu, D. Feng, X. H. Chen and Y. Zhang, *Nat. Nanotechnol.*, 2014, **9**, 372–377.
- 6 Z. Y. Zeng, C. L. Tan, X. Huang, S. Y. Bao and H. Zhang, *Energy Environ. Sci.*, 2014, **7**, 797–803.
- 7 F. Y. Li, K. X. Tu and Z. F. Chen, *J. Phys. Chem. C*, 2014, **118**, 21264–21274.
- 8 L. Ma, C. Ye, Y. Yu, X. F. Lu, X. Niu, S. Kim, D. Feng, D. Tomanek, Y. W. Son, X. H. Chen and Y. Zhang, *Nat. Commun.*, 2016, **7**, 10956.
- 9 X. Xi, L. Zhao, Z. Wang, H. Berger, L. Forro, J. Shan and K. F. Mak, *Nat. Nanotechnol.*, 2015, **10**, 765–769.
- 10 M. M. Ugeda, A. J. Bradley, Y. Zhang, S. Onishi, Y. Chen, W. Ruan, C. Ojeda-Aristizabal, H. Ryu, M. T. Edmonds, H. Z. Tsai, A. Riss, S. K. Mo, D. H. Lee, A. Zettl, Z. Hussain, Z. X. Shen and M. F. Crommie, *Nat. Phys.*, 2016, **12**, 92–97.
- 11 T. R. Chang, S. Y. Xu, G. Chang, C. C. Lee, S. M. Huang, B. Wang, G. Bian, H. Zheng, D. S. Sanchez, I. Belopolski, *et al.*, *Nat. Commun.*, 2016, **7**, 10639.
- 12 K. Sugawara, Y. Nakata, R. Shimizu, P. Han, T. Hitosugi, T. Sato and T. Takahashi, *ACS Nano*, 2016, **10**, 1341–1345.
- 13 Y. I. Joe, X. M. Chen, P. Ghaemi, K. D. Finkelstein, G. A. de la Pena, Y. Gan, J. C. T. Lee, S. Yuan, J. Geck, G. J. MacDougall, *et al.*, *Nat. Phys.*, 2014, **10**, 421–425.
- 14 B. Radisavljevic, A. Radenovic, J. Brivio, V. Giacometti and A. Kis, *Nat. Nanotechnol.*, 2011, **6**, 147–150.



- 15 W. Yang, J. Shang, J. Wang, X. Shen, B. Cao, N. Peimyoo, C. Zou, Y. Chen, Y. Wang, C. Cong, *et al.*, *Nano Lett.*, 2016, **16**, 1560–1567.
- 16 A. Thangaraja, S. M. Shinde, G. Kalita and M. Tanemura, *Appl. Phys. Lett.*, 2016, **108**, 053104.
- 17 K. F. Mak, K. He, J. Shan and T. F. Heinz, *Nat. Nanotechnol.*, 2012, **7**, 494–498.
- 18 J. K. Ellis, M. J. Lucero and G. E. Scuseria, *Appl. Phys. Lett.*, 2011, **99**, 261908.
- 19 T. Cheiwchanchamnangij and W. R. L. Lambrecht, *Phys. Rev. B*, 2012, **85**, 205302.
- 20 A. Splendiani, L. Sun, Y. Zhang, T. Li, J. Kim, C. Y. Chim, G. Galli and F. Wang, *Nano Lett.*, 2010, **10**, 1271–1275.
- 21 M. Amani, D. H. Lien, D. Kiriya, J. Xiao, A. Azcatl, J. Noh, S. R. Madhupathy, R. Addou, S. Kc, M. Dubey, *et al.*, *Science*, 2015, **350**, 1065–1068.
- 22 Z. Wang, Z. Dong, Y. Gu, Y. Chang, L. Zhang, L. Li, W. Zhao, G. Eda, W. Zhang, G. Grinblat, *et al.*, *Nat. Commun.*, 2016, **7**, 11283.
- 23 R. Cheng, D. Li, H. Zhou, C. Wang, A. Yin, S. Jiang, Y. Liu, Y. Chen, Y. Huang and X. Duan, *Nano Lett.*, 2014, **14**, 5590–5597.
- 24 O. Lopez-Sanchez, D. Lembke, M. Kayci, A. Radenovic and A. Kis, *Nat. Nanotechnol.*, 2013, **8**, 497–501.
- 25 X. Hong, J. Kim, S. F. Shi, Y. Zhang, C. Jin, Y. Sun, S. Tongay, J. Wu, Y. Zhang and F. Wang, *Nat. Nanotechnol.*, 2014, **9**, 682–686.
- 26 M. L. Tsai, S. H. Su, J. K. Chang, D. S. Tsai, C. H. Chen, C. I. Wu, L. J. Li, L. J. Chen and J. H. He, *ACS Nano*, 2014, **8**, 8317–8322.
- 27 G. Eda, H. Yamaguchi, D. Voiry, T. Fujita, M. Chen and M. Chhowalla, *Nano Lett.*, 2011, **11**, 5111–5116.
- 28 Q. Ji, Y. Zhang, Y. Zhang and Z. Liu, *Chem. Soc. Rev.*, 2015, **44**, 2587.
- 29 Y. Zhang, Y. Zhang, Q. Ji, J. Ju, H. Yuan, J. Shi, T. Gao, D. Ma, M. Liu, Y. Chen, *et al.*, *ACS Nano*, 2013, **7**, 8963–8971.
- 30 J. Yang, Y. Gu, E. Lee, H. Lee, S. H. Park, M. H. Cho, Y. H. Kim, Y. H. Kim and H. Kim, *Nanoscale*, 2015, **7**, 9311–9319.
- 31 K. N. Kang, K. Godin and E. H. Yang, *Sci. Rep.*, 2015, **5**, 13205.
- 32 Y. Lee, X. Zhang, W. Zhang, M. Chang, C. Lin, K. Chang, Y. Yu, J. Wang, C. Chang, L. Li and T. Lin, *Adv. Mater.*, 2012, **24**, 2320–2325.
- 33 D. Mahalu, M. Peisach, W. Jaegermann, A. Wold and R. Tenne, *J. Phys. Chem.*, 1990, **94**, 8012–8013.
- 34 Y. Rong, K. He, M. Pacios, A. W. Robertson, H. Bhaskaran and J. H. Warner, *ACS Nano*, 2015, **9**, 3695–3703.
- 35 Y. Liu, C. Tan, H. Chou, A. Nayak, D. Wu, R. Ghosh, H. Y. Chang, Y. Hao, X. Wang, J. Kim, *et al.*, *Nano Lett.*, 2015, **15**, 4979–4984.
- 36 J. Gao, B. Li, J. Tan, P. Chow, T. M. Lu and N. Koratkar, *ACS Nano*, 2016, **10**, 2628–2635.
- 37 J. D. Wood, S. A. Wells, D. Jariwala, K. S. Chen, E. Cho, V. K. Sangwan, X. Liu, L. J. Lauhon, T. J. Marks and M. C. Hersam, *Nano Lett.*, 2014, **14**, 6964–6970.
- 38 S. Ahn, G. Kim, P. K. Nayak, S. I. Yoon, H. Lim, H. J. Shin and H. S. Shin, *ACS Nano*, 2016, **10**, 8973–8979.
- 39 C. S. Woodhead, J. Roberts, Y. J. Noori, Y. Cao, R. Bernardo-Gavito, P. Tovee, A. Kozikov, K. Novoselov and R. J. Young, *2D Mater.*, 2017, **4**, 015032.
- 40 R. Yan, S. Bertolazzi, J. Brivio, T. Fang, A. Konar, A. G. Birdwell, N. V. Nguyen, A. Kis, D. Jena and H. Xing, 2013, arXiv:1211.4136.
- 41 D. R. Cote, G. Y. Lee, S. V. Nguyen, A. K. Stamper, D. S. Armbrust, D. Tobben, R. A. Conti and G. Y. Lee, *IBM J. Res. Dev.*, 1999, **43**, 5–38.
- 42 S. Ueno, Y. Konishi and K. Azuma, *ECS J. Solid State Sci. Technol.*, 2016, **5**, R21–R25.
- 43 J. Xiao, M. Yao, K. Zhu, D. Zhang, S. Zhao, S. Lu, B. Liu, W. Cui and B. Liu, *Nanoscale*, 2013, **5**, 11306–11312.
- 44 G. Kalita, H. R. Aryal, S. Adhikari, D. C. Ghimire, R. A. Afre, T. Soga, M. Sharon and M. Umeno, *Diamond Relat. Mater.*, 2008, **17**, 1697–1701.
- 45 Y. Kim, Y. Jhon, J. Park, C. Kim, S. Lee and Y. Jhon, *Sci. Rep.*, 2016, **6**, 21405.
- 46 F. Li, Y. Yan, B. Han, L. Li, X. Huang, M. Yao, Y. Gong, X. Jin, B. Liu, C. Zhu, Q. Zhou and T. Cuia, *Nanoscale*, 2015, **7**, 9075–9082.
- 47 G. Plechinger, F. X. Schrettenbrunner, J. Eroms, D. Weiss, C. Schuller and T. Korn, *Phys. Status Solidi RRL*, 2012, **6**(3), 126–128.
- 48 C. Hu, X. Zeng, J. Cui, H. Chen and J. Lu, *J. Phys. Chem. C*, 2013, **117**, 20998–21005.

