

Cite this: *RSC Adv.*, 2017, 7, 35283Received 9th June 2017
Accepted 25th June 2017

DOI: 10.1039/c7ra06453e

rsc.li/rsc-advances

The exceptional adsorption ability and gas-detection sensitivity of Cu₂O with tunable morphologies†

Xing Wang, Xuejian Huo, Eliséé Muhire and Meizhen Gao *

The systematic and delicate geometry control of Cu₂O nanostructures with different size can be achieved by simply tuning the dropping speed of NH₂OH HCl, the volume of solvent and the concentration of NaOH. All the Cu₂O nanocrystals exhibit better adsorption ability than granular active carbon on MO (methyl orange), and the hollow sphere Cu₂O shows the best adsorption performance, MO with a concentration of 15 mg L⁻¹ is completely absorbed by the hollow sphere Cu₂O in 5 min. The gas detection sensitivity is also investigated and the hollow sphere structures exhibit the best gas-sensing response, and more importantly, at low temperature and even ppb level it also exhibits a remarkably high response, fast response, recovery time and good stability.

Introduction

It is well known that the electronic structure, bonding, surface energy and chemical reactivity of nanomaterials are closely associated with their shape.^{1–4} These nanomaterials with well-defined shapes and different crystallographic characters have enormous potential applications such as gas sensor, catalysis, magnetic photodecomposition and molecular adsorption.^{5–8} Therefore, it is important to delicately control the synthesized morphologies with different surfaces structures and study their corresponding chemical and physical properties.

As an excellent p-type semiconductor material with a direct band gap of 2.17 eV, the inexpensive, non-toxic and abundantly available Cu₂O has excellent optical and magnetic properties, which can be applied in lithium-ion batteries,^{9,10} gas sensors,^{11–15} organic dye adsorption,^{16,17} catalysis,^{1,18–20} glucose sensing,²¹ and solar-energy conversion.^{22,23} Considering their wide applications, great efforts have been devoted to investigate the controlled synthesis of various shapes of Cu₂O. By adjusting the added PVP, Zhang *et al.* reported a variety of Cu₂O nanocrystals from cubes to octahedrons, and the as-prepared products demonstrate crystallography-dependent adsorption ability with MO.¹⁷ Lee *et al.* reported one-pot synthesis phase-pure Cu₂O nanocubes with a hierarchical structure by employing PEG and glucose as structure-directing agent and reductant, respectively, and the hierarchical nanocubes offer improved accessible surface active sites and optical/electronic

properties.²⁴ Huang *et al.* synthesized sub-100 nm Cu₂O nanocrystals with shape evolution from cubic to octahedral by adjusting the hydrazine volume, and all the samples display high product yields as catalysts in the direct synthesis of 1,2,3-triazoles.²⁵ However, few reports about systematic study of the effect of reductant NH₂OH HCl and precursors on geometry and size of Cu₂O have been found. Therefore a simple and mild way to synthesize Cu₂O nanostructures without any surfactant additives is put forward in this work, and the effect of reducing agent NH₂OH HCl and precursors on Cu₂O morphology and size is discussed in detail. This work provides a straightforward green synthesis method to synthesize the Cu₂O crystals with shape evolution from cube to octahedron and hollow sphere structure, and the size evolution from 800 nm to 100 nm. The as-prepared Cu₂O exhibits excellent adsorption ability to MO and noticeable response to ethanol even at low temperature of 140 °C and 1 ppm.

Experimental

Materials synthesis

All the chemicals, CuCl₂·2H₂O (99%, Reagent), NaOH (96%, Reagent), NH₂OH HCl (98.5%, Kemiou), MO (Ind), powdered active carbon (Kemiou) and granular active carbon (ZK 4.0, Guangzhou Hong Carbon) are analytical grade and used without further purification. For the synthesis of Cu₂O nanocrystals, 0.34 g (2 mmol) CuCl₂ was dissolved into 60 mL distilled water, after stirring for 10 min, a amount of NaOH added into the solution followed by stirring for another 10 min. After that, 10 mL of 2 M NH₂OH HCl solution was slowly added into the blue aqueous as reductant and stirred for 30 min. All the synthesized precipitates were collected and washed with ethanol and water thoroughly, and dried at 50 °C for 12 hours.

Key Laboratory for Magnetism and Magnetic Materials of MOE, School of Physical Science and Technology, Lanzhou University, 730000 Lanzhou, China. E-mail: gaomz@lzu.edu.cn; Tel: +86 931 8914160

† Electronic supplementary information (ESI) available. See DOI: 10.1039/c7ra06453e



Characterization

The structures of Cu_2O are characterized by X-ray diffraction (XRD, X'Pert Philips diffractometer, $\text{Cu K}\alpha$). The morphologies are investigated by field emission scanning electron microscopy (SEM, Hitachi S-4800). Transmission electron microscopic (TEM) images and high-resolution transmission electron microscopic (HRTEM) images are performed on a FEI, Tecnai G2 F30. UV-Vis spectra are recorded on Shimadzu UV 3600.

Adsorption properties research

The adsorption activities of the different shaped Cu_2O are explored using MO as pollutant. 0.05 g Cu_2O was dispersed into 80 mL MO solution, and stirred in the dark, 5 mL of the mixture solution was taken out at different intervals. The concentration of MO is recorded by the UV-Vis spectrum.

Gas sensor application for ethanol detection

Cu_2O sensors are fabricated by depositing slurry of the samples onto the alumina tube with two platinum wires attaching to each gold electrode, and a Ni-Cr alloy filament is put through the alumina tube and used as a heater by tuning the voltage (V). The sensors are aged for 5 days on the measurement system before gas sensing test. Then the gas sensing tests are performed on a WS-30A gas sensing measurement system (Weisheng Instrument Co., Zhengzhou, China). A voltage (5 V) is applied to the heating resistor. The gas response is defined as R_g/R_a , where R_g and R_a are the resistances of the sensor in target gas and air atmosphere, respectively. The response time and recovery time are defined as the time taken by the sensor to achieve 90% of total resistance change in the case of adsorption and desorption, respectively. A working temperature of 240 °C is selected.

Results and discussion

Fig. 1 shows the X-ray diffraction patterns of different samples. All the diffraction peaks can be perfectly indexed to the (110), (111), (200), (211), (220), (311) and (222) planes of cubic phase of Cu_2O (JCPDS NO. 77-0199), and no peaks from other phase have been detected revealing the high purity of the synthesized samples. The average crystal size of cubic, truncated cubic, cubo-octahedral, truncated octahedral, octahedral, small octahedron and hollow sphere are calculated to be about 5.4, 5.1, 4.3, 4.0, 4.6, 2.5 and 2.9 nm, respectively. Which according to the Debye-Scherrer formula ($D = 0.89\lambda/\beta \cos \theta$, where λ is the X-ray wavelength (1.5418 Å), θ is the Bragg diffraction angle of the peak, and β is the peak width at half maximum).²⁶ Fig. 2a-h shows the different morphologies of Cu_2O synthesized at different conditions. When the solvent volume is 150 mL, cubic Cu_2O particles with diameters of about 700 nm with six exposed {100} facets^{17,27-29} can be formed (Fig. 2a). As the solvent volume decreased to 120 mL, the truncated cubic Cu_2O particles with diameter of about 700 nm can be formed. The Cu_2O crystals have 8 triangle {111} with edge length of about 200 nm and 6 square {100} facets with length of about 400 nm (Fig. 2b), and the edge length ratio of {111} facets and {100} facets is 1 : 2. The cubo-octahedral Cu_2O with diameter of about 700 nm as shown in

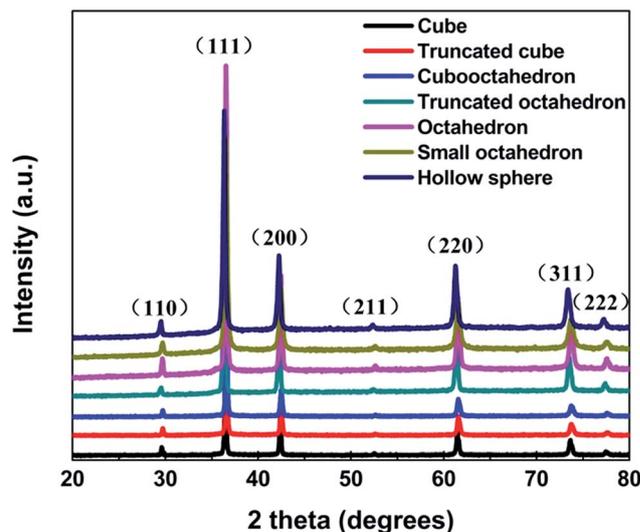
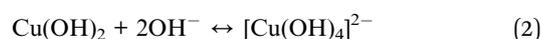


Fig. 1 The XRD patterns of Cu_2O nanocrystals with different geometries.

Fig. 2c can be formed by decreasing the solvent volume to 60 mL. As same as the truncated cubic Cu_2O , the cubo-octahedral Cu_2O crystals also have 8 {111} and 6 {100} facets, but the edge length of hexagonal {111} facets and square {100} facets is about 300 nm and the edge length ratio of {111} facets and {100} facets is about 1 : 1. As the dropping speed of $\text{NH}_2\text{OH HCl}$ solution increasing to 1 drop/1 s, truncated octahedral Cu_2O particles can be formed (Fig. 2d). The truncated octahedral Cu_2O crystals also have 8 {111} and 6 {100} facets, and the long edge length of {111} facet is about 500 nm, the length of square {100} facets is about 250 nm, the length ratio is about 2 : 1. Rapidly pouring the $\text{NH}_2\text{OH HCl}$ into the mixed solution, the high symmetry octahedral Cu_2O exposed only {111} facets can be formed and the edge length of octahedron is about 800 nm (Fig. 2e). With the increase of edge length ratio of {111} and {100} facets, the surface area of exposed {111} facets is also increasing. When the concentration of NaOH is 0.8 M, the final product contains different sizes of Cu_2O ranging from 500 nm to 100 nm (Fig. 2f). As the concentration of NaOH decreases to 0.6 M, the average size of the synthesized Cu_2O crystals is about 100 nm (Fig. 2g). When the concentration of NaOH is 0.5 M, the hollow sphere Cu_2O can be synthesized with the average diameters of 100 nm (Fig. 2h). With the concentration of NaOH decreasing, the size of Cu_2O particles tends to decrease from about 800 nm to 100 nm. The quantified crystallographic features regarding the exposed surfaces of the architectures are proved by TEM and the corresponding selected area electron diffraction (SAED), which is depicted in the ESI (Fig. S2-S4†). The Cu_2O particles can be formed through the following reaction [eqn (1)–(5)]:



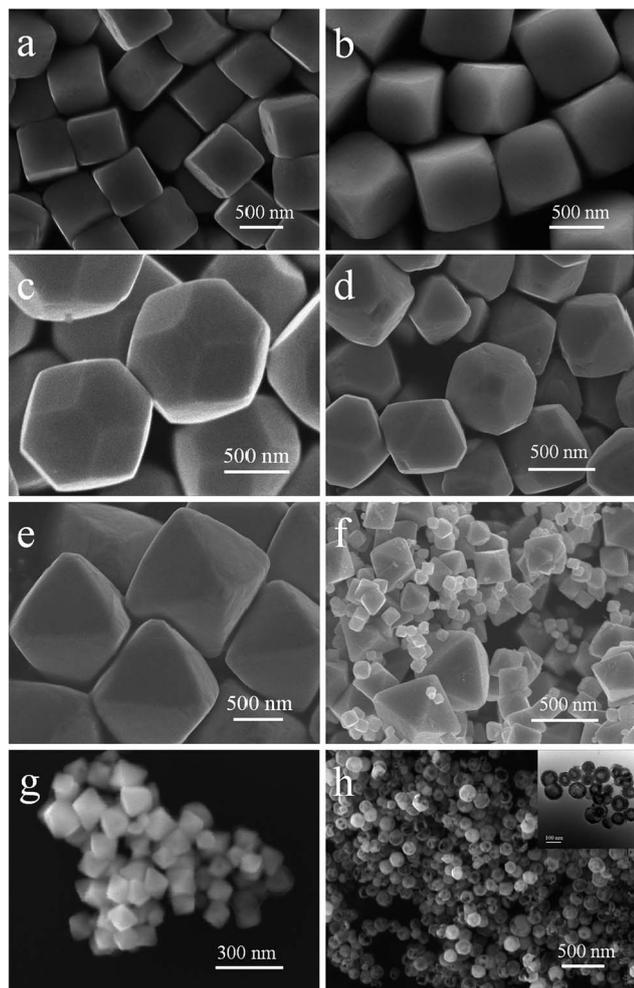
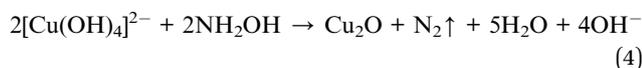


Fig. 2 SEM images of Cu_2O nanostructures with different shapes.



As shown in eqn (3), $\text{NH}_2\text{OH HCl}$ can decompose into NH_2OH and HCl , which is a reversible reaction, and NH_2OH can

deoxidize Cu^{2+} to Cu^+ [eqn (4) and (5)]. The redox reaction occurs as soon as adding the $\text{NH}_2\text{OH HCl}$ into the mixed solution under the vigorous stirring conditions, thus slowly dropping $\text{NH}_2\text{OH HCl}$ can make the concentration of $\text{NH}_2\text{OH HCl}$ in the mixed solution lower than rapidly injecting the $\text{NH}_2\text{OH HCl}$ into the mixed solution. Increasing the volume of solvent and slowly dropping $\text{NH}_2\text{OH HCl}$ into the solution can achieve low concentration of $\text{NH}_2\text{OH HCl}$ in the mixed solution. As illustrated in Fig. 3, increasing the dropping speed of $\text{NH}_2\text{OH HCl}$, the products experienced a shape evolution from cubo-octahedrons to truncated octahedron and finally to octahedron at the volume of 60 mL of solvent. And increasing the volume of solvent, the shape of Cu_2O evolved from cubo-octahedrons to truncated cube and finally to cube while the dropping speed of $\text{NH}_2\text{OH HCl}$ solution keeps 1 drop/2 s. This may due to the concentration of $\text{NH}_2\text{OH HCl}$ will influence the relative growth rate along the [100] direction to that of [111] direction.³⁰ With decreasing concentration of $\text{NH}_2\text{OH HCl}$, Cu_2O tend to grow along [100] direction to form cubo-octahedral, truncated cubic or cubic structure and *vice versa*, it is easier for Cu_2O to grow along the [111] direction to form truncated octahedral or octahedral structure with increasing the concentration of $\text{NH}_2\text{OH HCl}$.

As shown in Fig. 2e–g and 4a, the size of octahedral Cu_2O gradually decreases as the concentration of NaOH decreasing. When the concentration of NaOH is 2 M, almost all the precursor existing in the solution is $[\text{Cu}(\text{OH})_4]^{2-}$, then the octahedrons with uniform size about 800 nm are achieved (Fig. 2e). Visible $\text{Cu}(\text{OH})_2$ suspended particles appear in the solution when the concentration of NaOH is 0.8 M, the final product contains varying size of Cu_2O particles (Fig. 2f). When the concentration of NaOH is 0.6 M, a large amount of $\text{Cu}(\text{OH})_2$ precipitation appear, then small octahedral Cu_2O particles with a size of 100 nm can be formed (Fig. 2g). According to the above analysis, the size of Cu_2O particles may be determined by the different precursors $\text{Cu}(\text{OH})_2$ or $[\text{Cu}(\text{OH})_4]^{2-}$. Regarding the crystallographic surface of Cu_2O , every two 'Cu' atoms has a dangling bond perpendicular to the (111) plane, but (100) plane is saturated.¹⁷ As illustrated in Fig. 4b the octahedral Cu_2O possess positive charge surface {111}, $[\text{Cu}(\text{OH})_4]^{2-}$ is negative, but $\text{Cu}(\text{OH})_2$ is neutral. Thus it is easier for $[\text{Cu}(\text{OH})_4]^{2-}$ to gather around the newborn octahedral Cu_2O

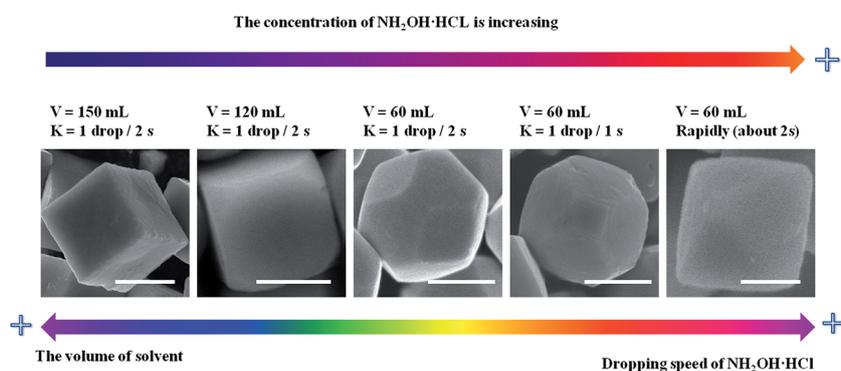


Fig. 3 The evolution of Cu_2O morphology with the change of the concentration of $\text{NH}_2\text{OH HCl}$. Scale bar is equal to 500 nm.



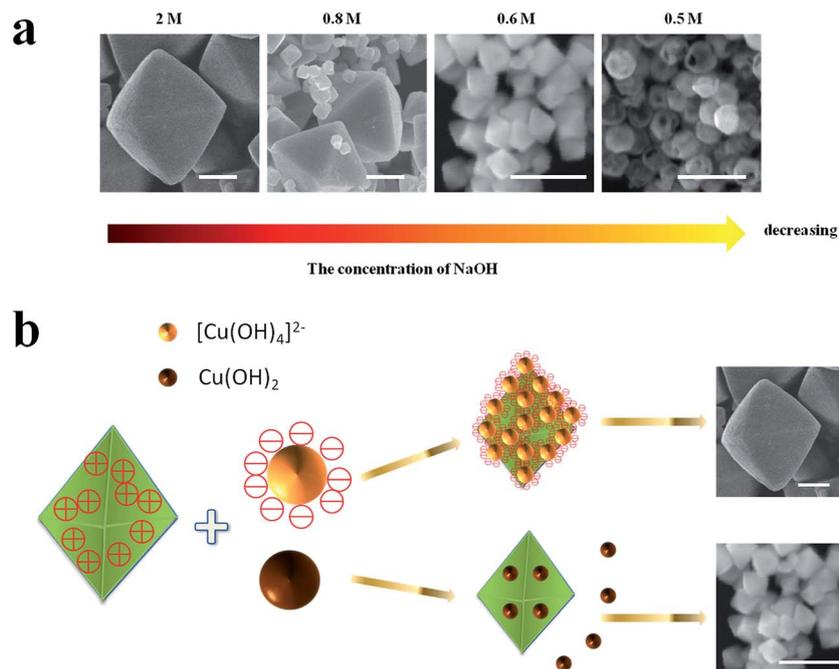


Fig. 4 (a) The size evolution of Cu₂O crystals with the decreasing of the concentration of NaOH, (b) the mechanism of Cu₂O size evolution. Scale bar is equal to 300 nm.

crystals and form large size of Cu₂O crystals. While Cu(OH)₂ disperses throughout the solution instead of gathering around the nuclei as it has no charge, then small size of Cu₂O can be formed. The formation of hollow sphere structure is due to the etching effect of HCl. When the amount of NaOH is 0.5 M, Cu₂O

will be inevitably etched by acid, thus the hollow sphere Cu₂O can be formed.

To explore the adsorption ability of the as-prepared Cu₂O nanocrystals, different concentration of MO is used as the pollutant. Dependence of MO adsorption amount on time is

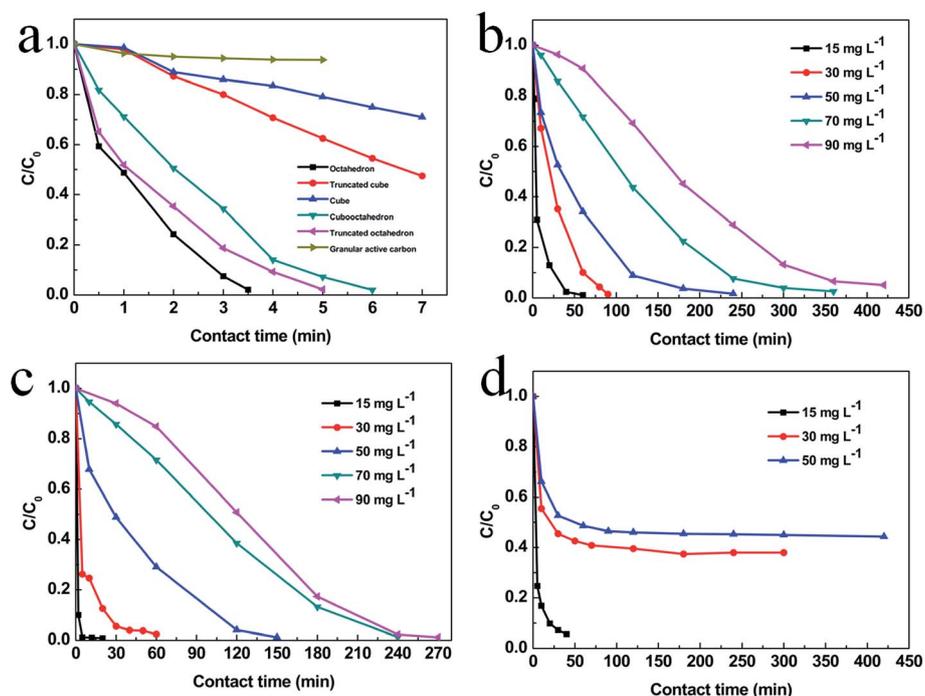


Fig. 5 Adsorption spectra of the MO solution in the presence of Cu₂O with different geometries or active carbon at different intervals: (a) cube to octahedron and granular active carbon, (b) small octahedral Cu₂O, (c) hollow sphere Cu₂O, (d) powdered active carbon.



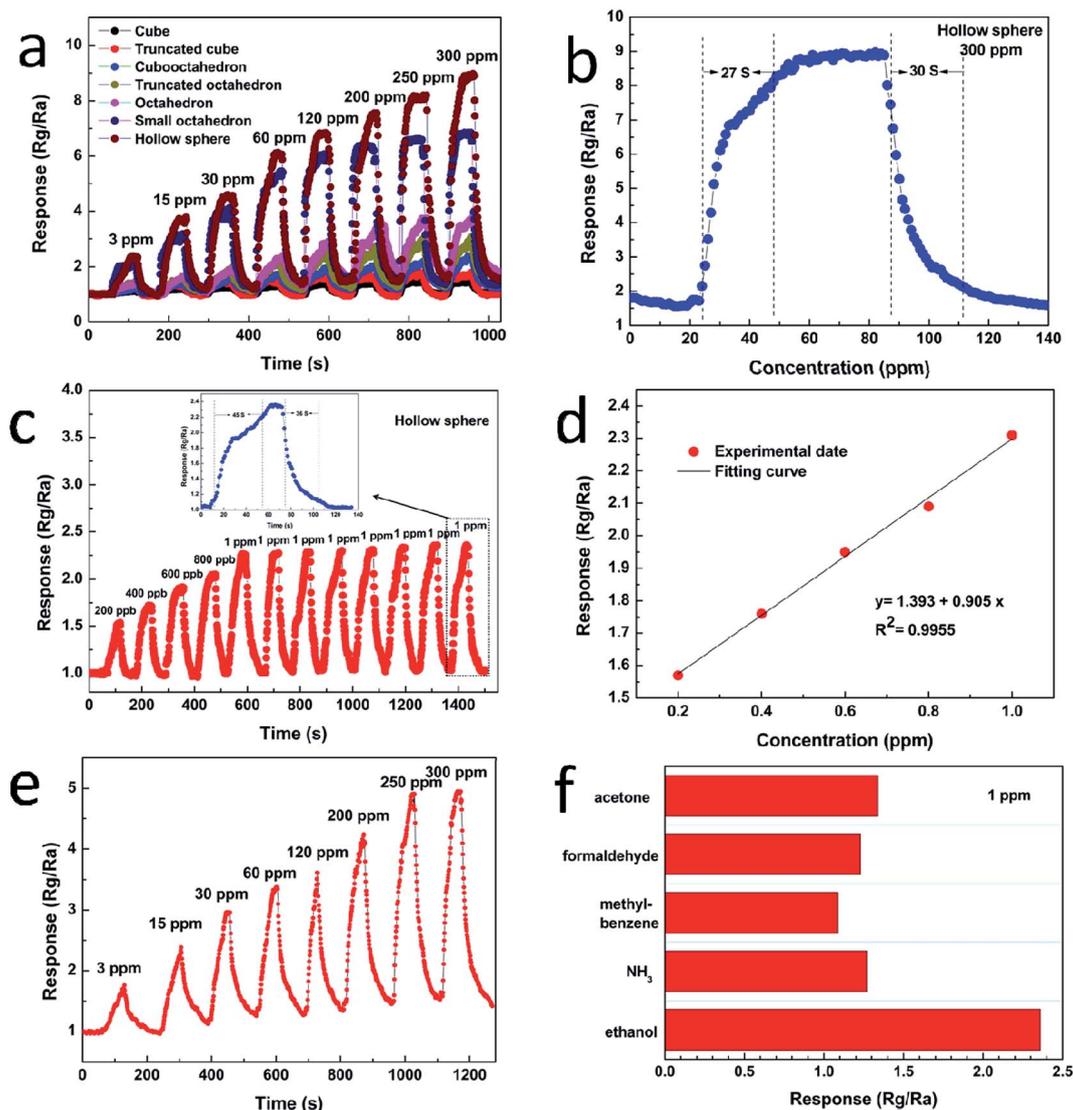


Fig. 6 (a) Dynamic response curves of the gas sensor fabricated by various Cu_2O morphologies to different concentration of ethanol at 240°C , (b) the response and recovery time to 300 ppm ethanol, (c) the repeatability of hollow sphere Cu_2O response to 1 ppm ethanol and response of the sensor to ethanol of 200–1000 ppb, (d) dependence of the response on gas concentration, (e) response curves of gas sensor based on hollow sphere Cu_2O at 140°C , (f) the response of sensor based on hollow sphere Cu_2O to various gases.

shown in Fig. 5. The MO with a concentration of 15 mg L^{-1} is completely adsorbed after 3.5 h, 5 h, 6 h for octahedral, truncated octahedral and cubooctahedral Cu_2O , respectively. For Cu_2O truncated cube and cube, the rest of MO is 41% and 72% respectively after 7 h. But after 7 h, the concentration of MO only reduced to 93% for granular active carbon. The absorbance versus time for small octahedral Cu_2O , hollow sphere Cu_2O and powdered active carbon are shown in Fig. 5b–d. It is found that MO with a concentration of 15 mg L^{-1} is completely adsorbed in 5 min for hollow sphere Cu_2O , and about 40 min are needed for small octahedron Cu_2O and powdered active carbon. Increasing the concentration of MO to 30 mg L^{-1} , 50 mg L^{-1} and 90 mg L^{-1} , the small octahedral and hollow sphere Cu_2O still show better adsorption than powdered active carbon. The corresponding adsorption kinetics of methyl orange on Cu_2O with

various geometries is shown in Fig. S7–S9 and Tables S1–S3.† And the kinetic parameters of various Cu_2O geometries has calculated according to the W – M intraparticle diffusion model and which is shown in Fig. S11 and S10† shows the typical plot of Weber and Moris's intraparticle diffusion model. The better adsorption performance can be attributed to their small grain size and large BET specific surface area, which are 1.7, 1.8, 2.7, 3.7, 2.2, 12.5 and $12.4\text{ m}^2\text{ g}^{-1}$ for cube, truncated cube, cubooctahedron, truncated octahedron, octahedron, small octahedron and hollow sphere Cu_2O respectively. Besides, the hollow sphere Cu_2O shows less package density than other geometries which enhance the adsorption performance further.³¹

Fig. 6a displays the dynamic response–recovery curves of the sensors toward ethanol with the concentration range from 3 to



300 ppm at 240 °C. It can be seen that the resistance of the sensor increases upon injection of ethanol and it is consistent with the sensing behavior of p-type semiconductor sensor. The response amplitude is highly dependent on gas concentration, and increases stepwise with increasing gas concentration. Once again, the hollow sphere Cu₂O exhibits the highest response amplitude to ethanol. Taking 300 ppm of ethanol as example, the response of the hollow sphere is 5.74, 4.96, 3.61, 2.90, 2.21, 1.30 times higher than that of cube, truncated cube, cubooctahedron, truncated octahedron, octahedron and the small octahedron, respectively. Fig. 6b shows the transient response of the sensor based on hollow sphere Cu₂O to 300 ppm ethanol at 240 °C. The response of hollow sphere Cu₂O is about 9.07–300 ppm ethanol. The response time and recovery time are about 27 s and 30 s, respectively. Fig. 6c shows the actual response of the hollow sphere Cu₂O sensor exposed to low-ppb-level ethanol, ranging from 200–1000 ppb. It is found that even a very low concentration of ethanol, *i.e.* 200 ppb, can be effectively detected. The corresponding responses are 1.57, 1.76, 1.95, 2.09 and 2.36 for 200, 400, 600, 800 and 1000 ppb ethanol, respectively. In addition, the reproducibility of hollow sphere Cu₂O sensor is studied with 1 ppm ethanol, the result demonstrates that the sensor nearly maintains its initial value of response during eight tests. The inset (Fig. 6c) is the enlarged response–recovery curve of the sensors to 1 ppm ethanol and the response and recovery times is about 45 s and 36 s, respectively. Fig. 6d shows the dependence of the sensor response on the ethanol gas concentration in the range of 200 ppb to 1 ppm, which is approximately linear, the linearity is $y = 1.393 + 0.905x$ with a relative correlative coefficient $R^2 = 0.9955$ by fitting of the experimental date. Fig. 6e shows the dynamic response of hollow sphere exposed to various concentration of ethanol from 3 to 300 ppm at a lower working temperature of 140 °C. It is exciting to find that the hollow sphere exhibits high sensitivity to ethanol even at lower temperature, the sensitivity increases from 1.62 to 4.91 with the concentration of ethanol increasing from 3 to 300 ppm. Fig. 6f shows the sensitivities of the hollow sphere sensors to 1 ppm of ethanol, NH₃, methylbenzene, formaldehyde and acetone. Obviously, the hollow sphere Cu₂O shows high sensitivity and good selectivity to ethanol.

Cu₂O is a typical p-type semiconductor and its conductivity is highly influenced by the surface depletion layer, which is characteristic of the surface-controlled type of sensing materials, especially for nanostructures.²⁶ According to this characteristic, the adsorbed oxygen molecules on the material surface are partially ionized into ionized oxygen species (O₂⁻, O⁻ and O²⁻) in air, and the sensor resistance stands at a relative lower resistance level. When the gas sensor is exposed to reducing gas *e.g.* ethanol, the electrons species would react with the gas molecules and release electrons back to the valence band, which reduces the electric charge of the depletion layer and increases the layer resistance.

Conclusions

In summary, a simple, surfactant-free synthetic approach has been put forward to synthesize a series of Cu₂O nanocrystals

with the morphology from cube to hollow sphere and the size from 800 nm to 100 nm. The concentration of NH₂OH HCl in the mixed solution can affect the growth rate along the [100] direction. The small octahedral and hollow sphere Cu₂O nanostructures can be formed by different precursors. All the Cu₂O show better adsorption performance than granular active carbon on MO. Furthermore the hollow sphere Cu₂O nanocrystals show the best adsorption performance. In addition, gas sensing investigation shows that the hollow sphere Cu₂O have excellent gas sensitivity which can be applied to ethanol sensor.

Acknowledgements

This work was financially supported by the NSFC (No. 11674143) and the MOE (No. IRT1251 & 20130211130003) of China.

References

- 1 L. Wang, J. Ge, A. Wang, M. Deng, X. Wang, S. Bai, R. Li, J. Jiang, Q. Zhang, Y. Luo and Y. Xiong, *Angew. Chem., Int. Ed.*, 2014, **53**, 5107–5111.
- 2 A. Tao, S. Habas and P. D. Yang, *Small*, 2008, **4**, 310–325.
- 3 Z. G. Liu, Y. F. Sun, W. K. Chen, Y. Kong, Z. Jin, X. Chen, X. Zheng, J. H. Liu, X. J. Huang and S. H. Yu, *Small*, 2015, **11**, 2493–2498.
- 4 X. B. Liu, H. J. Du, P. H. Wang, T. T. Lim and X. W. Sun, *J. Mater. Chem. C*, 2014, **2**, 9536–9542.
- 5 S. Rej, H. J. Wang, M. X. Huang, S. C. Hsu, C. S. Tan, F. C. Lin, J. S. Huang and M. H. Huang, *Nanoscale*, 2015, **7**, 11135–11141.
- 6 M. H. Huang, S. Rej and S. C. Hsu, *Chem. Commun.*, 2014, **50**, 1634–1644.
- 7 Y. Shang and L. Guo, *Adv. Sci.*, 2015, **2**, 1500140.
- 8 K. Mudiyansele, S. D. Senanayake, L. Faria, S. Kundu, A. E. Baber, J. Graciani, A. B. Vidal, S. Agnoli, J. Evans, R. Chang, S. Axnanda, Z. Liu, J. F. Sanz, P. Liu, J. A. Rodriguez and D. J. Stacchiola, *Angew. Chem., Int. Ed.*, 2013, **52**, 5101–5105.
- 9 L. Hu, Y. M. Huang, F. P. Zhang and Q. W. Chen, *Nanoscale*, 2013, **5**, 4186–4190.
- 10 J. C. Park, J. Kim, H. Kwon and H. Song, *Adv. Mater.*, 2009, **21**, 803–807.
- 11 H. Meng, W. Yang, K. Ding, L. Feng and Y. F. Guan, *J. Mater. Chem. A*, 2015, **3**, 1174–1181.
- 12 P. Rai, R. Khan, S. Raj, S. M. Majhi, K. K. Park, Y. T. Yu, I. H. Lee and P. K. Sekhar, *Nanoscale*, 2014, **6**, 581–588.
- 13 L. Guan, H. Pang, J. Wang, Q. Lu, J. Yin and F. Gao, *Chem. Commun.*, 2010, **46**, 7022–7024.
- 14 X. J. Wan, J. L. Wang, L. F. Zhu and J. N. Tang, *J. Mater. Chem. A*, 2014, **2**, 13641.
- 15 Y. M. Sui, Y. Zeng, W. T. Zheng, B. B. Liu, B. Zou and H. B. Yang, *Sens. Actuators, B*, 2012, **171–172**, 135–140.
- 16 X. Q. Ge, H. M. Hu, C. H. Deng, Q. Zheng, M. Wang and G. Y. Chen, *Mater. Lett.*, 2015, **141**, 214–216.
- 17 D. F. Zhang, H. Zhang, L. Guo, K. Zheng, X. D. Han and Z. Zhang, *J. Mater. Chem.*, 2009, **19**, 5220–5225.



- 18 B. Lu, A. Liu, H. Wu, Q. Shen, T. Zhao and J. Wang, *Langmuir*, 2016, **32**, 3085–3094.
- 19 Y. Zhang, B. Deng, T. R. Zhang, D. M. Gao and A. W. Xu, *J. Phys. Chem. C*, 2010, **114**, 5073–5079.
- 20 E. M. Zahran, N. M. Bedford, M. A. Nguyen, Y. J. Chang, B. S. Guiton, R. R. Naik, L. G. Bachas and M. R. Knecht, *J. Am. Chem. Soc.*, 2014, **136**, 32–35.
- 21 D. L. Zhou, J. J. Feng, L. Y. Cai, Q. X. Fang, J. R. Chen and A. J. Wang, *Electrochim. Acta*, 2014, **115**, 103–108.
- 22 L. I. Hung, C. K. Tsung, W. Y. Huang and P. D. Yang, *Adv. Mater.*, 2010, **22**, 1910–1914.
- 23 C. M. McShane and K. S. Choi, *J. Am. Chem. Soc.*, 2009, **131**, 2561–2569.
- 24 S. Kumar, C. M. A. Parlett, M. A. Isaacs, D. V. Jowett, R. E. Douthwaite, M. C. R. Cockett and A. F. Lee, *Appl. Catal., B*, 2016, **189**, 226–232.
- 25 Y. H. Tsai, K. Chanda, Y. T. Chu, C. Y. Chiu and M. H. Huang, *Nanoscale*, 2014, **6**, 8704–8709.
- 26 X. Zhou, J. Y. Liu, C. Wang, P. Sun, X. L. Hu, X. W. Li, K. Shimano, N. Yamazoe and G. Lu, *Sens. Actuators, B*, 2015, **206**, 577–583.
- 27 Q. Hua, T. Cao, X. K. Gu, J. Lu, Z. Jiang, X. Pan, L. Luo, W. X. Li and W. Huang, *Angew. Chem., Int. Ed.*, 2014, **53**, 4856–4861.
- 28 K. Chanda, S. Rej and M. H. Huang, *Chemistry*, 2013, **19**, 16036–16043.
- 29 Q. Hua, D. L. Shang, W. H. Zhang, K. Chen, S. J. Chang, Y. S. Ma, Z. Q. Jiang, J. L. Yang and W. X. Huang, *Langmuir*, 2011, **27**, 665–671.
- 30 C. H. Kuo and M. H. Huang, *J. Phys. Chem. C*, 2008, **112**, 18355–18360.
- 31 C. H. Kuo and M. H. Huang, *J. Am. Chem. Soc.*, 2008, **130**, 12815–12820.

