


Cite this: *RSC Adv.*, 2017, 7, 38387

Computational design of polymers: poly(ester amide) and polyurethane†

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A methodology for conformational analysis of polymers including both ester and amide groups in the repeating unit has been developed by exemplifying a poly(ester amide), poly(oxyethylene-iminosuccinyl) (abbreviated herein as PEA), and a polyurethane, poly(oxy-1,2-ethanedioxydicarbonylimino-1,2-ethanedioyliminocarbonyl) (PU). The repeating units have been divided into two parts, each of which has been represented by a small model compound with the same bond sequence that the polymer includes. To determine which molecular orbital (MO) theory is appropriate for the models, B3LYP, M06-2X, and MP2 calculations were compared with NMR observations. Consequently, the M06-2X or MP2 electronic energy with the B3LYP geometry was shown to yield reliable conformer free energies. Both PEA and PU were found to prefer bent conformations with a number of *gauche* states owing to intramolecular N–H...O and C–H...O=C attractions. The characteristic ratio and configurational entropy, calculated by the refined rotational isomeric state scheme with the MO energies, were, respectively, obtained as follows: 5.39 and $5.8 \times 10^{-2} \text{ cal K}^{-1} \text{ g}^{-1}$ (PEA); and 5.40 and $5.6 \times 10^{-2} \text{ cal K}^{-1} \text{ g}^{-1}$ (PU). Both polymers show equivalent characteristics, however, the ratios (f_U/f) of the internal energy contribution (f_U) to the total restoring force (f) in chain elasticity, evaluated from the temperature coefficient of the characteristic ratio, are greatly different: -0.13 (PEA); and 0.21 (PU). This means that the urethane chain is more likely to behave as an elastomer than the ester amide chain. The possibility that poly(ester amide)s and polyurethanes will become biodegradable is also discussed in terms of the primary structures.

Received 12th May 2017

Accepted 12th July 2017

DOI: 10.1039/c7ra05395a

rsc.li/rsc-advances

Introduction

In state-of-the-art polymer science and engineering, polymeric materials are often required to exhibit different physical properties and functions appropriate for specific applications. For instance, a gene delivery polymer is expected to be highly efficient at transfection, and possess proper hydrophilicity and weak cytotoxicity.¹ However, it is very difficult or essentially impossible to find a synthetic homopolymer that satisfies all the above requirements. As an alternative, one can choose a linear mosaic of different structural units, each of which shows a particular function. In the case of a gene delivery polymer, one may adopt poly(ethylene imine) for transfection, poly(ethylene oxide) for hydrophilicity, and a biodegradable polymer such as poly(L-lactide) or poly(ϵ -caprolactone) for non-cytotoxicity.^{1,2} Herein, such a structural unit is designated as a functional unit (FU)

regardless of whether it is a polymer, an oligomer, or a monomer. In expectation that all the desired functions would be realized, one often synthesizes block copolymers composed of different FUs. If the individual monomeric units keep their inherent characteristics, their random copolymers must also be worth synthesizing and utilizing. Table S1 (ESI†) is a list of various FUs that we have so far characterized in detail. Each FU exhibits distinctive attractive and repulsive interactions due to the electronic structure of the included heteroatom(s), and the interactions result in conformational characteristics of the polymer.

The interaction energies of a given FU can be derived from high-accuracy molecular orbital (MO) calculations on small model compounds that have the same bond sequence that the FU includes. Table S1 (ESI†) also shows types of interaction, values of characteristic ratios ($\langle r^2 \rangle_0/nl^2$) calculated from the rotational isomeric state (RIS) scheme for the polymeric FUs in the Θ state,^{3,4} and the MO theories, which provide reliable conformational energies. Here, r is the end-to-end distance of the polymer, n is the number of skeletal bonds, and l is the bond length. The $\langle r^2 \rangle_0/nl^2$ value is a criterion for the flexibility/rigidity of the FU.

In polymers, the FUs are connected to each other by covalent bonds, whereas supramolecules are assembled with the aid of non-covalent linkages such as hydrogen bonds and weak

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† Electronic supplementary information (ESI) available: Characteristics of various functional units (Table S1); observed vicinal coupling constants of EA-1 and EA-1-¹³C (Table S2); *trans* fractions of EA-1 determined from NMR experiments (Table S3); *trans* fractions of EA-1 evaluated from MO calculations (Table S4); observed vicinal coupling constants of U-1 and U-2 (Table S5); and *trans* fractions of U-1 and U-2 (Table S6). See DOI: 10.1039/c7ra05395a



attractive interactions that are reconfigurable and reconnectable.^{5,6} Therefore, to fabricate supramolecules as desired, we must recognize the attraction and repulsion of the FUs in advance and design geometrical relations between the non-covalent connections. It is particularly important to design the molecular architecture from a stereochemical viewpoint because we are too accustomed to two-dimensional structural formulas such as those in Fig. 1, where two FUs, a poly(ester amide) and a polyurethane, are depicted in the all-*trans* conformation and the carbonyl groups stick out alternately in opposite directions. As will be shown later, however, the most stable conformations of the two FUs are largely bent; each FU is represented by two model compounds, whose most stable forms are shown in Fig. 2.

The amide group ($-\text{NHC}(=\text{O})-$) forms comparatively strong hydrogen bonds in natural and synthetic products: in α -helices and β -sheets in polypeptides; in adenine–thymine and guanine–cytosine pairs in DNA; and in parallel and anti-parallel pleated sheets of nylons. The amide group plays a major role in the formation of the higher-order structure and acts as a juncture in supramolecules.⁷ The $\text{NHC}(=\text{O})$ atoms are located on a plane. However, if the carbonyl oxygen atoms of a planar aromatic amide $-\text{NHC}(=\text{O})-\text{C}_6\text{H}_4-\text{C}(=\text{O})\text{NH}-$ are replaced with sulfur, the resultant thioamide group, $-\text{NHC}(=\text{S})-\text{C}_6\text{H}_4-\text{C}(=\text{S})\text{NH}-$, adopts six rotational isomeric states around the benzene ring.⁸ As the energy differences between the six rotamers are as small as $\sim 0.1 \text{ kcal mol}^{-1}$, the thioamide group undergoes nearly free rotation. In addition, the $\text{N-H}\cdots\text{S}=\text{C}$ attraction is weaker than the $\text{N-H}\cdots\text{O}=\text{C}$ attraction. Therefore, we must investigate rotational isomeric states and interactions of the amide group and its variants individually.

This study deals with a poly(ester amide), poly(oxyethylene-iminosuccinyl) (abbreviated as PEA and illustrated in Fig. 1a), and a polyurethane, poly(oxy-1,2-ethanediylloxycarbonylimino-

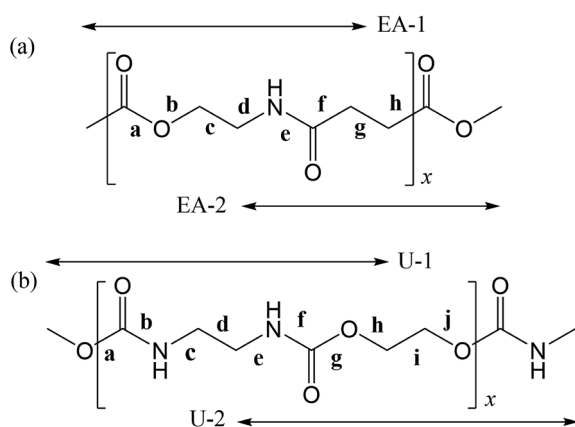


Fig. 1 (a) Poly(oxyethylene-iminosuccinyl) (abbreviated as PEA) and (b) poly(oxy-1,2-ethanediylloxycarbonylimino-1,2-ethanediylimino-carbonyl) (PU). The repeating units are divided into two parts, which are represented by small model compounds: 2-acetamidoethyl acetate (designated herein as EA-1) and methyl 4-(methylamino)-4-oxobutanoate (EA-2) for PEA; and dimethyl ethane-1,2-diyl dicarbamate (U-1) and ethane-1,2-diyl bis(methylcarbamate) (U-2) for PU. For the model compounds, see Fig. 2.

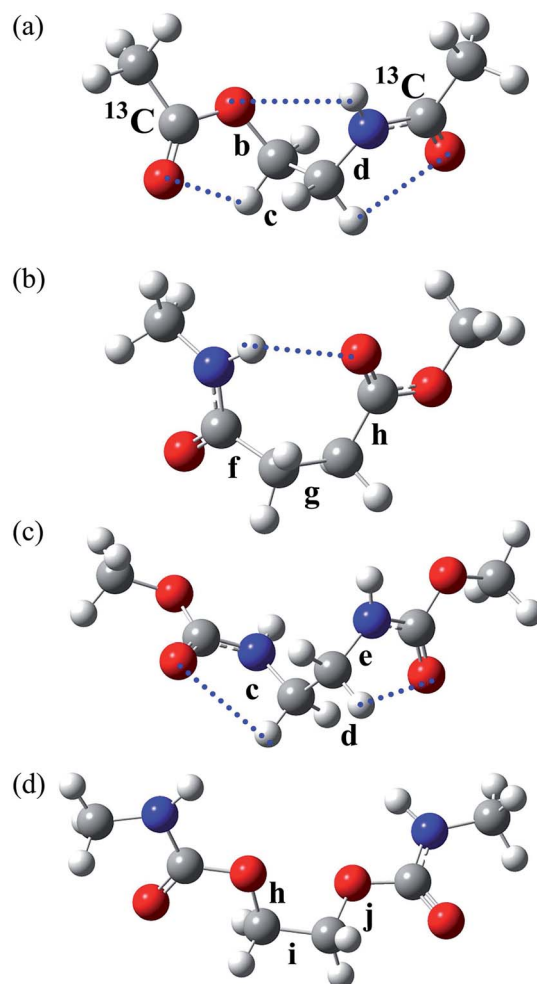


Fig. 2 The most stable conformers (with the lowest ΔG_R) of the model compounds: (a) EA-1, $g^+g^+g^+$; (b) EA-2, g^+g^-t ; (c) U-1, $g^+g^+g^+$; and (d) U-2, tg^+t . The rotatable bonds are designated as shown, and the symbols correspond to those shown in the polymers illustrated in Fig. 1. The EA-1 compound, whose carbonyl carbon atoms are labelled with ^{13}C , was also prepared for NMR measurements and designated as EA-1- ^{13}C . The dotted lines represent weak attractive interactions predicted by MO calculations.

1,2-ethanediyliminocarbonyl) (abbreviated as PU and illustrated in Fig. 1b). The monomeric unit of PEA consists of two ethylene groups, one amide group, and one ester group, and that of PU comprises two ethylene and two urethane groups. The urethane group corresponds to a combined form of an ester and an amide unit. Thus, PEA and PU are quite similar in terms of their primary structure. Both amide and urethane groups may form $\text{N-H}\cdots\text{O}=\text{C}$ hydrogen bonds, and the $\text{C}=\text{O}$ bond of the ester group would also participate in hydrogen bonding. Therefore, these FUs are useful for us to understand the conformational characteristics and physical properties of both ester and amide groups in these compounds. In this study, we eventually investigated some rather complicated FUs *via* a number of FUs listed in Table S1 (ESI[†]) and attempted to establish a method for conformational analysis of poly(ester amide)s and polyurethanes.



It is also of interest to us that poly(ester amide)s and polyurethanes have been used for practical materials and recently have become increasingly attractive, partly because some of them possess biodegradability.^{9,10} The PEA and PU polymers studied here were already synthesized but nevertheless have been hardly characterized: PEA was prepared from succinic anhydride and 2-amino-1-ethanol and was found to be semi-crystalline (weight-average molecular weight: 13.1 kDa and melting point (T_m): 190 °C).¹¹ Semi-crystalline PU was prepared by interfacial polycondensation between ethylene diamine and ethylene bis(chloroformate), and its T_m was reported as 225 °C.¹²

At the beginning of this study, we attempted to establish a method for MO calculations appropriate for the aliphatic ester-amide and urethane FUs. Our studies have proven that geometrical optimization by the B3LYP density functional and electronic energy calculation by the MP2 theory with basis sets of medium sizes usually yield satisfactory results, as shown in Table S1 (ESI†). As far as aromatic polyesters and polyamides are concerned, however, the MP2//B3LYP combination fails to reproduce NMR observations.^{8,13,14} This is because the MP2 theory tends to overestimate π/π and C-H/ π attractions of aromatic polyesters, and the self-consistent reaction field (SCRf) calculations can not satisfactorily reproduce the effects of irritant and polar solvents on hydrogen bonds of polyamides. Therefore, we have searched for a MO theory and solvation model suitable for PEA and PU through comparison with NMR experiments on the model compounds shown in Fig. 2. On the basis of the reliable MO calculations thus established, we have investigated the conformational characteristics and intramolecular interactions of the two FUs, and evaluated the configurational properties of the two polymers by the refined RIS scheme.¹⁵ This article reports the processes and results in detail and discusses the differences between PEA and PU, especially in terms of chain elasticity and biodegradability.

Methods

Synthesis of EA-1 and EA-1-¹³C

Acetyl chloride (0.91 mL, 13 mmol) was added dropwise with a syringe to 2-aminoethanol (0.31 mL, 5.1 mmol) and pyridine (4.1 mL, 51 mmol), and then the mixture was stirred under a nitrogen atmosphere at 40 °C for 27 h. After the solvent was removed on a rotary evaporator, diethyl ether was added to the residue to precipitate pyridinium chloride, which was removed by filtration. The filtrate was condensed, and the residue was dried *in vacuo* overnight to yield a yellowish viscous liquid, which was identified as EA-1 by ¹H and ¹³C NMR spectroscopy. In preparation of EA-1-¹³C, acetyl chloride-¹³C was used instead of acetyl chloride.

Synthesis of U-1

Ethylenediamine (1.2 mL, 18 mmol), dimethyl carbonate (3.0 mL, 36 mmol), sodium acetate (0.36 g, 4.4 mmol), and ethanol (9.0 mL) were mixed in a four-necked flask equipped with a reflux condenser connected to a drying tube of calcium chloride.¹⁶ The mixture was stirred under a nitrogen atmosphere at

75 °C for 15 h. After ethanol was removed on a rotary evaporator, the residue was dried under reduced pressure at 50 °C to yield a white solid. The crude product was dissolved in chloroform and filtrated to remove sodium acetate. The filtrate was condensed, dried *in vacuo* at 50 °C for 2 h, dissolved in toluene (50 mL) at 80 °C, and allowed to stand still to form white crystallites, which were collected and dried *in vacuo* at 50 °C for 2 h. The product was identified by ¹H and ¹³C NMR spectroscopy as U-1.

Synthesis of U-2

Ethylene carbonate (44 g, 0.5 mol) and methylamine aqueous solution (40%, 61.5 mL) were mixed in a four-necked flask equipped with a reflux condenser and a thermometer and stirred at 15–20 °C for 1 h.¹⁷ After water was removed on a rotary evaporator, the residue was distilled under reduced pressure to collect a fraction at *ca.* 170 °C and 7 mmHg. The product was 2-hydroxyethyl methylcarbamate.

The prepared 2-hydroxyethyl methylcarbamate (3.8 g, 32 mmol) and dibutyltin dilaurate (0.40 g, 0.63 mmol) were stirred in an eggplant flask at 170 °C for 1 h and then distilled at 170 °C and 30 mmHg to remove a by-product, ethylene glycol. The residue underwent suction filtration to be separated into a white cake and a yellow filtrate. The former was dissolved in toluene (50 mL), heated to 80 °C, cooled to room temperature, and allowed to stand still to form white crystallites, which were collected by filtration and dried *in vacuo* at 30 °C for 2 h. The product was identified by ¹H and ¹³C NMR spectroscopy as U-2.

NMR measurements

Proton (¹³C) NMR spectra were measured at 500 MHz (126 MHz) on a JEOL JNM-ECA500 spectrometer in the Center for Analytical Instrumentation of Chiba University. The sample temperature was changed step-wise from 15 or 25 to 55 °C at intervals of 10 °C and maintained there within ± 0.1 °C. Free induction decays of 32k (64k) points were accumulated 32 (128) times, fully zero-filled, and underwent Fourier transform. The pulse width and recycle delay were 5.7 (3.27) μ s and 3.0 (2.0) s, respectively. In the ¹³C NMR experiments, the gated decoupling technique was used under the conditions written in parentheses. The above numerical data are typical examples; the experimental conditions were adjusted to each sample. The NMR solvents were benzene-*d*₆ (C₆D₆), chloroform-*d* (CDCl₃), methanol-*d*₄ (CD₃OD), dimethyl-*d*₆ sulfoxide (DMSO-*d*₆, (CD₃)₂SO), and deuterium oxide (water-*d*₂, D₂O), and 5 mm NMR sample tubes were used. The ¹H and ¹³C NMR spectra thus obtained were simulated with the gNMR program¹⁸ to yield chemical shifts and coupling constants.

MO calculations on model compounds

Density functional theory and *ab initio* MO calculations were carried out with the Gaussian09 program¹⁹ installed on a Hitachi SR16000 server in the Institute of Management and Information Technologies of Chiba University. For each conformer of the model compounds, the geometrical parameters were fully optimized at the B3LYP/6-311+G(2d,p) level²⁰ under tight



convergence, and then the thermochemical energies at 25 °C and 1 atm were evaluated by the frequency computations at the same level. With the optimized geometry, the electronic energy was computed at the MP2/6-311+G(2d,p)²¹ and M06-2X/6-311+G(2d,p) levels.²² The Gibbs free energy was evaluated from the electronic and thermochemical energies, being expressed herein as the difference from that of a given conformer and represented as ΔG_k (k : conformer). The solvent effect on the electronic energy was evaluated by the polarizable continuum model (PCM) using the integral equation formalism variant (IEFPCM)²³ and the conductor-like PCM model (CPCM).²⁴

The geometrical parameters required for the refined RIS calculations on PEA and PU were chosen out of those optimized for the model compounds. Vicinal ^1H - ^1H and ^{13}C - ^1H coupling constants for the NMR analysis were calculated at the B3LYP/6-311++G(3df,3pd) level.²⁵

Herein, the dihedral angle is defined according to the tradition in polymer science:³ *trans* (t) $\sim \pm 0^\circ$ and *gauche*[±] (g^\pm) $\sim \pm 120^\circ$. The dihedral angle here (ϕ) is related to that (Φ) recommended by IUPAC²⁶ according to $\Phi = -\text{sign}(\phi)(180 - |\phi|)$, and *vice versa* $\phi = -\text{sign}(\Phi)(180 - |\Phi|)$, where the function, $\text{sign}(\phi)$, returns the sign (plus or minus) of ϕ .

Results and discussion

Conformational analysis of EA-1 and EA-2

NMR spectra of EA-1. Fig. 3 shows the ^1H NMR spectra observed from EA-1 and EA-1- ^{13}C dissolved in $(\text{CD}_3)_2\text{SO}$ at 25 °C along with the results of the gNMR simulations. Fig. 3 also shows the ^{13}C NMR spectra of EA-1- ^{13}C whose carbonyl carbon atoms are labelled with ^{13}C . All the spectra clearly show that the simulations satisfactorily reproduced the experiments, and the vicinal coupling constants thus derived are listed in Tables 1 and S2 (ESI†).

In Table 2, the vicinal coupling constants are expressed as a function of bond conformation, *i.e.* the *trans* (p_t) and *gauche* (p_g) fractions. The coefficients (J_C , J'_E , J_G , and J_T) of the equations, defined in Fig. 4, were chosen from the well-known Karplus equations, a cyclic compound with the same bond sequence that EA-1 has, and MO calculations on EA-1 itself. Their sources and numerical values are described in footnotes of Table 3. The *trans* fractions of the three bonds, O-C, C-C, and C-N, of EA-1, obtained from the equations in Table 2, are listed in Tables 3 and S3 (ESI†). The p_t values seem to depend somewhat on the used coefficients but generally stay within narrow ranges. The *trans* fractions of the C-N bond were derived from J_{CH}^{N} and J_{HH}^{N} ; however, the results do not agree well for the $(\text{CD}_3)_2\text{SO}$ solution. This is partly because the NH proton undergoes strong solvent effects when EA-1 is dissolved in polar solvents. Actually, the J_{HH}^{N} value could not be obtained from the D_2O solution. Nevertheless, we can clearly conclude that the O-C, C-C, and C-N bonds possess *trans*, *gauche*, and *gauche* preferences, respectively.

MO calculations for EA-1 and EA-2. As stated in the Introduction, we searched for the MO calculations appropriate for molecules including both amide and ester groups. The geometrical optimization at the B3LYP/6-311+G(2d,p) level was

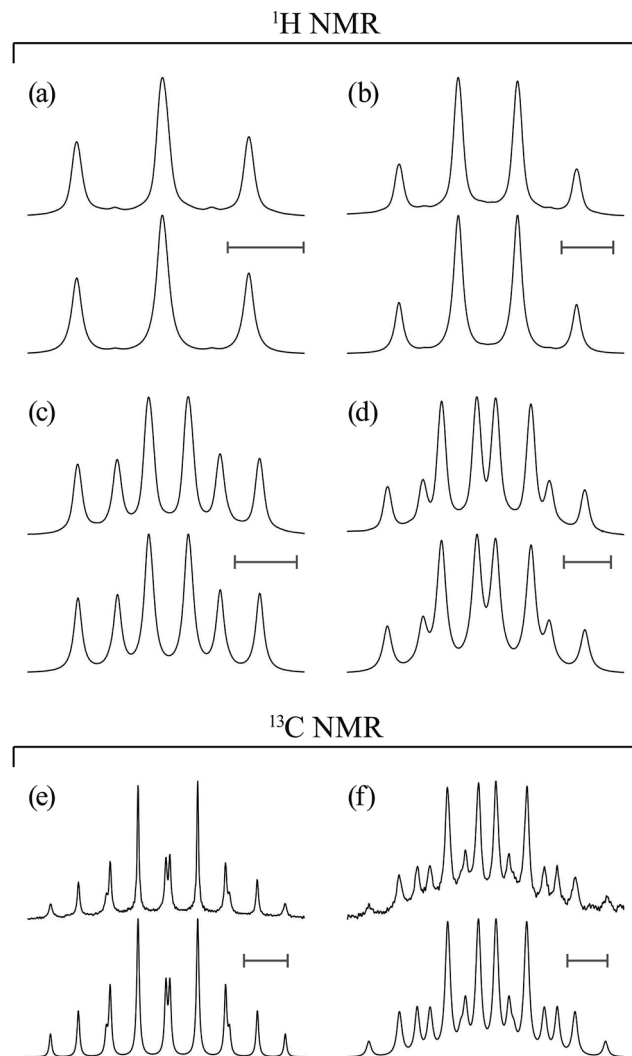


Fig. 3 ^1H and ^{13}C NMR spectra observed from (a and b) EA-1 and (c–f) EA-1- ^{13}C dissolved in $(\text{CD}_3)_2\text{SO}$ at 25 °C (above), together with the results of spectrum simulations (below): (a) $\text{C}(=\text{O})-\text{O}-\text{CH}_2$; (b) $\text{CH}_2-\text{NH}-\text{C}(=\text{O})$; (c) $^{13}\text{C}(=\text{O})-\text{O}-\text{CH}_2$; (d) $\text{CH}_2-\text{NH}-^{13}\text{C}(=\text{O})$; (e) $^{13}\text{C}(=\text{O})-\text{O}-\text{CH}_2$; and (f) $\text{CH}_2-\text{NH}-^{13}\text{C}(=\text{O})$. All the scale bars represent 5 Hz.

Table 1 Observed vicinal ^1H - ^1H and ^{13}C - ^1H coupling constants of EA-1 and EA-1- $^{13}\text{C}^a$

Solvent ^b	Temp. (°C)	J_{CH}^{O}	J_{HH}	J'_{HH}	J_{CH}^{N}	J_{HH}^{N}
C_6D_6	15	3.33	6.66	4.72	3.50	5.85
	25	3.33	6.66	4.72	3.50	5.85
	35	3.35	6.60	4.72	3.50	5.85
	45	3.36	6.60	4.78	3.50	5.85
	55	3.38	6.60	4.79	3.50	5.85
CDCl_3	25	3.18	6.78	3.90	3.32	5.74
$(\text{CD}_3)_2\text{SO}$	25	3.19	6.52	4.96	3.76	5.70
D_2O	25	3.13	6.76	3.80	3.80	— ^c

^a In Hz. For more detailed data, see Table S2 (ESI†). ^b Symbols: C_6D_6 , benzene- d_6 ; CDCl_3 , chloroform- d_3 ; $(\text{CD}_3)_2\text{SO}$, dimethyl- d_6 sulfoxide (abbreviated as DMSO- d_6); D_2O , deuterium oxide (water- d_2). ^c Not available.



Table 2 Vicinal coupling constants as a function of bond conformation

Compound	Bond ^a	Coupled nuclei	Equation ^b	Table	Figure
EA-1 & EA-1- ¹³ C	b: O–C	¹³ C(=O)O–CH ₂	$J_{CH}^O = [(J_G + J'_G)/2] p_t + [(J'_T + J'_G)/2] p_g$	1, S2 (ESI†)	4
		CH ₂ –CH ₂	$J_{HH} = J_G p_t + [(J'_T + J'_G)/2] p_g$	1, S2 (ESI†)	4
	c: C–C		$J'_{HH} = J_T p_t + [(J'_G + J''_G)/2] p_g$	1, S2 (ESI†)	
		CH ₂ –NH ¹³ C(=O)	$J_{CH}^N = [(J_G + J'_G)/2] p_t + [(J'_T + J'_G)/2] p_g$	1, S2 (ESI†)	4
		CH ₂ –NH	$J'_{HH} = [(J_C + J'_C)/2] p_t + [(J'_E + J'_C)/2] p_g$	1, S2 (ESI†)	
U-1	c & e: N–C	NH–CH ₂	$J_{HH}^N = [(J_C + J'_C)/2] p_t + [(J'_E + J'_C)/2] p_g$	S5 (ESI†)	7
U-1 & U-2	d & i: C–C	CH ₂ –CH ₂	$J_{HH} = J_G p_t + [(J'_T + J'_G)/2] p_g$	S5 (ESI†)	7
			$J'_{HH} = J_T p_t + J'_G p_g$	S5 (ESI†)	

^a See Fig. 2. ^b By definition, $p_t + p_g = 1$. For the coefficients (J_T , J_G , J_C , and J'_E) of the equations, see the footnotes of Tables 3 and 7.

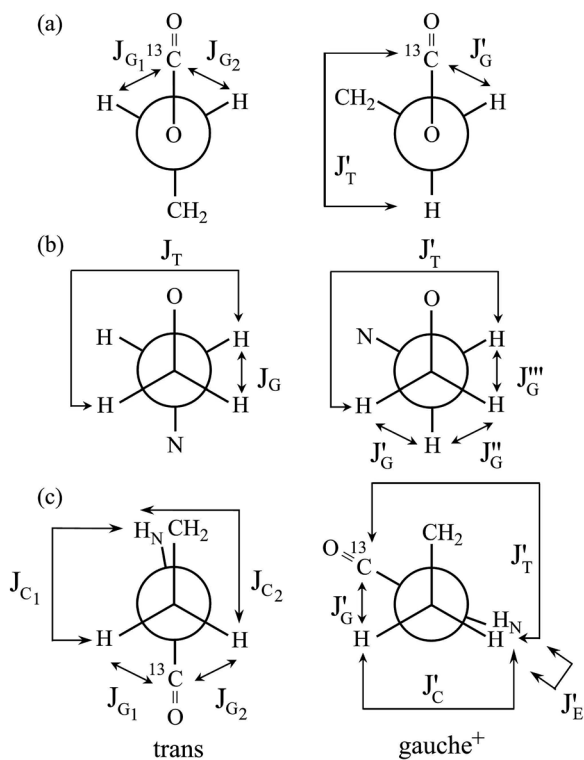


Fig. 4 Newman projections for rotamers around the (a) b: O–C, (b) c: C–C, and (c) d: C–N bonds of EA-1, illustrating the coefficients (J_C , J'_E , J_G , and J_T) of the equations given in Table 2. For the bond symbols, see Fig. 2a.

carried out for each conformer of EA-1. For the optimized geometry, single point energy calculations at the M06-2X/6-311+G(2d,p) and MP2/6-311+G(2d,p) levels with the IEFPCM and CPCM solvation models were carried out. The geometrical optimization for EA-1 yielded only six conformers, whose Gibbs free energies are given for each MO theory, solvent, and solvation model (Table 4). The free energies of the six conformers depend largely on the solvent but only slightly on the solvation model. As for the ΔG_k values, the M06-2X and MP2 calculations are consistent with each other and contradictory to the B3LYP results.

From the free energies, the bond conformations were calculated as listed in Tables 5 and S4 (ESI†). The M06-2X and MP2 results agree well with the NMR data (Tables 3 and S3 (ESI†)), regardless of the solvation model used. Therefore, we have decided to adopt the MP2/6-311+G(2d,p)/B3LYP/6-311+G(2d,p) combination and the IEFPCM model for all the model compounds of PEA and PU because this computation has yielded satisfactory results consistent with NMR experiments for most FUs that we have considered (Table S1 (ESI†)). The reliable MP2 calculations indicated that the first and second most stable conformers of EA-1 are $g^+g^+g^+$ and tg^+g^+ , respectively. As all the model compounds treated here are symmetrical, *i.e.* achiral, for example, $g^+g^+g^+$ and its mirror image, $g^-g^-g^-$, have the same free energy and hence exist at an identical probability. Therefore, the results and discussion herein are represented by either conformer.

In a previous study,²⁸ we carried out conformational analysis of *N*-(2-methoxyethyl)methylamine ($\text{CH}_3\text{OCH}_2\text{CH}_2\text{NHCH}_3$). Its most stable conformer is the *tgt* form with an intramolecular N–H...O attraction. It seems that the $g^+g^+g^+$ conformer of EA-1 has two C–H...O=C attractions as well as an N–H...O one (Fig. 2a); the outer carbonyl groups influence the conformational preference of the inner portion of EA-1.

The model compound, EA-1, represents bonds a–e of PEA (Fig. 1a). As a model for bonds f–h of PEA, we have employed EA-2 (Fig. 2b). As a result of the geometrical optimization for all conceivable conformers of EA-2, only five conformations remained. Around bonds f and h, unusual eclipsed forms as well as the three staggered states (*t*, g^+ , and g^-) were suggested to exist: *cis*_{N–C} in bond f; and *cis*_{H–O} and *cis*_{C–O} in bond h (Fig. 5). The ΔG_k values of the five conformers and the bond conformations of bonds f–h are given in the upper and lower parts of Table 6, respectively. In benzene and chloroform, the most stable conformer was indicated to be g^+g^-t , whereas in polar solvents such as DMSO and water, the *ttt* conformation would be more likely to exist. This probably depends on whether an intramolecular N–H...O=C attraction is formed (in g^+g^-t , Fig. 2b) or not (in *ttt*). The polar solvents may disturb the intramolecular interaction.

Conformational analysis of U-1 and U-2

NMR spectra of U-1 and U-2. In Fig. 6, the ¹H NMR spectra observed for U-1 and U-2 are shown, together with the gNMR



Table 3 *Trans* fractions of the O–C, C–C, and C–N bonds of EA-1, determined from NMR experiments^a

Solvent	Temp. (°C)	Bond ^b							
		b: O–C		c: C–C		d: C–N			
		J_{CH}^O		J_{HH} and J'_{HH}		J_{CH}^N		J_{HH}^N	
		Set A ^c	Set B ^d	Set C ^e	Set D ^f	Set E ^g	Set F ^h	Set G ⁱ	Set H ^j
C ₆ D ₆	15	0.40	0.42	0.14	0.22	0.10	0.16	0.10	0.11
	25	0.40	0.42	0.14	0.22	0.10	0.16	0.10	0.11
	35	0.40	0.42	0.14	0.23	0.10	0.16	0.10	0.11
	45	0.40	0.41	0.15	0.23	0.10	0.16	0.10	0.11
	55	0.39	0.41	0.15	0.23	0.10	0.16	0.10	0.11
CDCl ₃	25	0.45	0.46	0.07	0.15	0.16	0.22	0.15	0.14
(CD ₃) ₂ SO	25	0.44	0.46	0.17	0.25	0.01	0.08	0.17	0.15
D ₂ O	25	0.46	0.47	0.07	0.14	0.00	0.07	—	—

^a For more detailed data, see Table S3 (ESI†). ^b See Fig. 2a. ^c With the coupling constants calculated from a Karplus equation proposed by Tvaroška and Gajdoš:²⁷ $J_{G_1} + J_{G_2} = 2.42$ Hz and $J'_T + J'_G = 9.53$ Hz. ^d With the coupling constants calculated from MO calculations at the B3LYP/6-311++G(3df,3pd) level: $J_{G_1} + J_{G_2} = 2.26$ Hz and $J'_T + J'_G = 9.87$ Hz. ^e With the coupling constants of 2-methylmorpholine: e.g. $J_T = 11.62$, $J_G = 2.35$, $J'_T = 11.62$, $J'_G = 3.33$, $J''_G = 1.21$, and $J'''_G = 2.51$ Hz for the chloroform solution. For details, see ref. 28. ^f With the coupling constants calculated from MO calculations at the B3LYP/6-311++G(3df,3pd) level: $J_T = 11.12$, $J_G = 4.82$, $J'_T = 10.07$, $J'_G = 2.25$, $J''_G = 1.96$, and $J'''_G = 1.82$ Hz. ^g With the coupling constants calculated from a Karplus equation proposed by Kao and Barfield:²⁹ $J_{G_1} + J_{G_2} = 1.96$ Hz and $J'_T + J'_G = 7.56$ Hz. ^h With the coupling constants calculated from MO calculations at the B3LYP/6-311++G(3df,3pd) level: $J_{G_1} + J_{G_2} = 1.57$ Hz and $J'_T + J'_G = 8.03$ Hz. ⁱ With the coupling constants calculated from a Karplus equation proposed by Ludvigsen *et al.*:³⁰ $J_{C_1} + J_{C_2} = 8.13$ Hz and $J_E + J'_C = 12.09$ Hz. ^j With the coupling constants calculated from MO calculations at the B3LYP/6-311++G(3df,3pd) level: $J_{C_1} + J_{C_2} = 5.16$ Hz and $J'_E + J'_C = 12.48$ Hz.

simulations. Of the NMR data that the simulations yielded, only vicinal coupling constants used in the conformational analysis are listed in Table S5 (ESI†). As the model compounds are hardly soluble in non-polar solvents, the data are limited in number. In addition, the satellite of the methylene protons of U-

2 appears to be broadened. This may be due to some specific interaction of the C(=O)-NH group with the polar NMR solvents. For U-2, therefore, only the vicinal coupling constants at 15 and 25 °C are presented in Table S5 (ESI†). The equations for the analysis are given in Table 2, the coefficients ($J_C, J_E, J_G,$

Table 4 Conformer free energies (ΔG_k) of EA-1, evaluated from different MO calculations^a

MO theory ^c	Solvent	Solvation	ΔG_k^b (kcal mol ⁻¹)					
			Existing conformer (in bonds b-d) ^b					
			ttg^+	tg^+g^+	g^+tg^-	$g^+g^+g^+$	$g^+g^+g^-$	g^+g^-
M06-2X	Benzene	IEFPCM	0.00	-0.76	0.11	-0.96	0.53	-0.57
MP2		IEFPCM	0.00	-0.92	0.14	-1.04	0.30	-0.59
B3LYP	Chloroform	IEFPCM	0.00	-0.70	0.77	-0.25	2.22	0.39
		CPCM	0.00	-0.65	0.85	-0.22	2.19	0.56
M06-2X		IEFPCM	0.00	-0.60	0.34	-0.77	0.63	-0.16
		CPCM	0.00	-0.54	0.43	-0.73	0.58	0.02
MP2		IEFPCM	0.00	-0.76	0.35	-0.86	0.40	-0.21
		CPCM	0.00	-0.70	0.44	-0.83	0.36	-0.04
B3LYP	DMSO	IEFPCM	0.00	-0.52	1.07	-0.03	2.29	0.94
		CPCM	0.00	-0.51	1.09	-0.02	2.29	0.96
M06-2X		IEFPCM	0.00	-0.41	0.65	-0.54	0.66	0.39
		CPCM	0.00	-0.40	0.66	-0.54	0.65	0.42
MP2		IEFPCM	0.00	-0.56	0.64	-0.64	0.43	0.30
		CPCM	0.00	-0.55	0.66	-0.64	0.43	0.33
M06-2X	Water	IEFPCM	0.00	-0.39	0.66	-0.53	0.66	0.42
MP2		IEFPCM	0.00	-0.55	0.66	-0.63	0.43	0.33

^a Relative to the free energy of the ttg^+ conformer. Abbreviations: IEFPCM, the polarizable continuum model using the integral equation formalism variant; CPCM, the conductor-like polarizable continuum model; DMSO, dimethyl sulfoxide; B3LYP, Becke's three parameter hybrid functionals with the correlation functional of Lee, Yang, and Parr;²⁰ MP2, a Hartree-Fock calculation followed by the Moller-Plesset correlation energy correction truncated at the second-order;²¹ and M06-2X, the hybrid functional of Zhao and Truhlar.²² ^b The geometrical optimization was carried out at the B3LYP/6-311+G(2d,p) level. Of 27 (=3³) possible conformers, only 6 conformations remained. For the bond symbols, see Fig. 2a. ^c The basis set of 6-311+G(2d,p) was used for all the MO theories.



Table 5 *Trans* fractions of the O–C, C–C, and C–N bonds of EA-1, evaluated from MO calculations^a

Solvent	Solvation model	Temp. (°C)	Bond ^b								
			b: O–C			c: C–C			d: C–N		
			B3LYP ^c	M06-2X ^d	MP2 ^e	B3LYP ^c	M06-2X ^d	MP2 ^e	B3LYP ^c	M06-2X ^d	MP2 ^e
Benzene	IEFPCM	15		0.34	0.37		0.13	0.11		0.19	0.17
		25		0.34	0.37		0.14	0.11		0.20	0.17
		35		0.34	0.37		0.14	0.12		0.20	0.17
		45		0.34	0.37		0.15	0.12		0.20	0.17
		55		0.34	0.37		0.15	0.13		0.20	0.18
Chloroform	IEFPCM	25	0.65	0.39	0.41	0.19	0.16	0.14	0.08	0.14	0.12
	CPCM	25	0.66	0.40	0.41	0.20	0.17	0.14	0.06	0.11	0.10
DMSO	IEFPCM	25	0.70	0.45	0.45	0.24	0.20	0.17	0.04	0.08	0.08
	CPCM	25	0.70	0.45	0.45	0.24	0.20	0.17	0.04	0.08	0.07
Water	IEFPCM	25		0.45	0.45		0.20	0.17		0.08	0.07

^a For more detailed data, see Table S4 (ESI†). ^b See Fig. 2a. ^c At the B3LYP/6-311+G(2d,p)//B3LYP/6-311+G(2d,p) level. ^d At the M06-2X/6-311+G(2d,p)//B3LYP/6-311+G(2d,p) level. ^e At the MP2/6-311+G(2d,p)//B3LYP/6-311+G(2d,p) level.

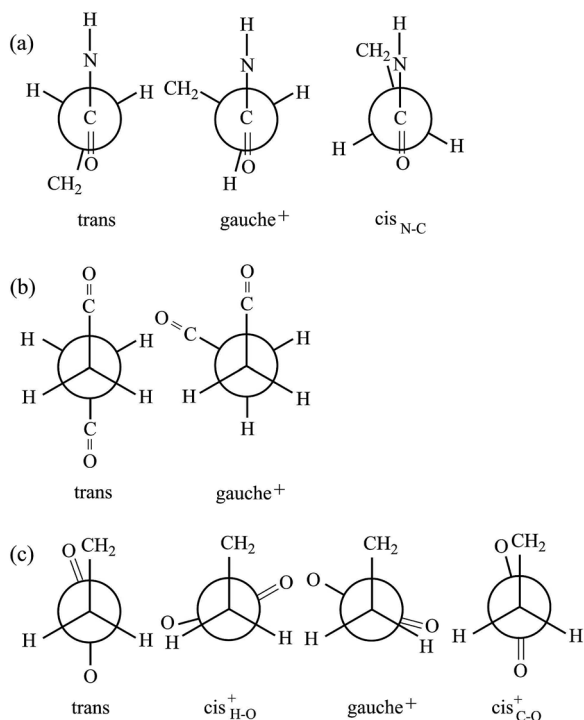


Fig. 5 Newman projections for rotamers around the (a) f: C(=O)–C, (b) g: C–C, and (c) h: C–C(=O) bonds of EA-2. For the bond symbols, see Fig. 2b.

Table 6 Conformer free energies (ΔG_k) and bond conformations (p_ξ , ξ : rotamer) of EA-2 at 25 °C, evaluated from MO calculations

Solvent	ΔG_k^a (kcal mol ^{−1})				
	<i>ttt</i>	<i>ttg</i> ⁺	<i>g</i> ⁺ <i>g</i> [−] <i>t</i>	<i>tg</i> ⁺ (<i>cis</i> _{C–O})	(<i>cis</i> _{N–C}) <i>g</i> ⁺ (<i>cis</i> _{H–O})
Benzene	0.00	0.57	−0.45	0.42	2.03
Chloroform	0.00	0.47	−0.18	0.24	1.95
DMSO	0.00	0.37	0.08	0.04	1.85
Water	0.00	0.37	0.09	0.03	1.85

Solvent	Bond ^c	p_ξ^b				
		<i>trans</i>	<i>gauche</i>	<i>cis</i> _{N–O}	<i>cis</i> _{C–O}	<i>cis</i> _{H–O}
Benzene	f	0.39	0.60	0.01		
	g	0.25	0.75			
	h	0.74	0.11		0.14	0.01
Chloroform	f	0.54	0.45	0.01		
	g	0.32	0.68			
	h	0.62	0.15		0.22	0.01
DMSO	f	0.68	0.30	0.02		
	g	0.36	0.64			
	h	0.48	0.19		0.32	0.01
Water	f	0.68	0.30	0.02		
	g	0.36	0.64			
	h	0.47	0.19		0.33	0.01

^a At the MP2/6-311+G(2d,p)//B3LYP/6-311+G(2d,p) level with the IEFPCM solvation model. The five conformations were obtained from the geometrical optimization. ^b For the rotamers (ξ) of each bond, see Fig. 5. ^c See Fig. 2b.

and J_T) are defined in Fig. 7, and their numerical values are written in the footnotes of Table 7. The *trans* fractions thus determined are listed in Tables 7 and S6 (ESI†). Despite the limited data, it is obviously indicated that the N–C and C–C bonds of U-1 and the C–C bond of U-2 prefer the *gauche* states.

MO calculations for U-1 and U-2. The conformer free energies of U-1 and U-2 were also calculated at the MP2/6-311+G(2d,p)//B3LYP/6-311+G(2d,p) level with the IEFPCM model, and the values are shown in Table 8. In contrast with EA-2, both U-1 and U-2 were optimized only in the normal

staggered forms (*t*, *g*⁺, and *g*[−]). The most stable conformers of U-1 and U-2 are indicated to be *g*⁺*g*⁺*g*⁺ and *tg*⁺*t*, respectively (Fig. 2c and d). In our previous study on poly(ethylene imine),³¹ its NH–CH₂–CH₂–NH bond sequence was found to be most stabilized in *meso* *tg*⁺*t* that exhibits an intramolecular N–H⋯N attraction. In U-1, the *g*⁺*g*⁺*g*⁺ stabilization may be partly due to a C=O⋯H–C–H attraction; the outer ester groups appear to influence the inner NH–CH₂–CH₂–NH portion. On the other hand, the *tg*⁺*t*



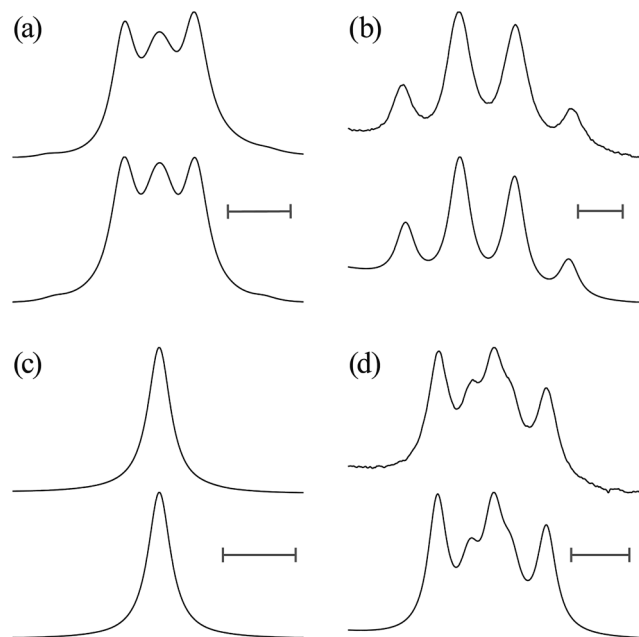


Fig. 6 ^1H NMR spectra observed from (a and b) U-1 dissolved in $(\text{CD}_3)_2\text{SO}$ at 25 $^\circ\text{C}$ and (c and d) U-2 in CD_3OD at 15 $^\circ\text{C}$ (above), together with the results of spectrum simulations (below): (a and c) CH_2 ; and (b and d) a satellite of CH_2 . All the scale bars represent 5 Hz.

stabilization in the $\text{C}(=\text{O})\text{O}-\text{CH}_2-\text{CH}_2-\text{OC}(=\text{O})$ bonds of U-2 has also been found for poly(ethylene terephthalate) ($\Delta G_k = -1.2 \text{ kcal mol}^{-1}$)³³ and poly(ethylene succinate) ($-1.3 \text{ kcal mol}^{-1}$).³⁴ The outer NH groups of U-2 seem not to disturb the tg^+t preference inherent to the diester bond sequence.

The *trans* fractions of U-1 and U-2, calculated from the ΔG_k values, are compared in Tables 7 and S6 (ESI[†]) with the NMR data. The agreement between theory and experiment is good enough for us to conclude that the MP2/6-311+G(2d,p)//B3LYP/6-311+G(2d,p) calculation with the IEFPCM model should also be applicable to the urethane models.

Configurational properties of PEA and PU

As shown above, the MO calculations at the MP2//B3LYP level are suitable for the ester–amide and urethane models. Therefore, the configurational properties of PEA and PU were calculated with the MO energies and geometrical parameters. What we call the refined RIS method¹⁵ has advantages compared with the conventional RIS scheme:^{3,4} all geometrical parameters are dependent on (variable with) conformations of the current and neighboring bonds. The ΔG_k energies of the model compounds, not being separated into first-, second-, and higher-order interaction energies, were directly introduced into the refined RIS computations.³⁵ The conformational energies over bonds a–e and f–h of PEA were taken from the ΔG_k values of EA-1 and EA-2 in DMSO, respectively (Fig. 1). In a similar manner, the conformational energies of PU were determined from the ΔG_k values of U-1 and U-2 in DMSO.

The results of the refined RIS calculations are summarized in Table 9. The characteristic ratio ($\langle r^2 \rangle_0/nl^2$) of PEA is 5.39, almost

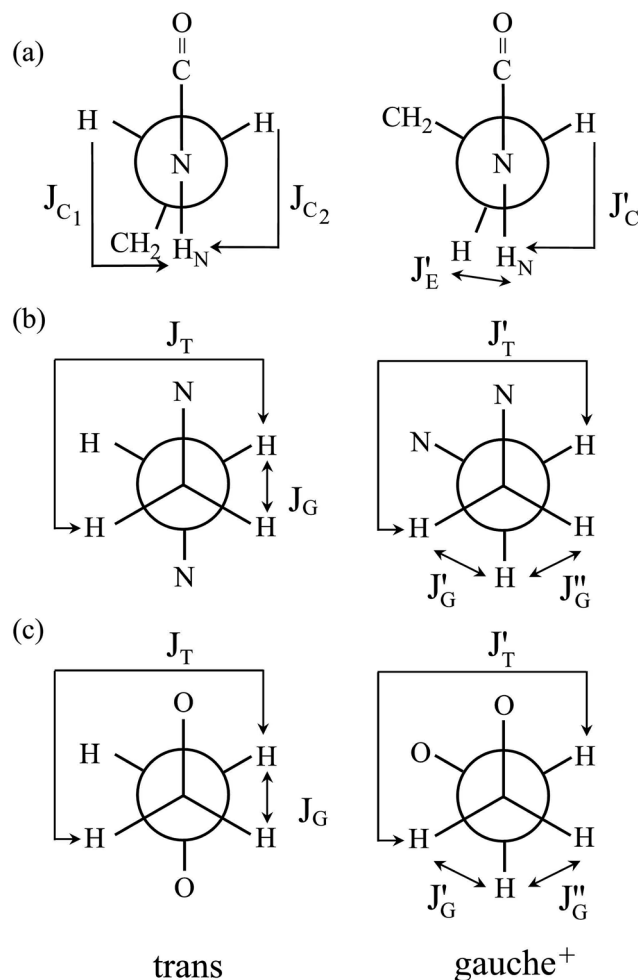


Fig. 7 Newman projections for rotamers around the (a) c and e: N–C and (b) d: C–C bonds of U-1 and the (c) i: C–C bond of U-2, illustrating the coefficients (J_C , J'_E , J_G , and J_T) of the equations given in Table 2. For the bond symbols, see Fig. 2c and d.

the same as that (5.40) of PU. However, the temperature coefficient ($d\ln\langle r^2 \rangle_0/dT \times 10^3$) of PEA is negative (-0.45), whereas that of PU is positive (0.69). The temperature coefficient at T_0 was calculated using the finite-difference method:

$$\frac{d\ln\langle r^2 \rangle_0}{dT}(T_0) \approx \frac{\ln[\langle r^2 \rangle_0(T_0 + \Delta T)/\langle r^2 \rangle_0(T_0 - \Delta T)]}{2\Delta T} \quad (1)$$

where T_0 and ΔT were set at 298.15 K (25 $^\circ\text{C}$) and 1.00 K, respectively.

The S_{conf} value (in units of calories per Kelvin per mole) represents the conformational freedom of the unperturbed polymeric chain (e.g. in the melt and the amorphous state). Because PEA and PU have different monomer sizes, it is preferable that the unit should be changed to $\text{cal K}^{-1} \text{g}^{-1}$. Then, PEA and PU are suggested to have close S_{conf} values of 5.8×10^{-2} and $5.6 \times 10^{-2} \text{ cal K}^{-1} \text{g}^{-1}$, respectively. The geometrical parameters averaged over all the possible conformations at 25 $^\circ\text{C}$ are also listed in Table 9.



Table 7 *Trans* fractions (p_t) of U-1 and U-2: comparison between NMR experiments and MO calculations^a

		U-1 ^b						U-2 ^b			
		c & e: N-C			d: C-C			h & j: O-C	i: C-C		
		NMR			NMR				NMR		
Medium	Temp. (°C)	Set A ^d	Set B ^e	MO ^c	Set C ^f	Set D ^g	MO ^c	MO ^c	Set E ^h	Set F ⁱ	MO ^c
Gas	25			0.16			0.14	0.46			0.24
Methanol	25			0.19	0.19	0.24	0.15	0.64	0.08	0.15	0.13
DMSO	25	0.18	0.17	0.19	0.18	0.21	0.15	0.64	0.04	0.11	0.13
	35	0.21	0.18	0.19	0.18	0.22	0.15	0.63	— ^j	— ^j	0.13
	45	0.23	0.20	0.19	0.19	0.23	0.16	0.63	—	—	0.14
	55	0.23	0.20	0.19	0.20	0.24	0.16	0.62	—	—	0.14

^a For more detailed data, see Table S6 (ESI†). ^b For the bond symbols, see Fig. 2c and d. ^c At the MP2/6-311+G(2d,p)/B3LYP/6-311+G(2d,p) level with the IEFPCM solvation model. ^d With the coupling constants calculated from a Karplus equation proposed by Ludvigsen *et al.*:³⁰ $J_{C_1} + J_{C_2} = 7.98$ Hz and $J'_E + J'_C = 12.44$ Hz. ^e With the coupling constants calculated from MO calculations at the B3LYP/6-311+G(3df,3pd) level: $J_{C_1} + J_{C_2} = 4.77$ Hz and $J'_E + J'_C = 13.03$ Hz. ^f With the coupling constants of 2-methylpiperazine:³¹ $J_T = 11.92$, $J_G = 2.77$, $J'_T = 11.92$, $J'_G = 3.19$, and $J''_G = 1.92$ Hz for the methanol solution; $J_T = 11.49$, $J_G = 2.74$, $J'_T = 11.49$, $J'_G = 3.05$, and $J''_G = 2.14$ Hz for the DMSO solution. ^g With the coupling constants calculated from MO calculations at the B3LYP/6-311+G(3df,3pd) level: $J_T = 10.33$, $J_G = 4.51$, $J'_T = 10.38$, $J'_G = 2.62$, and $J''_G = 1.65$ Hz. ^h With the coupling constants optimized for ethylene oxides:³² $J_T = J'_T = 11.4$ Hz and $J_G = J'_G = J''_G = 2.3$ Hz. ⁱ With the coupling constants calculated from MO calculations at the B3LYP/6-311+G(3df,3pd) level: $J_T = 11.33$, $J_G = 4.96$, $J'_T = 10.73$, $J'_G = 1.39$, and $J''_G = 2.49$ Hz. ^j The satellite of U-2 was broadened at temperatures above 25 °C.

Table 8 Conformer free energies of U-1 and U-2 at 25 °C

		ΔG^a (kcal mol ⁻¹)					
		U-1 ^b			U-2 ^c		
Conformer ^d		Gas	MeOH	DMSO	Gas	MeOH	DMSO
<i>t</i>	<i>t</i>				0.00	0.00	0.00
<i>t</i>	<i>t</i>				−0.49	−0.33	−0.33
<i>t</i>	<i>g</i> ⁺				−1.02	−1.56	−1.57
<i>t</i>	<i>g</i> ⁺				−0.67	−0.99	−1.00
<i>t</i>	<i>g</i> ⁺	0.64	0.37	0.37			
<i>g</i> ⁺	<i>t</i>	2.78	1.24	1.20			
<i>g</i> ⁺	<i>t</i>	0.77	0.74	0.75	0.03	0.67	0.69
<i>g</i> ⁺	<i>g</i> ⁺	0.00	0.00	0.00	−0.75	−0.41	−0.40
<i>g</i> ⁺	<i>g</i> ⁺	2.05	1.27	1.26	−0.32	−0.16	−0.16
<i>g</i> ⁺	<i>g</i> ⁺	1.64	1.56	1.56			

^a At the MP2/6-311+G(2d,p)/B3LYP/6-311+G(2d,p) level with the IEFPCM solvation model. ^b Relative to the *g*⁺*g*⁺ conformer. ^c Relative to the *ttt* conformer. ^d The blank represents that the potential minimum was not found by the geometrical optimization.

Chain elasticity

If the ester amide or urethane unit is introduced into a polymeric network as a functional unit, it may show an elastic behavior. We can discuss the potentiality on the basis of the pioneering work of Flory *et al.*^{36–40}

The tension (f) of an elastomer is known to be expressed as:⁴¹

$$f = f_U + f_S \quad (2)$$

where f_U and f_S are the internal energy (U) and entropy (S) terms, respectively, and

$$f_U = \left(\frac{\partial U}{\partial L} \right)_{T,V} \quad (3)$$

and

$$f_S = -T \left(\frac{\partial S}{\partial L} \right)_{T,V} \quad (4)$$

with L , T , and V being the length, absolute temperature, and volume, respectively. The S_{conf} value corresponds to the intramolecular entropy difference between the random coil and completely extended form (e.g. the all-*trans* conformation), thus giving a measure of the maximum entropic elasticity that the polymeric chain can exhibit. As PEA and PU have almost the same S_{conf} values, the difference in chain elasticity between the two polymers depends chiefly on f_U .

The ratio f_U/f can be related to the temperature coefficient of the unperturbed chain dimension by:^{36–40}

$$\frac{f_U}{f} = -T \left[\frac{\partial \ln(f/T)}{\partial T} \right]_{L,V} = T \frac{d \ln \langle r^2 \rangle_0}{dT} \quad (5)$$

From eqn (5), the f_U/f ratios of PEA and PU at 25 °C were evaluated as shown in Table 9. PEA has a negative f_U/f of −0.13, while PU shows a positive f_U/f of 0.21. The sign of f_U/f is related to conformational changes during contraction of the polymer.

Polyethylene is known to exhibit negative f_U/f values of ca. −0.4.^{36,38,42} As the temperature increases ($\Delta T > 0$, where Δ means the change), the *trans* conformations partly change to more distorted *gauche* states of higher energy ($\Delta \langle r^2 \rangle_0 < 0$), therefore $d \ln \langle r^2 \rangle_0 / dT < 0$. When the polyethylene chain shrinks ($\Delta L < 0$), the internal energy increases ($\Delta U > 0$) owing to the *trans*-to-*gauche* conversion: ($\Delta U / \Delta L \approx (\partial U / \partial L) < 0$). The negative f_U/f vitiates the entropic elasticity, because f_S must always be positive.

Positive f_U/f values have often been found for elastomers, for example, natural rubber (0.12–0.18), polydimethylsiloxane (0.13–0.30), and *cis*-1,4-polybutadiene (0.10–0.17).⁴² The PU



Table 9 Configurational properties and averaged geometrical parameters of PEA and PU at 25 °C, evaluated from the refined RIS calculations with MO parameters^a

PEA		PU	
$\langle r^2 \rangle_0 / nl^2 (x \rightarrow \infty)$	5.39		5.40
$d \ln \langle r^2 \rangle_0 / dT \times 10^3 (K^{-1})$	-0.45		0.69
$S_{\text{conf}}^b (\text{cal K}^{-1} \text{mol}^{-1})$	8.28		9.73
f_U / f	-0.13		0.21

Geometry ^c	Bond	\bar{l}	$\bar{\theta}$	$\bar{\phi}_{g^\pm}$	$\bar{\phi}_{cis_{N-C}^\pm}$	$\bar{\phi}_{cis_{H-O}^\pm}$	$\bar{\phi}_{cis_{C-O}^\pm}$	\bar{l}	$\bar{\theta}$	$\bar{\phi}_{g^\pm}$
	a	1.353	116.4					1.365	110.4	
	b	1.447	109.8	± 93.0				1.355	122.1	
	c	1.520	113.0	± 116.3				1.454	113.3	± 93.7
	d	1.452	122.6	± 72.7				1.530	113.3	± 115.8
	e	1.366	116.2					1.454	122.1	± 93.7
	f	1.526	112.5	± 90.3	± 174.1			1.355	110.4	
	g	1.529	114.3	± 106.4				1.365	115.6	
	h	1.512	111.8	± 13.2		± 66.9	± 172.4	1.437	109.6	± 90.0
	i							1.509	109.6	± 111.7
	j							1.437	115.6	± 90.0

^a Calculated from the MO energies including the solvent effect of DMSO. ^b $S_{\text{conf}} = (R/x)[\ln Z + T(d(\ln Z))/(dT)]$ where R is the gas constant, x is the degree of polymerization, T is the absolute temperature, and Z is the partition function of the whole chain. ^c Symbols: \bar{l} , averaged bond length (in Å); $\bar{\theta}$, averaged bond angle (in deg.); and $\bar{\phi}_\xi$, average dihedral angle (in deg.) of the ξ conformation.

chain here also shows a positive value of 0.21. This is because PU strongly prefers distorted conformations: in bonds c-e, $g^+g^+g^+$; and in bonds h-j, tg^+t , tg^+g^+ , and $g^+g^+g^+$. As temperature increases, the distortion relaxes and the chain dimension increases: $d \ln \langle r^2 \rangle_0 / dT > 0$. The stretching (contraction) of the PU chain increases (decreases) the internal energy: $(\partial U / \partial L) > 0$. This nature of PU supports the entropic elasticity. On the other hand, PEA will behave like polyethylene because of the negative f_U / f . Therefore, the urethane unit is more likely to behave as an elastomer than the ester amide unit.

Although it is well established that rubber-like elasticity is entirely of intramolecular origin,⁴³ the as-synthesized PU may not necessarily act as an elastomer. This is because the chain must also satisfy the following requirements to exhibit rubber-like elasticity: the chain must have a high degree of flexibility and mobility and, in addition, join a network structure.⁴³ Probably, as needed, the polymer must become amorphous (by *e.g.* quenching), cross-linked, diluted with a plasticizer to lower the glass transition temperature, and/or foamed. Or, as mentioned in the Introduction, if the oligo-urethane is incorporated into a polymeric mosaic, it may behave as an elastomeric functional unit. The repeating units of PEA and PU include the same atomic groups, two ethylene, ester, and amide groups, but their arrangements in the backbone are different. The cardinal principle of polymer science and molecular biology, “higher-order structures, physical properties, and functions of a polymer originate from its primary structure”, should be emphasized. The difference between PEA and PU in atomic arrangement generates the differences in physical properties.

Biodegradability

Poly(ester amide)s have often been rendered biodegradable by introducing degradable components such as lactic acid, glycolic acid, and ϵ -caprolactone units into the ester part of the main chain.^{9,44–47} Then, the ester and amide moieties of poly(ester amide)s are considered to be different FUs. If the former and latter units are, respectively, a biodegradable ester and an α -amino acid, the decomposed products would be non-toxic to humans. Therefore, such poly(ester amide)s may be used for biomedical materials.

The biodegradability of polyurethanes has been investigated from a microbiological viewpoint.^{10,48,49} The findings may be summarized as follows: (1) polyester-type polyurethanes are more susceptible to microbial attack than polyether-type ones; (2) polyurethanes with long repeating units are more readily degradable than those with short monomers; (3) the amorphous regions are degraded prior to the crystallites; and (4) the biodegradation of polyester-type polyurethanes is mainly due to the hydrolysis of the ester bonds by lipases and their homologs. In addition, aliphatic polyurethanes with different numbers of carbon atoms in the backbone, being close to the PU considered here in terms of the primary structure, have been reported to be biodegradable.^{10,48,49}

Previous studies on biodegradable polymers^{48,50–52} have indicated that lipases are apt to degrade aliphatic polyesters with comparatively many methylene groups. Our conformational analysis of poly(ethylene succinate) and poly(butylene succinate)³⁴ has indicated that such polyesters will selectively adopt extended and planar structures and can enter narrow crevices of lipases. The active site for hydrolyzing the ester bond is located inside the crevice. Therefore, the lipases may hardly



degrade the PEA and PU chain because they strongly prefer the distorted conformations. It is well-known that the $\text{CH}_2\text{-CH}_2$ bond adjacent to the O-C bond in $\text{O-(CH}_2)_z\text{-O}$ sequences of esters and ethers prefers the *gauche* form,⁵³ and that the *gauche* stability of the bond generally decreases with increasing number of methylene units between the two oxygen atoms.^{13,14,33,34,54–56} If the ester and amide groups of the poly(ester amide)s $[\text{-C(=O)O(CH}_2)_y\text{NHC(=O)(CH}_2)_z\text{-}]_x$ or urethane groups of the polyurethanes $[\text{-C(=O)NH(CH}_2)_y\text{NHC(=O)O(CH}_2)_z\text{-}]_x$ are separated by at least four methylene units ($z \geq 4$), these FUs would more readily form planar structures and possibly approach the active sites of lipases. Such poly(ester amide)s and polyurethanes are expected to exhibit some level of biodegradability. This prediction is consistent with the above-mentioned microbiological findings.

Summary

To elucidate the effects of ester and amide groups on the conformational characteristics and physical properties of polymers, this study has dealt with a simple poly(ester amide) and polyurethane, PEA and PU. Each repeating unit was divided into two moieties, which have been represented by model compounds: EA-1 and EA-2 for PEA; and U-1 and U-2 for PU. These compounds were prepared and subjected to NMR measurements to determine the bond conformations. To examine the applicability of MO theories to the models, B3LYP, M06-2X, and MP2 calculations with the IEFPCM and CPCM solvation models were applied to EA-1 and compared with the NMR experiments. The M06-2X and MP2 energies with the B3LYP geometry satisfactorily reproduced the NMR data, and the IEFPCM and CPCM models made no essential differences. Therefore, MP2/6-311+G(2d,p)//B3LYP/6-311+G(2d,p) calculations with the IEFPCM model were employed to calculate the conformer free energies of all the four model compounds and suggested that the models form intramolecular $\text{N-H}\cdots\text{O}$ and $\text{C-H}\cdots\text{O}=\text{C}$ attractions and strongly prefer distorted structures including *gauche* bonds.

The refined RIS calculations for the PEA and PU chains at 25 °C yield essentially the same characteristic ratios of *ca.* 5.4, whereas the temperature coefficients ($\text{dln}\langle r^2 \rangle_0/\text{dT} \times 10^3$) of PEA and PU were negative (−0.45) and positive (0.69), respectively. Therefore, in PEA, the energy elasticity works against the entropic elasticity, whereas in PU, the former works with the latter. The results suggest the possibility that the PU chain would act as an elastomer under suitable conditions. In addition, the biodegradability of poly(ester amide)s and polyurethanes has been discussed in terms of their primary structures, and chemical modifications to render these polymers biodegradable have also been proposed.

In conclusion, PEA and PU, if chemically modified as needed, are expected not only to play the role of junctures of polymer networks and architectures but also to become more functional. Computational science will be increasingly helpful for molecular and materials design.^{57–59}

Acknowledgements

This study was partially supported by the Grants-in-Aid for Scientific Research (C) (22550190 and 16K05906) from the Japan Society for the Promotion of Science.

References

- 1 E. Wagner and J. Kloeckner, *Adv. Polym. Sci.*, 2006, **192**, 135–173.
- 2 Y. Yuan, C.-J. Zhang and B. Liu, *Angew. Chem., Int. Ed.*, 2015, **54**, 11419–11423.
- 3 P. J. Flory, *Statistical Mechanics of Chain Molecules*, Wiley & Sons, New York, USA, 1969.
- 4 W. L. Mattice and U. W. Suter, *Conformational Theory of Large Molecules: The Rotational Isomeric State Model in Macromolecular Systems*, Wiley-Interscience, New York, USA, 1994.
- 5 J.-M. Lehn, *Supramolecular Chemistry: Concepts and Perspectives*, VCH, Weinheim, Germany, 1995.
- 6 I. M. Atkinson and L. F. Lindoy, *Self Assembly in Supramolecular Systems*, Royal Society of Chemistry, London, UK, 2000.
- 7 A. Greenberg, C. M. Breneman and J. F. Liebman, *The Amide Linkage: Selected Structural Aspects in Chemistry, Biochemistry, and Materials Science*, Wiley-Interscience, New York, USA, 2000.
- 8 M. Nagasawa, T. Ishii, D. Abe and Y. Sasanuma, *RSC Adv.*, 2015, **5**, 96611–96622.
- 9 A. C. Fonseca, M. H. Gil and P. N. Simões, *Prog. Polym. Sci.*, 2014, **39**, 1291–1311.
- 10 G. T. Howard, in *Microbial Degradation of Xenobiotics, Environmental Science and Engineering*, ed. S. N. Singh, Springer, Berlin, Germany, 2012, ch. 14.
- 11 T. Fey, M. Hölscher, H. Keul and H. Höcker, *Polym. Int.*, 2003, **52**, 1625–1632.
- 12 E. L. Wittbecker and M. Katz, *J. Polym. Sci., Part A: Polym. Chem.*, 1959, **40**, 367–375.
- 13 Y. Sasanuma and N. Suzuki, *Macromolecules*, 2009, **42**, 7203–7212.
- 14 Y. Sasanuma, Y. Wagai, N. Suzuki and D. Abe, *Polymer*, 2013, **54**, 3904–3913.
- 15 Y. Sasanuma, S. Asai and R. Kumagai, *Macromolecules*, 2007, **40**, 3488–3497.
- 16 D.-L. Sun, S.-J. Xie, J.-R. Deng, C.-J. Huang, E. Ruckenstein and Z.-S. Chao, *Green Chem.*, 2010, **12**, 483–490.
- 17 V. V. Mikheev, *Russ. J. Org. Chem.*, 1989, **25**, 2028–2029.
- 18 P. H. Budzelaar, *gNMR, version 5.0*, IvorySoft & Adept Scientific plc, Letchworth, U.K., 2004.
- 19 M. J. Frisch, G. W. Trucks, H. B. Schlegel, G. E. Scuseria, M. A. Robb, J. R. Cheeseman, G. Scalmani, V. Barone, B. Mennucci, G. A. Petersson, H. Nakatsuji, M. Caricato, X. Li, H. P. Hratchian, A. F. Izmaylov, J. Bloino, G. Zheng, J. L. Sonnenberg, M. Hada, M. Ehara, K. Toyota, R. Fukuda, J. Hasegawa, M. Ishida, T. Nakajima, Y. Honda, O. Kitao, H. Nakai, T. Vreven, J. A. Montgomery Jr, J. E. Peralta, F. Ogliaro, M. Bearpark, J. J. Heyd,



- E. Brothers, K. N. Kudin, V. N. Staroverov, R. Kobayashi, J. Normand, K. Raghavachari, A. Rendell, J. C. Burant, S. S. Iyengar, J. Tomasi, M. Cossi, N. Rega, J. M. Millam, M. Klene, J. E. Knox, J. B. Cross, V. Bakken, C. Adamo, J. Jaramillo, R. Gomperts, R. E. Stratmann, O. Yazyev, A. J. Austin, R. Cammi, C. Pomelli, J. W. Ochterski, R. L. Martin, K. Morokuma, V. G. Zakrzewski, G. A. Voth, P. Salvador, J. J. Dannenberg, S. Dapprich, A. D. Daniels, O. Farkas, J. B. Foresman, J. V. Ortiz, J. Cioslowski and D. J. Fox, *Gaussian 09 Revision B01*, Gaussian Inc., Wallingford CT, USA, 2009.
- 20 A. D. Becke, *J. Chem. Phys.*, 1993, **98**, 5648–5652.
- 21 C. Moller and M. S. Plesset, *Phys. Rev.*, 1934, **46**, 618–622.
- 22 Y. Zhao and D. G. Truhlar, *Acc. Chem. Res.*, 2008, **41**, 157–167.
- 23 E. Cancès, B. Mennucci and J. Tomasi, *J. Chem. Phys.*, 1997, **107**, 3032–3041.
- 24 M. Cossi, N. Rega, G. Scalmani and V. Barone, *J. Comput. Chem.*, 2003, **24**, 669–681.
- 25 T. Helgaker, M. Watson and N. C. Handy, *J. Chem. Phys.*, 2000, **113**, 9402–9409.
- 26 IUPAC, *Compendium of Chemical Terminology*, ed. A. D. McNaught and A. Wilkinson, Blackwell Scientific Publications, Oxford, UK, 2nd edn (the “Gold Book”), 1997.
- 27 I. Tvaroška and J. Gajdoš, *Carbohydr. Res.*, 1995, **271**, 151–162.
- 28 Y. Sasanuma, R. Kumagai and K. Nakata, *Macromolecules*, 2006, **39**, 6752–6764.
- 29 L.-F. Kao and M. Barfield, *J. Am. Chem. Soc.*, 1985, **107**, 2323–2330.
- 30 S. Ludvigsen, K. V. Andersen and F. M. Poulsen, *J. Mol. Biol.*, 1991, **217**, 731–736.
- 31 Y. Sasanuma, S. Hattori, S. Imazu, S. Ikeda, T. Kaizuka, T. Iijima, M. Sawanobori, M. A. Azam, R. V. Law and J. H. G. Steinke, *Macromolecules*, 2004, **37**, 9169–9183.
- 32 K. Tasaki and A. Abe, *Polymer J.*, 1985, **17**, 641–655.
- 33 Y. Sasanuma, *Macromolecules*, 2009, **42**, 2854–2862.
- 34 Y. Sasanuma, Y. Nonaka and Y. Yamaguchi, *Polymer*, 2015, **56**, 327–339.
- 35 Y. Sasanuma, Y. Ogawa and M. Matsumoto, *Phys. Chem. Chem. Phys.*, 2010, **12**, 14619–14628.
- 36 P. J. Flory, C. A. J. Hoeve and A. Ciferri, *J. Polym. Sci.*, 1959, **34**, 337–347.
- 37 P. J. Flory, A. Ciferri and C. A. J. Hoeve, *J. Polymer Sci.*, 1960, **45**, 235–236.
- 38 A. Ciferri, C. A. J. Hoeve and P. J. Flory, *J. Am. Chem. Soc.*, 1961, **83**, 1015–1022.
- 39 J. E. Mark, *J. Chem. Educ.*, 1981, **58**, 898–903.
- 40 J. E. Mark, *J. Chem. Educ.*, 2002, **79**, 1437–1443.
- 41 P. J. Flory, *Principles of Polymer Chemistry*, Cornell University Press, Ithaca, NY, USA, 1953.
- 42 J. E. Mark, *Macromol. Rev.*, 1976, **11**, 135–159.
- 43 J. E. Mark, in *Physical Properties of Polymers*, Cambridge University Press, Cambridge, UK, 2003, pp. 3–71.
- 44 J. Tuominen and J. V. Seppälä, *Macromolecules*, 2000, **33**, 3530–3535.
- 45 M. Vera, M. Admetlla, A. Rodríguez-Galán and J. Puiggali, *Polym. Degrad. Stab.*, 2005, **89**, 21–32.
- 46 E. Botines, M. T. Casas and J. Puiggali, *J. Polym. Sci., Part B: Polym. Phys.*, 2007, **45**, 815–825.
- 47 J. E. Báez, D. Ramírez, J. L. Valentín and Á. Marcos-Fernández, *Macromolecules*, 2012, **45**, 6966–6980.
- 48 Y. Tokiwa, B. P. Calabia, C. U. Ugwu and S. Aiba, *Int. J. Mol. Sci.*, 2009, **10**, 3722–3742.
- 49 G. T. Howard, *Int. Biodeterior. Biodegrad.*, 2002, **49**, 245–252.
- 50 H. Pranamuda, A. Tsuchii and Y. Tokiwa, *Macromol. Biosci.*, 2001, **1**, 25–29.
- 51 Y. Tokiwa and A. Jarerat, *Macromol. Symp.*, 2003, **201**, 283–290.
- 52 Y. Tokiwa and B. P. Calabia, *Biotechnol. Lett.*, 2004, **26**, 1181–1189.
- 53 Y. Sasanuma, in *Annual Reports on NMR Spectroscopy*, ed. G. A. Webb, Academic Press, Elsevier Science, Oxford, UK, 2003, vol. 49, ch. 5, pp. 213–280.
- 54 Y. Sasanuma, H. Ohta, I. Touma, H. Matoba, Y. Hayashi and A. Kaito, *Macromolecules*, 2002, **35**, 3748–3761.
- 55 Y. Sasanuma and A. Watanabe, *Macromolecules*, 2006, **39**, 1646–1656.
- 56 R. V. Law and Y. Sasanuma, *Macromolecules*, 1998, **31**, 2335–2342.
- 57 *Computer Simulation of Polymers*, ed. R. J. Roe, Prentice Hall, Upper Saddle River, New Jersey, USA, 1991.
- 58 *Computer Simulation of Polymeric Materials: Application of the OCTA System*, ed. Japan Association for Chemical Innovation, Springer, Singapore, 2016.
- 59 T. K. Mukhopadhyay and A. Datta, *J. Phys. Chem. C*, 2017, **121**, 10210–10223.

