



Cite this: RSC Adv., 2017, 7, 35666

Crystal growth and characterization of a mixed laser crystal: Nd-doped $\text{Gd}_{0.89}\text{La}_{0.1}\text{NbO}_4$

Shoujun Ding,^{ab} Qingli Zhang,^{ID *a} Wenpeng Liu,^a Jianqiao Luo,^a Fang Peng,^a Xiaofei Wang,^a Guihua Sun^a and Dunlu Sun^a

Herein, a Nd-doped GdLaNbO_4 (Nd:GLNO) single crystal has been grown successfully using the Czochralski method. The segregation coefficient of Nd^{3+} ions in the Nd:GLNO crystal is measured to be 0.71. The X-ray power diffraction patterns of the Nd:GLNO crystal confirm that the as-grown crystal has the same crystal structure as pure GdNbO_4 . The structural parameters of the as-grown crystal are obtained via an X-ray Rietveld refinement method. The absorption spectrum, fluorescence spectrum, and fluorescence decay curve of Nd:GLNO were obtained at room temperature. The small emission cross-section at 1065 nm and long fluorescence lifetime of Nd^{3+} ions (176.4 μs) indicate that the Nd:GLNO crystal is suitable for generating an ultrashort pulse laser. A maximum continuous wave (CW) laser output power of 957 mW is achieved at 1.06 μm , corresponding to an optical-to-optical conversion efficiency of 30.3% and slope efficiency of 34.2%. All the obtained results imply that this crystal is a novel mixed niobate-based laser host suitable for laser diode pumps.

Received 12th May 2017

Accepted 3rd July 2017

DOI: 10.1039/c7ra05380k

rsc.li/rsc-advances

1. Introduction

Solid state lasers are more perfect than free-electron lasers and gas lasers due to their significantly enhanced mobility and excellent performance.¹ Examples of these lasers include rod-, disk-, and slab-type lasers based on rare earth-doped crystals. Currently, diode-pumped solid-state lasers (DPSSLs) based on Nd-doped crystals have attracted significant attention because of their high efficiency, compactness, and high stability and have been widely applied in the fields of medical treatment, industrial processing, military, and optical communication.^{2–5} In this study, considerable efforts have been devoted towards exploring new single crystals with excellent performance and high quality for DPSSLs. For example, Nd-doped vanadate crystals (Nd:LnVO_4 , Ln = Y, Gd, and Lu), represented by Nd:YVO_4 , have been identified to be a series of excellent laser materials with high chemical stability, good laser properties, and high laser damage threshold and commercialized and broadly applied in low- and even moderate-power lasers.^{6–9} Recently, Nd-doped tantalate (Nd:LnTaO_4 , Ln = Y and Gd) and Nd-doped niobate (Nd:LnNbO_4 , Ln = Y and Gd) crystals have been proven to be novel laser materials with high laser efficiency and good thermal properties and suitable for growing large-sized single crystal via the Czochralski (Cz) method.^{10–12} However, rare earth-doped vanadate, niobate, and tantalate laser crystals always possess short fluorescence lifetimes and

narrow absorption bandwidths that limit their application in pulse lasers, especially in ultrashort pulse lasers.

Nowadays, ultrashort pulse laser technology is one of the research hotspots in the laser field, which has been broadly applied in laser micromachining, medical science, optical trapping, etc.^{13–16} For laser materials, the most important requirement to generate an ultrashort pulse laser is the presence of a broad fluorescence band.^{17,18} Generally, spectral bandwidth consists of homogeneous broadening and inhomogeneous broadening.¹⁹ The inhomogeneous broadening is mainly affected by the crystalline field surrounding the active ions.²⁰ Therefore, it has been theoretically and experimentally verified that a mixed crystal can effectively broaden the spectra and generate ultrashort pulse lasers. In 1996, the first mixed vanadate crystal ($\text{Nd:Gd}_{0.5}\text{La}_{0.5}\text{VO}_4$) with broad fluorescence spectrum was grown and investigated.²¹ Thereafter, many other Nd-doped mixed vanadate crystals, such as $\text{Nd:Y}_{1-x}\text{La}_x\text{VO}_4$, $\text{Nd:Gd}_x\text{Y}_{1-x}\text{VO}_4$, and $\text{Nd:Lu}_x\text{Gd}_{1-x}\text{VO}_4$, were also investigated.^{22–24} More recently, Nd-doped niobate (Nd:GdYNbO_4) and tantalate (Nd:GdYTaO_4) mixed laser crystals were successfully grown by the Cz method, with large inhomogeneous broadened spectra and good laser properties.^{25,26} Among these mixed laser crystals, the La^{3+} -doped crystal is most interesting because La is the first element in the lanthanide series and has the largest diameter. Therefore, La-doped mixed crystals may have a more disordered structure and broader fluorescence bandwidth. However, it is generally known that La-doped mixed crystals are very difficult to grow, and, to the best of our knowledge, there are no reports on La-doped niobate laser crystals.

^aAnhui Institute of Optics and Fine Mechanics, Chinese Academy of Sciences, Hefei 230031, PR China. E-mail: zql@aiofm.ac.cn; 18119616997@163.com

^bUniversity of Science and Technology of China, Hefei 230026, PR China



In this study, a Nd:GdLaNbO₄ mixed laser crystal was successfully grown *via* the Cz method for the first time. Its structure, crystalline quality, absorption spectrum, and fluorescence spectrum were systematically investigated. An efficient laser diode end-pumped CW laser operation at 1066 nm was also demonstrated.

2. Experiment details

2.1 Crystal growth

The compounds Nd₂O₃ (99.999%), Gd₂O₃ (99.999%), La₂O₃ (99.99%), and Nb₂O₅ (99.99%) were used as starting materials to grow the Nd:GdLaNbO₄ laser crystal *via* the conventional Cz technique using JGD-60 furnace (26th institute of CETC, China). The starting materials were accurately weighed according to the chemical formula Nd_{0.01}:Gd_{0.89}La_{0.1}NbO₄ (all components were dried before weighing). After being adequately mixed and pressed into tablets, the raw materials were loaded into an iridium crucible. To prevent iridium from oxidation, the crystal was grown in a sealed chamber under a nitrogen atmosphere. A *a*-orientated GdNbO₄ crystal bar was used as the seed crystal. The pulling rate and rotation speed were 0.5–1 mm h⁻¹ and 5–10 rpm, respectively. After the growth process ended, the crystal was cooled down to room temperature at a rate of 30–40 °C h⁻¹. Afterwards, a Nd:GLNO crystal boule was obtained, as shown in Fig. 1. As can be seen, the crystal is free from cracks and pink in color. Under a 532 nm laser, no obvious scattering was found for the as-grown crystal. Before cutting the as-grown crystal for experiments, it was oriented in three crystallographic orientations.

2.2 Characterization

The X-ray diffraction (XRD) patterns of the Nd:GLNO crystal powder were obtained *via* the Philips X'pert PRO X-ray diffractometer using Cu K α radiation in the 2 θ range of 10–90° at a step rate of 0.02° min⁻¹. The X-ray rocking curves were obtained by a high-resolution X'pert Pro MPD diffractometer with a hybrid K α_1 monochromator. X-ray fluorescence analysis (XRF-1800) was conducted to measure the elemental concentrations of Nd³⁺ in the as-grown crystal. The transmission spectrum of

Nd:GLNO in the wavelength range of 320–950 nm was obtained by a Perkin-Elmer lambda-950 spectrophotometer at a spectral interval of 1 nm. The fluorescence spectra with 0.5 nm spectral interval were obtained from 850 to 1400 nm by an Edinburgh fluorescence spectrometer (FLSP-920) with a Xe lamp as the excitation source. The fluorescence decay curve of Nd:GLNO was also obtained using an FLSP-920 spectrometer with a microsecond lamp as the excitation source.

2.3 Laser experiment

Diode pumped continuous wave laser experiment at 1.06 μ m was performed using an uncoated 2 mm × 2 mm × 5 mm Nd:GdLaNbO₄ crystal, in which the 2 mm × 2 mm faces were cut along the crystallographic *b*-orientation and carefully polished. The configuration for generating 1.06 μ m laser is shown in Fig. 2. A fiber-coupled CW laser diode with a maximum output power of 30 W (central wavelength at around 808 nm) was employed as the pumping source. The pump beam was focused on the Nd:GLNO crystal with a spot radius of about 0.22 mm through the focusing optics. M1 is a plane mirror, with an antireflection (AR) material coated at 808 nm on the pump side, a high reflection (HR) material coated at 1.06 μ m, and a high transmission (HT) material coated at 808 nm on the other face. M2 is the output couple, with a transmission of 5.4% at 1.06 μ m. During the laser experiment, the crystal was wrapped with an indium foil and mounted on a copper block cooled by deionized water. For cooling, deionized water was maintained at 20 °C. The output laser power was measured using an OPHIR 30A-BB-18 power meter.

3. Results and discussion

3.1 Crystal structure and quality

The XRD patterns of the as-grown Nd:GLNO crystal are shown in Fig. 3. As can be seen, the diffraction peaks in the XRD patterns of Nd:GLNO can be well indexed with those in ICSD#20408, suggesting that the as-grown crystal has the same crystal structure as the M-type GdNbO₄, with a space group of *I2/a*. The structural parameters of the Nd:GLNO crystal were obtained by fitting the XRD data using the Rietveld refinement method *via* the general structure analysis software (GSAS),²⁷ and the structural parameters of ICSD#20408 were used as the initial values. The Rietveld refinement results of the Nd:GLNO crystal are shown in Fig. 4. The residual R_p and R_{wp} are 5.86% and 4.20%, respectively, which indicate that the refined results are reliable. The unit cell parameters of Nd:GLNO are fitted to be $a = 5.381$ \AA , $b = 11.112$ \AA , $c = 5.112$ \AA , $\alpha = \gamma = 90^\circ$, and $\beta = 94.551^\circ$. Compared with Nd:GNO ($a = 5.372$ \AA , $b = 11.093$ \AA , $c = 5.107$)

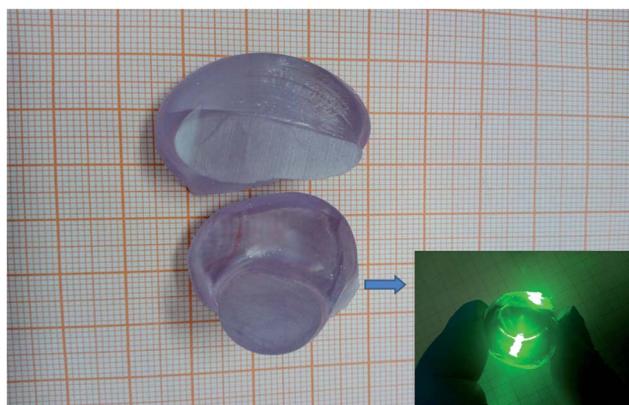


Fig. 1 Image of the as-grown Nd:GdLaNbO₄ laser crystal.

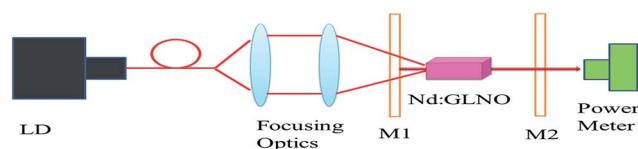


Fig. 2 Schematic of the experimental laser setup.



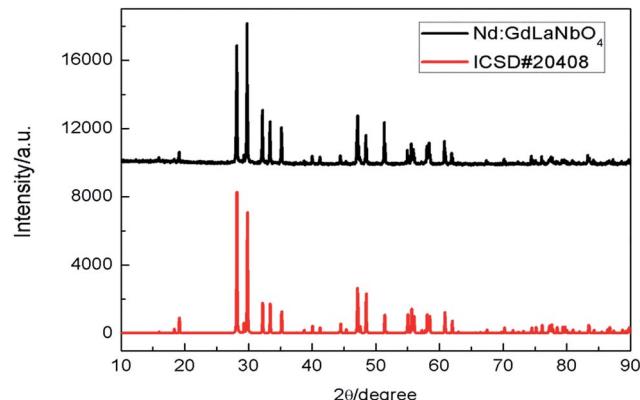


Fig. 3 XRD diffraction patterns of the Nd:GdLaNbO₄ crystal and GdNbO₄ obtained from ICSD#20408.

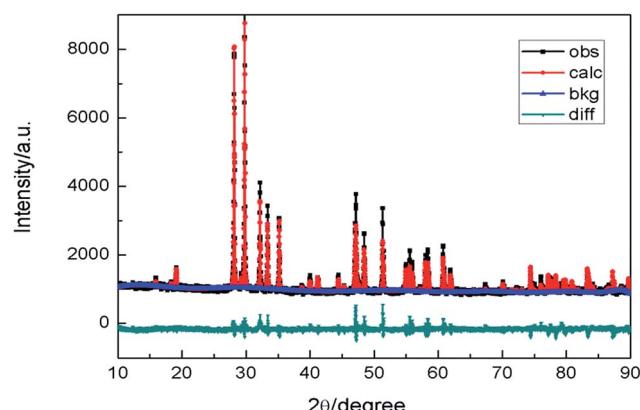


Fig. 4 The Rietveld refinement results of Nd:GLNO crystal (obs: the observed data; calc: calculated data; bkg: the background; diff: the difference between observed and calculated data).

and Nd:GYNO ($a = 5.349 \text{ \AA}$, $b = 11.049 \text{ \AA}$, $c = 5.097$) crystals, the unit-cell of Nd:GLNO is largest because the ionic radius of La³⁺ (1.15 Å) is larger than those of Gd³⁺ (0.94 Å) and Y³⁺ (0.9 Å).

The X-ray rocking curves of the as-grown crystal are shown in Fig. 5. The diffraction peak shows a symmetric shape without splitting, suggesting that the as-grown crystal does not have a twin structure. Additionally, the full width at half-maximum (FWHM) values of the diffraction peak are 0.053°, 0.101°, and 0.0457°, which indicate high crystalline quality of the as-grown crystal.

3.2 Effective segregation coefficients

The concentrations of Nd³⁺ and La³⁺ ions in the as-grown Nd:GLNO crystal were measured using the XRF analysis. A slice for the measurement was cut in the shoulder part of the as-grown crystal. The effective segregation coefficients of Nd³⁺ and La³⁺ are calculated to be 0.71 and 0.47, respectively, according to the equation $k_{\text{eff}} = C_s/C_l$, where C_s and C_l are the ion concentration in the crystal and melt, respectively. Generally, the radius difference between the doping and matrix ions have great influence on the effective segregation.²⁸ The radii of the

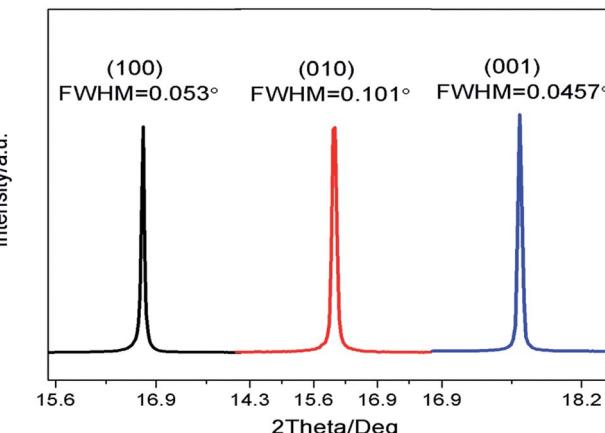


Fig. 5 X-ray rocking curve of the (100), (010), and (001) crystallographic faces.

doping ions are closer or smaller than those of the matrix ions, and the matrix ions are easier to be substituted by the doping ions. In this study, the radius of Nd³⁺ (0.98 Å) is closer to that of Gd³⁺ (0.94 Å) than that of La³⁺ (1.15 Å). Therefore, the effective segregation coefficient of Nd³⁺ in Nd:GLNO host is quite large and much larger than that of Nd:YAG (0.1–0.2). The large segregation coefficient of Nd³⁺ in Nd:GLNO host also means that high and uniform Nd³⁺-doped crystal can be grown *via* the Cz method.²⁹

3.3 Optical properties

3.3.1 Absorption spectrum. The samples for the absorption spectrum measurement were cut from the as-grown crystal along the crystallographic axes *a*, *b*, and *c*. All the samples were polished to a thickness of 2 mm. As can be seen, the room temperature absorption spectrum (Fig. 6) consists of 11 absorption bands in the measured wavelength range, which is associated with the observed transitions from the ⁴I_{9/2} ground state. *Via* comparing the absorption spectrum of Nd³⁺ in the

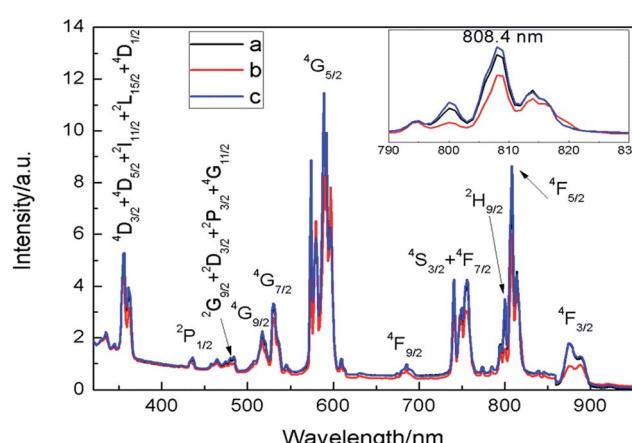


Fig. 6 Room temperature absorption spectra of the Nd:GLNO crystal along three crystallographic orientations. Inset: enlarged absorption spectra at around 808 nm.



Nd:GLNO crystal with that of Nd³⁺ in other tantalate and niobate crystals, all the absorption bands are assigned and denoted in Fig. 6. Moreover, the absorption spectra along *a*, *b*, and *c* axes show distinct differences, which can be attributed to the anisotropy of monoclinic crystals. The absorption of the Nd:GLNO crystal at 808 nm with *c*-orientation is strongest among the three orientations with an absorption coefficient of 8.97 cm⁻¹ and an FWHM of about 13 nm. The FWHM of Nd:GLNO is wider than that of Nd:GNO and Nd:YNO (much wider than the 2 nm of Nd:YAG), indicating that Nd:GLNO crystal can be better matched with commercial AlGaAs laser diode and is advantageous for improving the laser efficiency. Based on the equation $\sigma = \alpha(\lambda)/N_c$, where $\alpha(\lambda)$ is the absorption coefficient and N_c is the concentration of Nd³⁺ in the Nd:GLNO crystal, the absorption cross section of the Nd:GLNO crystal with *c*-orientation at 808 nm is estimated to be 10.49×10^{-20} cm². Compared with Nd:GNO and Nd:YNO, Nd:GLNO shows larger absorption cross section at 808 nm and thus has better absorption of the pumping energy.

3.3.2 Fluorescence spectrum. The room-temperature emission spectrum of the as-grown Nd:GLNO crystal in the wavelength range of 850–1400 nm is shown in Fig. 7. In the measured wavelength range, three emission bands (central wavelength at 885 nm, 1065 nm, and 1330 nm) were observed, which corresponded to the $^4F_{3/2} \rightarrow ^4I_J$ ($J = 9/2, 11/2$, and $13/2$, respectively) transitions of Nd³⁺. The comparison between the emission spectra of Nd:GLNO, Nd:GNO, and Nd:YNO crystal at around 1065 nm is shown in the inset of Fig. 7. Owing to inhomogeneous broadening in the Nd:GLNO crystal, the FWHM of Nd:GLNO (5 nm) is widest among these crystals. In addition, stimulated emission cross-section (σ_{em}) can be estimated from the fluorescence spectra using the Füchtbauer-Ladenburg equation³⁰

$$\sigma_{\text{em}}(\lambda) = \frac{\lambda^5 I(\lambda)}{8\pi n^2 c \tau_m \int \lambda I(\lambda) d\lambda}$$

where $I(\lambda)$ is the fluorescence intensity, n is the reflective index, as reported in,¹² c is the velocity of light, and τ_m is the measured lifetime, which is presented in the following section. Therefore,

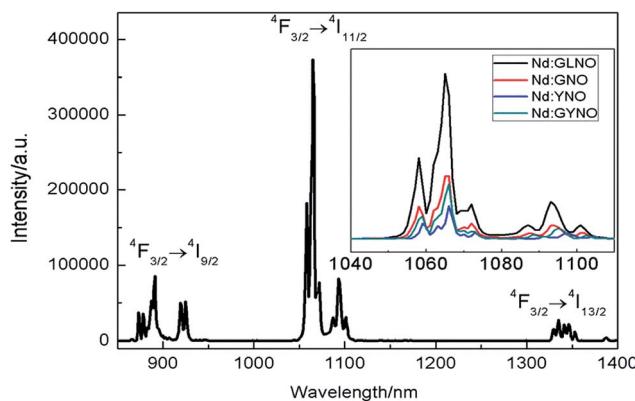


Fig. 7 Fluorescence spectra of the Nd:GLNO crystal at room temperature. Inset: comparison of the FWHM at around 1065 nm between Nd:GLNO, Nd:GNO, Nd:YNO, and Nd:GYNO crystals.

the stimulated emission cross-section value of the Nd:GLNO crystal at 1065 nm is estimated to be 18×10^{-20} cm², which is smaller than that of Nd:GYNO and Nd:YNO crystals.

3.3.3 Fluorescence decay curve. The fluorescence decay curves of $^4F_{3/2} \rightarrow ^4I_{11/2}$ transition at room temperature are shown in Fig. 8. The decay curve can be well fitted with a single exponential function, and fluorescence lifetime is fitted to be 176.4 μ s, which is longer than that of Nd:GYNO and Nd:YNO crystals. The small emission cross-section and long fluorescence lifetime indicates that the Nd:GLNO crystal possesses good energy storage capacity, which is advantageous for its application in a Q-switched laser.

The optical property comparison between Nd:GLNO crystal and other Nd-doped niobate laser crystals is listed in Table 1. As we can see, Nd:GLNO crystal shows good comprehensive spectroscopic and laser performance. Therefore, it is believable that Nd:GLNO crystal has great application potential in low- and moderate-power lasers.

3.4 Laser performance

Given that the strongest absorption of the Nd:GLNO crystal was along the *c*-orientation, the crystal with *c*-orientation was chosen to perform the CW laser characterization using a plano-plano resonator. The output power under various incident pump powers is shown in Fig. 9. It can be observed that the pump threshold is 204 mW, and a maximum output power of 957 mW at 1065 nm is obtained at a pump power of 3153 mW. Therefore, the optical-to-optical conversion efficiency was achieved is 30.3%, and a slope efficiency of 34.2% was derived from the linear fitting to the curve. As shown in Table 1, the slope efficiency of Nd:GLNO is comparable to that of other Nd-doped niobate laser crystals, if not better. However, the slope efficiency of Nd:GLNO is comprehensively lower than that of commercially Nd-doped vanadate laser crystals. The reason for the low laser output power and slope efficiency of Nd:GLNO crystal are listed below: first, the investigation on Nd:GLNO is just in the preliminary stage and higher quality crystals are expected to be obtained in the future studies. Second, the crystal used in the laser

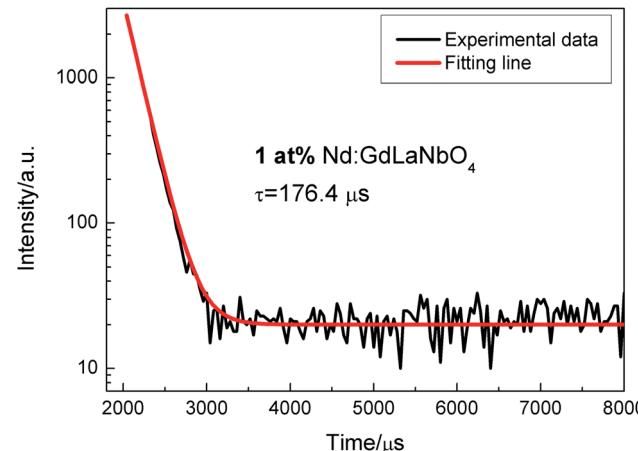


Fig. 8 Fluorescence decay curve of the $^4F_{3/2} \rightarrow ^4I_{11/2}$ transition.



Table 1 Optical property comparison between Nd:GLNO crystal and other Nd-doped niobate laser crystals

| Crystals | FWHM (808 nm) | σ_a (10^{-20} cm 2) (808 nm) | σ_{em} (10^{-20} cm 2) (1.06 μ m) | τ (μ s) | η (%) (slope efficiency) | Ref. |
|----------|---------------|--|---|-------------------|-------------------------------|-----------|
| Nd:GLNO | 13 | 10.49 | 18 | 176.4 | 34.2 | This work |
| Nd:GYNO | 5–14 | 11.6 | 20.5 | 156 | 30.4 | 25 |
| Nd:YNO | 6 | 8.7 | 22 | 152 | 24.0 | 12 |
| Nd:GTO | 6 | 5.1 | 39 | 178 | 36.0 (coated) | 11 |
| Nd:GYTO | 6–12 | 6.9 | 22 | 182 | 38.5 (coated) | 26 |

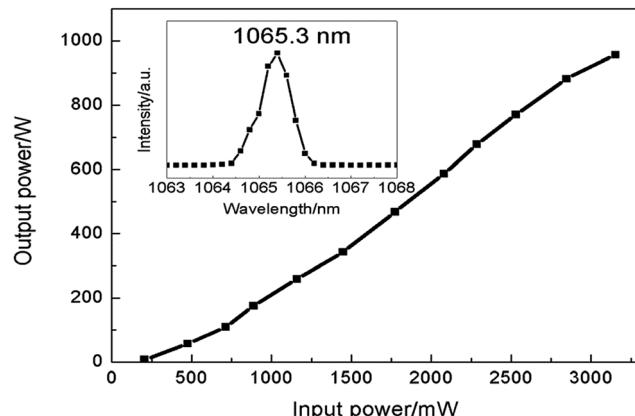


Fig. 9 Laser output power of Nd:GdLaNbO₄ versus incident power. Inset: laser spectrum of the Nd:GLNO crystal at around 1.06 μ m.

experiment is uncoated. Third, the concentration of Nd³⁺ in the GLNO host is not optimized. Therefore, the laser efficiency of the Nd:GLNO crystal is expected to improve in the future studies.

4. Conclusions

In this study, a new high-quality mixed laser crystal Nd:GdLaNbO₄ has been grown successfully by the Czochralski method. The unit cell parameters are obtained to be $a = 5.381$ \AA , $b = 11.112$ \AA , $c = 5.112$ \AA , $\alpha = \gamma = 90^\circ$, and $\beta = 94.551^\circ$ by the Rietveld refinement method. The effective segregation coefficient of Nd³⁺ ion in the Nd:GLNO crystal is measured to be 0.71. Room temperature absorption spectra and emission spectra show obvious inhomogeneous broadening, which can be attributed to the disordered structure of the Nd:GLNO crystal. The broad emission spectra and relatively long lifetime indicates that Nd:GLNO is suitable for generating ultrashort pulse lasers. Moreover, CW laser operation at 1.06 μ m is demonstrated with *c*-oriented Nd:GLNO crystal for the first time. A maximum output power of 957 mW is obtained corresponding to an optical-to-optical conversion efficiency of 30.3% and slope efficiency of 34.2%. All the obtained results indicate that Nd:GLNO is a very promising laser material.

Acknowledgements

This work was financially supported by the National Natural Science Foundation of China (Grants No. 51502292, and 61405206).

Notes and references

- J. Sanghera, W. Kim, G. Villalobos, B. Shaw, C. Baker, J. Frantz, B. Sadowski and I. Aggarwal, *Materials*, 2012, **5**, 258–277.
- S. M. Hooker, *Nat. Photon.*, 2013, **7**, 775–782.
- R. Ortiz, S. M. Flores, I. Quintana, M. Vivanco, J. R. Sarasua and J. L. T. Herrera, *Mater. Sci. Eng., C*, 2014, **37**, 241–250.
- J. Cheng, C. S. Liu, S. Shang, D. Liu, W. Perrie, G. Dearden and K. Watkins, *Opt. Laser Technol.*, 2013, **46**, 88–102.
- P. V. Zarubin, *Quantum Electron.*, 2002, **32**, 1048–1064.
- H. H. Yu, J. H. Liu, H. J. Zhang, A. A. Kaminskii, Z. P. Wang and J. Y. Wang, *Laser Photonics Rev.*, 2014, **8**, 847–864.
- H. J. Zhang, J. H. Liu, J. Y. Wang, C. Q. Wang, L. Zhu, Z. S. Shao, X. L. Meng, X. B. Hu and M. H. Jiang, *J. Opt. Soc. Am. B*, 2002, **19**, 18–27.
- H. H. Yu, H. J. Zhang, Z. P. Wang, J. Y. Wang, Z. S. Shao, M. H. Jiang and X. Y. Zhang, *Opt. Express*, 2007, **15**, 3206–3211.
- H. H. Yu, H. J. Zhang, Z. P. Wang, J. Y. Wang, Y. G. Yu, X. F. Cheng, Z. S. Shao, M. H. Jiang, Z. C. Ling and H. R. Xia, *J. Appl. Phys.*, 2007, **101**, 113109.
- S. J. Ding, F. Peng, Q. L. Zhang, J. Q. Luo, W. P. Liu, D. L. Sun, R. Q. Dou, J. Y. Gao, G. H. Sun and M. J. Cheng, *J. Alloys Compd.*, 2017, **693**, 339–343.
- F. Peng, H. J. Yang, Q. L. Zhang, J. Q. Luo, W. P. Liu, D. L. Sun, R. Q. Dou and G. H. Sun, *Appl. Phys. B*, 2015, **118**, 549–554.
- S. J. Ding, F. Peng, Q. L. Zhang, J. Q. Luo, W. P. Liu, D. L. Sun, R. Q. Dou and G. H. Sun, *Opt. Mater.*, 2016, **62**, 7–11.
- T. J. Yu, S. K. Lee, J. H. Sung, J. W. Yoon, T. M. Jeong and J. Lee, *Opt. Express*, 2012, **20**, 10807–10815.
- N. Tolstik, E. Sorokin and I. T. Sorokina, *Opt. Express*, 2014, **22**, 5564–5571.
- U. Anwar, C. W. Yi and M. Hiroshi, *Sci. Prog.*, 2013, **96**, 1–18.
- J. H. Ha, B. J. Lee, D. J. Hwang and D. Kim, *RSC Adv.*, 2016, **6**, 86232.
- H. H. Xu, S. Han, H. H. Yu, Z. P. Wang, J. Y. Wang, H. J. Zhang and D. Y. Tang, *J. Cryst. Growth*, 2014, **387**, 66–72.
- Z. B. Pan, J. Ma, H. H. Xu, D. Y. Tang, H. Q. Cai, H. H. Yu, H. J. Zhang and J. Y. Wang, *RSC Adv.*, 2015, **5**, 44137.
- B. L. Wang, L. Tian, H. H. Yu, H. J. Zhang and J. Y. Wang, *Opt. Lett.*, 2015, **40**, 3213–3216.
- H. H. Xu, H. H. Yu, Z. P. Wang, S. Han, Y. C. Wang, Z. B. Pan, Y. Y. Zhang, S. Q. Sun, J. Y. Wang and H. J. Zhang, *Opt. Express*, 2012, **20**, 16524–16531.

21 V. G. Ostroumov, G. Huber, A. L. Zagumenniy, Y. D. Zavartsev, P. A. Studenikin and I. A. Shcherbakov, *Opt. Commun.*, 1996, **124**, 63–68.

22 H. H. Yu, H. J. Zhang, Z. P. Wang, J. Y. Wang, Y. G. Yu, Z. S. Shao, M. H. Jiang and X. Y. Zhang, *Appl. Phys. Lett.*, 2007, **90**, 231110.

23 S. Han, H. H. Xu, Y. G. Zhao, L. J. Chen, Z. P. Wang, H. H. Yu, H. J. Zhang and X. G. Xu, *Infrared Phys. Technol.*, 2013, **60**, 66–70.

24 S. P. Ng, D. Y. Tang, A. Q. Liu, L. J. Qin and X. L. Meng, *Opt. Commun.*, 2006, **259**, 256–260.

25 S. J. Ding, Q. L. Zhang, F. Peng, W. P. Liu, J. Q. Luo, R. Q. Dou, G. H. Sun, X. F. Wang and D. L. Sun, *J. Alloys Compd.*, 2017, **698**, 159–163.

26 F. Peng, H. J. Yang, Q. L. Zhang, J. Q. Luo, D. L. Sun, W. P. Liu, G. H. Sun, R. Q. Dou, X. F. Wang and X. Xing, *Opt. Mater. Express*, 2015, **5**, 2536–2544.

27 A. C. Larson and R. B. V. Dreele, *General structure analysis system*, Los Alamos National Laboratory Report no. LAUR, 2004, vol. 86, p. 748.

28 H. L. Zhang, X. J. Sun, J. Q. Luo, Z. Q. Fang, X. Y. Zhao, M. J. Cheng, Q. L. Zhang and D. L. Sun, *J. Alloys Compd.*, 2016, **672**, 223–228.

29 J. Y. Gao, Q. L. Zhang, D. L. Sun, J. Q. Luo, W. P. Liu and S. T. Yin, *Opt. Commun.*, 2012, **285**, 4420–4426.

30 I. Sokolska, E. Heumann, S. Kuck and T. Lukasiewicz, *Appl. Phys. B*, 2002, **71**, 893–896.

