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# Facile synthesis of novel $\text{CuCo}_2\text{S}_4$ nanospheres for coaxial fiber supercapacitors†

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Flexible and highly efficient energy storage units act as one of the key components in portable electronics. In this work, based on  $\text{CuCo}_2\text{S}_4$  nanospheres, a flexible all-solid-state coaxial fiber supercapacitor is designed and produced via a low cost and facile method. The as-fabricated flexible device exhibits high specific capacitance, high energy density, excellent rate capability and outstanding long-term cycling stability. This approach can be an efficient strategy for the preparation of novel and low-cost electrodes for various applications.

## Introduction

Developing high-performance electrochemical energy storage devices has been one of the important issues in the energy strategic projects established by the governments of countries around the world.<sup>1</sup> Among various energy storage devices, rechargeable supercapacitors (SCs), with high power density, fast charge/discharge rate, and long lifespan, are considered typically as one of the most appropriate choices for energy storage and conversion.<sup>2,3</sup> Especially, flexible SC, such as fiber SCs, and flexible micro-SC, have been devoted a significant amount of effort, owing to their potential applications in the fields of portable electronic devices. To achieve flexible SCs with comprehensive advantage requirements, such as high charge storage capability and mechanical flexibility is the greatest challenge,<sup>4</sup> one of the most effective ways is to realize the perfect combination of bendable current collectors with good mechanical flexibility and electrode materials with a high capacitive performance. The commercial Ti wire own high conductivity and high flexibility. Fiber SC fabricated based on Ti wire can render the designer free from conventional constraints. Furthermore, the Ti wire based SC could be easily worn on human body, realization the practical wearable electronics.

Generally, the properties of SCs are influenced by the electrode material, electrolyte and assembly technology. Among them, the vital factor is electrode material, and therefore it has become a research focus for researchers. Transition metal oxides are widely considered as the most low cost, low toxicity, environmental friendliness, and multiple oxidation states. However, oxides usually have low electrical conductivity, which

limits their rate capability, another key parameter of supercapacitor.<sup>5,6</sup> In contrast, metal sulfides have excellent chemical stability, rich valence,<sup>7–9</sup> and better electrical conductivity.<sup>10–12</sup> Compared to the single-component metal sulfides, binary metal sulfides show richer redox reactions and higher electronic conductivity, resulting in the enhancement of the electrochemical performances. Dong *et al.* reported the fabrication of zinc cobalt sulfide ( $\text{Zn}_{0.76}\text{Co}_{0.24}\text{S}$ ) nanoartichokes by co-decomposing Zn and Co precursors in a hot oleylamine/oleic acid solution through a facile oil phase approach, which obtained a capacitance of  $486.2 \text{ F g}^{-1}$  at  $2 \text{ A g}^{-1}$ .<sup>13</sup> Xiao *et al.* reported the single crystalline nanotube arrays  $\text{NiCo}_2\text{S}_4$ , which manifested higher specific capacitance and rate capability than  $\text{NiCo}_2\text{O}_4$ .<sup>14</sup> Yuan *et al.* fabricated hollow hetero- $\text{NiCo}_2\text{S}_4/\text{Co}_9\text{S}_8$  spindles, and delivered a capacitance of  $749 \text{ F g}^{-1}$ .<sup>15</sup> Very recently,  $\text{CuCo}_2\text{S}_4$  nanoparticles have been synthesized in glycerol fulfills a capacitance of  $5030 \text{ F g}^{-1}$  at  $20 \text{ A g}^{-1}$  in a polysulfide electrolyte using a three-electrode system.<sup>16</sup>

Based on the above considerations, in this work, we successfully designed  $\text{CuCo}_2\text{S}_4$  nanospheres on Ti wire for the first time prepared by a simple template-free hydrothermal method as a novel binder-free electrode for high-performance flexible coaxial fiber SC. The synergistic contributions from high density of active sites, rapid transfer of electrons and fast diffusion of ions lead the nanospheres-assembled SC to have a high specific capacitance and excellent rate capability.

## Experimental

### Fabrication of $\text{CuCo}_2\text{S}_4$ nanowires on Ti wire

Prior to deposition, commercial Ti wires were cleaned by sonication sequentially in acetone, 1 M HCl solution, deionized water, and ethanol for 15 min, respectively. After being dried, the Ti wire was transferred into Teflon-lined stainless autoclave. In a typical synthesis of  $\text{CuCo}_2\text{S}_4$  nanosphere, 1 mmol  $\text{CuCl}_2 \cdot 6\text{H}_2\text{O}$ , 2 mmol  $\text{CoCl}_2 \cdot 6\text{H}_2\text{O}$ , 6 mmol  $\text{NH}_4\text{F}$  and 9 mmol

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$\text{CO}(\text{NH}_2)_2$  were dissolved in 30 mL deionized water by constant stirring. Then the solution was sealed in a Teflon-lined stainless-steel autoclave and kept at 120 °C for 6 h. After the reaction cooled to room temperature, the product was collected, washed, and then thermal treated at 300 °C for 3 h to obtain  $\text{CuCo}_2\text{O}_4$  nanospheres. Finally, the obtained  $\text{CuCo}_2\text{O}_4/\text{Ti}$  wire was immersed in 1 g  $\text{Na}_2\text{S}$  with 25 mL distilled water solution and kept at 140 °C for 10 h. After cooling slowly to room temperature, the Ti wire covered  $\text{CuCo}_2\text{S}_4$  nanospheres was washed several times under deionized water and ethanol, then dried under vacuum at 60 °C for 8 h.

### Three-electrode electrochemical measurements

As-prepared  $\text{CuCo}_2\text{S}_4/\text{Ti}$  wire was used as the working electrode directly, a platinum electrode, and a saturated calomel electrode (SCE) were used as counter electrode and reference electrode, respectively. The electrochemical test was carried out in 3 M KOH electrolyte solution.

### Fabrication of flexible all-solid-state coaxial SC

PVA/KOH gel electrolyte was prepared by mixing 6 g PVA (polyvinyl alcohol) and 3 g KOH with 60 mL deionized water and heated to 95 °C under vigorous stirring until it became clear. The electrodes were coated with the PVA/KOH and dried at 45 °C for 1 h. The final coaxial fiber SC was formed by wrapping the gel-coated  $\text{CuCo}_2\text{S}_4$  electrode around the other gel-coated  $\text{CuCo}_2\text{S}_4$  electrode.

### Characterization

The synthesized products were characterized with an X-ray diffractometer. The morphology of the samples was characterized by scanning electron microscopy (SEM) and transmission electron microscopy (TEM). Cyclic voltammetry (CV), electrochemical impedance spectroscopy (EIS), and galvanostatic charge/discharge measurements were carried out on an electrochemical workstation (Ivium, CompactStat. 10800).

### Calculation

The capacitance values were calculated from the discharge curve based on equation:

$$C_A = I \times \Delta t / (A \times \Delta V) \text{ (mF cm}^{-2}\text{)} \quad (1)$$

where  $\Delta V$  is the potential drop during discharge (in V),  $I$  is the discharge current density (mA),  $\Delta t$  is discharge time (in seconds),  $A$  is the area of the coaxial SC (in cm). The electrochemical performance of device shown in the Ragone plot was based on the areal capacitance and measured from the discharge curves. The energy density and power density of the device was obtained from the equations:

$$E = C_A \times \Delta V^2 / 7200 \quad (2)$$

$$P = E \times 3600 / \Delta t \quad (3)$$

where  $E$  is the energy density (in  $\text{W h cm}^{-2}$ ), and  $P$  is the power density (in  $\text{W cm}^{-2}$ ).

## Results and discussion

In this work,  $\text{CuCo}_2\text{S}_4$  was developed on Ti fiber by the following two steps: (1) the  $\text{CuCo}_2\text{O}_4$  nanosphere was hydrothermally grown on highly conductive Ti wire; (2) the hydrothermal treatment of such  $\text{CuCo}_2\text{O}_4$  in the presence of  $\text{Na}_2\text{S}$  lead to  $\text{CuCo}_2\text{S}_4$  *via* an anion-exchange reaction. The crystallographic structures of  $\text{CuCo}_2\text{S}_4$  was characterized by X-ray power diffraction (XRD). All diffraction patterns reveal the successful formation of the carrollite structure phase of  $\text{CuCo}_2\text{S}_4$  (according to the JCPDS card no. 42-1450), as shown in Fig. 1a. The main diffraction peaks appear at 26.6, 31.3, 38, 47, 50, and 54.8 respectively correspond to the (022), (113), (004), (224), (115) and (044) planes of the carrollite phase of the  $\text{CuCo}_2\text{S}_4$  structure. The XRD pattern of  $\text{CuCo}_2\text{O}_4$  was shown in Fig. S1.† To gain further information on the structure and composition of  $\text{CuCo}_2\text{S}_4$ , we resort to X-ray photoelectron spectroscopy (XPS) measurement and the results is shown in Fig. 1b–d.

The spin-orbit doublet corresponding to Cu  $2p_{1/2}$  and Cu  $2p_{3/2}$  is readily observed in the Cu 2p region at 952.2 and 932.3 eV ( $\Delta E = 19.9$  eV), respectively. As regards the Co 2p XPS spectrum of  $\text{CuCo}_2\text{S}_4$ , it shows a doublet containing a low energy band (Co  $2p_{3/2}$ ) and a high energy band (Co  $2p_{1/2}$ ) at 781.1 and 796.3 eV (Fig. 1), consistent with the results reported elsewhere.<sup>17</sup> The spin-orbit splitting value of Co  $2p_{1/2}$  and Co  $2p_{3/2}$  is over 15 eV, suggesting the coexistence of  $\text{Co}^{2+}$  and  $\text{Co}^{3+}$ .<sup>18</sup> In the S 2p spectrum of  $\text{CuCo}_2\text{S}_4$ , the peak at 162.4 and 161.3 eV are characteristics of  $\text{S}^{2-}$ ,<sup>19</sup> and the component 169.2 eV can be ascribed to the sulphur in low coordination at the surface.<sup>20</sup> According to the XPS analysis, the near-surface of the  $\text{CuCo}_2\text{S}_4$  sample has a composition of  $\text{Co}^{2+}$ ,  $\text{Co}^{3+}$ ,  $\text{Cu}^{2+}$ ,  $\text{Cu}^{3+}$ , and  $\text{S}^{2-}$ , which is in good agreement with the  $\text{CuCo}_2\text{S}_4$ .

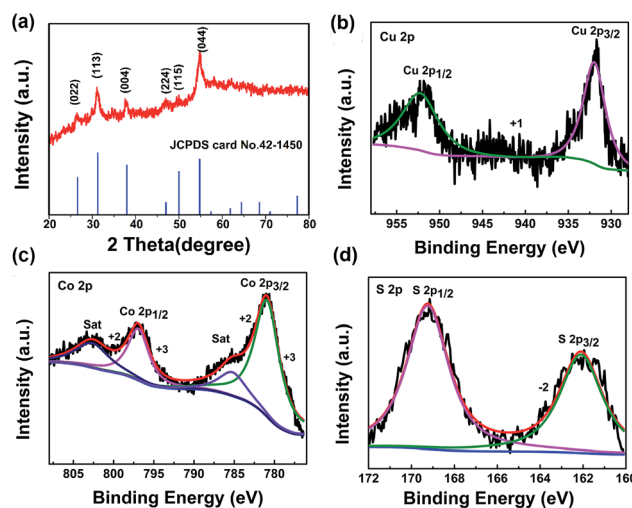
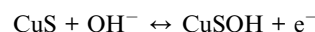
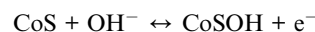


Fig. 1 (a) XRD pattern of the as-prepared  $\text{CuCo}_2\text{S}_4$ , (b–d) XPS spectra of Cu 2p, Co 2p and S 2p, respectively.



The morphologies of the  $\text{CuCo}_2\text{O}_4$  and  $\text{CuCo}_2\text{S}_4$  were investigated by SEM and TEM, as shown in Fig. 2. It can be seen that the Ti wire was covered by spherical  $\text{CuCo}_2\text{O}_4$  sample with diameter of about 15–20  $\mu\text{m}$  (Fig. 2a and b). Typical the  $\text{CuCo}_2\text{O}_4$  micro-spheres are composed of numerous small nanowires radially grown from the center. To better understand the formation mechanism of the as-synthesized urchin-like  $\text{CuCo}_2\text{O}_4$  nanostructures on Ti wire, we carried out an additional experiment by reducing the molar of urea to 7 mmol, the corresponding SEM is shown in Fig. S2,<sup>†</sup> from which we can see nanowires and spheres were both found. Based on the result, it is manifested that the spheres were evolved from nanowires. After being treated with  $\text{Na}_2\text{S}$ , the obtained  $\text{CuCo}_2\text{S}_4$  can well maintain the spheres nanostructures (Fig. 2c and d). The mass loading of  $\text{CuCo}_2\text{O}_4$  on Ti wire is 0.36 and 0.5  $\text{mg cm}^{-2}$  when the urea is 7 mmol and 9 mmol, respectively. It was expected that unique structure might have large surface area that could provide high specific capacitance due to the easy access of the active material in the redox process to their interface. More details of the morphological and structural features of the as-obtained sphere-like  $\text{CuCo}_2\text{S}_4$  nanostructures are studied by HRTEM and selected-area electron diffraction (SAED). Fig. 2e exhibits the inter-planar spacing of 0.17 and 0.28 nm, which are consistent with the (004) and (113) planes of  $\text{CuCo}_2\text{S}_4$ . Selected area electron diffraction (SAED) pattern of  $\text{CuCo}_2\text{S}_4$  in Fig. 2f confirms that the  $\text{CuCo}_2\text{S}_4$  nanospheres are of polycrystalline nature. According to the results, a possible mechanism is proposed. When the  $\text{CuCo}_2\text{O}_4$  reacted with  $\text{S}^{2-}$ ,  $\text{HS}^-$ , and  $\text{H}_2\text{S}$  in  $\text{Na}_2\text{S}$  solution, the ion-exchange reaction occurs slowly to convert all of the  $\text{CuCo}_2\text{O}_4$  to  $\text{CuCo}_2\text{S}_4$  and keep the morphology. The photos of pure Ti wire,  $\text{CuCo}_2\text{O}_4/\text{Ti}$  and  $\text{CuCo}_2\text{S}_4/\text{Ti}$  were demonstrated in Fig. S3,<sup>†</sup> respectively.

Regarding with such sphere-like nanostructure of Cu–Co sulfides, which are almost composed of surfaces with the most active sites exposed outside for the highly surface related faradaic reactions, their potential applications in pseudocapacitors are studied. Fig. 3a shows the cyclic voltammogram (CV) curves at different scan rates between  $-0.2$  to  $0.45$  V. The redox peaks are derived from reversible faradaic reactions including following equations.



To further investigated the electrochemical performance of  $\text{CuCo}_2\text{S}_4$  electrode, the galvanostatic charge/discharge measurements were conducted at different current densities between  $-0.2$  to  $0.45$  V (Fig. 3b), the charge/discharge curves of  $\text{CuCo}_2\text{S}_4$  show some curvature, which is due to redox transitions and corresponds to the typical redox couples in the CV curves.

The practical performance of the electrode in a full-cell setup was further evaluated by fabricating a  $\text{CuCo}_2\text{S}_4$ -based flexible coaxial fiber SC. Fig. 3c shows the CV curves under different scan rates between 0 to 1 V. Galvanostatic charge/discharge curves of the flexible coaxial fiber SC at a set of current densities were further illustrated in Fig. 3d. The internal resistance (IR) drop *versus* current density are plotted in Fig. S4,<sup>†</sup> and corresponding linear functions are fitted. The small slope values demonstrate the low resistance of the as-fabricated SC. Fig. 3e shows the CV curves at a scan rate of  $10 \text{ mV s}^{-1}$  of  $\text{CuCo}_2\text{O}_4$  and  $\text{CuCo}_2\text{S}_4$  based flexible fiber SC, respectively. The integrated CV area for the  $\text{CuCo}_2\text{S}_4$  fiber SC is significantly larger, this confirms that the  $\text{CuCo}_2\text{S}_4$  nanosphere based electrodes are superlative for pseudocapacitive devices. We have also measured the charge transport and ion diffusion of two-electrode materials using electrochemical impedance spectroscopy (EIS), as shown in Fig. 3f, which reveals that the  $\text{CuCo}_2\text{S}_4$  shows a much smaller  $R_{\text{ct}}$  in the Nyquist plots as compared to that of the  $\text{CuCo}_2\text{O}_4$ . The results clearly

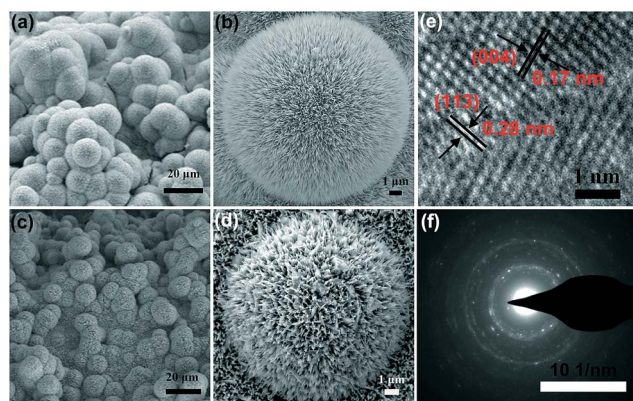


Fig. 2 SEM images of  $\text{CuCo}_2\text{O}_4$  (a and b), (c and d)  $\text{CuCo}_2\text{S}_4$ . (e) HRTEM image, (f) SAED pattern of  $\text{CuCo}_2\text{S}_4$ .

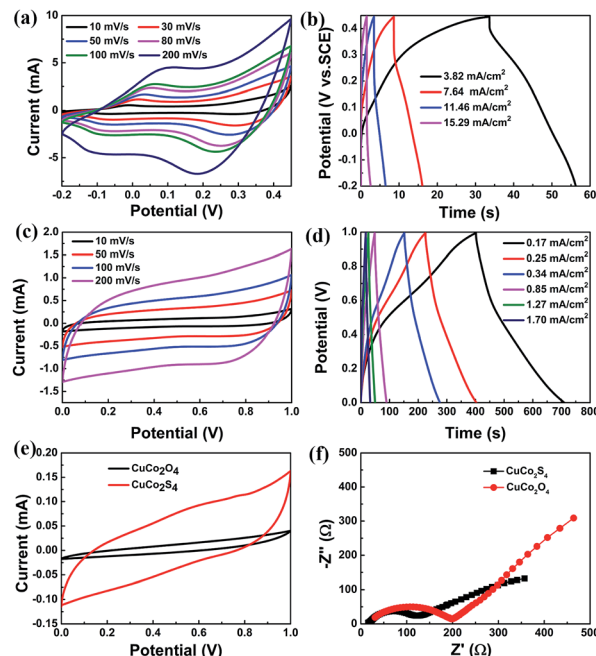


Fig. 3 (a and b) CV curves and galvanostatic charge/discharge curves of  $\text{CuCo}_2\text{S}_4$  electrode in 3 M KOH electrolyte between  $-0.2$  to  $0.45$  V. (c and d) CV curves and galvanostatic charge/discharge curves of as-fabricated flexible coaxial fiber SC device between 0 to 1 V. (e and f) The comparison of CV curve at  $10 \text{ mV s}^{-1}$  and EIS curve for flexible fiber SC based on  $\text{CuCo}_2\text{O}_4$  and  $\text{CuCo}_2\text{S}_4$  electrodes, respectively.



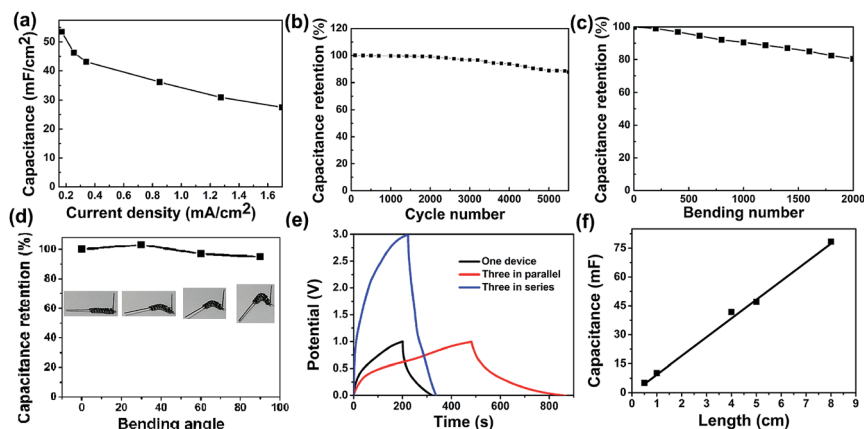


Fig. 4 (a) Areal capacitances as a function of current. (b) Capacitance retention *versus* cycle number. (c) Capacitance retention *versus* bending number. (d) Capacitance retention *versus* bending angles. Inset is the photos of fiber SC at different bending angles. (e) Galvanostatic charge/discharge curves for the fiber SCs with different configuration at a fixed current of  $0.34 \text{ mA cm}^{-2}$ . (f) Capacitance *versus* electrode length for fiber SCs.

demonstrate that the  $\text{CuCo}_2\text{S}_4$  display favorable charge-transfer kinetics and fast electron transport and thus exhibit the dramatically enhanced pseudocapacitive performance.

The specific capacitance values are calculated to be approximately at  $53.5 \text{ mF cm}^{-2}$  at  $0.17 \text{ mA cm}^{-2}$ , which are much higher than  $\text{MnO}_2/\text{graphene}$  ASC ( $4.57 \text{ mF cm}^{-2}$ ),<sup>21</sup>  $\text{MnO}_2/\text{CF}/\text{MoO}_3$  ASC ( $4.86 \text{ mF cm}^{-2}$ ),<sup>22</sup>  $\text{CNT}/\text{Co}_3\text{O}_4$  yarns SC ( $52.6 \text{ mF cm}^{-2}$ ),<sup>23</sup>  $\text{NiCo}_2\text{O}_4$  NG@CF ( $25.03 \text{ mF cm}^{-2}$ ).<sup>24</sup> Fig. 4b shows the capacitance retention evaluated at  $0.34 \text{ mA cm}^{-2}$ , after 5500 cycles, the capacitance retention is about 86% of the initial capacitance. The capacitance retention *versus* bending at  $30^\circ$  for 2000 cycles and bending different angles of the fiber SC is demonstrated in Fig. 4c and d, respectively. The results indicate the fiber SC has good flexibility and mechanical stability. Fig. 4e shows GCD curves of three SCs connected in parallel and in series. As compared with a single SC, the discharge time of the assembled parallel devices is about three times longer than that of a single device, which means approximately triple capacitance could be achieved. For three SCs combined in series, under the similar discharge time, the output voltage of this device can be extended to 3 V. Overall, these results illustrate that electrical performances of the series and parallel combinations of the SC devices, which show a good agreement with the theoretical models of series and parallel combined circuits, enabling them to combine multiply for practical applications. In Fig. S5,<sup>†</sup> the galvanostatic charge–discharge curves of the fiber SC with different length were shown. The calculated length specific capacitance has an approximately linear increase with the length increase of fiber SC (Fig. 4f), indicating the uniformity and the promising potential of large-scale fabrication of  $\text{CuCo}_2\text{S}_4$  nano-sphere fiber for flexible energy storage. Ragone plot (power density *vs.* energy density) of the flexible coaxial fiber SC describing was obtained and shown in Fig. S6.<sup>†</sup> The energy density decreased from  $7.29$  to  $3.55 \text{ } \mu\text{Wh cm}^{-2}$ , compared to the power density increased from  $0.08$  to  $0.82 \text{ mW cm}^{-2}$ , which is higher than that of those reported SC values.<sup>21–24</sup>

## Conclusions

In summary, we report the synthesis of sphere-like  $\text{CuCo}_2\text{S}_4$  nanostructure on flexible Ti wire for the first time *via* a two-step hydrothermal method. In this unique nanoarchitecture, the  $\text{CuCo}_2\text{S}_4$  spheres act as an excellent pseudocapacitive material that can be capable of fast electron conduction and ion diffusion. The as-fabricated flexible coaxial fiber SC based on  $\text{CuCo}_2\text{S}_4$  micro-sphere exhibited a high specific capacitance of  $53.5 \text{ mF cm}^{-2}$ , good rate capability, as well as excellent cycling life.

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## Notes and references

- (a) D. Qi, Y. Liu, Z. Liu, L. Zhang and X. Chen, *Adv. Mater.*, 2017, **29**, 1602802; (b) L. B. Liu, Y. Yu, C. Yan, K. Li and Z. J. Zheng, *Nat. Commun.*, 2015, **6**, 7260; (c) Y. Yang, Q. Y. Huang, L. Y. Niu, D. R. Wang, C. Yan, Y. Y. She and Z. J. Zheng, *Adv. Mater.*, 2017, 1606679, DOI: 10.1002/adma.201606679; (d) Y. B. Zhang, B. Wang, F. Liu, J. P. Cheng, X. W. Zhang and L. Zhang, *Nano Energy*, 2016, **27**, 627.
- X. Li, L. Y. Zhang, N. Li, T. Yi and Z. Q. Liu, *Adv. Mater.*, 2016, **28**, 7680–7687.
- Y. M. Li, Y. S. Jiang, X. M. Sun, B. J. Wang and H. S. Peng, *Adv. Mater.*, 2016, **28**, 8431–8438.
- M. Zhu, Y. Huang, Y. Huang, H. Li, Z. Wang, Z. Pei, Q. Xue, H. Geng and C. Zhi, *Adv. Mater.*, 2017, 1605137.
- Y. Q. Zhu, C. B. Cao, S. Tao, W. S. Chu, Z. Y. Wu and Y. D. Li, *Sci. Rep.*, 2014, **4**, 5787.
- Y. Q. Fan, G. J. Shao, Z. P. Ma, G. L. Wang, H. B. Shao and S. Yan, *Part. Part. Syst. Charact.*, 2014, **31**, 1079–1083.



- 7 W. C. Xu, Y. Q. Liang, Y. G. Su, S. L. Zhu, Z. D. Cui, X. J. Yang, A. Inoue, Q. Wei and C. Y. Liang, *Electrochim. Acta*, 2016, **211**, 891–899.
- 8 J. Xiao, L. Wan, S. Yang, F. Xiao and S. Wang, *Nano Lett.*, 2014, **14**, 831.
- 9 R. J. Zou, Z. Y. Zhang, M. F. Yuen, J. Q. Hu, C. S. Lee and W. J. Zhang, *Sci. Rep.*, 2015, **5**, 7862.
- 10 Z. M. Zhang, Q. Wang, C. J. Zhao, S. D. Min and X. Z. Qian, *ACS Appl. Mater. Interfaces*, 2015, **7**, 4861.
- 11 L. Yu, L. Zhang, H. B. Wu and X. W. Lou, *Angew. Chem., Int. Ed.*, 2014, **53**, 3711–3714.
- 12 Y. M. Gao, L. W. Wei, W. T. Cui, S. Z. Zheng, Z. Hou and H. W. Chen, *ACS Appl. Mater. Interfaces*, 2015, **7**, 4311–4319.
- 13 J. Yang, Y. Zhang, C. Sun, G. Guo, W. Sun, W. Huang, Q. Yan and X. Dong, *J. Mater. Chem. A*, 2015, **3**, 11462–11470.
- 14 J. W. Xiao, L. Wan, S. H. Yang, F. Xiao and S. Wang, *Nano Lett.*, 2014, **14**, 831–838.
- 15 L. R. Hou, Y. Y. Shi, S. Q. Zhu, M. Rehan, G. Pang, X. G. Zhang and C. Z. Yuan, *J. Mater. Chem. A*, 2017, **5**, 133–144.
- 16 J. Tang, Y. Ge, J. Shen and M. Ye, *Chem. Commun.*, 2016, **52**, 1509–1512.
- 17 J. Xiao, X. Zeng, W. Chen, F. Xiao and S. Wang, *Chem. Commun.*, 2013, **49**, 11734.
- 18 J. Xu, P. Gao and T. S. Zhao, *Energy Environ. Sci.*, 2012, **5**, 5333.
- 19 Q. Wang, L. Jiao, H. Du, Y. Si, Y. Wang and H. J. Yuan, *Mater. Chem.*, 2012, **22**, 21387.
- 20 H. Chen, J. Jiang, L. Zhang, H. Wan, T. Qi and D. Xia, *Nanoscale*, 2013, **5**, 8879.
- 21 N. Yu, H. Yin, W. Zhang, Y. Liu, Z. Y. Tang and M. Q. Zhu, *Adv. Energy Mater.*, 2016, **6**, 1501458.
- 22 J. Noh, C. M. Yoon, Y. K. Kim and J. Jang, *Carbon*, 2017, **116**, 470–478.
- 23 F. H. Su, X. M. Lv and M. H. Miao, *Small*, 2015, **11**(7), 854–861.
- 24 S. T. Senthilkumar, N. Q. Fu, Y. Liu, Y. Wang, L. M. Zhou and H. T. Huang, *Electrochim. Acta*, 2016, **211**, 411–419.

