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Theory guided systematic molecular design of benzothiadiazole–phenazine based self-assembling electron-acceptors†

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We report two new self-assembling n-type materials in which the design process was systematically analyzed using theoretical calculations prior to experimental synthesis. Benzothiadiazole (A') and alkoxyphenazine (A) serve as our basic electron-deficient building blocks to construct two candidate molecules with an acceptor (A) – acceptor (A') – acceptor (A) configuration. We conducted a computational characterization (B3LYP/6-31G*) of the electronic properties for structural subunits linked together and analyzed in a step-by-step fashion, thereby culminating in the target molecules of interest. We found that E_{LUMO} was controlled primarily by benzothiadiazole as was evident by orbital localization on this subunit. Meanwhile, E_{HOMO} was influenced by the dihedral angle between A and A'. The molecule with A and A' coupled with a C–C triple bond (BTD-P-T) was found to be planar with a more stabilized E_{LUMO} and a reduced E_{gap} when compared to its C–C single bond counterpart (BTD-P-S). The two molecules were synthesized and characterized to verify the theoretical findings. Optical, electrochemical, and thermal properties were characterized with UV-vis absorption and fluorescence spectroscopy, cyclic voltammetry, and differential scanning calorimetry, respectively. In addition, the molecular packing was examined by X-ray powder diffraction. As predicted by theory, BTD-P-T exhibited a lower E_{gap} based on the system's lower E_{LUMO} in comparison to BTD-P-S. BTD-P-T exhibited a higher T_m and crystallinity due to its planarity. Both BTD-P-S and BTD-P-T exhibited excellent fibrillation ability upon solvent casting. Bulk heterojunction (BHJ) organic solar cell (OSC) devices were fabricated using BTD-P-S or BTD-P-T as an acceptor and P3HT as a donor at different weight ratios. For both P3HT:BTD-P-S and P3HT:BTD-P-T BHJs, the weight ratio of 6 : 1 produced the highest power conversion efficiency (PCE) of 0.17% and 0.44%, respectively. These results are consistent with fluorescence quenching experiments in which 90% of P3HT fluorescence was quenched at that ratio. Nearly two times higher PCE was observed for the BTD-P-T based device compared to that of the BTD-P-S based system, mainly due to the higher J_{sc} . Presumably, the flat geometry of BTD-P-T allows for more efficient intermolecular π -orbital overlap, enhancing charge transport.

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Introduction

Developing organic semiconductors that have tailored HOMO and LUMO energy levels (E_{HOMO} and E_{LUMO}) as well as a controllable solid-state molecular arrangement has been an important field of research.^{1–5} Understanding how structural

constituents impact the electronic properties of whole molecules is an essential step for controlling and ultimately manipulating E_{HOMO} and/or E_{LUMO} . Current practices for designing organic semiconductors utilize structural entities with known electronic properties (either electron accepting or donating) as molecular building blocks. The typical design strategy is to combine building blocks with specific electronic properties in order to obtain a molecular combination that produces the desired electronic result. Numerous studies are available describing the role of certain structural units on electronic properties, making the design process more convenient and potentially predictable.^{6–13} However, consideration of individual subunit (building block) properties does not always produce the anticipated result in a larger system comprised of these individual units since overall molecular electronic properties do not always follow a simple additivity rule. This is more

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significant when the materials are composed of non-identical structural subunits, *e.g.*, donor (D)–acceptor (A) type molecules.^{11–19} To predict and control the electronic properties for molecules before embarking on time-consuming synthesis, it is imperative to understand and assess how each smaller structural constituent contributes to these properties in a systematic manner. To that end, theoretical calculations can serve as a valuable tool.^{20–23} Although quantum mechanical theory has become a useful tool for understanding organic π -conjugated materials, the role of computation has mostly been employed as an “after-the-fact” analysis as opposed to predicting electronic properties before synthesizing a molecule. As one would expect, theoretically calculating molecular electronic properties will not produce numerical agreement with results obtained from experiment. This is particularly problematic when the sample set is small. In our work, we have obtained and correlated experimental and theoretical data for a wide range of related molecular systems using density functional theory and a modest basis set (B3LYP/6-31G*). Thus, we have established a large data set of related molecules that exhibit significantly high levels of correlation specific to electronic properties. Thus, our prediction of electronic properties for new, related systems is possible, and includes an ability to anticipate error. Specifically, our research group has achieved outstanding levels of correlation when treating substituted pyrazine–acene molecules. This previous work has given us a powerful tool for predicting the electronic properties of pyrazine–acene type molecules. In this study, our previous theoretical expertise handling pyrazine–acene derivatives has guided our current materials development and aided in identifying systems worthy of experimental preparation.

In this paper, we developed two self-assembling electron-deficient organic semiconductors guided by theory. The title molecules have an A–A'–A configuration using benzothiadiazole (BTD) with two alkoxyphenazines. We chose BTD as a pendant (A') due to its electron-deficiency^{4,24–26} and structural simplicity, as any bulky pendant moiety may disrupt molecular packing. Alkoxyphenazine (A) was chosen for its self-assembly property, with an ability to form fibrous structures through nonbonding interactions such as hydrogen bonding, π – π , and van der Waals interactions.^{27–29} Due to the presence of imine nitrogens, phenazine is considered an electron-deficient structural subunit.³⁰ We also considered assembly properties since the molecular arrangement of organic semiconductors plays an important role in the solid-state properties. Both π -cores and alkyl side groups contribute to the assembly, and harmonious interplay between intermolecular forces is required to achieve well-defined molecular assembly.^{31–35} However, the assembly is very sensitive to many parameters including the type and location of alkyl side groups, therefore it is hard to predict the resulting morphology. To eliminate the uncertainty of structure-dependent molecular assembly, a reasonable strategy would be to start with structural components that have known assembly patterns.

Two molecules were designed and prepared with different connectivity between A and A', namely, C–C single (**BTD-P-S**) and C–C triple bonds (**BTD-P-T**). During the design process,

theoretical analysis was conducted for each intermediate, from the dialkoxyphenazine to the title molecules. Our goal was to assess the impact of these subunits on both electronic properties as well as structural features. In addition, experimental characterization for the two title compounds included photo-physical, electronic, thermal, and assembly properties. Furthermore, we investigated the efficacy of their function as an acceptor in BHJ OSCs to reveal the effect of small structural variations (connectivity). Detailed device characterizations are also presented.

Results and discussion

Theory assisted molecular design: understanding the role of each structural unit

Starting from the dialkoxyphenazine (**1**) as a base structure, the whole molecular build-up process to reach A–A'–A was calculated to assess the impact of each structural unit on the electronic properties (Fig. 1). First, when A' (BTD) was attached to A (dialkoxyphenazine) directly to create **2**, the major effect was a lowering of E_{LUMO} by 0.43 eV.

This is an expected result when one considers the electron affinity of BTD. In **1** and **2**, a negligible change was found for E_{HOMO} . Further addition of phenazine to produce the first A–A'–A molecule, **BTD-P-S**, influenced E_{HOMO} more than E_{LUMO} . E_{HOMO} was raised by 0.19 eV while E_{LUMO} was lowered by 0.09 eV. The same operation was carried out for **BTD-P-T** which connects phenazine and BTD through a C–C triple bond. Although structurally small, the influence of the triple bond was noticeable in terms of the electronic properties. The addition of a triple bond to **1** reduced E_{LUMO} by 0.19 eV. Interestingly,

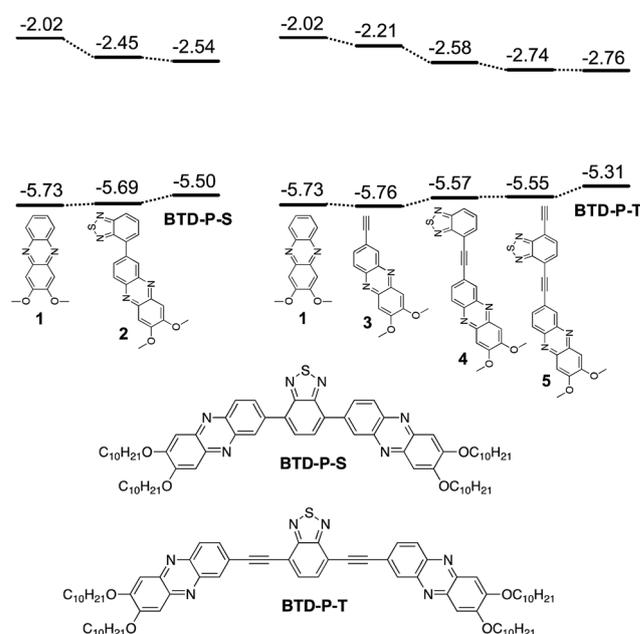


Fig. 1 The structures of BTD-P-S and BTD-P-T and their respective build-up intermediates with theoretical HOMO and LUMO energy levels. Theoretical E_{HOMO} and E_{LUMO} for BTD are -6.61 and -2.35 eV, respectively.



E_{HOMO} was negligibly affected. When BTD was added to 3 to form 4, E_{HOMO} was increased by 0.09 eV and E_{LUMO} was decreased far more significantly (-0.37 eV). Further addition of C–C triple bond to 4 produced the similar result as the process from 1 to 3, in which only E_{LUMO} was affected. Finally, dialkoxypyridazine was added to produce **BTD-P-T**. Surprisingly, E_{LUMO} remained almost the same, while E_{HOMO} was raised by 0.24 eV. Considering the electron deficiency of phenazine, E_{LUMO} was expected to be reduced, however, the computational result indicated otherwise. The changes in the electronic properties upon the structure build-up are presented in Table 1.

The comparative impacts of structural entities during the construction of two BTD-P molecules can be summarized as follows: (1) the addition of BTD lowered E_{LUMO} at a similar level, but the impact on E_{HOMO} was not consistent, (2) C–C triple bond reduced E_{LUMO} , and (3) the addition of phenazine at the final step raised E_{HOMO} while negligibly impacting E_{LUMO} .

To elucidate the origin of these results, two parameters were examined: frontier molecular orbital distributions and the dihedral angles between A and A'. A visual inspection of the HOMO and LUMO orbital distribution can give valuable insight into the subunits that impact the electronic properties. The orbital pictures of intermediates are shown in Fig. 2 for **BTD-P-S** and Fig. 3 for **BTD-P-T**. The most profound feature common to both compounds is the LUMO diagram for those systems where BTD is present. For example, the orbital coefficients are more localized on BTD for 2 and 4 when compared to 1 and 3. This accounts for the large E_{LUMO} lowering when BTD is added. The addition of phenazine to 2 and 5 that lead to **BTD-P-S** and **-T**, respectively, did not alter the LUMO pictures significantly in spite of the electron-deficiency of phenazine, which resulted in only small change in E_{LUMO} . Lower E_{LUMO} for **BTD-P-T** compared to **BTD-P-S** is due to the inclusion of the C–C triple bond which has sp-hybridized carbons. Increased E_{HOMO} when

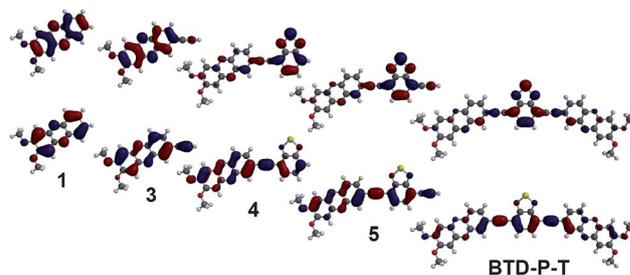


Fig. 3 HOMO (bottom row) and LUMO (top row) diagrams of build-up intermediates for **BTD-P-T**.

BTB is present is due to the delocalization of the HOMO orbital on the benzene in BTB. However, the degree of E_{HOMO} raising is more significant for the **BTD-P-T** case. To investigate this phenomenon, we examined the dihedral angle between A and A'.

The B3LYP/6-31G* optimized geometries resulted in a 36.40° dihedral angle between A (phenazine) and A' (BTD) for **BTD-P-S** while **BTD-P-T** was found to be completely planar (Fig. 4). The relatively large dihedral angle of **BTD-P-S** is due to the steric repulsion between two hydrogen atoms on BTB and phenazine. On the contrary, separation of the two hydrogen atoms by a C–C triple bond allows **BTD-P-T** to adapt a flat geometry. Such planarity enables more efficient orbital delocalization, which accounts for the higher E_{HOMO} .

In summary, **BTD-P-T** is superior to **BTD-P-S** with lower E_{LUMO} , more reduced HOMO–LUMO gap (E_{gap}), and potentially better self-assembly driven by molecular planarity that supports more efficient π – π interactions.

The unique finding from our bottom-up theoretical molecular build-up process is the discovery of individual structural component's impact on the electronic properties that defy prediction using a simple additivity rule. In fact, it would be difficult to elucidate the electronic contributions from the individual subunits by a single computation of the whole molecule only.

Synthesis, optical, electrochemical, thermal, and self-assembly properties

To verify theoretical findings experimentally, the two molecules (**BTD-P-S** and **-T**) were synthesized. The key reaction to prepare **BTD-P-S** was a Suzuki coupling between 2,3-bis(decyloxy)-7-iodophenazine and 2,1,3-benzothiadiazole-4,7-bis(boronic acid pinacol ester). Meanwhile, a Sonogashira

Table 1 The change in E_{HOMO} and E_{LUMO} during structure build-up from phenazine to BTD-P molecules

	E_{HOMO} (eV)	E_{LUMO} (eV)
1 to 2	+0.04	–0.43
2 to BTD-P-S	+0.19	–0.09
1 to 3	+0.03	–0.19
3 to 4	+0.19	–0.37
4 to 5	+0.02	–0.16
5 to BTD-P-T	+0.24	–0.02

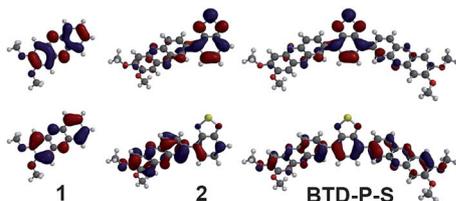


Fig. 2 HOMO (bottom row) and LUMO (top row) diagrams of build-up intermediates for **BTD-P-S**.

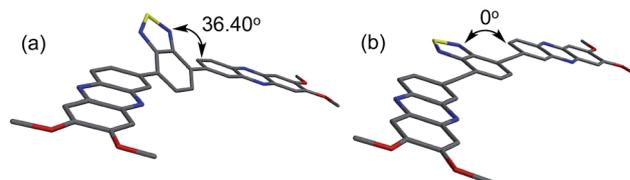


Fig. 4 Optimized geometry of **BTD-P-S** (a) and **BTD-P-T** (b) calculated by B3LYP/6-31G*.



coupling of 2,3-bis(decyloxy)-7-iodophenazine with 4,7-diethynyl-2,1,3-benzothiadiazole produced **BTD-P-T**. 2,3-Bis(decyloxy)-7-iodophenazine was prepared from a cyclization reaction between 4-iodobenzen-1,2-diamine and 2,5-dihydroxybenzoquinone followed by a Williamson ether synthesis with bromodecane. The yields for both of the final C–C coupling reactions were *ca.* 40%. More detailed synthetic procedures with full structural characterizations are provided in ESI.†

The optical properties of these title compounds were characterized by UV-vis and FL spectroscopy. As shown in Fig. 5, **BTD-P-S** showed a broad absorption caused by a π - π^* transition with λ_{max} at 426 nm in dichloromethane solution (5×10^{-6} M). In the case of **BTD-P-T**, a more structured absorption pattern appeared with a red-shifted λ_{max} at 446 nm which is accompanied by shoulders at *ca.* 470 nm and 420 nm. From the absorption edge, the optical E_{gap} was calculated. E_{gap} of **BTD-P-T** was found to be 2.51 eV, which is more compressed than that of **BTD-P-S** by *ca.* 0.1 eV. In the solid-state, the two compounds showed distinctively different absorption behaviors. The λ_{max} of the **BTD-P-S** cast film was found to be the same as in solution, however the absorption became broader with tailing at longer wavelengths. On the contrary, the **BTD-P-T** cast film showed a blue-shifted λ_{max} at 421 nm relative to the solution λ_{max} and red-shifted shoulder at *ca.* 485 nm. A similar behaviour has been assigned to pseudo H-aggregates.³⁶ E_{gap} 's in the solid-state showed the same trend as those in solution; though more reduced for **BTD-P-T**. Meanwhile, the fluorescence maxima (λ_{em}) for **BTD-P-S** and **BTD-P-T** were nearly the same: 505 nm and 508 nm in dichloromethane (5×10^{-6} M), when excited at 430 nm and 446 nm, respectively (Fig. 6). The fluorescence quantum efficiencies for both compounds were quite high, reaching 58% for **BTD-P-S** and 66% for **BTD-P-T** which were measured using 9,10-diphenylanthracene as a standard reference ($\Phi_{\text{F}} = 90\%$ in cyclohexane).

The electrochemical properties of the title compounds were investigated with cyclic voltammetry (CV) (Fig. 7). Our primary

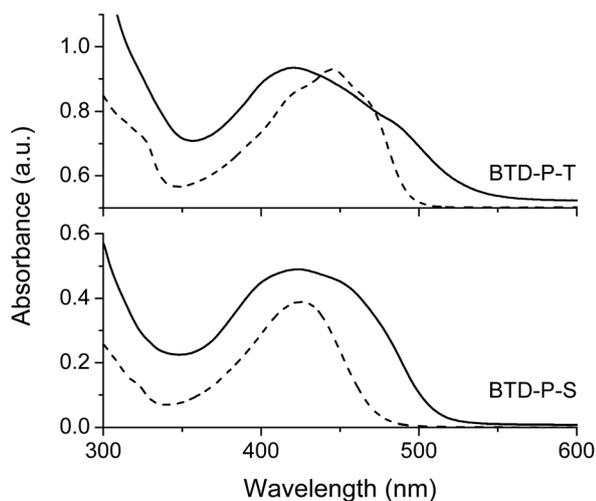


Fig. 5 UV-vis absorption spectra of **BTD-P-S** and **-T** in dichloromethane (5×10^{-6} M) (broken line) and as a film (solid line).

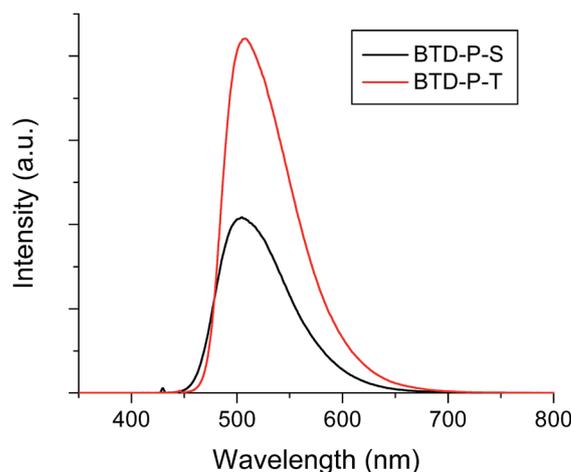


Fig. 6 Fluorescence spectra of **BTD-P-S** and **BTD-P-T** in dichloromethane (5×10^{-6} M). $\lambda_{\text{ex}} = 430$ nm for **BTD-P-S** and 446 nm for **BTD-P-T**.

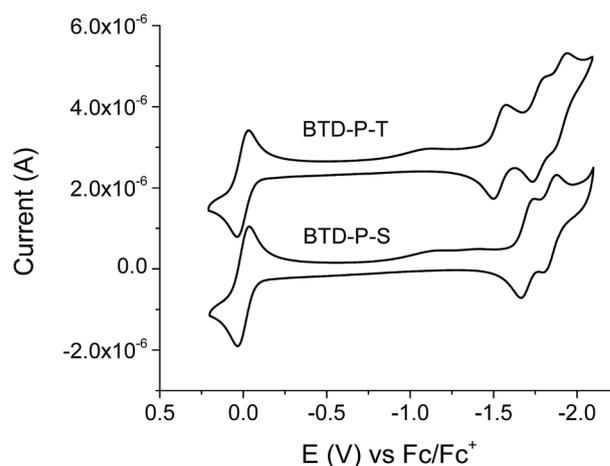


Fig. 7 Cyclic voltammograms of **BTD-P-S** and **BTD-P-T** in dichloromethane with tetrabutylammonium hexafluorophosphate (TBAPF_6) as a supporting electrolyte. Scan rate = 100 mV s^{-1} .

focus was on the reduction potential, which is directly related to the electron affinity. Both compounds showed one reversible reduction potential within the characterized potential window. From the onset of the reduction potential, we deduced the lowest unoccupied molecular orbital energy level (E_{LUMO}).

The CV experiments were conducted using ferrocene as an internal reference, therefore E_{LUMO} was calculated relative to the oxidation potential of ferrocene to ferrocenium, which is -4.8 eV from vacuum. Due to the electron-deficient nature of the title compounds, it was difficult to observe the oxidation potential, and therefore, E_{HOMO} was calculated from E_{LUMO} (from CV) and optical E_{gap} (from UV-vis). These values are listed in Table 2. In solution, E_{LUMO} of **BTD-P-T** was found to be -3.33 eV, which is more stabilized than that of **BTD-P-S** by 0.16 eV. In the case of E_{HOMO} , the difference was marginal.

A differential scanning calorimetry (DSC) was employed to characterize thermal properties (Fig. 8).



Table 2 Electronic properties of BT-D-P compounds

	E_{LUMO}^a	E_{HOMO}^b	E_{gap}^c
BT-D-P-S	-3.17	-5.80	2.63
BT-D-P-T	-3.33	-5.84	2.51

^a $E_{\text{LUMO}} = -(E_{\text{red}}^{\text{onset}} + 4.8 \text{ eV})$. ^b $E_{\text{HOMO}} = E_{\text{LUMO}} - E_{\text{g}}^{\text{optical}}$. ^c Optical HOMO-LUMO gap.

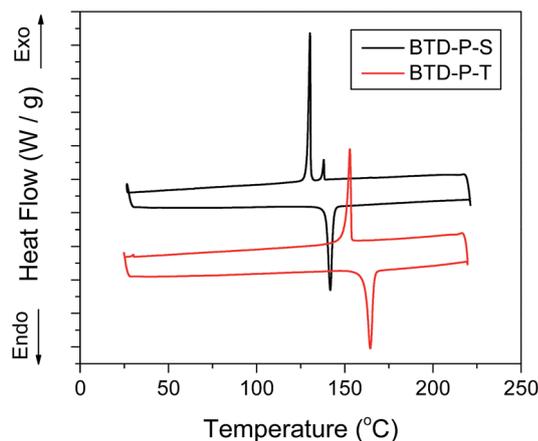


Fig. 8 DSC thermograms of title compounds. 2nd heating/cooling scans at 5 °C min⁻¹.

Both compounds showed reversible melting and crystallization behaviors. A sharp melting transition appeared at 142 °C for **BT-D-P-S** and 164 °C for **BT-D-P-T**. The heat of fusion of **BT-D-P-T** (33.3 J g⁻¹) was clearly higher than that of **BT-D-P-S** (26.4 J g⁻¹), indicating higher crystallinity of **BT-D-P-T**. While **BT-D-P-T** revealed only one crystallization transition at 153 °C, **BT-D-P-S** had a small exothermic peak at 138 °C in addition to the major crystallization peak at 130 °C. This appeared repeatedly only at the cooling scan, which suggests monotropic liquid crystalline behavior.

To examine the molecular packing, XRD experiments were conducted (Fig. 9). The XRD samples were prepared by drop-

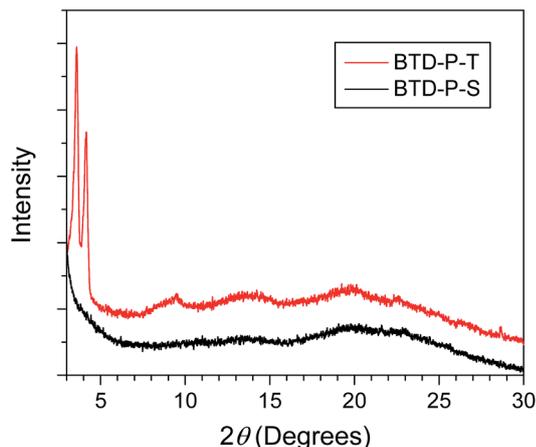


Fig. 9 X-ray powder diffraction patterns of BT-D-phenazine compounds.

casting chlorobenzene solutions (1 mg mL⁻¹) on a Si wafer in a glove box. These samples were annealed at 160 °C for 10 min. The annealing process was included to mimic the OSC device preparation which will be discussed later. The most distinctive difference in the diffraction pattern was found at 2θ below 5°. **BT-D-P-T** showed sharp primary diffractions at $2\theta = 3.58^\circ$ and 4.16° , which corresponds to d -spacings of 24.84 Å and 21.39 Å, respectively, in addition to broad peaks at higher angles. It should be noted that the theoretically calculated π -core length excluding alkyl side groups is ca. 25 Å (Fig. S7, ESI†) which corresponds to d -spacing from primary diffraction. Absence of strong, higher order peaks of the primary d -spacing indicates that long-range order is not present in the film. In the case of **BT-D-P-S**, the diffraction was quite featureless, indicating less crystalline character.

More significant π - π interactions from the solid-state absorption, higher melting transition with larger heat of fusion, and more crystalline character suggest that **BT-D-P-T** has tighter molecular arrangement than **BT-D-P-S**. Solving the crystal structure of both systems would provide direct verification of such a hypothesis, however single crystals of the title compounds were not available since they both have a tendency to fibrillate. Presumably, the planarity of **BT-D-P-T** as ascertained by computation allows for more compact molecular packing, which explains the observed difference in optical, thermal, and diffraction properties.

Both **BT-D-P-S** and **BT-D-P-T** showed excellent fibrillation capability. As presented in Fig. 10, cast films of the two molecules from dichloromethane solutions (3 mg mL⁻¹) exhibit morphologies which are composed of long fibers. In particular, **BT-D-P-T** formed longer, less branched fibers compared to **BT-D-P-S**. The initial assembly results support our strategy of incorporating dialkoxyphenazine to stimulate 1D assembly. Further detailed study on the self-assembly including organogelation is in progress.

We note that optical, electronic, thermal, and assembly properties of BT-D-P compounds are better understood when coupled with our theoretical findings.

Application to organic solar cell as electron acceptors

We investigated the efficacy of the title compounds as electron acceptors in BHJ OSCs. In recent years, there has been significant advancement in small-molecule-based non-fullerene acceptors for OSCs.³⁷⁻⁴⁵ We note that our objective in this

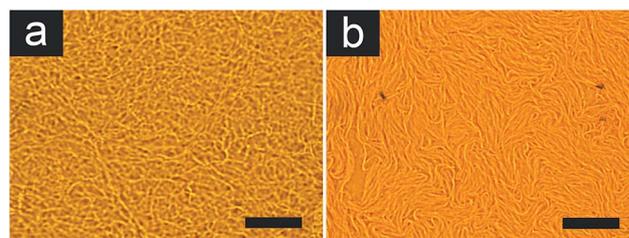


Fig. 10 Optical micrographs of cast films: (a) **BT-D-P-S** and (b) **BT-D-P-T**. Scale bar: 20 μm .



work is to investigate the impact of small structural differences on device performance, rather than to achieve the state-of-art PCE. Before an OSC device was fabricated, fluorescence quenching experiments were conducted to determine the optimum ratio between P3HT and BTDP-P compounds. An optimum ratio of these two compounds facilitates maximum electron transfer. The blended films were fabricated *via* spin-coating from a chlorobenzene solution (20 mg mL^{-1}) at 1200 rpm for 40 s. The chlorobenzene solution contained DIO at 3 v/v% to mimic the actual layer of the OSC device that will be fabricated. The films were further annealed at 160°C for 10 min. Interestingly, the fluorescence quenching was efficient even with a small content of BTDP-P acceptors. For both P3HT:BTDP-P-S (Fig. S8, ESI†) and P3HT:BTDP-P-T (Fig. S9, ESI†) blend films, 90% of P3HT FL was quenched at a weight ratio of 6 : 1. Further increase in the BTDP-P content did not facilitate more FL quenching. This result indicates that the electron transfer from P3HT to the acceptor is most efficient when the wt% of acceptor is only *ca.* 14.

To evaluate the performance of the BTDP-P molecules as acceptors in organic solar cell (OSC) devices, inverted bulk heterojunction (BHJ) OSC devices were fabricated with a structure of ITO/ZnO (40 nm)/active layer (P3HT:BTDP-P blend) (100 nm)/MoO₃ (10 nm)/Ag (100 nm). Detailed descriptions on the device fabrication process are provided in ESI. For both P3HT:BTDP-P-S and P3HT:BTDP-P-T, four different donor : acceptor ratios were tested. Representative characteristics of inverted BHJ OSCs such as open circuit voltage (V_{oc}), short circuit current (J_{sc}), fill factor (FF), and power conversion efficiency (PCE) are summarized in Table 3, and current density–voltage (J – V) characteristics of the BHJ OSC devices under AM 1.5G illumination at 100 mW cm^{-2} are shown in Fig. 11. In both cases, the donor : acceptor ratio of 6 : 1 gave the highest PCE. The device based on P3HT:BTDP-P-S BHJ showed a V_{oc} of 0.49 V, a J_{sc} of 0.57 mA cm^{-2} , a FF of 60.38% and a PCE of 0.17%. On the other hand, the device based on P3HT:BTDP-P-T BHJ exhibited highly improved photovoltaic performance with a V_{oc} of 0.63 V, a J_{sc} of 1.18 mA cm^{-2} , a FF of 58.53% and a PCE of 0.44%. Compared to the devices with P3HT:BTDP-P-S BHJ, the improvement in the photovoltaic performance of the devices with P3HT:BTDP-P-T BHJ mainly resulted from an improved J_{sc} value.

Surprisingly, increased acceptor content from 14% (6 : 1) to 50% (1 : 1) decreased J_{sc} by *ca.* 70% for both BHJs. Meanwhile,

Table 3 OSC device parameters of BHJ based on P3HT donor and BTDP-P acceptors

Acceptor	Ratio ^a	V_{oc} (V)	J_{sc} (mA cm^{-2})	FF (%)	PCE (%)
BTDP-P-S	1 : 1	0.33	0.17	47.67	0.03
	2 : 1	0.41	0.35	55.06	0.08
	4 : 1	0.48	0.50	57.47	0.14
	6 : 1	0.49	0.57	60.38	0.17
BTDP-P-T	1 : 1	0.56	0.41	42.92	0.10
	2 : 1	0.61	0.83	50.04	0.26
	4 : 1	0.62	0.73	57.81	0.25
	6 : 1	0.63	1.18	58.53	0.44

^a P3HT : acceptor (weight ratio).

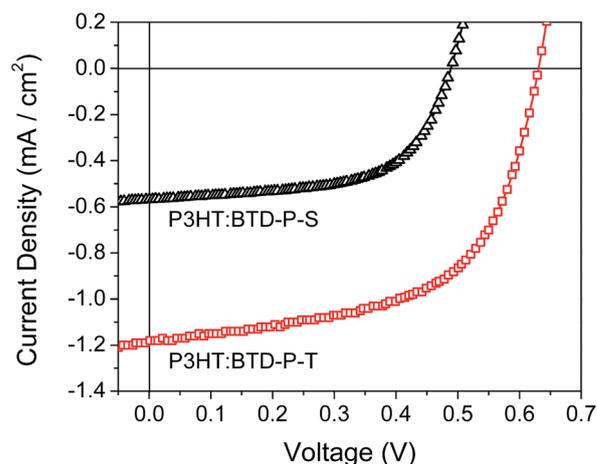


Fig. 11 J – V curves of P3HT : BTDP-P BHJs (6 : 1 wt ratio for both devices) under AM 1.5G condition.

V_{oc} decreased by *ca.* 30% for P3HT : BTDP-P-S and *ca.* 10% for P3HT : BTDP-P-T when the ratio changed from 6 : 1 to 1 : 1. Lower PCE at higher acceptor content may be due to the increased phase separation and poor light harvesting as P3HT is the major contributor for the light absorption.

To further confirm the accuracy of the measurements, the external quantum efficiency (EQE) of P3HT:BTDP-P-T BHJ OSC devices at different ratios was measured as shown in Fig. 12. All the tested donor : acceptor ratios exhibited a maximum EQE at *ca.* 500 nm, which is consistent with the absorption maximum of P3HT (Fig. S10, ESI†). The highest EQE was found at the P3HT : BTDP-P-T ratio of 6 : 1, which gave the highest PCE. The morphology of the active layers (P3HT:BTDP-P-T) were investigated by atomic force microscopy (AFM). The AFM images with donor : acceptor ratios of 1 : 1 and 6 : 1 did not exhibit any drastic morphological differences (Fig. S11, ESI†). Both blends exhibited smooth surfaces with root mean square roughness of 2.8 nm for 1 : 1 blend and 3.5 nm for 6 : 1 blend. Due to the low content of the acceptor, it would be difficult to observe fibrous textures seen in the cast films.

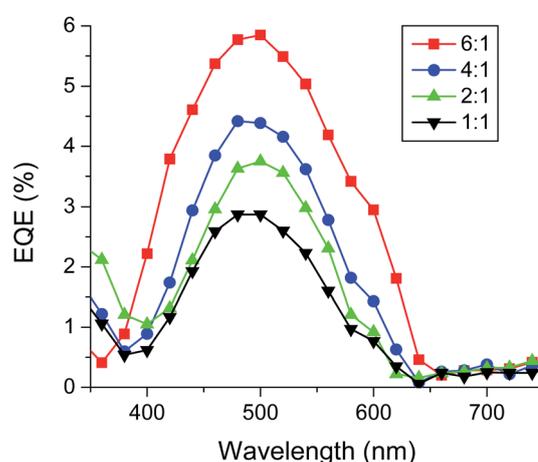


Fig. 12 EQE spectra of P3HT:BTDP-P-T based BHJ.



Conclusions

In this work, we demonstrated how structural entities impact the electronic properties of two A-A'-A systems by analyzing each structural intermediate theoretically during the design process. The most profound effect on E_{LUMO} was found when more electron-deficient BTD was added to the less electron-deficient phenazine. The connectivity between A and A' affected the dihedral angle significantly which influenced E_{HOMO} since the HOMO orbital was distributed along the whole molecule while the LUMO orbital was localized more on the electron-deficient component. Both **BTD-P-T** and **BTD-P-S** were successfully synthesized and characterized. From optical, electrochemical, and thermal property characterizations, **BTD-P-T** showed lower E_{gap} and E_{LUMO} , and higher T_{m} with larger heat of fusion. X-ray powder diffraction also revealed higher crystallinity for **BTD-P-T**. Both **BTD-P-S** and **BTD-P-T** showed excellent 1D assembling properties by producing fibers upon solvent casting. Theoretical calculations revealed a planar conformation for **BTD-P-T**, which accounts for more enhanced intermolecular π - π interactions. Even with moderate electron deficiency compared to fullerene, the title compounds functioned successfully as an electron-acceptor for bulk heterojunction OSCs with P3HT as a donor. For both P3HT : **BTD-P-S** and P3HT : **BTD-P-T** BHJs, the weight ratio of 6 : 1 produced the highest PCE of 0.17% and 0.44%, respectively. The higher J_{sc} of the **BTD-P-T** device was a major contribution for the PCE, and nearly double that of **BTD-P-S**. We hypothesize that the flat geometry of **BTD-P-T** allows for more efficient intermolecular π -orbital overlap, thus enhancing charge transport. The theory guided molecular building process we present in this work should make the experimental design process more effective and efficient by screening and optimizing molecular architectures before embarking on lengthy synthetic procedures.

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Notes and references

- 1 J. Roncali, *Chem. Rev.*, 1997, **97**, 173.
- 2 C. R. Newman, D. D. Frisbie, D. A. da Silva Filho, J.-L. Brédas, P. C. Ewbank and K. R. Mann, *Chem. Mater.*, 2004, **16**, 4436.
- 3 H. Usta, A. Facchetti and T. J. Marks, *Acc. Chem. Res.*, 2011, **44**, 501.
- 4 J. E. Anthony, *Chem. Mater.*, 2011, **23**, 583.
- 5 P. M. Beaujuge and J. M. J. Fréchet, *J. Am. Chem. Soc.*, 2011, **133**, 20009.
- 6 A. P. Kulkarni, C. J. Tonzola, A. Babel and S. A. Jenekhe, *Chem. Mater.*, 2004, **16**, 4556.
- 7 J. E. Anthony, A. Facchetti, M. Heeney, S. R. Marder and X. Zhan, *Adv. Mater.*, 2010, **22**, 3876.
- 8 P. Sonar, J. P. F. Lim and K. L. Chan, *Energy Environ. Sci.*, 2011, **4**, 1558.
- 9 Y. Lin, Y. Li and X. Zhan, *Chem. Soc. Rev.*, 2012, **41**, 4245.
- 10 A. Facchetti, *Mater. Today*, 2013, **16**, 123.
- 11 H. Zhou, L. Yang and W. You, *Macromolecules*, 2012, **45**, 607.
- 12 T. Xu and L. Yu, *Mater. Today*, 2014, **17**, 11.
- 13 L. Lu and L. Yu, *Adv. Mater.*, 2014, **26**, 4413.
- 14 H. A. M. van Mullekom, J. A. J. M. Vekemans, E. E. Havinga and E. W. Meijer, *Mater. Sci. Eng., R*, 2001, **32**, 1.
- 15 A. Ajayaghosh, *Acc. Chem. Res.*, 2005, **38**, 449.
- 16 Y.-J. Cheng, S.-H. Yang and C.-S. Hsu, *Chem. Rev.*, 2009, **109**, 5868.
- 17 J. Peet, A. J. Heeger and G. C. Bazan, *Acc. Chem. Res.*, 2009, **42**, 1700.
- 18 O. Inganäs, F. Zhang and M. R. Andersson, *Acc. Chem. Res.*, 2009, **42**, 1731.
- 19 S. C. Rasmussen, R. L. Schwiderski and M. E. Mulholland, *Chem. Commun.*, 2011, **47**, 11394.
- 20 L. Pandey, C. Risko, J. E. Norton and J.-L. Brédas, *Macromolecules*, 2012, **45**, 6405.
- 21 D.-C. Lee, K. Jang, K. K. McGrath, R. Uy, K. A. Robins and D. W. Hatchett, *Chem. Mater.*, 2008, **20**, 3688.
- 22 L. V. Brownell, K. Jang, K. A. Robins, I. C. Tran, C. Heske and D.-C. Lee, *Phys. Chem. Chem. Phys.*, 2013, **15**, 5967.
- 23 L. V. Brownell, K. A. Robins, Y. Jeong, Y. Lee and D.-C. Lee, *J. Phys. Chem. C*, 2013, **117**, 25236.
- 24 Y. Cao, T. Lei, J. Yuan, J.-Y. Wang and J. Pei, *Polym. Chem.*, 2013, **4**, 5228.
- 25 D. Mori, H. Benten, H. Ohkita, S. Ito and K. Miyake, *ACS Appl. Mater. Interfaces*, 2012, **4**, 3325.
- 26 C. Guo, Y.-H. Lin, M. D. Witman, K. A. Smith, C. Wang, A. Hexemer, J. Strzalka, E. D. Gomez and R. Verduzco, *Nano Lett.*, 2013, **13**, 2957-2963.
- 27 J.-L. Pozzo, G. M. Clavier and J.-P. Desvergne, *J. Mater. Chem.*, 1998, **8**, 2575.
- 28 D.-C. Lee, B. Cao, K. Jang and P. M. Forster, *J. Mater. Chem.*, 2010, **20**, 867.
- 29 K. Jang, L. V. Brownell, P. M. Forster and D.-C. Lee, *Langmuir*, 2011, **27**, 14615.
- 30 T. Yamamoto, T. Maruyama, Z. Zhou, T. Ito, T. Fukuda, Y. Yoneda, F. Begamn, T. Ikeda, S. Sasaki, H. Takezoe, A. Fukuda and K. Kubota, *J. Am. Chem. Soc.*, 1994, **116**, 4832.
- 31 F. J. M. Hoeben, P. Jonkheijm, E. W. Meijer and A. P. H. J. Schenning, *Chem. Rev.*, 2005, **105**, 1491.
- 32 A. Ajayaghosh and V. K. Praveen, *Acc. Chem. Res.*, 2007, **40**, 644.
- 33 L. Zang, Y. Che and J. S. Moore, *Acc. Chem. Res.*, 2008, **41**, 1596.
- 34 L. C. Palmer and S. I. Stupp, *Acc. Chem. Res.*, 2008, **41**, 1674.
- 35 S. S. Babu, V. K. Praveen and A. Ajayaghosh, *Chem. Rev.*, 2014, **114**, 1973.



- 36 A. Ajayaghosh, C. Vijayakumar, R. Varghese and S. J. George, *Angew. Chem., Int. Ed.*, 2006, **45**, 456.
- 37 H. Li, T. Earmme, G. Ren, A. Saeki, S. Yoshikawa, N. M. Murari, S. Subramaniyan, M. J. Crane, S. Seki and S. A. Jenekhe, *J. Am. Chem. Soc.*, 2014, **136**, 14589.
- 38 Z. Mao, W. Senevirathna, J.-Y. Liao, J. Gu, S. V. Kesava, C. Guo, E. D. Gomez and G. Sauvé, *Adv. Mater.*, 2014, **26**, 6290.
- 39 H. Li, Y.-J. Hwang, B. A. E. Courtright, F. N. Eberle, S. Subramaniyan and S. A. Jenekhe, *Adv. Mater.*, 2015, **27**, 3266.
- 40 Y. Lin, Z.-G. Zhang, H. Bai, J. Wang, Y. Yao, Y. Li, D. Zhu and X. Zhan, *Energy Environ. Sci.*, 2015, **8**, 610.
- 41 D. Sun, D. Meng, Y. Cai, B. Fan, Y. Li, W. Jiang, L. Huo, Y. Sun and Z. Wang, *J. Am. Chem. Soc.*, 2015, **137**, 11156.
- 42 Y.-J. Hwang, H. Li, B. A. E. Courtright, S. Subramaniyan and S. A. Jenekhe, *Adv. Mater.*, 2016, **28**, 124.
- 43 Y. Lin, Q. He, F. Zhao, L. Huo, J. Mai, X. Lu, C.-J. Su, T. Li, J. Wang, J. Zhu, Y. Sun, C. Wang and X. Zhan, *J. Am. Chem. Soc.*, 2016, **138**, 2973.
- 44 Y. Lin, F. Zhao, Q. He, L. Huo, Y. Wu, T. C. Parker, W. Ma, Y. Sun, C. Wang, D. Zhu, A. J. Heeger, S. R. Marder and X. Zhan, *J. Am. Chem. Soc.*, 2016, **138**, 4955.
- 45 S. Dai, F. Zhao, Q. Zhang, T.-K. Lau, T. Li, K. Liu, Q. Ling, C. Wang, X. Lu, W. You and X. Zhan, *J. Am. Chem. Soc.*, 2017, **139**, 1336.

