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High performance anion exchange ionomer for anion exchange membrane fuel cells

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The anion exchange ionomer incorporated into the electrodes of an anion exchange membrane fuel cell (AEMFC) enhances anion transport in the catalyst layer of the electrode, and thus improves performance and durability of the AEMFC. In this work, a novel ionomer based on a triblock copolymer with high conductivity and good durability is synthesized successfully. The spectroscopy (such as ¹H-NMR, FT-IR) results of the ionomer indicate that the functional group is grafted onto the poly(styrene-ethylene/butylene-styrene) (SEBS) successfully and the OH⁻ conductivity of the ionomer is 30 mS cm⁻¹ at 75 °C. Besides, quaternary ammonium SEBS (QASEBS) is used as the ionomer in a H₂/O₂ AEMFC and exhibits a significant durability of 500 h at a constant current density of 100 mA cm⁻², moreover, the degradation rate of voltage is only 0.22 mV h⁻¹ during the 500 h durability test. In addition, the peak power density of the membrane electrode assembly (MEA) with the QASEBS ionomer reaches 375 mW cm⁻² at 50 °C, which is 3 times than that of the MEA using the commercially available Acta I2 ionomer (124 mW cm⁻²) for comparison.

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1. Introduction

Recently, to deal with the energy crisis and environment issues, fuel cell technology has drawn more academic and industrial interest. Among all fuel cell technology, proton exchange membrane fuel cells (PEMFCs) have been recognized as promising energy source systems because of their high power density, high energy conversion efficiency, and low pollutant emission.¹⁻³ Even though PEMFCs still have a strong dependence on platinum group metal catalysts, thus increasing the cost and hindering the large scale commercialization of fuel cells. A promising solution to the problem is to develop anion exchange membrane fuel cells (AEMFCs), which adjust the conditions of electrochemical reaction.^{4,5} On one hand, non-precious metal or compounds can be used as stable and efficient catalysts under high pH for electrode reactions.^{6,7} On the other hand, the electrolyte leakage issue that occurs in the traditional alkaline fuel cells has been mitigated due to the use of a solid alkaline electrolyte membrane.^{8,9}

AEMFCs have been a new focus in the field of fuel cell research over the past decades. The tremendous efforts have been put into the development of alkaline anion exchange membrane (AAEM)^{1,10-12} and the catalyst materials,¹³⁻¹⁵ and the research on the AAEM have acquired some encouraging results

that the conductivity of the AAEM has been more than 100 mS cm⁻¹,¹⁶ which is comparable to the commercial Nafion® membrane used in PEMFC. However, the performance of the AEMFC has not yet satisfied the industrial and commercial application, one of the most significant problems for AEMFC is a lack of a soluble ionomer in the catalyst layer to build an efficient three-phase boundary and thus dramatically increase the utilization of the catalyst particles and decrease the internal resistance.

The ionomer serves as the physical binder and ion conductor transporting the OH⁻ ion in the catalyst layer, playing a key role in the development of the fuel cell performance.¹⁷ Thus, to the ionomer, the high ion conductivity and alkaline stability are needed. Besides, high solubility in low boiling point water soluble solvents is also one of the most desirable properties for an ionomer applying in the catalyst layer, because these solvents are easy and safe to handle and remove during the electrode preparation. Recently, some ionomers from the industrial and academia have been reported,¹⁸⁻³² however neither the formulation of the ionomers nor their functioning mechanisms were disclosed. The polysulfone (PSf) pendent quaternary ammonium ionomer in recent reports can only dissolve in high boiling point solvent, such as dimethylformamide (DMF) and dimethylacetamide (DMAc),^{1,5,33} which is considered difficult to remove the solvents in the presence of finely dispersed catalysts. Yan *et al.* prepared a quaternary phosphonium functionalized PSf ionomer soluble in low boiling point solvents, which exhibited exceptional high hydroxide conductivity and good alkaline stability. And a peak power density of 138 mW cm⁻² in a H₂/O₂ AEMFC was

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achieved at 50 °C, which was 3.5 times higher for the MEA without ionomers.³⁴ Sun and co-workers chose a novel block copolymer SEBS as the backbone which was a thermoplastic elastomer with good chemical, thermal and mechanical stability and synthesized quaternary ammonium based ionomers with different ion exchange capacity. A peak power output of 210 mW cm⁻² in a H₂/O₂ AEMFC was reached at 50 °C with 2 atm back pressure, which was much higher than that (<50 mW cm⁻²) of an MEA with PSf-based ionomers.²³ Although these ionomers exhibited an acceptable ion conductivity and solubility, the performance of the MEA with these ionomers was not comparable to state-of-the-art PEMFCs.^{24,35,36} Besides, the literatures about the AEMFC performance, especially durability of the AEMFC, are few in the public publications.

Although the SEBS was chosen to prepare the anion exchange membrane and anion exchange ionomer, the performance of AEMFC was still very low due to the bad solubility of the ionomer which lead to bad MEA fabrication. In this work, chloromethyl groups were grafted onto the SEBS, and the non-ionic CMSEBS had good solubility in tetrahydrofuran. The home-made AEH9620 (ref. 37) and CMSEBS were chosen as the AAEM and ionomer binder to prepare the electrode, and then the electrode was immersed in trimethylamine solution for amination and soaked in KOH solution for alkalization. Using the post-ammoniated optimized fabrication approach of the MEA, a higher peak power output and longer lifetime are realized in the H₂/O₂ AEMFC.

2. Experimental

2.1. Preparation of the QASEBS ionomer

The Tuftec H1031 SEBS with 30 wt% of styrene structural units as the raw material was purchased by AsahiKASE. 1,4-Bis(chloromethoxy)butane (BCMB) was provided by Xi'an Lanjing LLC. Chloroform, tetrahydrofuran, methanol, ethanol, tin(IV) chloride, phenolphthalein, trimethylamine solution (33 wt%), hydrochloric acid, sodium hydroxide and potassium hydroxide were from Tianjing Damao Chemical Works.

Firstly, the SEBS was purified before functionalization using the method mentioned in Sun's report.²³ For chloromethylation, 2 g SEBS was dissolved in 60 mL chloroform in a three-neck flask

equipped with a mechanical and a condenser agitation in an oil bath at 55 °C, then 4.1 mL tin(IV) chloride and 27.6 mL BCMB were added in the reaction system in turn. The solution was stirred for 12 h at 55 °C. The chloromethylated SEBS (CMSEBS) was purified by precipitation from 5 wt% tetrahydrofuran solution to methanol. Then, the CMSEBS was dissolved in tetrahydrofuran to prepare a 5 wt% polymer solution to use. The synthetic process of the quaternary ammoniated SEBS for the ionomer was shown in Scheme 1.

2.2. NMR characterization

The SEBS, CMSEBS and QASEBS were analysed on a Bruker 400 MHz instrument. Due to the QASEBS poor solubility, an indirect method is employed. The experimental details are as follows. Firstly, the chloromethylated polymer CMSEBS was dissolved in the chloroform-d, then the trimethylamine water solution (33 wt%) was added in the CMSEBS solution. After 2 hours, the mixture and the trimethylamine solution (*n*-propanol as the solvent) was analysed by a Bruker 400 MHz instrument, respectively. The chloroform-d was used as the solvent and the tetramethylsilane (TMS) was used as the internal reference.

2.3. FTIR characterization

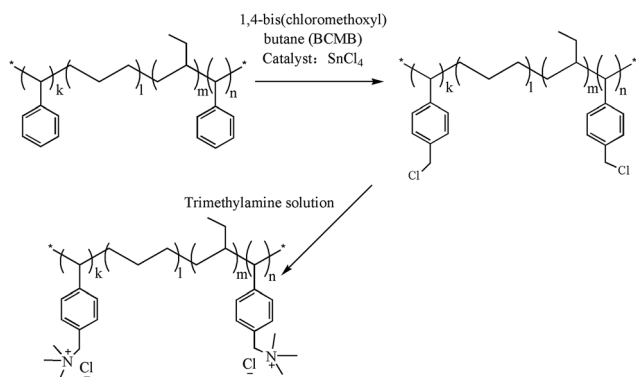
TR-FTIR of the SEBS, CMSEBS and QASEBS membranes was obtained on a Bruker Tensor 27 spectrometer with an ATR accessory containing a Ge crystal, and the wavenumber resolution was 4 cm⁻¹ and the range was 4000–400 cm⁻¹.

2.4. Ion exchange capacity, water uptake and swelling behaviour

The ionomers solution can't be directly measured some properties, so the ion exchange capacity (IEC), swelling ratio and water uptake of the ionomer were characterized in the form of membrane. During the test, the sample were all in hydroxide form. In order to convert the ionomer to hydroxide form, we immersed the ionomer in the 1 M KOH solution for 24 hours, and replaced the solution with the fresh 1 M KOH every 8 hours. Then, the ionomer was washed by DI water (the conductivity is 0.19 μS cm⁻¹ and the water is saturated by the nitrogen before using) until the pH of the filtrate water was seven. At last, the ionomer was put into the DI water and stored for later use. On the other hand, to weaken the effect of the CO₂ during the test of the ionomer in hydroxide form, DI water was saturated by the nitrogen before using. Besides, all the test of the properties (water uptake, swelling, hydroxide conductivity) were all done in a big box full of nitrogen.

The IEC of the ionomers were determined using the back titration method.³⁸ The dry ionomer with OH⁻ form was immersed in 30 mL of 0.01 M HCl solution for 48 h at 30 °C, then the solution was titrated against 0.01 M NaOH solution, and the phenolphthalein was used as indicator. The IEC was calculated as follows:

$$\text{IEC} = \frac{M_{1,\text{HCl}} - M_{2,\text{HCl}}}{m} \quad (1)$$



Scheme 1 The synthetic procedure of the quaternary ammonium SEBS block copolymer in Cl⁻ form.



where $M_{1,\text{HCl}}$ and $M_{2,\text{HCl}}$ are the amount of HCl in mmol before and after titration with NaOH, respectively, and m is the dry mass of the ionomer membrane.

The dimension of the wet and dry samples were measured, and the swelling ratio SR (%) could be calculated by:

$$\text{SR (\%)} = \frac{L_{\text{wet}} - L_{\text{dry}}}{L_{\text{dry}}} \times 100\% \quad (2)$$

where L_{dry} and L_{wet} are the dimension of sample measured in the OH^- form under fully hydrated and dried conditions, respectively. The drying sample was achieved in vacuum at 60 °C for 24 h.

To measure the water uptake of the ionomer, firstly, the ionomer membrane was immersed into deionized water at 30 °C for 24 h. Then the membranes were taken out, and surface attached water was wiped with a tissue paper, and immediately weighed on a microbalance. The dry membrane was obtained by drying in vacuum at 60 °C until constant weight. The water uptake Wu (wt%) could be calculated by:

$$\text{Wu (wt\%)} = \frac{W_{\text{wet}} - W_{\text{dry}}}{W_{\text{dry}}} \times 100\% \quad (3)$$

where W_{dry} and W_{wet} are the weights of the samples before and after hydration, respectively.

2.5. Hydroxide ion conductivity

The conductivity of ionomer membrane was measured following a standard four-point probe technique by an AC impedance spectroscopy using a Solartron 1260 frequency response analyser coupled to a Solartron 1287 potentiostat. The measurement used the potentiostat mode with the frequency range from 1 MHz to 0.1 Hz and the oscillating voltage of 10 mV. Before the test, specimens were soaked in DI water for at least 24 h. Then ionomer membrane was taken out quickly and clamped between two platinum electrodes, and the measurement was in DI water with relative humidity of 100% at different temperature. The ionomer conductivity (σ , S cm^{-1}) could be calculated by the following equation.

$$\sigma = \frac{L}{RS} \quad (4)$$

where L (cm) and S (cm^2) are the length between potential sensing platinum electrode and the membrane cross sectional area, respectively, and R (Ω) is the membrane resistance derived from high frequency region (at zero phase angle).

2.6. MEA fabrication

To characterize the performance of the ionomer in AEMFC, the homemade membrane AEH9620 was chosen as the anion exchange membrane to prepare the MEA in this paper, and its preparation procedure was disclosed in the previous work.³⁷ The key properties of the AEH9620 membrane were listed in Table 1. The commercial ionomer I2 (Acta S.p.A. Co.) and prepared CMSEBS were chosen as the ionomer during the preparing the catalyst ink for both the anode and cathode, respectively. The preparation of catalyst ink as followed: a commercial catalyst Pt/C

Table 1 Basic properties of the AEH9620 membrane at 30 °C (ref. 25)

Thickness (μm) in dry state	IEC (mmol g^{-1})	OH^- conductivity (S cm^{-1})	Mechanical strength (MPa)
22	1.35	0.035	99.7

(70%, JM Co.) was mixed with a certain amount of water, tetrahydrofuran (or propyl alcohol) and CMSEBS solution (or I2 solution), and the ionomer was in an 80 : 20 as catalyst: ionomer weight ratio. The homogenous ink was obtained by sonication, then the ink was brushed onto the two sides of the AEH9620 by thermal spray casting, and the Pt loading in the anode and cathode were both 0.5 mg cm^{-2} . Besides, the MEA using the CMSEBS as the ionomer needed to immerse in trimethylamine solution 24 h for quaternary amination and then to soak in 1 M KOH solution 24 h for alkalization. The MEA was sandwiched between two pieces of gas diffuse layers, and hot-pressed at 60 °C, 1 MPa for 2 min to assemble a single cell with an active electrode area of 5 cm^2 .

2.7. Fuel cell test

A homemade fuel cell test system was used for controlling humidity, the cell temperature, H_2 and O_2 flow rate, and back-pressures. For operation of AEMFC, the temperature of H_2 and O_2 was 50 °C and relative humidity was 100%. The flow rate of H_2 and O_2 was 100, 200 mL min^{-1} at 0.20 MPa, respectively. An electric load system (KMF2030, Kikusui Electronics Corp.) and Solartron 1260 workstation were used to measure and record the i - V curves and ohmic resistance of the AEMFC. The EIS experiments were carried out at a current density of 100 mA cm^{-2} and 200 mA cm^{-2} with a small amplitude alternating voltage of 10 mV and the frequency range was from 10 kHz to 0.1 Hz. In order to further characterize the performance of the ionomer in AEMFC, the operation time of the cell under the constant current density 100 mA cm^{-2} was tested and recorded by the electric load system.

2.8. Characterization of CL microstructure

The surface morphology of the electrodes were investigated using a scanning electron microscope (SEM) (JEOL JSM-6700F) with an acceleration voltage 3.0 kV.

2.9. Transmission electron microscopy (TEM) observation

The TEM of the QASEBS (Cl type) was observed. The operation details are as follows. Firstly, the membrane was solidified using the epoxy resin; then the solidified membrane was sliced with the thickness of *ca.* 100 nm; at last, the slices were placed on a Cu grid, followed by exchanging the anions for PtCl_6^{2-} , and then subjected to TEM observations.

3. Results and discussion

3.1. Characterization of the QASEBS ionomer

The SEBS with 30 wt% styrene units is chosen as the raw material to prepare ionomers. The process of functioning SEBS



as follows: firstly, the chloromethylation reaction of the SEBS occurs in a diluted chloroform solution by BCMB under catalysis of tin(IV) chloride at the temperature of 55 °C for 12 h. Then the chloromethylated SEBS is precipitated from a 5 wt% tetrahydrofuran solution to methanol for purification. Lastly, the CMSEBS is dissolved in tetrahydrofuran to prepare a 5 wt% ionomer solution for using in preparation of the MEA.

The substance structures are characterized by ^1H NMR and FT-IR. The chloromethylation reaction is monitored by ^1H NMR, as shown in Fig. 1. Compared with the Fig. 1(a) of the SEBS, a new peak 9 occurs in the Fig. 1(b) (^1H NMR (400 MHz, chloroform- d , 25 °C, TMS): $\delta = 4.5$ (s, 2H, CH_2)), which indicates that the chloromethyl groups are successfully grafted onto the SEBS. Besides, the ^1H NMR of QASEBS was also conducted. Due to the QASEBS bad solubility, an indirect method is employed. The results were shown in Fig. 1(c) and (d). Compared with Fig. 1(b) and (d), the peak 9 of $\sigma = 4.5$ disappeared and a new peak 9' of $\sigma = 4.67$ occurred in Fig. 1(c), and the peak 9' was attributed to the H of $-\text{CH}_2-\text{N}^+$, which demonstrated the CMSEBS was successfully quaternized.³⁹ Then the solid membranes structure is characterized by FTIR spectra. As shown in the Fig. 2, the peaks at 3025 and 3059 cm^{-1} in Fig. 2(a) were associated with the stretching vibration of $\text{CH}=\text{}$ groups of aromatic hydrocarbon,⁴⁰

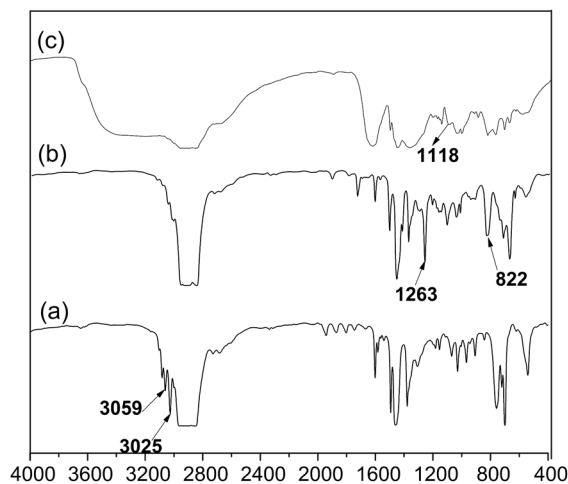


Fig. 2 (a) FTIR spectra of Asahi KASE Tuftec SEBS H1031 (b) CMSEBS and (c) QASEBS (CI type, thin film).

but they disappeared in Fig. 2(b) due to reaction of the styrene units. Furthermore, the peaks at 822 cm^{-1} , 1263 cm^{-1} in the Fig. 2(b) also demonstrated the existence of C-Cl bond, and a characteristic peak at 1118 cm^{-1} was observed, which represented the stretching vibration of C-N groups in Fig. 2(c).⁴⁰ The observation suggested the quaternary ammonium groups being successfully introduced into the SEBS, which accorded with the results of the ^1H NMR.

3.2. Physical properties of the QASEBS ionomer

Firstly, the water uptake has an observable effect on the transport behaviour and mechanical properties of a membrane. It is well known that the higher water uptake can offer more opportunity for ions to transfer through the membrane, leading to a higher conductivity, but excessive water uptake will cause serious dimensionality deformation, and even reduce the mechanical stability of the membrane.⁴¹ Similarly, it is an important factor for ionomer to use in the catalyst layer. The water uptake, IEC and swelling ratio of the QASEBS ionomer with OH^- form are listed in Table 1. The water uptake of the sample is only 57%, and the swelling ratio of the ionomer is 20.4%, 19.1%, 14.5% in the length, width and thickness direction at room temperature, respectively. The results demonstrate that the ionomeric membrane possesses good dimensionality stability.

As a key parameter of the membrane, the IEC represents the hydroxide ion concentration in a polymer matrix, which has an important effect on the water uptake and hydroxide ion conductivity of the ionomer. As shown in Table 2, the IEC of the ionomer is only 0.21 mmol g^{-1} because the content of the styrene units in the raw material SEBS which can be functioned to the quaternary ammonium groups are very low and only 30 wt%.²³

Besides, the hydroxide ion conductivity of the ionomer is also an important parameter, which affects directly the performance of fuel cell.³⁷ The ion conductivity of the ionomer at different temperature is characterized. As shown in Fig. 3(a), the ion conductivities of the QASEBS ionomer are measured in

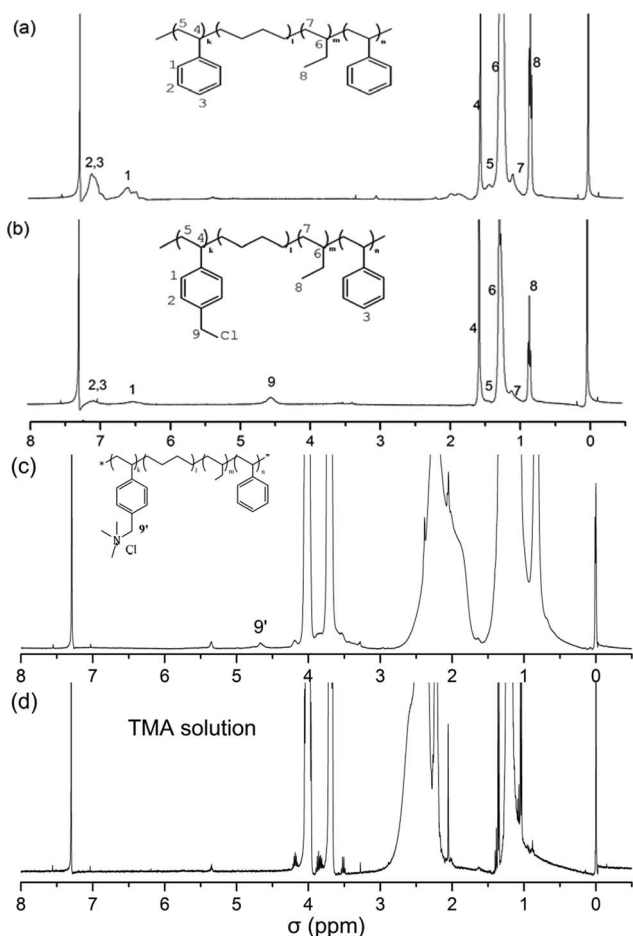


Fig. 1 (a) ^1H NMR spectra of Asahi KASE Tuftec SEBS H1031, (b) CMSEBS, (c) QASEBS (CI type) and (d) TMA solution in chloroform- d .

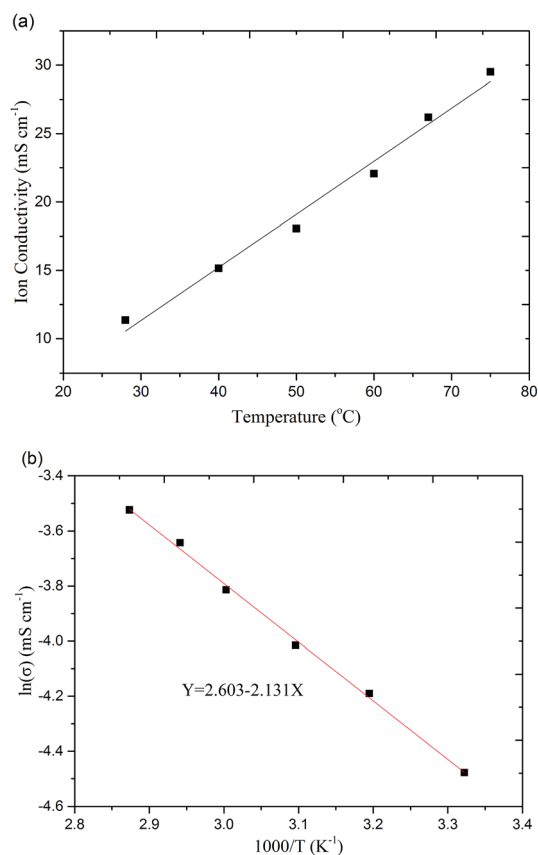
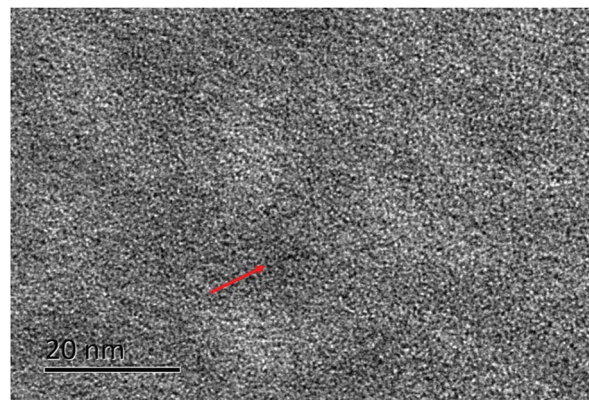


Table 2 Physical properties of the QASEBS ionomer with OH⁻ form at room temperature

Water uptake [wt%]	Swelling ratio [%]			IEC [mmol g ⁻¹]
	Length	Width	Thickness	
57.0	20.4	19.1	14.5	0.21

deionized water at a temperature range from 25 to 75 °C, and the ion conductivity of the ionomer increase with the temperature. Assuming the ion conductivity of the ionomer followed the Arrhenius behaviour, the activation energy E_a of the OH⁻ transportation in the ionomer could be derived from Fig. 3(a) and calculated as 17.52 kJ mol⁻¹, by the curve fitting with the equation $E_a = -b \times R$, where b is the slope of the fitting curve of $\ln \sigma$ vs. $1000/T$ (Fig. 3(b)), R is the universal gas constant (8.314 J K⁻¹ mol⁻¹), σ is the OH⁻ conductivity, and T is absolute temperature (K). The ionomer membrane shows a higher OH⁻ transportation activation energy than that obtained in the A901 (Tokuyama Co. Japan) membrane (14.10 kJ mol⁻¹), which indicates that the ionomer has a weak effect on the change of temperature.

To further analyse the performance of the membrane with the low IEC, we explored whether the microphase separation existed in the membrane or not, for that some publications has

**Fig. 3** Ionic conductivity (a) and Arrhenius plot (b) of the QASEBS with OH⁻ form at different temperatures.**Fig. 4** The TEM image of QASEBS. The dark spots represent the hydrophilic domains (as shown in red arrow).

demonstrated the microphase separation in the membrane could promote the improvement of the ion conductivity.^{16,46} Therefore, in order to demonstrate the existence of the hydrophilic/hydrophobic phase separation micro-structure in QASEBS and the transmission electron microscopy (TEM) of the membrane was conducted on JEM-2100 microscope with an accelerating voltage of 200 kV. The result was shown in Fig. 4. The dark and bright spots in the image represented the hydrophilic and hydrophobic micro-phase, respectively, and the clear hydrophilic/hydrophobic micro-phase separation was observed in the image, which was contributed to form efferent ion transport channels and could promote the OH⁻ conducting.¹⁶ The formation of hydrophilic/hydrophobic micro-phase separation were attributed to the change that the poly styrene block of the triblock copolymers SEBS became the hydrophilic block after functionalization, which could form phase separation micro-structure with the hydrophobic alkyl chains.⁴⁷

3.3. Fuel cell performance

The performance of the ionomer is further studied in AEMFCs, and for the comparison, the performance of AEMFCs assembled with Acta I2 ionomer is also tested. The polarization and power curves of the MEAs are presented in Fig. 5. All the MEAs are assembled with the homemade membrane AEH9620; only the ionomer binders used in the anodes and cathodes are different. Operating at 50 °C and 200 kPa backpressure for H₂/O₂, the AEMFCs with QSEBS ionomer in both the anode and cathode outperform that with the Acta I2 ionomer under the same test conditions. As shown in Fig. 5, the open circuit voltage of the AEMFCs with Acta I2 and QASEBS ionomer is 1.03 V and 1.07 V, respectively. The peak power density of the MEA with the QASEBS ionomer reaches 375 mW cm⁻² when the current density is 688 mA cm⁻² and the cell voltage is 0.545 V, which is better than that of MEA with the commercial Acta I2 ionomer (124 mW cm⁻²). These results indicate that the QASEBS ionomer has lower ionic transport resistance in the electrode than that of the Acta I2 ionomer due to the higher ion conductivity, which is proven by EIS in the later discussion. Although the IEC and the conductivity of the QASEBS ionomer was not very high,



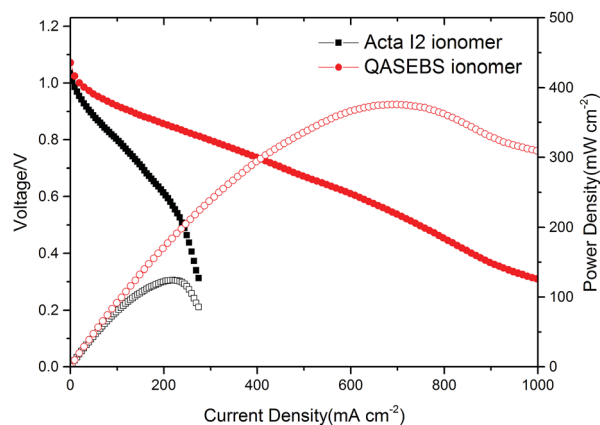


Fig. 5 The cell performance of the different electrode with the Acta I2 ionomer and QASEBS ionomer. Test conditions: cell and H_2/O_2 temperatures of $50\text{ }^\circ\text{C}$; flow rates of 0.1 L min^{-1} and 0.2 L min^{-1} for H_2 and O_2 , respectively; and back pressure of 200 kPa for H_2 and O_2 .

the performance of the AEMFC with the QASEBS ionomer was good. The results of the cell performance show that the conductivity of the ionomer in the catalyst layers is not the only role to play, in other words, and the other properties including water uptake, swelling and the comparability of the ionomer with membrane are also critical factors to be considered,⁴⁸ which is contributed to building an efficient three-phase boundary in the catalyst layers, as mentioned in Yang's report.⁴⁹

Besides, the QASEBS ionomer was compared with the literature reported ionomer with TMA based quaternary ammonium in terms of the properties (IEC and conductivity) and cell performance, and the Table 3 presented the results of these ionomers. Although the ionomer in this work has low IEC, the conductivity of the QASEBS ionomer is comparable with the ionomer reported. Besides, the low IEC could limit the water uptake and swelling of the ionomer, which was benefit to construct and stabilize the ionomer-catalyst interface and develop performance of the AEMFC.⁵⁰ Furthermore, the most of ionomers reported didn't conduct the durability test of the H_2/O_2 AEMFC.

To further investigate the effects of the ionomer on the cell performance, associating with the Tafel plots of the AEMFCs with

the QASEBS ionomer and the Acta I2 ionomer after the iR -correction, the electrode kinetic parameters are obtained by fitting the i - V curves at low current density ($<50\text{ mA cm}^{-2}$).^{49,51} As shown in the Fig. 6(a), the Tafel slopes of the AEMFCs with the QASEBS ionomer and the Acta I2 ionomer are 32.2 mV dec^{-1} and 43.5 mV dec^{-1} , respectively. The results reveal that the ionomers have effects on the active polarization losses, and furthermore, in the high current density, the AEMFC with the Acta I2 ionomer have more serious voltage degradation than that with the QASEBS ionomer due to the ohmic overpotential and mass diffusion polarization losses.⁴⁹ Besides, to understand the polarization behaviours of various ionomer in the MEAs, the *in situ* EIS measurements are carried out by discharging AEMFC at various current density, including 100 and 200 mA cm^{-2} , respectively. Compared with the results of the MEAs with various ionomer, the obvious effects are made on the high frequency and mid-frequency resistance referred to the ohmic polarization and charge-transfer limitations. As shown in the Fig. 6(b) and (c), the ohmic resistance of AEMFC with the QASEBS ionomer is $0.07\text{ }\Omega\text{ cm}^2$, while that of AEMFC with the Acta I2 ionomer reaches $0.14\text{ }\Omega\text{ cm}^2$, twice of the former. Furthermore, when the current density vary from 100 to 200 mA cm^{-2} , the mid-frequency of AEMFC with QASEBS ionomer remarkably decreases due to the current addition leading to the water content of the MEA increasing and further the ohmic resistance decreasing. On the other hand, the EIS data of AEMFC with Acta I2 ionomer was not changing, and it's mainly caused by the high hydrophilicity and is easy to be flooded at high current density.⁴⁹ In addition, EIS impedance spectra of MEAs under 100 and 200 mA cm^{-2} were fitted using the equivalent circuit illustrated in Fig. 6(d). R_{ohm} , R_a and R_c represented ohmic resistance, charge transfer resistance of anode and cathode and were shown in Table 4, respectively. It is noteworthy that the R_a and R_c of the MEA with the QASEBS ionomer is lower than that of the MEA with the Acta I2 ionomer. Furthermore, the charge transfer resistance and the mass diffusion resistance will increase because the flooding leads to the deterioration of the TPBs and the mass transportation and further the voltage degradation, which is in accord with the results of the i - V measurements.

Furthermore, the durability of the electrode with QASEBS ionomer is also investigated. As shown in Fig. 7, the MEA's

Table 3 The properties and cell performance of the ionomers reported

Ionomers	Ionic group	IEC ^b , mmol g ⁻¹	Conductivity, mS cm ⁻¹	P_{max} ^c , mW cm ⁻²	Durability ^f , hours	Ref.
QAPS	QA _{TMA} ^a	1.18	20 (30 °C)	100	n.a. ^d	42
AS-4	QA _{TMA}	1.30	13 (30 °C)	338	234@100 mA cm ⁻²	27
FAA	QA _{TMA}	1.20	17 (20 °C)	200	146@100 mA cm ⁻²	34
QPMBV	QA _{TMA}	1.35	10 (30 °C)	180	n.a.	43
ETFE-g-poly(VBTMAC)	QA _{TMA}	1.24	n. a.	1160	n.a.	36
QAPVA	QA _{TMA}	1.20	23 (r. t.) ^e	172.8	n.a.	22
PTFE-VBTMAC	QA _{TMA}	n. a.	n. a.	130	n.a.	44
S-PFSO ₂ NH-GCI-CI	QA _{TMA}	n. a.	n. a.	138	n.a.	45
QASEBS	QA _{TMA}	0.21	11 (30 °C)	375	550@100 mA cm ⁻²	This work

^a QA_{TMA} = TMA based quaternary ammonium. ^b IEC, the ion exchange capacity. ^c P_{max} , the peak power density of H_2/O_2 AEMFC. ^d Not completed. ^e r. t. = room temperature. ^f Durability, the operation time of the H_2/O_2 AEMFC under 100 mA cm^{-2} .



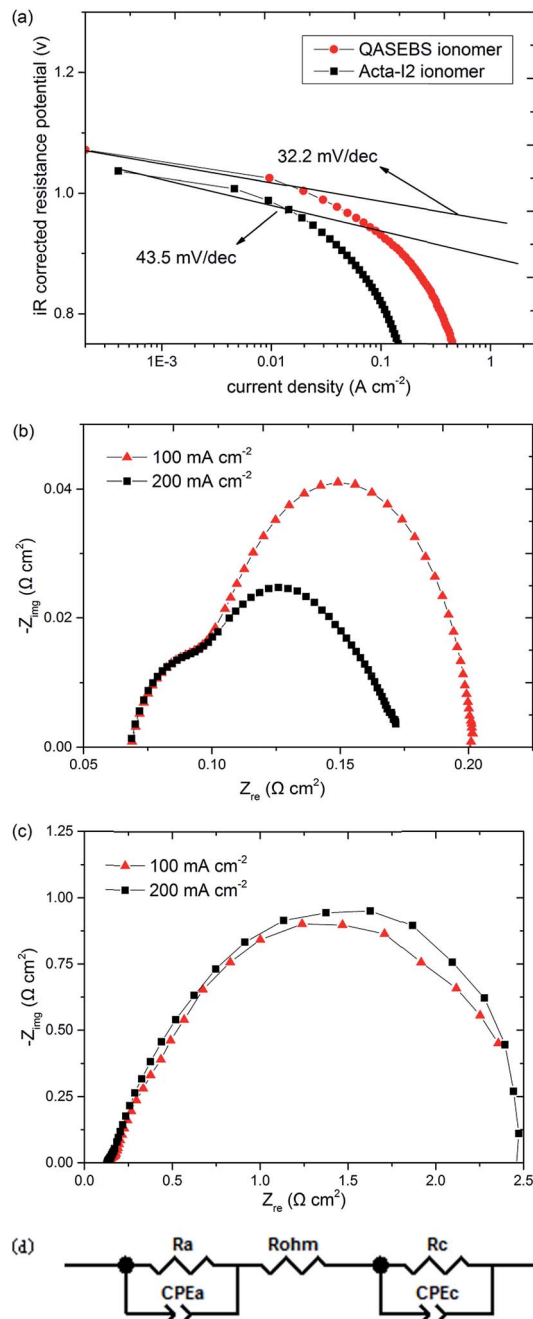


Fig. 6 The Tafel plot (a) of the AEMFCs with the QASEBS ionomer and the Acta I2 ionomer; the electrochemical impedance spectra of the AEMFCs with the QASEBS ionomer (b) and the Acta I2 ionomer (c) at 100, 200 mA cm⁻² current density, respectively. (d) The equivalent circuit for modelling the EIS. The test conditions: cell and H₂/O₂ temperatures of 50 °C; flow rates of 0.1 L min⁻¹ and 0.2 L min⁻¹ for H₂ and O₂, respectively; and back pressure of 200 kPa for H₂ and O₂.

lifetime has been over 550 h under the constant current density 100 mA cm⁻², which is more than that reported in the public literatures.^{19,43} Moreover, the voltage of the MEA remains above 0.6 V, and the rate of the average voltage degradation is about 0.22 mV h⁻¹. During the test, the voltage quickly dropped from 0.8 V to 0.7 V at first of 50 h, which is ascribed to the polarization loss of AEMFC. At 200 h and 400 h, the operation

Table 4 The fitting parameters using the equivalent circuit for the EIS modelling

		$R_{ohm}, \Omega \text{ cm}^2$	$R_a, \Omega \text{ cm}^2$	$R_c, \Omega \text{ cm}^2$
QASEBS ionomer	100 mA cm ⁻²	0.07144	0.02442	0.1065
	200 mA cm ⁻²	0.07052	0.01073	0.09139
Acta I2 ionomer	100 mA cm ⁻²	0.1407	0.2204	2.435
	200 mA cm ⁻²	0.1372	0.2186	2.361

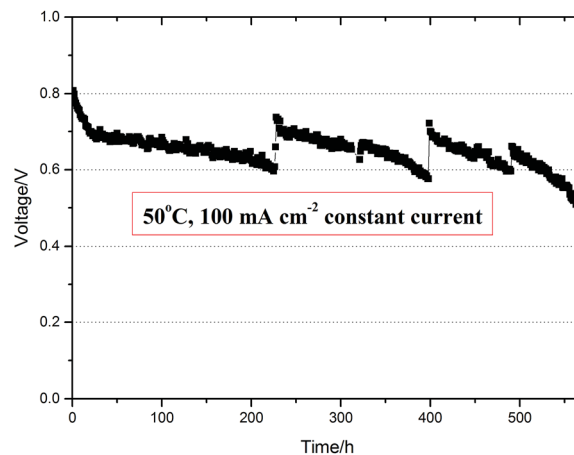


Fig. 7 Durability test for MEA with QASEBS ionomer. Test conditions: constant current density 100 mA cm⁻², cell and H₂/O₂ temperatures of 50 °C; flow rates of 0.1 L min⁻¹ and 0.2 L min⁻¹ for H₂ and O₂, respectively; and back pressure of 200 kPa for H₂ and O₂.

pressure of AEMFC is unloaded and the voltage increases about 100 mV, while the liquid drops expel from the cell inside. These results demonstrated that the liquid water will accumulate during the process of the AEMFC operation, and the relief pressure is benefit to the evaporation and exhaust of the liquid water and further improves the water management of AEMFC. On the other hand, the flow rate of H₂ and O₂ increased for that the pressure difference increased, and increased the ratio of reactant gases and promoted the electrochemical reaction, which led to the increase of cell voltage.

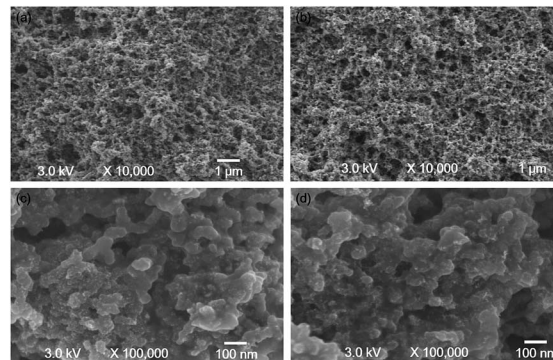


Fig. 8 SEM of the electrode with the QASEBS ionomer before (a and c) and after (b and d) of the durability test.



Besides, the SEM of the electrode before and after test were performed, as shown in Fig. 8. Comparing the SEM spectrum of before test electrode with that of after test electrode, we found that the SEM images of the catalyst layer before and after test were similar, which also might explain the good durability of the AEMFC with the QASEBS ionomer.

4. Conclusions

For AEMFC, the ionomer is the key material of the catalyst layer, which determines the property of the MEA to a large extent. In this work, the triblock copolymer SEBS with good chemical and thermal stability was chosen raw material, and the CMSEBS ionomer that could dissolve in tetrahydrofuran which benefit MEA fabrication process was synthesized. By $^1\text{H-NMR}$, FT-IR characteristic, it was found that the functional groups was grafted onto the SEBS successfully. Moreover, the physical properties of the QASEBS ionomer, such as water uptake, swelling ratio, conductivity and IEC, were measured, and the OH^- conductivity of the ionomer reached 30 mS cm^{-1} at 75°C . Besides, the QASEBS was used as the ionomer in the H_2/O_2 anion exchange membrane fuel cell and showed a peak power density of 375 mW cm^{-2} at 50°C , which was about 3 times higher than that of the MEA using the commercially available Acta I2 ionomer (124 mW cm^{-2}). In addition, only 0.22 mV h^{-1} degradation rate of voltage was seen during durability test at a constant current density 100 mA cm^{-2} for 550 h.

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Notes and references

- 1 S. F. Lu, J. Pan, A. B. Huang, L. Zhuang and J. T. Lu, *Proc. Natl. Acad. Sci. U. S. A.*, 2008, **105**, 20611–20614.
- 2 J. R. Varcoe and R. C. T. Slade, *Fuel Cells*, 2005, **5**, 187–200.
- 3 J. R. Varcoe, P. Atanassov, D. R. Dekel, A. M. Herring, M. A. Hickner, P. A. Kohl, A. R. Kucernak, W. E. Mustain, K. Nijmeijer, K. Scott, T. Xu and L. Zhuang, *Energy Environ. Sci.*, 2014, **7**, 3135–3191.
- 4 G. Merle, M. Wessling and K. Nijmeijer, *J. Membr. Sci.*, 2011, **377**, 1–35.
- 5 J.-S. Park, S.-H. Park, S.-D. Yim, Y.-G. Yoon, W.-Y. Lee and C.-S. Kim, *J. Power Sources*, 2008, **178**, 620–626.
- 6 Q. G. He and E. J. Cairns, *J. Electrochem. Soc.*, 2015, **162**, F1504–F1539.
- 7 M. R. Tarasevich and O. V. Korchagin, *Russ. J. Electrochem.*, 2013, **49**, 600–618.
- 8 J. Cheng, G. H. He and F. X. Zhang, *Int. J. Hydrogen Energy*, 2015, **40**, 7348–7360.
- 9 Y. J. Wang, J. L. Qiao, R. Baker and J. J. Zhang, *Chem. Soc. Rev.*, 2013, **42**, 5768–5787.
- 10 Z. Y. Luo, Y. J. Gong, X. F. Liao, Y. J. Pan and H. W. Zhang, *RSC Adv.*, 2016, **6**, 13618–13625.
- 11 L. Zhu, J. Pan, Y. Wang, J. J. Han, L. Zhuang and M. A. Hickner, *Macromolecules*, 2016, **49**, 815–824.
- 12 Q. Ge, Y. Liu, Z. Yang, B. Wu, M. Hu, X. Liu, J. Hou and T. Xu, *Chem. Commun.*, 2016, **52**, 10141–10143.
- 13 Z. B. Zhuang, S. A. Giles, J. Zheng, G. R. Jenness, S. Caratzoulas, D. G. Vlachos and Y. S. Yan, *Nat. Commun.*, 2016, **7**, 8.
- 14 L. M. Dai, Y. H. Xue, L. T. Qu, H. J. Choi and J. B. Baek, *Chem. Rev.*, 2015, **115**, 4823–4892.
- 15 Q. Li, R. Cao, J. Cho and G. Wu, *Adv. Energy Mater.*, 2014, **4**, 19.
- 16 J. Pan, C. Chen, Y. Li, L. Wang, L. S. Tan, G. W. Li, X. Tang, L. Xiao, J. T. Lu and L. Zhuang, *Energy Environ. Sci.*, 2014, **7**, 354–360.
- 17 J. R. Varcoe, R. C. T. Slade and E. Lam How Yee, *Chem. Commun.*, 2006, 1428–1429, DOI: 10.1039/b600838k.
- 18 H. Yanagi and K. Fukuta, *ECS Trans.*, 2008, **16**, 257–262.
- 19 M. Piana, M. Boccia, A. Filpi, E. Flammia, H. A. Miller, M. Orsini, F. Salusti, S. Santuccioli, F. Ciardelli and A. Pucci, *J. Power Sources*, 2010, **195**, 5875–5881.
- 20 M. Tanaka, M. Koike, K. Miyatake and M. Watanabe, *Polym. Chem.*, 2011, **2**, 99–106.
- 21 F. X. Zhang, H. M. Zhang, C. Qu and J. X. Ren, *J. Power Sources*, 2011, **196**, 3099–3103.
- 22 M.-S. Shin, Y.-J. Byun, Y.-W. Choi, M.-S. Kang and J.-S. Park, *Int. J. Hydrogen Energy*, 2014, **39**, 16556–16561.
- 23 L. Sun, J. S. Guo, J. Zhou, Q. M. Xu, D. Chu and R. R. Chen, *J. Power Sources*, 2012, **202**, 70–77.
- 24 E. E. Switzer, T. S. Olson, A. K. Datye, P. Atanassov, M. R. Hibbs, C. Fujimoto and C. J. Cornelius, *Electrochim. Acta*, 2010, **55**, 3404–3408.
- 25 L. Zeng and T. S. Zhao, *Electrochem. Commun.*, 2013, **34**, 278–281.
- 26 Y. Zhao, H. M. Yu, D. M. Xing, W. T. Lu, Z. G. Shao and B. L. Yi, *J. Membr. Sci.*, 2012, **421**, 311–317.
- 27 Y. Zhao, H. M. Yu, D. L. Yang, J. Li, Z. G. Shao and B. L. Yi, *J. Power Sources*, 2013, **221**, 247–251.
- 28 N. W. Li, Y. J. Leng, M. A. Hickner and C. Y. Wang, *J. Am. Chem. Soc.*, 2013, **135**, 10124–10133.
- 29 L. Zeng, T. S. Zhao, L. An, G. Zhao, X. H. Yan and C. Y. Jung, *J. Power Sources*, 2015, **275**, 506–515.
- 30 J. A. Kerres, *J. Membr. Sci.*, 2001, **185**, 3–27.
- 31 Y. Leng, L. Wang, M. A. Hickner and C.-Y. Wang, *Electrochim. Acta*, 2015, **152**, 93–100.
- 32 S. D. Poynton, R. C. T. Slade, T. J. Omasta, W. E. Mustain, R. Escudero-Cid, P. Ocon and J. R. Varcoe, *J. Mater. Chem. A*, 2014, **2**, 5124–5130.
- 33 J. Pan, S. Lu, Y. Li, A. Huang, L. Zhuang and J. Lu, *Adv. Funct. Mater.*, 2010, **20**, 312–319.
- 34 S. Gu, R. Cai, T. Luo, Z. W. Chen, M. W. Sun, Y. Liu, G. H. He and Y. S. Yan, *Angew. Chem., Int. Ed.*, 2009, **48**, 6499–6502.
- 35 J. R. Varcoe, R. C. T. Slade, G. L. Wright and Y. Chen, *J. Phys. Chem. B*, 2006, **110**, 21041–21049.
- 36 L. Wang, E. Magliocca, E. L. Cunningham, W. E. Mustain, S. D. Poynton, R. Escudero-Cid, M. M. Nasef, J. Ponce-



- Gonzalez, R. Bance-Souahli, R. C. T. Slade, D. K. Whelligan and J. R. Varcoe, *Green Chem.*, 2017, **19**, 831–843.
- 37 Y. Zhao, H. M. Yu, F. Xie, Y. X. Liu, Z. G. Shao and B. L. Yi, *J. Power Sources*, 2014, **269**, 1–6.
- 38 Y. S. Kim and K. S. Lee, *Polym. Rev.*, 2015, **55**, 330–370.
- 39 C. R. Yang, S. L. Wang, W. J. Ma, L. H. Jiang and G. Q. Sun, *J. Membr. Sci.*, 2015, **487**, 12–18.
- 40 Q. H. Zeng, Q. L. Liu, I. Broadwell, A. M. Zhu, Y. Xiong and X. P. Tu, *J. Membr. Sci.*, 2010, **349**, 237–243.
- 41 Q. J. Duan, S. H. Ge and C. Y. Wang, *J. Power Sources*, 2013, **243**, 773–778.
- 42 J. Pan, S. F. Lu, Y. Li, A. B. Huang, L. Zhuang and J. T. Lu, *Adv. Funct. Mater.*, 2010, **20**, 312–319.
- 43 Y. T. Luo, J. C. Guo, C. S. Wang and D. Chu, *Electrochem. Commun.*, 2012, **16**, 65–68.
- 44 J. R. Varcoe and R. C. T. Slade, *Electrochem. Commun.*, 2006, **8**, 839–843.
- 45 X. D. Liu, H. R. Gao, X. H. Chen, Y. Hu, S. P. Pei, H. Li and Y. M. Zhang, *J. Membr. Sci.*, 2016, **515**, 268–276.
- 46 J. Han, Q. Liu, X. Li, J. Pan, L. Wei, Y. Wu, H. Peng, Y. Wang, G. Li, C. Chen, L. Xiao, J. Lu and L. Zhuang, *ACS Appl. Mater. Interfaces*, 2015, **7**, 2809–2816.
- 47 Q. Li, L. Liu, Q. Q. Miao, B. K. Jin and R. K. Bai, *Polym. Chem.*, 2014, **5**, 2208–2213.
- 48 R. Zeng, J. Handsel, S. D. Poynton, A. J. Roberts, R. C. T. Slade, H. Herman, D. C. Apperley and J. R. Varcoe, *Energy Environ. Sci.*, 2011, **4**, 4925–4928.
- 49 D. L. Yang, H. M. Yu, G. F. Li, Y. Zhao, Y. X. Liu, C. K. Zhang, W. Song and Z. G. Shao, *J. Power Sources*, 2014, **267**, 39–47.
- 50 V. R. Stamenkovic, D. Strmcnik, P. P. Lopes and N. M. Markovic, *Nat. Mater.*, 2017, **16**, 57–69.
- 51 D. S. Kim, C. H. Fujimoto, M. R. Hibbs, A. Labouriau, Y. K. Choe and Y. S. Kim, *Macromolecules*, 2013, **46**, 7826–7833.

