



Cite this: *RSC Adv.*, 2017, 7, 25019

A Cr₂O₃/MWCNTs composite as a superior electrode material for supercapacitor†

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It has been a challenge to achieve a good capacitive performance for Cr-based oxides, and only few studies have been reported on these oxides. Herein, a composite of chromium oxide (Cr₂O₃) and multi-walled carbon nanotubes (MWCNTs) was prepared by a simple hydrothermal procedure followed by thermal decomposition. The Cr₂O₃ nanoparticles were evenly decorated on the surface of MWCNTs. The prepared Cr₂O₃/MWCNTs composite exhibited superior electrochemical performance in a 1 M KOH electrolyte, *i.e.*, 257 F g⁻¹ at the current density of 0.25 A g⁻¹, and the capacity fade was only 12% after 3000 cycles. In addition, the Cr₂O₃/MWCNTs//AC asymmetric capacitor has high energy density (15.2 W h kg⁻¹ at 266 W kg⁻¹). The abovementioned results for the first time indicate the potential application of Cr₂O₃ as an electrode material for supercapacitors.

Received 16th February 2017
 Accepted 12th April 2017

DOI: 10.1039/c7ra01954h

rsc.li/rsc-advances

Introduction

Recently, the need of modern society and emerging energy shortage have stimulated untiring research effort into novel, cheap, and eco-friendly energy storage systems.¹ Lithium-ion batteries² (LIBs) and supercapacitors^{3,4} are two major energy storage devices. Supercapacitors have fast charge–discharge capability, high power density, and excellent cycling stability in comparison with LIBs.⁵

Generally, on the basis of operating principle, supercapacitors can be classified into two types. One is electrical double-layer capacitors (EDLCs), which store energy by accumulating charges at the surface of the electrodes. Carbon materials including activated carbon (AC), carbon aerogel, carbon nanotubes (CNTs), and graphene have been widely applied in EDLCs. These EDLCs have outstanding cycling performance as no chemical reaction occurs in them.⁶ However, major drawbacks of these supercapacitors are their low energy density and specific capacitance. The other is faradaic supercapacitors that store energy through reversible redox reactions.^{7,8}

Transition metal oxides (TMO) such as MnO₂,^{8–10} RuO₂,¹¹ V₂O₅,^{12,13} MoO₃,^{14,15} Co₃O₄,¹⁶ and NiO¹⁷ possess a higher pseudocapacitance and energy density compared with carbon

materials. In general, use of metal oxides as supercapacitors is limited by their poor conductivity and lower specific surface area (SSA).

CNTs, known for their diverse and promising mechanical and electrical features, have been widely investigated and applied in various energy storage devices. Due to their excellent properties such as suitable pore structure, high SSA, superior conductivity as well as stability, CNTs act as an ideal skeleton for metal oxide nanoparticles to grow. At present, the primary way to grow nanoparticles on CNTs is their surface functionalization by efficient oxidizer to produce hydrophilic groups. These groups can act as nucleation centers for *in situ* assembling of nanoparticles on the surface of CNTs. The CNTs have been combined with various metal oxides to form composite electrodes. For instance, CNTs could be used as a conducting network to improve the electrochemical reversibility of some oxides, such as MnO₂, and enhance their capacitive enhancement.^{9,18}

As a p-type TMO semiconductor with a wide band gap ($E_g \sim 3.4$ eV), Cr₂O₃ is widely applied in optical and electronic devices, catalysts, gas sensors, and colorants due to its high electrical conductivity, special optoelectronic characteristics, as well as high thermal and chemical stability.¹⁹ Cr₂O₃ is cheap and easily available. Therefore, it is a promising candidate for supercapacitor electrode materials. As for the fourth period transition metal elements, it is known that oxides of Ti, V, (Cr), Mn, Fe, Co, and Ni can be utilized as supercapacitors, electrode materials, *etc.* To date, very few studies have been reported on oxides of Cr, thereby indicating that it is a challenge to achieve a good capacitive performance for Cr-based oxides.

In this study, we successfully synthesized a Cr₂O₃/MWCNTs composite. Cr₂O₃ nanoparticles were well-distributed on the

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† Electronic supplementary information (ESI) available. See DOI: 10.1039/c7ra01954h



MWCNTs. In particular, the composite electrode delivered a high specific capacitance of 257 F g^{-1} at the current density of 0.25 A g^{-1} . The capacity fade was only 12% after 3000 cycles, indicating good cyclic stability of the composite. In addition, energy density of the $\text{Cr}_2\text{O}_3/\text{MWCNTs}/\text{AC}$ asymmetric capacitor (ASC) could reach up to 15.2 W h kg^{-1} at the power density of 266 W kg^{-1} . The present study provides an effective strategy to design new materials based on the combination of CNTs and Cr_2O_3 for energy storage devices.

Experimental

Preparation of the composite

The MWCNTs, 30–50 nm in diameter, were purchased from Chengdu Organic Chemicals Co. Ltd., Chinese Academy of Science. The AC with a specific surface area of $1600 \text{ m}^2 \text{ g}^{-1}$ was purchased from Ningde Xinsen Chemicals Co. Ltd. All other chemical reagents used in this study were of analytical grade purity and were purchased from Sinopharm Chemical Reagent Co. Ltd, and used without any purification. The MWCNTs were treated with 6 M HNO_3 solution to purify and render hydrophilic groups on the surface.^{9,20} Next, 20 mg acid-treated MWCNTs were immersed in a 0.03 M $\text{Cr}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$ absolute ethanol solution (35 mL). Then, the solution was transferred to a Teflon-lined stainless steel autoclave of 40 mL volume and heated at $180 \text{ }^\circ\text{C}$ for 12 h. After filtration and washing with deionization water, the product was dried at $80 \text{ }^\circ\text{C}$ overnight. The formed precipitate was finally annealed at $500 \text{ }^\circ\text{C}$ for 3 h under the protection of N_2 . Moreover, pristine Cr_2O_3 nanoparticles were prepared by calcifying $\text{Cr}_2\text{O}_3/\text{MWCNTs}$ composite at $500 \text{ }^\circ\text{C}$ for 2 h in air.

Characterization

The structures and morphologies of the materials were analyzed by X-ray diffraction (XRD, Bruker D4), scanning electron microscopy (SEM, Philip XL 300 microscope), transmission electron microscopy (TEM, JEOL JEM-2010), and N_2 adsorption-desorption test (TRISTAR3000, MICROMERITICS). Thermal properties of the composite were investigated by thermogravimetric analysis (TGA, Du-Pont TGA-2950).

Electrochemical tests

The working electrode was fabricated by mixing the active material (80 wt%), acetylene black (10 wt%), and polytetrafluoroethylene (PTFE) (10 wt%). Then, the mixture was coated on a Ni grid with the active material mass loading of 0.8 mg cm^{-2} . After being dried at $80 \text{ }^\circ\text{C}$ overnight, they were used for electrochemical measurements.

The electrochemical measurements were evaluated by cyclic voltammetry (CV) and electrochemical impedance spectroscopy (EIS) using a CHI440B electrochemical working station (Shanghai, China). The galvanostatic charge/discharge (GCD) measurement was conducted using an automatic LAND battery (Wuhan, China). A three-compartment cell was assembled by the abovementioned working electrode, a Ni grid (counter electrode), and a saturated calomel electrode (SCE, reference

electrodes). The specific capacitance was calculated according to the following equation:

$$C_s = \int \frac{I dt}{m dV} \quad (1)$$

where C_s is the specific capacitance (F g^{-1}), I is the current (A), dt is the time (s), m is the mass of the active materials (g), and dV is the voltage change.

Results and discussion

Morphology and structure

The XRD patterns of the $\text{Cr}_2\text{O}_3/\text{MWCNT}$ nanocomposite, virginal Cr_2O_3 , and MWCNTs are shown in Fig. 1a. The $\text{Cr}_2\text{O}_3/\text{MWCNT}$ nanocomposite shows peaks at around 25° , 34° , 37° , 42° , 51° , 55° , 64° , and 66° corresponding to the (012), (104), (110), (113), (204), (116), (214), and (300) crystal planes of Cr_2O_3 (JCPDS no. 06-0504), respectively. The peak at around 26° indicates the existence of MWCNTs. For the $\text{Cr}_2\text{O}_3/\text{MWCNT}$ composite, the intensity of the (110) plane is obviously stronger than that of the (012) plane as the $\text{Cr}_2\text{O}_3/\text{MWCNT}$ composite has a crystal orientation along the (110) plane. The (110) plane has more Cr atoms compared to other planes, which can provide more active sites during the OH^- insertion/extraction process (Fig. S1†). Moreover, other impurity peaks were not detected. The content of Cr_2O_3 in $\text{Cr}_2\text{O}_3/\text{MWCNTs}$ was

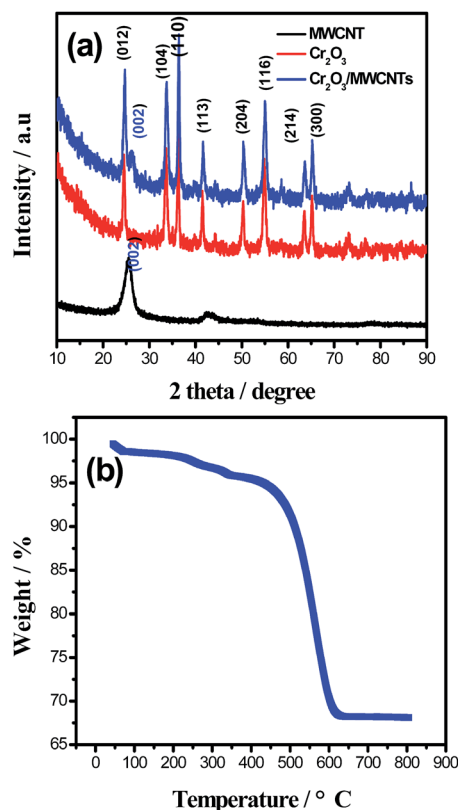


Fig. 1 (a) XRD pattern and (b) TG analysis of the $\text{Cr}_2\text{O}_3/\text{MWCNT}$ composite.



accurately calculated by TGA, as shown in Fig. 1b, as 68 wt% in the composite.

The porosity of Cr₂O₃ and Cr₂O₃/MWCNT composite was analyzed by N₂ adsorption–desorption test, as shown in Fig. 2a and b, respectively. The relevant parameters including Brunauer–Emmett–Teller (BET) surface area (S_{BET}), total pore volume (V_{Tot}), and average pore size (D_{av}) are listed in Table 1. The Cr₂O₃/MWCNT composite demonstrated a type-IV isotherm, and a small hysteresis loop can be observed at the relative pressure between 0.7 and 0.95, which implied the presence of mesopores in the surface structure. The Cr₂O₃/MWCNT composite possesses a S_{BET} of 225 m² g⁻¹ and higher V_{Tot} of 0.741 cm³ g⁻¹. In general, S_{BET} is a vital factor influencing the specific capacitance since charges accumulate on the surface of the supercapacitor electrodes and a higher SSA of the electrode usually result in an increased specific capacitance.²¹ According to the Barrett–Joyner–Halenda (BJH) pore size distribution plot, the average pore size of the Cr₂O₃/MWCNT composite was calculated to be 6.67 nm. The increase of the pore size in the mesoporous region indicates improved energy and power outputs of the supercapacitors.²²

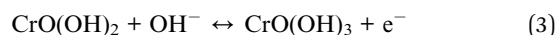
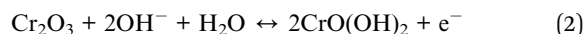
The SEM (Fig. 3a) and TEM (Fig. 3b and c) images of the Cr₂O₃/MWCNT composite show that the Cr₂O₃ nanoparticles are evenly decorated on the surface of the MWCNTs. The size of the Cr₂O₃ nanoparticles are in the range from 5 to 10 nm. The abovementioned results confirmed the successful synthesis of the Cr₂O₃/MWCNT composite. In contrast, the virginal Cr₂O₃

Table 1 The relevant porosity parameters for virginal Cr₂O₃ and its composite

Sample	S_{BET} (m ² g ⁻¹)	V_{Tot} (cm ³ g ⁻¹)	D_{av} (nm)
Cr ₂ O ₃ /MWCNTs	224.9	0.741	6.76
Cr ₂ O ₃	45.7	0.362	5.93

particles were agglomerated (Fig. 3d). Perhaps, the MWCNT framework could distinctly impede the aggregation of the Cr₂O₃ nanoparticles during the crystal growth process, resulting in their smaller diameter and better dispersion on the MWCNTs.

The CV curves of the two samples at the scan rate of 10 mV s⁻¹ between -0.04 and 0.46 V (vs. SCE) in a 1 M KOH electrolyte are shown in Fig. 4a. The composite exhibited two redox peaks at about 0.1/0.08 V and 0.4/0.33 V (vs. SCE). The possible reversible redox reactions have been presented as follows:²³



Moreover, larger area of the Cr₂O₃/MWCNT composite indicated that the composite had a higher capacitance than Cr₂O₃, which was due to the incorporation of the MWCNT. This can improve the conductivity and the rate of ion transport. Another important reason can be owed to smaller size of the Cr₂O₃ nanoparticles, which can provide more active sites during the reaction process as well as shorten the distance for ion diffusion.

As shown in the CV curves of the two samples at various scan rates (Fig. 4b and c), compared with those of virginal Cr₂O₃, the peak positions of the composite shifted a little because of their different crystal orientation. As for the composites, the location of the redox peaks slightly changed with the increasing scan rate, indicating good reversibility of the redox reactions.

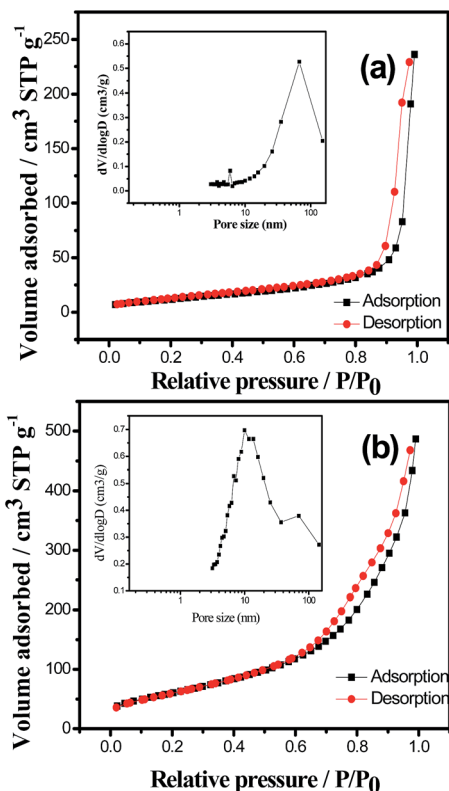


Fig. 2 N₂ adsorption–desorption isotherms and pore size distribution (inset) of (a) virginal Cr₂O₃ and (b) Cr₂O₃/MWCNT composite.

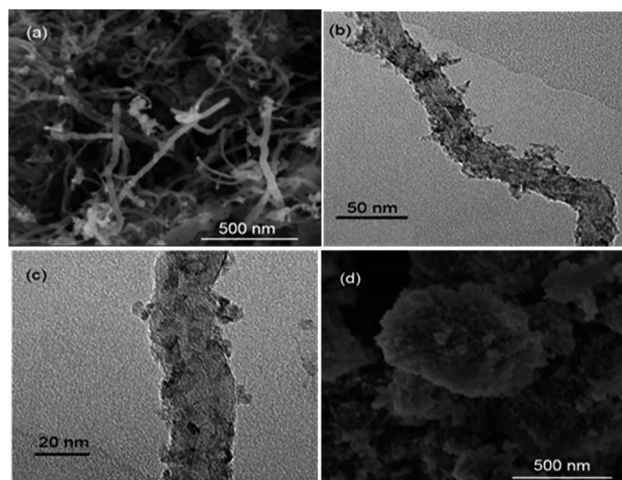


Fig. 3 Morphology of the samples: SEM image (a) and TEM images (b and c) of Cr₂O₃/MWCNTs composite; SEM image of Cr₂O₃ (d).



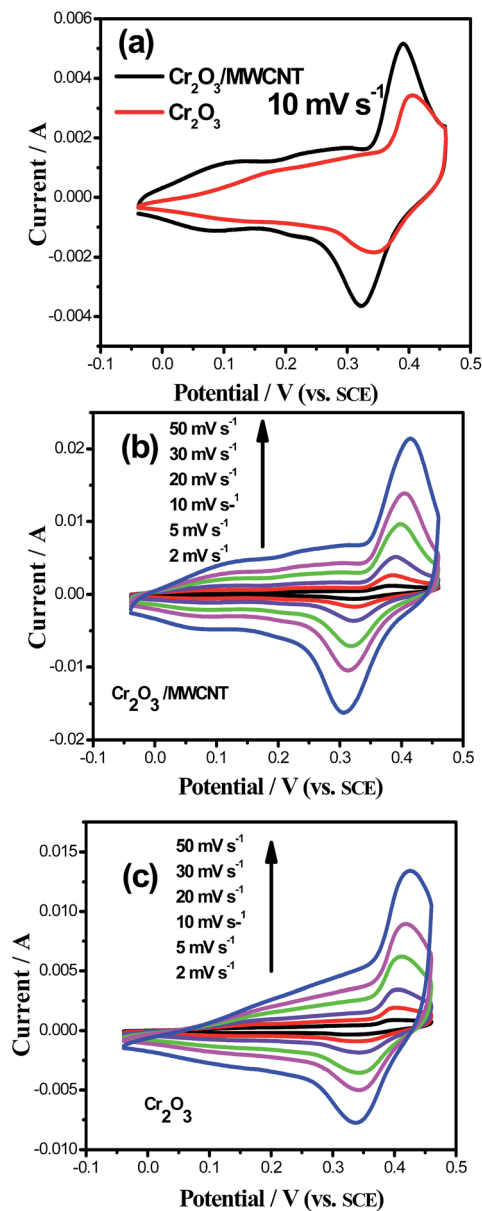


Fig. 4 CV curves of (a) Cr_2O_3 and $\text{Cr}_2\text{O}_3/\text{MWCNT}$ composite at the scan rate of 10 mV s^{-1} , (b) Cr_2O_3 , and (c) $\text{Cr}_2\text{O}_3/\text{MWCNT}$ composite at various scan rates.

As shown in the GCD curves of the two samples (Fig. 5a), a discharge flat voltage at around 0.3 V was observed for both samples, which corresponds well to the CV results, indicating a typical pseudocapacitive property of Cr_2O_3 . Note that the $\text{Cr}_2\text{O}_3/\text{MWCNT}$ composite has much longer discharge time than Cr_2O_3 , confirming its higher specific capacitance. The GCD curves of the $\text{Cr}_2\text{O}_3/\text{MWCNT}$ composites at various current densities are shown in Fig. 5b. The calculated specific capacitance values were 257, 189, 164, and 153 F g^{-1} , at 0.25, 0.5, 1, and 1.5 A g^{-1} , respectively. Even at the high current density of 2 A g^{-1} , the composite possessed a high specific capacitance of 138 F g^{-1} . However, the capacitance value of virginal Cr_2O_3 was only 42 F g^{-1} (Fig. 5c). The cycling stability of both the electrodes were evaluated using GCD at 0.5 A g^{-1} (Fig. 5d).

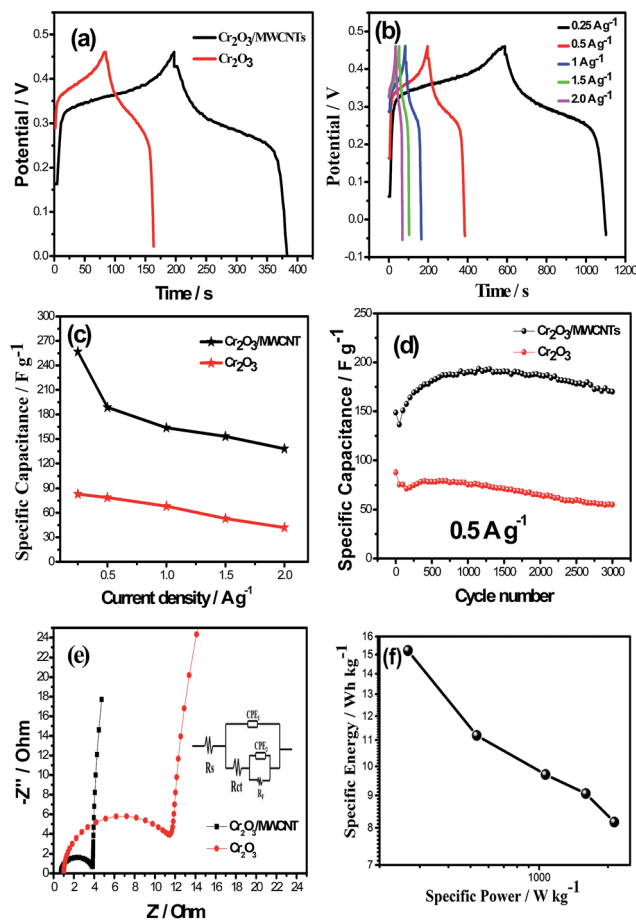


Fig. 5 Electrochemical characteristics of the materials: (a) GCD curves of the two samples at 0.5 A g^{-1} , (b) GCD curves of the $\text{Cr}_2\text{O}_3/\text{MWCNT}$ composite at various current densities, (c) specific capacitance of the two materials vs. current densities, (d) cycling performance of the two samples at 0.5 A g^{-1} , and (e) EIS curve of the two samples at the frequency of 0.1 Hz and equivalent circuit used for fitting the impedance spectra. (f) Ragone plot related to energy and power densities of the $\text{Cr}_2\text{O}_3/\text{MWCNT}/\text{AC ASC}$.

Obviously, at the beginning of the cycling, the specific capacitance gradually increased due to an initial activation process.²⁴ After 3000 cycles, the capacity fade of the $\text{Cr}_2\text{O}_3/\text{MWCNT}$ composite and virginal Cr_2O_3 were 12% and 38% of their highest capacitance, respectively. These values are superior to those reported in the literature.^{23,25} Moreover, there was a remarkable increase in the specific capacitance due to the introduction of the MWCNTs. Specifically, at the constant stage, the composite electrode possessed a high specific capacitance of 180 F g^{-1} , nearly 110 F g^{-1} higher than that of the virginal Cr_2O_3 electrode. The good cycling stability and high pseudocapacitance of the composite can be explained as follows:⁹ the MWCNTs can form an interconnected conductive network, which can increase the conductivity.

The EIS curves of both the samples are shown in Fig. 5e. The amplitude of the perturbation was 5 mV and the data were obtained in the frequency range from 0.01 Hz to 10 kHz. The equivalent circuit used for fitting the impedance spectra is



presented in the inset of Fig. 5e, which consists of the bulk solution resistance R_s , charge transfer resistance R_{ct} , pseudo-capacitive element R_p , and constant phase element (CPE).²⁶ The semicircle in the high frequency region resulted from R_{ct} .^{9,27} The R_{ct} was estimated to be 3.3 Ω for the $\text{Cr}_2\text{O}_3/\text{MWCNT}$ composite, which was lower than 11 Ω for virginal Cr_2O_3 . This result suggests that the reversible reactions on the electrode surface of the $\text{Cr}_2\text{O}_3/\text{MWCNT}$ composite are much facile than those of virginal Cr_2O_3 .²⁸ Moreover, the intercept on Z' indicates R_s . Clearly, it was evident that R_s of the composite (0.7 Ω) was smaller than that of virginal Cr_2O_3 (1.0 Ω), which means that the incorporation of the MWCNTs can decrease the R_s of the composite and that was similar to the actions of CNTs in other composites such as with MnO_2 and V_2O_5 .²⁹ These results can justify the abovementioned results indicating that the incorporation of MWCNTs remarkably enhanced the electrochemical performance of the $\text{Cr}_2\text{O}_3/\text{MWCNT}$ composite.

The applicability of the composite was investigated in an ASC ($\text{Cr}_2\text{O}_3/\text{MWCNTs}/\text{AC}$), in which the composite/Ni grid acted as a positive electrode and AC (mixed with 10% PTFE)/Ni grid acted as the negative electrode. The electrochemical measurement was carried out in a 1 M KOH electrolyte, and the potential ranged from 0 to 1.46 V. The energy and power densities of the ASC systems were calculated according to the following equations:

$$E = C_s(\Delta V)^2/7.2 \quad (4)$$

$$P = 3600E/\Delta t \quad (5)$$

where E (W h kg^{-1}) is the specific energy density, P (W kg^{-1}) is the specific power density, C_s (F g^{-1}) is the specific capacitance of the total ASC system, ΔV (V) is the potential range, and Δt (s) is the discharge time.

Fig. 5f shows the Ragone plot related to the energy and power densities of the $\text{Cr}_2\text{O}_3/\text{MWCNTs}/\text{AC}$ ASC. At the power density of 266 W kg^{-1} , its energy density can reach up to 15.2 W h kg^{-1} ; even at the high power density of 2130 W kg^{-1} , the energy density could still remain 8.17 W h kg^{-1} .

The outstanding electrochemical performance of the composite can be attributed to the following reasons: (1) MWCNTs provide a framework for the Cr_2O_3 nanoparticles and can prevent their agglomeration, which enable the nanoparticles to retain high specific surface area, thereby shortening the diffusion distance during the charge-discharge progress; (2) MWCNTs provide good conductivity; thus, the composite has fast electron transfer rate and low charge transfer resistance, which may accelerate the kinetic process in the redox reaction; (3) MWCNTs have excellent mechanical properties and good chemical and thermal stability, which provide a matrix for long term cycling stability; and (4) another reason is perhaps the preferred crystal orientation due to the MWCNTs template.^{16,30}

Conclusion

In summary, a composite of Cr_2O_3 with MWCNTs was successfully synthesized. The diameter of the Cr_2O_3

nanoparticles was only several nanometers, and these nanoparticles were strongly anchored on the MWCNTs. The specific capacitance of the composite was 257 F g^{-1} at the current density of 0.25 A g^{-1} . The composite also exhibited excellent cycling stability with the capacity fade of 12% after 3000 cycles. Moreover, the energy density of the $\text{Cr}_2\text{O}_3/\text{MWCNTs}/\text{AC}$ ASC could reach up to 15.2 W h kg^{-1} (at 266 W kg^{-1}). The outstanding electrochemical performance of the composite can be ascribed to good conductivity of the MWCNTs, high specific surface area of the composite, and preferred orientation of the crystal planes. Therefore, the $\text{Cr}_2\text{O}_3/\text{MWCNT}$ composite can be applied as a potential electrode material for aqueous supercapacitor. This study also provides a strong basis to exploit the Cr-based composite for supercapacitors.

Acknowledgements

Financial support received from the National Materials Genome Project (2016YFB0700600), China National Distinguished Youth Scientists (51425301), NSFC (U1601214), and Sanyo Chemical Co. Ltd., Japan, is greatly appreciated.

Notes and references

- 1 A. S. Arico, P. Bruce, B. Scrosati, J. M. Scrosati and W. V. Schalkwijk, *Nat. Mater.*, 2005, **4**, 366–377.
- 2 J. Hassoun, F. Bonaccorso, M. Agostini, M. Angelucci, M. G. Betti, R. Cingolani, M. Gemmi, C. Mariani, S. Panero, V. Pellegrini and B. Scrosati, *Nano Lett.*, 2014, **14**, 4901–4906.
- 3 F. Li, G. Li, H. Chen, J. Q. Jia, F. Dong, Y. B. Hu, Z. G. Shang and Y. X. Zhang, *J. Power Sources*, 2015, **296**, 86–91.
- 4 J. L. Zhang, G. L. Chen, Q. Zhang, F. Kang and B. You, *ACS Appl. Mater. Interfaces*, 2015, **7**, 12760–12766.
- 5 L. L. Zhang and X. S. Zhao, *Chem. Soc. Rev.*, 2009, **38**, 2520–2531.
- 6 M. J. Zhi, C. C. Xiang, J. T. Li, M. Li and N. Q. Wu, *Nanoscale*, 2013, **5**, 72–88.
- 7 S. Bose, T. Kuila, A. K. Mishra, R. Rajasekar, N. H. Kim and J. H. Lee, *J. Mater. Chem. A*, 2012, **22**, 767–784.
- 8 J. G. Wang, F. Y. Kang and B. Q. We, *Prog. Mater. Sci.*, 2015, **74**, 51–124.
- 9 Q. T. Qu, P. Zhang, B. Wang, Y. H. Chen, S. Tian, Y. P. Wu and R. Holze, *J. Phys. Chem. C*, 2009, **113**, 14020–14027.
- 10 W. Tang, Y. Y. Hou, X. J. Wang, Y. Bai, Y. S. Zhu, H. Sun, Y. B. Yue, Y. P. Wu, K. Zhu and R. Holze, *J. Power Sources*, 2012, **197**, 330–333.
- 11 F. Yang, L. B. Zhang, A. Zuzuarregui, K. Gregorczyk, L. Li, M. Beltran, C. Tollan, J. Brede, C. Rogero, A. Chuvilin and M. Knez, *ACS Appl. Mater. Interfaces*, 2015, **7**, 20513–20519.
- 12 Q. T. Qu, Y. Shi, L. L. Li, W. L. Guo, Y. P. Wu, H. P. Zhang, S. Y. Guan and R. Holze, *Electrochem. Commun.*, 2009, **11**, 1325–1328.
- 13 Y. J. Wu, G. H. Gao, H. Y. Yang, W. C. Bi, X. Liang, Y. R. Zhang, G. Y. Zhang and G. M. Wu, *J. Mater. Chem. A*, 2015, **3**, 15692–15699.



- 14 W. Tang, L. L. Liu, S. Tian, L. Li, Y. B. Yue, Y. P. Wu and K. Zhu, *Chem. Commun.*, 2011, **47**, 10058–10060.
- 15 Y. Liu, B. H. Zhang, Y. Q. Yang, Z. Chang, Z. B. Wen and Y. P. Wu, *J. Mater. Chem. A*, 2013, **1**, 13582–13587.
- 16 X. W. Wang, M. X. Li, Z. Chang, Y. Q. Yang, Y. P. Wu and X. Liu, *ACS Appl. Mater. Interfaces*, 2015, **7**, 2280–2285.
- 17 K. Y. Tao, P. Y. Li, L. T. Kang, X. R. Li, Q. F. Zhou, L. Dong and W. Liang, *J. Power Sources*, 2015, **293**, 23–32.
- 18 W. Chen, Z. L. Fan, L. Gu, X. H. Bao and C. L. Wang, *Chem. Commun.*, 2010, **46**, 3905–3907.
- 19 S. Lei, X. Peng, Z. Liang, X. Li, C. Wang, B. Cheng, Y. Xiao and L. Zhou, *J. Mater. Chem.*, 2012, **22**, 1643–1651.
- 20 W. Tang, Y. Y. Hou, F. X. Wang, L. L. Liu, Y. P. Wu and K. Zhu, *Nano Lett.*, 2013, **13**, 2036–2040.
- 21 Z. N. Yu, L. Tetard, L. Zhai and J. Y. Thomas, *Energy Environ. Sci.*, 2015, **8**, 702–730.
- 22 J. Zhi, Y. F. Wang, S. Deng and A. Hu, *RSC Adv.*, 2014, **4**, 40296–40300.
- 23 C. Y. Cummings, G. A. Attard, J. M. Mitchells and F. Marken, *Aust. J. Chem.*, 2012, **65**, 65–71.
- 24 R. B. Rakhi, W. Chen, D. Cha and H. N. Alshareef, *Nano Lett.*, 2012, **12**, 2559–2567.
- 25 X. Y. Xu, J. Z. Wu, N. A. Yang, H. Na, L. Z. Li and J. P. Gao, *Mater. Lett.*, 2015, **142**, 172–175.
- 26 Y. Liu, Z. B. Wen, X. W. Wu, X. W. Wang, Y. P. Wu and R. Holze, *Chem. Commun.*, 2014, **50**, 13714–13717.
- 27 L. J. Wang, K. Zhang, Z. Hu, W. C. Duan, F. Y. Cheng and J. Chen, *Nano Res.*, 2013, **7**, 199–208.
- 28 L. B. Ma, H. Zhou, X. P. Shen, Q. R. Chen, G. X. Zhu and Z. Y. Ji, *RSC Adv.*, 2014, **4**, 53180–53187.
- 29 W. Tang, X. W. Gao, Y. S. Zhu, Y. B. Yue, Y. Shi, Y. P. Wu and K. Zhu, *J. Mater. Chem.*, 2012, **22**, 20143–20145.
- 30 F. X. Wang, X. W. Wang, Z. Chang, Y. S. Zhu, L. J. Fu, X. Liu and Y. P. Wu, *Nanoscale Horiz.*, 2016, **1**(1), 272–289.

