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# 1 Introduction

Hydrogen is one of the most promising fuels for the future.<sup>1</sup> Most hydrogen is produced from the steam reforming of natural gas, which requires an intense thermal energy input. Since Fujishima and Honda discovered the photocatalytic production of hydrogen from water on TiO<sub>2</sub> electrodes, the efficient photoconversion of water to hydrogen has been a long-term goal in the areas of energy science and engineering.<sup>2</sup> In general, the capability of a photocatalyst depends heavily on its structure because the structure can influence the three major steps of photocatalysis: photon absorption, charge carrier transfer, and catalytic surface reactions.3 Currently, photocatalysts with three-dimensional (3D) structures have been proposed, and the use of a 3D structure demonstrated a promising photocatalytic performance because of the interconnected open-framework, high specific surface area and more active sites for photocatalytic reactions.4-7

In this regard,  $TiO_2$  nanotube arrays (TNA) have a typical 3D framework with a high mechanical strength. Meanwhile, the excellent electron transport ability was observed for the 3D TNA because of the minimized grain boundaries.<sup>8</sup> However, since  $TiO_2$  could only be used under UV light irradiation due to its wide band gap, one of possible approaches has been made to combine narrow band gap photocatalyst with  $TiO_2$  to induce the absorption into the visible light region.<sup>9</sup> Recently, graphitic

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# Three-dimensional TiO<sub>2</sub> nanotube arrays combined with $g-C_3N_4$ quantum dots for visible light-driven photocatalytic hydrogen production

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A three-dimensional (3D) nanostructured photocatalyst combined with  $g-C_3N_4$  quantum dots (QDs) and a TiO<sub>2</sub> nanotube array (TNA) was fabricated to form a 3D  $g-C_3N_4/TNA$  nanocomposite by a facile electro-deposition process. The photocatalytic ability of the 3D  $g-C_3N_4/TNA$  was evaluated by measuring the amount of hydrogen generated from water splitting under visible light irradiation. Benefiting from an attractive heterostructure between  $g-C_3N_4$  and TiO<sub>2</sub> leading to a unique photogenerated charge separation, as well as a distinctive 3D well-ordered nanotube structure, this 3D  $g-C_3N_4/TNA$  exhibited an average H<sub>2</sub> production of approximately 243 µmol h<sup>-1</sup> g<sup>-1</sup>, which was approximately 4.7 times higher than that of sole  $g-C_3N_4$  under the same experimental conditions. Therefore, this work could offer a prospective 3D nanostructure for visible light-driven photocatalytic applications.

> carbon nitride  $(g-C_3N_4)$ , an metal-free photocatalyst, has attracted increasing attention because of its narrow band gap.10 In particular, the  $g-C_3N_4$  quantum dots (QDs) have a shorter transferring distance of photoinduced charge carriers and strong quantum confinement.11 Most recently, Li and coworkers reported that size-controllable g-C3N4 QDs modified single-crystalline TNA could exhibit high activity in synergetic H<sub>2</sub> evolution and organic pollutant degradation.<sup>12</sup> Su and coworkers also reported that modification of TNA by the ultrasmall g-C<sub>3</sub>N<sub>4</sub> QDs could improve the photoelectrochemical activity significantly.13 However, until now, the TNA combined with g-C<sub>3</sub>N<sub>4</sub> QDs has a Ti substrate, which restricted its photocatalytic application. Here, the 3D TNA without the Ti substrate was prepared by an electrochemical anodization, and then modified with g-C<sub>3</sub>N<sub>4</sub> QDs to fabricate a 3D g-C<sub>3</sub>N<sub>4</sub>/TNA by a facile electro-deposition. The performance of this 3D  $g-C_3N_4/$ TNA photocatalyst was investigated for visible light-driven hydrogen production from water.

# 2 Experimental

#### 2.1 Preparation of 3D g-C<sub>3</sub>N<sub>4</sub>/TNA

The method for preparation of  $g-C_3N_4$  QDs based on a previous report.<sup>11</sup> Meanwhile, 3D TNA with Ti substrate was fabricated by potentiostatic anodization as Liu's report.<sup>14</sup> Then electrodeposition of  $g-C_3N_4$  QDs on the 3D TNA was performed using an electrophores apparatus (DYY-6C) where the TNA and Pt plate were connected with the negative pole and positive pole, respectively. The electrolyte solution consisted of isopropyl alcohol (100 mL), magnesium nitrate (5 mg), and  $g-C_3N_4$  QDs (0.14 g L<sup>-1</sup>, 10 mL), and the electro-deposition was maintained

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at 20 V for 3 h. Subsequently, the sample annealed at 500 °C for 2 h. Finally, the annealed sample was anodized again in the same stock electrolyte at 60 V for 4 h to remove Ti substrate. The obtained free-standing 3D g-C<sub>3</sub>N<sub>4</sub>/TNA was cleaned in ethanol. Fig. 1 shows a schematic of the 3D g-C<sub>3</sub>N<sub>4</sub>/TNA preparation.

#### 2.2 Characterization

The morphology of the 3D g-C<sub>3</sub>N<sub>4</sub>/TNA was analyzed using scanning electron microscopy (SEM, Quanta 200 FEG) and transmission electron microscopy (TEM, JEM-2100F, JEOL, Japan). The chemical composition was determined using energy dispersive X-ray (EDX) spectroscopy, which was combined with SEM. The crystallinity of the samples was determined by X-ray diffraction (XRD) using a diffractometer with Cu Ka radiation (Shimadzu LabX XRD-6000). X-ray photoelectron spectroscopy (XPS, ESCALAB250, Thermo-VG Scientific) was used to analyze the elementary composition of the samples. The UV-vis absorption spectra of the samples were investigated as a function of the incident wavelength using UV-vis diffuse reflectance spectroscopy (DRS, Shimadzu UV-2450 spectrophotometer). The Brunauer-Emmett-Teller (BET) surface area was analyzed using an automated surface area and pore size analyzer (Quantachrome Autosorb-1 MP).

#### 2.3 Photocatalytic hydrogen production

The photocatalytic production of hydrogen was performed in a glass closed-gas circulation and evacuation system. A 300 W Xe lamp with an ultraviolet filter ( $\lambda \ge 400$  nm) was used as the light source and was positioned 10 cm from the cylindrical quartz photoreactor (300 mL). The light source provided an incident light with an intensity of 100 mW cm<sup>-2</sup>, which was measured by a radiometer (model FZ-A, Photoelectric Instrument Factory Beijing Normal University, China). Photocatalytic hydrogen production was carried out in the water/methanol mixture (v/v, 9/1), where methanol acted as a sacrificial reagent to capture photogenerated holes. In a typical photocatalytic experiment, 0.1 g of the 3D g-C<sub>3</sub>N<sub>4</sub>/TNA catalyst was added to a mixed solution containing 90 mL of water and 10 mL of methanol. The loading of 1.2 wt% Pt co-catalyst was conducted by photoreduction of H<sub>2</sub>PtCl<sub>6</sub>. To remove the dissolved oxygen and to ensure the anaerobic conditions, the reaction system was purged with argon for 1 h at a flow rate of 70 sccm controlled by mass flow controller unit before irradiation. The reaction temperature was maintained at 10 °C by a flow of cooling water. The amount of evolved H<sub>2</sub> was detected using an on-line gas chromatograph (Shimadzu, GC-14C).

# 3 Results and discussion

Fig. 2 shows SEM images of the 3D g-C<sub>3</sub>N<sub>4</sub>/TNA. Fig. 2a shows a highly ordered 3D nanotube structure, and the diameter of the nanotubes was about 160 nm. All of the 3D g-C<sub>3</sub>N<sub>4</sub>/TNA nanocomposites were open at both ends. The length of the 3D g-C<sub>3</sub>N<sub>4</sub>/TNA was approximately 35  $\mu$ m (Fig. 2c). The BET surface areas of the 3D g-C<sub>3</sub>N<sub>4</sub>/TNA were analyzed using the automated surface area and pore size analyzer. The calculated specific surface area of the 3D g-C<sub>3</sub>N<sub>4</sub>/TNA was approximately 103.6 m<sup>2</sup> g<sup>-1</sup>.

Fig. 3a shows a representative TEM image of the as-prepared 3D g- $C_3N_4$ /TNA, in which the hollow structures with a smooth tube wall can be clearly seen. The high-resolution TEM image of the 3D g- $C_3N_4$ /TNA nanocomposites exhibited that the material



Fig. 1 Schematic of the 3D g-C<sub>3</sub>N<sub>4</sub>/TNA preparation.



Fig. 2 SEM images of the 3D g-C<sub>3</sub>N<sub>4</sub>/TNA: (a) top view, (b) bottom view, and (c) cross-sectional view.



Fig. 3 TEM image of (a) the 3D g-C<sub>3</sub>N<sub>4</sub>/TNA and (b) the g-C<sub>3</sub>N<sub>4</sub> QDs in 3D g-C<sub>3</sub>N<sub>4</sub>/TNA. (c) The size distribution of the g-C<sub>3</sub>N<sub>4</sub> QDs. (d) EDX spectrum of the 3D g-C<sub>3</sub>N<sub>4</sub>/TNA.



was crystalline with interplanar spacing of 0.353 nm (inset of Fig. 3a), which corresponded to the (101) plane of anatase TiO<sub>2</sub>. Meanwhile, the lattice spacing for g-C<sub>3</sub>N<sub>4</sub> was 0.336 nm (inset of Fig. 3b), which corresponded to the (002) plane of hexagonal g-C<sub>3</sub>N<sub>4</sub> (JCPDS 87-1526).<sup>15</sup> The size distribution of more than 90% of the g-C<sub>3</sub>N<sub>4</sub> QDs fell in the range of 7–10 nm, and the mean particle diameter was about 7.6 nm (Fig. 3c). Fig. 3d presents the EDX spectrum of the 3D g-C<sub>3</sub>N<sub>4</sub>/TNA; Ti, O, C, and N were observed in the 3D g-C<sub>3</sub>N<sub>4</sub>/TNA.

Fig. 4 shows the XRD patterns of the 3D g-C<sub>3</sub>N<sub>4</sub>/TNA. The diffraction peaks at  $2\theta$  values of  $25.33^{\circ}$ ,  $37.84^{\circ}$ ,  $48.07^{\circ}$ ,  $53.95^{\circ}$ , and  $55.11^{\circ}$  were observed, which correspond to the plane diffractions of (101), (004), (200), (105), and (211), respectively. These results correspond to the reflections of the crystalline anatase-TiO<sub>2</sub> planes (JCPDS 21-1272). Meanwhile, the characteristic peak of g-C<sub>3</sub>N<sub>4</sub> at 27.5° conforms to the (002) plane of g-C<sub>3</sub>N<sub>4</sub>.

The XPS spectra of the 3D g- $C_3N_4$ /TNA were recorded to analyze the chemical composition. The peaks corresponding to



Fig. 5 XPS spectra of the full scan for the 3D  $g-C_3N_4/TNA$  (inset: (a) C 1s and (b) N 1s).

Ti, O, C, and N were observed for the 3D g- $C_3N_4$ /TNA from Fig. 5. The Ti  $2p_{3/2}$  and Ti  $2p_{1/2}$  binding energies were observed at 458.8 eV and 464.7 eV, respectively, which indicated that the Ti<sup>4+</sup> species were present in the form of TiO<sub>2</sub> clusters.<sup>16</sup> The position of the O 1s band at  $\sim$ 529.7 eV slightly deviated from the standard position at ~530.0 eV,17 which was probably caused by the chemical configuration, and it further confirms the Ti-O bond at TiO2.18 As shown in Fig. 5a, a high resolution C 1s spectrum was recorded for the 3D g-C<sub>3</sub>N<sub>4</sub>/TNA and found to have three peaks. The peak located at 285.1 eV was due to C-C and/or adventitious carbon, and the following peaks at 286.4 and 288.4 eV could be accounted for by the C-N-C and C-(N)<sub>3</sub> groups of g-C<sub>3</sub>N<sub>4</sub>, respectively.<sup>18</sup> No chemical interaction occurred between  $g-C_3N_4$  and TNA. Fig. 5b shows the regional spectrum of N 1s for the 3D g-C<sub>3</sub>N<sub>4</sub>/TNA. The spectrum could be deconvolved into three peaks, viz. 398.9, 399.9, and 400.9 eV, which could account for the C-N=C, N-(C)<sub>3</sub>, and -NH<sub>2</sub> or =NH of the uncondensed terminal amino groups of the material, respectively.19

UV-vis diffuse reflection spectra are shown in Fig. 6 to understand the optical absorption properties of the 3D g- $C_3N_4/$ TNA. The band gap of anatase TiO<sub>2</sub> is about 3.2 eV, so its



Fig. 6 UV-vis diffuse reflection spectra of the 3D  $g-C_3N_4/TNA$ .



Fig. 7 (a) The time for  $H_2$  production and (b) the corresponding rate for the 3D g-C<sub>3</sub>N<sub>4</sub>/TNA.

absorption onset is located at ~388 nm. The band gap for g- $C_3N_4$  is 2.71 eV, corresponding to the absorption onset of ~458 nm.<sup>15</sup> In comparison with g- $C_3N_4$  and TNA, the 3D g- $C_3N_4$ /TNA has an absorption in the visible range that accompanies a red shift in the absorption edge due to the synergistic effect between TiO<sub>2</sub> and g- $C_3N_4$ .<sup>20</sup> These results indicate that the obtained 3D g- $C_3N_4$ /TNA could be employed under visible light irradiation.

In this study, 3D g-C<sub>3</sub>N<sub>4</sub>/TNA as well as sole g-C<sub>3</sub>N<sub>4</sub> was tested for photocatalytic H<sub>2</sub> production through water splitting under visible light irradiation. Time dependent production of H<sub>2</sub> and corresponding rate of H<sub>2</sub> production are given in Fig. 7. It can be seen that all the H<sub>2</sub> production increased with respect to the irradiation time (Fig. 7a). As shown in Fig. 7b, the average  $H_2$ production observed by sole g-C<sub>3</sub>N<sub>4</sub> was 52  $\mu$ mol h<sup>-1</sup> g<sup>-1</sup>. However, the 3D g-C<sub>3</sub>N<sub>4</sub>/TNA produced 243  $\mu$ mol h<sup>-1</sup> g<sup>-1</sup>, which are 4.7 times higher than that of sole g-C<sub>3</sub>N<sub>4</sub>. Such an improvement in H<sub>2</sub> production is attributed to the distinguish nanostructure of 3D g-C<sub>3</sub>N<sub>4</sub>/TNA. During photocatalytic reaction, benefiting from an attractive heterostructure between g-C<sub>3</sub>N<sub>4</sub> and TiO<sub>2</sub> leading to a unique photogenerated charge separation. Meanwhile, the highly ordered 3D TNA nanostructures can promote better dispersion of g-C<sub>3</sub>N<sub>4</sub> QDs and extend the number of active sites, which may be another factor to the increase in corresponding rate of H<sub>2</sub> production. In addition, the stability of the 3D g-C<sub>3</sub>N<sub>4</sub>/TNA was investigated in four consecutive runs of the accumulation of 64 h of irradiation. No deactivations occurred for the 3D g- $C_3N_4$ /TNA, indicating a considerable photostability for  $H_2$  production.

### 4 Conclusions

In summary, a novel 3D g- $C_3N_4$ /TNA nanostructure was developed by a simple electro-deposition of g- $C_3N_4$  QDs on TNA. The 3D g- $C_3N_4$ /TNA contributed to enhanced visible-light harvesting and reduced recombination of photogenerated electron-hole pairs due to the synergistic effect of heterojunction structure between g- $C_3N_4$  QDs and TNA, thereby leading to a high photocatalytic activity for H<sub>2</sub> production under visible light irradiation. Meanwhile, the well-ordered 3D TNA nanostructure was also beneficial to the H<sub>2</sub> production. Herein, compared with sole g- $C_3N_4$ , the average H<sub>2</sub> production increased 4.7 times using the 3D g- $C_3N_4$ /TNA nanocomposite. This simplified strategy reported here can be applied to form other 3D nanostructures, and nanoscale heterojunction structure based on  $TiO_2$  nanotube array.

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