



CrossMark
click for updates

Cite this: *RSC Adv.*, 2017, 7, 12329

High performance MWCNT–Pt nanocomposite-based cathode for passive direct methanol fuel cells†

Longjuan Pu, Liangliang Zou, Yi Zhou,* Zhiqing Zou and Hui Yang*

Reduction of the Pt loading required in cathodes is crucial for the development of passive direct methanol fuel cells (DMFCs). Herein, a novel membrane electrode assembly (MEA) that utilizes a MWCNT–Pt nanocomposite cathodic catalyst layer (CCL) with a 3D network structure is shown to require significantly less Pt loading. With a CCL Pt loading of 0.5 mg cm^{-2} , the maximum power density of the prepared DMFC is $19.2 \pm 0.4 \text{ mW cm}^{-2}$ using 2.0 M methanol solution at $25 \pm 1 \text{ }^\circ\text{C}$, which is higher than that of the power density by a conventional MEA with twice the Pt loading (1.0 mg cm^{-2}). Electrochemical tests show that the structure of the CCL decreases the charge transfer resistance of the cathode reaction and greatly increases the cathode catalyst utilization in comparison with the conventional MEA. The enhanced MEA performance is attributed to the discontinuous distributions of the Pt MWCNT structures and the formation of a cross-twined network within the CCL. This study could provide a promising way to reduce the cost of future commercialized DMFCs.

Received 26th December 2016

Accepted 9th February 2017

DOI: 10.1039/c6ra28703d

rsc.li/rsc-advances

1. Introduction

Passive direct methanol fuel cells (DMFCs) are promising power sources for portable applications due to their high energy density, small amount of pollutant byproducts, simplicity and convenience.^{1–6} However, the high noble metal content (*i.e.* Pt) that is required in membrane electrode assemblies (MEAs) hinders the development of passive DMFCs.^{7–9} The development of new methods to prepare Pt-based nanocatalysts with high activity is desirable to reduce the need for high noble metal loading.^{10–14}

Previous work has been done to reduce the need for high catalyst loadings in passive DMFCs. Some have been to develop low Pt content or Pt-free catalysts that have high activities,^{10,15–17} to use nanomaterials as the catalyst or catalyst support,^{17–19} and to use an ordered nanostructured diffusive layer or catalyst layer.^{20–25} These approaches have effectively reduced the anodic catalyst loading from $4.0 \text{ mg}_{(\text{Pt-Ru})} \text{ cm}^{-2}$ to $1.0 \text{ mg}_{(\text{Pt-Ru})} \text{ cm}^{-2}$ for passive DMFCs. However, the techniques to reduce the Pt loading for the cathode of DMFCs are still limited.

In an attempt to modify the morphology and architecture of Pt-based catalysts for increasing catalyst utilization and eventually reducing the catalyst loading, various Pt nanocrystals and Pt-based nanostructures have been reported, such as 0D Pt nanoparticles,^{26–29} 1D Pt nanostructures (nanowires,^{30,31} nanorods,^{32–34} nanotubes^{12,19,35–38}), 2D Pt nanoscale membranes,³⁹ and 3D Pt

nano-assemblies.⁴⁰ Pt-based carbon nanotubes (CNT) composites for fuel cells have been of great interest due to the unique electronic, thermal, and mechanical properties of CNTs.^{41–51} For examples, Fang *et al.* have used functionalized MWCNTs to fabricate MWCNT-supported, high Pt loading electrocatalyst for proton exchange membrane fuel cells (PEMFCs).⁴⁵ Tian *et al.* used vertically aligned carbon nanotubes as highly ordered catalyst supports to prepare ultra-low Pt loading MEA for applications in PEMFCs.⁴⁸ These MEAs have shown excellent performance with Pt loading as low as $35 \text{ } \mu\text{g cm}^{-2}$, which is comparable to the commercial Pt catalyst on carbon powder loading level ($400 \text{ } \mu\text{g cm}^{-2}$). A low-defect MWCNT–Pt nanocomposite has been synthesized by a soft chemical route and showed promising electrocatalytic activity and CO-poisoning tolerance for potential use in DMFCs.⁴⁹ Yang *et al.* reported the polymer-wrapped MWCNTs as support for Pt nanoparticles as the cathode catalyst, leading to an improved MEA performance of a DMFC.^{50,51}

In our previous work, a Pt nanorod assembly-based double-layered cathode was prepared which required a significant reduction in catalyst loading (up to 50% less) by using platinum carbonyl complexes as the cathode catalyst raw material compared to conventional cathode.⁵² Herein, we report the development of a method to fabricate a cross-twined MWCNT–Pt nanocomposite-based cathodic catalyst layer (CCL) for passive DMFCs by controlling the mass ratio of Pt and MWCNT in a solution containing platinum carbonyl complexes and MWCNTs. Compared with the maximum power density, we found that a 1 : 2 mass ratio of Pt and MWCNTs shows the best MEA performance, including the power density of $19.2 \pm 0.4 \text{ mW cm}^{-2}$ using 2.0 M methanol solution and a 0.5 mg cm^{-2} Pt

Shanghai Advanced Research Institute, Chinese Academy of Sciences, Shanghai 201210, China. E-mail: yangh@sari.ac.cn; zhouy@sari.ac.cn; Fax: +86-21-20321112

† Electronic supplementary information (ESI) available. See DOI: 10.1039/c6ra28703d



loading within the MWCNT–Pt nanocomposite-based CCL. The formation of the MWCNT–Pt nanocomplex-based CCL with a 3D network structure leads to a significant decrease in Pt loading, while maintaining a similar performance compared to a conventional MEA. Therefore, we envisioned that this method may provide a new way to reduce Pt loading and potentially help lead to the commercialization of DMFCs.

2. Experimental

2.1 Materials

Sodium acetate (CH_3COONa , AR), methanol (CH_3OH , AR), ethanol (EtOH, AR), and isopropyl alcohol ($(\text{CH}_3)_2\text{CHOH}$, AR) were supplied by Sinopharm Chemical Reagent Co., Ltd. Sodium chloroplatinate ($\text{Na}_2\text{PtCl}_6 \cdot 6\text{H}_2\text{O}$) was purchased from Alfa Aesar. Multi-walled carbon nanotube solution (2.96 wt% MWCNT in methanol, 40–50 nm nanotube diameter, 10–30 μm nanotube length) was purchased from Beijing DK nano S&T Ltd. Nafion® 115 membrane, Nafion® solution (5.3 wt%), and PTFE solution (60 wt%) were purchased from DuPont. Pt–Ru/C (Pt loading: 40%, Ru loading: 20%; HiSpec 10 000), Pt/C (Pt loading: 60 wt%, HiSpec 9100), Pt–Ru black (atomic ration 1 : 1, HiSpec 6000), and Pt black (HiSpec 1000) were purchased from Johnson Matthey. The gas diffusion layers (GDLs) for DMFCs were prepared with carbon paper (TGPH060, 20 wt% PTFE Toray), XC-72R carbon black (Cabot), and 10 wt% PTFE solution (diluted from 60 wt% PTFE solution). All reactants were used as received.

2.2 Formation of MWCNT–Pt nanocomposite

Platinum carbonyl complex solutions with a platinum concentration of 32.0 mM were prepared according to our previous work.⁵² Platinum carbonyl complex solutions and multi-walled carbon nanotube solutions were mixed with certain Pt/MWCNT mass ratios (3 : 1, 1 : 1, 1 : 2, 1 : 3) in a glove box under an argon atmosphere. The as-prepared Pt/MWCNT mixed solutions were quickly sprayed onto a substrate such as a silicon wafer under air. The MWCNT–Pt nanocomposite was prepared by evaporation of methanol, drying overnight at 50 °C, washing with ultrapure water several times and drying under air.

2.3 MEA preparation

A conventional MEA was prepared according to the literature with a Nafion® 115 membrane, a cathode with 1.0 $\text{mg}_{(\text{Pt})} \text{cm}^{-2}$ and an anode with 4.0 $\text{mg}_{(\text{Pt-Ru})} \text{cm}^{-2}$ using a Sono-Tek Ultrasonic Atomizing Nozzle System.⁵² In this work, conventional anodes with 4.0 $\text{mg}_{(\text{Pt-Ru})} \text{cm}^{-2}$ were used. The cathode based on the MWCNT–Pt nanocomposite was prepared by spreading a certain volume of as-prepared Pt/MWCNT mixed solution (mass ratio of Pt and MWCNT were 3 : 1, 1 : 1, 1 : 2, and 1 : 3, respectively) onto a cathode GDL as fabricated above under air. The solvent was allowed to evaporate under air, and the cathode was dried overnight at 50 °C, washed several times with ultrapure water, and dried in air to yield a new-structured cathode. The Pt loading in the new-structured cathode was 1.0 or 0.5 $\text{mg}_{(\text{Pt})} \text{cm}^{-2}$. A 1.0 wt% Nafion® solution was prepared by diluting the original Nafion® solution with a 1 : 1 (v/v) mixture of isopropanol and water and sonicating. Then, the 1.0

wt% Nafion® solution followed by a 20 wt% Nafion® solution was sprayed onto the new-structured cathode before preparation of the MEA. MEAs with an active area of $1 \times 1 \text{ cm}^2$ were fabricated by hot-pressing method (130 °C, 2 MPa) for 3 min and subsequently stored in a 2.0 M methanol solution before the test.

2.4 Physical and electrochemical characterization

Scanning electron microscopy (SEM) images were collected at an electron acceleration voltage of 10.0 kV using a scanning electron microscope (FESEM, Hitachi S-4800). Transmission electron microscopy (TEM) images and high resolution transmission electron microscopy (HRTEM) images were taken using a JEOL 2100F at an accelerating voltage of 200 kV.

Polarization curves of the passive DMFCs were recorded on an Arbin FCT testing system (Arbin Instrument Inc. USA) in an air-breathing mode and fed with 5 mL of 2.0 M methanol. A two minute waiting time was used to obtain a stable voltage for each discharging current point along the polarization curve. The stability of the DMFC was evaluated at a constant current density of 40 mA cm^{-2} for a period of time by discontinuously feeding fuel with 2.0 M methanol solution roughly every 12 h. Cyclic voltammograms were obtained using a Solartron SI 1287 Potentiostat/Galvanostat at a scan rate of 20 mV s^{-1} between 0.05 and 0.75 V. The cathode fed with humidified N_2 formed the working electrode and the anode fed with humidified H_2 was denoted as the counter electrode and reference electrode. Electrochemical impedance spectra (EIS) were obtained using a Solartron SI 1287 & 1255B instrument at a frequency range between 100 kHz and 0.01 Hz, and the cathode was applied as the working electrode whereas the anode was used as the counter electrode and reference electrode during the test. The data were obtained at a constant current density of 40 mA cm^{-2} and the amplitude of the sinusoidal voltage signal was 10 mV.⁵²

3. Results and discussion

New nanostructured cathode-based MEAs with varying mass ratios of Pt and MWCNTs (MEAs with cathode based on mass

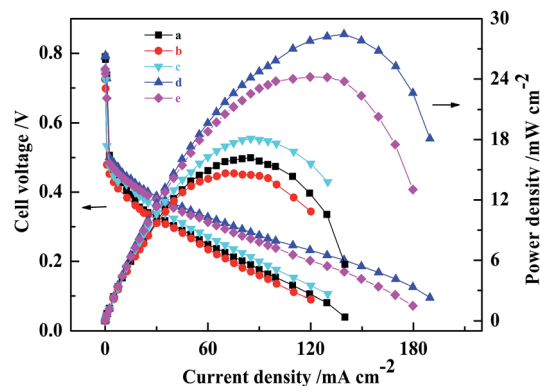


Fig. 1 Polarization curves of five MEAs at the loading of 1.0 $\text{mg}_{(\text{Pt})} \text{cm}^{-2}$ and 4.0 $\text{mg}_{(\text{Pt-Ru})} \text{cm}^{-2}$, fed with 2.0 M methanol solution at 25 ± 1 °C. (a) Conventional MEA, (b–e) MEAs with CCL based on the mass ratio (3 : 1, 1 : 1, 1 : 2 and 1 : 3) of Pt and MWCNTs.



ratio of Pt and MWCNT of 3 : 1, 1 : 1, 1 : 2, and 1 : 3 were named as MEA_{3:1}, MEA_{1:1}, MEA_{1:2} and MEA_{1:3}, respectively) were fabricated and the results of the performance and polarization curves of the fabricated MEAs with CCLs of different mass ratios are shown in Fig. 1. When the Pt loading was 1.0 mg_(Pt) cm⁻² and 4.0 mg_(PtRu) cm⁻², the maximal power density of all passive DMFCs tested was achieved using a CCL with a 1 : 2 mass ratio of Pt and MWCNTs and found to be 28.5 ± 0.2 mW cm⁻². The

MEA_{3:1}, MEA_{1:1}, and MEA_{1:3} and conventional MEA were found to give power densities of 14.7 ± 0.3, 18.1 ± 0.2, 24.2 ± 0.2, and 16.2 ± 0.3 mW cm⁻², respectively. The use of a nanostructured cathode with a suitable Pt : MWCNT mass ratio led to enhanced MEA performance. In addition, at the low current density region (≤30 mA cm⁻²), MEA_{1:2} and MEA_{1:3} exhibited higher power densities than those of conventional MEA, MEA_{3:1} and MEA_{1:1}. In the low current density region, the polarization is mainly

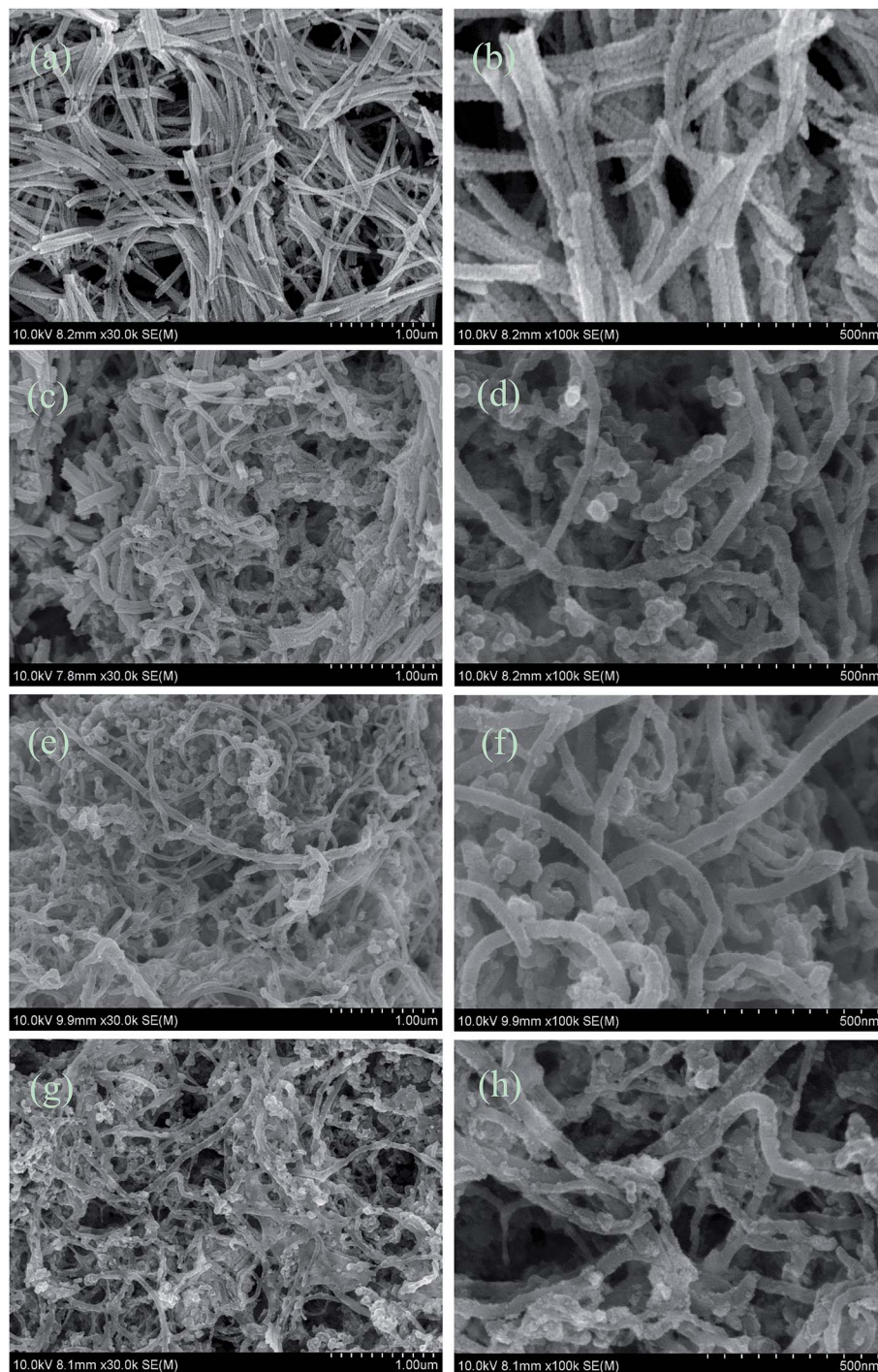


Fig. 2 SEM images of cathode catalyst layers based on various mass ratios of Pt and MWCNTs: (a) MEA_{3:1}, (c) MEA_{1:1}, (e) MEA_{1:2}, (g) MEA_{1:3}. (b, d, f and h) are enlarged images corresponding to the images of (a, c, e and g).



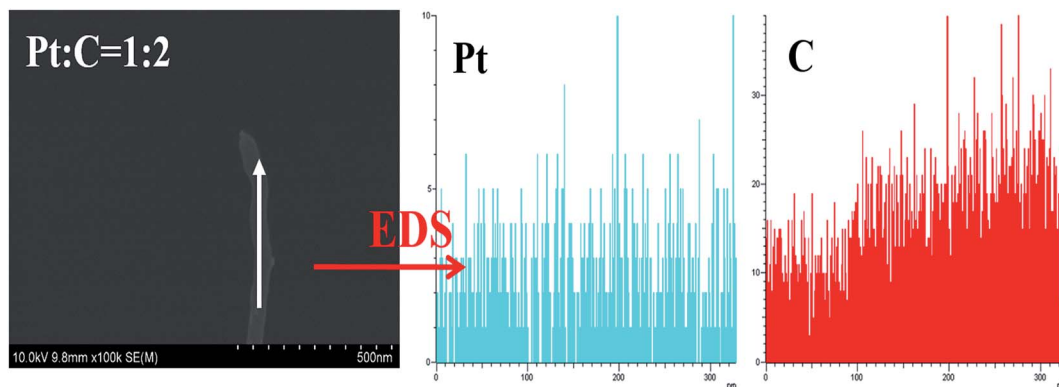


Fig. 3 SEM image and EDS line scanning of MWCNT–Pt nanocomposite.

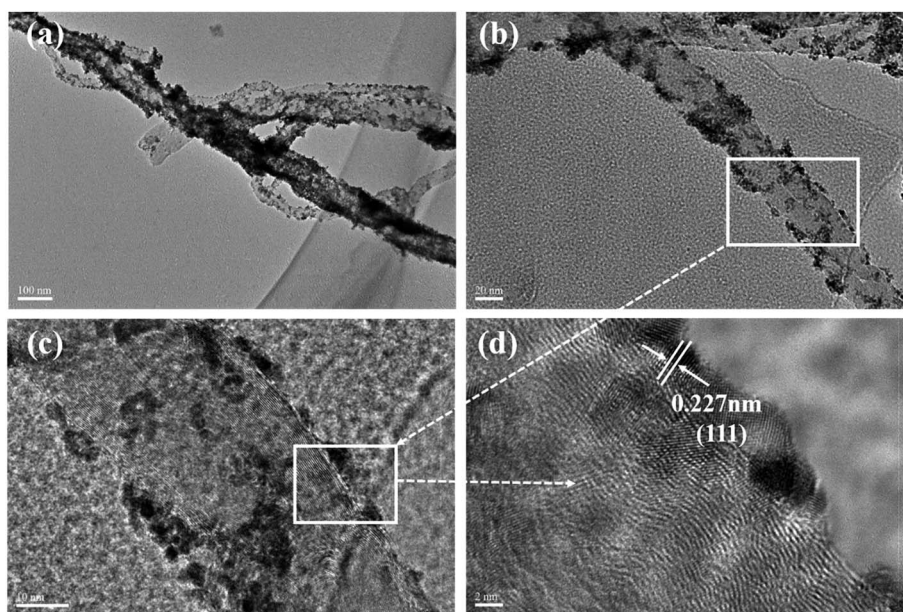


Fig. 4 TEM and HRTEM images of MWCNT–Pt nanocomposite: (a) TEM image of MWCNT–Pt nanocomposite, (b) TEM image of single MWCNT–Pt nanocomposite, (c and d) HRTEM image of MWCNT–Pt nanocomposite.

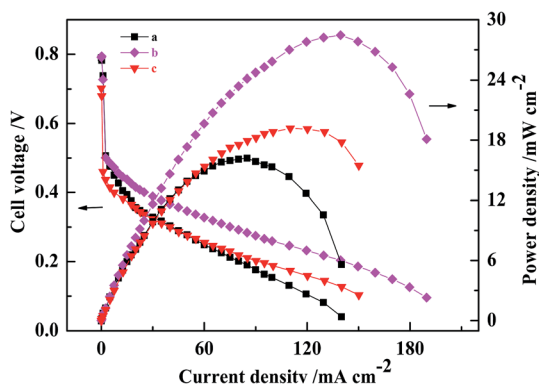


Fig. 5 Polarization curves of MEAs fed with 2.0 M methanol solution at 25 ± 1 °C. (a) The conventional MEA at the loading of $1.0 \text{ mg}_{(\text{Pt})} \text{ cm}^{-2}$, (b and c) $\text{MEA}_{1:2}$ at the loading of 1.0 and $0.5 \text{ mg}_{(\text{Pt})} \text{ cm}^{-2}$, respectively.

related to catalytic activity, which demonstrates that $\text{MEA}_{1:2}$ and $\text{MEA}_{1:3}$ have higher catalytic activity than the other MEAs. In the secondary current density region ($30\text{--}130 \text{ mA cm}^{-2}$), $\text{MEA}_{1:2}$ and $\text{MEA}_{1:3}$ showed a smaller ohmic resistance compared to the other MEAs. This is attributed to the excellent electrical conductivity of MWCNTs. However, in the high current density region ($\geq 130 \text{ mA cm}^{-2}$), there is a rapid decrease in cell voltage for the conventional MEA compared with that of $\text{MEA}_{1:2}$. This is likely a result of having different types of cathode mass transport. In short, the great performance of $\text{MEA}_{1:2}$ is an optimized result from the new-structured CCL.

SEM top view images of the CCLs based on various mass ratios of Pt and MWCNTs were taken to explain the MEA performance shown in Fig. 1. The CCLs of $\text{MEA}_{3:1}$ (Fig. 2a and b), $\text{MEA}_{1:1}$ (Fig. 2c and d), $\text{MEA}_{1:2}$ (Fig. 2e and f), and $\text{MEA}_{1:3}$ (Fig. 2g and h) were imaged. Fig. 2a and b displays a CCL that consists of Pt nanorods and a rare MWCNT.⁵² Fig. 2c and



d shows a CCL that consists of Pt nanorods and a higher MWCNT content. Fig. 2e and h shows a CCL that consists of uniformly network-distributed MWCNTs without Pt nanorods. In general, the higher the MWCNT content in the CCL (Fig. 2), the higher amount of cross-twined MWCNTs in the CCL. This type of MWCNT offers an unobstructed mass transport and a higher degree of catalyst utilization in the cathode, and eventually results in a significant improvement in performance as shown in Fig. 1. Moreover, the enlarged image of the MWCNTs in Fig. 2f and h shows the rough surface morphology caused by nano-particle clusters on the MWCNT. The thinner CCL in MEAs with Pt : MWCNT mass ratios of MEA_{1:2} and MEA_{1:3} leads to a shorter channel for mass transportation and good performance values. The SEM image and EDS line scanning of a single MWCNT from MEA_{1:2} was obtained as shown in Fig. 3. The carbon element continuously exists and platinum discontinuously exists along the nanotube axis (diameter: ~45 nm). According to the nanotube data and the element distribution, the nanotube is considered a MWCNT–Pt nanocomposite with Pt nano-particles discontinuously distributed on the MWCNTs. TEM & HRTEM images and particle size distributions of different mass ratios of Pt and MWCNT are shown in Fig. S1 of the ESI.† The average particle sizes for all the samples are around 2 nm. According to the HRTEM image in Fig. 4d, the small dots in Fig. S1† are Pt nanoparticles. When the mass ratios of Pt and MWCNT are 3 : 1 and 1 : 1, both Pt nanoparticles wrapped MWCNTs and Pt nanoparticles are clearly observed (Fig. S1a and S1b†) because of excess Pt content. When the mass ratios of Pt and MWCNT are 1 : 2 and 1 : 3, only Pt nanoparticles wrapped MWCNTs are found (Fig. S1c and S1d†), leading to an improved Pt utilization, which could explain the enhanced performance of the MEA_{1:2} and MEA_{1:3}.

The MWCNT–Pt CCLs were fabricated by depositing a platinum carbonyl complex solution and a MWCNT solution on a silicon wafer and evaporation of the solvent under air. The TEM images further verified the structure of a MWCNT–Pt CCL (Fig. 4). The nanotubes in Fig. 4a and b have a diameter in the range of 40–50 nm which matches the diameter of the MWCNTs. The 2–4 nm diameter nano-particles are well distributed on the MWCNTs (Fig. 4c). A lattice spacing of 0.227 nm (Fig. 4d, HRTEM) for nano-particles on the MWCNTs is attributed to the Pt (111) lattice fringe. Thus, Pt nano-particles are distributed on the MWCNT surface along the axis which leads to a higher catalyst utilization compared to agglomerated Pt nanorods and commercial Pt/C. This clearly explains why MEA_{1:2} and MEA_{1:3} exhibit better performance than other

MEAs. Owing to the best MEA performance of MEA_{1:2}, further work focused on the MEA.

A comparison of the polarization curves for the three passive DMFCs with CCLs fabricated by conventional catalyst and by MWCNT–Pt CCL with varying Pt loadings is shown in Fig. 5. When Pt loading is decreased to 0.5 mg cm⁻², the maximum power density of the DMFC based on Pt–MWCNT CCL is 19.2 ± 0.4 mW cm⁻², which is higher than that of a conventional MEA with a Pt loading of 1.0 mg cm⁻². This shows that using MWCNT–Pt nanocomposite-based DMFCs require substantially less Pt loading compared to DMFCs based on conventional MEAs. At a catalyst loading of 1.0 mg_(Pt) cm⁻², the ohmic resistance of the MWCNT–Pt MEA was 427 mΩ while that of the conventional MEA was 630 mΩ (Table 1). This is due to the excellent electrical conductivity of MWCNT. This reduced ohmic resistance leads to better MEA performance.

Possible mechanisms for the improved performance of the DMFCs based on MWCNT–Pt nanocomposite were investigated further. Cyclovoltammograms were collected in order to quantify the ESAs of the cathodes at the same loading of 1.0 mg_(Pt) cm⁻² (Fig. 6). The ESAs were calculated by integrating the H adsorption/desorption areas⁵³ of the cathodes used in the conventional CCL and the MWCNT–Pt CCLs, and found to be ca. 24.1 ± 0.4 and 43.6 ± 0.5 m² g⁻¹, respectively (Table 1). These values indicate that the MWCNT–Pt CCLs utilize the catalyst more efficiently and are consistent with the MEA performance as shown in Fig. 5. The performance enhancement is largely attributed to an increase in ESA when there is a MWCNT–Pt nanocomposition in the CCL.

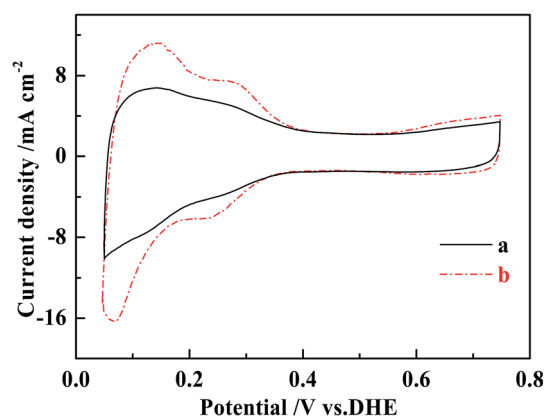


Fig. 6 Cyclic voltammograms of the cathodes of MEAs with the loading of 1.0 mg_(Pt) cm⁻² at a scan rate of 20 mV s⁻¹ at 25 ± 1 °C. (a) The conventional MEA, (b) MEA_{1:2}.

Table 1 Maximal power densities, ohmic resistance, and ESAs of the three MEAs

Cathode structure	Cathode loading/ mg cm ⁻²	Maximum power density/mW cm ⁻²	Ohmic resistance/mΩ	ESAs by H desorption/m ² g _(Pt) ⁻¹
Conventional MEA	1.0	16.2 ± 0.3	630	24.1 ± 0.4
MEA _{1:2}	1.0	28.5 ± 0.2	427	43.6 ± 0.5
	0.5	19.2 ± 0.4	601	—



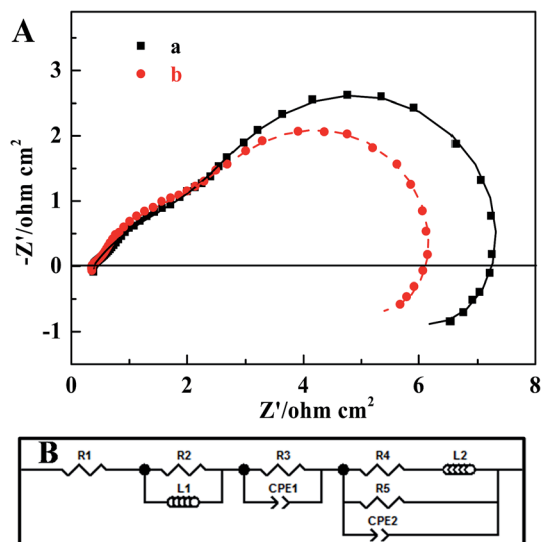


Fig. 7 Electrochemical impedance spectra of the MEAs ($1.0 \text{ mg}_{(\text{Pt})} \text{ cm}^{-2}$) operated at a constant current density of 40 mA cm^{-2} fed with 2.0 M methanol solution at $25 \pm 1^\circ \text{C}$ with fitted curves (A) based on an equivalent circuit model (B). (a) The conventional MEA, (b) $\text{MEA}_{1:2}$.

The EIS of two MEAs were collected in order to identify the effect that the MWCNT–Pt nanocomposite-based CCL had on the impedance of the respective MEAs. Fig. 7 shows the MEA impedance spectra and fitted curves based on an equivalent circuit model, as described in our previous work.⁵² The marked points are the experimental data and the solid lines are the fitted curves. The fitted parameters are provided in Table 2. From these spectra, it is clear that the charge transfer resistance (R_3 ; $2.555 \Omega \text{ cm}^2$) of the MWCNT–Pt nanocomposite-based MEAs is smaller than that of the conventional MEA (R_3 ; $2.923 \Omega \text{ cm}^2$) at the same loading ($1.0 \text{ mg}_{(\text{Pt})} \text{ cm}^{-2}$). This improved MEA performance as shown in Fig. 5 may be a result of the cross-twined MWCNT–Pt nanocomposite in the CCL. This nano-composition provides more catalytically active sites and an unobstructed mass transport for the cathode.

The application of MWCNT–Pt CCLs in MEAs significantly improves the performance of the corresponding passive

Table 2 CPE-based equivalent circuit model fitted parameters of the MEAs at the loading of $1.0 \text{ mg}_{(\text{Pt})} \text{ cm}^{-2}$ and $4.0 \text{ mg}_{(\text{PtRu})} \text{ cm}^{-2}$ operated at a constant current density of 40 mA cm^{-2} and at $25 \pm 1^\circ \text{C}$

Parameter	Conventional MEA	$\text{MEA}_{1:2}$
R_1 ($\Omega \text{ cm}^2$)	0.372	0.351
R_2 ($\Omega \text{ cm}^2$)	0.263	0.254
L_1 (H cm^{-2})	6.021×10^{-8}	6.962×10^{-8}
R_3 ($\Omega \text{ cm}^2$)	2.923	2.555
$\text{CPE}_{1-\text{T}}$ (F cm^{-2})	0.100	0.090
$\text{CPE}_{1-\text{P}}$	0.564	0.663
R_4 ($\Omega \text{ cm}^2$)	3.630	3.200
L_2 (H cm^{-2})	95.252	89.403
R_5 ($\Omega \text{ cm}^2$)	4.441	3.532
$\text{CPE}_{2-\text{T}}$ (F cm^{-2})	0.340	0.430
$\text{CPE}_{2-\text{P}}$	0.991	0.993

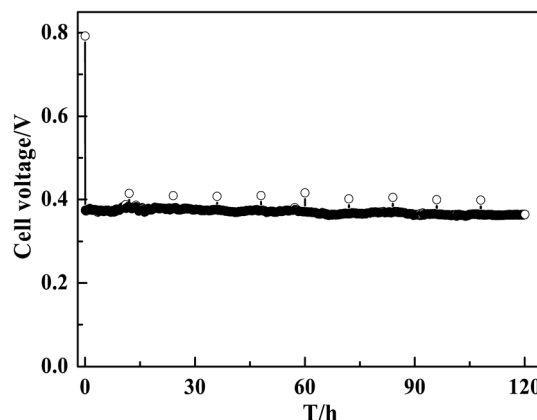


Fig. 8 Discharging curve of a DMFC with $\text{MEA}_{1:2}$ ($1.0 \text{ mg}_{(\text{Pt})} \text{ cm}^{-2}$) at a constant current of 40 mA cm^{-2} .

DMFCs. Therefore, the durability of these devices over long-term operation was tested. As shown in Fig. 8, a steady cell performance was observed for 120 h at a current density of 40 mA cm^{-2} , which indicates that the stability of this MEA is suitable for practical applications.

4. Conclusions

Newly structured cathodes based on Pt and MWCNTs (1 : 2 by mass) were prepared and used in the fabrication of MEAs, with the best performing MEAs having Pt loadings of $1.0 \text{ mg}_{(\text{Pt})} \text{ cm}^{-2}$ and $4.0 \text{ mg}_{(\text{PtRu})} \text{ cm}^{-2}$. The 3D network structure of the nanocomposite-based CCLs used in these MEAs leads to a decrease in charge transfer resistance of the cathode reaction, a large increase in the utilization of cathode catalyst, and ultimately, an enhanced performance in the DMFC. The discontinuous distribution of Pt nano-particles on the MWCNT and the cross-twined networks within the CCL may also contribute. It is highly desirable to develop new ways to reduce the cost and expedite the commercialization of DMFC.

Acknowledgements

We would like to thank the National '863' High-Tech. Research Programs of China (2015AA043503-2) and the National Natural Science Foundation of China (51506213, 21533005) for support of this work.

References

- 1 A. Heinzl and V. M. Barragán, *J. Power Sources*, 1999, **84**, 70–74.
- 2 E. Gulzow, T. Kaz, R. Reissner, H. Sander, L. Schilling and M. von Bradke, *J. Power Sources*, 2002, **105**, 261–266.
- 3 C. Y. Chen and P. Yang, *J. Power Sources*, 2003, **123**, 37–42.
- 4 L. Jianguo, S. Gongquan, Z. Fengliang, W. Guoxiong, Z. Gang, C. Likang, Y. Baolian and X. Qin, *J. Power Sources*, 2004, **133**, 175–180.



- 5 D. Dunwoody and J. Leddy, *Electrochem. Soc. Interface*, 2005, **14**, 37–39.
- 6 S. K. Kamarudin, W. R. W. Daud, S. L. Ho and U. A. Hasran, *J. Power Sources*, 2007, **163**, 743–754.
- 7 H. I. Joh, S. Y. Hwang, J. H. Cho, T. J. Ha, S.-K. Kim, S. H. Moon and H. Y. Ha, *Int. J. Hydrogen Energy*, 2008, **33**, 7153–7162.
- 8 W. W. Yang and T. S. Zhao, *Electrochim. Acta*, 2007, **52**, 6125–6140.
- 9 F. Xie, C. Chen, H. Meng and P. K. Shen, *Fuel Cells*, 2007, **7**, 319–322.
- 10 Z.-B. Wang, G.-P. Yin, P.-F. Shi, B.-Q. Yang and P.-X. Feng, *J. Power Sources*, 2007, **166**, 317–323.
- 11 H. M. Chen, R. S. Liu, M. Y. Lo, S. C. Chang, L. D. Tsai, Y. M. Peng and J. F. Lee, *J. Phys. Chem. C*, 2008, **112**, 7522–7526.
- 12 S. Jiang, Y. Ma, G. Jian, H. Tao, X. Wang, Y. Fan, Y. Lu, Z. Hu and Y. Chen, *Adv. Mater.*, 2009, **21**, 4953–4956.
- 13 K. Gong, F. Du, Z. Xia, M. Durstock and L. Dai, *Science*, 2009, **323**, 760–764.
- 14 H.-W. Liang, S. Liu, J.-Y. Gong, S.-B. Wang, L. Wang and S.-H. Yu, *Adv. Mater.*, 2009, **21**, 1850–1854.
- 15 X. Wang, J. Liao, C. Liu, W. Xing and T. Lu, *Electrochem. Commun.*, 2009, **11**, 198–201.
- 16 K. W. Park, B. K. Kwon, J. H. Choi, I. S. Park, Y. M. Kim and Y. E. Sung, *J. Power Sources*, 2002, **109**, 439–445.
- 17 Y.-J. Wang, D. P. Wilkinson and J. Zhang, *Chem. Rev.*, 2011, **111**, 7625–7651.
- 18 J. Cao, C. Du, S. C. Wang, P. Mercier, X. Zhang, H. Yang and D. L. Akins, *Electrochem. Commun.*, 2007, **9**, 735–740.
- 19 C. H. Wang, H. Y. Dub, Y. T. Tsai, C. P. Chen, C. J. Huang, L. C. Chen, K. H. Chen and H. C. Shih, *J. Power Sources*, 2007, **171**, 55–62.
- 20 T. Yuan, Z. Zou, M. Chen, Z. Li, B. Xia and H. Yang, *J. Power Sources*, 2009, **192**, 423–428.
- 21 H. Wu, T. Yuan, Q. Huang, H. Zhang, Z. Zou, J. Zheng and H. Yang, *Electrochim. Acta*, 2014, **141**, 1–5.
- 22 Q. Huang, J. Jiang, J. Chai, T. Yuan, H. Zhang, Z. Zou, X. Zhang and H. Yang, *J. Power Sources*, 2014, **262**, 213–218.
- 23 P. Chen, H. Wu, T. Yuan, Z. Zou, H. Zhang, J. Zheng and H. Yang, *J. Power Sources*, 2014, **255**, 70–75.
- 24 M. Michel, A. Taylor, R. Sekol, P. Podsiadlo, P. Ho, N. Kotov and L. Thompson, *Adv. Mater.*, 2007, **19**, 3859–3864.
- 25 M.-L. Lin, C.-C. Huang, M.-Y. Lo and C.-Y. Mou, *J. Phys. Chem. C*, 2008, **112**, 867–873.
- 26 H. Lee, S. E. Habas, S. KweSkin, D. Butcher, G. A. Somorjai and P. Yang, *Angew. Chem., Int. Ed.*, 2006, **45**, 7824–7828.
- 27 T. K. Sau and A. L. Rogach, *Adv. Mater.*, 2010, **22**, 1781–1804.
- 28 W.-N. Wang, W.-J. An, B. Ramalingam, S. Mukherjee, D. M. Niedzwiedzki, S. Gangopadhyay and P. Biswas, *J. Am. Chem. Soc.*, 2012, **134**, 11276–11281.
- 29 J. Suntivich, Z. Xu, C. E. Carlton, J. Kim, B. Han, S. W. Lee, N. Bonnet, N. Marzari, L. F. Allard, H. A. Gasteiger, K. Hamad-Schifferli and Y. Shao-Horn, *J. Am. Chem. Soc.*, 2013, **135**, 7985–7991.
- 30 E. P. Lee, Z. Peng, D. M. Cate, H. Yang, C. T. Campbell and Y. Xia, *J. Am. Chem. Soc.*, 2007, **127**, 10634–10635.
- 31 C. Koenigsmann, W.-p. Zhou, R. R. Adzic, E. Sutter and S. S. Wong, *Nano Lett.*, 2010, **10**, 2806–2811.
- 32 J. Mao, X. Cao, J. Zhen, H. Shao, H. Gu, J. Lu and J. Y. Ying, *J. Mater. Chem.*, 2011, **21**, 11478–11481.
- 33 L. Yang, X. Song, M. Qi, L. Xia and M. Jin, *J. Mater. Chem. A*, 2013, **1**, 7316.
- 34 C. Li, T. Sato and Y. Yamauchi, *Angew. Chem., Int. Ed.*, 2013, **52**, 8050–8053.
- 35 T. Kijima, T. Yoshimura, M. Uota, T. Ikeda, D. Fujikawa, S. Mouri and S. Uoyama, *Angew. Chem., Int. Ed.*, 2004, **43**, 228–232.
- 36 C. Chen and P.-k. Shen, *Battery Bimon.*, 2009, **39**, 335–337.
- 37 T. V. Reshchenko, H.-T. Kim and H.-J. Kweon, *Electrochim. Acta*, 2008, **53**, 3043–3049.
- 38 S. Takenaka, H. Miyamoto, Y. Utsunomiya, H. Matsune and M. Kishida, *J. Phys. Chem. C*, 2014, **118**, 774–783.
- 39 B. Y. Xia, H. B. Wu, Y. Yan, X. W. Lou and X. Wang, *J. Am. Chem. Soc.*, 2013, **135**, 9480–9485.
- 40 B. Y. Xia, W. T. Ng, H. B. Wu, X. Wang and X. W. Lou, *Angew. Chem., Int. Ed.*, 2012, **51**, 7213–7216.
- 41 S. Wang, S. P. Jiang, T. J. White, J. Guo and X. Wang, *Adv. Funct. Mater.*, 2010, **20**, 1368.
- 42 S. Wang, F. Yang, S. P. Jiang, S. Chen and X. Wang, *Electrochem. Commun.*, 2010, **12**, 1646–1649.
- 43 S. Wang, S. P. Jiang, T. J. White, J. Guo and X. Wang, *J. Phys. Chem. C*, 2009, **113**, 18935–18945.
- 44 W. Zhang, A. I. Minett, M. Gao, J. Zhao, J. M. Razal, G. G. Wallace, T. Romeo and J. Chen, *Adv. Energy Mater.*, 2011, **1**, 671–677.
- 45 B. Fang, M.-S. Kim, J. H. Kim, M. Y. Song, Y.-J. Wang, H. Wang, D. P. Wilkinson and J.-S. Yu, *J. Mater. Chem.*, 2011, **21**, 8066–8073.
- 46 Y.-J. Wang, B. Fang, H. Li, X. T. Bi and H. Wang, *Prog. Mater. Sci.*, 2016, **82**, 445–498.
- 47 Y.-J. Wang, N. Zhao, B. Fang, H. Li, X. T. Bi and H. Wang, *Chem. Rev.*, 2015, **115**, 3433–3467.
- 48 Z. Q. Tian, S. H. Lim, C. K. Poh, Z. Tang, Z. Xia, Z. Luo, P. K. Shen, D. Chua, Y. P. Feng, Z. Shen and J. Lin, *Adv. Energy Mater.*, 2011, **1**, 1205–1214.
- 49 H. Huang, D. Sun and X. Wang, *J. Phys. Chem. C*, 2011, **115**, 19405–19412.
- 50 Z. Yang and N. Nakashima, *Sci. Rep.*, 2015, **5**, 12236.
- 51 Z. Yang and N. Nakashima, *ChemCatChem*, 2016, **8**, 600–606.
- 52 L. Pu, H. Zhang, T. Yuan, Z. Zou, L. Zou, X.-M. Li and H. Yang, *J. Power Sources*, 2015, **276**, 95–101.
- 53 A. N. Correia, L. H. Mascaró, S. A. S. Machado and L. A. Avaca, *Electrochim. Acta*, 1997, **42**, 493–495.

