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# A promising vanadium sulfide counter electrode for efficient dye-sensitized solar cells

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In this study, we demonstrated the synthesis of vanadium sulfide ( $VS_2$ ) via an *in situ* hydrothermal route, which was subsequently employed as a counter electrode (CE) for Pt-free dye-sensitized solar cells (DSSCs) for the first time. It was demonstrated from scanning electron microscopy that the size of  $VS_2$  increased with the increasing temperature, and the morphology was also affected by temperature. Extensive electrochemical performance analysis, including cyclic voltammetry, electrochemical impedance, and Tafel polarization, revealed that the  $VS_2$  CE possesses a high electrocatalytic activity for the reduction of triiodide to iodide and a low charge-transfer resistance at the electrolyte/CE interface. The DSSC based on the  $VS_2$  CE exhibited a conversion efficiency of 6.24% under an illumination of 100 mW cm<sup>-2</sup> as compared to the DSSC based on the Pt CE.

## 1. Introduction

Dye-sensitized solar cells (DSSCs) have received extensive interest due to their facile fabrication, sustainability, low-cost, and environmentally friendliness.<sup>1-4</sup> The typical structure of a DSSC consists of  $TiO_2$  nanocrystallines as the photoanode, dyes, and a platinum (Pt)-coated tin oxide transparent (FTO) substrate as the counter electrode (CE) fabricated with an  $I^-/I_3^-$  redox couple liquid electrolyte.<sup>5</sup> Although Pt is one of the most selected materials for catalyzing the reduction of  $I_3^-$  to  $I^-$  due to its superior electrocatalytic ability, stability, and conductivity, as a noble metal, its high cost restricts the scale up production for DSSCs. To resolve this issue, many researchers are concentrating on CE catalytic materials, including carbon-based materials, metal sulfides, nitrides, polymers, and oxides,<sup>8-10</sup> with a high conductivity, large specific surface areas, and a good catalytic ability.<sup>6,7</sup> Among these, metal sulfides with two-dimensional (2D) permeable channels possess the properties of ideal CE materials and are considered as promising electrode materials.<sup>11,12</sup> Lin *et al.*<sup>13</sup> prepared a molybdenum disulfide CE for DSSCs and obtained a significant improvement in the power conversion efficiency. Moreover, we also successfully prepared a nickel disulfide CE with very promising results for DSSCs.<sup>13,14</sup> Vanadium sulfide ( $VS_2$ ) has been proven to be an ideal material platform due to its synergic properties of metallic nature brought about by the conducting S-V-S layers stacked up via weak van der Waals interlayer interactions, offering great potential as high-performance in-plane supercapacitor electrodes.<sup>15</sup> Therefore, it is interesting and significant to investigate

the potential applications of  $VS_2$  as a CE catalyst in DSSCs for low-cost and efficient photoelectric conversion efficiency.

Herein, we designed and synthesized  $VS_2$  nanofibres as a CE material via an *in situ* hydrothermal route for DSSCs, hoping that the  $VS_2$  nanofibres could promote the catalytic activity and improve the photovoltaic properties of the DSSC. The DSSC based on the  $VS_2$  nanofibre CE exhibited a high power conversion efficiency of 7.40%. This study may broaden the potential applications of two-dimensional layered transition-metal dichalcogenides in the area of photoelectrochemistry.

## 2. Experimental

### 2.1 Preparation of $VS_2$ CEs

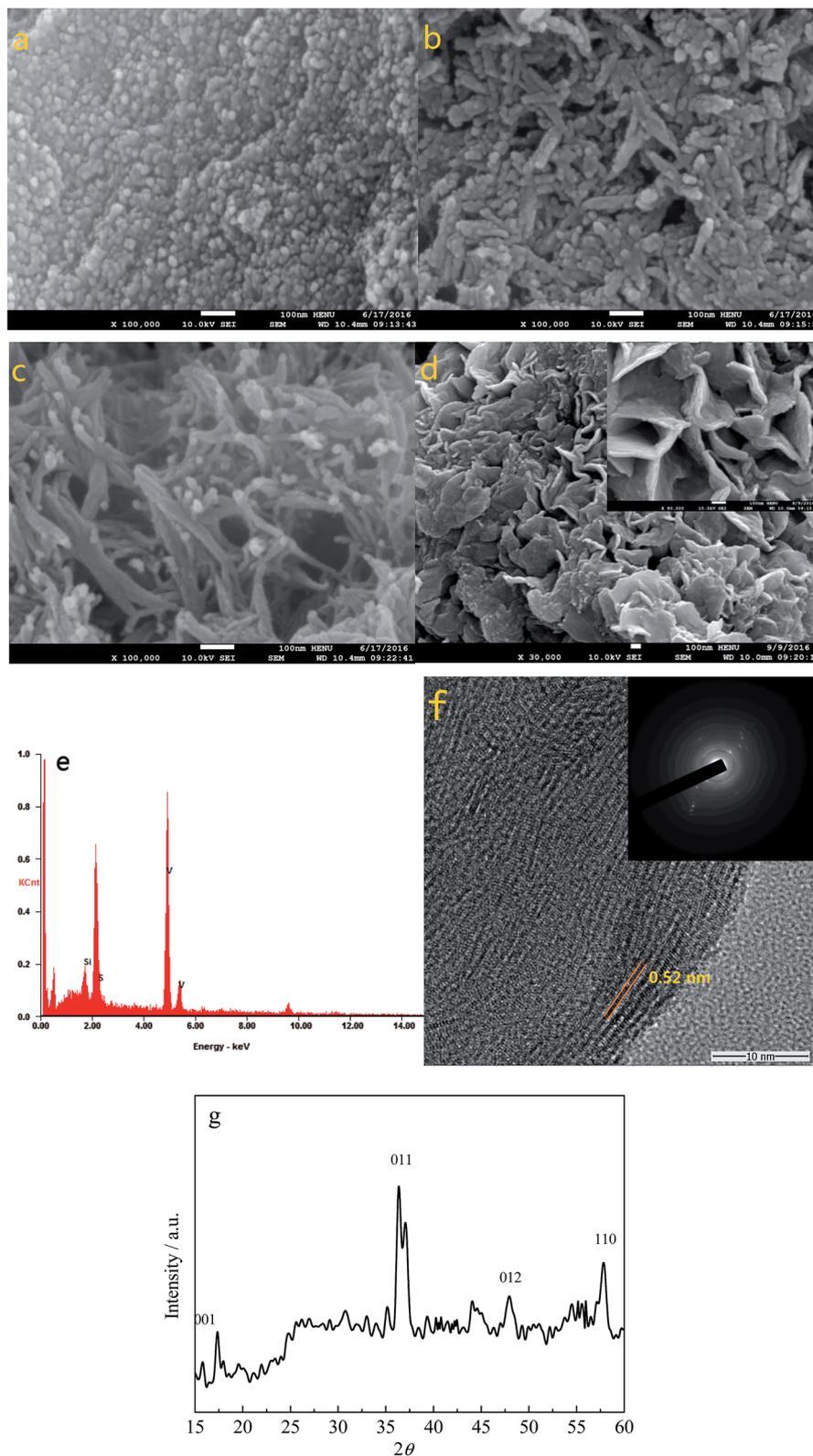
$VS_2$  was prepared via modifying the procedure reported by Feng *et al.*<sup>15</sup> A 3 mmol sodium orthovanadate and 15 mmol thiourea were dissolved in 40 ml distilled water; the mixture was then stirred for 1 h to form a homogeneous solution and transferred to a 50 ml Teflon-lined autoclave. It was then heated in an oven at 140, 160, 180, and 200 °C for 24 h. The product was collected by centrifugation, washed at least 5 times with ethanol and distilled water, and dried in a vacuum oven at 80 °C for 12 h. The slurry of the  $VS_2$  CE was composed of nanofibre structure  $VS_2$ , acetylene black, and polyvinylidene fluoride (weigh ratio = 8 : 1 : 1), which were dissolved in *N*-methyl-2-pyrrolidinone. Then, the slurry was ultrasonicated for 30 min and was stirred for 12 h. Subsequently, the as-prepared slurry was coated on the FTO substrates using a doctor blade method. The coated CEs were dried at 100 °C for 24 h in a vacuum oven.

### 2.2 Fabrication of the DSSCs

A  $TiO_2$  anode was prepared according to a previously reported procedure.<sup>16,17</sup> The dye-sensitized  $TiO_2$  photoanode was

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**Fig. 1** The SEM images of VS<sub>2</sub> obtained at (a) 140 (b) 160, (c) 180, and (d) 200 °C; (e) the EDS spectrum of VS<sub>2</sub>; (f) TEM image of the VS<sub>2</sub> obtained after the 180 °C hydrothermal synthesis; (g) XRD pattern of the VS<sub>2</sub> nanofibers.

constructed by immersing the TiO<sub>2</sub> photoanode in a 0.3 mM dye Z907 ethanol solution for 24 h. Thus, a dye-sensitized TiO<sub>2</sub> photoanode with a total thickness of 6–8 μm was obtained. After

this, the dye-sensitized TiO<sub>2</sub> photoanode and the CE were clipped together and wrapped with the thermoplastic hot-melt Surlyn. The liquid electrolyte contained 0.05 M of iodine,



0.1 M of lithium iodide, 0.6 M of tetrabutylammonium iodide, and 0.5 M of 4-*tert*-butyl-pyridine in acetonitrile and was injected into the aperture between the two electrodes.

### 2.3 Characterization

The surface morphology of the sample was observed using a JSM-7001F field emission scanning electron microscope (SEM). Energy dispersive spectroscopy analysis (EDS) was carried out using a Bruker-ASX (Model Quan-Tax 200). A field emission transmission electron microscope (TEM; JEOL JEM-2100F, operated at 200 kV with a point-to-point resolution of 0.19 nm) was used to obtain information about the microstructures. The crystalline structures of the composites were investigated by glancing incidence X-ray diffractometer (X'Pert Pro, PANalytical B.V., the Netherlands). Electrochemical impedance spectroscopy (EIS) was carried out using a CHI660E (Shanghai Chenhua Device Company, China) electrochemical measurement system at a constant temperature of 25 °C in an ambient atmosphere under dark conditions, leaving an exposed area of 0.8 cm<sup>2</sup>. The frequency of the applied sinusoidal AC voltage signal was varied from 0.1 Hz to 10<sup>5</sup> Hz,

and the corresponding amplitude was set at 5 mV in all the cases.

The photovoltaic test of the DSSC with an exposed area of 0.2 cm<sup>2</sup> was carried out by measuring photocurrent–photovoltage (*J*–*V*) character curves under a white light irradiation of 100 mW cm<sup>-2</sup> (AM 1.5 G) from a solar simulator (CEL-S500, Beijing China Education Au-light Co., Ltd) in an ambient atmosphere.

## 3. Results and discussion

Fig. 1a–d show the SEM images of VS<sub>2</sub> obtained at 140, 160, 180, and 200 °C, respectively. It can be seen that the size of the VS<sub>2</sub> nanoparticles increased with the increasing temperature and the morphology also changed, from nanoparticles to nanofibers and nanosheets. This phenomenon indicates that a higher reaction temperature can promote the growth of VS<sub>2</sub> crystals. The EDS patterns of VS<sub>2</sub> prepared at 180 °C are shown in Fig. 1e, in which the V and S elements with an almost 1 : 1 ratio can be detected, and the Si element originates from the Si substrate. Fig. 1f presents the TEM image of VS<sub>2</sub>, and the lattice spacing has been estimated to be 0.52 nm, which is in accordance with the literature parameter for VS<sub>2</sub> (0.573 nm). To further identify the composition of the sample, Fig. 1g shows the XRD pattern of the VS<sub>2</sub> nanofibers obtained from the 180 °C hydrothermal synthesis. As can be seen, although there are some impurity peaks appearing in the sample, the (001), (011), (012), and (110) peaks of the VS<sub>2</sub> all correspond to the JCPDS database card no. 89-1640. As a consequence, the results demonstrated that the VS<sub>2</sub> has been successfully prepared *via* the facile hydrothermal synthesis at 180 °C.

Fig. 2 presents the cyclic voltammograms of various CEs measured using a three-electrode system. In Fig. 2, the pair of peaks in the low potential area has a significant impact on the photovoltaic properties of the DSSCs between the two pairs of redox peaks.<sup>18</sup> As observed from the inset of Fig. 2, the Pt and VS<sub>2</sub> (180 °C preparation) CEs have a similar cathodic peak current density (*J*<sub>pc</sub>) and cathodic peak potential, indicating that the VS<sub>2</sub> CE is as good conductive and catalytic material as Pt. The VS<sub>2</sub> CEs prepared at temperatures from 140 to 200 °C possess a similar cathodic peak potential, and the cathodic peak current density follows the order VS<sub>2</sub> (180 °C) > VS<sub>2</sub> (160

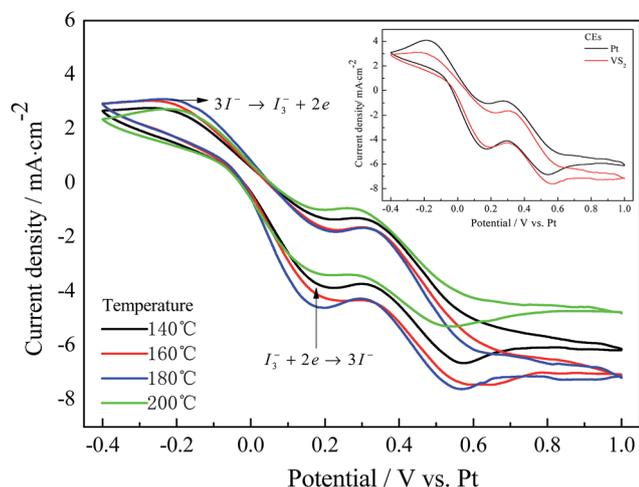


Fig. 2 Cyclic voltammograms for the Pt and VS<sub>2</sub> (prepared at 140 °C to 200 °C) CEs.

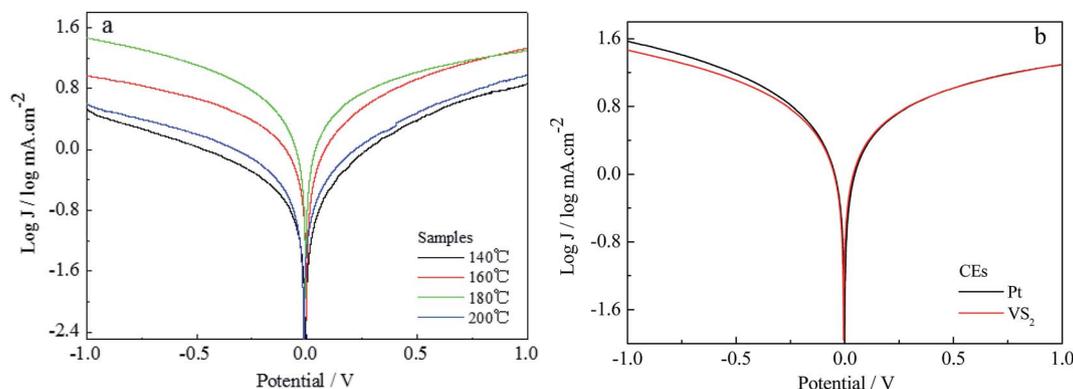


Fig. 3 Tafel curves for (a) various VS<sub>2</sub> CEs and (b) the Pt CE and the VS<sub>2</sub> CE prepared at 180 °C.



$^{\circ}\text{C}) > \text{VS}_2 (140^{\circ}\text{C}) > \text{VS}_2 (200^{\circ}\text{C})$  CEs, suggesting that the  $\text{VS}_2$  CE prepared at  $180^{\circ}\text{C}$  has a better catalytic activity and conductivity. This indicates that the electron transport was affected by the surface morphologies of the samples.  $\text{VS}_2$  nanofibers with a large specific surface area can vastly enhance the accessibility of the electrolyte to the electrode, thus improving interfacial charge transfer and increasing the number of active catalytic sites.<sup>19,20</sup>

Tafel curves for the  $\text{VS}_2$  CEs prepared at different temperatures are shown in Fig. 3a. The exchange current density ( $J_0$ ), obtained as the intercept of the extrapolated linear region of the curve when the overpotential was zero, is positively correlated to the reduction capability of the CE materials in an  $\Gamma/\text{I}_3^-$  electrolyte. The  $\text{VS}_2$  CEs prepared at temperatures from  $140$  to  $200^{\circ}\text{C}$  exhibit the same change tendencies as for the CVs. Under the optimizing conditions, the  $J_0$  of the  $\text{VS}_2$  CE ( $180^{\circ}\text{C}$  preparation) is similar to that of the Pt CE, as shown in Fig. 3b. This is mainly because  $\text{VS}_2$  ( $180^{\circ}\text{C}$  preparation) with nanofibre structure is better for electron transport than  $\text{VS}_2$  nanoparticles. Thus, this also indicates that the electrochemical catalytic activity of the  $\text{VS}_2$  CE is significantly affected by the morphology.

Fig. 4 displays the electrochemical impedance spectroscopy (EIS) and equivalent circuit models of the Pt and the  $\text{VS}_2$  CEs synthesized at various temperatures, in which the first semi-circle in the high frequency region denotes the charge-transfer resistance ( $R_{ct}$ ) at the CE/electrolyte interface<sup>21,22</sup> and the

corresponding EIS parameters are listed in Table 1. The smaller  $R_{ct}$  exhibits a faster electron transfer from the CE to the electrolyte, which is a significant parameter for evaluating the performance of CEs. In Fig. 4, the values of  $R_{ct}$  for the  $\text{VS}_2$  CEs synthesized at  $140$ ,  $160$ ,  $180$  and  $200^{\circ}\text{C}$  are  $6.437$ ,  $4.318$ ,  $3.360$ , and  $8.435 \Omega \text{ cm}^2$ , respectively; and the  $R_{ct}$  for the Pt CE is  $3.432 \Omega \text{ cm}^2$ . The  $R_{ct}$  value for the  $\text{VS}_2$  CE prepared at  $180^{\circ}\text{C}$  is comparable to that of the Pt CE. The results are in agreement with the CVs and Tafel curves, and this can be attributed to the same reasons as for the CVs and Tafel curves.

Fig. 5 shows the photocurrent density–voltage curves for the DSSCs based on the Pt and  $\text{VS}_2$  ( $180^{\circ}\text{C}$  preparation) CEs under the irradiation of  $100 \text{ mW cm}^{-2}$ , and the photovoltaic parameters for the DSSCs are also summarized in Table 1. From Fig. 5, it can be observed that the DSSC based on the  $\text{VS}_2$  CE synthesized at  $180^{\circ}\text{C}$  achieved a power conversion efficiency (PCE) of  $6.24\%$ , an open-circuit voltage ( $V_{oc}$ ) of  $0.726 \text{ V}$ , a short-circuit current density ( $J_{sc}$ ) of  $13.65 \text{ mA cm}^{-2}$ , and a fill factor (FF) of  $0.63$ , which are almost the same as those for the DSSC based on the Pt electrode (PCE =  $6.44\%$ ,  $V_{oc} = 0.717 \text{ V}$ ,  $J_{sc} = 14.03 \text{ mA cm}^{-2}$ , and FF =  $0.64$ ). The DSSCs based on the  $\text{VS}_2$  CEs prepared at  $140$ ,  $160$  and  $200^{\circ}\text{C}$  have much lower PCEs than the DSSC with the  $\text{VS}_2$  CE synthesized at  $180^{\circ}\text{C}$ . The positive effect in the performance of the DSSC based on the  $\text{VS}_2$  ( $180^{\circ}\text{C}$ ) CE possibly results from the following aspects. First, the contact frequency between the  $\Gamma/\text{I}_3^-$  redox couple in the electrolyte and the CE can be quickened because of the large specific surface area of

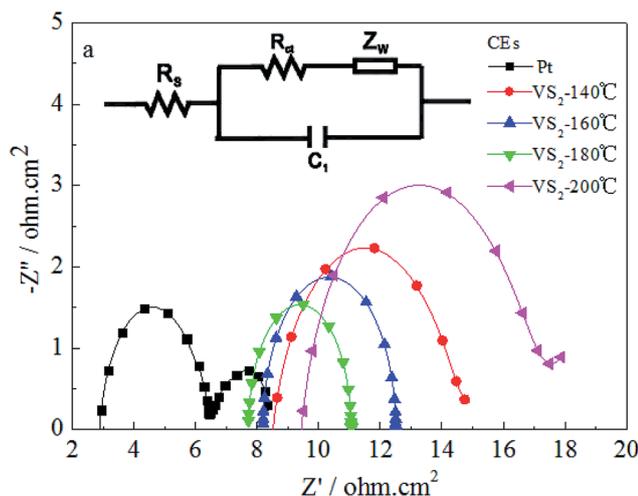


Fig. 4 EIS spectra for the Pt and  $\text{VS}_2$  CEs. The inset (a) shows the equivalent circuit model used for fitting the resultant impedance spectra.

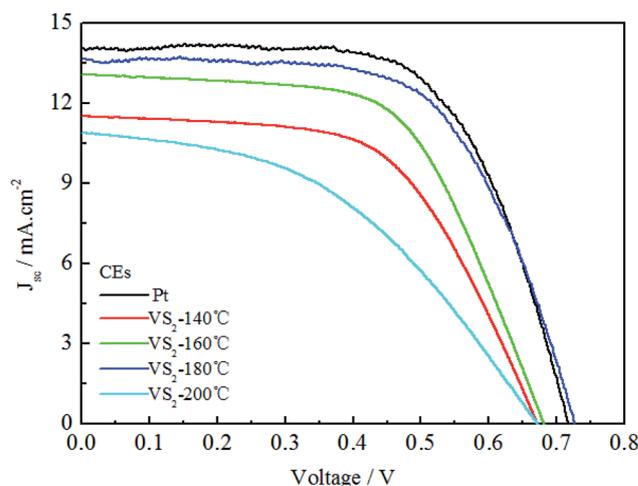


Fig. 5 Photocurrent density–voltage curves of the DSSCs based on Pt and various  $\text{VS}_2$  CEs.

Table 1 Electrochemical parameters for the Pt and  $\text{VS}_2$  CEs and the corresponding photocurrent–voltage parameters of the DSSCs

Temp. ( $^{\circ}\text{C}$ )	$R_{ct}$ ( $\Omega \text{ cm}^2$ )	$ J_{pc} $ ( $\text{mA cm}^{-2}$ )	$V_{oc}$ (V)	$J_{sc}$ ( $\text{mA cm}^{-2}$ )	FF	$\eta$ (%)
140	6.437	3.79	0.671	11.52	0.58	4.48
160	4.318	4.20	0.681	13.06	0.60	5.34
180	3.360	4.55	0.726	13.65	0.63	6.24
200	8.435	3.33	0.672	10.89	0.46	3.37
Pt	3.432	4.65	0.717	14.03	0.64	6.44



the VS<sub>2</sub> nanofibers, such that to provide a good reaction speed for the reduction of I<sub>3</sub><sup>-</sup> to I<sup>-</sup>. In addition, the VS<sub>2</sub> nanofibers have synergic advantages of a high conductivity and 2D permeable channels guarantee the rapid transmission of electrons, thereby showing efficient PCEs for the DSSCs based on the VS<sub>2</sub> CE. In addition, it has been reported in previous studies that VS<sub>2</sub> was successfully assembled for constructing the electrodes of in-plane supercapacitors.<sup>23,24</sup> To the best of our knowledge, it is the first time that VS<sub>2</sub> has been reported as a CE material for DSSCs.

## 4. Conclusion

VS<sub>2</sub> nanofibers were prepared *via in situ* hydrothermal techniques and were employed as a CE in the Pt-free DSSCs for the first time. The sizes of VS<sub>2</sub> increase and the morphology is also affected with the increasing temperature, whereby nanoparticles were observed to changed into nanofibers and even nanosheets. Extensive electrochemical and photoelectric chemical experiments indicate that the VS<sub>2</sub> nanofibers prepared at 180 °C have the synergic advantages of high conductivity, large specific surface area, and 2D permeable channels and provide the most excellent catalytic activity for the reduction of triiodide compared to VS<sub>2</sub> prepared at 140, 160, and 200 °C. Under the optimum conditions, the PCE of the DSSC based on the VS<sub>2</sub> nanofiber CE (6.24%) is as high as that of the DSSC based on the Pt electrode (6.44%). This study offers a new and effective material substitution to Pt, which will broaden the application field of 2D sulfides.

## Conflict of interest

The authors declare that they have no competing interests.

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## References

- B. O'Regan and M. Grätzel, *Nature*, 1991, **353**, 737–740.
- A. Yella, H. W. Lee, H. N. Tsao, C. Y. Yi, A. K. Chandiran, M. K. Nazeeruddin, E. W. G. Diau, C. Y. Yeh, S. M. Zakeeruddin and M. Grätzel, *Science*, 2011, **334**, 629–634.
- A. Hagfeldt, G. Boschloo, L. Sun, L. Kloo and H. Pettersson, *Chem. Rev.*, 2010, **110**, 6595–6663.
- S. Mathew, A. Yella, P. Gao, R. Humphry-Baker, B. F. E. Curchod, N. Ashari-Astani, I. Tavernelli, U. Rothlisberger, M. K. Nazeeruddin and M. Grätzel, *Nat. Chem.*, 2014, **6**, 242–247.
- J. H. Wu, Z. Lan, J. M. Lin, M. L. Huang, Y. F. Huang, L. Q. Fan and G. G. Luo, *Chem. Rev.*, 2015, **115**, 2136–2173.
- Y. J. Li, Q. W. Tang, L. M. Yu, X. F. Yan and L. Dong, *J. Power Sources*, 2016, **305**, 217–224.
- G. T. Yue, X. P. Ma, W. F. Zhang, F. M. Li, J. H. Wu and G. Q. Li, *Nanoscale Res. Lett.*, 2015, **10**, 1–9.
- Z. Q. Li, F. Gong, G. Zhou and Z. S. Wang, *J. Phys. Chem. C*, 2013, **117**, 6561–6566.
- H. C. Sun, D. Qin, S. Q. Huang, X. Z. Guo, D. M. Li, Y. H. Luo and Q. B. Meng, *Energy Environ. Sci.*, 2011, **4**, 2630–2637.
- G. T. Yue, P. Li, F. M. Li and C. Chen, *RSC Adv.*, 2016, **6**, 61278–61283.
- M. Mulazzi, A. Chainani, N. Katayama, R. Eguchi, M. Matsunami, H. Ohashi, Y. Senba, M. Nohara, M. Uchida, H. Takagi and S. Shin, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2010, **82**, 075130.
- K. S. Kim, Y. Zhao, H. Jang, S. Y. Lee, J. M. Kim, K. S. Kim, J. H. Ahn, P. Kim, J. Y. Choi and B. H. Hong, *Nature*, 2009, **457**, 706–710.
- J. Y. Lin, A. L. Su, C. Y. Chang, K. C. Hung and T. W. Lin, *ChemElectroChem*, 2015, **2**, 720–725.
- G. T. Yue, F. R. Tan, F. M. Li, C. Chen, W. F. Zhang, J. H. Wu and Q. H. Li, *Electrochim. Acta*, 2014, **149**, 117–125.
- J. Feng, X. Sun, C. Z. Wu, L. L. Peng, C. W. Lin, S. L. Hu, J. L. Yang and Y. Xie, *J. Am. Chem. Soc.*, 2011, **133**, 17832–17838.
- G. T. Yue, J. H. Wu, Y. M. Xiao, M. L. Huang, J. M. Lin and J.-Y. Lin, *J. Mater. Chem. A*, 2013, **1**, 1495–1501.
- Z.-Q. Li, Y.-P. Que, L.-E. Mo, W.-C. Chen, Y. Ding, Y.-M. Ma, L. Jiang, L.-H. Hu and S.-Y. Dai, *ACS Appl. Mater. Interfaces*, 2015, **7**, 10928–10934.
- G. T. Yue, J. H. Wu, J.-Y. Lin, Y. M. Xiao, J. M. Lin, M. L. Huang and Z. Lan, *Carbon*, 2013, **55**, 1–9.
- C.-T. Li, Y.-L. Tsai and K.-C. Ho, *ACS Appl. Mater. Interfaces*, 2016, **8**, 7037–7046.
- H. Sun, J. Deng, L. Qiu, X. Fang and H. Peng, *Energy Environ. Sci.*, 2015, **8**, 1139–1159.
- Y. Xiao, G. Han, Y. Li, M. Li and Y. Chang, *J. Mater. Chem. A*, 2014, **2**, 3452–3460.
- G. Yue, J. Wu, Y. Xiao, J. Lin, M. Huang and Z. Lan, *J. Phys. Chem. C*, 2012, **116**, 18057–18063.
- D. Pech, M. Brunet, H. Durou, P. Huang, V. Mochalin, Y. Gogotsi, P. L. Taberna and P. Simon, *Nat. Nanotechnol.*, 2010, **5**, 651–654.
- J. J. Yoo, K. Balakrishnan, J. Huang, V. Meunier, B. G. Sumpter, A. Srivastava, M. Conway, A. L. Mohana Reddy, J. Yu, R. Vajtai and P. M. Ajayan, *Nano Lett.*, 2011, **11**, 1423–1427.

