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# Preparation and evaluation of modified cyanobacteria-derived activated carbon for H<sub>2</sub> adsorption†

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A series of porous carbon sorbents obtained by carbonization and activation by KOH/ZnCl<sub>2</sub> of the abundant waste material cyanobacteria was utilized for the adsorption of H<sub>2</sub>. During experiments to determine the optimal activation parameters, the activation time had little effect, whereas the KOH/C and ZnCl<sub>2</sub>/C mass ratios and activation temperature had a significant impact on adsorption performance. Samples activated by KOH exhibited better adsorption characteristics (such as higher values of  $S_{\text{BET}}$ ,  $V_{\text{total}}$ ,  $V_{\text{micro}}$ , etc.) than those activated by ZnCl<sub>2</sub>. In particular, ACK-2-8, which had an  $S_{\text{BET}}$  of 1951 m<sup>2</sup> g<sup>-1</sup>, displayed H<sub>2</sub> uptake capacities of nearly 17.3 mg g<sup>-1</sup> at -196 °C and 1 bar. In addition, for evaluation of the H<sub>2</sub> adsorption performance, N-doped and P-doped porous carbons were synthesized using HNO<sub>3</sub>, NH<sub>3</sub>·H<sub>2</sub>O and H<sub>3</sub>PO<sub>4</sub>, respectively, to modify the abovementioned cyanobacteria-derived activated carbon ACK-2-8, and exhibited H<sub>2</sub> adsorption amounts that were 16.8%, 31.8%, and 45.7% higher, respectively, than those of undoped ACK-2-8. Furthermore, it was remarkable that P-doping in conjunction with a moderately high  $S_{\text{BET}}$  enhanced the uptake of H<sub>2</sub> by ACK-2-8-2, which had a smaller  $S_{\text{BET}}$  than ACK-2-8-3 but the best uptake capacity of up to 25.2 mg g<sup>-1</sup>, and thereby showed that the relationship between the H<sub>2</sub> adsorption capacity and  $S_{\text{BET}}$  of these materials was scarcely linear and was therefore noticeably different from previously reported results for the uptake of H<sub>2</sub> by other carbon sorbents. Hence, it is probable that the considerable H<sub>2</sub> adsorption properties of the material with a moderate  $S_{\text{BET}}$  have great potential applications for hydrogen capture or storage.

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## 1 Introduction

Hydrogen, which is considered to be an environmentally friendly fuel, is also an energy vector that should be extensively used in the future.<sup>1,2</sup> It is the molecule that delivers the largest amount of energy on combustion with oxygen, in which only water is generated. Moreover, it is nontoxic and its high flammability is counterbalanced by its ultrafast dissipation in the atmosphere in the event of a leakage.<sup>3</sup> The main drawback of hydrogen arises from its very low density, which is only 90 g m<sup>-3</sup> at 1 bar and 273 K.<sup>4</sup> Storing the maximum amount of hydrogen in the smallest possible volume is thus a real challenge. On the other hand, high-performance energy storage systems have been extensively investigated in recent decades to meet the increasing energy demands of the world.<sup>5</sup> Hydrogen storage is one of the most important technologies in energy storage.<sup>6,7</sup>

Thus, there is great interest in the development of suitable adsorbents for gas capture/storage that can be used for solving energy-related issues. Hydrogen storage can be achieved by adsorption, liquefaction, compression, or chemical bonding in the form of metal hydrides.<sup>8,9</sup>

Recently, abundant research into porous materials including carbon materials,<sup>10–13</sup> zeolites,<sup>14,15</sup> metal–organic frameworks (MOFs),<sup>16–18</sup> and covalent organic frameworks (COFs)<sup>19,20</sup> has been widely undertaken. Among these, carbon materials have been extensively studied as candidates for hydrogen storage owing to their large surface area, well-developed pore structure, low cost, low mass density, and so on.<sup>21,22</sup> In general, a large surface area and large micropore volume are essential for enhancing the uptake of hydrogen.<sup>23–26</sup> Activated carbons based on carbon materials can be prepared from a large number of precursors, which include vegetable, mineral, and purely synthetic sources such as organic gels and polymers.<sup>4</sup> However, from the viewpoint of sustainability, environment-friendliness, and continuous availability, biomass-based materials are more promising sources of carbon for porous carbons.<sup>5,27–31</sup> Cyanobacteria are phototrophic bacteria that exist in several ecosystems across the planet and are key contributors to global photosynthesis.<sup>32</sup> However, not all cyanobacteria are beneficial to the environment and human health; some cyanobacteria

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grow and expand too fast, contaminating freshwater and creating toxins.<sup>33</sup> Owing to the abundance and accessibility of cyanobacteria, it is desirable to confirm the suitability of converting this form of biomass into active carbon for energy storage. Thus far as we know, few studies have attempted to use cyanobacteria as a precursor for activated carbon in materials science other than as supercapacitor electrodes.<sup>32</sup> In this study, for use as low-cost, abundant and available carbon precursors, cyanobacteria (CB) obtained from Dianchi in Yunnan Province, China were prepared for precursors. As hydrogen adsorption performance requires very large surface areas and very narrow pores, activation by KOH is known to be among the best methods for obtaining carbon materials having such characteristics. Thus, cyanobacteria-based activated carbons prepared by activation by KOH and also other carbons prepared using ZnCl<sub>2</sub> were used for comparison. As a number of studies have reported,<sup>34–37</sup> the weight ratio of the activating agent to the precursor and the activation temperature have a great effect on the uptake of hydrogen. The influence of the activation parameters will be investigated in the following research.

The adsorption of H<sub>2</sub> could be characterized as physisorption with weak van der Waals interaction forces. Much effort has been devoted to increasing the specific surface area because a highly porous structure is indispensable for a high adsorption capacity.<sup>38</sup> Moreover, it is necessary to improve the adsorption efficiency and enhance the interactions between the gas and carbon material, which is of great importance for the adsorption performance of carbon materials. Heteroatom doping is widely used to modify carbon materials for efficient adsorption.<sup>39</sup> For the adsorption of different gases, the presence of heteroatoms leads to a remarkable performance.<sup>5,38–41</sup> Some studies suggest that the surface polarity created by the discrepancy in electronegativity between heteroatoms and carbon could enhance the interactions between gas molecules and the carbon surface.<sup>5,38</sup> For the adsorption of hydrogen, Li *et al.* investigated the influence of doping nitrogen, sulfur, and phosphorus into activated carbons on the adsorption of H<sub>2</sub>, CH<sub>4</sub> and CO<sub>2</sub> gases, which exhibited a favorable effect on the adsorption of H<sub>2</sub>.<sup>38</sup> In order to achieve further enhancements in adsorption efficiency, further modifications of cyanobacteria-derived activated carbon were performed by impregnation with HNO<sub>3</sub>, H<sub>3</sub>PO<sub>4</sub>, and NH<sub>3</sub>·H<sub>2</sub>O, respectively. The H<sub>2</sub> adsorption properties were investigated systematically and the possible effects and causes were also discussed.

## 2 Experimental

### 2.1 Materials and reagents

Cyanobacteria (CB) obtained from Dianchi in Yunnan Province, China were subjected to a natural drying process for 72 h, crushed and systematically sieved to give particles with a size of 30 mesh. All chemicals were purchased from Sinopharm Chemical Reagent Co., Ltd. and were of analytical grade. Water was filtered and deionized with a Millipore Milli-Q system (Milford, MA, USA).

### 2.2 Synthesis

Cyanobacteria-derived activated carbon (AC) was prepared by heating cyanobacteria to 400 °C at a ramp rate of 20 °C min<sup>-1</sup> and allowing the product to dwell for 1 h in a tube furnace with N<sub>2</sub> as flushing gas. By a subsequent activation process, a series of specific activated carbons were synthesized *via* the following method. AC (1 g) was first impregnated with a solution of KOH or ZnCl<sub>2</sub> (50 mL) at predetermined KOH/C or ZnCl<sub>2</sub>/C mass ratios, and the mixture was shaken at 170 rpm at 25 °C for 24 h in a constant-temperature shaker and then dried at 105 °C for 24 h in a vacuum drying oven. The resulting dry materials were placed in a tubular furnace, followed by heating at a ramp rate of 20 °C min<sup>-1</sup> to a predetermined activation temperature, which was maintained for 2 h. The heating process was conducted under protection by N<sub>2</sub> flow. Finally, the activated carbon particles were washed using aq. HCl (1 mol L<sup>-1</sup>), followed by washing with deionized water until the pH value of the wash water was neutral, and the resulting samples were dried in a vacuum oven at 105 °C for 24 h. The cyanobacteria-derived activated carbons were denoted as ACK-X-Y or ACZ-X-Y, where X represents the KOH/C or ZnCl<sub>2</sub>/C mass ratio, respectively, and Y denotes the activation temperature in multiples of 100 °C.

In turn, further modifications of the samples were carried out by adding ACK-2-8 to 50 mL of the corresponding 10 wt% solution of HNO<sub>3</sub>, H<sub>3</sub>PO<sub>4</sub>, or NH<sub>3</sub>·H<sub>2</sub>O, respectively, in a water bath at 60 °C for 3 h, followed by washing with distilled water until the wash water reached a neutral pH and drying at 105 °C for 24 h in a vacuum drying oven. These samples were respectively named as ACK-2-8-1, ACK-2-8-2 and ACK-2-8-3 for simplicity.

### 2.3 Characterization and analysis of the samples

A Diamond TG/DTA (thermogravimetric/differential thermal analysis) integrated thermal analyzer (PerkinElmer Instruments) was used to investigate the pyrolysis characteristics of the cyanobacteria. All the TGA tests were performed in high-purity nitrogen (99.99%) with a flow rate of 50 mL min<sup>-1</sup>. The test temperature ranged from room temperature to 800 °C at a heating rate of 20 °C min<sup>-1</sup>. The mass of the sample used in each experiment was 3.0 ± 0.1 mg. The elemental abundances of C, H, O, N and S were determined using a CHNS/O elemental analyzer (vario MICRO cube, Elementar). This analysis gave the mass percentages of carbon, hydrogen, nitrogen, and sulfur in the samples simultaneously, and the mass percentage of oxygen was determined by subtraction. A thermogravimetric analyzer (SDT Q600, TA Instruments) provided proximate analyses of the sample contents (that is, the moisture, volatile matter, fixed carbon, and ash contents of the material). An X-ray fluorescence spectrometer (Eagle III XRF, EDAX) was further employed for component analysis of the cyanobacteria ash. The surface functional groups of the samples were analyzed using the classic Boehm titration method.<sup>42</sup> The acidic surface functional groups were quantified using a solution of NaHCO<sub>3</sub>, Na<sub>2</sub>CO<sub>3</sub>, and NaOH (total acidic groups), whereas the total basic surface functional groups were quantified using hydrochloric acid.<sup>43</sup> Parallel control experiments were conducted without carbon.



The physical parameters of the initial and modified AC samples, including their adsorption–desorption characteristics, BET surface area ( $S_{\text{BET}}$ ), total pore volume ( $V_{\text{total}}$ ), micropore volume ( $V_{\text{micro}}$ ) and average pore size ( $\bar{D}$ ), were determined using an automatic specific surface area and porosity analyzer (JW-BK122W, JWGB Ltd.). The adsorbed volume of high-purity  $\text{N}_2$  was measured under different pressures at  $-196^\circ\text{C}$ . The  $S_{\text{BET}}$  value was calculated using the Brunauer–Emmett–Teller method. The pore size distribution was computed using the Barrett–Joyner–Halenda method from the adsorption branch of the isotherm.<sup>44</sup> The micropore volume ( $V_{\text{micro}}$ ) and micropore-specific surface area ( $S_{\text{micro}}$ ) were determined using the  $t$ -plot method.<sup>43</sup> All samples were characterized by microscopy techniques. Scanning electron microscopy (SEM) was performed using a Jeol JSM-6301F microscope. X-ray photoelectron spectroscopy (XPS) was carried out using a Thermo Scientific Al  $K\alpha$  source. A static gas adsorption instrument (NOVA 3200e, Quantachrome) was used to measure the adsorption of  $\text{H}_2$  in the pressure range of 0.01–1 bar at  $-196^\circ\text{C}$ .

## 3 Results and discussion

### 3.1 Chemical composition and thermogravimetric analysis

Regarding the characteristics of the cyanobacteria (CB) raw material, the results of proximate and ultimate analysis of the CB are shown in Table 1. As is observed, the mass percentages of carbon and oxygen reached 43.28% and 35.81%, respectively, whereas the moisture, volatile matter, fixed carbon, and ash contents of the material were 15.03%, 68.04%, 10.96% and 5.97%, respectively. The percentages of 43.28% C and 10.96% fixed carbon were indispensable for providing the original fundamental material framework of the modified ACs. In comparison with lignocellulosic materials,<sup>45</sup> the higher oxygen content facilitated the formation of extensive porous architectures *via* the release of activated oxygen during the pyrolysis or carbonization process of CB. It can also be seen that the moisture content was much larger than those of wheat straw, rice straw or candlenut wood,<sup>45</sup> which was ascribed to the characteristics of CB and the different preprocessing methods (only dried naturally). XRF experiments were performed on CB ash, and the corresponding component analysis is shown in Table S1 (ESI†). It is noticeable that  $\text{Al}_2\text{O}_3$ ,  $\text{P}_2\text{O}_5$ ,  $\text{CaO}$ ,  $\text{SiO}_2$  and  $\text{K}_2\text{O}$  were the main components (more than 80%) of all the ashes.

Table 1 Proximate and ultimate analysis of cyanobacteria

| Proximate analysis <sup>a</sup> (wt%) |       | Ultimate analysis (wt%) |       |
|---------------------------------------|-------|-------------------------|-------|
| Moisture                              | 15.03 | C                       | 43.28 |
| Volatiles                             | 68.04 | H                       | 6.36  |
| Ash                                   | 5.97  | O <sup>b</sup>          | 35.81 |
| Fixed carbon                          | 10.96 | N                       | 7.7   |
|                                       |       | S                       | 0.88  |

<sup>a</sup> Dry basis. <sup>b</sup> Subtraction calculations.

TG/DTG experiments on CB were performed at a heating rate of  $20^\circ\text{C min}^{-1}$  and the thermal decomposition was studied from ambient temperature to  $800^\circ\text{C}$ , and the corresponding curves are shown in Fig. 1. It was demonstrated that the thermal decomposition of CB could be divided into three main stages. Stage I occurred in the low-temperature region (below  $250^\circ\text{C}$ ) and resulted in low mass loss (TG curves). It was found to be attributed to the evaporation of physically adsorbed water and light volatile compounds. Stage II, which covered the temperature range from  $250^\circ\text{C}$  to  $400^\circ\text{C}$  and exhibited a high mass loss percentage (approximately 50%), was ascribed to the decomposition or devolatilization of CB. The thermal degradation behavior in stage III, which was characterized by low mass loss, was probably due to the decomposition of carbonaceous materials retained in char residues. Overall, the mass losses of CB amounted to over 60% during stages I and II of the thermal decomposition process. The reason was that CB contained large amounts of volatile matter and moisture but less ash content. From the DTG curve, it could be observed that the sharp peak that appeared near  $300^\circ\text{C}$  indicated the maximum mass loss rate, and then the mass loss rate practically remained unchanged with an increase in temperature above  $400^\circ\text{C}$ . Therefore, this would be appropriate for selection of the carbonization temperature during the preparation of activated carbon.

### 3.2 Influence of activation parameters

The parameters of the porous structure determined from the nitrogen adsorption isotherms and pore size distributions are important for adsorbents and could also differ considerably as the conditions used for the preparation of the activated carbon changed. Therefore, the influence of the activation parameters on the adsorption performance was evaluated using  $\text{N}_2$  adsorption/desorption isotherms (NADIs) and pore size distribution (PSD) experiments. Among all the various preparation conditions, the KOH/C or  $\text{ZnCl}_2/\text{C}$  mass ratio, activation temperature, and activation time had a significant effect on activated carbons.<sup>46</sup> To find the optimum activation time for the preparation of activated carbons, the results of preliminary experiments performed to determine the maximum BET surface areas using ACK-2-8 and ACZ-3-8, respectively, are presented in

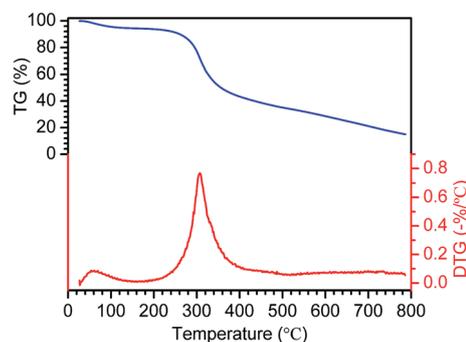


Fig. 1 TG and DTG curves of cyanobacteria.



Fig. S1 (ESI†). The optimum activation time was confirmed to be 2 h, which was therefore used in the subsequent experiments.

When the activation temperature and time were fixed at 600 °C and 2 h, respectively, the NADIs and PSD curves for some materials are presented in Fig. 2, and the corresponding parameters of the porous architectures are summarized in Table 2. In comparison with AC (Fig. S2 (ESI†)), the data for the activated counterparts ACK-X-6 and ACZ-X-6 all exhibited a steep increase, which indicated better absorption characteristics, *i.e.*,  $S_{\text{BET}}$ ,  $V_{\text{total}}$  and  $V_{\text{micro}}$  all displayed evident increases. The two kinds of sample with activation temperatures of 600 °C exhibited highly microporous structures with type I adsorption curves. The sharp increases in the adsorption curves at low relative pressures were indicative of the development of micro- and supermicropores. Specifically, the increase in the amount of  $\text{N}_2$  adsorbed by the samples with an increase in relative pressure exhibited a nearly linear relationship, with a hysteresis loop occurring at high relative pressures. This phenomenon implied the existence of a considerable number of mesopores, in which capillary condensation enabled the adsorption capacity to rise rapidly with respect to an increase in pressure.<sup>43</sup>

In addition, the amount of  $\text{N}_2$  adsorbed on activated carbon increased with an increase in the KOH/C or  $\text{ZnCl}_2/\text{C}$  mass ratio (Fig. 2a and c). The maximum value was achieved at a KOH/C or  $\text{ZnCl}_2/\text{C}$  mass ratio of 3, but a remarkable difference was observed in that the adsorbed amounts were nearly identical at KOH/C mass ratios of 2 and 3. Therefore, in the subsequent

Table 2 Structural parameters of ACK-X-6 and ACZ-X-6

| Sample    | $S_{\text{BET}}$<br>( $\text{m}^2 \text{g}^{-1}$ ) | $V_{\text{total}}$<br>( $\text{cm}^3 \text{g}^{-1}$ ) | $V_{\text{micro}}$<br>( $\text{cm}^3 \text{g}^{-1}$ ) | $\bar{D}$<br>(nm) |
|-----------|--|---|---|-------------------|
| ACK-0.5-6 | 644  | 0.404   | 0.312   | 4.81              |
| ACK-1-6   | 705  | 0.429   | 0.344   | 4.62              |
| ACK-2-6   | 1006   | 0.578   | 0.505   | 2.30              |
| ACK-3-6   | 970  | 0.545   | 0.454   | 2.23              |
| ACZ-1-6   | 901  | 0.514   | 0.428   | 2.28              |
| ACZ-2-6   | 996  | 0.640   | 0.496   | 2.31              |
| ACZ-3-6   | 1268   | 0.760   | 0.531   | 2.39              |

experiments KOH/C mass ratios of 2 and 3 and a  $\text{ZnCl}_2/\text{C}$  mass ratio of 3 were utilized, respectively, to determine the optimum activation temperature.

Similarly, the effect of the activation temperature on samples (denoted as ACK-2-Y, ACK-3-Y and ACZ-3-Y) was studied *via*  $\text{N}_2$  adsorption-desorption experiments, in which the activated carbon was prepared with KOH/C mass ratios of 2 and 3 or a  $\text{ZnCl}_2/\text{C}$  mass ratio of 3 and an activation time of 2 h. The results are illustrated in Fig. 3, whereas the structural parameters are listed in Table 3. The NADIs and PSD curves almost all exhibited a type I nitrogen adsorption isotherm, which indicated the development of micro- and supermicropores at low relative pressures. The remarkable hysteresis loop that occurred at high relative pressures with some samples (*i.e.*, ACK-2-7, ACK-2-8, and ACZ-3-8) provided evidence of the presence of some

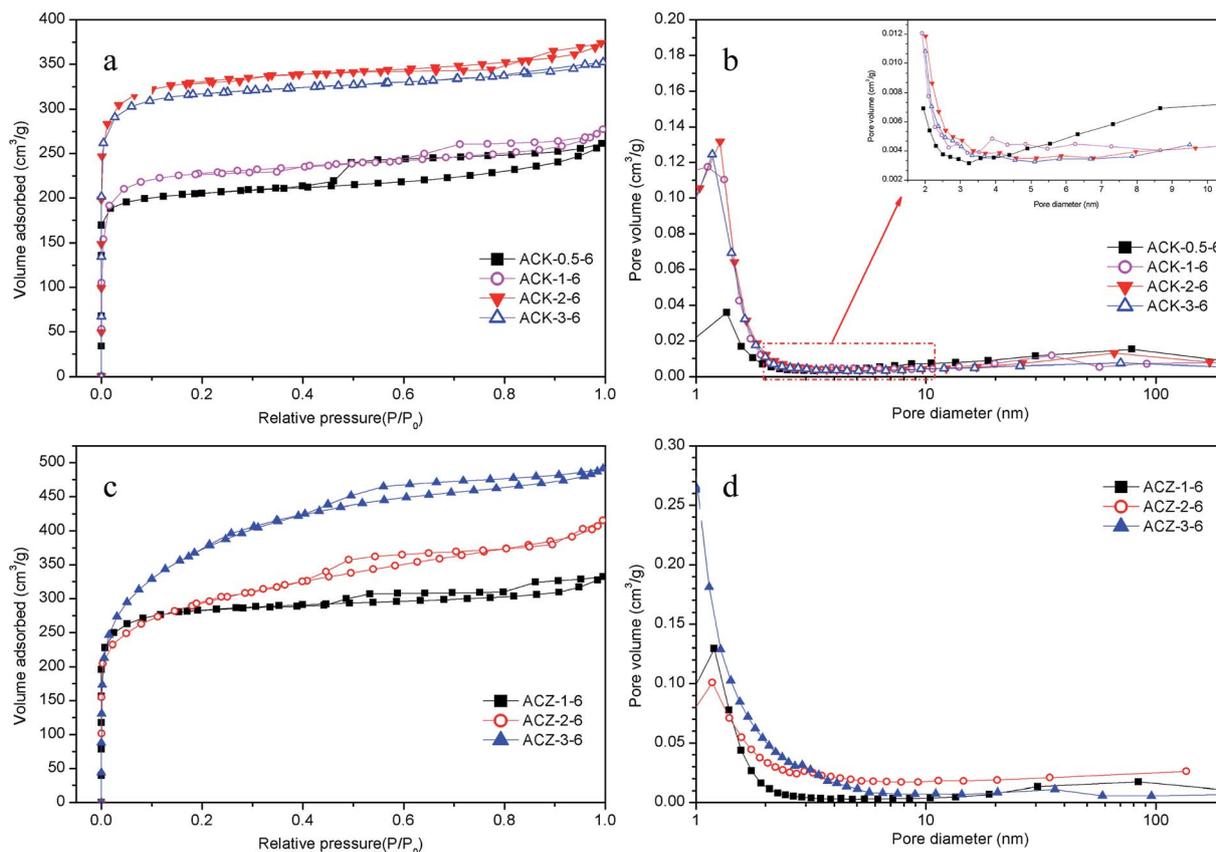


Fig. 2 NADIs (a, c) and PSDs (b, d) of ACK-X-6 and ACZ-X-6.



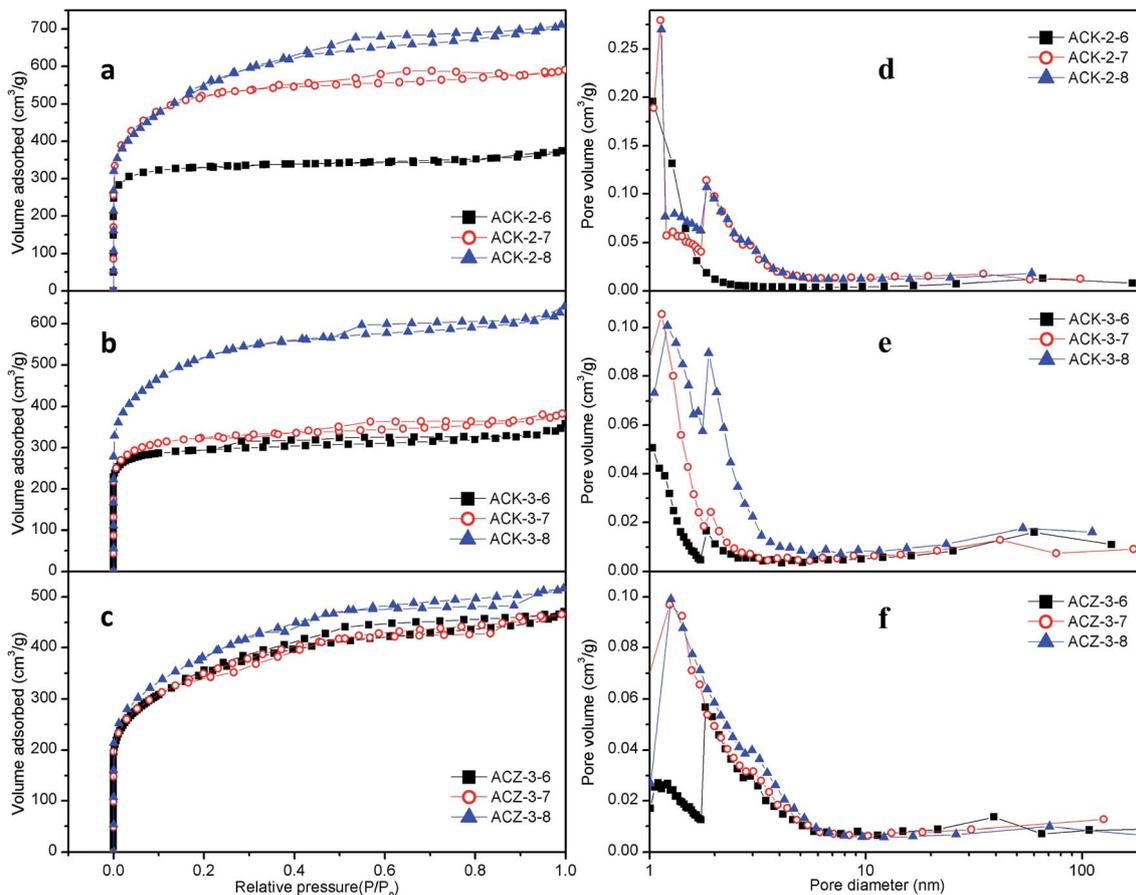


Fig. 3 NADIs (a–c) and PSDs (d–f) of ACK-2-Y, ACK-3-Y and ACZ-3-Y.

Table 3 Structural parameters of ACK-2-Y, ACK-3-Y and ACZ-3-Y

| Sample  | $S_{\text{BET}}$<br>( $\text{m}^2 \text{g}^{-1}$ ) | $V_{\text{total}}$<br>( $\text{cm}^3 \text{g}^{-1}$ ) | $V_{\text{micro}}$<br>( $\text{cm}^3 \text{g}^{-1}$ ) | $\bar{D}$<br>(nm) |
|---------|--|---|---|-------------------|
| ACK-2-6 | 1006   | 0.578   | 0.505   | 2.30              |
| ACK-2-7 | 1694   | 0.887   | 0.700   | 2.10              |
| ACK-2-8 | 1951   | 1.092   | 0.756   | 2.27              |
| ACK-3-6 | 970  | 0.545   | 0.454   | 2.23              |
| ACK-3-7 | 1111   | 0.592   | 0.487   | 2.13              |
| ACK-3-8 | 1854   | 0.992   | 0.724   | 2.26              |
| ACZ-3-6 | 1268   | 0.760   | 0.531   | 2.39              |
| ACZ-3-7 | 1176   | 0.720   | 0.505   | 2.45              |
| ACZ-3-8 | 1310   | 0.798   | 0.547   | 2.49              |

mesopores. For the materials activated by KOH (Fig. 3a–e), the amount of nitrogen adsorbed increased and reached a maximum at an activation temperature of 800 °C, whereas the volumes of micropores and mesopores underwent appreciable increases with an increase in activation temperature at KOH/C mass ratios of both 2 and 3. As can be clearly seen, samples with a KOH/C mass ratio of 2 achieved greater nitrogen adsorption performance than those with a KOH/C mass ratio of 3 at a certain activation temperature (Fig. 3a and b). For the samples activated by  $\text{ZnCl}_2$ , the amount of nitrogen adsorbed first decreased in the

activation temperature range of 600–700 °C and then increased with an increase in activation temperature from 700 °C to 800 °C (Fig. 3c and f). The optimal temperature was 800 °C, and the sample was denoted as ACZ-3-8.

In order to obtain better pore structural parameters and greater adsorption of  $\text{H}_2$ , further modified the activated carbons derived from cyanobacteria (denoted as ACK-2-8-1, ACK-2-8-2, and ACK-2-8-3, respectively) were characterized using the  $\text{N}_2$  adsorption–desorption method. The results shown in Fig. 4 and Table 4 imply that the other three samples exhibited type I nitrogen adsorption isotherms that were unexpectedly the same as that of ACK-2-8, and moreover the development of micro- and supermicropores at low relative pressures and some mesopores at higher relative pressures occurred for all the samples. It was easily found that ACK-2-8-3 exhibited the greatest adsorbed amount of  $\text{N}_2$  and the largest  $S_{\text{BET}}$  of  $2125 \text{ m}^2 \text{g}^{-1}$ . The samples ACK-2-8-1, ACK-2-8-2 and ACK-2-8-3 on  $\text{H}_2$  adsorption would be advanced undergone succeeding study.

### 3.3 Analysis of the surface morphology and functional groups

Scanning electron microscopy (SEM) was used to investigate the morphology of the cyanobacteria-derived activated carbons that underwent carbonization alone (AC), activation by KOH (ACK-2-



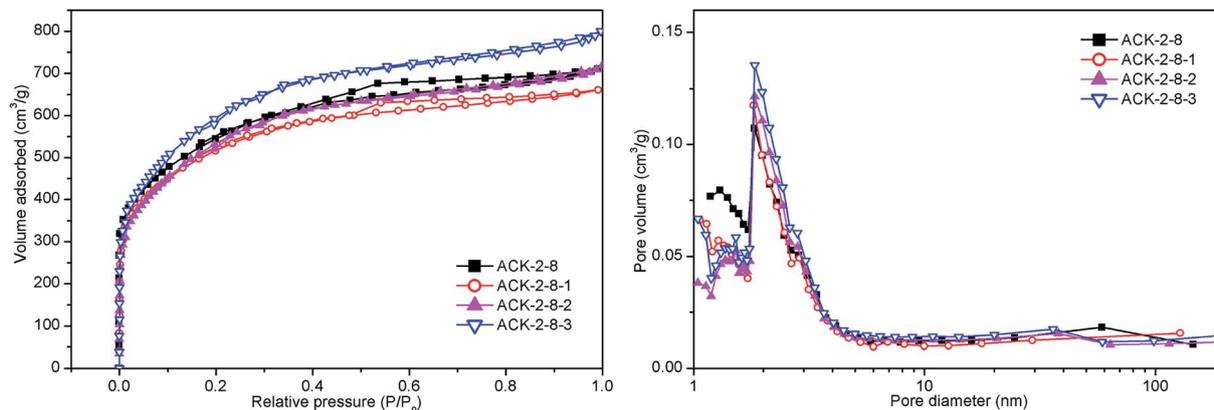


Fig. 4 NADIs and PSDs of ACK-2-8, ACK-2-8-1, ACK-2-8-2 and ACK-2-8-3.

Table 4 Structural parameters of ACK-2-8, ACK-2-8-1, ACK-2-8-2 and ACK-2-8-3

| Sample    | $S_{\text{BET}}$<br>( $\text{m}^2 \text{g}^{-1}$ ) | $V_{\text{total}}$<br>( $\text{cm}^3 \text{g}^{-1}$ ) | $V_{\text{micro}}$<br>( $\text{cm}^3 \text{g}^{-1}$ ) | $\bar{D}$<br>(nm) |
|-----------|--|---|---|-------------------|
| ACK-2-8   | 1951   | 1.092   | 0.756   | 2.27              |
| ACK-2-8-1 | 1846   | 1.023   | 0.735   | 2.22              |
| ACK-2-8-2 | 1901   | 1.026   | 0.726   | 2.26              |
| ACK-2-8-3 | 2125   | 1.238   | 0.834   | 2.33              |

8) and activation by  $\text{ZnCl}_2$  (ACZ-3-8). As can be seen in Fig. 5, the morphology was influenced regardless of the activation reagent used. AC retained an irregular plate-like morphology with relatively smooth surfaces (Fig. 5a and b). After the activation process, the plate-like structure of AC was destroyed, and the porous carbons were composed of a variety of sheets with

irregular shapes (Fig. 5c-f). These carbons contained uniform worm-like micropores with a random distribution, as illustrated by the SEM images. For the sample of ACK-2-8 (Fig. 5c and d), the carbon displayed an irregular worm-like structure that contained a considerable number of irregularly shaped pores of different sizes, which might be due to polymerization by KOH of the outer surface of AC when AC was impregnated with KOH *via* thermal treatment at high temperatures. In addition, KOH strongly etched and destroyed the plate-like morphology and the resulting activated carbon took the form of many small irregularly interconnected pores with pore walls. In comparison with ACK-2-8, ACZ-3-8 exhibited a sponge-like morphology and had wider pore sizes and fewer pores. This observation corresponded to the PSDs of the carbons deduced from the analysis of nitrogen sorption isotherms. The sponge-like morphology and wider pore size of ACZ-3-8 predicted that the  $\text{H}_2$  adsorption

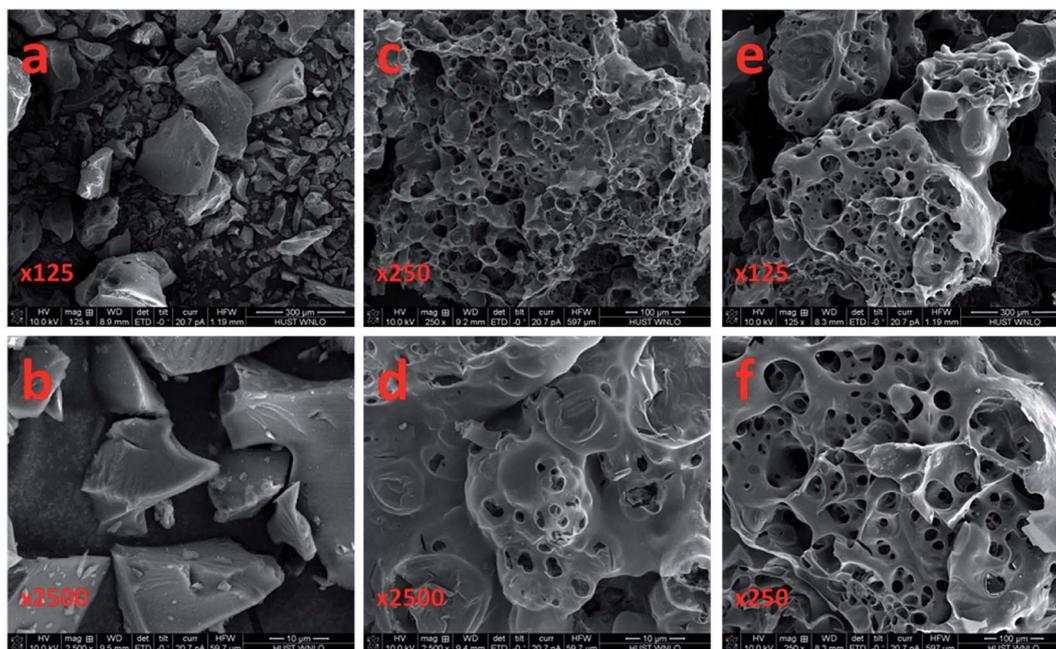


Fig. 5 Scanning electron micrographs of samples of AC (a and b), ACK-2-8 (c and d) and ACZ-3-8 (e and f).



capacity would be greater in comparison with that of AC, which had a plate-like morphology with smooth surfaces. The interesting structure of ACK-2-8, with its irregular worm-like morphology and many small irregularly interconnected pores with pore walls, predicted that it would be expected to give the best H<sub>2</sub> adsorption performance.

To determine the reason for the subsequent adsorption of H<sub>2</sub> on some cyanobacteria-derived activated carbons, the classic Boehm method was adopted to characterize the chemical properties of the adsorbent surface, and moreover X-ray photoelectron spectroscopy (XPS) was carried out to identify the nature of the nitrogen or phosphorus species that were impregnated into ACK-2-8 *via* modification with HNO<sub>3</sub>, H<sub>3</sub>PO<sub>4</sub> and NH<sub>3</sub>·H<sub>2</sub>O, respectively. As can clearly be seen in Fig. S3 (ESI<sup>†</sup>), the surface functional groups on the adsorbents were mainly acidic groups except in the case of ACK-2-8-3, which after impregnation with ammonium hydroxide contained 2.7 times more basic groups than ACK-2-8. Acid groups were particularly rich in ACK-2-8-1 and were approximately 20% more abundant than in ACK-2-8. In comparison, ACK-2-8-2 displayed a moderate decrease in the number of acidic groups, and the number of basic groups scarcely varied with respect to ACK-2-8. The XPS spectra of the doped elements used to further modify the carbon material ACK-2-8 are illustrated in Fig. 6. The resulting N 1s spectra (Fig. 6a and c) and P 2p spectrum (Fig. 6b) proved that nitrogen and phosphorus were successfully integrated into the porous carbons in a systematic way. In the N 1s spectrum of ACK-2-8-1, three peaks were observed at 398.55 eV, 400.2 eV, and 401.1 eV, which corresponded to pyridinic N, pyrrolic/pyridonic N and graphitic N, respectively.<sup>47–49</sup> However, pyrrolic N and pyridonic N species are difficult to distinguish within the range of accuracy of XPS measurements.<sup>5</sup> The nitrogen species of ACK-2-8-3 were represented (Fig. 6c) by peaks at 398.55 and 400.2 eV, which were assigned to pyridinic N and pyrrolic/pyridonic N, respectively. Furthermore, the single peak observed at 133.1 eV in the P 2p spectrum of ACK-2-8-2 indicated that phosphorous was doped in the form of P–O.<sup>50</sup>

### 3.4 Hydrogen adsorption performance

The adsorption of H<sub>2</sub> was measured at –196 °C and pressures of up to 1 bar. The hydrogen adsorption isotherms and hydrogen uptake of cyanobacteria-derived activated carbon activated by

KOH or ZnCl<sub>2</sub> at 1 bar are shown in Fig. 7 and Table 5, respectively. On comparing all four samples indicated in Fig. 7, in general the adsorbents activated by KOH exhibited higher hydrogen uptake than ACZ-3-8. Moreover, it is noticeable that ACK-2-8 displayed the highest hydrogen uptake of 17.3 mg g<sup>–1</sup>, which was 49.1% higher than that of the sample of ACZ-3-8. This result was expected because ACK-2-8 also had the largest BET surface area. Furthermore, the relationship between the BET surface area and the hydrogen uptake, which is given in Fig. 7b, was close to a linear relation, as the solid line represented simple linear regression with a correlation coefficient of  $R^2 = 0.9163$ . The strong correlation between the adsorption of H<sub>2</sub> and the BET surface area demonstrated that the adsorption of H<sub>2</sub> on cyanobacteria-derived carbons activated by KOH or ZnCl<sub>2</sub> is mainly governed by their specific surface area, which was in accordance with previous research.<sup>5,8,38</sup>

Similarly, Fig. 8 shows H<sub>2</sub> adsorption isotherms measured at –196 °C for further modified cyanobacteria-derived activated carbons. In comparison to ACK-2-8, the H<sub>2</sub> uptake at 1 bar increased in order from ACK-2-8-1 *via* ACK-2-8-3 to ACK-2-8-2, which represented the ACK-2-8 sample modified by HNO<sub>3</sub>, NH<sub>3</sub>·H<sub>2</sub>O and H<sub>3</sub>PO<sub>4</sub>, respectively. The H<sub>2</sub> uptake at –196 °C and a pressure of 1 bar was found to reach the highest value of 25.2 mg g<sup>–1</sup> for ACK-2-8-2, which was 45.7% higher than that for ACK-2-8 (Table 5). Nonetheless, it was remarkable that the  $S_{\text{BET}}$  of ACK-2-8-2 was not the largest of all four samples, as the  $S_{\text{BET}}$  of ACK-2-8-3 was 2125 m<sup>2</sup> g<sup>–1</sup>. A weaker linear relationship was found between the adsorption of H<sub>2</sub> and the  $S_{\text{BET}}$  with a correlation coefficient  $R^2$  of only 0.0214, as shown in Fig. S4 (ESI<sup>†</sup>). According to the prior study of surface functional groups (Fig. S3<sup>†</sup>), it could be concluded that the presence of basic groups would inhibit the uptake of H<sub>2</sub>, whereas the amount of acidic surface groups might favor the adsorption of H<sub>2</sub> to some extent. In comparison with the surface groups, the  $S_{\text{BET}}$  should exert a much greater effect on the uptake of H<sub>2</sub>. Regarding the ACK-2-8-1 and ACK-2-8-3 samples, in addition to the sufficiently large  $S_{\text{BET}}$  doping with nitrogen, as the XPS results showed, would favor the uptake of H<sub>2</sub>. This result was different from prior reports. Jiang *et al.*<sup>51</sup> considered that doping with nitrogen had no obvious positive influence on the adsorption of hydrogen at 77 K. Xia *et al.*<sup>52</sup> concluded that the effect of N doping on the uptake of hydrogen was only apparent when

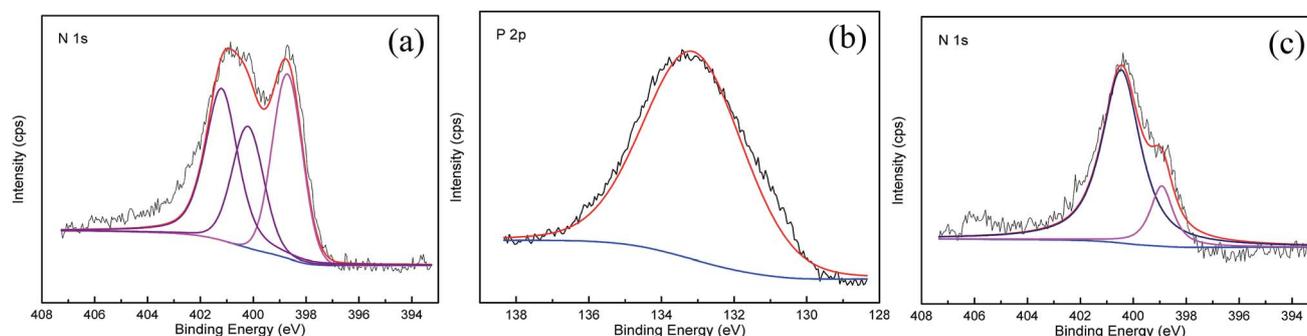


Fig. 6 XPS spectra of ACK-2-8 carbon materials with further modifications: (a) ACK-2-8-1, modified by HNO<sub>3</sub>, (b) ACK-2-8-2, modified by H<sub>3</sub>PO<sub>4</sub>, and (c) ACK-2-8-3, modified by NH<sub>3</sub>·H<sub>2</sub>O.



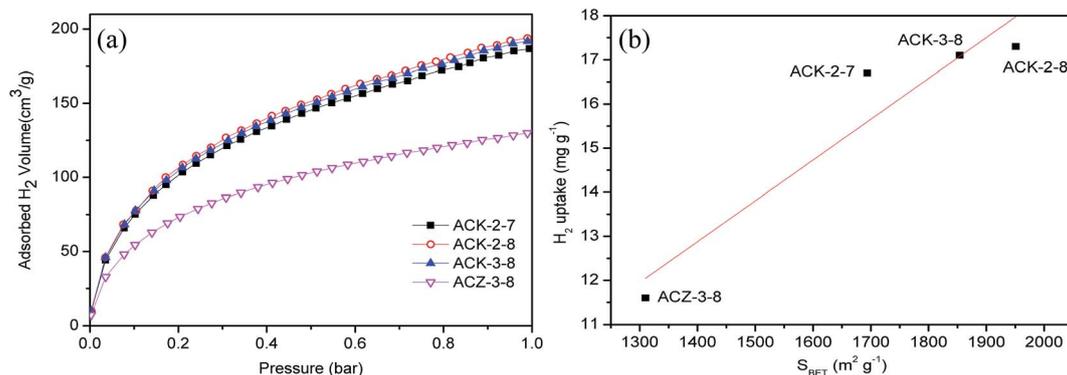


Fig. 7 (a) Hydrogen adsorption isotherms at  $-196\text{ }^{\circ}\text{C}$  and (b) relationship between the amount of  $\text{H}_2$  adsorbed (at  $-196\text{ }^{\circ}\text{C}$  and 1 bar) and the BET surface area.

Table 5 Comparison of  $\text{H}_2$  adsorption and  $S_{\text{BET}}$  of various materials

| Sample            |                                       | $S_{\text{BET}}$<br>( $\text{m}^2\text{ g}^{-1}$ ) | $\text{H}_2$ uptake<br>( $\text{mg g}^{-1}$ ) | $\text{H}_2$ uptake<br>per $S_{\text{BET}}$<br>( $\text{mg m}^{-2}$ ) |
|-------------------|---------------------------------------|--|---|---|
| This publication  | ACZ-3-8                               | 1310   | 11.6  | 0.0088  |
|                   | ACK-3-8                               | 1854   | 17.1  | 0.0092  |
|                   | ACK-2-7                               | 1694   | 16.7  | 0.0099  |
|                   | ACK-2-8                               | 1951   | 17.3  | 0.0089  |
|                   | ACK-2-8-1                             | 1846   | 20.2  | 0.0109  |
|                   | ACK-2-8-2                             | 1901   | 25.2  | 0.0133  |
| Ref.              | ACK-2-8-3                             | 2125   | 22.8  | 0.0107  |
|                   | C-4 (ref. 8)                          | 3870   | 39  | 0.01  |
|                   | P2 (ref. 31)                          | 1807   | <20   | <0.0111   |
|                   | Carbide-derived carbons <sup>10</sup> | <3000  | 3–27  | <0.01   |
|                   | MOFS <sup>7</sup>                     | 1132–4346  | 11.7–25.4                                     | <0.01   |
|                   | Zeolite-based carbons <sup>43</sup>   | 2254   | 22.7  | 0.0101  |
|                   | PC-2-800 (ref. 5)                     | 2919   | 27.1  | 0.0093  |
|                   | ACs <sup>28</sup>                     | <2500  | <20   | 0.008   |
|                   | ACs <sup>3</sup>                      | 2849   | <30   | <0.01   |
| CAC <sup>44</sup> | 1767–3711                             | 19.2–32.1  | 0.0086  |   |
| UCs <sup>45</sup> | 1127.2                                | 1.43   | 0.0013  |   |

related to the surface area and pore volume associated with micropores rather than the total porosity. The differences were probably due to the different carbon precursors and doping methods used. Doping with phosphorus in ACK-2-8-2 promoted the adsorption of  $\text{H}_2$  in this study, which was in accordance with a previous investigation.<sup>38</sup> Furthermore, when taking into account the uptake of  $\text{H}_2$  at the same time as the specific surface area of the material, it can clearly be seen from Table 5 that ACK-2-8-2 exhibited a noteworthy adsorption capacity of  $25.2\text{ mg g}^{-1}$  with a modest  $S_{\text{BET}}$   $1901\text{ m}^2\text{ g}^{-1}$  and holds great potential for the adsorption of  $\text{H}_2$ . In other words, ACK-2-8-2 displayed evidently superior hydrogen storage density (*i.e.*, hydrogen uptake per surface area).

## 4 Conclusions

Cyanobacteria, which are highly abundant waste materials, have been used as a cost-effective precursor to prepare a series of effective porous carbon sorbent materials by means of controlled activation with  $\text{KOH}/\text{ZnCl}_2$  and further modification by  $\text{HNO}_3$ ,  $\text{H}_3\text{PO}_4$  and  $\text{NH}_3\cdot\text{H}_2\text{O}$ , respectively. The resulting carbons activated by  $\text{KOH}$  exhibited larger amounts of micropores and mesopores and a larger specific surface area than those activated by  $\text{ZnCl}_2$ , even with the optimal activation parameters, *i.e.*, the  $\text{KOH}/\text{C}$  or  $\text{ZnCl}_2/\text{C}$  mass ratio, activation temperature, and activation time. In particular, ACK-2-8, which had an  $S_{\text{BET}}$  of  $1951\text{ m}^2\text{ g}^{-1}$ , displayed  $\text{H}_2$  uptake capacities of nearly  $17.3\text{ mg g}^{-1}$  at  $-196\text{ }^{\circ}\text{C}$  and 1 bar. Moreover, we demonstrated controlled doping with nitrogen or phosphorus and achieved increases in  $\text{H}_2$  uptake *via* the use of moderate concentrations of  $\text{HNO}_3$ ,  $\text{NH}_3\cdot\text{H}_2\text{O}$  and  $\text{H}_3\text{PO}_4$ , respectively. In comparison with the surface groups, the  $S_{\text{BET}}$  should have a much greater effect on the uptake of  $\text{H}_2$ . It was remarkable that the poor linear relationship between the adsorption of  $\text{H}_2$  and the  $S_{\text{BET}}$  was very different from the results reported for other carbon sorbents used for the uptake of  $\text{H}_2$ . Doping with phosphorus, in conjunction with a moderately large  $S_{\text{BET}}$ , was very effective for increasing the amount of  $\text{H}_2$  adsorbed, with the result that the ACK-2-8-2 sample, which had an  $S_{\text{BET}}$  of  $1901\text{ m}^2\text{ g}^{-1}$ , exhibited the adsorption of the greatest amount of  $\text{H}_2$ ,

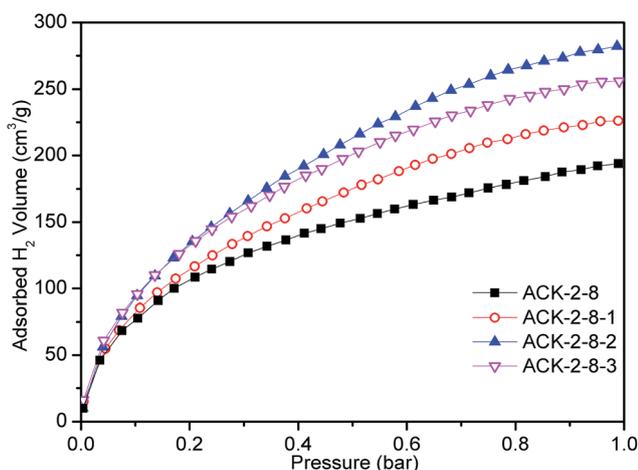


Fig. 8 Hydrogen adsorption isotherms normalized to the adsorption capacity from 0 to 1 bar for different samples.



namely, 25.2 mg g<sup>-1</sup> at -196 °C and 1 bar, and displayed evidently superior hydrogen storage density. Therefore, the better H<sub>2</sub> adsorption properties of this material with a moderate specific surface area hold great potential for hydrogen capture or storage in environmental and energy-related applications.

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## References

- 1 C. Robertson and R. Mokaya, *Microporous Mesoporous Mater.*, 2013, **179**, 151.
- 2 S. Schaefer, V. Fierro, M. T. Izquierdo and A. Celzard, *Int. J. Hydrogen Energy*, 2016, **41**, 12146.
- 3 V. Fierro, A. Szczurek, C. Zlotea, J. F. Mareche, M. T. Izquierdo, A. Albinia, M. Latroche, G. Furdin and A. Celzard, *Carbon*, 2010, **48**, 1902.
- 4 W. Zhao, V. Fierro, N. Fernandez-Huerta, M. T. Izquierdo and A. Celzard, *Int. J. Hydrogen Energy*, 2012, **37**, 14278.
- 5 Z. Q. Wang, L. X. Sun, F. Xu, H. Y. Zhou, X. J. Peng, D. L. Sun, J. C. Wang and Y. Du, *Int. J. Hydrogen Energy*, 2016, **41**, 8489.
- 6 P. Moriarty and D. Honnery, *Int. J. Hydrogen Energy*, 2009, **34**, 31.
- 7 J. L. C. Rowsell and O. M. Yaghi, *J. Am. Chem. Soc.*, 2006, **128**, 1304.
- 8 J. Choma, Ł. Osuchowski, M. Marszewski and M. Jaroniec, *RSC Adv.*, 2014, **4**, 14795.
- 9 H. Wang, Q. Gao and J. Hu, *J. Am. Chem. Soc.*, 2009, **131**, 7016.
- 10 M. Sevilla and R. Mokaya, *Energy Environ. Sci.*, 2014, **7**, 1250.
- 11 F. Cheng, J. Liang, J. Zhao, Z. Tao and J. Chen, *Chem. Mater.*, 2008, **20**, 1889.
- 12 M. T. Zheng, H. R. Zhang, Y. Xiao, H. W. Dong, Y. L. Liu, R. C. Xu, Y. K. Hu, B. Y. Deng, B. F. Lei and X. T. Liu, *Mater. Lett.*, 2013, **109**, 279.
- 13 C. Liu, Y. Chen, C. Z. Wu, S. T. Xu and H. M. Cheng, *Carbon*, 2010, **48**, 452.
- 14 G. Yang, L. Zhou, X. Liu, X. Han and X. Bao, *Microporous Mesoporous Mater.*, 2012, **161**, 168.
- 15 K. H. Chung, *Energy*, 2010, **35**, 2235.
- 16 A. G. Wong-Foy, A. J. Matzger and O. M. Yaghi, *J. Am. Chem. Soc.*, 2006, **128**, 3494.
- 17 Y. H. Hu and L. Zhang, *Adv. Mater.*, 2010, **22**, 117.
- 18 J. W. Ren, N. M. Musyoka, H. W. Langmi, B. C. North, M. Mathe, X. D. Kang and S. J. Liao, *Int. J. Hydrogen Energy*, 2015, **40**, 10542.
- 19 F. Gao, J. T. Sun and S. Meng, *Nanoscale*, 2015, **7**, 6319.
- 20 H. Furukawa and O. M. Yaghi, *J. Am. Chem. Soc.*, 2009, **131**, 8875.
- 21 M. Hirscher and B. Panella, *J. Alloys Compd.*, 2005, **404–406**, 399.
- 22 Q. Hu, Y. Lu and G. P. Meisner, *J. Phys. Chem. C*, 2008, **112**, 1516.
- 23 Z. Yang, Y. Xia and R. Mokaya, *J. Am. Chem. Soc.*, 2007, **129**, 1673.
- 24 E. Masika and R. Mokaya, *J. Phys. Chem. C*, 2012, **116**, 25734.
- 25 A. Pacuła and R. Mokaya, *J. Phys. Chem. C*, 2008, **112**, 2764.
- 26 W. C. Xu, K. Takahashi, Y. Matsuo, Y. Hattori, M. Kumagi, S. Ishiyama, K. Kaneko and S. Iijima, *Int. J. Hydrogen Energy*, 2007, **32**, 2504.
- 27 J. C. Wang, I. Senkowska, S. Kaskel and Q. Liu, *Carbon*, 2014, **75**, 372.
- 28 Y. Sun and P. A. Webley, *Chem. Eng. J.*, 2010, **162**, 883.
- 29 M. Sevilla, A. B. Fuertes and R. Mokaya, *Energy Environ. Sci.*, 2011, **4**, 1400.
- 30 L. Wei, M. Sevilla, A. B. Fuertes, R. Mokaya and G. Yushin, *Adv. Energy Mater.*, 2011, **1**, 356.
- 31 B. Hu, K. Wang, L. H. Wu, S. H. Yu, M. Antonietti and M. M. Titirici, *Adv. Mater.*, 2010, **22**, 813.
- 32 K. L. Wang, Y. H. Cao, X. M. Wang, Q. H. Fan, W. Gibbons, T. Johnson, B. Luo and Z. R. Gu, *Energy*, 2016, **94**, 666.
- 33 D. J. Barrington and A. Ghadouani, *Environ. Sci. Technol.*, 2008, **42**, 8916.
- 34 D. Lozano-Castello, M. A. Lillo-Rodenas, D. Cazorla-Amoros and A. Linares-Solano, *Carbon*, 2001, **39**, 741.
- 35 W. Zhao, V. Fierro, C. Zlotea, E. Aylon, M. T. Izquierdo, M. Latroche and A. Celzard, *Int. J. Hydrogen Energy*, 2011, **36**, 5431.
- 36 D. Lozano-Castello, D. Cazorla-Amoros, A. Linares-Solano and D. F. Quinn, *Carbon*, 2002, **40**, 989.
- 37 M. Armandi, B. Bonelli, K. Cho, R. Ryoo and E. Garrone, *Int. J. Hydrogen Energy*, 2011, **36**, 7937.
- 38 D. Li, W. B. Li, J. S. Shi and F. W. Xin, *RSC Adv.*, 2016, **6**, 50138.
- 39 J. P. Paraknowitsch and A. Thomas, *Energy Environ. Sci.*, 2013, **6**, 2839.
- 40 X. Ma, M. Cao and C. Hu, *J. Mater. Chem. A*, 2013, **1**, 913.
- 41 B. Ashourirad, A. K. Sekizkardes, S. Altarawneh and H. M. ElKaderi, *Chem. Mater.*, 2015, **27**, 1349.
- 42 H. P. Boehm, *Carbon*, 1994, **32**, 759.
- 43 H. Hu, X. Lu, F. Wang, J. J. He, J. Li and M. H. Fan, *Carbon*, 2011, **49**, 2197.
- 44 M. S. Kim, J. B. Jeon and J. Y. Chang, *J. Mater. Chem.*, 2012, **22**, 20713.
- 45 L. Deng, T. Zhang and D. F. Che, *Fuel Process. Technol.*, 2013, **106**, 712.
- 46 H. R. Wei, S. B. Deng, B. Y. Hu, Z. H. Chen, B. Wang, J. Huang and G. Yu, *ChemSusChem*, 2012, **5**, 2354.
- 47 Y. J. Qiu, J. Yu, T. N. Shi, X. S. Zhou, X. D. Bai and J. Y. Huang, *J. Power Sources*, 2011, **196**, 9862.
- 48 J. Yin, Y. J. Qiu and J. Yu, *Electrochem. Commun.*, 2013, **30**, 1.
- 49 F. Kapteijn, J. A. Moulijn, S. Matzner and H. P. Boehm, *Carbon*, 1999, **37**, 1143.
- 50 C. H. Choi, S. H. Park and S. I. Woo, *ACS Nano*, 2012, **6**, 7084.
- 51 J. H. Jiang, Q. M. Gao, Z. J. Zheng, K. S. Xia and J. Hu, *Int. J. Hydrogen Energy*, 2010, **35**, 210.



- 52 Y. Xia, G. S. Walker, D. M. Grant and R. Mokaya, *J. Am. Chem. Soc.*, 2009, **131**, 16493.
- 53 J. J. Cai, L. J. Li, X. X. Lv, C. P. Yang and X. B. Zhao, *ACS Appl. Mater. Interfaces*, 2014, **6**, 167.
- 54 Z. Geng, C. M. Zhang, D. B. Wang, X. Y. Zhou and M. Cai, *J. Energy Chem.*, 2015, **24**, 1.
- 55 J. J. Cai, M. D. Yang, Y. L. Xing and X. B. Zhao, *Colloids Surf., A*, 2014, **444**, 240.

