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# Ideal strength and elastic instability in single-layer 8-*Pmmn* borophene†

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Using first principle calculations based on density functional theory (DFT), we have investigated the mechanical properties of monolayer 8-*Pmmn* borophene, including ideal strength and critical strain. It was found that monolayer 8-*Pmmn* borophene can withstand stress up to 21.59 N m<sup>-1</sup> and 27.79 N m<sup>-1</sup> in *a* and *b* directions, respectively. The ultimate tensile strain along the *a* direction is about 0.155, and 0.16 along the *b* direction, which is larger than that of stripe borophene. Furthermore, the phonon dispersions under the three applied strains indicate that the mechanical failure in 8-*Pmmn* borophene is likely to originate from the elastic instability. Our study shows that 8-*Pmmn* borophene is a hard but flexible 2D material, and may potentially be useful in high-strain engineering applications.

## 1 Introduction

Beyond the scope of group-IV and group-V elementary two-dimensional (2D) materials,<sup>1–10</sup> a novel 2D material of group-III named borophene (2D boron sheet) has been grown successfully on Ag (111) substrates very recently.<sup>11,12</sup> Three types of borophene sheets are have been fabricated in the experiments, *i.e.* stripe,  $\beta_{12}$ , and  $\chi_3$  borophene. The variety of the geometric structures of borophene stems from boron's unique physical properties. For example, the monolayer borophene exhibits highly anisotropic metallic characteristic and unique mechanical properties, such as high ideal strength and critical strain.<sup>13–15</sup> On the other hand, the stripe borophene possesses an extremely large Young's modulus of 398 GPa nm along the *a* direction, much larger than that of graphene.<sup>11,15</sup> More interestingly, the stripe borophene has an out-of-plane negative Poisson's ratio, similar to black phosphorus (BP).<sup>15</sup>

Although most forms of borophene are good metals, striped borophene exhibits a direction-dependent Dirac cone when fully hydrogenated. In addition, the intrinsic band structure of 8-*Pmmn* borophene also has a Dirac cone.<sup>16</sup> 8-*Pmmn* borophene has been predicted to be highly stable by several theoretical studies,<sup>17,18</sup> but has not been synthesized experimentally yet. Previous calculations reveal that strong structural anisotropy exists in 8-*Pmmn* borophene, similar to the cases of stripe,  $\beta_{12}$ , and  $\chi_3$  borophene. Hence, 8-*Pmmn* borophene may be transformed into an indirect band gap semiconductor by means of external shear stress.<sup>17</sup> These discoveries make 8-*Pmmn*

borophene a very promising candidate for next generation flexible electronics. Moreover, due to the lattice mismatch between the materials and the substrates or external environmental factors, an adventitious strain is almost unavoidable experimentally. Thus, to investigate the failure mechanisms of 2D materials under tension strain is very important for the research and application.<sup>19,20</sup> Also as different applications fields have different performance requirements for materials, thus it is necessary to selectively control the properties of the two-dimensional material before it is prepared as an electronic device.<sup>21</sup> Meanwhile, apply strains have been confirmed to be an efficient way to adjust the electronic structure of 2D materials.<sup>22–25</sup> Although the elastic constants and the intrinsic electronic structure of the 8-*Pmmn* borophene have been studied by very few studies, the mechanical properties of 8-*Pmmn* borophene have not been comprehensive investigated, especially the elastic limit, how its electronic structure varies with strain and failure mechanisms under tension strain. Therefore, it is essential to investigate the mechanical properties of 8-*Pmmn* borophene.

In this work, we have conducted a comprehensive investigation on the strain-induced mechanical properties of monolayer 8-*Pmmn* borophene based on density functional theory (DFT). The ultimate stress and critical strain, as well as the failure mechanism when approaching the limit strain have been discussed. Our calculation shows that the 8-*Pmmn* borophene exhibit superior mechanical properties. The monolayer 8-*Pmmn* borophene can withstand up to 21.59 N m<sup>-1</sup> and 27.79 N m<sup>-1</sup> in *a* and *b* directions, which is much higher than that of stripe borophene,  $\beta_{12}$  borophene,  $\chi_3$  borophene, silicene, MoS<sub>2</sub> and BP, indicating 8-*Pmmn* borophene a promising flexible material. The calculated Young's moduli of 8-*Pmmn* borophene are 241.08 N m<sup>-1</sup> (*a* direction) and 305.20 N m<sup>-1</sup> (*b* direction), respectively, which is comparable to that of steel, make it a hard

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and flexible 2D material which can withstand strong strains compared with other 2D materials. The strain also has a significant effect on the electronic structure of 8-*Pmmn* borophene.

## 2 Method and computational details

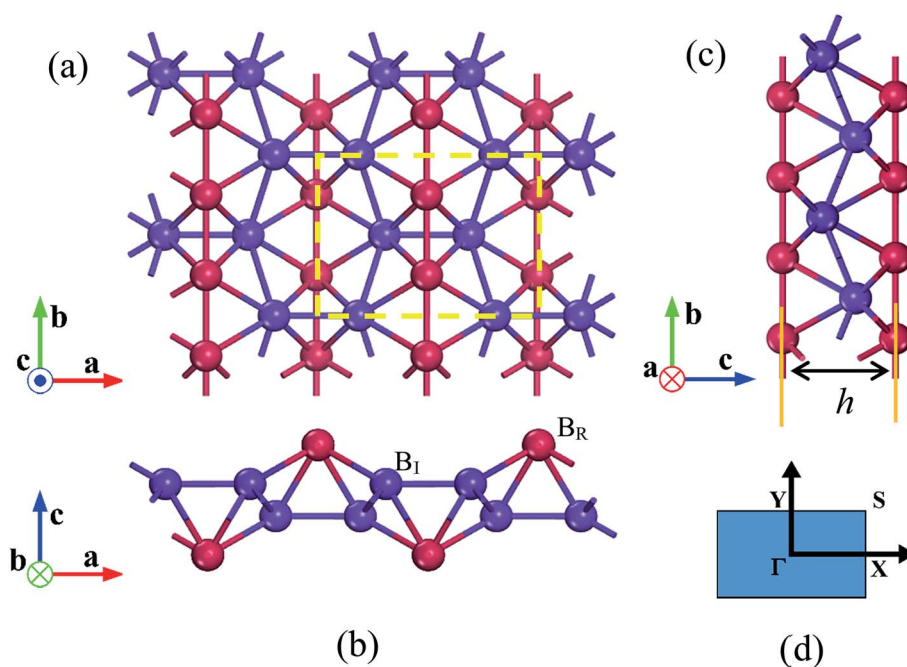
All DFT calculations were performed using plane-wave-based Vienna *Ab initio* Simulation Package (VASP) code.<sup>26,27</sup> The generalized gradient approximation (GGA) within the Perdew–Burke–Ernzerhof (PBE) functional form was used for the exchange–correlation energy,<sup>28</sup> and projector augmented-wave (PAW) pseudopotentials were used to replace the core electrons.<sup>29,30</sup> The plane wave energy cutoff was fixed to 500 eV. For all structural relaxations, the convergence criteria for total energy was set to  $1.0 \times 10^{-7}$  eV, and structural optimization was carried out until the Hellmann–Feynman force acting on each atom was less than  $0.001 \text{ eV \AA}^{-1}$ . A  $15 \times 20 \times 1$  *k*-point mesh generated by the scheme of Monkhorst–Pack was chosen for Brillouin zone sampling.<sup>31</sup> In order to reduce the interaction between neighboring layers, a vacuum slab of  $15 \text{ \AA}$  along the *z*-axis was introduced. The phonon dispersion was calculated with the density functional perturbation theory, using the phonopy code.<sup>32</sup>

The theoretical stress–strain relation was predicted by following a standard method as follows.<sup>33,34</sup> To compute the stress–strain relation as defined in Fig. 1, we applied a series of incremental tensile strains to the monolayer 8-*Pmmn* borophene and relaxed on the rectangular directions until the corresponding conjugate stress components were less than 0.01 GPa. The engineering tensile strain is defined as  $\varepsilon = -(l - l_0)/l_0$ ,

where *l* is the strained lattice constants and *l*<sub>0</sub> is the original lattice constants, respectively. Since the interlayer distance of borophene has not been determined experimentally, we used the in-plane stress *f* (2D force per length with a unit of  $\text{N m}^{-1}$ ) to represent the strength of the structure.<sup>35</sup> The 2D stress can be expressed by multiplying the Cauchy stresses and the thickness of the unit cell.

## 3 Results and discussion

The optimized structures of 8-*Pmmn* borophene are shown in Fig. 1. Compared with the other three types of borophene (stripe,  $\beta_{12}$  and  $\chi_3$ , see ESI, Fig. ESI1†), the 8-*Pmmn* borophene has a much different geometric structure with four boron atomic layers. The calculated lattice constants and buckling height are  $a = 4.523 \text{ \AA}$ ,  $b = 3.258 \text{ \AA}$  and  $h = 2.18 \text{ \AA}$ , in good agreement with the previous theoretical results.<sup>16–18</sup> 8-*Pmmn* borophene has a highly anisotropic crystal structure with two types of nonequivalent B atoms in a unit cell, namely ridge atoms ( $B_R$ ) and inner atoms ( $B_I$ ), as shown in Fig. 1b. The distinct local environments of  $B_R$  and  $B_I$  atoms lead to different local physical and chemical properties.<sup>17</sup> As the 8-*Pmmn* borophene has not been experimentally realized for now, so we firstly discussed the relative stability of the four types of borophene (8-*Pmmn*, stripe,  $\beta_{12}$  and  $\chi_3$  borophene) by comparing the total energy (see ESI, Table ESI1†). The calculated total energy *E* of 8-*Pmmn*, stripe,  $\beta_{12}$  and  $\chi_3$  borophene are  $-6.325$ ,  $-6.187$ ,  $-6.232$  and  $-6.244$  eV per atom, respectively, which is in line with the previous theory studies.<sup>13,36</sup> Remarkably, the calculated results show that 8-*Pmmn* borophene is 0.138, 0.093 and 0.081 eV per atom, respectively, lower in energy than that of



**Fig. 1** The optimized geometric structures of 8-*Pmmn* borophene. (a) Top view, (b) and (c) side views of 8-*Pmmn* borophene. The color scheme distinguishes the two types of nonequivalent B atoms, namely the atoms along the nanostructure ridges ( $B_R$ ), and the atoms bonded to the them in inner positions ( $B_I$ ). Dashed rectangles delimit the unit cell; (d) Brillouin zone of 8-*Pmmn* borophene.



the most recently synthesized stripe,  $\beta_{12}$  and  $\chi_3$  borophene, indicating its high possibility for experimental realization. The calculated cohesive energies ( $E_c$ ) confirmed the previous results as well (see ESI, Table ESI1†). Therefore, the 8-*Pmmn* structure of borophene is more energetic favourable than other three forms of borophene.

Fig. 2 shows the calculated ideal tensile stress–strain relations for 8-*Pmmn* borophene under both biaxial and uniaxial strains. By fitting the initial stress–strain curves based on linear regression up to 2% along the *a* and *b* directions (Fig. 2, inset), we obtained the corresponding elastic moduli. The Young's moduli are  $E_a = 241.08 \text{ N m}^{-1}$  and  $E_b = 305.20 \text{ N m}^{-1}$ , in line with the previous theoretical results.<sup>37</sup> The stiffness of 8-*Pmmn* borophene along *a* direction is substantially lower than that along *b* direction, indicating the strong anisotropy of the crystal structure. In addition, the stiffness of 8-*Pmmn* borophene is much higher than that of  $\chi_3$  and  $\beta_{12}$  borophene,<sup>13</sup> as shown in Table 1. Compared with the stripe borophene, though the Young's modulus along *a* direction ( $E_a$ ) of 8-*Pmmn* borophene is lower than that of stripe borophene, the Young's modulus along

*b* direction ( $E_b$ ) is much higher than that of stripe borophene. Compared to other 2D materials, such as graphene,<sup>39</sup> BN,<sup>40</sup> silicene,<sup>41</sup> and MoS<sub>2</sub>,<sup>42</sup> the Young's moduli of graphene are higher than that of 8-*Pmmn* borophene in both directions, but the Young's moduli of silicene and MoS<sub>2</sub> are much lower than that of 8-*Pmmn* borophene. The Young's moduli of BN is slightly higher than that of 8-*Pmmn* borophene along *a*-axis, but lower than that of 8-*Pmmn* borophene along *b*-axis. In addition, our calculated Poisson's ratios of 8-*Pmmn* borophene (as listed in Table 1) are around 0.042 and 0.062 along the *a* and *b* directions, respectively, also in good agreement with the previous study.<sup>37</sup> Compared with the other two types of borophenes, the Poisson's ratios of 8-*Pmmn* borophene is much smaller than that of  $\chi_3$  and  $\beta_{12}$  borophene, indicating the higher stiffness of 8-*Pmmn* borophene in consistent with its Young's modulus.

The calculated strain–stress relation becomes nonlinear as the applied strain increases, as shown in Fig. 2. The stress–strain curve under the uniaxial strain in *a* or *b* direction follows the similar tendency as that under the biaxial strain until  $\epsilon > 0.12$ . The maximum stress for biaxial strain is  $21.89 \text{ N m}^{-1}$ , and the corresponding critical strain is 0.13. The tensile strength value is much larger than that of stripe borophene,<sup>15</sup> silicene,<sup>41</sup> and MoS<sub>2</sub> (also shown in Table 2).<sup>42</sup> When the biaxial strain exceeds 0.13, the stress curve decreased rapidly. Along the *a* and *b* directions, the critical tensile strains are 0.155 and 0.16, respectively. The corresponding ideal strengths are 21.59 and  $27.79 \text{ N m}^{-1}$ , both higher than that of stripe borophene,<sup>15</sup>  $\beta_{12}$  borophene,<sup>38</sup>  $\chi_3$  borophene,<sup>38</sup> silicene,<sup>41</sup> MoS<sub>2</sub>,<sup>42</sup> and BP,<sup>43</sup> but much lower than that of graphene.<sup>39</sup> Compared to the biaxial strain or uniaxial strain along *a* direction, 8-*Pmmn* borophene shows superior flexibility when tension is applied along the *b* direction. The high ultimate strains make 8-*Pmmn* borophene potentially useful in high-strain engineering applications.

Fig. 3 shows the dependence of buckling height *h* of 8-*Pmmn* borophene on the three types of tensions, where the tension dependent buckling heights are monotonic. For the biaxial strain, the buckling height decreased significantly, especially when the strain  $\epsilon > 0.13$ . On the other hand, when the tension is applied along the *a* or *b* direction, the buckling height decreased monotonously and almost linearly before the strain

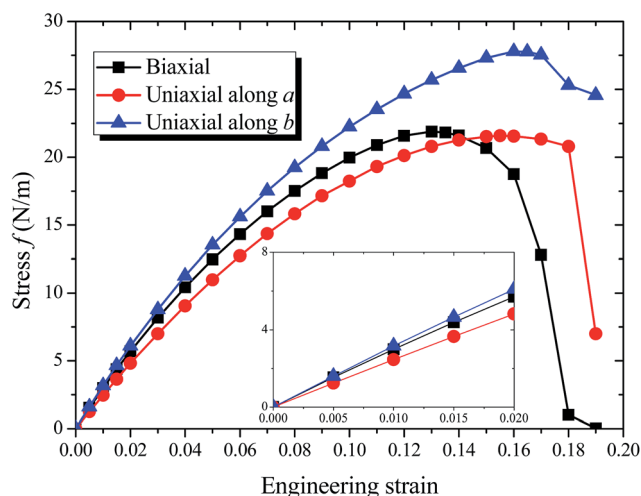


Fig. 2 The calculated stress–strain relationships of borophene under uniaxial *a*, *b* and biaxial tension. Insert shows the linear regression of the initial stress strain curves for the corresponding elastic moduli.

Table 1 Calculated lattice constant *a* and *b*, buckling height *h*, and Young's modulus ( $E_a$  and  $E_b$ ) and Poisson's ratio  $\nu$  of 8-*Pmmn* borophene. Previous theoretical data are also listed for comparison

	<i>a</i> (Å)	<i>b</i> (Å)	<i>h</i> (Å)	$E_a$ (N m <sup>-1</sup> )	$E_b$ (N m <sup>-1</sup> )	$\nu_a$	$\nu_b$
8- <i>Pmmn</i> borophene	4.523	3.258	2.18	241.08	305.20	0.042	0.062
Ref. 18	4.52	3.25	2.21	—	—	—	—
Ref. 17	4.51	3.25	2.20	—	—	—	—
Ref. 37	—	—	—	248.26	321.54	0.048	0.062
Stripe borophene <sup>15</sup>	1.614	2.866	0.911	389	166	-0.012	-0.005
$\chi_3$ borophene <sup>13</sup>	4.45	4.45	0	187.8	180.8	0.19	0.19
$\beta_{12}$ borophene <sup>13</sup>	5.07	2.93	0	182.0	207.5	0.17	0.19
Graphene <sup>39</sup>	—	—	—	342.2	342.2	0.175	0.175
BN <sup>40</sup>	—	—	—	275.8	275.8	—	—
Silicene <sup>41</sup>	—	—	—	60.6	60.6	—	—
MoS <sub>2</sub> (ref. 42)	—	—	—	197.9	200.3	—	—

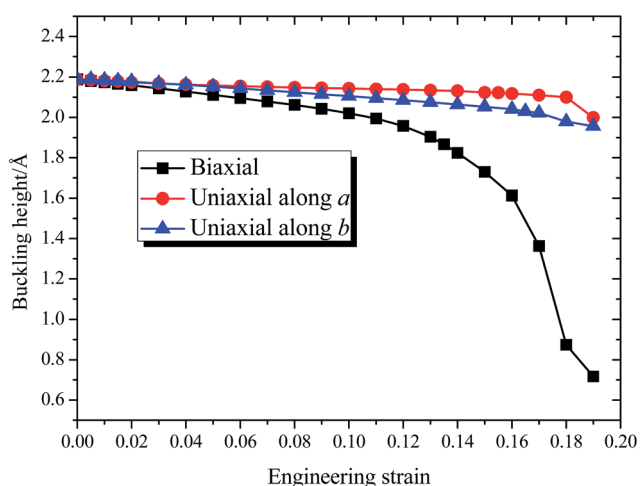


**Table 2** Summary of the ideal tensile strength, critical strains, and failure mechanisms of monolayer 8-*Pmmn* borophene under three strain paths and comparison with stripe borophene,  $\beta_{12}$  borophene,  $\chi_3$  borophene, graphene, silicene, MoS<sub>2</sub> and BP

	Direction	$f$ (N m <sup>-1</sup> )	$\epsilon$	Failure mechanism
8- <i>Pmmn</i> borophene	<i>a</i>	21.59	0.155	Elastic instability
	<i>b</i>	27.79	0.165	Elastic instability
	Biaxial	21.89	0.13	Elastic instability
Stripe borophene <sup>15</sup>	<i>a</i>	20.26	0.08	Phonon instability
	<i>b</i>	12.98	0.15	Elastic instability
	Biaxial	14.75	0.08	Phonon instability
$\beta_{12}$ borophene <sup>38</sup>	Armchair	19.97	0.2	—
	Zigzag	20.38	0.21	—
	Armchair	19.91	0.11	—
$\chi_3$ borophene <sup>38</sup>	Zigzag	20.18	0.21	—
	Armchair	40.41	0.266	Elastic instability
	Zigzag	36.74	0.194	Elastic instability
Silicene <sup>41</sup>	Armchair	7.59	0.17	Elastic instability
	Zigzag	5.26	0.136	Elastic instability
	Biaxial	6.76	0.16	Elastic instability
MoS <sub>2</sub> (ref. 42)	Armchair	14.75	0.256	Phonon instability
	Zigzag	9.59	0.18	Elastic instability
	Biaxial	14.63	0.195	Phonon instability
BP <sup>43</sup>	Armchair	9.99	0.27	—
	Zigzag	4.44	0.33	—

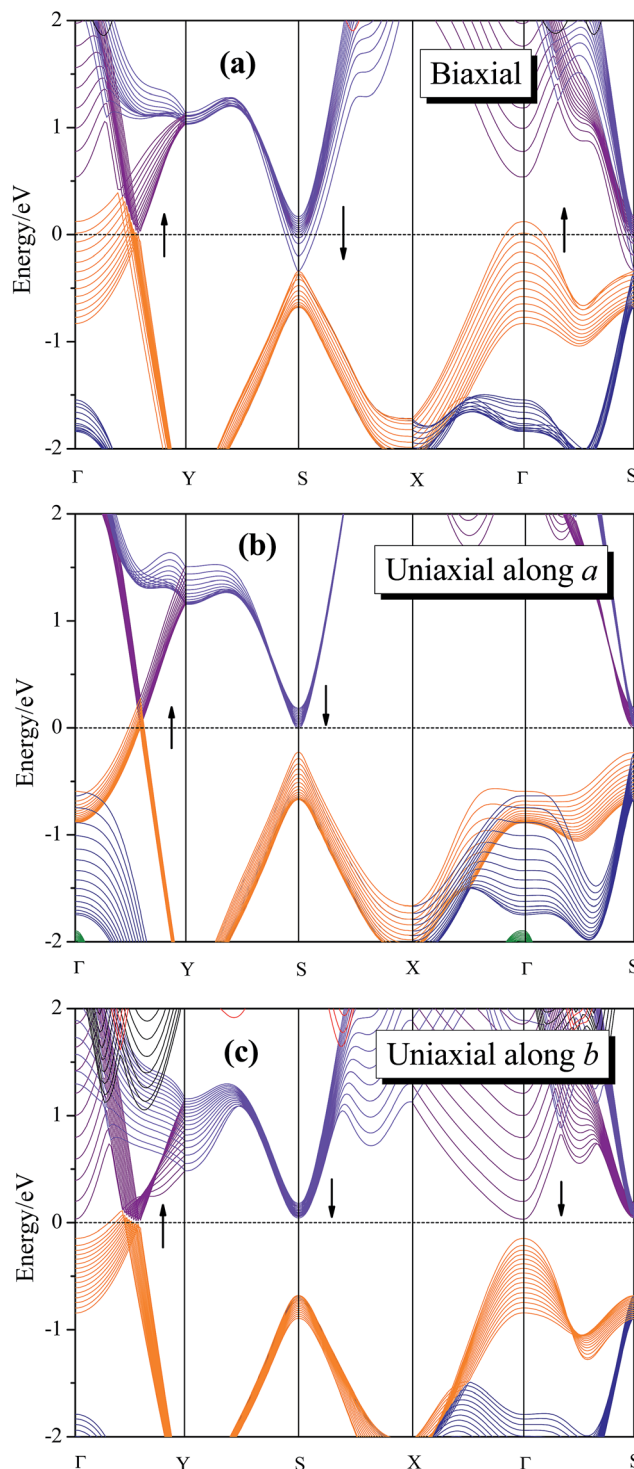
approaches the critical value. Note that the buckling height of the 8-*Pmmn* borophene does not decrease to zero within the applied strain range, in contrast to stripe borophene, which turns from buckled structure to planar structure when a biaxial strain of 0.13 is applied.<sup>15</sup>

Applying mechanical strain is proved to be a powerful method for modulating the electronic properties of 2D materials.<sup>22–25</sup> The intrinsic band structure of 8-*Pmmn* borophene has a Dirac cone located along  $\Gamma$ -*Y* direction.<sup>16–18</sup> The spatial inversion symmetry is of great significance for the sake of Dirac cone formation, and the broken of the spatial inversion symmetry may have great effects on characteristic of the Dirac



**Fig. 3** The calculated dependence of buckling height of 8-*Pmmn* borophene on the three types of tensions.

fermion in the system. For example, graphene will break its symmetry after applying a certain strain, which yields a band gap and destroys its Dirac cone.<sup>44</sup> Therefore, it is important to study the electronic structure of 8-*Pmmn* borophene under



**Fig. 4** The calculated dependence of band structure of 8-*Pmmn* borophene on the three types of tensions. (a) Biaxial strain (0.01–0.13); (b) uniaxial strain along *a* direction (0.01–0.15); (c) uniaxial strain along *b* direction (0.01–0.16).



strain. The calculated dependence of band structure of 8-*Pmmn* borophene on the three types of tensions are shown in Fig. 4. Fig. 4a shows the band structure of 8-*Pmmn* borophene under biaxial strain range from 0.01 to 0.13. According to our calculation, 8-*Pmmn* borophene will open a tiny energy gap (about 44 meV) with a small ( $\epsilon = 0.02$ ) strain along  $\Gamma$ -Y direction. However, within the increase of strain, the valence band maximum (VBM) along  $\Gamma$ -Y direction will cross over the Fermi level. Meanwhile, the conduction band at high symmetry point *S* and the valence band located at  $\Gamma$  will also pass through the Fermi energy, as marked by the arrows in the Fig. 4a. Thus, the metallic characteristics of 8-*Pmmn* borophene will be enhanced by applying a large biaxial strain. When the applied uniaxial strain increases along the *a* direction, similar to the biaxial strain condition, the VBM located at  $\Gamma$ -Y direction will cross the

Fermi level, and the conduction band located at *S* will gradually decrease, approaching the Fermi surface (as drawn in Fig. 4b), both tend to enhance the metallic behavior of 8-*Pmmn* borophene. Finally, when the applied uniaxial strain is along the *b* direction, 8-*Pmmn* borophene will open a small energy gap (about 23 to 48 meV), in contrast to the case of *a*-axis strain. However, when the *b*-axis strain reach 7%, the VBM along  $\Gamma$ -Y direction will cross the Fermi level, which transforms 8-*Pmmn* borophene from a semiconductor to a metal (Fig. 4c).

The 8-*Pmmn* borophene possesses good mechanical properties with relatively high critical strain under the three types of strain. Thus, it is important to verify whether the monolayer 8-*Pmmn* borophene remains stable before approaching the maximum stress. Phonon dispersion is an effective way to estimate structural stability of crystal structures. Accordingly,

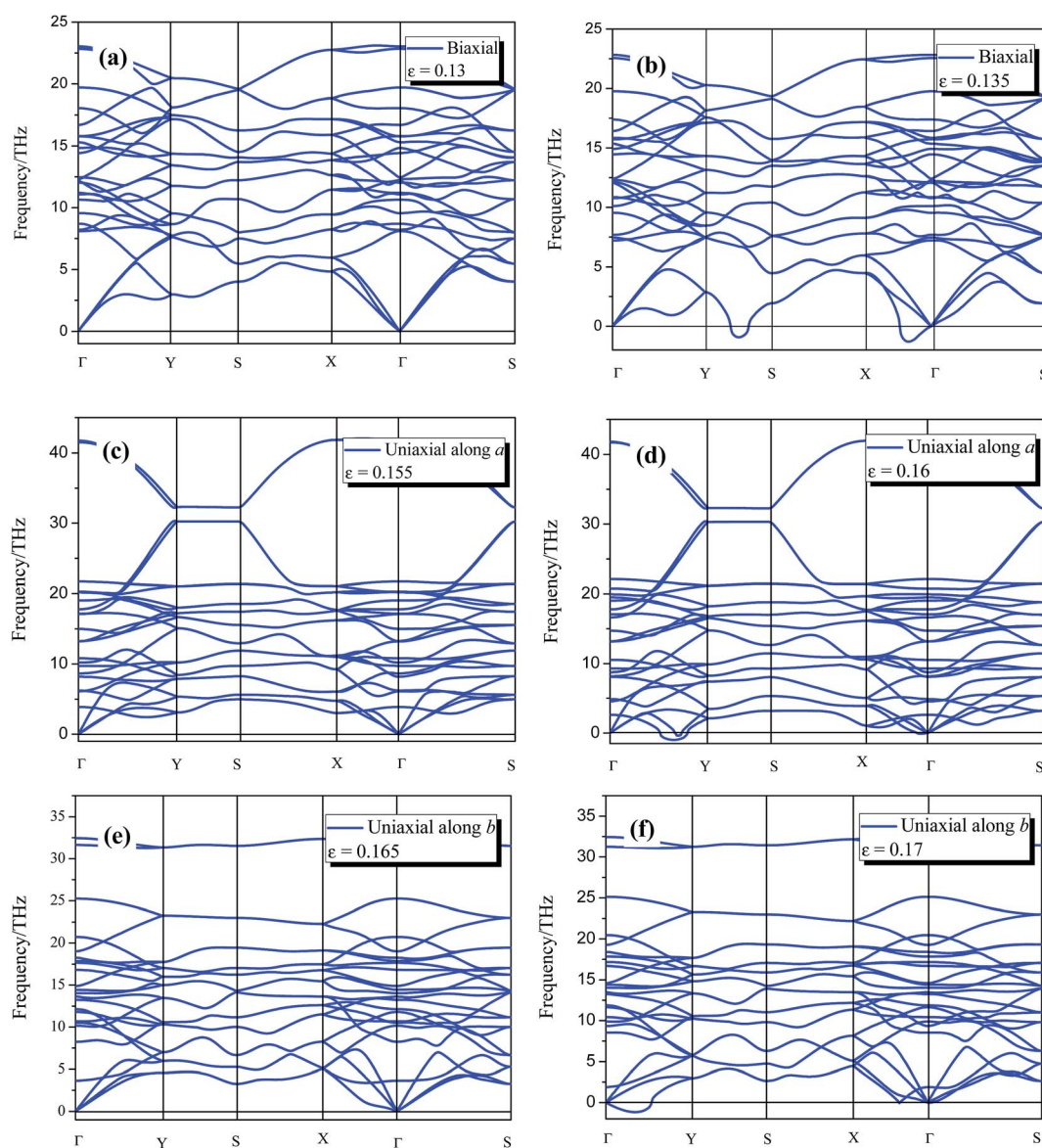


Fig. 5 The phonon dispersion of 8-*Pmmn* borophene under a biaxial strain of (a)  $\epsilon = 0.13$ , (b)  $\epsilon = 0.135$ ; under a uniaxial strain of (c)  $\epsilon = 0.155$ , (d)  $\epsilon = 0.16$  in the *a*-direction; and under a uniaxial strain of (e)  $\epsilon = 0.165$ , (f)  $\epsilon = 0.17$  in the *b*-direction.



we performed phonon dispersion calculations for the systems in order to examine the kinetic stability. The calculated phonon dispersions of 8-*Pmmn* borophene under the three types of applied strains are shown in Fig. 5. Under the biaxial strain  $\varepsilon = 0.13$ , the system exhibits good stability due to the phonon dispersion without imaginary frequencies. Nevertheless, as the biaxial strain increases to 0.135, imaginary frequencies appear in the  $Y-S$  and  $\Gamma-X$  direction. Under the uniaxial strain in the  $a$ -direction, the imaginary frequencies appear when the strain reaches  $\varepsilon_a = 0.16$ . We also noticed that top-most phonon branches are extremely flat between  $Y$  and  $S$ , as shown in Fig. 5c and d. These phonon dispersion curves (flat phonon dispersion) mean the localization of phonons and may cause the low thermal conductivity.<sup>45–48</sup> Under  $b$ -axis uniaxial strain, imaginary frequencies appear at  $\varepsilon_b = 0.17$ . The phonon dispersions of 8-*Pmmn* borophene under tensile strains indicate that 8-*Pmmn* borophene can withstand up to 15.5% and 16.5% uniaxial tensile strains along the  $a$  and  $b$  directions, respectively, and 13% biaxial tensile strain. It is generally accepted that imaginary frequencies of phonon dispersion are signs of instability. Hence, 8-*Pmmn* borophene would remain stable when reaching the ultimate tensile strain limits. In other words, the failure mechanisms of 8-*Pmmn* borophene under biaxial and uniaxial tensions are elastic instability.

To understand the chemical bonding of 8-*Pmmn* borophene under three types of strains, we calculated its bond lengths (see ESI, Fig. ESI2†) and valence charge density (see ESI, Fig. ESI3a–d†) as well. The B–B bond lengths ( $l_1$ – $l_5$ , see ESI, Fig. ESI2d†) show a linear variation under the biaxial ( $\varepsilon < 0.13$ ) and uniaxial strains. As the biaxial strain goes beyond 0.13, the bond lengths change rapidly, which explains the change of the buckling height  $h$  (as drawn in Fig. 3). The applied strain has a great influence on the B–B bond of 8-*Pmmn* borophene. As the strain increases, the electron around boron atom becomes more localized (especially in the case of biaxial strain, see ESI, Fig. ESI3b†) and the B–B bonds are transformed from a strong covalent bond to a weak ionic bond, as shown in ESI, Fig. ESI3.† The weakening of the B–B bonding also led to the structural instability of 8-*Pmmn* borophene under sufficiently large strain, as shown in Fig. 5.

Within the context of novel 2D materials, we also compared the mechanical responses of 8-*Pmmn* borophene with stripe borophene,<sup>15</sup>  $\beta_{12}$  borophene,<sup>38</sup>  $\chi_3$  borophene,<sup>38</sup> graphene,<sup>39</sup> silicene,<sup>41</sup> MoS<sub>2</sub>,<sup>42</sup> and BP,<sup>43</sup> as shown in Table 2. Compared with the other three types of borophene, silicene, MoS<sub>2</sub> or BP, the ideal tensile strengths of monolayer 8-*Pmmn* borophene are substantially larger, but much smaller than that of graphene, due to the C–C bonds are stronger than the B–B bonds. The corresponding strain along the  $a$  direction (15.5%) is also much larger than that of stripe (8%) and  $\chi_3$  borophene (11%), but slightly smaller than that of  $\beta_{12}$  borophene. For the tension applied in  $b$  direction, the 8-*Pmmn* borophene shows superior mechanical flexibility as well, just like the other three types of borophene. The failure mechanisms of 8-*Pmmn* borophene are similar to graphene and silicene, *i.e.*, under uniaxial and biaxial tensions their failure mechanism is elastic instability. Such is

quite different from the failure mechanisms of stripe borophene and MoS<sub>2</sub> upon tensions, where in one direction ( $b$  or zigzag direction) attributed to elastic instability, while in the other direction ( $a$  or armchair direction) due to phonon instability.

## 4 Conclusion

We have investigated the ideal strength and critical strain of monolayer 8-*Pmmn* borophene by first-principles calculations. The monolayer 8-*Pmmn* borophene can withstand up to 21.59 N m<sup>-1</sup> and 27.79 N m<sup>-1</sup> stresses in  $a$  and  $b$  directions, respectively. The corresponding critical strains are 15.5% ( $a$  direction) and 16.5% ( $b$  direction). For biaxial tension, it also can withstand up to 21.89 N m<sup>-1</sup> and the corresponding critical strain is 13%. The calculated Young's moduli are 241.08 N m<sup>-1</sup> ( $a$  direction) and 305.20 N m<sup>-1</sup> ( $b$  direction), respectively, revealing that 8-*Pmmn* borophene is a hard but also flexible 2D material compared with other common 2D materials. It has been found that the tensile strains have a significant effect on the electronic structures of 8-*Pmmn* borophene. Furthermore, phonon dispersions under various applied strains imply that the mechanical failure in 8-*Pmmn* borophene is likely to originate from the elastic instability. Our calculation indicates that 8-*Pmmn* borophene may find its applications in high-strain engineering electronics area.

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