1. Introduction

Thermoelectric materials have received global attention because of their potential application in heat-electricity energy conversion, power generation and cooling thermoelectric devices. Many specialized applications have been realized in the thermoelectric devices due to high reliability, good scalability and simple device structure. The efficiency of thermoelectric devices can be determined by a dimensionless figure of merit $ZT = TS^2/σκ$, where $T$, $S$, $κ$ and $σ$ are the absolute temperature, the Seebeck coefficient, electrical conductivity and thermal conductivity (as the sum of the lattice and electronic contributions), respectively. The product of $S^2/σκ$ is the thermoelectric power factor (PF) which should be maximized and the thermal conductivity $κ$ should be minimized to achieve the expected high thermoelectric performance. The typical ways to optimize $ZT$ are alloying, pressure & strain engineering and low dimensioning. Besides, the investigations of novel thermoelectric materials with intrinsic high $ZT$ value are still anticipated.

Since 1950, the $\text{Bi}_2\text{Te}_3$-based chalcogenides have been developed and proposed as the best materials for room-temperature thermoelectric applications. The $z\text{-MgAgSb}$-based materials with the high $ZT \sim 1$ near room-temperature has been recently reported to be the candidates of the next generation thermoelectric energy production materials. Unfortunately, the high-cost and scarcity of Te and Ag elements hinder the commercial applications of these thermoelectric materials in large scale. Recently, the semiconducting alkaline-earth metals (AEMs) silicides have gained great attention for their potential applications in the thermoelectric power generator utilizing waste heat sources. $\text{Ca}_2\text{Si}$ is one of the environmental friendly and inexpensive AEMs, which can be synthesized by the heat treatment from the Mg$_2$Si powders under Ca vapor. It is interesting to see that the Seebeck coefficient of $\sim 300 \mu\text{V K}^{-1}$ with p-type conduction nature of $\text{Ca}_2\text{Si}$ was experimentally measured. However, in ambient condition $\text{Ca}_2\text{Si}$ is stabilized in the orthorhombic phase with a space group of $\text{Pnma}$. Considering the fact that the high symmetry cubic phase present larger number of equivalent degenerated valleys of the electronic structure than the low symmetry orthorhombic phase, better performance of the thermoelectric properties of cubic $\text{Ca}_2\text{Si}$ is anticipated. And the low symmetry of the orthorhombic phase leads to the anisotropic mechanical and optical properties which is not desired associated with the Cannikin law. From the self-consistent scalar-relativistic full potential linearized augmented plane wave (FPLAPW) calculation, cubic $\text{Ca}_2\text{Si}$ shows smaller bandgap value than the orthorhombic phase. The authors also assumed that suitable cubic substrate like diamond can help the growth of the cubic $\text{Ca}_2\text{Si}$. It is worth noting that as the silicon source and crystal template, Mg$_2$Si is crystallized in the cubic phase with the space group of $\text{Fm\overline{3}m}$, which is isotropic. But till now, the cubic $\text{Ca}_2\text{Si}$ is not experimentally observed. Hence theoretically, a more complete picture is required to elucidate the existence condition of cubic $\text{Ca}_2\text{Si}$. Moreover, by combining with Boltzmann transport theory and the phonon Boltzmann transport equation, we have predicted that the figure of merit $ZT$ for the cubic $\text{Fm\overline{3}m}$ $\text{Ca}_2\text{Si}$ reaches the maximum value of 0.52 by p-type doping. Our results provide an interesting insight and feasible guidelines for the potential applications of cubic $\text{Fm\overline{3}m}$ $\text{Ca}_2\text{Si}$ and related alkaline-earth metals silicides as the thermoelectric materials for heat-electricity energy convertors.
Applying external pressure is an effective way to stabilize the metastable or unstable phases of materials.\textsuperscript{29-31} In this work, by means of the \emph{ab initio} evolutionary algorithm calculations, we have revealed the stable conditions for the Ca\textsubscript{2}Si isomers under a wide (including positive and negative) pressure range. We found that the cubic phase of Ca\textsubscript{2}Si is a metastable phase under negative pressure and can be stabilized to the ambient condition, such negative-pressure condition could be probably achieved by increasing the synthetic temperature. Based on further density functional theory calculations combined with Boltzmann transport theory, we have systematically analyzed the electronic structure, lattice dynamical properties and the thermoelectric properties for cubic Ca\textsubscript{2}Si.

2. Computational details

We studied Ca\textsubscript{2}Si based on density functional theory (DFT) using the Vienna \emph{ab initio} simulation package\textsuperscript{32} (VASP) in conjunction with projector augmented wave (PAW) pseudopotentials within the generalized gradient approximations\textsuperscript{33} (GGA) of Perdew–Burke–Ernzerhof\textsuperscript{34} (PBE). Structure searches for the Ca\textsubscript{2}Si isomers under positive and negative pressures were carried out through the \emph{ab initio} evolutionary algorithm (EA) using the USPEX code\textsuperscript{35,36} The valence electron configuration for Ca and Si were 3s\textsuperscript{2}3p\textsuperscript{6}4s\textsuperscript{2} and 3s\textsuperscript{2}3p\textsuperscript{2} respectively. The cutoff energy for the plane-wave was set to 600 eV. The first Brillouin zone (1BZ) was sampled with a grid of 11 × 11 × 11 for the cubic phase and equivalent dense for the other phases in the structure optimization and energy calculation processes. The YPHON\textsuperscript{37} code was applied to obtain the phonon frequencies through the density functional perturbation theory (DFPT) methods\textsuperscript{38} with 2 × 2 × 2 supercells and 3 × 3 × 3 \emph{K}-points sampling. The thermodynamic properties were gained by using the quasi-harmonic approximation (QHA) method.\textsuperscript{39} The relaxation convergence for both ions and electrons were 1 × 10\textsuperscript{-6} eV. The crystal structures were visualized using the VESTA\textsuperscript{40} tool. It is well known that PBE calculations normally underestimate the electronic band gap, and the hybrid function with the mixing of the Hartree–Fock and DFT exchange terms is thought to be a practical solution to solve the band gap problem.\textsuperscript{41,42} Herein, we introduced the Heyd–Scuseria–Ernzerhof (HSE06) hybrid functional\textsuperscript{43} to precisely evaluate the bandgap of Ca\textsubscript{2}Si. The thermoelectric properties were computed from the semi-classical Boltzmann transport theory within the constant relaxation time approximation (\(\tau = 10^{-15}\) s, which is a normal magnitude for relaxation time in semiconductors\textsuperscript{44}) and a rigid band approach using the BoltzTraP code.\textsuperscript{45} For the prediction of the thermoelectric properties, denser \emph{K}-point meshes (twice as large as for the geometry optimization) were adopted for Brillouin zone integration to ensure the convergence and accuracy of eigenvalues. The second harmonic force constants from the Phonopy\textsuperscript{46} package and the third order anharmonic interatomic force constants created by the ShengBTE\textsuperscript{47} code with 4 × 4 × 4 supercells and gamma \emph{K}-point sampling were introduced to evaluate the lattice thermal conductivity.
3. Results and discussions

We have found four possible structures for Ca$_2$Si, which belong to the space groups of Pnma (Fig. 1(a)), Fm3m (Fig. 1(b)), R3m (Fig. 1(c)) and Pn3m (Fig. 1(d)), respectively. The equilibrium lattice parameters and atomic Wycko positions for the predicted phases at the ambient condition are listed in Tables 1 and 2. Herein the Pnma phase refers to the orthorhombic phase which has been experimentally synthesized. At the ambient condition, our GGA-PBE relaxed lattice parameters for orthorhombic Pnma Ca$_2$Si are $a = 7.605$, $b = 4.821$, $c = 9.038$, agree well with the experimental results within less than 1% mismatch. The Fm3m and Pn3m are two cubic phases. It is obvious that the Fm3m phase is more close-packing than the Pnma phase, indicating that the Fm3m phase is more stable. In fact, the Fm3m phase shows the same crystal structure as stable Mg$_2$Si. Our calculated lattice constant $a$ is 7.165 Å for the cubic Fm3m Ca$_2$Si at the ambient condition, which is in good agreement with the previous theoretical prediction. Fig. 1(e) illustrates the enthalpies vs. pressure curves of Ca$_2$Si in different space groups. As we can see from the figure that Pnma and Fm3m phases hold very close enthalpy, which locate at obviously lower level than the R3m and Pn3m cases. Hence in the full pressure range, the Pnma and Fm3m phases are favorable. Moreover, one can see that the Pnma phase is the most stable structure at the ambient condition and the full positive pressure range. The Fm3m phase become the most stable one only under a negative external pressure of $-0.82$ GPa. To achieve such negative pressure condition, we can experimentally enhance the synthesis temperature. As seen in Fig. 1(f), the equilibrium volume of cubic Fm3m Ca$_2$Si gradually increases with the increase of the temperature. The volume of cubic Fm3m Ca$_2$Si at $-0.82$ GPa (the transition pressure) is 94.22 Å$^3$, corresponding to the environmental temperature around 480 K. Hence we assume that the synthetic temperature higher than 480 K is necessary to achieve the cubic Fm3m Ca$_2$Si. The insert figures in Fig. 1(e) show the pressure dependence of the lattice constants.

### Table 1 The calculated cohesive energy and equilibrium lattice parameters for the predicted phases of Ca$_2$Si at the ambient condition

<table>
<thead>
<tr>
<th>Phase</th>
<th>$E_0$ (eV per u.c.)</th>
<th>$a_0$ (Å)</th>
<th>$b_0$ (Å)</th>
<th>$c_0$ (Å)</th>
<th>$\alpha (\degree)$</th>
<th>$\beta (\degree)$</th>
<th>$\gamma (\degree)$</th>
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<tbody>
<tr>
<td>Pnma</td>
<td>$-10.667$</td>
<td>7.605</td>
<td>4.821</td>
<td>9.038</td>
<td>90</td>
<td>90</td>
<td>90</td>
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<tr>
<td>Pnma$^*$</td>
<td>$-10.773$</td>
<td>7.618</td>
<td>4.793</td>
<td>9.001</td>
<td>90</td>
<td>90</td>
<td>90</td>
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<tr>
<td>Pnma$^{17}$</td>
<td>Expt.</td>
<td>7.691</td>
<td>4.816</td>
<td>9.035</td>
<td>90</td>
<td>90</td>
<td>90</td>
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<tr>
<td>Fm3m</td>
<td>$-10.623$</td>
<td>7.165</td>
<td>7.165</td>
<td>7.165</td>
<td>90</td>
<td>90</td>
<td>90</td>
</tr>
<tr>
<td>Fm3m$^{25}$</td>
<td>$-10.740$</td>
<td>7.148</td>
<td>7.148</td>
<td>7.148</td>
<td>90</td>
<td>90</td>
<td>90</td>
</tr>
<tr>
<td>R3m</td>
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<td>3.819</td>
<td>3.819</td>
<td>22.748</td>
<td>90</td>
<td>90</td>
<td>120</td>
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<tr>
<td>Pn3m</td>
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<td>6.435</td>
<td>6.435</td>
<td>90</td>
<td>90</td>
<td>90</td>
</tr>
</tbody>
</table>

### Table 2 The atomic Wyckoff positions for the predicted phases of Ca$_2$Si at the ambient condition

<table>
<thead>
<tr>
<th>Phase</th>
<th>Atom</th>
<th>Site</th>
<th>x</th>
<th>y</th>
<th>z</th>
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</thead>
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<tr>
<td>Pnma</td>
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<td>0.3459</td>
<td>0.2500</td>
<td>0.0742</td>
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<tr>
<td></td>
<td>Ca2</td>
<td>4c</td>
<td>0.9802</td>
<td>0.2500</td>
<td>0.8221</td>
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<tr>
<td></td>
<td>Si1</td>
<td>4a</td>
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<td>0.2500</td>
<td>0.1052</td>
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<tr>
<td>Fm3m</td>
<td>Ca1</td>
<td>6c</td>
<td>0.2500</td>
<td>0.2500</td>
<td>0.2500</td>
</tr>
<tr>
<td></td>
<td>Ca2</td>
<td>6c</td>
<td>0.7500</td>
<td>0.7500</td>
<td>0.7500</td>
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<tr>
<td></td>
<td>Si1</td>
<td>4a</td>
<td>0.0000</td>
<td>0.0000</td>
<td>0.0000</td>
</tr>
<tr>
<td>R3m</td>
<td>Ca1</td>
<td>3a</td>
<td>0.0000</td>
<td>0.0000</td>
<td>0.2500</td>
</tr>
<tr>
<td></td>
<td>Ca2</td>
<td>3a</td>
<td>0.0000</td>
<td>0.0000</td>
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<tr>
<td></td>
<td>Si1</td>
<td>3a</td>
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<tr>
<td>Pn3m</td>
<td>Ca1</td>
<td>4b</td>
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<td>0.7500</td>
<td>0.2500</td>
</tr>
<tr>
<td></td>
<td>Ca2</td>
<td>4b</td>
<td>0.2500</td>
<td>0.2500</td>
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</tr>
<tr>
<td></td>
<td>Si1</td>
<td>2a</td>
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Fig. 2 The calculated phonon dispersion curves of cubic Fm3m Ca$_2$Si (a–c) without and (e–g) with LO–TO splitting, where (a) and (e) at $-0.82$ GPa external pressure, (b) and (f) at the ambient condition, (c) and (g) at 7.8 GPa external pressure.
of Pnma and Fm3m Ca2Si, we found that the lattice constants of both the phases decrease gradually with the increase of the external pressure. Considering the fact that the enthalpy difference between the Pnma and Fm3m phases is as small as 44 meV per u.c. (unit cell), it is natural to expect that the cubic Fm3m phase might be stabilized at the ambient condition as well as under finite positive external pressure.

To investigate the stability of cubic Fm3m Ca2Si under different pressures, we calculated the phonon dispersion curves both without and with LO–TO splitting in the first Brillouin zone (1BZ), which are illustrated in Fig. 2. As we can see, no negative or imaginary frequency was found in cubic Fm3m Ca2Si for all the cases. The results suggest that cubic Fm3m Ca2Si shows very good lattice dynamical stability at the pressure range we have studied. Hence the Fm3m structure is a metastable phase of Ca2Si under pressure higher than −0.82 GPa. With the increase of pressure, the Ca–Si bonds were compressed, resulting in the increase of the maximum dispersion frequencies of the optical modes. Without LO–TO splitting, the calculated maximum frequencies of the twofold degenerated optical modes at the Æ point are 6.2 THz, 6.5 THz and 8.1 THz under −0.82 GPa negative pressure, at the ambient condition and under a positive pressure of 7.8 GPa, respectively. The LO–TO splitting is accounted for by collecting the Born effective charge tensor and dielectric constant tensor. Under the influence of LO–TO splitting, we found that the twofold degenerated optical modes at the Æ point split into a higher frequency discontinuous LO branch and a lower frequency TO branch. The TO branch holds the frequency of the optical branches without LO–TO splitting. The discontinuity of the LO branch is due to the difference in the magnitude of LO–TO splitting along the L–Æ and Æ–X directions. It is well known that the soft mode of the acoustic branch is the critical feature of a material with low lattice thermal conductivity, which is expected to achieve high ZT. We have found that cubic Fm3m Ca2Si represents very similar soft mode characteristics to cubic Fm3m Mg2Si. It is worth noting that the low frequency optical branches overlap with the acoustic branches along the X–W–K path are softer than Mg2Si. Hence lower lattice thermal conductivity in cubic Fm3m Ca2Si is anticipated. On the other hand, similar to Mg2Si, the increase of external pressure leads to the frequency enhancement of the soft modes, which is not expected for thermoelectric applications.

In additions, to further confirm the stability of cubic Fm3m Ca2Si, we predicted the mechanical stability via the elastic constants criterion. For a cubic crystal, there are three independent elastic constants C11, C12 and C44. The mechanical stability can be judged by C11 > 0, C44 > 0, C11 > |C12| and (C11 + 2C12) > 0. Fig. 3(a) shows the calculated pressure dependence of elastic constants by a step by step stress-strain method for cubic Fm3m Ca2Si. It is obviously that the elastic constants satisfy all the stability conditions shown above, indicating that cubic Fm3m Ca2Si is mechanical stable in the pressure range from −2 to 8 GPa. The positive correlation between the elastic constants and external pressure can be found, which is due to the enhanced interaction between atoms under external pressure. The discontinuity of the LO branch is due to the decrease of the high temperature bipolar conduction modulus E, and shear modulus G from the elastic constants as a function of pressure. As we can see, with the increase of pressure, the mechanical properties of B, E, G increase gradually, which are similar to the trend of elastic constants. Moreover, according to the calculated Poisson’s ratio presented in Fig. 3(c), we found that the Poisson’s ratio is very small, although it is increasing with the growth of pressure, indicating cubic Fm3m Ca2Si is relatively stable against the shear strain.

In order to get better understanding, we introduced both the standard GGA-PBE functional as well as the HSE06 hybrid functional to calculate the electronic structures. Fig. 4(a) illustrates the total density of states (DOS) for cubic Fm3m Ca2Si. As we can see from the figure, we have obtained 0.59 eV band gap using GGA-PBE functional, agrees well with the previous FPLOPW values of 0.56 eV. It is noted that HSE06 functional enlarges the bandgap to 1.13 eV. The band gap value is obviously larger than the narrow gap Bi2Te3-based alloys, which is beneficial for the high temperature thermoelectric applications due to the decrease of the high temperature bipolar conduction
effect.\textsuperscript{25} By the analysis of the bandgap at various conditions in Fig. 4(b), we found that both GGA-PBE and HSE06 bandgaps reduce linearly with the increase of the pressure. The fitting slope for the GGA-PBE and HSE06 results are $-0.0326$ and $-0.0283$ eV GPa$^{-1}$, respectively. We have found that GGA-PBE and HSE06 methods present very similar band gap changing tendency of cubic $\text{Fm\textsuperscript{3}m Ca\textsubscript{2}Si}$ under pressure, where the HSE06 gaps are about 0.6 eV larger than the PBE gaps. According to the projected band structure at the ambient condition in Fig. 4(c) and (e), we found that cubic $\text{Fm\textsuperscript{3}m Ca\textsubscript{2}Si}$ has been theoretically predicted to be a direct band gap semiconductor using PBE, where both the valance band maximum (VBM) conduction band minimum (CBM) are located at the $X$ point. For both the valence and conduction states are not sensitive to the external, however, with the increasing of the pressure up to 7.8 GPa, the effective masses of the electron along the $X$–$\Gamma$ direction using PBE $m^*_{\text{e},\text{PBE}} = 0.18 m_e$ and using HSE06 $m^*_{\text{e},\text{HSE06}} = 0.21 m_e$. The effective masses of the hole along the $X$–$\Gamma$ direction using PBE $m^*_{\text{h},\text{PBE}} = 1.15 m_e$ and using HSE06 $m^*_{\text{h},\text{HSE06}} = 2.05 m_e$. The outstanding effective masses of the hole indicating the potential high thermoelectric properties with $p$ type doping of cubic $\text{Fm\textsuperscript{3}m Ca\textsubscript{2}Si}$. According to the projected band structure under 7.8 GPa external pressure in Fig. 4(d) and (f), we found that the effects of the external pressure pull up the energy level of VBM and push down the energy level of CBM at the $X$ point for both PBE and HSE06. As a result, the $X$ to $X$ direct gap nature of cubic $\text{Fm\textsuperscript{3}m Ca\textsubscript{2}Si}$ can be protected. The Ca-s dominated conduction band state at the $\Gamma$ point and the Si-s dominated conduction band state at the $L$ point move upwards to higher energy levels under certain pressure. Besides, the other valence and conduction states are not sensitive to the external pressure. Herein, the stability of the electronic structure can protect the structure stability of cubic $\text{Fm\textsuperscript{3}m Ca\textsubscript{2}Si}$. Although the effective masses of the electron are insensitive to the external, however, with the increasing of the pressure up to 7.8 GPa, the effective masses of the hole along the $X$–$\Gamma$ direction reduce to $m^*_{\text{h, PBE}} = 0.66 m_e$ and $m^*_{\text{h, HSE06}} = 0.91 m_e$. The decreasing of the hole effective mass indicating the increasing of the external pressure cannot improve the thermoelectric properties of cubic $\text{Fm\textsuperscript{3}m Ca\textsubscript{2}Si}$. Hence combining with the lattice dynamic analysis, our further thermoelectric property study is focused on cubic $\text{Fm\textsuperscript{3}m Ca\textsubscript{2}Si}$ at the ambient conduction.
By solving the Boltzmann transport equation based on the HSE06 electronic eigenvalues, we first analyzed the thermoelectric power factor (PF) $S^2\sigma$ to examine the thermoelectric properties of cubic $Fm\bar{3}m$ $Ca_2Si$. The Seebeck coefficients $S$ as a functional of the chemical potential at different temperature are illustrated in Fig. 5(a). We found that the Seebeck coefficient is very sensitive to the chemical potential. Hence the efficient thermoelectric properties can be tuned via the carrier concentration. Generally, we found that the Seebeck coefficient of cubic $Fm\bar{3}m$ $Ca_2Si$ decreases with the increase of temperature. Nevertheless, it is higher than the experimental measured peak value 300 $\mu$V K$^{-1}$ for the p-type doped orthorhombic $Ca_2Si$ in most of the chemical potential and temperature range, which indicates that cubic $Fm\bar{3}m$ $Ca_2Si$ is expected to show better thermoelectric performances than the orthorhombic phase. Moreover, we found that the p-type doping is preferred for the thermoelectric applications of cubic $Fm\bar{3}m$ $Ca_2Si$ with larger optimized thermoelectric PF $S^2\sigma$.

In the denominator of the figure of merit $ZT$, the thermal conductivity can be divided to the electronic ($k_e$) and lattice ($k_l$) contribution terms. The electronic thermal conductivity in Fig. 6(a) can also be solved from the well-known Wiedemann–Franz law $\kappa_e = \frac{\pi^2}{3} \left(\frac{k_B}{\sigma}\right) T$. Hence as a function of the chemical potential, the electronic thermal conductivity curves present very similar trend to the electrical conductivity. Meanwhile, in the whole chemical potential range, positive correlation can be found between the electronic thermal conductivity and temperature. The lattice thermal conductivity in Fig. 6(b) is estimated by the phonon Boltzmann transport equation.

We found that the lattice thermal conductivity of cubic $Fm\bar{3}m$ $Ca_2Si$ close to 0, indicating the finite carrier concentration is needed for the optimized thermoelectric PF. As seen from the calculated thermoelectric PF in Fig. 5(c), the thermoelectric PF reaches a peak around $-0.5$ to $-0.6$ eV for the p-type doping and 0.4–0.6 eV for the n-type doping at different temperatures. We found that the thermoelectric PF increases with the increase of temperature, which is contrary to the Seebeck coefficient. It is because the electrical conductivity is gradually increased when the temperature is increasing at the chemical potential larger than $-0.64$ eV. Moreover, we found that the p-type doping is preferred for the thermoelectric applications of cubic $Fm\bar{3}m$ $Ca_2Si$ with larger optimized thermoelectric PF $S^2\sigma$. 
show the same order of magnitude to the \textit{Pnma} \textit{Ca}_2\textit{Si} and \textit{Bi}_2\textit{Te}_3-based chalcogenides around serval W m\(^{-1}\) K\(^{-1}\), which is lower than \textit{Mg}_2\textit{Si} as we have anticipated. Considering the large differences between the covalent radii of Ca (1.76 Å) and Si (1.11 Å)\(^7\) as well as the soft mode nature of the phonon dispersion curves, the low lattice thermal conductivities can be understood. As seen in the insert figure of Fig. 6(b), the lattice thermal conductivity can be linearly fitted to the reciprocal of temperature, indicating the fact that the phonon scattering is dominated by the anharmonic phonon–phonon interactions. To the better understanding of the low thermal conductivity, we analyze the Grüneisen parameter to quantify the anharmonicity. The average Grüneisen parameter is 1.97 for cubic \textit{Fm\textit{3m} Ca}_2\textit{Si}, which is comparable with typical thermoelectric materials \textit{PbTe} of 1.96 with low thermal conductivity.\(^5\)

After all, we can evaluate the thermoelectric figure of merit \(ZT\) based on the above results. Fig. 7(a) illustrates the calculated \(ZT\) as a function of the chemical potential. We found that the plot of \(ZT\) shows very similar feature to the plot of PF, and it is interesting that \(ZT\) increases with the increase of temperature up to 1000 K. The peak values of \(ZT\) at different temperatures are plotted in Fig. 7(b). The maximal \(ZT\) value of 0.07 and 0.52 are observed with the chemical potential of \(-0.628\) eV and \(-0.533\) eV at 300 K and 1000 K, corresponding to the hole concentration of \(2.45 \times 10^{18}\) cm\(^{-3}\) and \(2.83 \times 10^{18}\) cm\(^{-3}\), respectively. The \(ZT\) value of the p-type doped \textit{Pnma} \textit{Ca}_2\textit{Si} is \(2 \times 10^{-5}\) around 350 K (ref. 21) (marked as the violet star in Fig. 7(b)), which is much smaller than our predicted values for the cubic \textit{Fm\textit{3m} Ca}_2\textit{Si}. As seen the \(ZT\) of intrinsic \textit{Mg}_2\textit{Si} in Fig. 7(b), we found that the thermoelectric properties of cubic \textit{Fm\textit{3m} Ca}_2\textit{Si} are comparable to intrinsic \textit{Mg}_2\textit{Si}.\(^9,6\) Since the solid solution treatment and doping can effectively enhance the figure of merit \(ZT\) of \textit{Mg}_2\textit{Si} to as large as 1.1 (ref. 61), further optimizing of the thermoelectric properties of the cubic \textit{Fm\textit{3m} Ca}_2\textit{Si} is anticipated.

4. Conclusion

In summary, based on the \textit{ab initio} evolutionary algorithm structure searching and density functional theory calculations, we have found that the orthorhombic \textit{Pnma} phase is the most stable structure at pressure higher than \(-0.82\) GPa, while the cubic \textit{Fm\textit{3m} Ca}_2\textit{Si} phase is a meta-stable one in certain conditions. The stability of the cubic \textit{Fm\textit{3m} Ca}_2\textit{Si} has been confirmed by the lattice dynamical phonon dispersion calculations and the mechanic criterion studies. The direct bandgap nature of cubic \textit{Fm\textit{3m} Ca}_2\textit{Si} at various conditions was explored. Combining the HSE06 hybrid functional calculations, Boltzmann transport equation and the phonon Boltzmann transport equation, we have theoretically predicted the thermoelectric properties of cubic \textit{Fm\textit{3m} Ca}_2\textit{Si}, which present larger Seebeck coefficient than orthorhombic \textit{Ca}_2\textit{Si}. The optimized thermoelectric figure of merit \(ZT\) value of 0.52 is achieved for the p-type doping with the hole concentration \(2.83 \times 10^{18}\) cm\(^{-3}\) at 1000 K. We expect our findings will facilitate the experimental synthesis of \textit{Ca}_2\textit{Si} isomers as well as their practical applications in thermoelectric devices.

Acknowledgements

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