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Two environmentally and thermally stable [1]benzothieno[3,2-*b*][1]benzothiophene (BTBT) derivatives, BOP-BTBT and DBOP-BTBT are successfully synthesized and analyzed as active layers in organic thin film transistors. The effects of methoxy on BTBT based OTFT materials are reported for the first time. The experimental results show an excellent optimization influence of methoxy group on the OTFT performance with an improved hole transport mobility up to $0.63\text{ cm}^2\text{ V}^{-1}\text{ s}^{-1}$ (BOP-BTBT) and $3.57\text{ cm}^2\text{ V}^{-1}\text{ s}^{-1}$ (DBOP-BTBT). Meantime the mono- and bis-substituted derivatives are compared in terms of their physical properties and device performance. We find that the threshold voltage decreases when more methoxy groups are introduced.

Introduction

Organic thin film transistors (OTFTs) have attracted much attention due to their potential applications in optoelectronic devices including sensors, electronic paper, transparent circuits and low-cost and flexible displays.¹ In recent decades, enormous progress has been made in organic semiconductors (OSCs) and different kinds of OSCs with mobility exceeding $1\text{ cm}^2\text{ V}^{-1}\text{ s}^{-1}$, higher than that of amorphous silicon, have been developed.² Among the most efficient OSCs, [1]benzothieno[3,2-*b*][1]benzothiophene (BTBT) derivatives show excellent OTFT performance because of the strong intermolecular interactions between sulfur atoms and large electron cloud overlap between OSC molecules. In 2006, BTBT core was used in organic semiconductors for the first time and diphenyl-substituted BTBT (DPh-BTBT) was synthesized. The highest charge mobility of DPh-BTBT achieved was $2.0\text{ cm}^2\text{ V}^{-1}\text{ s}^{-1}$ which was a dramatically high value at that time.³ In 2007, dialkyl-substituted BTBT

was reported with good solubility in normal solvents. When the compound, $\text{R}=\text{C}_{13}\text{H}_{25}$ showed excellent performance in the solution processed OTFT devices with $\mu = 2.75\text{ cm}^2\text{ V}^{-1}\text{ s}^{-1}$ and $I_{\text{on}}/I_{\text{off}} > 10^7$.⁴ Also in 2007, phenyl ring was fused to BTBT core and formed another star structure DNTT.⁵ Four years later, naphthyl-fused BTBT was reported called DATT⁶ and in 2012, mono functionalized BTBT using alkyl chain of $\text{C}_{13}\text{H}_{25}$ was studied and high hole mobility over $17.2\text{ cm}^2\text{ V}^{-1}\text{ s}^{-1}$ was attained by using $\text{AlO}_x/\text{C}_{14}\text{-PA}$ as dielectric layer.⁷ In 2015, 2,7-di-*tert*-butyl-BTBT was reported as isotropic OTFT material with $\mu = 17\text{ cm}^2\text{ V}^{-1}\text{ s}^{-1}$ in single crystal FETs.⁸ Up to now, almost all BTBT derivatives are modified using alkyl chains and aromatic rings, however, the effect of hetero atoms on the BTBT derivatives is seldom studied.

BTBT derivatives usually exhibit good air stability due to the low lying highest occupied molecular orbital (HOMO) energy levels. For example, the HOMO level of DPh-BTBT is -5.6 eV .⁹ Meanwhile, the large difference between the HOMO level of OSCs and the work function of normally used gold source electrode (-5.1 eV) results in an obvious contact resistance between the active layer and the electrode which limits the OTFT device performance, especially in the linear region.¹⁰ In order to explore the effects generated by heteroatoms on BTBT derivatives and reduce the contact resistance in relevant devices, electron-donating methoxy group is introduced into phenyl-substituted BTBT. In our previous work,¹¹ we introduced methoxy group into diphenyl-substituted anthracene to obtain high mobility ($2.96\text{ cm}^2\text{ V}^{-1}\text{ s}^{-1}$) semiconductor along with strong fluorescence and isotropic properties. The designed target material is also expected to have excellent isotropic properties.

Herein, we successfully synthesize two novel materials, 2-(4-methoxyphenyl)benzo[*b*]benzo[4,5]thieno[2,3-*d*]thiophene (BOP-BTBT) and 2,7-bis(4-methoxyphenyl)benzo[*b*]benzo[4,5]thieno[2,3-*d*]thiophene (DBOP-BTBT) by Suzuki coupling. Optical and electrical properties of BOP-BTBT and DBOP-BTBT are measured and OTFT devices based on these two materials are successfully fabricated and tested. All the devices show

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apparently improved FETs performance compared to the phenyl-substituted ones, with an increase in mobility from $0.02 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ (Ph-BTBT)¹² to $0.63 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ (BOP-BTBT) and from $2.0 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ (DPh-BTBT) to $3.57 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ (DBOP-BTBT). Meantime, we find that the threshold voltage decreases when more methoxy groups are introduced into the semiconductor structure.

Results and discussion

Synthesis

The aimed compound is synthesized through Suzuki–Miyaura coupling reaction¹³ using (4-methoxyphenyl)boronic acid and bromo substituted [1]benzothieno[3,2-*b*]benzothiophene (BTBT).¹⁴ The later one was obtained according to the reported procedure¹⁵ (details in ESI†). The coupling reaction mixture was purified through vacuum sublimation three times at 260°C for BOP-BTBT and 290°C for DBOP-BTBT, to get pure materials as white powder and pale-yellow powder respectively. Finally, the chemical structures were confirmed by NMR and MS (ESI†). While the DBOP-BTBT was too hard to dissolve, so we used elemental analysis and single crystal X-ray analysis to confirm the structure.

Thermal and photo physical properties

As shown in Fig. 1, BOP-BTBT and DBOP-BTBT exhibit good thermal stability with 5% weight loss at temperatures of 325°C and 415°C , respectively, as measured by thermogravimetric analysis (TGA). The bis-substituted derivative is more thermally stable than the mono-substituted derivative. This high decomposition temperature ensures potential OTFT availability under high temperature conditions.

Thin films of BOP-BTBT and DBOP-BTBT with a thickness of $\sim 50 \text{ nm}$ were vacuum-deposited on quartz substrates for measurement of UV-vis absorption spectra (Fig. 2). BOP-BTBT and DBOP-BTBT exhibit an absorption onset of 400 nm and 420 nm in solid state, respectively, which shows an apparent absorption red-shift compared to that in dichloromethane solution (360 nm and 382 nm). This indicates the strong intermolecular interactions in the thin films. The optical HOMO–LUMO gaps estimated from the absorption edge in thin films are 3.10 eV (BOP-BTBT) and 2.95 eV (DBOP-BTBT). The optical band gap of DBOP-BTBT is narrowed as compared to that of BOP-BTBT, which is consistent with the introduction of electron donating methoxy phenyl units. Based on the

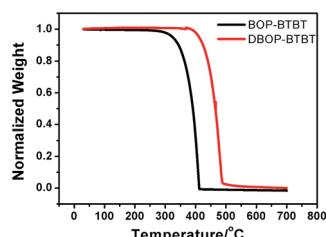


Fig. 1 TGA plots of BOP-BTBT and DBOP-BTBT in nitrogen with a heating rate of $10^\circ\text{C min}^{-1}$.

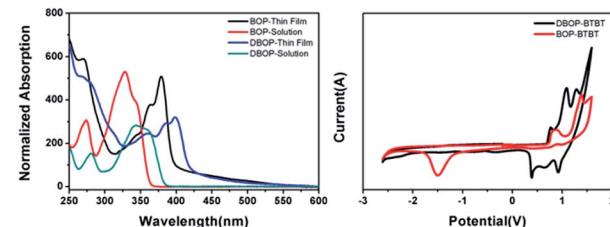


Fig. 2 The UV-vis absorption spectra and cyclic voltammogram of BOP-BTBT and DBOP-BTBT.

calibration of redox potentials of an internal ferrocene reference (F_c/F_c^+), the thermally evaporated thin films of the two materials on GCE electrode were subjected to cyclic voltammetry (CV) (Fig. 2). The cyclic voltammetry of DBOP-BTBT in acetonitrile shows onset of the first oxidation potential at $+0.69 \text{ V}$ vs. F_c/F_c^+ , corresponding to HOMO energy level of -5.49 eV ,¹⁶ which is 0.11 eV higher than that of the reported DPh-BTBT (HOMO = -5.6 eV). The results are summarized in Table 1.

Charge transport properties

To investigate and compare the field-effect properties of the two materials, BOP-BTBT and DBOP-BTBT, organic thin-film transistors with top contact/bottom gate configuration were fabricated by vacuum deposition. The heavily doped silicon wafers with oxide thickness of 300 nm were firstly cleaned with the acetone, deionized water and isopropanol in the ultrasonic bath for 15 minutes, respectively. These substrates were then blown dry with nitrogen gas and further cleaned in oxygen plasma for 15 min. They were then treated in 0.03 g mL^{-1} concentration of octadecyl trichlorosilane (OTS) solution in toluene at 60°C for 20 min to form self-assembled monolayer. The OTS-modified SiO_2/Si substrates were rinsed with fresh toluene in the sonic bath for 1 min and blown dry with nitrogen gas. The organic active layer was then evaporated onto these substrates to a thickness of approximately 50 nm under the pressure of $1 \times 10^{-4} \text{ Pa}$ for each substrate temperature condition. Gold source and drain electrodes ($W/L = 10$) were deposited on the organic active layer using a shadow mask. Electrical characteristics of the OTFT devices were measured by Agilent B1500A in the ambient air. The hole mobility (μ_h) was obtained from the saturation regime using the following equation:

Table 1 Electrochemical and optical properties of BOP-BTBT and DBOP-BTBT

| | E_{ox}^a (V) | E_{HOMO}^b (eV) | λ_t^c (nm) | λ_s^d (nm) | E_g^e (eV) |
|-----------|-----------------------|--------------------------|--------------------|--------------------|--------------|
| BOP-BTBT | $+0.73$ | -5.53 | 400 | 360 | 3.10 |
| DBOP-BTBT | $+0.69$ | -5.49 | 420 | 382 | 2.95 |

^a Estimated from electrochemical determination versus F_c/F_c^+ . ^b $E_{\text{HOMO}} = -(E_{\text{onset,ox}} \text{ vs. } \text{F}_c/\text{F}_c^+ + 5.1)$ (eV). ^c Estimated from thin film absorption spectra. ^d Solution absorption spectra in dichloromethane. ^e Energy band gap calculated from the absorption edge of the thin-film.



$$I_{ds} = C_i \mu_e (W/2L)(V_g - V_{th})^2$$

In the above equation, I_{ds} , W , L , V_g , V_{th} and C_i represent the drain-source current, channel width, channel length, gate voltage, threshold voltage and the capacitance per unit area of the gate dielectric layer, respectively.

We tried different substrate temperatures for BOP-BTBT (RT, 60 °C) and DBOP-BTBT (RT, 60 °C, 80 °C, 100 °C, 120 °C). Table 2 summarizes the performance of the thin film transistors fabricated at different substrate temperatures (T_{sub}).

The output curves show standard linear and saturation regions. While the feature of injection barriers is not observed in the low- V_D region. According to the table, DBOP-BTBT shows better overall OTFT performance than BOP-BTBT, especially at elevated substrate temperatures. When T_{sub} = RT, mobility of BOP-BTBT and DBOP-BTBT is almost the same around $0.6 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$, which is higher than Ph-BTBT ($0.012 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$)¹² and DPh-BTBT ($0.36 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$). With T_{sub} increasing, mobility of DBOP-BTBT simultaneously increases up to $3.57 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ (at T_{sub} = 100 °C) while mobility of BOP-BTBT drops by half to only 0.28 (at T_{sub} = 60 °C). The highest mobility obtained from Ph-BTBT¹² is only $0.016 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$, an order of magnitude lower than that of BOP-BTBT. The highest hole mobility for DBOP-BTBT is $3.57 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ with V_{th} = -25.3 V and the relative I_{on}/I_{off} = 2.13×10^7 . This mobility is almost two times higher than that reported for DPh-BTBT ($2.0 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$), which was obtained by addition of only two methoxy groups. Noticeably, the V_{th} of BOP-BTBT is much lower than that of Ph-BTBT (almost -50 V). Also, V_{th} of DBOP-BTBT (RT, -11.4 V) is lower than BOP-BTBT (RT, -32.6 V) (Fig. 3).

Morphological characterizations

The OTFT performance is greatly affected by the substrate temperature through influencing the grain size, as shown in the atomic force microscopy (AFM) images in Fig. 4. For DBOP-BTBT, the polycrystalline domain size in the thin film deposited at room temperature is only $0.1 \mu\text{m}$ while the grain size grows to $1\text{--}1.5 \mu\text{m}$ in the films deposited at $100 \text{ }^\circ\text{C}$. When DBOP-BTBT is deposited at $120 \text{ }^\circ\text{C}$, we get larger but discontinuous polycrystals compared to those at $100 \text{ }^\circ\text{C}$ with the emergence of deep cracks, which explains the decrease of transistor performance. However, for BOP-BTBT, the polycrystalline domain size is only $0.3 \mu\text{m}$ at T_{sub} = RT, but when the substrate is merely heated to $60 \text{ }^\circ\text{C}$ the grains grow and morphology of the thin

Table 2 Experimental characteristics of polycrystalline thin film device at different substrate temperatures. All the data are reported in average values from more than 10 devices

| | $T_{sub}/^\circ\text{C}$ | Mobility/ $\text{cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ | I_{on}/I_{off} | Average V_{th}/V |
|-----------|--------------------------|---|------------------|---------------------------|
| DBOP-BTBT | RT | 0.54–0.68 | $>10^6$ | –11.4 |
| | 60 | 0.66–0.72 | $>10^6$ | –20.8 |
| | 100 | 2.92–3.57 | $>10^7$ | –25.3 |
| | 120 | 1.81–2.15 | $>10^6$ | –32.2 |
| BOP-BTBT | RT | 0.51–0.63 | $>10^6$ | –32.6 |
| | 60 | 0.18–0.35 | $>10^6$ | –35.8 |

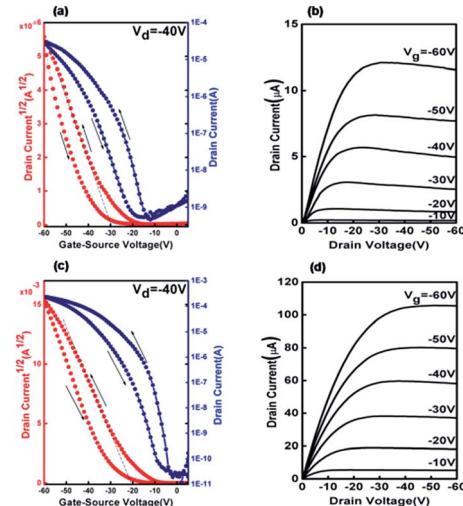


Fig. 3 Typical OTFT characteristics (transfer and output curves) for BOP-BTBT and DBOP-BTBT thin films grown on the OTS-modified SiO_2/Si substrates. (a) and (b) represent BOP-BTBT devices at T_{sub} = RT. (c) and (d) represent DBOP-BTBT devices at T_{sub} = $100 \text{ }^\circ\text{C}$.

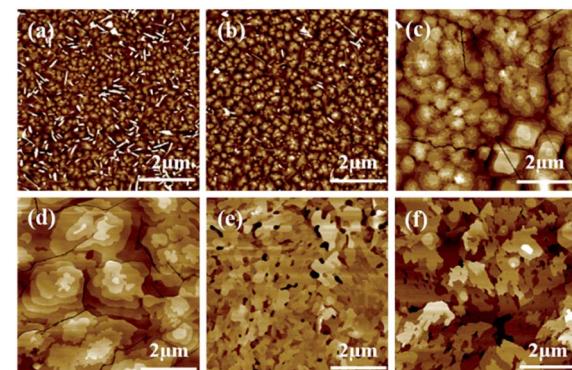


Fig. 4 AFM images ($5 \mu\text{m}$) of DBOP-BTBT (a–d) and BOP-BTBT (e and f) based polycrystalline thin-films on the OTS-modified SiO_2/Si substrate at various substrate temperatures. (a) RT; (b) $60 \text{ }^\circ\text{C}$; (c) $100 \text{ }^\circ\text{C}$; (d) $120 \text{ }^\circ\text{C}$; (e) RT; (f) $60 \text{ }^\circ\text{C}$.

films becomes very chaotic which is the cause of the obvious decay in the relative performance of the OTFT devices.

X-ray analysis

To investigate the molecular packing, single crystals of DBOP-BTBT (CCDC 1518254) with sufficient quality for X-ray structural analysis were grown by vacuum sublimation under 10^{-5} Pa . But for BOP-BTBT, sublimation could not form large enough crystals for X-ray structural analysis. Also, the method of slow evaporation of solvents (benzene, chlorobenzene, chloroform) tend to give out twin crystals which could not be used.

Single-crystal X-ray analysis of DBOP-BTBT demonstrate that the compound exhibits a typical herringbone packing structure with crystal parameters of: $a = 7.5863 \text{ \AA}$, $b = 6.0552 \text{ \AA}$, $c = 46.173 \text{ \AA}$ and $\alpha = \beta = \gamma = 90^\circ$. The intermolecular distance between two adjacent DBOP-BTBT molecules along the π -



stacking direction is only 6.06 Å (6.19 Å for DNTT). The torsion angle of phenyl rings out of BTBT core planes is measured to be 14.75° which indicates an almost planar structure of DBOP-BTBT. Two main kinds of sulfur–sulfur van der Waals interactions exist in *a*–*b* plane and the distance between sulfur atoms is 3.40 Å and 3.77 Å. Also, the methoxy group causes strong contacts between O and H with a distance of 2.582 Å between oxygen and hydrogen atoms. This flat molecular structure and multi strong intermolecular interactions lead to dense packing of DBOP-BTBT and higher mobility compared to that reported for DPh-BTBT (Fig. 5).

As shown in Fig. 6, the X-ray diffraction (XRD) measurements of the BOP-BTBT and DBOP-BTBT thin films on Si/SiO₂ substrates show a series of peaks assignable to (00*h*) reflections. For DBOP-BTBT, a primary diffraction peak appears at $2\theta = 3.80^\circ$, with fourth diffraction peak at $2\theta = 7.73^\circ$ and sixth diffraction peak at $2\theta = 11.50^\circ$. The *d*-spacing of DBOP-BTBT

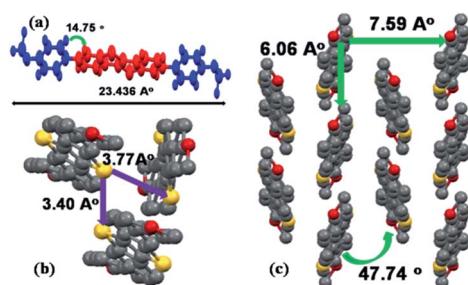


Fig. 5 Single crystal structure of DBOP-BTBT. (a) Molecular length and torsion angle. (b) Two types of S–S contacts. (c) Herringbone packing.

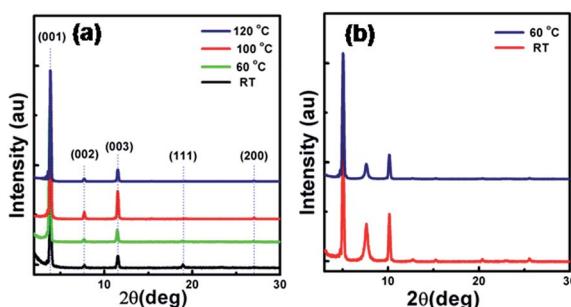


Fig. 6 XRD pattern of the thin films of (b) BOP-BTBT and (a) DBOP-BTBT as deposited at various substrate temperatures.

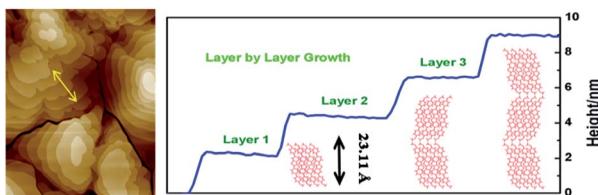


Fig. 7 The step heights of crystalline terrace layers as measured along the line marked in left and DBOP-BTBT layer-by-layer packing mode in the films.

obtained from the first reflection peak is 23.22 Å, which is very close to the length of DBOP-BTBT molecule (23.436 Å), indicating that DBOP-BTBT molecules are grown perpendicular to the substrate. This inference is further confirmed by the step height of the 2 µm thin film terraces which is 23.11 Å, as shown in Fig. 7. For BOP-BTBT, primary diffraction peak appears at $2\theta = 2.53^\circ$, indicating that the *d*-spacing of BOP-BTBT is 34.91 Å, which is much larger than that of DBOP-BTBT. Compared to thin films deposited at room temperature, thin films at 60 °C display stronger diffraction peaks. This reveals that the crystal quality of thin films drops with the increase of substrate temperature.

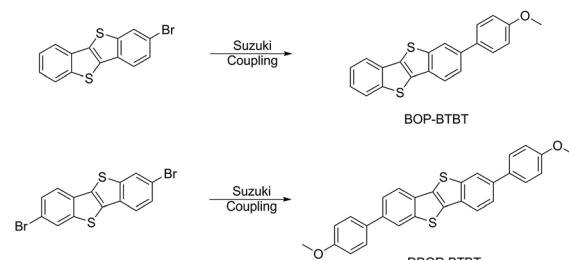
Conclusions

Two environmentally and thermally stable BTBT derivatives BOP-BTBT and DBOP-BTBT are designed and synthesized. This is the first time that methoxy group is introduced into BTBT core. All of the devices show apparently improved FETs performance compared to those based on the phenyl-substituted molecules. An improvement in the mobility from $0.02 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ (Ph-BTBT) to $0.63 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ (BOP-BTBT) and from $2.0 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ (DPh-BTBT) to $3.57 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ (DBOP-BTBT) has been demonstrated in the fabricated OTFTs. Meantime, through the comparison of derivatives containing different numbers of methoxy group, we find that the threshold voltage decreases with more methoxy groups. Photophysical, morphological and single crystal characterizations explain that the methoxy groups improve the charge transport performance by affecting the HOMO level and intermolecular interactions. The method of introducing methoxy groups into organic semiconductor structures could be a promising molecular design strategy.

Experimental section

General

All chemicals and solvents were analytical reagents unless otherwise noted. All of the reactions were performed under a nitrogen atmosphere in dry solvent. The synthetic route of BOP-BTBT and DBOP-BTBT is shown Scheme 1. Thermogravimetric analysis (TGA) was carried out on a TA Instruments TA2950 TGA system at a heating rate of $10 \text{ }^\circ\text{C min}^{-1}$ and a nitrogen flow rate of $60 \text{ cm}^3 \text{ min}^{-1}$. Differential scanning calorimetry (DSC) was run on a TA Instrument DSC Q1000 at a heating or cooling rate of $5 \text{ }^\circ\text{C min}^{-1}$ under a nitrogen flow.



Scheme 1 Synthetic route to BOP-BTBT and DBOP-BTBT.

XRD patterns were recorded using a Bruker D8 advance X-ray diffractometer with a Cu K α source ($\lambda = 1.541$ Å). Film morphology was studied with atomic force microscopy (AFM) on a SPA400HV instrument with a SPI 3800 controller (Seiko Instruments).

BOP-BTBT. To a 100 mL flask, 1.6 g (5 mmol) 2-bromo-BTBT, 1.28 g (10 mmol) 4-methoxyphenyl boronic acid and Pd(PPh₃)₄ 200 mg was added under nitrogen. Then 15 mL ethanol, 40 mL toluene and 10 mL 2 M K₂CO₃ aqueous solution was added. The whole system was heated to 120 °C and kept overnight. Then the whole system was cooled down and 50 mL methanol was added. After filtration, the filtrate was washed with water and methanol successively, then dried under vacuum and further purified by sublimation. BOP-BTBT was obtained as a white solid with a yield of 65% (1.12 g). ¹H NMR (400 MHz, CDCl₃) δ 8.10 (d, 1H), 7.99–7.87 (m, 3H), 7.72–7.66 (m, 3H), 7.52–7.39 (m, 2H), 7.05 (d, 2H), 3.90 (s, 3H).

DBOP-BTBT. This was achieved in the same manner as the synthesis of BOP-BTBT. DBOP-BTBT was obtained as a yellow solid with a yield of 52% (1.4 g). Anal. calcd for C₂₈H₂₀O₂S₂: C, 74.31; H, 4.45; O, 7.07; S, 14.17. Found: C, 74.22; H, 4.47; S, 14.31.

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