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Interface engineering of an AlNO/AlGaN/GaN MIS diode induced by PEALD alternate insertion of AlN in Al₂O₃

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In this paper, AlNO nano-films have been deposited on an AlGaN/GaN heterojunction by alternating growth of AlN and Al₂O₃ using plasma enhanced atomic layer deposition (PEALD). With optimized AlN layer insertion in Al₂O₃, the oxygen is effectively blocked from diffusing to the AlGaN surface and the formation of detrimental Ga–O bonds is significantly suppressed. Owing to the negative fixed charges in Al₂O₃, provided by the incorporated nitrogen, the flat band voltage (V_{fb}) of the AlNO/AlGaN/GaN metal–insulator–semiconductor (MIS) diode exhibits a positive shift of 1.50 V, compared with the Al₂O₃/AlGaN/GaN MIS diode. Markedly reduced hysteresis and frequency-dispersion in the C – V characteristics have also been observed at the AlNO/AlGaN interface. Furthermore, the interface states density (N_{it}) at the AlNO/AlGaN interface has been reduced by one order of magnitude compared with the N_{it} at the Al₂O₃/AlGaN interface, and the border traps density (N_{bt}) near the AlNO/AlGaN interface is also identified to be reduced by the insertion of AlN layers into Al₂O₃. The PEALD induced optimization of AlNO deposition on the AlGaN/GaN heterojunction provides a pathway to the fabrication of AlGaN/GaN high electron mobility transistors (HEMTs) with low interface trap density.

Received 23rd November 2016
Accepted 12th February 2017

DOI: 10.1039/c6ra27190a

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1. Introduction

AlGaN/GaN high electron mobility transistors (HEMTs) are ideal for high-frequency, high temperature and high-voltage power switching applications, due to their superior material and device properties, such as high breakdown electric field, low on-resistance, high switching frequency and high temperature operation.^{1–4} However, there remain urgent issues of large gate leakage and current collapse in Schottky-gate HEMTs (S-HEMTs). The current collapse phenomenon is a temporary reduction of drain-current (I_D) immediately after the application of both gate stress and drain stress, which will result in prominently lowered output power than expected from the dc characteristics of S-HEMTs.^{5,6} In addition, the large gate leakage current in GaN-based S-HEMTs will lead to inferior noise characteristics, larger power consumption and smaller capable gate voltage swing.⁷ In order to address these issues, HEMTs fabricated with a gate dielectric between AlGaN and gate metals, which are referred to as metal–insulator–semiconductor high electron mobility transistors (MIS-HEMTs) have been proposed. High- κ dielectrics such as Al₂O₃,^{8,9} HfO₂,¹⁰ ZrO₂ (ref. 11) and

Ta₂O₅ (ref. 12) are of vital importance to MIS-HEMTs, and among the above high- κ dielectrics, Al₂O₃ has been well accepted as the gate dielectric due to its large band gap (8.7 eV), high breakdown field (5–10 MV cm^{−1}) and relatively high dielectric constant ($\kappa \sim 9$).^{13–15} However, the dielectric/AlGaN interface presents new challenges in suppressing/reducing the interface states or border traps and obtaining stable device operation. It has been revealed that there exist large amounts of interface traps with long and short emission time constant (τ_{it}) at the dielectric/AlGaN interface due to the presence of detrimental Ga–O bonds, leading to the degradation of device performance or reliability problems.^{16,17} The dynamic capture/emission processing of these interface traps, especially the ones relatively deep (with longer τ_{it}) below the AlGaN conduction band may not be in synchrony with the switching gate control signals, resulting in a threshold voltage instability issue.^{18–20} In order to overcome this problem, Zhu *et al.* used AlN to replace high- κ oxides and act as the dielectric in AlGaN/GaN MIS-HEMTs.²¹ However, the leakage current of AlN was high due to the easy crystallization of AlN. In order to reduce the leakage current, Al₂O₃/AlN double-layer could be performed to act as the dielectric, in which AlN was a passivation layer and Al₂O₃ acted as the insulator. Nevertheless, a lower gate capacitance would be obtained because of the series capacitance of the double layers, resulting in the degradation of gate controlled capacity for the device. For the purpose of increasing the insulating property and control capacity of dielectrics on AlGaN/GaN

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heterojunction, Liu *et al.* proposed to reduce the thickness of AlN to 0.5 nm and then deposit 25 nm Al₂O₃ onto it.¹⁸ Nevertheless, 0.5 nm AlN was too thin to isolate oxygen diffusion during 25 nm Al₂O₃ deposition, leading to a low breakdown voltage of the dielectric. In a recent report, nitrogen was incorporated in the bulk of the Al₂O₃ dielectrics using a 1.5 kW inductively coupled remote plasma to dissociate N₂ gas producing activated nitrogen species after each cycle of Al₂O₃ deposition.²² However, this method of nitrogen incorporation in the Al₂O₃ would damage the quality of films and result in oxidation of nitrogen. In this work, alternant AlN incorporation in Al₂O₃ to form AlNO nano-films is proposed as a path to reduce interface trap density and suppress the gate leakage current. This method of alternant growth of two materials is a widely accepted method to deposit a composite films by ALD, such as HfAlO films, AlTiN films and HfZrO films.^{23–26} In addition, nitrogen can incorporate on either cation/anion sites or interstitial sites and thus become a source of negative fixed charge within Al₂O₃, which could contribute to positive shifting of flat band voltage (V_{fb}) (from -8 V to -6 V).²⁷ Insertion of a ~ 0.7 nm AlN interlayer in each ~ 1.6 nm Al₂O₃ deposition alternation can decrease the interface trap density, suppress the voltage hysteresis and reduce the frequency dispersion of gate capacitance compared to the pristine Al₂O₃/AlGaIn/GaN MIS diode.

2. Experimental

The AlGaIn/GaN heterojunction was formed on Si substrate by metal organic chemical vapor deposition (MOCVD) and the bottom-up structure included a 3.9 μ m C-doped GaN buffer layer, a 300 nm GaN channel layer, a 1 nm AlN spacer layer and a 25 nm AlGaIn barrier layer with Al mole fraction of 25%. The AlGaIn/GaN MIS diode (shown in Fig. 1(a)) process began with Ti (20 nm)/Al (100 nm)/Ni (50 nm)/Au (100 nm) deposition and ohmic contact formation achieved by thermal annealing at 870 $^{\circ}$ C for 30 s after mesa isolation. Afterwards, the samples were cleaned with acetone to remove the organic contamination, rinsed with deionized water and blow dried in N₂ before depositing the gate dielectric by plasma enhanced atomic layer deposition (PEALD). The AlNO/AlGaIn/GaN structure was fabricated by alternate growth of AlN and Al₂O₃ nano-lamination on AlGaIn/GaN in a PEALD process. (Al(CH₃)₃) TMA and NH₃/O₂ plasma were adopted as the Al and N/O precursors, respectively.

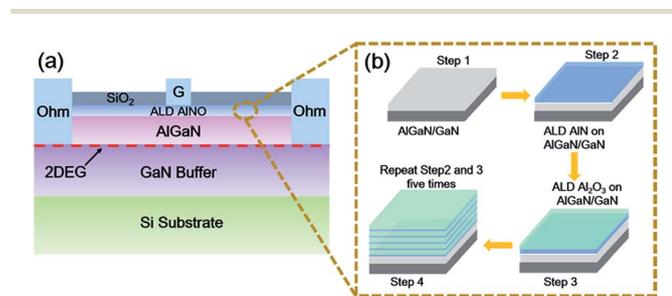


Fig. 1 (a) Schematic cross sectional view of AlGaIn/GaN MIS diode with AlNO nano-films as the gate dielectrics. (b) The process flow charts of AlNO nano-films on AlGaIn/GaN heterojunction.

One ALD cycle of AlN/Al₂O₃ was executed with the completion of following four steps: (1) a 1 s pulse of TMA in duration; (2) a 5 s purge of excess TMA and any byproducts; (3) a 1 s supply of ammonia/oxygen plasma; (4) a 5 s purge of excess ammonia and any byproducts. The chemical reaction equations of TMA and NH₃/O₂ plasma are as follows:

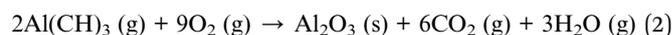


Fig. 1(b) shows the process flow charts of AlNO nano-films on AlGaIn/GaN heterojunction. In total, 5 periods of AlNO nano-films were deposited and each period contained 7 cycles of AlN followed by 14 cycles of Al₂O₃. The thicknesses of one PEALD cycle of AlN and Al₂O₃ were 0.09 nm and 0.12 nm, respectively, confirmed by the spectroscopic ellipsometer (SE). As reference, 90 cycles of Al₂O₃ was also deposited on AlGaIn/GaN heterojunction. Post deposition annealing (PDA) process was performed in nitrogen at 600 $^{\circ}$ C for 30 s to promote inter-diffusing of AlN and Al₂O₃. Then, electron beam evaporation (EBE) was performed to deposit the gate-electrode Ni (30 nm)/Au (100 nm) on AlNO or Al₂O₃. Post metallization annealing (PMA) was implemented in a 95% N₂ and 5% H₂ mixed atmosphere at 400 $^{\circ}$ C for 3 minutes to improve the interface between the dielectric layers and Ni/Au. Finally, 200 nm SiO₂ was deposited by plasma enhanced chemical vapor deposition (PECVD) as the insulator between electrodes.

X-ray photoelectron spectroscopy (XPS) was performed with Al K α X-ray from Axis Ultra DLD equipment to analyze the elemental compositions of AlNO nano-films on AlGaIn/GaN heterojunction, and the binding energy (BE) was calibrated to the position of the C 1s peak at 284.8 eV. The spectra were curve-fitted with a combination of Gaussian and Lorentzian line shapes using a Shirley-type background subtraction. For the investigation of dielectrics/AlGaIn interface, a period of AlNO nano-films and 18 PEALD cycles of Al₂O₃ nano-films were deposited on the AlGaIn/GaN heterojunction, respectively. High resolution transmission electron microscopy (HRTEM) was carried out to show the cross-section microstructure of the dielectric/AlGaIn interface. The leakage current (I - V) and capacitance-voltage (C - V) measurements were carried out to reveal the high quality of AlNO nano-films on AlGaIn/GaN heterojunction. In addition, the frequency- and voltage-dependent conductance method was utilized to characterize the interface trap density at the dielectric/AlGaIn interface.

3. Results and discussion

The XPS Al 2p, O 1s, N 1s and O 1s loss energy spectra of AlNO nano-films on AlGaIn/GaN heterojunction were illustrated in Fig. 2. The peaks of Al 2p spectra, as shown in Fig. 2(a) could be fitted by two sub-peaks located at 74.1 eV and 74.8 eV, which were corresponding to Al-N and Al-O bonds, respectively.^{28,29} Fig. 2(b) showed the symmetric O 1s peak located at 530.9 eV corresponding to the O-Al bonds, indicating nitrogen was not



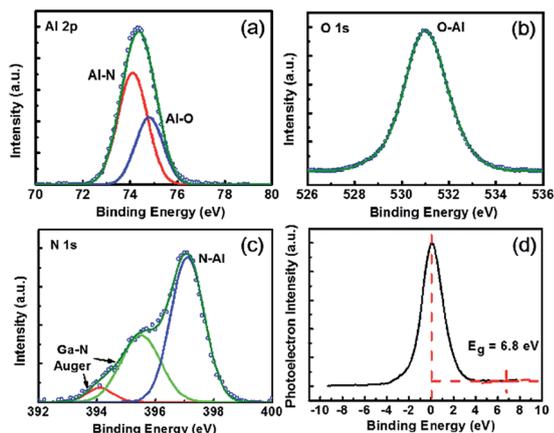


Fig. 2 XPS spectra of (a) Al 2p, (b) O 1s, (c) N 1s, and (d) O 1s energy-loss spectra of AlNO nano-films on AlGaIn/GaN heterojunction.

oxidized during the PEALD and the subsequent annealing processes. As shown in Fig. 2(c), the N 1s peak could be fitted by three sub-peaks located at 394.1, 395.5 and 397.1 eV, respectively. The sub-peak located at 397.1 eV was assigned to the N-Al bonds,²² whereas the peaks located at 394.1 and 395.5 eV were identified as the Ga-N Auger peaks, which were originated from the AlGaIn substrate.^{30,31} It was worth to mention that no N-O bonds was detected, further confirming the nitrogen was not oxidized. Furthermore, XPS quantitative analysis was employed to determine the chemical composition of the AlNO nano-films. The atomic fractions in AlNO nano-films was Al (41.5%)–N (5.2%)–O (53.3%). O 1s energy-loss spectrum was also performed to calculate the band gap of AlNO nano-films on AlGaIn/GaN heterojunction, as is shown in Fig. 2(d). The obtained band gap of AlNO was 6.8 eV, which was between the band gap of pure Al₂O₃ (7.3 eV) and AlN (6.4 eV) reported in the literature.^{32,33}

Fig. 3 showed the leakage current density (J_G) versus gate voltage (V_G) for the AlNO/AlGaIn/GaN and Al₂O₃/AlGaIn/GaN MIS diodes. Compared to the AlGaIn/GaN MIS diode with Al₂O₃ nano-films, the AlNO nano-films exhibited a well-suppressed gate leakage current density of 3.45×10^{-8} A cm⁻² up to a forward bias of 3.5 V and the corresponding the breakdown

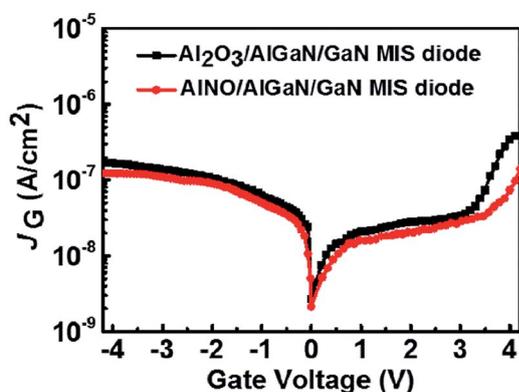


Fig. 3 J_G - V_G characteristics of AlNO/AlGaIn/GaN and Al₂O₃/AlGaIn/GaN MIS diodes.

electrical field was up to 3.2 MV cm⁻¹. Specially, the reported breakdown electrical field of AlNO deposited by N₂ plasma incorporation in ALD Al₂O₃ was only 1.5 MV cm⁻¹.²² These data implied that oxygen diffusion could be effectively suppressed by alternant AlN incorporation in Al₂O₃, resulting in a lower leakage current density and a higher breakdown electrical field of AlNO/AlGaIn/GaN MIS diode than that for Al₂O₃/AlGaIn/GaN MIS diode.

Fig. 4 showed the hysteresis and multi-frequency C - V curves of Al₂O₃/AlGaIn/GaN and AlNO/AlGaIn/GaN MIS diodes. There existed two abrupt slopes in all C - V curves, one was at negative voltage corresponding to the accumulation at 2DEG interface, and the other one was at positive voltage corresponding to the gate dielectric/AlGaIn interface. The dielectric constant of the AlNO nano-films deduced from the C - V curves was 6.9, which was slightly less than that of Al₂O₃ (7.4). Due to the smaller

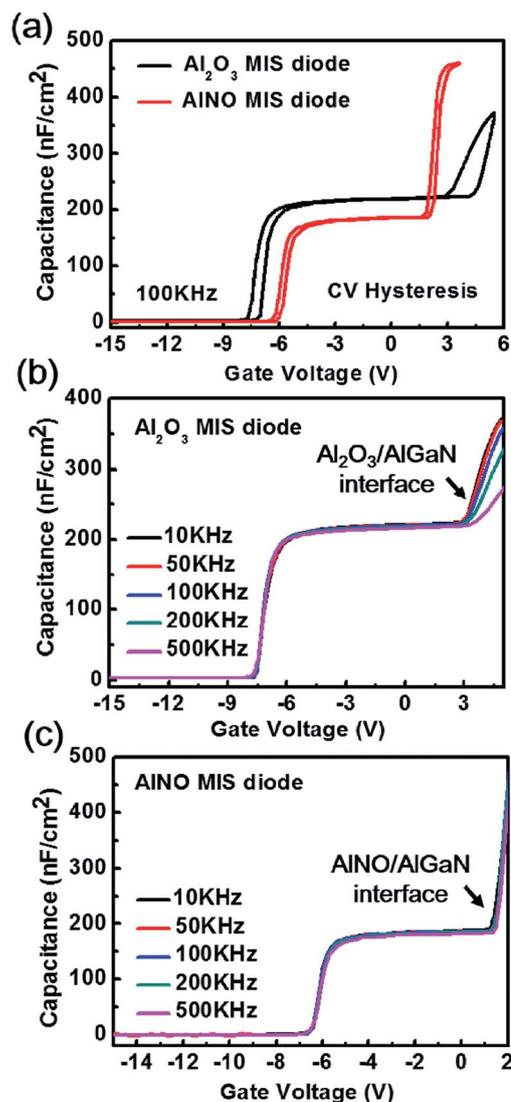


Fig. 4 (a) Hysteresis C - V curves of AlNO/AlGaIn/GaN and Al₂O₃/AlGaIn/GaN MIS diodes; (b) multi-frequency C - V curves of Al₂O₃/AlGaIn/GaN MIS diode; (c) multi-frequency C - V curves of AlNO/AlGaIn/GaN MIS diode.



dielectric constant of AlNO, it had a smaller capacitance than Al₂O₃ under the same thickness and area. In addition, dielectrics were in series to AlGa_{0.5}N on GaN. Therefore, a smaller voltage would be sustained by the AlNO with respect to the Al₂O₃ when the same voltage was applied at the gate, leading to a smaller bend up voltage in the second abrupt slope of the *C*-*V* curves in the AlNO/AlGa_{0.5}N/GaN MIS diode. The hysteresis *C*-*V* curves, measured at 100 kHz from all samples were shown in Fig. 4(a). Compared to the Al₂O₃/AlGa_{0.5}N/GaN MIS diode, the diode with AlNO nano-films achieved a positive *V*_{fb} shift of 1.50 V and a negligible *V*_{fb} hysteresis, which indicated that AlN incorporating into Al₂O₃ would compensate the positive fixed charge within Al₂O₃.^{34,35} The second step reflected the state of gate-dielectrics/AlGa_{0.5}N interface, and compared to the Al₂O₃/AlGa_{0.5}N/GaN MIS diode, the AlNO/AlGa_{0.5}N/GaN MIS diode exhibited a significant reduced hysteresis, indicating there was an excellent interface quality at the AlNO/AlGa_{0.5}N interface. In addition, the dynamic capacitance dispersion measurement was also accomplished to further analyze the dielectric/AlGa_{0.5}N interface. The measured *C*-*V* curves with frequency varied from 10 kHz to 500 kHz were shown in Fig. 4(b) and (c). The frequency dispersion of capacitance in *C*-*V* curves due to the dielectric/AlGa_{0.5}N interface trap response was observable in the second slope region for both samples. For one given *f*, only those interface traps with energy level (*E*_T) aligned to Fermi level (*E*_F) and τ_{it} shorter than $1/f$ could respond to ac anode signal and contribute additional capacitance with *V*_G increasing. With a higher frequency *f*, larger *V*_G was required to raise the *E*_F towards conduction band (*E*_C) so that the shallower interface traps with smaller τ_{it} could respond, resulting in frequency dispersion of capacitance in the second slope bias region.¹⁸ Severe frequency dispersion of capacitance in the Al₂O₃/AlGa_{0.5}N/GaN MIS diode was attributed to high-density interface trap because of the presence of detrimental interfacial layer Ga-suboxide (GaO_x) shown in Fig. 4(b). As shown in Fig. 4(c), the frequency dispersion was effectively suppressed in the AlNO/AlGa_{0.5}N/GaN MIS diode, indicating the interface trap density was significantly reduced. As a result, insertion of AlN in Al₂O₃ could potentially suppress the formation of unstable interfacial layers such as GaO_x at the AlNO/AlGa_{0.5}N interface.

The cross-section microstructures of AlNO/AlGa_{0.5}N and Al₂O₃/AlGa_{0.5}N interface were both investigated by HRTEM, as shown in Fig. 5(a) and (b), respectively. There was a rough and nearly ~2 nm interfacial layer at the Al₂O₃/AlGa_{0.5}N interface, which was originated from the GaO_x formed during the Al₂O₃ deposition. Nevertheless, in the AlNO/AlGa_{0.5}N/GaN MIS diode, the AlN layer not only could act as a passivation layer on AlGa_{0.5}N, but also could serve as a separation membrane to suppress the oxidation process during subsequent Al₂O₃ deposition. By alternative deposition of AlN and Al₂O₃ layer to form AlNO nano-films on AlGa_{0.5}N/GaN heterojunction, a uniform and sharp AlNO/AlGa_{0.5}N interface could be obtained. It was worth to mention that intermixed Al₂O₃ and AlN rather than Al₂O₃ and AlN overlays was obtained in our work. This conjecture was firstly supported by the fact that no layered structures were observed by HRTEM (Fig. 5(a)). Furthermore, it could also be confirmed by the huge difference in per-unit-length capacitance between AlNO nano-

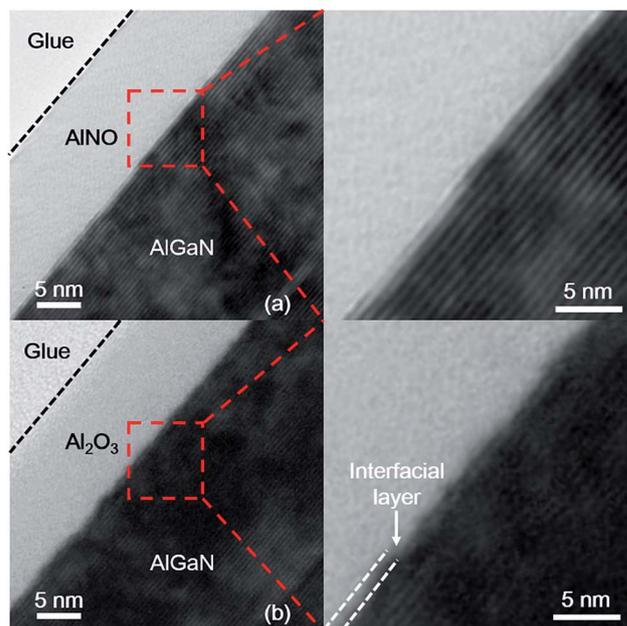


Fig. 5 High resolution transmission electron microscopy (HRTEM) micrographs of (a) Al₂O₃/AlGa_{0.5}N and (b) AlNO/AlGa_{0.5}N interfaces.

films and segregated Al₂O₃/AlN layers. The per-unit-length capacitances of pure Al₂O₃ and AlN on AlGa_{0.5}N/GaN were 573 nF cm⁻² and 1270 nF cm⁻², respectively,³³ and the calculated per-unit-length capacitance of segregated Al₂O₃ (8 nm)/AlN (3 nm) layers was 126 nF cm⁻², which was much less than the measured capacitance of AlNO nano-films (580 nF cm⁻²). The high per-unit-length capacitance of AlNO nano-films was due to incorporation of AlN into Al₂O₃ and this result agreed well with the reports of M. Cho *et al.* and C. An *et al.*^{36,37} We thus proposed that the AlNO nano-films fabricated using the approach described in this paper was effectively Al₂O₃-AlN alloy. In addition, the PDA process at 600 °C could indeed promote interdiffusion of AlN and Al₂O₃, but the aim of performing 600 °C PDA was to prove that AlNO nano-films could maintain the amorphous state even at 600 °C annealing due to Al₂O₃ incorporation. AlN was known to crystallize even at 500 °C and lead to a serious gate leakage current.³³

To investigate the difference of interface chemistry and bonding states between AlNO/AlGa_{0.5}N and Al₂O₃/AlGa_{0.5}N interfaces, the evolution of Ga 3d core-level spectra of AlNO/AlGa_{0.5}N and Al₂O₃/AlGa_{0.5}N interfaces were both demonstrated. As shown in Fig. 6(a), the Ga-O peak of Al₂O₃/AlGa_{0.5}N interface was obvious while a lower intensity of Ga-O peak at the AlNO/AlGa_{0.5}N interface was obtained (shown in Fig. 6(b)), manifesting that the GaO_x were effectively suppressed by the alternative growth of AlN and Al₂O₃ in AlNO nano-films.

The frequency- and voltage-dependent conductance measurement was carried out to quantitative deduce the interface trap distribution.³⁸ With the MIS diode biased at the second slope in the *C*-*V* curves, AlGa_{0.5}N/GaN heterojunction interface trap located in the band gap were far below the Fermi level, thus only the dielectric/AlGa_{0.5}N interface trap could respond to the ac signal. The



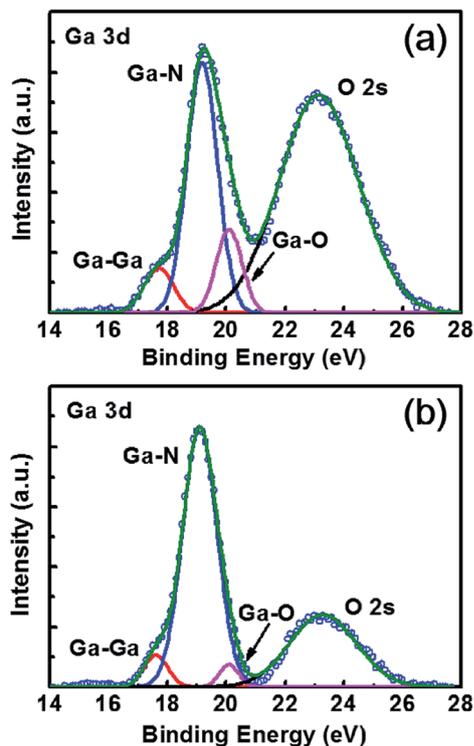


Fig. 6 XPS spectra of Ga 3d core levels of (a) 18 cycles growth of Al_2O_3 films on AlGaIn/GaN heterojunction, and (b) one period cycles growth of AlInO nano-films on AlGaIn/GaN heterojunction.

measurement frequency was ranging from 1 kHz to 1 MHz at room temperature. The equivalent parallel conductance G_p/ω was given by the following equation,

$$\frac{G_p}{\omega} = \frac{q\omega\tau_{it}D_{it}}{1 + (\omega\tau_{it})^2} \quad (3)$$

where $\omega = 2\pi f$ was the radial frequency, τ_{it} was trap time constant, and D_{it} was the interface trap density. The parallel

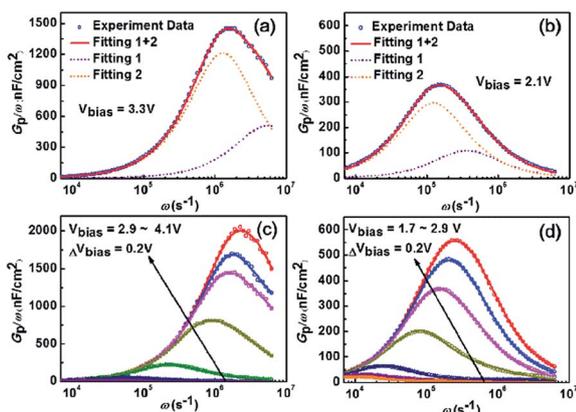


Fig. 7 The measured curves and fitting curves of G_p/ω vs. ω at the V_{bias} of 3.3 V and 2.1 V for (a) $\text{Al}_2\text{O}_3/\text{AlGaIn}/\text{GaN}$ and (b) $\text{AlInO}/\text{AlGaIn}/\text{GaN}$ MIS diodes, respectively. Measured curves and fitting curves of G_p/ω vs. ω at the V_{bias} of 2.9 to 4.1 V and 1.7 to 2.9 V for (c) $\text{Al}_2\text{O}_3/\text{AlGaIn}/\text{GaN}$ and (d) $\text{AlInO}/\text{AlGaIn}/\text{GaN}$ MIS diodes, respectively. Each fitting curve was the superposition of fitting 1 and fitting 2 for both diodes.

conductance G_p/ω was related to the measured capacitance C_m and conductance G_m , and could be calculated from the relation:

$$\frac{G_p}{\omega} = \frac{\omega G_m C_b^2}{G_m^2 + \omega^2 (C_b - C_m)^2} \quad (4)$$

where C_b represented the static-state capacitance of MIS diodes, and the measured capacitance C_m represented the series connection of the trap capacitance C_{trap} to the dielectric layer capacitance $C_{\text{dielectrics}}$ and the AlGaIn layer capacitance C_{AlGaIn} .

The D_{it} and the corresponding τ_{it} at the dielectric/AlGaIn interface could be extracted by fitting the experimental data using eqn (3). As shown in Fig. 7(a) and (b), both the G_p/ω curves for $\text{Al}_2\text{O}_3/\text{AlGaIn}/\text{GaN}$ and $\text{AlInO}/\text{AlGaIn}/\text{GaN}$ MIS diodes could be resolved into the superposition of two fitting curves at the

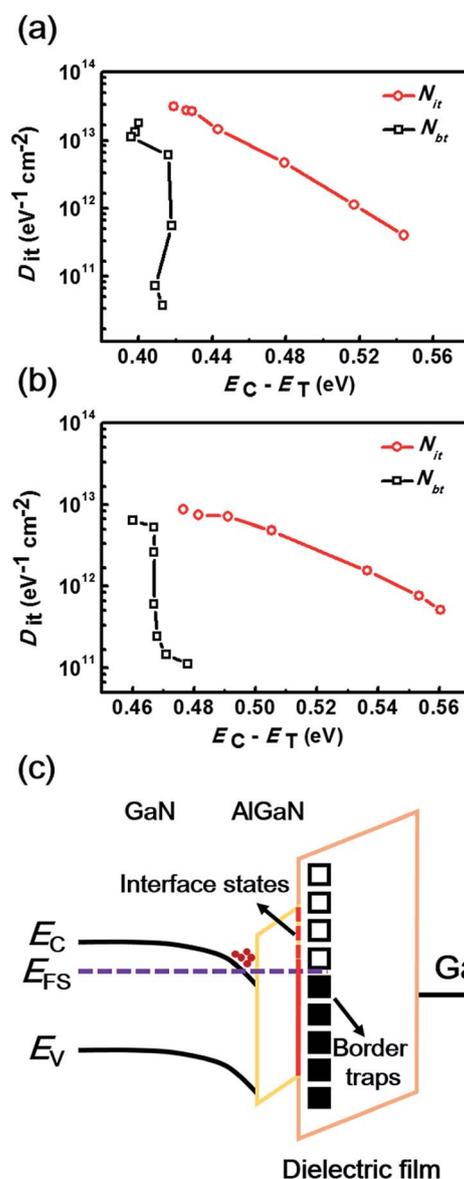


Fig. 8 Dependence of interface trap density on trap energy level below AlGaIn conduction band at (a) $\text{Al}_2\text{O}_3/\text{AlGaIn}$ and (b) $\text{AlInO}/\text{AlGaIn}$ interfaces. (c) Schematic band diagram of interface states and border traps in the dielectric/AlGaIn/GaN MIS diodes.



given bias voltage of 3.3 V and 2.1 V, respectively, indicating there were two types of trap states presenting at the dielectrics/AlGa_n/Ga_n interface, *i.e.*, interface states and border traps. Border traps were the near interfacial oxide traps and could be regarded as either interface traps or bulk oxide traps, relating to the gate bias, voltage ramp rate and measurement frequency. The densities of both interface states (N_{it}) and border traps (N_{bt}) were determined to characterize the interface properties. As exhibited in Fig. 7(c) and (d), similar fitting results could be obtained in the voltage bias range of 2.9 V to 4.1 V and 1.7 V to 2.9 V for both AlNO/AlGa_n/Ga_n and Al₂O₃/AlGa_n/Ga_n MIS diodes, respectively. For clarity, only superposed fitting curves were presented.

The energy levels E_T of traps with respect to E_C , *i.e.* $E_C - E_T$, were related to it through the following equation,³⁹

$$\tau_{it} = \frac{1}{\nu_{th}\sigma_n N_c} \exp\left(\frac{E_C - E_T}{kT}\right) \quad (5)$$

where T was room temperature (300 K), $\sigma_n = 4 \times 10^{-13} \text{ cm}^2$ was the electron capture cross section, $N_c = 2.2 \times 10^{18} \text{ cm}^{-3}$ represented the effective density of states in Al_{0.25}Ga_{0.75}N conduction band, and $\nu_{th} = 2.6 \times 10^7 \text{ cm s}^{-1}$ was the thermal velocity.⁴⁰ The dependence of D_{it} on trap energy level $E_C - E_T$ could be extracted by using eqn (3) and (5) and shown in Fig. 8. The interface states exhibited strong continuous distribution in energy level, and the energy level of border traps were located at a very narrow energy range. Compared to the Al₂O₃/AlGa_n/Ga_n MIS diode with a N_{it} of 10^{13} to $10^{11} \text{ eV}^{-1} \text{ cm}^{-2}$ in the energy level range of 0.42 eV to 0.54 eV (Fig. 8(a)), the N_{it} in the AlNO/AlGa_n/Ga_n MIS diode varied from 10^{12} to $10^{11} \text{ eV}^{-1} \text{ cm}^{-2}$ with the energy level range of 0.48 eV to 0.56 eV (Fig. 8(b)), indicating the interface states density was significantly reduced and the GaO_x was effectively suppressed at the AlNO/AlGa_n interface. In addition, the introduced AlN layer could also separate oxide border traps near the interface in the AlNO nano-films. As a result, both high quality border region and reduced N_{bt} could be obtained. The schematic band diagram of interface states and border traps was shown in Fig. 8(c). For a positive gate bias voltage, the conduction band of the AlGa_n barrier could be pulled down and the electrons would transfer from the channel to the dielectrics/AlGa_n interface, where they were trapped or detrapped by interface states or border traps.

4. Conclusions

In conclusion, by alternating growth of AlN and Al₂O₃ nanolamination using NH₃ and O₂ plasma in a PEALD process, AlGa_n/Ga_n MIS diode with AlNO nano-films as the gate dielectric has been fabricated. With the insertion of AlN layer, the AlNO/AlGa_n features a sharp interface with effective suppression of AlGa_n surface oxidation, leading to significantly reduced hysteresis and frequency-dispersion of $C-V$ characteristics. In addition, nitrogen incorporation can produce negative fixed charge in Al₂O₃ gate dielectric, which was contributed to positive shift in the flat band voltage. Compared with the Al₂O₃/AlGa_n/Ga_n MIS diode, an improved interface with a lower interface states and reduced border traps was obtained in the

AlNO/AlGa_n/Ga_n MIS diode. The manufacture of AlNO nano-films by PEALD could provide a pathway to achieve the operation of enhancement mode AlGa_n/Ga_n MIS-HEMTs with low interface trap density.

Acknowledgements

This work is funded by the National Natural Science Foundation of China (Grant No. 16ZR1442300) and the Natural Science Foundation of Shanghai (Grant No. Y52GXAJ01).

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