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# Stable ambipolar organic–inorganic heterojunction field-effect transistors and inverters with Cytop interlayer†

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In this study, ambipolar field-effect transistors (FETs) based on organic–inorganic bilayer structures were investigated. InO<sub>x</sub> and pentacene were selected as n-type and p-type semiconductors, respectively. To improve the hole mobility, poly(perfluorobutenylvinylether) (Cytop) layer was introduced to modify the surface of InO<sub>x</sub>. The ambipolar FETs showed hole and electron mobilities of 1.1 and 0.1 cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup>, respectively and excellent electrical stability under gate bias stress. Furthermore we found that ambipolar FETs could be integrated into functional complementary metal oxide semiconductor (CMOS)-like inverters and exhibited high peak gain (>50). This work provides a general method for realizing ambipolar FETs based on organic–inorganic hybrid structure.

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## Introduction

In recent years, ambipolar field-effect transistors (FETs) have been extensively studied for technological applications in complementary logic circuits due to their unique operation, which allows both holes and electrons to be injected and transported depending on the gate bias in a single device.<sup>1–5</sup> Organic semiconductors and metal-oxide semiconductors are two families of active materials that are currently drawing much attention in FETs.<sup>6–8</sup> However, they usually exhibit unipolar charge transport, which limits their applications in ambipolar FETs. Although organic semiconductors can transport both holes and electrons at the same time, the electron transport for most of the organic semiconductor is poor compared to the hole transport.<sup>9</sup> In contrary to the organic semiconductors, most metal-oxide semiconductors have only n-type characteristics.<sup>10,11</sup> Therefore, it is possible to achieve ambipolar transport by using organic semiconductor/oxide semiconductor bilayer structure. Nakanotani *et al.* developed an ambipolar FET based on an organic–inorganic hybrid structure containing indium zinc oxide and pentacene bilayer with hole and electron mobilities of 0.14 and 13.8 cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup>, respectively.<sup>12</sup> This work opened up a new field in ambipolar FETs fabrication. Smith *et al.* reported an air-stable solution-processed hybrid transistors with hole and electron mobilities exceeding 2 cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup>, but the device showed ambipolar performance only in the dual-

gated model, which is complicated and costly.<sup>13</sup> Yang *et al.* reported ambipolar FETs based on pentacene/ZnO hybrid structures and got balance hole and electron mobilities of 0.34 and 0.38 cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup> respectively.<sup>14</sup> Kim *et al.* reported ambipolar FETs based on carbon nanotube/zinc tin oxide heterostructures by inkjet printing with hole and electron mobilities exceeding 2 cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup>, but the processing temperature is as high as 500 °C, which was too high to fabricate on most of the flexible substrates.<sup>15</sup> Ambipolar FETs have also been realized by several approaches, such as bilayer and blend. But most work can't exhibit good stability.<sup>16–20</sup>

Herein, we report on a high-stable ambipolar FET based on an organic–inorganic bilayer structure channel. For the p-type organic semiconductor, we chose pentacene which was well studied and exhibit stable performance. InO<sub>x</sub> was selected as the n-type metal oxide semiconductor. In our previous work, directly patternable solution-processed InO<sub>x</sub> films could be fabricated with an economic and environmentally friendly approach at low temperature.<sup>21</sup> To improve the p-type performance, the surface of the InO<sub>x</sub> layer was modified by Cytop.<sup>22</sup>

## Experimental section

Fig. 1 shows the schematic structure of the ambipolar FET. Firstly, a 300 nm thick Al–Nd (3 wt%) film was deposited with DC sputtering and patterned with conventional photolithography method. Afterwards, Al:Nd film was anodized for 1.5 h to produce a layer of 200 nm thick AlO<sub>x</sub>:Nd (41 nF cm<sup>-2</sup>) on the surface of the Al:Nd film as the gate insulator. After 5 min oxygen plasma treatment, the InO<sub>x</sub> film was spin-coated and patterned in a facile method, following which the InO<sub>x</sub> film underwent an annealing process at 220 °C for 2 h in air. After

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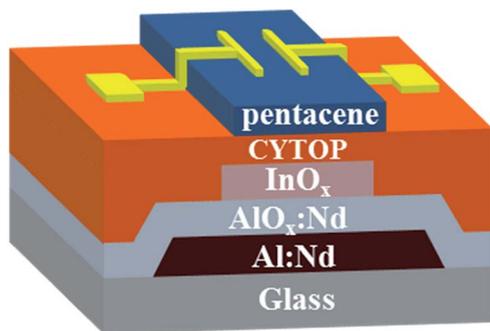


Fig. 1 Schematic diagram of ambipolar FETs based on InO<sub>x</sub>/Cytop/pentacene hybrid structure.

that, the 45 nm thick Cytop film was spin-coated as an interlayer to modify the growth of pentacene. After that, a layer of 30 nm thick pentacene film was thermally evaporated onto the insulator at room temperature under a vacuum pressure of  $\sim 3 \times 10^{-4}$  Pa with a deposition rate of  $0.1 \text{ \AA s}^{-1}$ . Then a 45 nm thick Au film was thermally evaporated onto the pentacene film through a shadow mask, defining a channel width/length ( $W/L$ ) of  $500/70 \text{ \mu m}$ . The TFT characterizations were measured with a semiconductor parameter analyzer (Agilent 4155C) and a probe station at room temperature (RT) in vacuum environment.

## Results and discussion

Fig. 2 shows the output and transfer characteristics of the ambipolar FETs based on the organic-inorganic heterostructures with the Cytop layer. The FETs exhibited ambipolar performance with hole and electron mobilities of  $1.1$  and  $0.1 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ , respectively, which were extracted from the following equation,

$$I_{\text{DS}} = \mu \frac{W}{2L} C_i (V_G - V_T)^2, \quad (1)$$

where  $I_{\text{DS}}$  is drain-source current,  $V_G$  is gate voltage,  $V_T$  is the threshold voltage,  $C_i$  is capacitance density. Specifically,  $C_i$  ( $22 \text{ nF cm}^{-2}$ ) for hole mobility and  $C_i$  ( $41 \text{ nF cm}^{-2}$ ) for electron mobility were different, because the 20 nm thick InO<sub>x</sub> and 45 nm thick Cytop layer were acted as the dielectric layers when the FET was operated in the p-channel mode, as shown in Fig. 3.  $C_i$  for hole mobility was constant as gate voltage varies from 0 V to  $-30 \text{ V}$ , as shown in Fig. S3.†

It could be also seen from Fig. 2 the hysteresis between the forward and reverse sweeps in the transfer curve is too little to be detected. It implies that the amount of fast traps in the bulk or at the interface is small. It is worth noting that the ambipolar FETs without Cytop or with 20 nm Cytop exhibited large hysteresis, as shown in Fig. S1,† which was attributed to the large amount of hole traps on the surface of InO<sub>x</sub> (and 20 nm Cytop was not thick enough to block the holes from trapping into InO<sub>x</sub>). Besides Cytop, polymethyl methacrylate (PMMA) and polystyrene (PS) were also employed as the modification layer. It can be seen from Fig. S2† that the PMMA and PS layer can also improve FET performance, but not as good as Cytop.<sup>23,24</sup> Compared to PMMA and PS, Cytop film is smoother and more hydrophobic. Besides, the density of states (DOS) of Cytop film is low due to the low dielectric constant, resulting in higher mobility and better stability.

Fig. 4(a) and (b) show the atomic force microscopy (AFM) images of pentacene films on InO<sub>x</sub> and Cytop/InO<sub>x</sub>, respectively. The pentacene grain size on InO<sub>x</sub> was much larger than that on Cytop/InO<sub>x</sub>. The grain size was based on surface roughness and surface energy. It is known that the pentacene film tends to grow into larger size on the surface of low roughness and high surface energy.<sup>25</sup> The roughness of InO<sub>x</sub> (1.2 nm) is similar as that of Cytop (1.4 nm), tested by AFM. The surface energy of InO<sub>x</sub> ( $45.6 \text{ mN m}^{-1}$ ) is larger than that of Cytop ( $13.5 \text{ mN m}^{-1}$ ), which was calculated by OWRK (Owens-Wendt-Rabel-Kaelble) equation using contact angles of water, ethylene glycol and diiodomethane on the surface of InO<sub>x</sub> and Cytop.<sup>26,27</sup> Although the pentacene grain size on InO<sub>x</sub> is larger than that on Cytop/InO<sub>x</sub>, the hole transport of the FETs without Cytop can hardly be

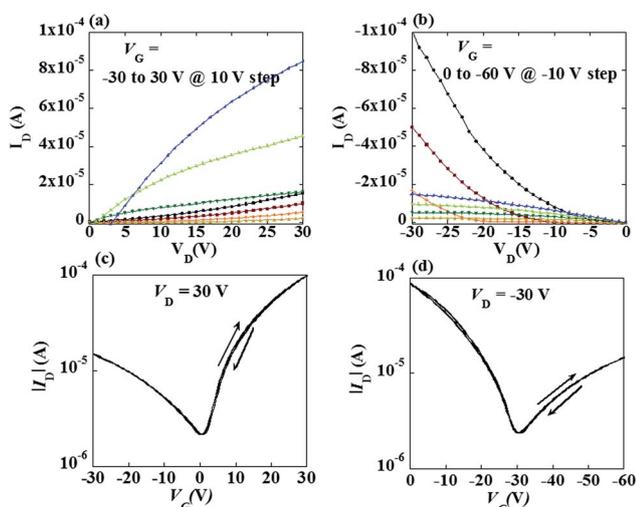


Fig. 2 Output characteristics of ambipolar FETs based on InO<sub>x</sub>/Cytop/pentacene hybrid structure: (a) in electron enhancement mode and (b) in hole enhancement mode. Transfer characteristics of ambipolar FETs based on InO<sub>x</sub>/Cytop/pentacene hybrid structure: (c)  $V_D = 30 \text{ V}$  and (d)  $V_D = -30 \text{ V}$ . Every curve includes forward and reverse sweeps.

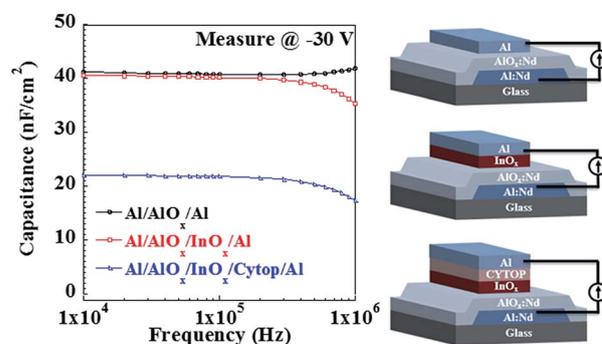


Fig. 3 Capacitance vs. frequency plots measured at  $-30 \text{ V}$  for AlO<sub>x</sub>:Nd, AlO<sub>x</sub>:Nd/InO<sub>x</sub> and AlO<sub>x</sub>:Nd/InO<sub>x</sub>/Cytop; inset: the structures of these three capacitances.



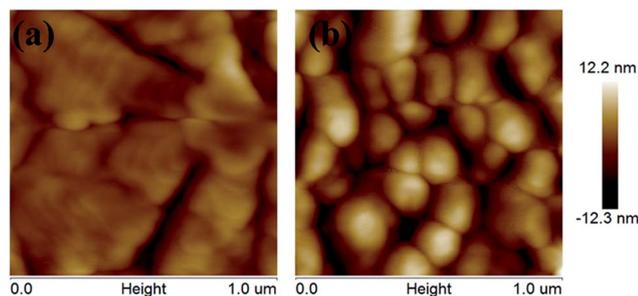


Fig. 4 AFM images of pentacene films on (a) InO<sub>x</sub> and (b) Cytop/InO<sub>x</sub>.

detected. This phenomenon was ascribed to the high electron density of InO<sub>x</sub>. In FETs without Cytop, the large amount of the undepleted electrons can act as the hole traps, so the holes in the organic semiconductor are trapped. After inserted with a Cytop interlayer, the undepleted electrons from InO<sub>x</sub> are blocked by Cytop, so the holes in pentacene cannot be trapped by the electrons from InO<sub>x</sub>. As a result, the hole transport of the FETs with Cytop layer improves.

The electrical stability under gate bias stress is an important parameter in FETs. Fig. 5 and 6 show negative bias stress (NBS) and positive bias stress (PBS) stability of the ambipolar FETs, respectively. Almost no threshold voltage shift was found during the PBS test, while a little positive shift was found during the NBS test. The high bias stability was ascribed to the Cytop modification layer. For electron transport, the Cytop acted as the passivation layers to prevent the back channel of InO<sub>x</sub> from interacting with oxygen and moisture from the air. For hole transport, the Cytop layer can passivated the OH groups on the InO<sub>x</sub> surface. It is known that the OH groups at the insulator/semiconductor interface would cause device instability. There are no OH groups in the Cytop molecule, and the Cytop films are highly hydrophobic with a water contact angle of 106° which is better than those of many other polymers. Thus, the Cytop modification will improve the electrical stability for both electron and hole transport.<sup>22</sup>

Finally, a complementary-like inverter was fabricated with the integration of two identical pentacene/Cytop/InO<sub>x</sub> ambipolar FETs. Fig. 7(a) shows the voltage transfer characteristics. The input voltage was applied through the Al:Nd bottom gate

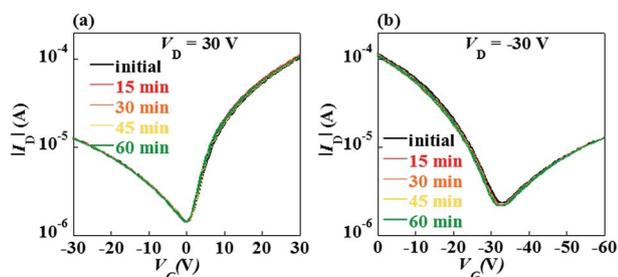


Fig. 5 The time-dependent transfer property ((a)  $V_D = 30$  V and (b)  $V_D = -30$  V) under negative gate-bias-stress for ambipolar FETs stress condition:  $V_G = 30$  V,  $V_D = 30$  V. Every curve includes forward and reverse sweeps.

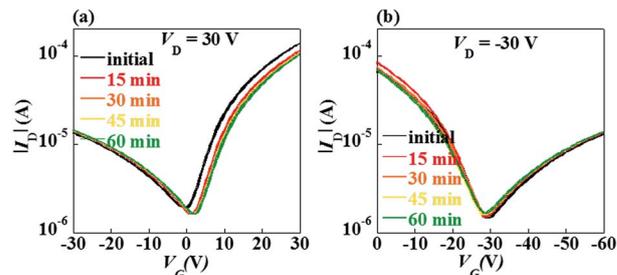


Fig. 6 The time-dependent transfer property ((a)  $V_D = 30$  V and (b)  $V_D = -30$  V) under negative gate-bias-stress for ambipolar FETs stress condition:  $V_G = -30$  V,  $V_D = -30$  V. Every curve includes forward and reverse sweeps.

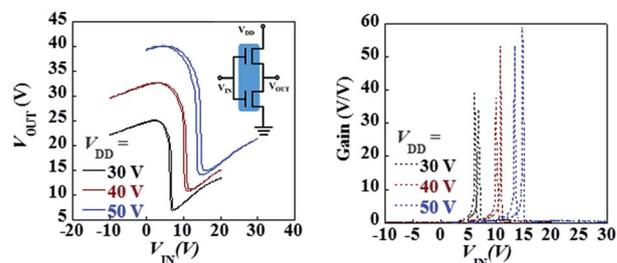


Fig. 7 (a) The voltage transfer characteristics of the CMOS-like inverter at different supply voltages of 30 V, 40 V, 50 V. Inset: the circuit diagram of the inverter. (b) The corresponding voltage gains of the inverter.

electron. Three supply voltages ( $V_{DD}$ ) of 30 V, 40 V and 50 V were biased to the inverter respectively. The output voltage ( $V_{OUT}$ ) varied from high to low in response to the swing of the  $V_{IN}$  from low to high, and sharp inversions were found to be near 5 V, 10 V and 15 V, respectively. The voltage grain, defined as  $|dV_{OUT}/dV_{IN}|$  was calculated from the voltage transfer curves, as shown in Fig. 7(b). The voltage gains reached maximum value of 40, 54 and 58. The  $V_{OUT}$  was not constant at both high and low values of  $V_{IN}$ , where  $V_{OUT}$  increased slightly with  $V_{IN}$ . This common feature of ambipolar inverters arose because one of the ambipolar FETs is not completely turned off while the second FET was turned on.<sup>15</sup> Overall, the inverter exhibited better electrical stability and voltage gain compared to most of other inverters based on unipolar p-type and n-type FETs and ambipolar FETs.<sup>28–30</sup>

## Conclusions

In conclusion, the stable ambipolar FETs and CMOS-like inverters based on organic–inorganic heterostructures were fabricated. Pentacene and InO<sub>x</sub> as the semiconducting p-type and n-type layers composed the hybrid structure. Cytop was inserted to modify the organic–inorganic interface by reducing the surface energy to improve the hole mobility and the stability. As a result the ambipolar FETs showed hole and electron mobilities of 1.1 and 0.1 cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup>, respectively and excellent electrical stability under gate bias stress. COMS-like



inverters showed clear signal inverting operation with a high gain of 58. These results suggest that FETs based on organic-inorganic heterostructures are promising candidates for realizing high stable ambipolar logic devices.

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