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Self-immolative polymers as novel pH-responsive gate keepers for drug delivery†

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A novel pH-sensitive nanocarrier based on mesoporous silica nanoparticles with self-immolative polymers blocking the pore openings is presented. Triggered release by acid pH is demonstrated, together with their *in vitro* biocompatibility and effective cell internalisation, which makes this new material a promising candidate for future applications in cancer treatment.

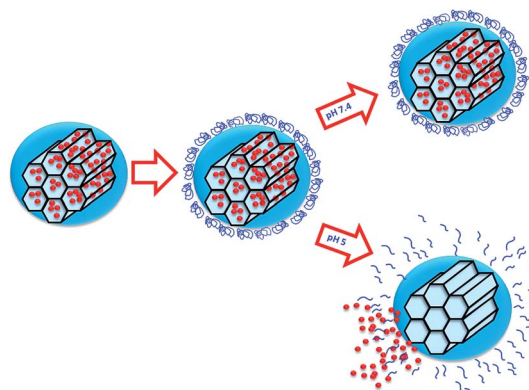
Nanotechnology, which has the potential to impact nearly every area of modern society, is transforming medicine thanks to the development of new smart drug delivery systems.^{1–3} Those nanocarriers are expected to bring breakthroughs in terms of detecting, diagnosing and treating different forms of cancer.⁴ Recently, Mesoporous Silica Nanoparticles (MSNs) have drawn much attention for many biomedical applications, especially as drug delivery nanocarriers.^{5–10} The reason for this relies on their unique properties, such as a straightforward synthesis, a network of cavities ready available to be filled with different payloads, tuneable pore diameters and morphologies, facile surface functionalisation and hemo- and biocompatibility.^{11,12}

The textural properties of MSNs, such as large surface areas (*ca.* 1000 m² g⁻¹) and large pore volume (*ca.* 1 cm³ g⁻¹), are responsible for their high loading capacities. In fact, conventional polymeric nanoparticles suffer from the drawback of low drug capacity, usually less than 5% of total weight, while MSNs allow much greater values.^{13,14}

However, MSNs present an open structure of the pores, which means that is simple to introduce molecules into their network of cavities, but it is also easy for them to diffuse out. This is the main reason to place gatekeepers on the entrance of

the pores, so they can be closed once the cargo is loaded, and then opened to allow the drug release. The former process can be carried out during the production of the nanocarrier, but the later event, opening the gates of the pores, must be done on-demand inside the body. This approach leads to stimuli-responsive nanocarriers,¹⁵ in which the release of the cargo is triggered by a stimulus than can be exogenous (temperature, light, magnetic fields or ultrasounds) or endogenous (pH, redox potential or specific enzymes or analytes). Those systems allow tailoring the release profiles with temporal and dosage control.

Among the available stimuli, pH variations offer the possibility of controlling the delivery of drugs to different regions of the body such as the gastrointestinal tract, the endosomes or lysosomes, or the tumour microenvironment. In particular, the slightly acidic lysosomal pH value (4.5–5)¹⁶ has been used for the design of pH-targeted nanocarriers that release their cargo once inside the cell.^{17,18} Then, an efficient pH-responsive system should respond to this subtle change of pH within this organelle, triggering the release of the cargo only inside the cells (Scheme 1).



Scheme 1 Representation of the operating principle of our pH sensitive nanocarrier. A change in the pH initiates the disassembling of the self-immolative polymer allowing cargo molecules to diffuse out of the MSNs.

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There are many polymeric systems with acid-sensitive bonds that can be exploited for designing responsive systems. In this sense, there are a new class of polymers, the so-called Self-Immolative Polymers (SIPs), that disassemble from head to tail completely when a specific functional group is cleaved from the polymer in response to certain stimuli.^{19,20} One potential advantage is that it is possible to incorporate prodrugs into the polymer chain as pendant groups. These groups would turn into actual drugs as the self-immolation takes place.²¹ It is also possible to modify the rate at which the chain disassembles to achieve a more controlled and sustained drug release.²² There are many triggers that can initiate the fragmentation of the polymer into its building blocks, depending on the molecule placed at the end of the polymer. Thus, it is possible to design a linear polymer based on a polyurethane backbone with a pH sensitive capping molecule as trigger. Dropping the pH would cleavage the carbamate linkage of the trigger, which would start the sequential 1–6 elimination and decarboxylation reactions yielding CO₂ and the starting monomer.

Combining both materials, MSNs and SIPs, in a single functional responsive nanocarrier would provide access to the benefits of both materials, *i.e.*, the high loading capacity of a variety of cargoes into MSNs, and the responsiveness of pH-sensitive SIPs to control the release. Decoration of the MSNs pore entrances with pH-responsive SIP gatekeepers that avoid premature release of the cargo and unblock the pores upon pH drops, represent a new and promising concept that is explored here for the first time (Scheme 2).

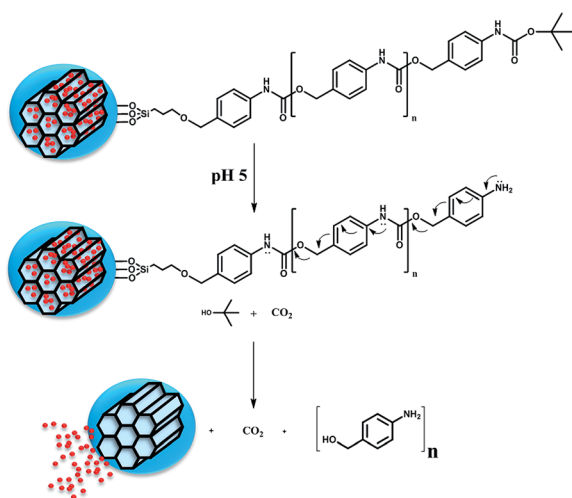
In this work we have developed a smart nanocarrier able to respond to changes in the pH for potential applications in cancer treatment. For this purpose, we have selected MSNs as carriers and a SIP with a trigger sensitive to acid pH to block the pore entrances and prevent premature release. The pH sensitivity of the nanosystem was evaluated *in vial* showing that cargo

release took place in acidic conditions, while at physiological pH there was almost no release. The biocompatibility of the system was evaluated with different cells, and internalisation of the nanoparticles was also demonstrated.

MSNs were synthesised *via* sol-gel method under basic conditions using hexadecyltrimethylammonium bromide (CTAB) as structure directing agent following a previously published method.²³ The sol-gel method allows a fine control on the final composition of the nanoparticles under mild conditions.^{24,25} The spherical morphology of the as-produced MSNs was confirmed through Scanning Electron Microscopy (SEM) and Transmission Electron Microscopy (TEM), as can be observed in Fig. 1a and b, with average sizes of *ca.* 150 nm (see ESI, Fig. S3† with Dynamic Light Scattering, DLS, studies).

Typical adsorption-desorption isotherms of MCM-41-like materials with cylindrical pores were observed (Fig. 1c), with pore diameters centred at *ca.* 2.7 nm. The characteristic hexagonal arrangement of the pores was confirmed through low angle X-Ray Diffraction (XRD) analyses (Fig. 1d), where typical well resolved maxima from well-ordered 2D-hexagonal structure with *p6mm* space group were observed.

A linear self-immolative polymer was produced to cover the pore entrances and avoid premature release of the cargo. We employed a polyurethane backbone with a BOC protecting group on the terminal amine that acted as a trigger. A pH drop would cleave the protecting group and initiate the disassembly of the polymer. The SIP was synthesised by a modification of a previously published method (Fig. 2a),¹⁹ producing the monomer **1** from 4-aminobenzyl alcohol and phenyl chloroformate. Then, the polymerisation process using DBTL as catalyst and **2** as end-group led to the polymeric chain **3** with an average of about 20 monomers (determined by NMR, through the ratio benzylic hydrogens at the molecule tail *vs.* those in the polymeric backbone).



Scheme 2 Schematic representation of the nanocarrier response to acid pH. Cleavage of the protecting group generates an intermediate that undergoes sequential 1,6-elimination and decarboxylation reactions to form the initial monomer, disassembling the immolative polymer and allowing the cargo release.

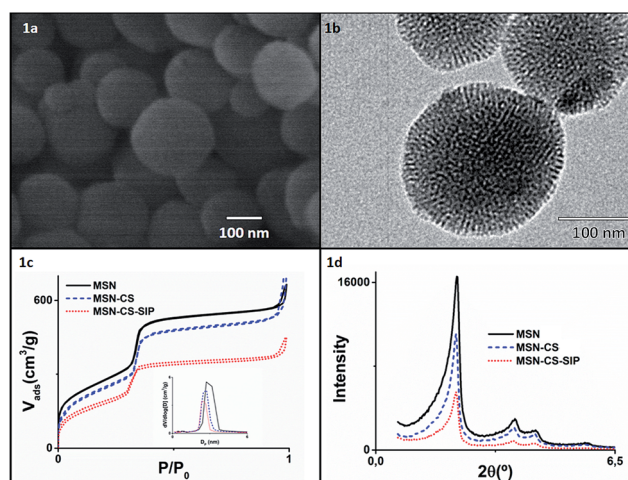


Fig. 1 (a) SEM images of MSNs; (b) TEM images of MSNs; (c) nitrogen adsorption and desorption isotherms of MSN (black solid line), MSN-CS (blue dashed line) and MSN-CS-SIP (red dotted line) (inset: pore size distribution determined by the BJH method from the desorption branch of the isotherm); (d) XRD patterns of MSN (black solid line), MSN-CS (blue dashed line) and MSN-CS-SIP (red dotted line).



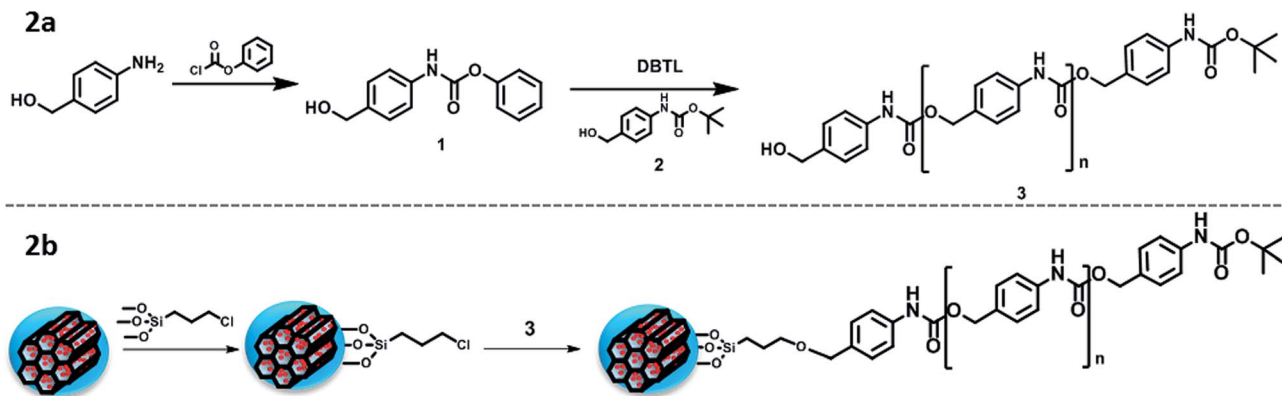


Fig. 2 (a) Schematic representation of the SIP synthesis; (b) schematic representation of the SIP grafting to the MSNs.

The self-immolative behaviour of the SIP was confirmed by treating the polymer with trifluoroacetic acid in CH_2Cl_2 , which are the standard conditions for removing BOC protecting groups. After the acid treatment, all characteristic NMR peaks of SIP had disappeared (Fig. S8†), as expected.

The successful attachment of the SIP to the MSNs was achieved through a previous modification of the nanoparticles with an alkoxy silane bearing chloride groups at the other end (denoted CS) (Fig. 2b). Those modified-particles (MSN-CS) were characterised through thermogravimetric (TG) analysis (Fig. S1†), where the increase of weight loss confirmed the presence of organic matter, together with elemental analysis (Table 1), where the increment of C% verifies the linker grafting. The C–H stretching bands from Fourier transform infrared spectroscopy (FTIR, Fig. S2†) also confirmed the presence of the linker.

Addition of polymer 3, previously activated with DIPEA, to the MSN-CS led to SIP coating (MSN-CS-SIP). XRD analyses after functionalisation (Fig. 1d) showed that the ordered meso-structure survived the process, although the progressive reduction of the intensity might be ascribed to the partial filling of the pores by some polymer chains, as it was previously observed when functionalising MSNs with large dendrimers.²⁶ The successful SIP coating onto MSNs was confirmed through different characterisation techniques. The great weight loss observed in TG analyses (Fig. S1†) is indicative of the great amount of SIP present. FTIR spectrum of MSN-CS-SIP (Fig. S2†) exhibits additional bands to those typical from silica materials

at $\text{ca. } 1650 \text{ cm}^{-1}$, typical from C–O stretching vibrations from carbamates, verifying the presence of the polymer. This is in agreement with the data obtained from N_2 analyses (Table 1), in which the reduction of the surface area and pore volume is indicative of the presence of the polymer blocking the pore entrances. Finally, the increase of N% in the elemental analysis in MSN-CS-SIP confirms the presence of the polymer.

Finally, the changes in the zeta potential to less negative values confirm the correct functionalisation. However, due to the small molecular weight of SIP the size was not affected by the functionalisation, either at pH 7.4 or pH 5.

After confirmation of the polymer grafting on the surface of MSN, drug delivery capabilities of the system were evaluated. Fluorescent cargo, tris(2,2'-bipyridine)dichloro ruthenium(II) denoted as Ru, was loaded into the network of cavities of the mesoporous nanoparticles overnight before the blocking the pore entrances with the SIP polymer (for further details see ESI†). The successful Ru loading was confirmed by the decrease in the pore volume measured through N_2 adsorption (data not shown).

The responsiveness to pH of the system here developed was evaluated through an *in vial* Ru release from sample MSN-CS-SIP at 37 °C using Transwells at different pHs, *i.e.* 7.4 and 5. The data from the release kinetics, Fig. 3, showed that there was almost no cargo release at physiological pH, which means that the nanocarrier would travel through the blood stream without premature release of the payload. However, when the pH of the solution dropped to pH 5, there was a fast release of the loaded Ru, as it can be observed in Fig. 3. This behaviour confirmed that the polymer was disassembling as a consequence of the acid pH, opening the pore entrances and favouring the release of their cargo only at acid environments.

Moreover, after being exposed to acid pH, the zeta potential of MSN-CS-SIP increased from -23.8 mV to -10.7 mV . That would confirm that the self-immolation actually took place, as its residue is a molecule of 4-aminobenzyl alcohol.

Once the MSNs covered with the SIPs nanocarrier were fully characterised and their pH-sensitive release capabilities demonstrated *in vial*, the next step was analysing their behaviour with cells. The *in vitro* cytotoxicity study was determined by the exposition of MC3T3-E1 pre-osteoblastic cells to different

Table 1 Main properties of the materials produced in this work

Material	MSN	MSN-CS	MSN-CS-SIP
Organic matter by TG (%)	5.3	14.4	20
Elemental analysis (%)	C: 3.95	C: 9.34	C: 10.76
	H: 2.09	H: 2.34	H: 2.46
	N: 0.06	N: 0.11	N: 0.53
Surface area ($\text{m}^2 \text{ g}^{-1}$)	1053	914	671
Pore volume ($\text{cm}^3 \text{ g}^{-1}$)	1.05	1.02	0.60
Pore width (nm)	2.74	2.66	2.19
Size (nm); PDI	190; 0.467	190; 0.467	190; 0.358
Zeta potential (mV)	-35.3	-29.0	-23.8



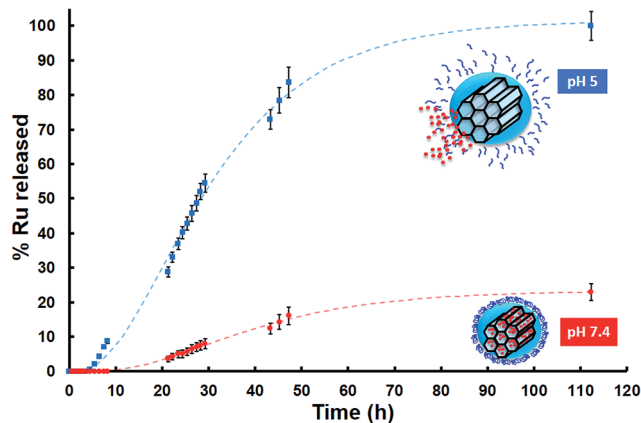


Fig. 3 *In vitro* cumulative Ru release (in percentage) vs. time from MSN-CS-SIP at physiological and acid pHs (repeated 3 times each measurement; Standard Error of the Mean (SEM)).

amounts of nanoparticles (50 , 100 , 200 and $300 \mu\text{g mL}^{-1}$). Fig. 4 shows that none of the studied concentrations, except $300 \mu\text{g mL}^{-1}$, induced significant cytotoxicity measured by an Alamar Blue assay (cell viability was $>99\%$ that of the control). Similar results were obtained in LNCaP tumoral cells (Fig. S9†).

Cellular internalisation of the nanoparticles was observable by fluorescence microscopy and analysed by flow cytometry in LNCaP cells in contact with the nanoparticles ($100 \mu\text{g mL}^{-1}$) for 2 hours. No fluorescence was observed in the end slices corresponding to the external cellular surfaces, suggesting that the nanoparticles did not adsorb on the cell membranes. Therefore,

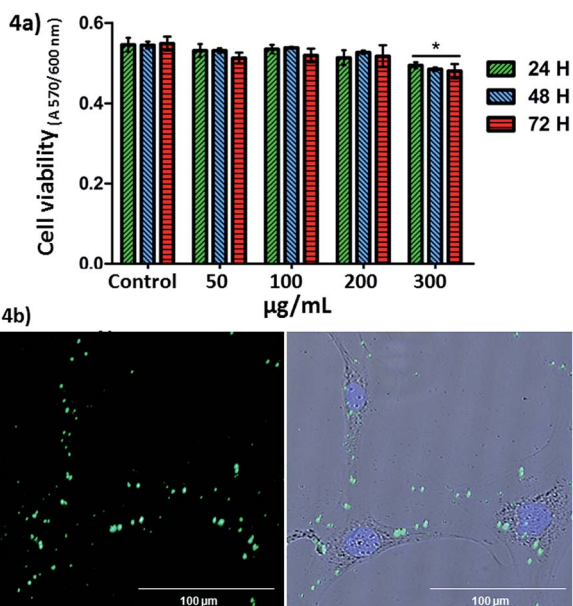


Fig. 4 (a) MC3T3-E1 cell viability in contact with different concentrations of nanoparticles at 24, 48 and 72 h of cell culture. Similar proliferation results were obtained in LNCaP cells. $*p < 0.05$ vs. control without nanoparticles (student's *t*-test). (b) Fluorescence microscopy images of LNCaP cells incubated with nanoparticles at 2 hours of cell culture. From left to right: green fluorescence (nanoparticles with fluorescein) and overlay image with blue fluorescence (nuclei).

we observed that the nanoparticles at $100 \mu\text{g mL}^{-1}$ were internalised by LNCaP cells. The cellular uptake observed by fluorescence microscopy was also confirmed by flow cytometry ($19.25 \pm 1.5\%$ of nanoparticles incorporation) in the same cells, at 2 hours. The low cellular internalisation would be in agreement with the fact that positively charged²⁷ or targeted²⁸ nano-carriers show better cellular internalization than those negative, as in the case of MSN-CS-SIP.

Conclusions

In conclusion, we have developed for the first time mesoporous silica nanoparticles capped with acid responsive self-immolative polymers. The cargo release can be triggered by acid pH, typical from tumours microenvironments. The biocompatibility of this new delivery nanoplatform has been demonstrated with different cell types, and the effective nanoparticle internalization has been achieved in tumoral cells. We envision that the synergy from MSNs carriers and SIPs could be of a great interest for future applications in nanomedicine to fight against cancer.

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Notes and references

- O. C. Farokhzad and R. Langer, *ACS Nano*, 2009, **3**, 16–20.
- J. D. Byrne, T. Betancourt and L. Brannon-Peppas, *Adv. Drug Delivery Rev.*, 2008, **60**, 1615–1626.
- S. Mura, J. Nicolas and P. Couvreur, *Nat. Mater.*, 2013, **12**, 991–1003.
- R. Misra, S. Acharya and S. K. Sahoo, *Drug Discovery Today*, 2010, **15**, 842–850.
- A. Baeza, M. Manzano, M. Colilla and M. Vallet-Regí, *Biomater. Sci.*, 2016, **4**, 803–813.
- F. Tang, L. Li and D. Chen, *Adv. Mater.*, 2012, **24**, 1504–1534.
- Z. Li, J. C. Barnes, A. Bosoy, J. F. Stoddart and J. I. Zink, *Chem. Soc. Rev.*, 2012, **41**, 2590–2605.
- C. E. Ashley, E. C. Carnes, G. K. Phillips, D. Padilla, P. N. Durfee, P. A. Brown, T. N. Hanna, J. Liu, B. Phillips, M. B. Carter, N. J. Carroll, X. Jiang, D. R. Dunphy, C. L. Willman, D. N. Petsev, D. G. Evans, A. N. Parikh, B. Chackerian, W. Wharton, D. S. Peabody and C. J. Brinker, *Nat. Mater.*, 2011, **10**, 389–397.
- I. I. Slowing, J. L. Vivero-Escoto, B. G. Trewyn and V. S.-Y. Lin, *J. Mater. Chem.*, 2010, **20**, 7924–7937.
- M. Manzano and M. Vallet-Regí, *J. Mater. Chem.*, 2010, **20**, 5593–5604.
- B. G. Trewyn, I. I. Slowing, S. Giri, H. Chen and V. S. Lin, *Acc. Chem. Res.*, 2007, **40**, 846–853.
- J. L. Vivero-Escoto, I. I. Slowing, B. G. Trewyn and V. S.-Y. Lin, *Small*, 2010, **6**, 1952–1967.



- 13 C. Argyo, V. Weiss, C. Bräuchle and T. Bein, *Chem. Mater.*, 2014, **26**, 435–451.
- 14 Y. Zhang, Z. Zhi, T. Jiang, J. Zhang, Z. Wang and S. Wang, *J. Controlled Release*, 2010, **145**, 257–263.
- 15 A. Baeza, M. Colilla and M. Vallet-Regí, *Expert Opin. Drug Delivery*, 2015, **12**, 319–337.
- 16 T. Kolter and K. Sandhoff, *FEBS Lett.*, 2010, **584**, 1700–1712.
- 17 Z. Li, H. Li, L. Liu, X. You, C. Zhang and Y. Wang, *RSC Adv.*, 2015, **5**, 77097–77105.
- 18 Y. Yan, J. Fu, X. Liu, T. Wang and X. Lu, *RSC Adv.*, 2015, **5**, 30640–30646.
- 19 A. Sagi, R. Weinstein, N. Karton and D. Shabat, *J. Am. Chem. Soc.*, 2008, **8**, 5434–5435.
- 20 W. Wang and C. Alexander, *Angew. Chem., Int. Ed.*, 2008, **47**, 7804–7806.
- 21 S. Gnaim and D. Shabat, *Acc. Chem. Res.*, 2014, **47**, 2970–2984.
- 22 R. A. McBride and E. R. Gillies, *Macromolecules*, 2013, **46**, 5157–5166.
- 23 A. Baeza, E. Guisasola, A. Torres-Pardo, J. M. González-Calbet, G. J. Melen, M. Ramirez and M. Vallet-Regí, *Adv. Funct. Mater.*, 2014, **24**, 4625–4633.
- 24 A. S. Timin, A. R. Muslimov, K. V. Lepik, N. N. Saprykina, V. S. Sergeev, B. V. Afanasyev, A. D. Vilesov and G. B. Sukhorukov, *J. Mater. Chem. B*, 2016, **18**, 33–37.
- 25 A. S. Timin, H. Gao, D. V. Voronin, D. A. Gorin and G. B. Sukhorukov, *Adv. Mater. Interfaces*, 2016, DOI: 10.1002/admi.201600338.
- 26 B. Gonzalez, M. Colilla, C. L. de Laorden and M. Vallet-Regí, *J. Mater. Chem.*, 2009, **19**, 9012–9024.
- 27 A. Verma and F. Stellacci, *Small*, 2010, **6**, 12–21.
- 28 K. Ulbrich, K. Holá, V. Šubr, A. Bakandritsos, J. Tuček and R. Zbořil, *Chem. Rev.*, 2016, **116**, 5338–5431.

