

CrossMark
click for updatesCite this: *RSC Adv.*, 2017, 7, 7718

Morphology evolution of Ag/Au nanocomposites via temperature-controlled galvanic exchange to enhance catalytic activity†

Tae-Hyeon Park, Hyeri Lee, Jaewon Lee and Du-Jeon Jang*

Morphology-controlled Ag/Au nanocomposites have been fabricated facilely via a modified galvanic replacement reaction using Ag nanocubes as sacrificial templates. The structures of the Ag/Au nanocomposites produced at the galvanic reaction temperatures of 0, 25, 55, 85, and 105 °C were found to be Au-decorated Ag nanocubes, well-defined nanoboxes, truncated nanoboxes, porous nanoboxes, and broken nanoboxes, respectively. Compared to pristine Ag nanocubes or Ag/Au nanocomposites without cavities, Ag/Au nanocatalysts with hollow interiors have been found to catalyze the reduction of rhodamine B much more rapidly in the presence of KBH_4 . In particular, Ag/Au nanocomposites fabricated at 85 °C show the most efficient catalytic performance as they have the largest nanocavities, which are surrounded by the most porous walls. The high enhancement of the catalytic performance is attributed to the facilitation of rapid electron relay from BH_4^- to rhodamine B via the catalytic surface, consequently lowering the kinetic barrier of the catalytic reaction. Overall, the temperature of the galvanic replacement reaction has been varied to optimize the morphology and the subsequent catalytic performance of Ag/Au nanocomposites.

Received 3rd November 2016
Accepted 11th January 2017

DOI: 10.1039/c6ra26249j

www.rsc.org/advances

Introduction

Metallic nanostructured materials have received a great deal of research effort owing to their outstanding physical and chemical properties depending on size, shape, and composition.^{1–3} In particular, there has been a growing interest in nanostructures comprising noble metals due to their unusual physicochemical characteristics which are considerably distinct from their bulk-sized analogues.^{4,5} In this respect, the fabrication of nanometer-sized noble metals has led to intriguing applications in various fields such as photography, spectroscopy, optoelectronics, sensing, biological labeling, and catalysis.^{6–8} Bimetallic nanostructures consisting of noble metals such as silver, gold, palladium and platinum have received special attention because their unique optical, electronic, and catalytic properties, which lead to a broad range of applications, including localized surface-plasmon resonance, surface-enhanced Raman scattering, biosensing, and catalysis.^{9,10} For example, by combining Ag and Au into a single entity, catalytic properties can be further improved and localized surface-plasmon absorption can be varied continuously to the limits of the corresponding monometallic particles.^{11,12} Diverse approaches to synthesize the bimetallic nanomaterials have been studied: the

simultaneous chemical reduction of mixed metal ions, the electrochemical reduction, and the galvanic replacement reaction on the surface of sacrificial nanoparticles.^{13,14}

The galvanic replacement reaction is basically a redox process between two metals with distinct reduction potentials. Oxidation takes place in one metal with a low reduction potential which is referred to as a sacrificial template, and reduction coincides with the deposition of the other metal with a high reduction potential.^{15–17} This simple process is controlled solely by a favorable difference between the reduction potentials of two metals. The galvanic exchange provides a simple and flexible route to engineering a range of advanced multifunctional nanostructures such as hollow cavities and porous walls.^{14,18,19} Among various noble metal-based nanostructures, hollow architectures are a particularly interesting class of materials that have unusual chemical and physical properties compared to the respective ones of solid counterparts.^{20–22} The galvanic replacement process has been employed as a useful method for preparing hollow metallic composite nanostructures including spherical nanoshells, nanodendrites, nanoboxes, nanocages, nanotubes, and nanoframes consisting of Au, Pd, and/or Pt by using Ag, Cu, or more reactive metals as sacrificial templates.^{23–25} These hollow metallic composite nanostructures have a great deal of promise for diverse fields including photothermal therapy, biomedical imaging, drug delivery, and electrochemical catalysis. In particular, catalytic performances can be optimized by controlling surface compositions, sizes, and morphologies carefully.^{14,25,26} Hollow-

Department of Chemistry, Seoul National University, NS60, Seoul 08826, Korea.
E-mail: djjang@snu.ac.kr

† Electronic supplementary information (ESI) available: Supplementary figures (Fig. S1–S10). See DOI: 10.1039/c6ra26249j



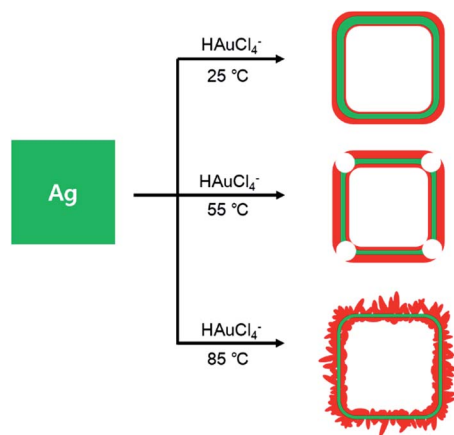


Fig. 1 Illustration for the temperature-controlled morphology evolution of Ag/Au nanocomposites.

structured Ag/Au nanocomposites exhibit high catalytic activity as they have high surface-to-volume ratios, low densities, and two different types of surfaces, which are the interior and the exterior walls of hollow nanostructures.

Herein, we have controlled the morphologies of Ag/Au nanocomposites *via* modified galvanic replacement reactions involving Ag nanocubes as sacrificial templates; the variation of reaction temperatures has resulted in the nanostructure modification of Ag/Au nanocomposites. Whereas Ag/Au nanocomposites prepared at 25 and 55 °C look like nanoboxes and truncated nanoboxes, respectively, Ag/Au nanocomposites fabricated at 85 °C resemble porous nanoboxes, as shown in Fig. 1. The catalytic activities of Ag/Au nanocomposites prepared at various reaction temperatures have been investigated by measuring the catalytic reduction rate constants and other kinetic parameters of rhodamine B (RhB) in the presence of BH_4^- . It has been found that the catalytic rate constant *via* Ag/Au nanocomposites prepared at 85 °C is higher by factors of 2.2, 1.5, and 100 than the rate constants of Ag/Au nanocomposites prepared at 25, 55, and 105 °C, respectively. This indicates that we can control the morphologies of Ag/Au nanocomposites to enhance their catalytic performances readily by adjusting the temperatures of galvanic replacement reactions.

Experimental section

Materials

Chemicals were used as purchased without further purification: $\text{HAuCl}_4 \cdot 3\text{H}_2\text{O}$ (s, $\geq 99.9\%$), CF_3COOAg (s, $\geq 99.9\%$), polyvinylpyrrolidone (PVP, s, MW = 55 000), $\text{Na}_2\text{S} \cdot 9\text{H}_2\text{O}$ (s, $\geq 99.9\%$), 37% HCl(aq) , and KBH_4 (s, $\geq 95\%$) from Sigma-Aldrich; ethylene glycol (EG, l, 99%) from Daejung Chemicals; rhodamine B (RhB, s) from Wako Chemicals. Deionized water ($>15 \text{ M}\Omega \text{ cm}$) from an Elga PURELAB Option-S system was used throughout the experiments.

Synthesis of Ag nanocubes

Pristine Ag nanocubes (Ag(pr)) were synthesized *via* a sulfide-mediated polyol process, with EG as the solvent and

CF_3COOAg as a precursor of elemental silver.^{21,22,27} In a typical synthesis, 20 mL of EG was added into a 100 mL round-bottom flask and heated in an oil bath at 160 °C for 30 min under magnetic stirring to remove residual water. 0.24 mL of 3.0 mM $\text{Na}_2\text{S} \cdot 9\text{H}_2\text{O(EG)}$ was then injected quickly into the preheated EG. After 2 min, 2.0 mL of 3.0 mM HCl(EG) and 5.0 mL of 20 mg mL^{-1} PVP(EG) were quickly injected into the mixture. After additional 2 min, 1.6 mL of 282 mM $\text{CF}_3\text{COOAg(EG)}$ was added into the mixture, which was then reacted for 20 min at 160 °C. During the entire reaction, the flask was capped with a rubber stopper except for the injection of reagents. The reaction mixture was quenched by placing the flask in an ice-water bath and added with acetone. The produced colloid was centrifuged, rinsed once with acetone and three times with water to remove remaining EG and PVP, and re-dispersed in water, yielding an aqueous colloidal solution of Ag(pr).

Preparation of Ag/Au nanocomposites

Ag/Au nanocomposites were prepared *via* galvanic replacement reactions between Ag(pr) and HAuCl_4 in water. 0.30 mL of the Ag(pr) colloid was dispersed in 5.0 mL of 1.0 mg mL^{-1} PVP(aq) and heated to a specific reaction temperature. 1.0 mL of 0.50 mM $\text{HAuCl}_4(\text{aq})$ was then added into the mixture, which was maintained for 20 min at the temperature. The produced Ag/Au nanocomposites were centrifuged, rinsed with water three times, and then dispersed in water. To adjust the morphology of Ag/Au nanocomposites, the temperatures of galvanic replacement reactions were varied from 0 to 105 °C; hereafter, Ag/Au nanocomposites prepared at 0, 25, 55, 85, and 105 °C will be described as Ag/Au(0), Ag/Au(25), Ag/Au(55), Ag/Au(85), and Ag/Au(105), respectively.

Characterization

The samples for transmission electron microscopy (TEM) were prepared by dropping the aqueous colloidal solution of nanocatalysts onto a carbon-coated Cu grid. Whereas TEM images were taken using a Hitachi H-7600 microscope operated at 80 kV, energy-dispersive X-ray (EDX) elemental profiles and maps, as well as scanning transmission electron microscopy (STEM) images, were measured using a JEOL JEM-2100F microscope operated at 200 kV. UV-visible absorption spectra were recorded using a Scinco S-3100 spectrometer with a temperature controller.

Measurement of catalytic properties

The catalytic performances of Ag/Au nanocomposites were investigated by measuring the reduction reaction of RhB in $\text{BH}_4^-(\text{aq})$. 0.10 mL of a nanocatalyst colloid was added to 0.90 mL of water and 1.7 mL of 20 μM RhB(aq) contained in a quartz cuvette having a path length of 10 mm. The absorption changes of RhB(aq) were confirmed using the above described temperature-controllable spectrometer at scheduled intervals after addition of 0.40 mL of 10 mM $\text{KBH}_4(\text{aq})$. Note that the concentrations of Ag and Au, calculated based on the respective precursor amounts, in a cuvette of catalysis experiments were 20 and 8.1 μM , respectively. The catalytic activity was also



investigated by measuring the reduction reaction of 4-nitrophenol (4-NP) into 4-aminophenol (4-AP); the experimental procedure was the same except that 1.7 mL of 20 μM RhB(aq), 0.90 mL of water, and 0.40 mL of 10 mM $\text{KBH}_4(\text{aq})$ were replaced by 50 μL of 2.0 mM 4-NP ($s, \geq 99\%$, purchased from Sigma-Aldrich), 2.25 mL of water, and 0.6 mL of 10 mM $\text{KBH}_4(\text{aq})$. In order to investigate the reusability of the nanocatalysts, all the experimental parameters were kept constant. The used nanocatalysts were separated from the mixture solution by centrifugation at 13 000 rpm, rinsed with water, and redispersed into water for the next cycle of catalysis. Then, the same processes were repeated three times under the same conditions.

Results and discussion

TEM characterization

Fig. 2 and S1† indicate that the evolution of different morphologies of Ag/Au nanocomposites has been induced by varying the temperatures of Ag templates-engaged galvanic replacement reactions. Truncated nanocubes of Ag(pr) (Fig. 2a) with an average edge length of 43.1 ± 5.1 nm (Fig. S2a†) were prepared for sacrificial nanotemplates *via* a typical sulfide-mediated polyol procedure.^{21,22,27} As Ag(pr) was mixed with $\text{HAuCl}_4(\text{aq})$, the galvanic replacement reaction took place immediately at the sites of the highest surface energy including defects, steps, or stacking faults.¹⁸ Thus, Ag atoms were oxidized to dissolve into the aqueous solution, producing small holes on the surfaces of nanocubes. Concurrently, Au ions were reduced to become Au atoms. Since Ag and Au have the same face-centered cubic structure and similar lattice constants, Au atoms are prone to be epitaxially deposited on Ag surfaces, forming an incomplete thin layer.^{17,18} Small holes formed in the Au layer act as primary sites for the continuous dissolution of Ag. As a result, Ag(pr) templates were partially dissolved to produce a cavity inside each nanoparticle. The temperature at which the galvanic replacement reaction takes place plays an important role in the formation of Ag/Au nanocomposites since

the rate of the galvanic replacement reaction increases exponentially with temperature. Fig. S2† shows that the observed mean edge lengths of Ag/Au(0), Ag/Au(25), Ag/Au(55), Ag/Au(85), and Ag/Au(105) are 48.7 ± 10.8 , 46.6 ± 4.8 , 46.9 ± 5.3 , 49.8 ± 7.2 , and 47.5 ± 7.1 nm, respectively, supporting that reduced Au atoms have been deposited on the exterior surfaces of Ag nanotemplates. Fig. 2 and S1† indicate that the structures of Ag/Au(0), Ag/Au(25), Ag/Au(55), Ag/Au(85), and Ag/Au(105) are Au-decorated Ag nanocubes, well-defined nanoboxes, truncated nanoboxes, and porous nanoboxes, and amorphous nanocomposites, respectively. The hollow interiors of Ag/Au nanocomposites have been found in Ag/Au(25), Ag/Au(55), and Ag/Au(85); the average cavity sizes of Ag/Au(25), Ag/Au(55), and Ag/Au(85) have been estimated as 29.8, 33.4, and 39.3 nm, respectively. The roughness and porosity of nanoboxes also increase with the increase of the reaction temperature. Thus, nanoboxes of Ag/Au(85) have the largest cavities surrounded by the roughest and the most porous walls among our prepared Ag/Au nanocomposites. Note that nanobox structures having hollow interiors were not formed when the temperature was too low as 0 °C or too high as 105 °C. Overall, Fig. 2, S1, and S2† indicate the morphologies of Ag/Au nanocomposites have been controlled facilely by adjusting the temperatures of galvanic replacement reactions.

EDX and STEM analysis

The EDX line-scanned elemental profiles have been explored to confirm the elemental distributions and exact structures of Ag/Au nanocomposites. Fig. 3a reveals that Ag(pr) is a solid Ag nanocube with an edge size of 45.3 nm. Meanwhile, the line-scanned EDX elemental profiles of Ag/Au(25) (Fig. 3b), Ag/Au(55) (Fig. 3c), and Ag/Au(85) (Fig. 3d) show that two characteristic sharp peaks of Au are similar to the respective ones of Ag, indicating that the walls of a nanobox compositely consist of Ag and Au. The elemental profiles also exhibit that the Ag edge sizes of Ag/Au(0) (Fig. S3†), Ag/Au(25), Ag/Au(55), and Ag/Au(85) are 31.2, 48.5, 45.8, and 49.1 nm, respectively, whereas the Au edge sizes of Ag/Au(0), Ag/Au(25), Ag/Au(55), and Ag/Au(85) are 47.1, 50.1, 46.1, and 49.2 nm, respectively. Thus, the Au edge size is larger than the Ag edge size in each examined Ag/Au nanocomposite, indicating that reduced Au atoms deposit on the exterior surfaces of Ag nanotemplates preferentially. As the reaction temperature increases, the mobility of atoms can be promoted rapidly by heat so that atomic exchange between Ag cores and Au shells takes place rapidly to form Ag/Au nanoalloys gradually. The STEM images and EDX elemental maps of Fig. 3 and S3† also support that Ag/Au(0), Ag/Au(25), Ag/Au(55), and Ag/Au(85) have nanostructures of Au-decorated Ag nanocubes, well-defined nanoboxes, truncated nanoboxes, and porous nanoboxes.

Absorption spectra

Variation in the surface-plasmon resonance (SPR) spectra of Ag(pr). Ag/Au(25), Ag/Au(55), and Ag/Au(85) depicted in Fig. 4 agrees well with the variation of morphologies, sizes, and compositions observed Fig. 2 and 3. The SPR spectrum of Ag(pr)

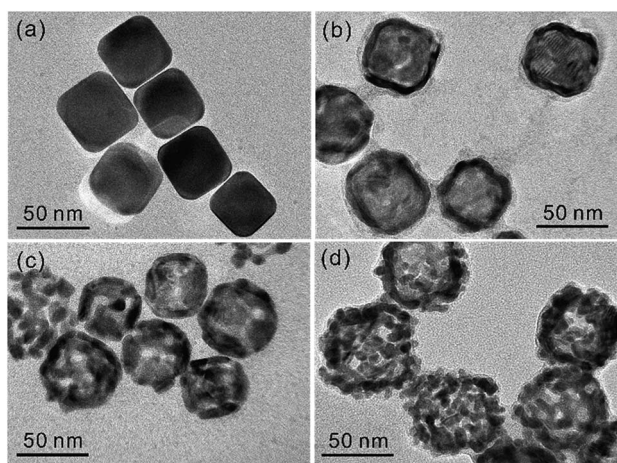


Fig. 2 TEM images of (a) Ag(pr), (b) Ag/Au(25), (c) Ag/Au(55) and (d) Ag/Au(85).



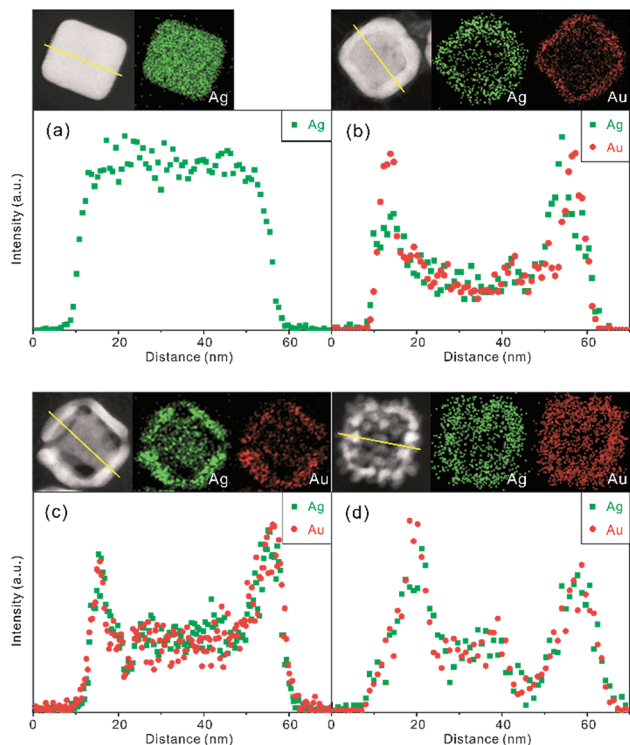


Fig. 3 STEM images, EDX elemental maps and area normalized line-scanned profiles of (a) Ag(pr), (b) Ag/Au(25), (c) Ag/Au(55) and (d) Ag/Au(85).

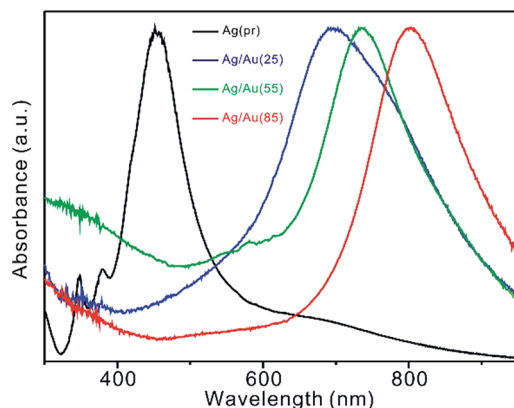


Fig. 4 Maximum-normalized surface-plasmon resonance spectra of indicated nanocatalysts in water.

shows three characteristic peaks of Ag nanocubes with an average edge size of 43.1 ± 5.1 nm, including the major SPR peak at 456 ± 33 nm. Meanwhile, the SPR spectra of Ag/Au(25), Ag/Au(55), and Ag/Au(85) have the absorption maxima at 693 ± 102 , 735 ± 62 , and 802 ± 77 nm, respectively, indicating that the average length-to-thickness value for the walls of Ag/Au nanoboxes increases with the increase of the reaction temperature.^{28–30} As shown in Fig. S1, S2, and S3,[†] the SPR spectra of Fig. S4[†] also suggest that Au nanoplates and elliptical nanoparticles have been deposited on the surfaces of Ag nanotemplates for Ag/Au(0) while Ag/Au broken nanoboxes

consisting of numerous thin nanoplates and nanodiscs with diverse sizes have been produced for Ag/Au(105).

Catalytic performances

The catalytic performances of Ag/Au nanocomposites have been evaluated by measuring the time-dependent absorption spectra of RhB reduced catalytically by as-prepared Ag/Au nanocomposites in the presence of BH_4^- as shown in Fig. 5 and S5,[†] indicating that the catalytic activity of Ag/Au nanocomposites depends highly on their morphologies,^{26,30,31} which have been varied by adjusting the temperature of the galvanic replacement reaction. RhB, which consists of a central xanthene ring connected to four *N*-ethyl groups, two on each side, exhibits high resistance to photodegradation.^{32,33} Thus, the rapid reduction of RhB is important in regard to the purification of dye effluents. The residual concentration of RhB reduced catalytically *via* Ag/Au nanocomposites in the presence of KBH_4 at a specific time has been estimated by monitoring the optical density of RhB at 554 nm. Fig. 5 and S5[†] show that Ag/Au nanocomposites have much higher catalytic activities than Ag(pr); whereas only 2% has been reduced in 40 min in the presence of Ag(pr), 70, 35, 49, 97 and 9% of RhB have been reduced within 40 min in the presence of Ag/Au(0), Ag/Au(25), Ag/Au(55), Ag/Au(85), and Ag/Au(105), respectively. This also indicates that the catalytic activity is dependent strongly on the structures of Ag/Au nanocomposites. In particular, Ag/Au(85) having the largest cavities surrounded by the most porous walls has shown the most efficient catalytic performance.

Fig. 6 and S6[†] show that the time-dependent pseudo-first-order reduction profiles of RhB catalyzed by Ag(pr), Ag/Au(25), Ag/Au(55), and Ag/Au(85) in the presence of KBH_4 have been used to elicit observed reduction constants (k_{obs}); the plots of $\ln(A/A_0) = -k_{\text{obs}}t$,^{20,33} where A and A_0 are the optical densities of RhB at 554 nm at times t and 0, respectively, have been used to extract k_{obs} presented in Table 1. The rate constant of a Ag/Au nanocomposite catalyst (k) has been extracted by deducting

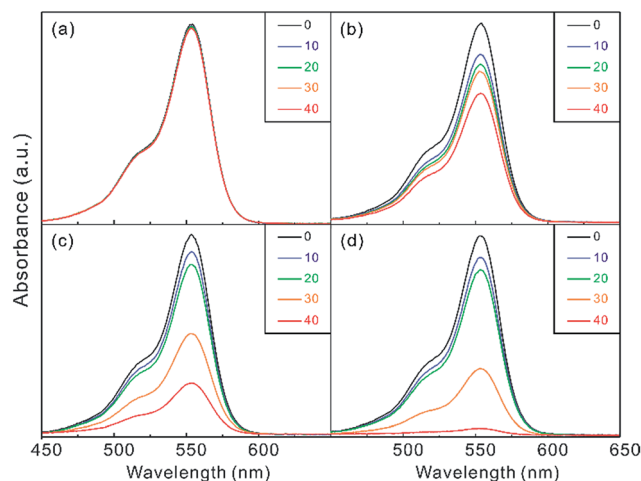


Fig. 5 Absorption spectra at 25 °C of 11 μM RhB(aq) in the presence of 1.3 mM KBH_4 , monitored at indicated times in the units of min after addition of (a) Ag(pr), (b) Ag/Au(25), (c) Ag/Au(55) and (d) Ag/Au(85).



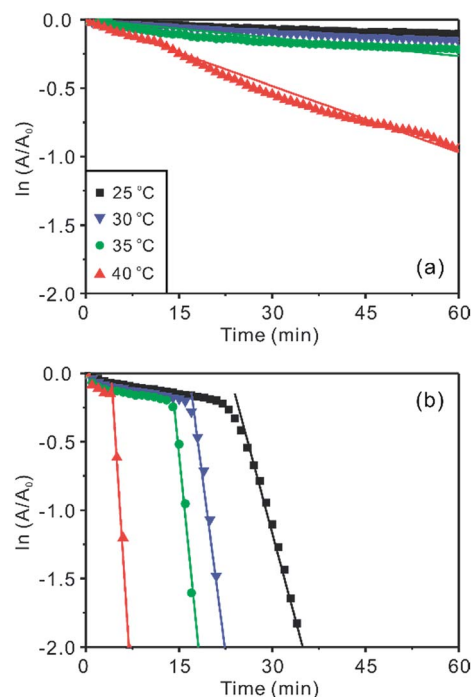


Fig. 6 $\ln(A/A_0)$ vs. t for the catalytic reduction of $11 \mu\text{M}$ RhB(aq) via (a) Ag(pr) (b) Ag/Au(85) in the presence of 1.3 mM KBH_4 at indicated temperatures.

the reduction rate constant of RhB with no nanocatalysts (k_0) from k_{obs} ; $k = k_{\text{obs}} - k_0$.^{20,34} The catalytic rate constants of nanocatalysts conducted at four different reaction temperatures indicate that the catalytic performances of nanocatalysts increase rapidly with increment of temperature. Fig. 6 and Table 1 exhibit that a certain period of time was required for a certain Ag/Au nanocatalyst to start catalysis. As observed in a number of systems,^{8,35–37} this period, defined as the induction time (t_0), has been attributed to a slow diffusion of reactants onto the catalytic surfaces. Table 1 indicates that the induction time is necessary for Ag/Au nanoboxes with hollow interiors, suggesting that the fast catalytic reduction of RhB molecules takes place in the cavities of Ag/Au nanoboxes and that it takes t_0 for RhB to diffuse through the walls of the nanoboxes. Table 1 also suggests that t_0 decreases gradually as the walls of the nanoboxes become more porous. Meanwhile, none of Ag(pr), Ag/Au(0), and Ag/Au(105) show the induction time because none

of them have hollow interiors. In addition, the reduction kinetic data of 4-NP into 4-AP in Fig. S7† also demonstrate clearly that the catalytic activity of Ag/Au(85) ($k = 0.10 \text{ min}^{-1}$) is much higher than that of Ag(pr) ($k = 0.0013 \text{ min}^{-1}$). The catalytic reduction kinetic data and the induction times of 4-NP are almost consistent with the respective ones of RhB.

Fig. 7a shows that Arrhenius plots have been used to extract the activation energies (E_a) and frequency factors (A) of Ag(pr) and Ag/Au nanocatalysts. Table 1 indicates that Ag/Au nanocatalysts with hollow interiors have much smaller E_a and A values than nanocatalysts without cavities. As already discussed with Fig. 6, this observation also suggests that the catalysis reaction taking place in cavities is much faster than the reaction occurring at exterior surfaces as the confinement effect reduces the energy barrier. A close examination of Table 1 reveals that among Ag/Au nanoboxes having hollow structures, in particular, Ag/Au(85) has the smallest E_a value to show the highest catalytic activity because the Ag/Au(85) nanocatalyst has the largest nanocavity surrounded by the most porous walls. In addition, Fig. S8† also indicates that Ag/Au nanocomposites have the different amount of Au atoms as the galvanic replacement has taken place at diverse temperatures, implying that with the increase of temperature, the more Au atoms have been reduced and deposited onto the surface of Ag(pr). Thus, the active sites to induce the catalytic reaction on Ag/Au nanocomposites are quite different depending on the molar fraction of Au. With the increase of Au atoms, the catalytic efficiency was steadily enhanced²¹ since the Au has been known well to have better catalytic activity than Ag.^{38,39} Overall, the lowest activation energy for the catalytic reaction of Ag/Au(85) nanocomposites is attributed to not only the largest amount of Au atoms but also the largest nanocavity, which finally contribute to the higher catalytic activity of the nanocomposites. Furthermore, the activation enthalpies (ΔH^\ddagger) and activation entropies (ΔS^\ddagger) listed in Table 1 have been elicited from Eyring plots in Fig. S6†: $\ln(k/T)$ vs. $1/T$ to exhibit $\ln(k/T) = -\Delta H^\ddagger/(RT) + \Delta S^\ddagger/R + \ln(k_B/h)$.^{34,40} Table 1 shows that not only ΔH^\ddagger but also ΔS^\ddagger are smallest for Ag/Au(85), suggesting that activated-complex formation for the catalytic reduction reaction of RhB via Ag/Au(85) is most favorable in energy but most unfavorable in degree of freedom. This suggests that the most efficient catalytic performance and the smallest E_a value of Ag/Au(85) can be attributed to the energetically most favorable formation of the activated complex within the cavity of a Ag/Au(85) nanobox. Overall, Fig. 7a and

Table 1 Observed rate constants (k_{obs}) at 25°C , induction time (t_0) at 25°C , activation energies (E_a), frequency factors (A), activation enthalpies (ΔH^\ddagger), and activation entropies (ΔS^\ddagger) for the catalytic reduction of RhB catalyzed by nanocomposites in KBH_4 (aq)

Catalyst	d_c^a (nm)	k_{obs}^b (min^{-1})	t_0 (min)	E_a (kcal mol^{-1})	A (min^{-1})	ΔH^\ddagger (kcal mol^{-1})	ΔS^\ddagger ($\text{cal mol}^{-1} \text{ K}^{-1}$)
Ag(pr)	— ^c	0.0014	0	34.4	3.1×10^{21}	33.8	29.6
Ag/Au(0)	—	0.035	0	28.0	1.2×10^{19}	27.4	18.5
Ag/Au(25)	30	0.072	35	21.7	6.7×10^{14}	21.1	−0.9
Ag/Au(55)	33	0.11	27	19.6	2.3×10^{13}	19.0	−7.2
Ag/Au(85)	39	0.16	22	15.9	9.1×10^{10}	15.4	−18.6
Ag/Au(105)	—	0.0016	0	32.9	5.0×10^{20}	32.3	26.0

^a Average cavity diameters estimated using HRTEM images. ^b The value in the absence of any catalysts (k_0) is 0.0012 min^{-1} . ^c Not observed.



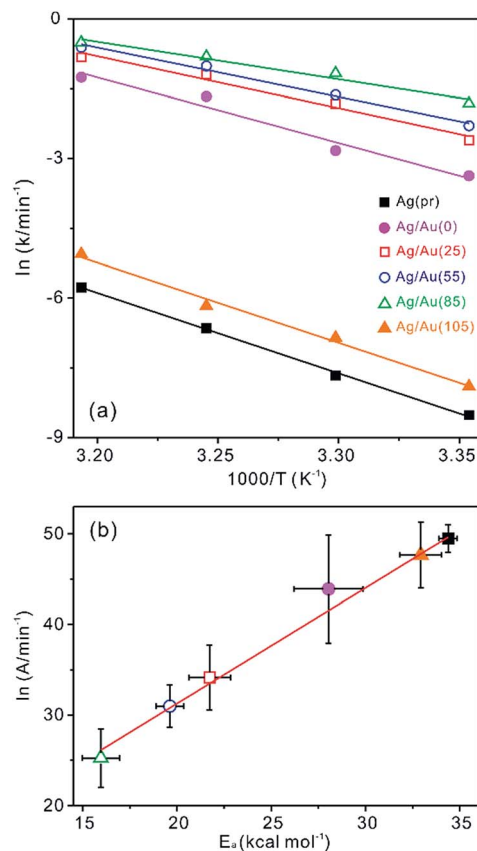


Fig. 7 (a) Arrhenius plots for the reduction reaction of 11 μM RhB(aq) catalyzed by indicated nanocatalysts in 1.3 mM $\text{KBH}_4(\text{aq})$. (b) Compensation law plot of the frequency factors and the activation energies extracted from the Arrhenius plots.

Table 1 have revealed that the catalytic activity depends on the morphologies of Ag/Au nanocomposites, which can be controlled by adjusting the galvanic replacement temperature. In particular, Ag/Au(85) that has been fabricated at 85 °C shows the most efficient catalytic performance.

Fig. 7b shows the compensation law plot of $\ln A = \alpha + E_a/(RT_0)$, where α is a constant and T_0 is the fictitious isokinetic temperature, where the catalytic rates of all the nanocomposites become the same.^{26,41,42} The empirical linear relationship between $\ln A$ and E_a of Fig. 7b clearly demonstrates the compensation effect in our catalytic system, where T_0 has been found as 395 K. The compensation effect is relevant to a switching from a regime where the overall rate is controlled by the activation of the reactant to a regime where the stability of the product becomes more important.^{26,42} Thus, we consider that for the reaction with Ag/Au nanoboxes having cavities, E_a was lower while the lower density of active sites would be compensated largely the stability of adsorption.

The catalytic reduction mechanism of RhB *via* nanocatalysts can be described as follows. The nucleophilic BH_4^- ions supply electrons to Ag/Au nanocatalysts and then electrophilic RhB molecules adsorbed on Ag/Au nanocatalysts take electrons from Ag/Au nanocatalysts, facilitating electron transfer from BH_4^- to RhB through their catalytic surfaces.^{20,31,34} Hence, Ag/Au

nanocatalysts act as electron relays for the reduction reaction of RhB in $\text{KBH}_4(\text{aq})$.^{8,31,34} With respect to the higher catalytic activity of Ag/Au nanoboxes, the nanoreactor confinement effect of Ag/Au nanocatalysts with hollow interiors is considered to expedite electron relays from BH_4^- to RhB enormously by reducing E_a extensively. Although catalytic stability during the reduction of RhB is an important factor, it is well-known that hollow structures have tendency to be unstable due to their high surface energy, resulting in the aggregation form as a stable state.^{21,24} In order to check the stability of Ag/Au(85) nanocatalyst, reusability tests have been conducted for three times. As shown in Fig. S10,[†] the catalytic activity of our nanocatalyst has decreased significantly after repeated cycles of catalysis due to the intrinsically unstable nature of the hollow structure. We consider that although the catalytic activity of Ag/Au(85) nanocomposites is considerably high, their stability is relatively low. Thus, we recognize that some ways to extend the limited reusability of nanocatalyst should be found to lower the overall price of noble-metal catalysts.

Conclusions

Morphology-controlled Ag/Au nanocomposite have been fabricated *facilely via* a modified galvanic replacement reaction using Ag nanocubes as sacrificial templates. The morphologies of Ag/Au nanocomposites have been readily controlled by adjusting the temperature of the galvanic reaction; the structures of Ag/Au nanocomposites prepared at 0, 25, 55, 85, and 105 °C are Au-decorated Ag nanocubes, well-defined nanoboxes, truncated nanoboxes, and porous nanoboxes, and broken nanoboxes, respectively. The catalytic activities of Ag nanocubes and Ag/Au nanocomposites have been investigated by measuring the time-dependent reduction of RhB in the presence of KBH_4 ; Ag/Au nanocomposites with hollow interiors have shown much higher catalytic activity with a smaller activation energy than pristine Ag nanocubes or Ag/Au nanocomposites without cavities. Among Ag/Au nanoboxes having hollow structures, Ag/Au nanocomposites fabricated at 85 °C show the most efficient catalytic performances as their largest nanocavities are surrounded by the most porous walls. Regarding to the higher catalytic activity of Ag/Au nanoboxes, the nanoreactor confinement effect of Ag/Au nanocatalysts with hollow interiors is considered to expedite electron transfer from KBH_4 to RhB enormously by reducing E_a largely; the energetically favorable formation of the activated complex within the cavities has lowered the energy barrier subsequently. Overall, the temperatures of galvanic replacement reactions have been varied to optimize the morphologies and the subsequent catalytic performances of Ag/Au nanocomposites.

Acknowledgements

This work was supported by research grants through the National Research Foundation of Korea funded by the Korea government (2014-057382 and 2015-051798).



References

- 1 G. Collins and J. D. Holmes, *Adv. Mater.*, 2016, **28**, 5689–5695.
- 2 M. Salerno, J. R. Krenn, B. Lamprecht, G. Schider, H. Ditlbacher, N. Felidj, A. Leitner and F. R. Aussenegg, *Opto-Electron. Rev.*, 2002, **10**, 217–224.
- 3 L. Wang and Y. Yamauchi, *J. Am. Chem. Soc.*, 2013, **135**, 16762–16765.
- 4 P. K. Jain, X. Huang, I. H. El-Sayed and M. A. El-Sayed, *Acc. Chem. Res.*, 2008, **41**, 1578–1586.
- 5 M. R. Langille, M. L. Personick, J. Zhang and C. A. Mirkin, *J. Am. Chem. Soc.*, 2012, **134**, 14542–14554.
- 6 Y. Yin, C. Erdonmez, S. Aloni and A. P. Alivisatos, *J. Am. Chem. Soc.*, 2006, **128**, 12671–12673.
- 7 Y. Sun, B. Wiley, Z.-Y. Li and Y. Xia, *J. Am. Chem. Soc.*, 2004, **126**, 9399–9406.
- 8 M. Son, J. Lee and D.-J. Jang, *J. Mol. Catal. A: Chem.*, 2014, **385**, 38–45.
- 9 S. Duan and R. Wang, *Prog. Nat. Sci.: Mater. Int.*, 2013, **23**, 113–126.
- 10 W. He, X. Wu, J. Liu, X. Hu, K. Zhang, S. Hou, W. Zhou and S. Xie, *Chem. Mater.*, 2010, **22**, 2988–2994.
- 11 Q. Zhang, J. Xie, J. Liang and J. Y. Lee, *Adv. Funct. Mater.*, 2009, **19**, 1387–1398.
- 12 P. Raveendran, J. Fu and S. L. Wallen, *Green Chem.*, 2006, **8**, 34–38.
- 13 X. Xia, Y. Wang, A. Ruditskiy and Y. Xia, *Adv. Mater.*, 2013, **25**, 6313–6333.
- 14 M. R. Kim, D. K. Lee and D.-J. Jang, *Appl. Catal., B*, 2011, **103**, 253–260.
- 15 F. Bellucci, *Corrosion*, 1991, **47**, 808–819.
- 16 C. Yang, Q. B. Zhang and A. P. Abbott, *Electrochem. Commun.*, 2016, **70**, 60–64.
- 17 E. González, J. Arbiol and V. F. Puntes, *Science*, 2011, **334**, 1377–1380.
- 18 S. E. Skrabalak, J. Chen, Y. Sun, X. Lu, L. Au, C. M. Copley and Y. Xia, *Acc. Chem. Res.*, 2008, **41**, 1587–1595.
- 19 X. Gong, Y. Yang and S. Huang, *J. Phys. Chem. C*, 2010, **114**, 18073–18080.
- 20 J.-A. Kwak, D. K. Lee and D.-J. Jang, *Appl. Catal., B*, 2013, **142**, 323–328.
- 21 J. Lee, K. Han and D.-J. Jang, *Appl. Catal., A*, 2014, **469**, 380–386.
- 22 M. A. Mahmoud and M. A. El-Sayed, *Langmuir*, 2012, **28**, 4051–4059.
- 23 H. Zhang, M. Jin, J. Wang, W. Li, P. H. C. Camargo, M. J. Kim, D. Yang, Z. Xie and Y. Xia, *J. Am. Chem. Soc.*, 2011, **133**, 6078–6089.
- 24 C. M. Kisukuri, D. J. Palmeira, T. S. Rodrigues, P. H. C. Camargo and L. H. Andrade, *ChemCatChem*, 2016, **8**, 171–179.
- 25 X. Hong, D. Wang, S. Cai, H. Rong and Y. Li, *J. Am. Chem. Soc.*, 2012, **134**, 18165–18168.
- 26 J. Zeng, Q. Zhang, J. Chen and Y. Xia, *Nano Lett.*, 2010, **10**, 30–35.
- 27 Q. Zhang, W. Li, L.-P. Wen, J. Chen and Y. Xia, *Chem.–Eur. J.*, 2010, **16**, 10234–10239.
- 28 M. Hu, H. Petrova, A. R. Sekkinen, J. Chen, J. M. McLellan, Z.-Y. Li, M. Marguez, X. Li, Y. Xia and G. V. Hartland, *J. Phys. Chem. B*, 2006, **110**, 19923–19928.
- 29 S. E. Skrabalak, L. Au, X. Li and Y. Xia, *Nat. Protoc.*, 2007, **2**, 2182–2190.
- 30 H. Wu, P. Wang, H. He and Y. Jin, *Nano Res.*, 2012, **5**, 135–144.
- 31 J. Huang, S. Vongehr, S. Tang, H. Lu, J. Shen and X. Meng, *Langmuir*, 2009, **25**, 11890–11896.
- 32 F. H. AlHamedi, M. A. Rauf and S. S. Ashraf, *Desalination*, 2009, **239**, 159–166.
- 33 Y. Kim, H.-B. Kim and D.-J. Jang, *J. Mater. Chem. A*, 2014, **2**, 5791–5799.
- 34 J. Lee and D.-J. Jang, *J. Phys. Chem. C*, 2016, **120**, 4130–4138.
- 35 P. Hervés, M. Pérez-Lorénzo, L. M. Liz-Marzán, J. Dzubielia, Y. Lu and M. Ballauff, *Chem. Soc. Rev.*, 2012, **41**, 5577–5587.
- 36 M. A. Mahmoud and M. A. El-Sayed, *Nano Lett.*, 2011, **11**, 946–953.
- 37 S. Wunder, Y. Lu, M. Albrecht and M. Ballauff, *ACS Catal.*, 2011, **1**, 908–916.
- 38 W. Li, L. Kuai, L. Chen and B. Geng, *Sci. Rep.*, 2013, **3**, 2377.
- 39 R. Liu, J. Guo, G. Ma, P. Jiang, D. Zhang, D. Li, L. Chen, Y. Guo and G. Ge, *ACS Appl. Mater. Interfaces*, 2016, **8**, 16833–16844.
- 40 K. D. Zimmer, R. Shoemaker and R. R. Ruminski, *Inorg. Chim. Acta*, 2006, **359**, 1478–1484.
- 41 M. A. Mahmoud, F. Saira and M. A. El-Sayed, *Nano Lett.*, 2010, **10**, 3764–3769.
- 42 T. Bligaard, K. Honkala, A. Logadottir, J. K. Nørskov, S. Dahl and C. J. H. Jacobsen, *J. Phys. Chem. B*, 2003, **107**, 9325–9331.

