



Electrochemical and electromechanical behavior of Nafion-based soft actuators with PPy/CB/MWCNT nanocomposite electrodes

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In this work, as an alternative to precious platinum electrodes in IPMC actuators, PPy/CB/MWCNT electrode actuators were successfully fabricated by electropolymerization of PPy on both sides of the CB/MWCNT-coated Nafion membranes. Electropolymerization of PPy was performed using cyclic voltammetry (CV) and chronopotentiometry (CHP) in water and acetonitrile (AN) to study the effect of the synthesis parameters on the physical, electrochemical and electromechanical characteristics of actuators. The electrochemical characteristics of the actuators were investigated using cyclic voltammetry (CV) and electrochemical impedance spectroscopy (EIS). The physical and structural properties of the actuators were characterized using scanning electron microscopy (SEM), water uptake (WUP) and ion exchange capacity (IEC) determination methods. SEM analysis revealed that the surface morphology of actuators changes considerably by altering the synthesis parameters and that the finest cauliflower-like structure of the PPy film is formed in actuators prepared in AN using the CHP technique. These metal-free actuators showed ionic conductivity (29.88 mS cm^{-1}) and capacitive characteristics ($39.23 \mu\text{F}$) about eleven and thirty six times higher than that considered for Pt-based IPMC actuators, respectively. The deposition of PPy/CB/MWCNT electrodes on the Nafion membrane also resulted in higher WUP (79.9%) and IEC (6.7 meq g⁻¹) of the actuators, confirming the hydrophilic nature of the PPy-based electrodes. The maximum tip displacement of these actuators (at optimized conditions) was however about 26% lower than that of Pt-based IPMC actuators which can be improved by decreasing the thickness and surface resistance of the PPy film via altering the synthesis parameters.

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1. Introduction

Ionic polymer–metal composites (IPMCs)¹ which are classified as ionic electroactive polymers (EAPs) exhibit bending deformation when a small electrical potential (1–5 V) is applied across their thickness. The fascinating characteristics of IPMCs including softness, lightweight, biocompatibility, flexibility and large displacement² have made them an attractive research subject for use in a wide range of industrial and medical applications as soft actuators.^{3–5} An IPMC actuator is typically composed of a thin cation-exchange polymer membrane (mostly Nafion), through which ions of single or multiple charge can pass, and a pair of electrodes (generally Pt or Au) which are chemically plated on both surfaces.^{6,7} Prior to use, the IPMC is soaked in an aqueous ionic solution to hydrate the polymer and to allow the exchange of the desired cation (Li^+ in this work). In the absence of an applied potential, the anionic groups ($-\text{SO}_3^-$ group, in Nafion) are

attached to the polymer backbone and the cations are associated. When a potential is applied, the anionic groups are unable to move far towards the positively charged anode, whereas the cations are able to break away from the anionic groups and move towards the negative cathode, forming volumetric gradient or pressure gradient between electrodes on both sides. This gradient is responsible for bending motion of the IPMC actuator towards anode.⁸ Therefore, IPMCs store electrical energy in a double interface and convert it into mechanical output by reversible ion intercalation and de-intercalation of electrodes.^{9,10} Ion exchange membranes,¹¹ the charge and mobility of the cations¹² the solvent content of the polymer,¹³ the structure and capacitance of the electrodes¹⁴ and the specific interactions between the electrode material and the cations^{12,15} are significant factors that largely influence the actuation performance of IPMCs. Under long-term actuation in open air, however, Nafion-based IPMC actuators become dehydrated due to the leakage of the inner electrolyte and hydrated cations through cracks in the metallic electrodes.¹⁶ There have been many studies to overcome the mentioned drawbacks by modifying the ion exchange membrane,^{17,18} replacing water with non-volatile solvents,¹⁶ and replacing conventional metal electrodes with flexible conducting materials. In previous works, we illustrated that the incorporation of Li^+ ions

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to the ionic liquid-based IPMCs¹² and also the addition of sulfonated graphene oxide to Nafion membrane¹⁶ promoted the actuation response of Pt-based IPMC actuators by enhancing the capacitance, ionic conduction, operating life and the displacement rate of the actuator.

It has been established that the actuation performance of IPMC actuators is mainly dominated by electrochemical and electromechanical process of the electrode layer.^{19,20} Thus, the electrode material and the electrode structure become more crucial to achieve actuators with high performances. Surely, electrode materials with lower cost, higher flexibility and appropriate electrochemical and electromechanical properties are of great importance for various potential applications. Researchers have shown that introducing carbon nanostructures²¹ into the electrode layer results in a significant improvement of the actuation performance of the IPMC actuators.^{22,23} Carbon nanotubes (CNTs)²⁴ and carbon black (CB)²¹ possess high surface area, and exceptional electrical, mechanical and chemical characteristics and their application in electrochemical double-layer capacitors has been studied in detail. Investigating polymer-based actuators containing single-wall carbon nanotubes (SWCNTs),²⁵ activated multi-walled carbon nanotubes (MWCNTs),²⁶ and a combination of non-activated MWCNTs and a metal oxide²⁷ showed that by introducing carbon nanostructures into the electrode, the actuation performances including respond speed, actuation displacement and strain rate could be significantly improved.²⁰ The volume change of carbon nanostructures-electrode based ionic actuators is caused by the intercalation of ions due to the electric double layer charging of the electrodes.^{28,29} However, carbon nanostructures have low capacitance which can be improved by compositing with other conducting materials.^{30,31} Conducting polymers (CPs) are soft actuation and sensing materials with promising applications in biomedical devices³² and micromanipulation systems³³ due to their good electrical conductivity, large pseudo-capacitance, and relatively low cost. The most commonly used CPs include polyaniline (PANI),³⁴ polypyrrole (PPy)^{35,36} and poly[3,4-ethylenedioxothiophene] (PEDOT).³⁷ CPs have been combined with carbon nanostructures³⁸⁻⁴⁰ and characterized for electromechanical actuator applications⁴¹ and electrochemical capacitors.^{42,43} These composites are interesting since they can combine two relatively low-cost materials to gain the large pseudo-capacitance of the CPs coupled with the conductivity and mechanical strength of the carbon nanostructures. The volume change of CPs is caused by insertion and extraction of ions and solvent from the electrolyte when material is electrochemically switched between the oxidized and reduced states. Mostly, PPy is used as actuator material since it is easy to fabricate and actuate and it possesses high conductivity, environmental stability and ability to form freestanding films.^{44,45} For actuators, electrochemical polymerization is most commonly utilized to prepare PPy film since it is a stable and well-reproducible synthetic method and can rapidly synthesize polymer film with different properties by adjusting the procedure parameters.^{46,47} In general, galvanostatic or potentiostatic method is applied to synthesize the CPs, although more elaborate deposition methods have been reported, such as a sequence of potential steps.⁴⁴ The choice of solvent in electropolymerization method is of critical

importance, determining both the physical characteristic (*i.e.* porosity and conductivity) and the actuation performance of CPs.

Due to the hydrophobic nature and strong van der Waals interactions between MWCNTs, they intend to pack into crystalline ropes and tangle networks. This diminishes the special mechanical and electrical properties of the individual tubes.⁴⁸ In this work, we constructed novel metal-free ionic actuators by sandwiching Nafion-117 membrane between two pieces of CB/MWCNT/PPy nanocomposite film based electrode, in which the PPy films were uniformly deposited onto the CB/MWCNT-coated Nafion membrane through electropolymerization method. A combination of the electrical conductivity, large surface area of the individual CB and MWCNT nanoparticles and excellent electrochemical properties of the PPy films resulted in the composite electrode having good physical properties and a high electrochemical capacitance. The properties of PPy film and the electromechanical performance of resulting actuators were manipulated by altering the electropolymerization parameters; including solvent type and potential regime. To this end, electropolymerization of PPy was performed using cyclic voltammetry (CV) and chronopotentiometry (CHP) in water and acetonitrile (AN). The effects of the above-mentioned parameters were followed using physical, electrochemical and electromechanical measurements and were compared with the corresponding behavior of Pt electrode-based IPMC actuators. SEM analysis revealed that the surface morphology of actuators changes considerably by altering the synthesis parameters and that the finest cauliflower-like structure of electrode layer forms in actuators prepared in AN using CHP technique. These actuators showed the highest water uptake (WUP), ion exchange capacity (IEC), hydrated thickness, electrode capacitance, ionic conduction and tip displacement. The actuation performance of the fabricated metal-free actuators was comparable to the conventional Pt-electrode based IPMC actuators.

2. Experimental

2.1. Materials

Pyrrole (98%) and acetonitrile (AN, 99.9%) were acquired from Merck and vacuum-distilled prior to use. Lithium chloride (LiCl, 99%) and oxalic acid ($H_2C_2O_4$, 2H₂O, 99%) were obtained from Fluka and used as received. Isopropyl alcohol (C₃H₈O, 99%) and nitric acid (HNO₃, 65%) were received from Merck. Distilled water, activated CB, Vulcan XC-72R carbon, was purchased from Cabot Company. Functionalized multi walled carbon nanotubes (MWCNTs, 99%) was purchased from Sigma Aldrich (COOH content 8 wt%). DuPont liquid Nafion solution (5 wt%), in a mixture of lower aliphatic alcohols and water (45 wt%), with an equivalent weight of 1100 and Nafion 117 sheet of thickness 0.18 mm in protonated form were purchased from Aldrich.

2.2. Preparation of CB/MWCNT-coated Nafion membrane

To prepare CB/MWCNT ink 60 mg Vulcan XC-72R carbon, 30 mg MWCNT and 30 mg of Nafion solution 5 wt% was added in 20 mL mixture of isopropanol-DI water (3 : 1 v/v). The mixture was then ultrasonicated for 1.5 h and stirred at 2000 rpm for 24 h. The



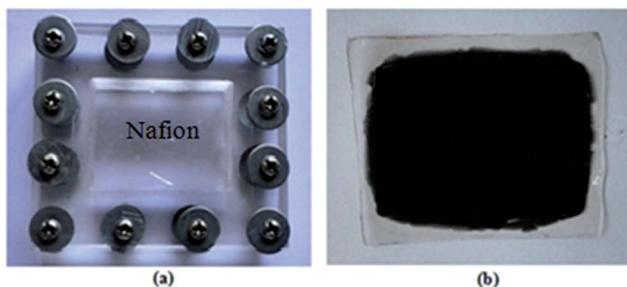


Fig. 1 Digital images of (a) Nafion membrane in the Plexiglas fram, (b) CB/MWCNT-coated Nafion membrane.

Nafion solution in the prepared ink serves as the binder and ion conductor and provides a suitable interface between the Nafion membrane and electrode materials.

A 3×4 cm sample of commercial Nafion membrane was roughened at both surfaces using abrasive paper to increase the electrode–polymer interface area. To remove all impurities remaining from the manufacturing process the membrane was washed using an extensive cleaning process described elsewhere.^{12,49} The pre-cleaned Nafion membrane was placed between two sintered glass plates (to avoid membrane deformation during drying) and put in an oven at $70\text{ }^{\circ}\text{C}$ for 1 h. Nafion membrane have a tendency to curl or be pushed sideways when absorb water or other solvents. This limits the accuracy of the ink coating process. To avoid the effects of Nafion membrane deformation, the membrane was mounted in a $3\text{ cm} \times 4\text{ cm}$ Plexiglas holder shown in Fig. 1(a) and then put on the hot plate at $60\text{ }^{\circ}\text{C}$ prior to ink coating process. A same volume of the prepared CB/MWCNT ink was carefully sprayed on both sides of the membrane using a jet spray. The CB/MWCNT-coated membrane was then placed in an oven at $80\text{ }^{\circ}\text{C}$ for 15 min to remove excess solvents. The CB/MWCNT-coated Nafion membrane (shown in Fig. 1(b)) was cut in $3\text{ cm} \times 0.5\text{ cm}$ strips and stored in DI water for use in actuator preparation.

2.3. Electrodeposition of PPy films

To prepare the actuators, PPy was electrochemically grown on both faces of the CB/MWCNT-coated Nafion membrane immersed as anode in solutions containing 0.1 M pyrrole and 0.1 M LiCl. These experiments were carried out in a conventional three-electrode cell, using platinum plate and saturated calomel electrode (SCE) as counter and reference electrodes, respectively. Prior to the experiment, the electrolyte solution was purged with nitrogen to remove oxygen, because films prepared in the presence of oxygen have inferior properties. It has been established that the mechanical properties of the PPy film, their morphology and electrochemical behavior depend on the conditions of preparation, including the applied potential regime, the nature of solvent, the purity and concentration of the initial monomer, and the nature and concentration of electrolytic salt.⁵⁰ Therefore, electropolymerization of pyrrole was performed using cyclic voltammetry (CV) and chronopotentiometry (CHP) in water and a mixture of acetonitrile–water (97 : 3 v/v) to study the effect of these parameters on the physical, electrochemical and electro-mechanical characteristic of the actuators. According to the extensively used procedure of Diaz,⁵⁰ adding a small amount of water to acetonitrile solution results in improved properties (adhesion to electrode, uniformity) of PPy film. The mixture of acetonitrile–water was named as AN in this text. Fig. 2 illustrates the fabrication process of PPy-electrode actuators. The electro-polymerization on the CB/MWCNT-coated membrane was performed by continuous potential cycling at a sweep rate of 0.03 V s^{-1} and in the potential range of -0.2 to 1 V through 35 scans. The PPy synthesis on the CB/MWCNT-coated membrane was also achieved using CHP technique in similar solutions, at current density of 2 mA cm^{-2} and duration of 3600 s. The total electro-polymerization time was calculated according to the particular current density to ensure that the total charge passed was the same for each sample. All experiments were performed on a potentiostat/galvanostat (Autolab) controlled by NOVA software at ambient temperature. The membranes were then rinsed with ethanol and DI water repeatedly, immersed in 1 M HNO_3 solution

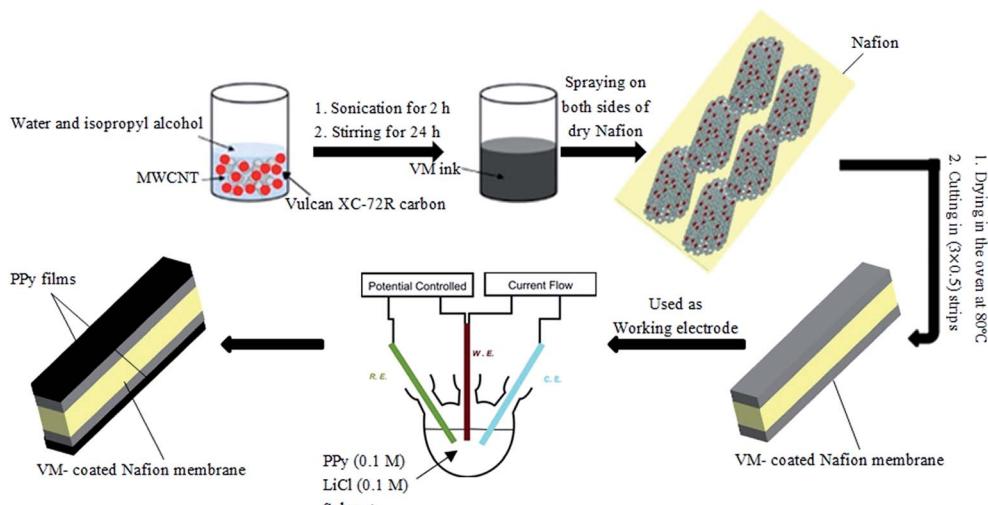


Fig. 2 Schematic diagram of the fabrication process of PPy/CB/MWCNT electrode-based actuators.



Table 1 Naming system of actuators

Sample name	LiCl (M)	Pyrrole (M)	Solvent	Synthesis method
IP	—	—	—	—
IP-W-CV	0.1	0.1	H ₂ O	CV
IP-W-CHP	0.1	0.1	H ₂ O	CHP
IP-AN-CV	0.1	0.1	AN : H ₂ O	CV
IP-AN-CHP	0.1	0.1	AN : H ₂ O	CHP

for 12 h, rinsed again and stored in aqueous 1 M LiCl solution to exchange all protons in polymer with Li⁺ ions. The actuators were named to indicate the type of the solvent and electrochemical technique used in their fabrication; IP-W-CV and IP-AN-CV for actuators prepared using CV technique and IP-W-CHP and IP-AN-CHP for actuators prepared using CHP technique in water and AN, respectively. Table 1 gives the names and the detailed process information for all actuators. The physical, electrochemical and electromechanical behavior of the fabricated PPy/CB/MWCNT-electrode based actuators were compared with Pt-electrode based IPMC actuator prepared by depositing Pt electrodes on the faces of the cleaned Nafion membrane, using an electroless plating method described in our previous work.¹⁶ The Pt-based IPMC actuator was named as IP in this work.

2.4. Physical and morphological characterization of actuators

Water uptake (WUP) of the actuators was calculated using eqn (1):

$$\% \text{WUP} = 100 \times (W_{\text{wet}} - W_{\text{dry}}) / W_{\text{dry}} \quad (1)$$

where W_{wet} and W_{dry} are wet weight and dry weight of the actuators, respectively. To determine W_{wet} of the actuators, they were equilibrated in DI water at room temperature for 24 h. The fully hydrated actuators were then removed from water and the excess water was wiped off from their surface using a tissue paper and weighted instantly. The actuators were then vacuum dried in an oven at 80 °C overnight and their W_{dry} was measured.

The hydrated thickness of the actuators was measured at multiple points using a digital micrometer. The ionic-exchange capacity (IEC) of the prepared actuators was measured by classical titration method, described elsewhere.¹⁶

Scanning electron microscopy (SEM) was employed to study the morphology of the electrode region of actuators using a TESCAN-MIRA3 SEM instrument. SEM was performed at an accelerating voltage of 20.0 kV at 500 nm and 2 μm magnification levels.

2.5. Electrochemical characterization of actuators

The ionic conduction and capacitive characteristics of the prepared actuators were studied using electrochemical impedance spectroscopy (EIS) and cyclic voltammetry (CV). To this end, square elements of 5 mm × 5 mm were cut from the actuators and sandwiched between a pair of platinum contacts (with the same surface area) embedded within a Plexiglas clamp. The measurements were performed on a potentiostat/galvanostat (EG&G) controlled by PowerSuite software at ambient

temperature. CV scans were carried out in the potential range of −0.2 to 1 V at a scan rate of 0.03 V s^{−1}. EIS measurements were performed in the frequency range of 100 kHz to 100 mHz by superimposing an a.c. voltage of 5 mV and the results were analyzed using Z-view software. Proton conductivity (σ) was obtained using eqn (2), where σ is the proton conductivity (S cm^{−1}), l the measured thickness of the actuators (cm), R is the resistance (Ω) and A is the area of the actuators (cm²).

$$\sigma = l / RA \quad (2)$$

2.6. Actuation measurements of actuators

Maximum tip displacement of the free end of the actuators was measured by applying a d.c. voltage across the thickness of the sample under study and the effect of varying amplitude of the input voltage was investigated. These measurements were carried out in an open air environment (25 °C) of 40% humidity using the arrangement shown in our previous work.¹² To this end, the actuator of standard dimensions of 0.5 cm wide by 3 cm long was held at one end in a sample holder consisting of a pair of platinum contacts within an electrically insulating clamp. The metal contacts in turn were connected to the two terminals of a d.c. power supply using thin wires. All actuators were subjected to d.c. voltages from 1 to 8 V. Video images of the bending deformation of the free end of the sample were captured by the digital camera. The images were then transferred to a computer and the average maximum peak tip displacement of the actuator was determined by measuring the linear distance between the starting point and end point of the free end of the actuators, using image analysis software.

3. Results

3.1. Electropolymerization of PPy

Generally, in CHP technique, a constant current pulse is applied to the working electrode and its resulting potential is measured against the reference electrode as a function of time. Fig. 3

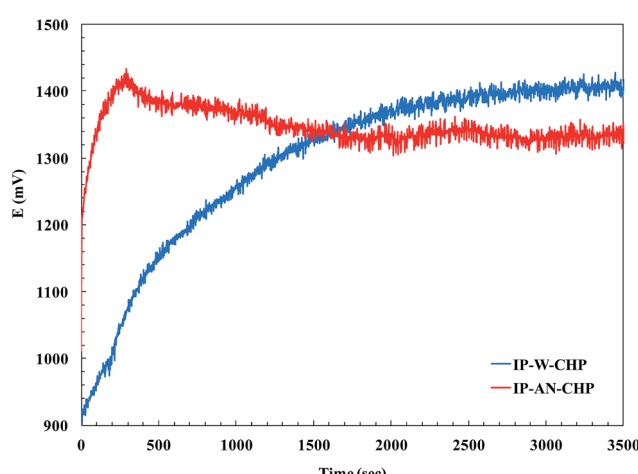


Fig. 3 The effect of solvent on the chronopotentiograms of polymerization of PPy in fabrication of IP-AN-CHP and IP-W-CHP actuators.



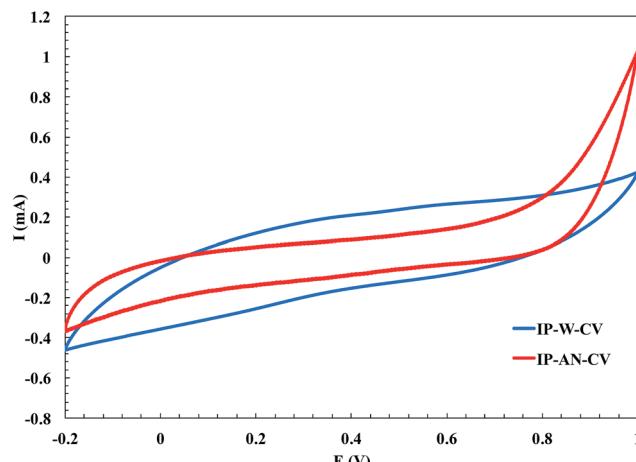


Fig. 4 The effect of solvent on the cyclic voltammograms (average of 35 cycles) of polymerization of PPy in fabrication of IP-W-CV and IP-AN-CV actuators.

compares the $E-t$ curve during CHP electropolymerization of pyrrole in water and AN. In both cases, as the current was applied, the measured potential was abruptly increased due to the ohmic loss, and afterward it gradually changed and stabilized. However, the mentioned potential change occurred in a significantly shorter time in the case of electropolymerization in AN. In other words, the time elapsed for achieving to the desired anodic potential was much longer in water compared to that observed in AN, indicating a higher rate of nucleation and polymerization of PPy in AN (IP-AN-CHP actuator). This was attributed to the higher donor number of AN in comparison with water.⁵⁰ The earlier electrochemical studies have reported that the first stage of the oxidation reaction of pyrrole to PPy involves the electrode oxidation of monomer yielding radical cations.³⁵ The stability of radical cations is an important factor in electropolymerization. The radical cations with moderate stability possess the chemical selectivity which favors radical coupling in the film formation.⁵¹ However, the radical cations with lower stability are more reactive and react with solvent or other nucleophiles close to the electrode surface. Therefore, the yield of polymer and its conductivity can grow with increasing donor number of the solvent. The results shown in Fig. 3 suggest that the radical cations formed in AN possessed higher stability and could react selectively with the MWCNTs^{52,53} and pyrrole monomers, resulting in the faster PPy film formation. Similar trend was observed in electropolymerization of pyrrole using CV technique. Fig. 4 compares the cyclic voltammograms of electropolymerization of pyrrole on the CB/MWCNT-coated Nafion membrane in water and AN. In CV measurements, the current is recorded as a function of the potential applied to the working electrode, and at any electrode potential the measured current is the sum of the capacitive current and the faradaic current.⁵⁴ The capacitive current is originated from the structure of the electrode–solution interface, while the faradaic current is measured when charges flow across the interface. Rectangular region of CV voltammograms are related to the

capacitive current and slope regions are referred to the faradaic current. As can be seen in Fig. 4, a higher faradaic current was achieved during electropolymerization of pyrrole in AN (IP-AN-CV), revealing a faster red/ox reactions across the electrode–solution interface, and so confirming a higher nucleation and polymerization rate. However, a slightly higher capacitive current was achieved in the case of electropolymerization in water (IP-W-CV) because of higher dielectric constant of water. The observed differences in the electropolymerization of PPy in AN and water (using both techniques) was in good agreement with the results obtained from SEM study of the samples, which would be discussed in next section.

3.2. Surface morphology of actuators

As discussed earlier, the magnitude and speed of the deformation of Nafion-based actuators depends on the structure and capacitance of the electrodes and also the nature of the interface between the electrode and the polymer.¹⁶ Fig. 5(a–d) shows longitudinal SEM images of the electrode region of the actuators; (a) IP-W-CV, (b) IP-W-CHP, (c) IP-AN-CV, and (d) IP-AN-CHP. The surface morphology is an important parameter as the PPy films are used in electromechanical applications. A uniform cauliflower-like deposition of PPy layer can be observed for all actuators. The functional groups on MWCNTs are likely to offer the interfacial interaction between the polymer and the nanotubes due to the hydrogen bonds formed between the MWCNTs and NH groups of the electrode layer.⁴⁸ Comparison of the actuators in each row reveals the effect of the applied regime (using the same solvent) on the nanostructure of the PPy electrode region. As the images show, the surface morphology of the actuators changed considerably as a function of the applied regime. It is important to note that the size of the PPy grains was significantly reduced in actuators prepared using CHP technique, shown in Fig. 5(b) and (d). The largest PPy grains were formed in IP-W-CV actuator prepared in water using CV technique. Comparison of the actuators in each column also shows that the size of the PPy grains decreased considerably in actuators prepared in AN, shown in Fig. 5(c) and (d). The finest cauliflower-like PPy grains were formed in IP-AN-CHP actuator prepared in AN using CHP technique. This can be explained by the higher rate of nucleation and polymerization of pyrrole radical cations in AN (see Fig. 3). Fig. 6(a) and (b) shows cross sectional SEM images of the electrode region of the actuators; (a) IP-W-CHP and (b) IP-AN-CHP. The dark regions in these images show the Nafion membrane and the bright layer running along the surface of the membrane is assigned as the PPy-coated CB/MWCNT electrode region. The surface layer in the IP-AN-CHP (Fig. 6(b)) actuator seems to be smoother and more uniform. The electrode layer appeared to have a higher thickness (~2 times) in the IP-AN-CHP actuators (0.55 μ m) compared to that in IP-W-CHP actuator (0.25 μ m). Electropolymerization of pyrrole in water can involve the side reaction of the cation radicals with water, yielding undesirable electrochemically inert products,⁵⁰ resulting in the formation of shorter polymer chains with poorer adhesion properties compared with PPy film prepared in AN.



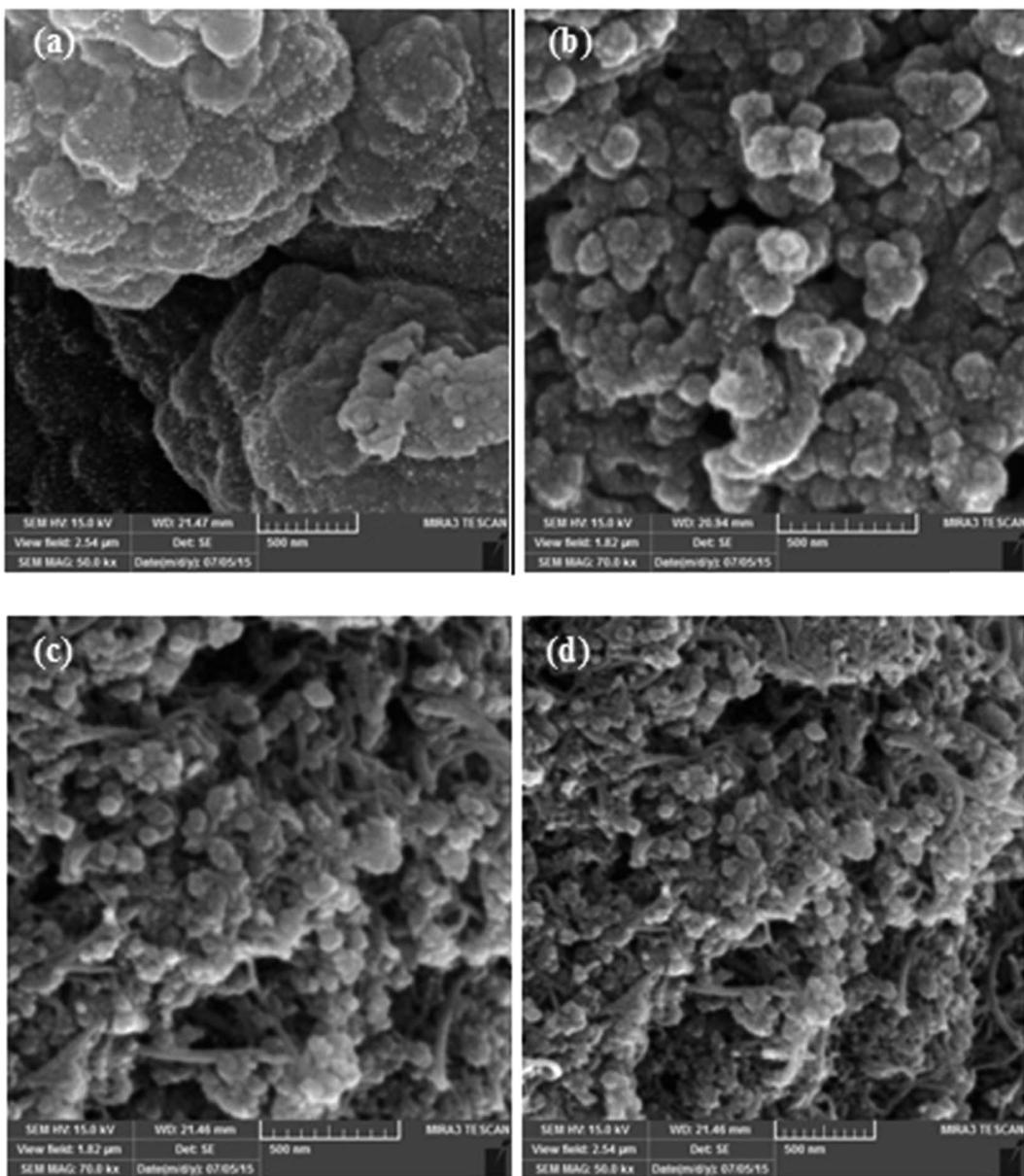


Fig. 5 Longitudinal SEM images of the electrode region of the actuators; (a) IP-W-CV, (b) IP-W-CHP, (c) IP-AN-CV, and (d) IP-AN-CHP.

3.3. Physical characterization of actuators

The data obtained from the evaluation of WUP and IEC of the prepared actuator (shown in Table 2) revealed that the metal-free actuators have higher water and ionic content compared to the Pt-based actuators. The hydrated thickness, WUP and IEC of IP-W-CV actuator, which appeared to have the largest PPy grains at the electrode region (see Fig. 5), was about 28%, 60% and 50% higher than that obtained for Pt-based IP actuator, respectively. These parameters increased significantly for the IP-W-CHP actuator, prepared in water using CHP technique, indicating the effect of the applied regime on the physical properties of the actuators. Replacing water with AN, resulted in a considerable increase in the hydrated thickness, WUP and IEC of the actuators, in which IP-AN-CHP showed the highest

values. Furthermore, the IP-AN-CHP actuator showed to have higher IEC (~134%) and WUP (~53%) compared to that of the IP-W-CHP sample, indicating the effect of the applied solvent. The results demonstrate that the physical characteristics of the actuators change considerably as a function of the applied regime and the solvent. The variation in IEC and WUP of the actuators as a function of the applied synthesis parameters can be followed more clearly in Fig. 7. The observed changes could be attributed to the morphological differences of electrode layer in the fabricated actuators, as shown in Fig. 5. The electrode layer in the IP-AN-CHP actuators appeared to have the finest structure among all actuators. This facilitates the processes of water absorption and ion exchange in these actuators. It is interesting to note that the IP-AN-CHP actuator have WUP and IEC of about 3 and 6 times higher than that considered for Pt-

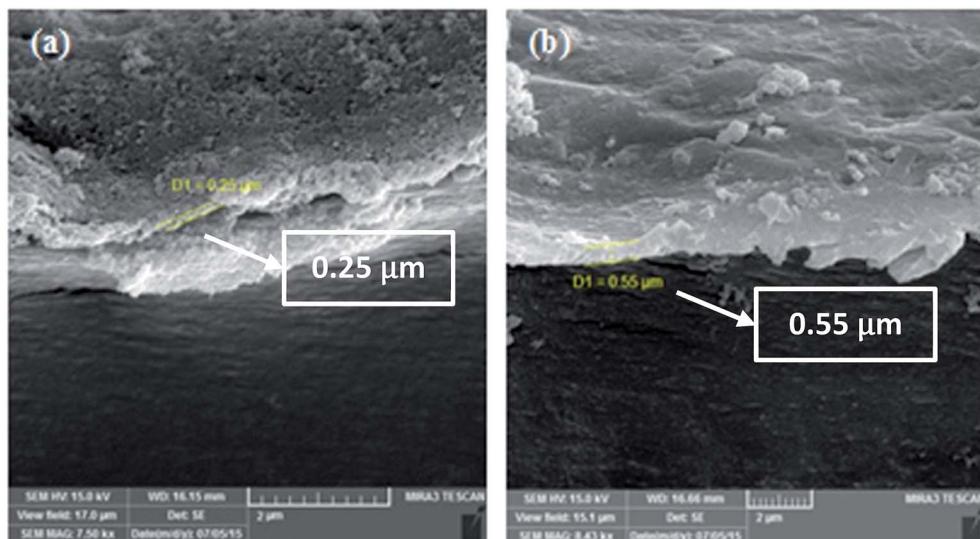


Fig. 6 Cross sectional SEM images of the electrode region of the actuators; (a) IP-W-CHP and (b) IP-AN-CHP.

Table 2 Hydrated thickness, ion exchange capacity (IEC) and water uptake (WUP) of actuators

Sample name	Hydrated thickness (mm)	WUP (%)	IEC (meq g ⁻¹)
IP	0.25	26	1.11
IP-W-CV	0.32	41.7	1.67
IP-W-CHP	0.36	52.0	2.85
IP-AN-CV	0.38	66.7	5.33
IP-AN-CHP	0.40	79.9	6.66

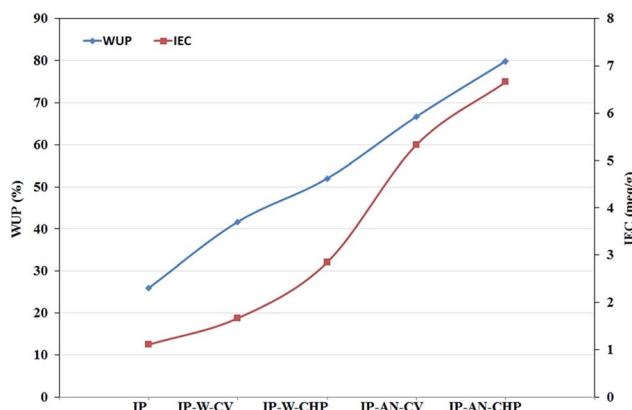


Fig. 7 Comparison of the IEC and WUP of the fabricated actuators.

electrode based IP actuator, respectively. This emphasize that the deposition of PPy film and the CB/MWCNT ink on the Nafion membrane not only led to no restriction in water absorption into the Nafion membrane but also it increased the water content of these actuators compared to Pt-based actuators. This can be attributed to the higher porosity and hydrophilicity of the electrode layer in IP-AN-CHP actuator compared to the metallic platinum electrode.

3.4. Electrochemical characterizations of actuators

The actuation performance of conventional Nafion-based actuators is directly related to the capacitance of the electrodes. The ability of electrodes to exhibit capacitive characteristic strongly depends on the pore structure and chemical affinity of the applied electrode material. As SEM images of the actuators (shown in Fig. 5 and 6) revealed the surface morphology and pore structure of the electrodeposited PPy film changed considerably by varying the synthesis parameters. Fig. 8 compares the cyclic voltammograms of the fabricated actuators to examine the effect of the electrode layer structure on the capacitive characteristic of the actuators. As can be seen, all CV voltammograms are approximately similar in shape but differ in the magnitude of the current. The symmetric shape of the CV curves revealed that the electropolymerization of PPy has given rise to two equivalent electrodes in all actuators, providing excellent charge distribution in the electrode regions. Pt-based IP actuator showed the narrowest CV loop and the current response increased in the order: IP-AN-CHP > IP-AN-CV

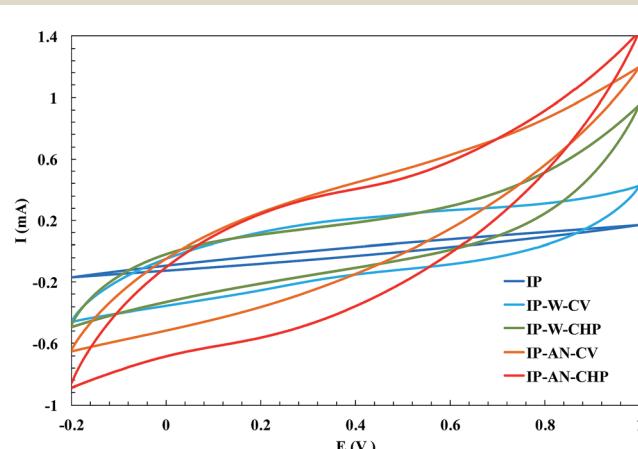


Fig. 8 CV response of the fabricated actuators.

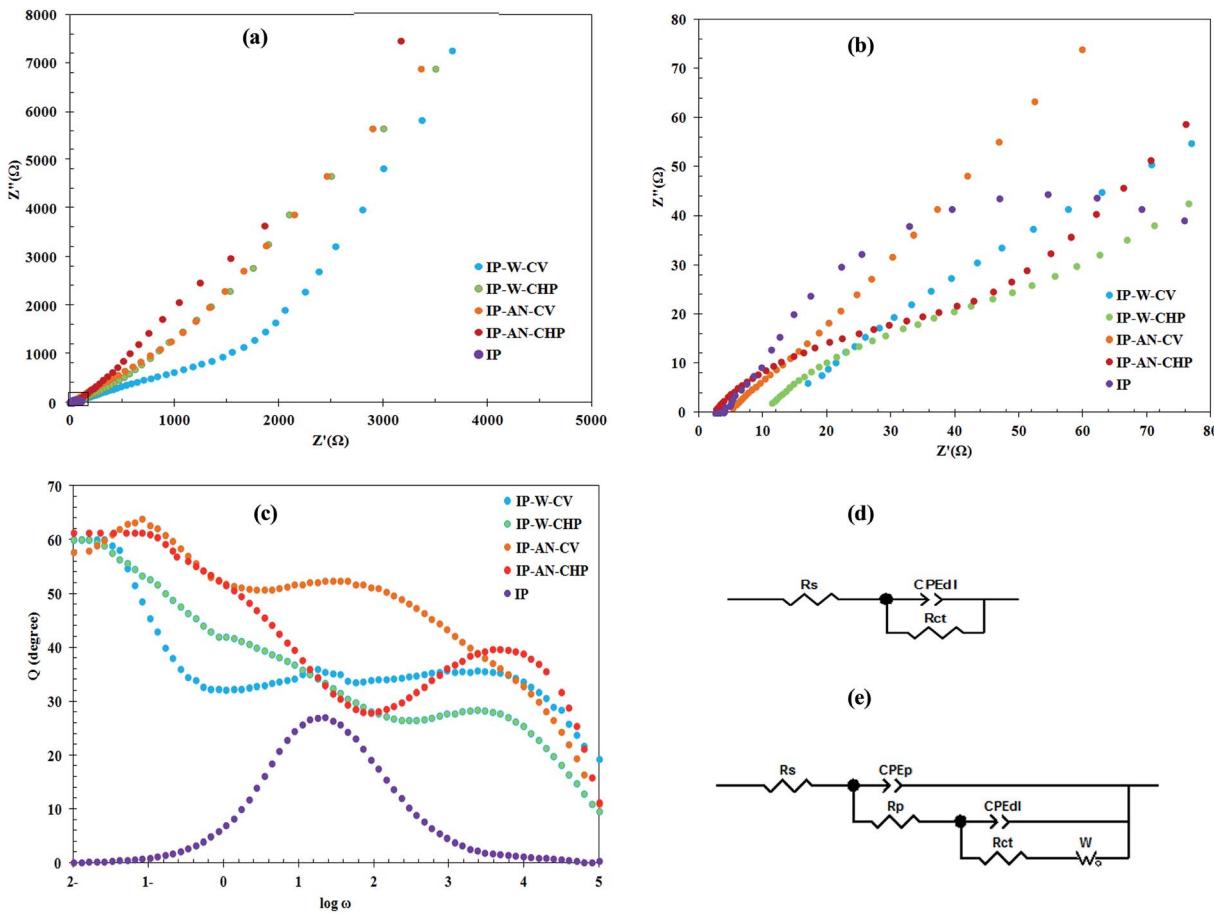


Fig. 9 Comparing Nyquist plots of actuators (a) in full view and (b) in the selected region shown in first part, (c) comparing Bode phase plots of actuators, (d) and (e) the proposed equivalent circuit model for Pt- and PPy/CB/MWCNT-electrode based actuators, respectively.

> IP-W-CHP > IP-W-CV > IP. This indicates that the PPy/CB/MWCNT-electrode based actuators generally show higher current response – and therefore higher capacitive characteristic. The results showed that the formation of the finest structure of electrode layer in the IP-AN-CHP actuator, which contributed in higher WUP and IEC of this actuator, also led to the highest current response. The more easily polarization mechanisms act, the larger current response and capacitive characteristic of the actuators would be.¹² Therefore, it is expected that the polarization mechanisms happen easier in the IP-AN-CHP actuator since it has a more porous electrode structure which is more accessible to the solvated ions. The CV results were confirmed with the data obtained from EIS studies on the actuators.

Fig. 9(a-d) demonstrates the Nyquist plots, Bode phase plots and the equivalent circuits obtained by EIS for the fabricated actuators. As can be seen, Nyquist plots (shown in Fig. 9(a)) for all actuators are nearly linear in the low frequency region and have a semicircle shape in the high frequency region. Amongst all actuators, the steepest vertical line and the largest semicircle region was observed for IP-AN-CHP and IP-W-CV actuators, respectively. This suggests that the IP-AN-CHP actuators possess the highest charge storage capacity and the IP-W-CV

actuators have the highest charge transfer resistance. For IP-AN-CHP and IP-AN-CV actuators, the semicircle region appeared in higher frequency ranges compared to that of the IP-W-CHP and IP-W-CV actuators, indicating lower ionic resistance of the actuators prepared in AN. This was attributed to the higher surface area of PPy film formed in AN and also to the higher accessibility of the ions to the more porous electrode structure in this actuator. The smallest semicircle was however obtained for Pt-based IP actuators at the far left of the x -axis, implying the lowest polarization resistance of IP among all actuators. Fig. 9(b) shows that the intercepts of the plots on the x -axis shifted to higher frequencies when water was replaced with AN, and CHP was used instead of CV technique. The high frequency intercepts on the real axis (x -axis) were used to calculate the ionic resistance of the actuators. IP-AN-CHP actuators showed to have the lowest ionic resistance due to its higher IEC and WUP (see Table 2). EIS results also revealed that replacing water with AN in electropolymerization of PPy film has an enhancing impact on the capacitive characteristic and the ionic conduction of the actuators, particularly when CHP method is applied. Fig. 9(c) compares the Bode phase plots of EIS data to study the frequency response of actuators as a function of the electrode structure. Except for Pt-based IP

Table 3 Equivalent circuit parameters of actuators obtained from EIS analysis

Sample name	R_S (ohm cm ²)	R_{ct} (ohm cm ²)	R_P (ohm cm ²)	C_P (μF)	C_{dl} (μF)	$T_P \times 10^{-4}$ (ohm ⁻¹ cm ⁻¹ S ^ϕ)	ϕ_P	Error (%)
IP	2.05	195	—	—	1.08	—	—	1.54
IP-W-CV	10.93	5000	591.3	11.22	4.08	0.13	0.80	1.83
IP-W-CHP	9.26	2817	109.3	16.43	16.48	0.14	0.60	0.80
IP-AN-CV	5.20	500	80.0	24.71	25.99	0.23	0.51	0.40
IP-AN-CHP	2.23	268	57.9	41.87	39.23	0.44	0.48	0.30

actuator, all Bode phase plots are approximately similar in shape and exhibit two peaks, which were assigned as two time constants. The intercept of the plots on the y-axis was used to recognize the electrochemical behavior of the actuators in the low frequencies. PPy-based actuators showed a phase angle in the range of 59° to 61°, indicating the diffusion-limited mass transport of these actuators at low frequency regions.⁵⁵ The impedance data was then fit with an equivalent circuit model to estimate contributions of the electrode components to the overall electrical properties. The chosen equivalent electrical circuit model for IP and PPy-based actuators are depicted in Fig. 9(d) and (e), respectively. The Randles model (shown in Fig. 9(d)) was found to give excellent fit for EIS data of IP actuators. The chosen model presented for PPy-based actuators composed of R_S , CPE_P, R_P , CPE_{dl}, R_{ct} and W elements that represent the solution resistance, pseudo-capacitance, charge transfer resistance of the polymer, double layer capacitance, charge transfer resistance of double layer and Warburg diffusion element of the actuators, respectively. The CPE_{dl}|| R_{ct} component is originated from the charging-discharging process at the electrolyte/electrode interface. The CPE_P|| R_P component is arisen from the presence of counter-anions besides the available positive charges in the electrode layer. CPE is non-ideal capacitor which its impedance can be calculated according to eqn (3):^{31,56}

$$Z_{CPE} = 1/T(j\omega)^\phi \quad (3)$$

where T and ϕ are the parameters that are independent of the frequency. T is the amplitude comparable to a capacitance (admittance) and the exponent, ϕ , shows the roughness of the electrode with values from 0 to 1. An ideal CPE shows $\phi = 1$, while $\phi = 0$ shows resistance and for the Warburg diffusion behavior $\phi = 0.5$.⁵⁷ The parameters of the equivalent circuit for all actuators are shown in Table 3. The mean error for all actuators was less than 2%, while the lowest mean error was obtained for the IP-AN-CHP actuator (0.3%), indicating the perfect fitting of the proposed equivalent circuit model to the EIS data of the PPy-based actuators. IP-W-CV actuators showed the highest ϕ , indicating the lowest surface roughness (porosity) of the electrode layer in this actuator. The ϕ decreased in the order: IP-W-CV > IP-W-CHP > IP-AN-CV > IP-AN-CHP. The lowest ϕ_P belonged to the IP-AN-CHP actuator with the finest electrode structure and the highest surface porosity. The observed trend for ϕ was reversed for the calculated T in which the IP-AN-CHP actuator showed the highest T , due to its highest capacitive

characteristic. For actuators, the true capacitance (C) was calculated by the following equation:^{58,59}

$$C = [T/(R_S^{-1} + R_{ct}^{-1})^{(1 - \phi)}]^{1/\phi} \quad (4)$$

The variation in the double layer capacitance (C_{dl}), pseudo capacitance (C_P) and ionic conductivity (σ) of the prepared actuators can be followed more clearly in Fig. 10. As can be seen, the IP-AN-CHP actuator exhibited the highest C_P , C_{dl} and σ values. This was attributed to the high surface porosity of the electrode region and high ionic and water content of this actuator. In contrast, the IP-W-CV actuator showed the highest R_{ct} and R_P values due to the formation of large PPy grains with lower surface area at both faces of Nafion membrane. The results also revealed that the IP-AN-CHP actuator possessed σ , C_{dl} and C_P of about twenty three, ten and four times higher than that of IP-W-CV actuators, respectively. Increasing the C_P in these actuators can be explained by $C_P = Ae/d$, where ϵ , A and d are the permittivity coefficient of the Nafion membrane, the surface area of the electrode and distance between the electrodes (the thickness of Nafion), respectively.⁵⁶ The values of ϵ and d can be considered approximately constant for all actuators. The SEM images (Fig. 5 and 6) showed that the surface morphology of the actuators changed considerably by altering the synthesis parameters. Thus, C_P of actuators can be directly related to the surface area of the electrode region. The observed higher C_P of IP-AN-CHP actuators can be attributed to their finest electrode structure, as shown in Fig. 5. This actuator also

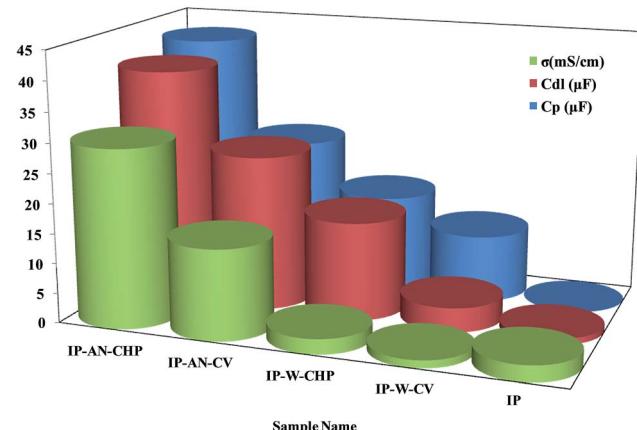


Fig. 10 The variation in the double layer capacitance (C_{dl}), pseudo capacitance (C_P) and ionic conductivity (σ) of all actuators.



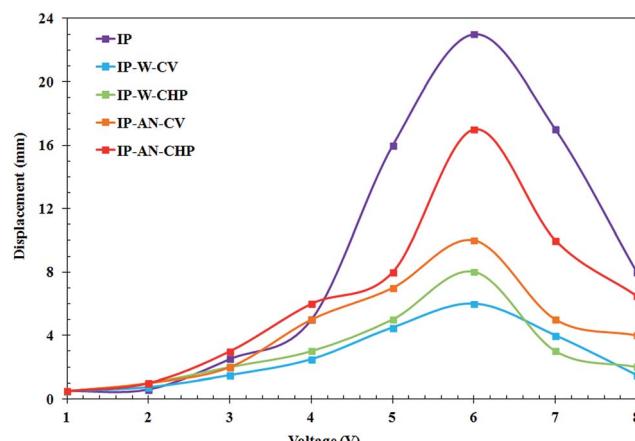


Fig. 11 Comparison of the dependence of tip displacement on amplitude of the input d.c. voltage for the fabricated actuators.

showed to have C_{dl} and σ of about thirty six and eleven times higher than considered for Pt-based IP actuator, respectively, while R_{ct} of the IP-AN-CHP actuator was about 37% higher. This suggests that the more porous and flexible structure of the electrode layers in the PPy-based actuator contributed in the formation of larger double layers, accommodating a higher amount of electrolyte and facilitating the insertion and extraction of ions at the electrode regions. It has been established there are some associations among the quinoid rings of PPy and MWCNTs which facilitate the charge transfer between the two components; PPy behaves as a good electron donor while MWCNTs act as good electron acceptors.⁶⁰ Thus, it is expected that the surface conductivity of the PPy-electrode based actuators can be improved by varying the synthesis parameters to provide a stronger coupling between the MWCNTs and the PPy chains.

3.5. Actuation performance of actuators

The average maximum tip displacement of the free end of the fabricated actuators was measured as a function of the amplitude of the applied d.c. voltage. Fig. 11 demonstrates that the tip displacement of all actuators increased as a function of input voltage in potential range of 1 to 6 V, and afterward decreased by

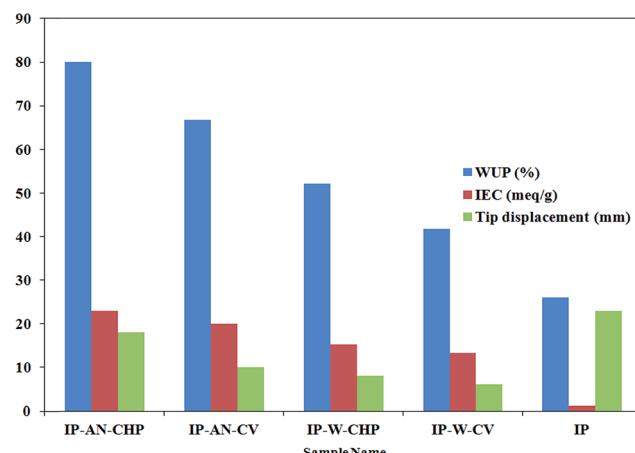


Fig. 13 The average maximum tip displacement, IEC (meq g⁻¹) and WUP (%) of all actuators, in response to the same d.c. voltage of 6 V in amplitude.

applying higher voltages. The comparison of the tip displacement of actuators also revealed that the magnitude of tip displacements were dependent on the applied synthesis parameters for the electropolymerization of the PPy film. The corresponding overlaid digital images captured at the starting point and end point for actuators (a) IP-W-CV, (b) IP-W-CHP, (c) IP-AN-CV and (d) IP-AN-CHP (with the corresponding SEM images) in response to d.c. voltage of 6 V in amplitude are shown in Fig. 12. The maximum tip displacement of actuators followed the order of IP-AN-CHP > IP-AN-CV > IP-W-CHP > IP-W-CV. The IP-AN-CHP actuator which showed to have the highest WUP, IEC, ionic conductivity and electrode capacitance presented the highest tip displacements (~17 mm) under 6 V d.c. potential which was about two and three order of magnitude higher than that considered for IP-AN-CV (~10 mm) and IP-W-CV (~6 mm) actuators. The variation in the WUP, IEC and the maximum tip displacement of actuators can be followed in Fig. 13. It is important to note that the IP-AN-CHP actuator with the highest hydrated thickness showed the highest tip displacement. This shows that the deposition of a thicker PPy film (see Fig. 6) promotes higher water absorption while it does not restrict the bending deformation of the actuator. The

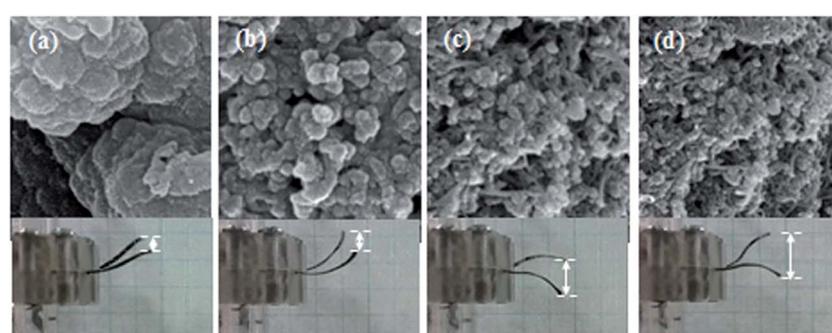


Fig. 12 The corresponding overlaid digital images captured at the starting point and end point for actuators (a) IP-W-CV, (b) IP-W-CHP, (c) IP-AN-CV and (d) IP-AN-CHP (with their corresponding SEM images) in response to d.c. voltage of 6 V in amplitude.



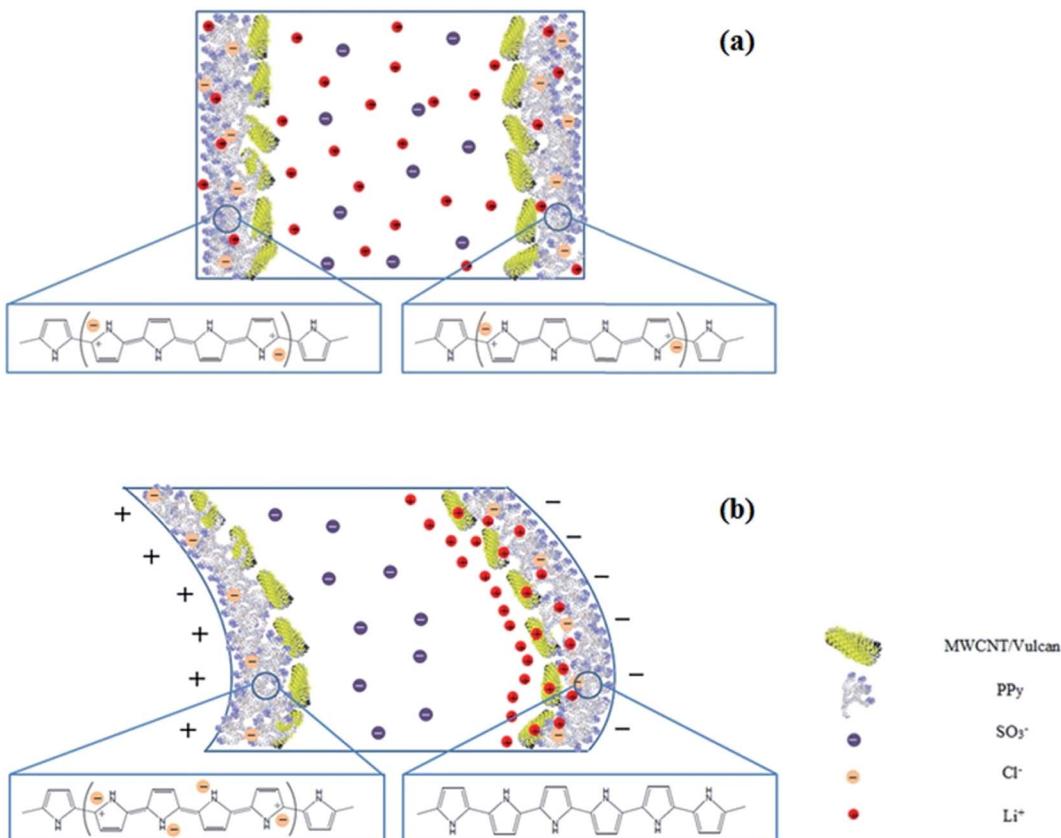


Fig. 14 Schematic representation of ion-redistribution in PPy-electrode actuators (a) with no potential applied and (b) in response to a d.c. voltage.

maximum tip displacement of this actuator was however about 26% lower than that of the conventional Pt electrode-based IPMC actuators fabricated in previous studies¹¹ which can be improved by decreasing surface resistance of the electrode layer by changing synthesis parameters of PPy film. In addition, it is expected that the IP-AN-CHP actuator to show a higher durability (life time) compared to the conventional IPMC actuators since it can absorb water in significantly higher extent (~3 times). Fig. 14 illustrates a schematic depiction of the redistribution of ions under influence of applied electric field and formation of electric double layer at the interface of Nafion and PPy/CB/MWCNT electrode. As Fig. 14(a) demonstrates when no potential is applied Li⁺ ions are mainly located within the Nafion membrane while a smaller portion of these ions are trapped in porous structure of the electrode layer at both sides. On application of an electric voltage, V , opposite charges of equal magnitude, Q , accumulate on the electrodes. The charge can be calculated according to eqn (5), where C is the capacitance of the actuators.⁷

$$Q = CV \quad (5)$$

Accumulation of opposite charges on the electrodes produces an electric field gradient developing across the thickness of the actuators. It is expected that the PPy chains

become reduced at the negatively charged electrode (cathode), shown in Fig. 14(b). In this case, the hydrated Li⁺ ions begin to break away from the sulfonate groups within Nafion membrane and move towards the cathode, while the sulfonate groups remain fixed to the molecular backbone. Since Nafion only pass cations, the chloride ions cannot diffuse electrically towards the anode and remain in the electrode region. Moreover, it has been established that MWCNTs change dimensions upon applying an external potential.⁶¹ The predominant actuation mechanism for the MWCNTs is the electrostatic repulsion between CNTs that are equally charged and the establishment of an electrostatic double layer which prompts the CNT layer to swell.⁶¹ This results in the formation of a thicker double layer near the cathode and bending deformation of the sample towards the anode. On application of a higher voltage to actuators with higher capacitance, more charge accumulates on the electrodes and a greater number of hydrated ions diffuse electrically towards the cathode. This produces a larger tip displacement in actuators with the highest capacitive characteristic (according to CV and EIS data, IP-AN-CHP actuator).

CNTs^{20,62} and CPs²⁶ have been proposed as alternative electrode materials for the fabrication of ionic actuators. The work presented by Lee *et al.*⁶³ is a procedure for the fabrication of IPMC actuators with electrospun Nafion/MWCNT composite electrodes, in which the surfaces of composite electrodes were



ion-beam coated with a thin gold layers to reduce the surface resistance. The proposed ionic actuator showed a rapid and large bending motion. They showed that the energy efficiency of strain was over 10 times higher than that of the conventional IPMC actuators.⁶³ Various efforts have also been made to incorporate PPy into the Nafion membrane^{64,65} in order to improve the performance of IPMC actuators. Although there are reports on the actuation performance of CP-based actuators³⁴ the PPy-electrode Nafion-based actuators have not been reported previously. The attractive points of PPy includes low operating voltage, large electrochemical strain, and long cycle life⁶⁶ Furthermore, PPy film is non-toxic, lightweight, flexible, biocompatible and can provide sufficient surface conductivity and prevent dehydration of actuator. These are the requirements for a suitable electrode material for ionic actuators. As mentioned earlier, the magnitude and speed of the bending deformation of water-based actuators depends strongly on the IEC, WUP and σ of the actuator. Compared to the conventional Pt-based IPMC actuator,⁷ the current work proposes a simple method for manufacturing high-performance ionic actuators using PPy-CB/MWCNT-coated Nafion membrane with improved hydrophilic, IEC, σ and C_{dl} characteristic. This paper also provides new insights in the relationship between the CB/MWCNT/PPy electrode nanostructure and the synthesis parameters of PPy film and their effect on the actuation performance of the metal-free ionic actuators.

4. Conclusion

We have demonstrated for the first time the fabrication and characterization of PPy/CB/MWCNT-electrode Nafion-based actuators. PPy film was electro-polymerized on the CB/MWCNT-coated Nafion membrane using CHP and CV techniques. The effect of solvent type and applied regime on the PPy film nanostructure and also the actuation performance of the actuators were followed using physical, electrochemical and electromechanical experiments and were compared with the corresponding behavior of Pt electrode-based IPMC actuators. WUP and IEC of actuators were determined and the impact of the applied technique and the solvent type on the morphology of PPy-based electrode layer was studied using SEM. Furthermore, CV responses, EIS data and actuation performance of actuators were compared to follow the effect of varying synthesis parameters on R_{ct} , C_{dl} and σ of actuators. Generally, actuators prepared using CHP technique in AN as solvent showed the highest hydrated thickness, IEC and WUP, which was about 3%, 17% and 6% higher than that considered for conventional Pt-based actuators, respectively. SEM analysis showed that the electrodeposition of PPy on CB/MWCNT-coated Nafion membrane resulted in the formation of PPy film with good surface coverage. The surface morphology of the electrode layer was found to be dependent on the solvent type and the applied regime. The CHP technique gave rise to finer and denser PPy grains in actuators prepared in AN. Furthermore, the electrode layer appeared to have higher thickness (~ 2 times) in these actuators compared to the polymer film electro-deposited in water using CHP technique. CV studies of

actuators revealed that applying PPy film on CB/MWCNT-coated Nafion membrane generally led to a higher current response – and therefore higher characteristic capacitance compared to the conventional Pt-based IPMC actuators. The actuators prepared using CHP technique in AN showed the highest C_{dl} and σ , and also the lowest R_{ct} among all actuators. Moreover, these actuators showed the highest ionic conductivity which was about twenty three and eleven times higher than that considered for actuators prepared using CV technique in water and Pt-based IPMC actuators, respectively. In electromechanical tests the maximum tip displacement were recorded for actuators. For all actuators, displacement increased by increasing applied voltage. The PPy-electrode actuators prepared using CHP technique in AN showed the highest tip displacement in all applied voltages. Moreover, the tip displacement of these actuators appeared to be about two times higher than the actuators prepared in water using CV technique. The maximum tip displacement of these actuators was however about 26% lower than that of the conventional Pt electrode-based IPMC actuators which could be improved by decreasing surface resistance of the PPy-based electrode layer by changing synthesis parameters.

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