



Cite this: RSC Adv., 2017, 7, 3249

Synthesis of titanosilicate pillared MFI zeolite as an efficient photocatalyst

Laleh Emdadi,^a Dat T. Tran,^b Junyan Zhang,^a Wei Wu,^a Haomin Song,^c Qiaoqiang Gan^c and Dongxia Liu^{*a}

A titanosilicate pillared MFI zeolite (abbreviated as Si/Ti-PMFI) has been synthesized by sequential infiltration of a mixed tetraethyl orthosilicate (TEOS) and tetrabutyl orthotitanate (TBOT) solvent into and hydrolysis of the TEOS/TBOT solvent entrapped between the multilamellar MFI layers. The Ti-species present in the titanosilicate pillars, acid sites supplied by the framework Al-O(H)-Si sites in MFI layers, and mesoporosity collectively endow Si/Ti-PMFI with efficient adsorption assisted photocatalytic functionality, which was exemplified by the degradation of methyl orange (MO) in water. The synthesis of Si/Ti-PMFI zeolites containing Al-O(H)-Si acid sites within MFI layers and Ti-sites between layers diversified the structure, physicochemical properties, and catalytic functionality of hierarchical lamellar zeolites. The Si/Ti-PMFI lamellar zeolites were studied as adsorptive catalysts for photocatalysis, especially when bulky molecules are involved.

Received 27th September 2016

Accepted 15th December 2016

DOI: 10.1039/c6ra23959e

www.rsc.org/advances

Introduction

Zeolites have wide applications as solid catalysts in many chemical and petrochemical processes because of their unique properties such as high surface area, high adsorption capacity, and well-defined micropores that are responsible for shape selectivity.^{1–5} The relatively small size of micropores (typically less than 2 nm), however, limits the catalytic performance of the zeolites and induces low catalyst utilization when the reaction of bulky molecules is concerned.^{6–8} Inclusion of the mesoporosity to the microporous zeolites, *i.e.*, synthesis of hierarchical meso-/microporous zeolites, is a strategy to improve the catalyst utilization by promoting active site accessibility and mass transport for reactions involving bulky molecules.^{7,9–11}

Different methods have been developed to create hierarchical meso-/microporous zeolites during the past decade.^{12–17} Synthesis of two-dimensional (2D) layered zeolites followed by pillaring of the resulted 2D structures is an example of these synthesis methods.^{18–22} The guest species can be intercalated between interlayer spacing of 2D zeolites to preserve their ordered layered structure. The void space created between the layers forms a mesopore system in 2D zeolites. As a result, new structural and physicochemical properties can be introduced into the 2D hierarchical meso-/microporous zeolites in addition to their intrinsic microporous features.^{23–25}

MCM-36, reported by Mobil scientists in 1990's,¹⁹ is the first pillared 2D zeolite. In the synthesis of MCM-36, a layered precursor, MCM-22 (P), was intercalated by silica species, which led to silica pillared meso-/microporous MCM-36 zeolite, with micropores within the zeolitic layers and mesopores between the layers. In recent years, the layered precursor versions of other zeolite frameworks such as FER,^{26–28} SOD,²⁹ MFI,¹³ and IPC^{30,31} have been synthesized. Silica pillared 2D structures built from these frameworks have been demonstrated, most notable structures like ITQ-36 (silica pillared FER)²⁶ and pillared MFI²¹ zeolites. In addition, pillaring of the 2D zeolites with inorganic compounds other than silica material has been reported.^{18,32–34} For example, alumina and magnesia–alumina have been applied for pillaring of the MCM-22 (P), which resulted in production of MCM-36 with variable acid-base and ion-exchange properties.³³

Advancing the derivatives of 2D zeolites that contain Ti-species will create opportunities for endowing these materials with new structural and catalytic properties. The active sites from Ti-species are grafted on the external surface or encapsulated between the 2D zeolite layers, together with the hierarchical meso-/microporosity in the zeolites, are expected to expand the application of 2D zeolites for a range of oxidation and photocatalytic reactions. The Ti-species have been introduced onto ITQ-2 zeolite by anchoring the titanocene precursor on the external surface of ITQ-2 material,³⁵ a delaminated format of the pure siliceous MWW polymorph (ITQ-1).³⁶ The resultant Ti/ITQ-2 zeolite enabled efficient reactions for epoxidation of olefins.³⁷ Similarly, Ti-species from the tetrabutyl orthotitanate (TBOT) precursor have been placed on the external surface of 2D MFI zeolite, which exhibited as excellent catalyst for epoxidation of bulky olefins.³⁸ The 2D zeolite with

^aDepartment of Chemical and Biomolecular Engineering, University of Maryland, College Park, 4418 Stadium Dr., College Park, MD 20742, USA. E-mail: liud@umd.edu

^bRDRL-SEC-C, US Army Research Laboratory, 2800 Powder Mill Road, Adelphi, MD 20783, USA

^cDepartment of Electrical Engineering, University at Buffalo, 230 M Davis Hall, Buffalo, NY 14260, USA



Ti-containing pillars (*i.e.*, titanosilicate pillared MCM-36) has also been recently synthesized by pillaring of MCM-22 (P) or ERB-1 (borosilicate and aluminum-free analogue of MCM-22 (P)) with a mixed tetraethyl orthosilicate (TEOS) and TBOT solvent.^{39,40} The same protocol has been applied to synthesis of titanosilicate pillared multilamellar titanium containing siliceous MFI (TS-1) zeolites.⁴¹ The resultant samples were active catalysts for either epoxidation of organic molecules^{39–41} or selective oxidation of bulky organic sulphides.⁴² To the best of our knowledge, the synthesis of titanosilicate pillared lamellar MFI zeolite with Al–O(H)–Si acid sites within microporous MFI layers has not been reported. Additionally, the application of the Ti-containing derivatives of 2D lamellar zeolites as catalysts for photocatalytic reactions has not been explored.

In this work, we aim to synthesize a titanosilicate pillared MFI (designated as Si/Ti-PMFI) zeolite and explore its application as a catalyst for photocatalytic reactions. The precursor of Si/Ti-PMFI is 2D multilamellar MFI, prepared using the method reported in Ryoo's group.¹³ The pillaring of 2D multilamellar MFI zeolite was carried out using a mixed TEOS and TBOT solvent, following a recent reported method for pillaring of multilamellar TS-1 zeolite.^{41,42} The TEOS/TBOT solvent was infiltrated into the gallery between two adjacent MFI zeolite nanosheet layers and was converted into titanosilicate pillars through the consecutive hydrolysis and calcination steps. The morphology and physicochemical properties of Si/Ti-PMFI zeolites were characterized. The photocatalytic degradation of methyl orange (MO) due to the presence of Ti-species in titanosilicate pillars and Al–O(H)–Si acid sites in MFI layers was studied to exemplify the application of 2D Si/Ti-PMFI catalysts in photocatalytic water treatment.

Experimental section

Materials

Titanium(IV) *n*-butoxide (TBOT, 99% purity) and methyl orange ($C_{14}H_{14}N_3NaO_3S$) were supplied by Alfa Aesar. Tetraethyl orthosilicate (TEOS, 98% purity), sodium hydroxide (NaOH, ≥97.0% purity, pellets), sulfuric acid (H_2SO_4 , 95.0–98.0%), ammonium nitrate (NH_4NO_3 , ≥99.0%), and titanium(IV) oxide (P25 titania ≥99.5% trace metal basis) were purchased from Sigma-Aldrich. Aluminum sulfate hydrate ($Al_2(SO_4)_3 \cdot 16H_2O$, 98.0–102.0%) was supplied by Mallinckrodt Chemicals. Deionized (DI) water was used throughout the experiment. Diquaternary ammonium surfactant ($[C_{22}H_{45}N^+(CH_3)_2-C_6H_{12}-N^+(CH_3)_2-C_6H_{13}]Br_2$, (C_{22-6-6})) was synthesized based on the reported method¹³ and the synthesis procedure has also been explained in our previous publications.^{43,44}

Synthesis of multilamellar MFI precursor

The Si/Ti-PMFI zeolite was synthesized from the multilamellar MFI precursor (designated as 2D MFI (P)), which was prepared following the recipe reported by Choi *et al.*¹³ Typically, the synthesis was performed by dissolving 0.7 g NaOH into 3.1 g DI water, dissolving 0.4 g H_2SO_4 in 4.2 g DI water, and subsequently adding the basic solution dropwise to the acidic solution under vigorous stirring. After cooling the mixture to

ambient temperature, 0.19 g $Al_2(SO_4)_3 \cdot 16H_2O$ was dissolved in the mixture. Then, 6.3 g TEOS was added to the mixture and the mixture was stirred vigorously at room temperature for 20 h using a magnetic stirrer. Finally, this solution was mixed with a C_{22-6-6} solution that was prepared by dissolving 2.2 g C_{22-6-6} in 15 g DI water at 333 K. After continuous mixing for 2 h at room temperature, the resultant gel was transferred into a Teflon-lined stainless-steel autoclave, followed by crystallization for 5 days by tumbling the autoclave vertically at 30 rpm in an oven heated at 423 K. After crystallization, the zeolite product was filtered, washed with DI water, and dried at 343 K overnight.

Pillaring of multilamellar MFI precursor

The pillaring of 2D MFI (P) was carried out by dispersing the precursor in a mixed TEOS and TBOT solvent, in which TEOS/TBOT molar ratio was varied from 5 : 1, 20 : 1 to 40 : 1, respectively. For comparison purpose, the pillaring in sole TEOS (TEOS/TBOT ratio = ∞) or TBOT (TEOS/TBOT ratio = 0) solvent was also conducted. In each case, the weight ratio between the pillaring solvent agent and 2D MFI (P) was controlled as 5. The resultant suspension was sealed in a flask under Ar atmosphere, heated to 351 K in an oil bath and kept at 351 K for 24 h. Finally, the solid sample was collected by filtration and dried under ambient condition.

Synthesis of Si/Ti-PMFI zeolite

Hydrolysis of 2D MFI (P) that contains entrapped TEOS/TBOT solvent was carried out by dispersing the sample in an aqueous solution with pH of 8 adjusted by NaOH. The weight ratio of aqueous solution to solid sample was 10. After stirring the mixture at 313 K for 6 h, the sample was collected by centrifugation, washed with DI water, and then dried under ambient condition. Finally, the sample was calcined at 723 K for 6 h under N_2 (100 mL min⁻¹) and at 823 K for 12 h under air (100 mL min⁻¹). The resultant pillared MFI was designated as x Si/Ti-PMFI, where x = 0, 5, 20, 40, and ∞ , respectively, denoting for the Si/Ti molar ratio used in the mixed TEOS/TBOT solvent. For comparison purpose, the 2D MFI (P) was calcined using the same protocol as that of x Si/Ti-PMFI, which was named as 2D MFI. The titanium containing microporous TS-1 zeolite (Si/Ti ratio of 50) was prepared following a reported procedure.⁴⁵ The as-synthesized samples were directly used as catalysts for photocatalytic reaction tests discussed above.

To synthesize the proton-form (H^+ -form) titanosilicate pillared 2D Si/Ti-PMFI zeolites for photocatalytic reaction tests, a portion of the calcined Si/Ti-PMFI samples was ion-exchanged three times using 1 M aqueous NH_4NO_3 (weight ratio of zeolite to NH_4NO_3 solution = 1 : 10) at 353 K for 12 h, and subsequently, collected by centrifugation, washed with DI water three times, and dried at 343 K overnight. The second calcination was conducted at 823 K for 4 h in dry air (100 mL min⁻¹, ultrapure, Airgas) to decompose the NH_4^+ cations to form the proton-form zeolite.

Catalyst characterization

Powder X-ray diffraction (XRD) patterns were recorded using a Bruker D8 Advance Lynx Powder Diffractometer (LynxEye PSD

detector, sealed tube, Cu K α radiation with Ni β -filter). Transmission electron microscopy (TEM) images were captured on a JEM 2100 LaB₆ electron microscope. The argon (Ar) isotherms were measured at 87 K on an Autosorb-iQ analyzer (Quantachrome Instruments). Prior to the measurement, samples were evacuated overnight at 623 K and 1 mm Hg. Si, Ti, and Al contents of zeolite samples were determined by inductively coupled plasma optical emission spectroscopy (ICP-OES, Perkin Elmer Optima 7000). Diffuse reflectance (DR) ultraviolet-visible (UV-Vis) spectra were obtained using an Ocean Optics USB2000+ spectrometer equipped with an IS200-4 integrating sphere detector, and the white high reflectance sphere material (manufactured from polytetrafluoroethylene (PTFE) based bulk material) was used as the reference.

Adsorption and photocatalytic decomposition of methyl orange

The liquid phase catalytic degradation of MO molecules under light irradiation was employed to examine the catalytic performance of the x Si/Ti-PMFI zeolites. The reaction set-up consisted of a 200 W HBO mercury (Hg) lamp (OSRAM, Germany) located next to a 50 mL quartz flask with the distance of 15 cm. In a typical experiment, 20 mL of 0.05 g L⁻¹ MO aqueous solution was mixed with 0.02 g of zeolite in the flask. The mixture was stirred at 500 rpm in the darkness for 2 hours and then the Hg lamp was turned on. This moment of turning on the Hg lamp was taken as the initial reaction time and 2 hours of mixing time guaranteed that the adsorption reached saturation. Liquid samples were withdrawn at regular time intervals and analyzed by a Lambda 40 UV/Vis spectrophotometer (PerkinElmer Instruments) at the maximum absorption wavelength of MO, *i.e.*, 468 nm.

Results and discussion

Formation of Si/Ti-PMFI zeolites tracked by XRD patterns

To study the influences of TBOT content in the mixed TEOS and TBOT solvent on synthesis of Si/Ti-PMFI, the molar ratio of TEOS/TBOT was controlled at 0 (*i.e.*, without TEOS in the mixed solvent), 5, 20, 40 and ∞ (*i.e.*, without TBOT in the mixed solvent), respectively, in the pillaring process. Fig. 1 shows the XRD patterns of the Si/Ti-PMFI zeolites from variant TEOS/TBOT moieties. For comparison, the 2D MFI (P) and 2D MFI, *i.e.*, multilamellar MFI before and after calcination treatment, respectively, were included in Fig. 1. The wide-angle XRD peaks (Fig. 1(B)) in spectra of 5Si/Ti-PMFI, 20Si/Ti-PMFI, and 40Si/Ti-PMFI are nearly the same as those in spectra of 2D MFI (P), 2D MFI, and ∞ Si/Ti-PMFI, which are characteristics of a crystalline MFI zeolite.^{13,21} This indicates that the pillaring of 2D MFI (P) with mixed TEOS/TBOT solvent neither scarified the crystalline structure nor introduced new crystalline phase in the MFI zeolite. The XRD spectrum of 0Si/Ti-PMFI has the characteristic peaks of MFI and a new peak around $2\theta = 25.4^\circ$. The new diffraction peak can be assigned to anatase phase (JCPDS file no. 84.1286) of titania, which suggests that Ti-species from the TBOT solvent were not completely converted into titanosilicate.

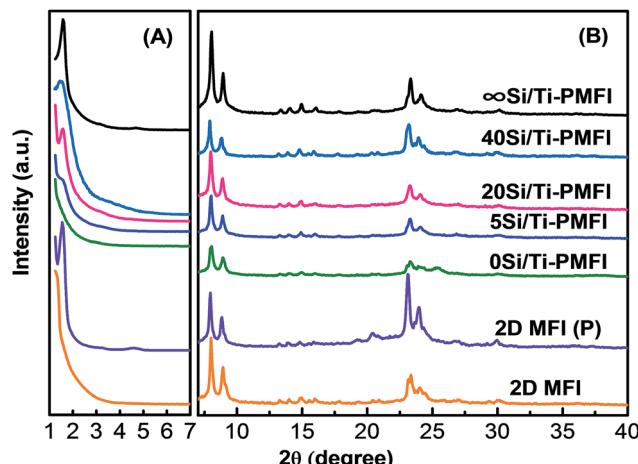


Fig. 1 Low-angle (A) and wide-angle (B) XRD patterns of the synthesized 0Si/Ti-PMFI, 5Si/Ti-PMFI, 20Si/Ti-PMFI, 40Si/Ti-PMFI, and ∞ Si/Ti-PMFI samples, respectively. 2D MFI (P) and 2D MFI have been shown here for comparison purpose.

The diffraction peaks in the low-angle range of XRD spectra (Fig. 1(A)) demonstrate the interlamellar structural coherence of the 2D MFI layers in the Si/Ti-PMFI samples. The 2D MFI (P) exhibits two low-angle peaks at $2\theta = 1.54^\circ$ and 4.58° , respectively, which are assigned to the first- and third-order reflections in the multilamellar MFI zeolite.²¹ The removal of the C₂₂₋₆₋₆ template by calcination in 2D MFI (P) led to collapse of the ordered MFI layers, indicated by the disappearance of low-angle diffraction peaks in 2D MFI in Fig. 1(A). The silica pillared multilamellar MFI (∞ Si/Ti-PMFI) apparently retained the long-range ordering of the nanosheet layers in 2D MFI (P) since two diffraction peaks at $2\theta = 1.57^\circ$ and 4.65° , respectively, were observed. The pillaring of 2D MFI (P) with the solvent consisting of TBOT precursor resulted in different features in the low-angle XRD spectra of the Si/Ti-PMFI samples. For example, the low-angle XRD peak was absent in 0Si/Ti-PMFI (Fig. 1(A)), which hints the failure in pillaring of 2D MFI (P) zeolite with TBOT alone. With a decrease in TBOT quantity in the mixed TEOS/TBOT solvent in the pillaring process, the low-angle XRD peaks start to appear and their intensities increase. For example, 5Si/Ti-PMFI shows an obscure peak at $2\theta \sim 1.5^\circ$ compared to 20Si/Ti-PMFI and 40Si/Ti-PMFI, but more obvious than 0Si/Ti-PMFI. The 20Si/Ti-PMFI and 40Si/Ti-PMFI show the clear low-angle diffraction peaks in Fig. 1(A), indicating that the long-range structural ordering of MFI nanosheet layers was well preserved in these two Si/Ti-PMFI samples.

The XRD data in Fig. 1 suggest that TBOT alone is not an efficient solvent for synthesis of Si/Ti-PMFI zeolite. On the contrary, TEOS is very suitable for pillaring of 2D MFI (P) zeolite. In the mixed TEOS/TBOT solvent, the decrease in the concentration of TBOT leads to an improvement in preservation of the long-range structural ordering of zeolite nanosheets in Si/Ti-PMFI. It should be noted that TBOT is bulkier in the molecular configuration and more viscous than TEOS. Both factors could contribute to the failure in synthesis of pillared 0Si/Ti-PMFI with



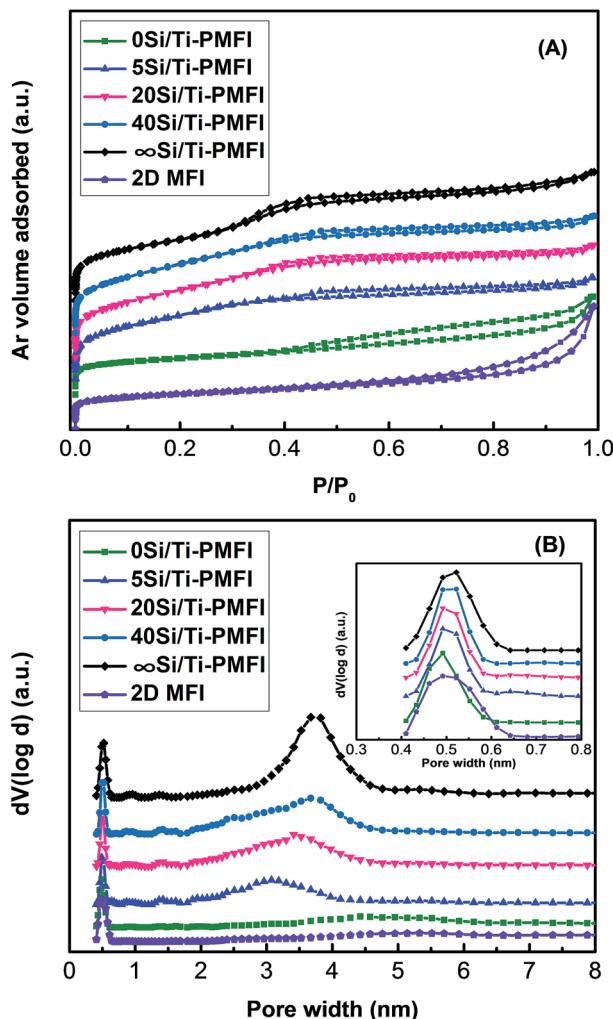


Fig. 2 Ar isotherms (A) and the corresponding NLDFT pore size distributions (B) of 0Si/Ti-PMFI, 5Si/Ti-PMFI, 20Si/Ti-PMFI, 40Si/Ti-PMFI, ∞ Si/Ti-PMFI, and 2D MFI samples, respectively. The inset image in (B) shows the NLDFT micropore size distributions.

mixed TEOS/TBOT solvent demonstrates that the molecular size of TBOT is not a barrier for incorporation of Ti-moiety into Si/Ti-PMFI. The mixed TEOS/TBOT solvent has lower viscosity than TBOT, which facilitates the infiltration of the solvent into gallery space of 2D MFI (P) and thus results in the successful synthesis of Si/Ti-PMFI zeolites. Similar results have been

reported for synthesis of titanosilicate pillared MCM-36 by pillaring of MCM-22(P) or ERB-1³⁹ and titanosilicate pillared TS-1 lamellar zeolites⁴¹ using the mixed TEOS/TBOT solvent.

Textural properties of Si/Ti-PMFI zeolites

Fig. 2 shows the Ar isotherms and pore size distributions of the Si/Ti-PMFI zeolites. At the relatively low pressures ($P/P_0 < 0.02$) in Fig. 2(A), all the Si/Ti-PMFI samples together with 2D MFI have similar Ar uptakes, which indicates all the presently studied zeolite samples have similar microporosity. The micropore surface areas (S_{micro}) and volumes (V_{micro}) in Table 1 confirm the similarity in microporosity of these zeolites. At a higher relative pressure range ($0.02 < P/P_0 < 0.9$), the Ar adsorption volume was increased in the order of 2D MFI \sim 0Si/Ti-PMFI \ll 5Si/Ti-PMFI $<$ 20Si/Ti-PMFI $<$ 40Si/Ti-PMFI $<$ ∞ Si/Ti-PMFI. The increase in Ar uptake in this pressure range indicates the increasing of mesoporosity in these samples. The external surface areas (S_{ext}) and total pore volumes (V_t) of these samples in Table 1 indicate that the mesoporosity increases following the above rank of these samples.

The non-local density functional theory (NLDFT) pore size distributions in Fig. 2(B) further underscore the changes in the textural properties of the Si/Ti-PMFI zeolites. The plots in Fig. 2(B) were determined from the adsorption branches of Ar isotherms using cylindrical-type pore and silica adsorbent model. All the zeolite samples exhibit a similar strong and sharp peak centered at about 0.54 nm (inset in Fig. 2(B)), which is attributed to the 10-membered ring micropores of MFI zeolite.⁴⁶ Additionally, broad peaks in the range of 2–5 nm with different intensities are seen for these zeolite samples, reflecting the existence of different mesoporosities. The 0Si/Ti-PMFI has a broad and less intense peak, analogous to that of 2D MFI, suggesting the failure in pillaring of the 2D MFI (P) using TBOT alone. The peak corresponding to mesopores in Fig. 2(B) becomes narrower and sharper when the material transits from 5Si/Ti-PMFI, 20Si/Ti-PMFI, and 40Si/Ti-PMFI to ∞ Si/Ti-PMFI. This reflects the increase in mesopore volume and uniformity of mesopore sizes with decreasing TBOT content in the mixed solvent in the synthesis.

The textural property analysis of Si/Ti-PMFI zeolites indicates that the preservation of mesoporosity (*i.e.*, the ordered layered structure of the 2D MFI (P)) is directly correlated to the TBOT content in the mixed TEOS/TBOT solvent. The similarity in micropore sizes and volumes of these Si/Ti-PMFI zeolites to

Table 1 Textural properties of 2D MFI and Si/Ti-PMFI zeolites determined from Ar isotherms

Zeolite	V_{micro}^a ($\text{cm}^3 \text{g}^{-1}$)	S_{micro}^a ($\text{m}^2 \text{g}^{-1}$)	S_{ext}^a ($\text{m}^2 \text{g}^{-1}$)	V_t^b ($\text{cm}^3 \text{g}^{-1}$)	V_{meso}^c ($\text{cm}^3 \text{g}^{-1}$)	S_{BET}^d ($\text{m}^2 \text{g}^{-1}$)
2D MFI	0.063	141	169	0.28	0.21	310
0Si/Ti-PMFI	0.058	88	195	0.39	0.33	284
5Si/Ti-PMFI	0.087	125	420	0.44	0.35	547
20Si/Ti-PMFI	0.067	100	476	0.45	0.38	576
40Si/Ti-PMFI	0.079	84	505	0.47	0.40	589
∞ Si/Ti-PMFI	0.060	89	520	0.54	0.48	609

^a Determined from *t*-plot method. ^b Determined by NLDFT method. ^c Determined from $V_{\text{meso}} = V_t - V_{\text{micro}}$. ^d Determined from Brunauer, Emmett, and Teller (BET) method.

Table 2 Composition analysis for 2D MFI and Si/Ti-PMFI zeolites

Zeolite	Bulk composition			Composition of pillars in zeolites				Si/Ti ratio ^f	TEOS/TBOT ratio ^g
	Si/Al ratio ^a	Si/Ti ratio ^a	SiO ₂ ^b (mmol g ⁻¹)	SiO ₂ pillar ^c (mmol g ⁻¹)	SiO ₂ pillar ^d (wt%)	TiO ₂ pillar ^e (wt%)			
2D MFI	49	—	16.3	0	0	0	—	—	—
0Si/Ti-PMFI	48	4.3	16.0	0	0	31.0	0	0	0
5Si/Ti-PMFI	59	22.6	19.7	3.3	17	6.0	4	5	
20Si/Ti-PMFI	69	61.0	23.0	6.7	29	2.2	18	20	
40Si/Ti-PMFI	74	118.0	25.0	8.3	34	1.1	40	40	
∞Si/Ti-PMFI	63	—	21.0	4.7	22	0	∞	∞	

^a Determined by ICP-OES analysis. ^b SiO₂ concentration in zeolite calculated from Si/Al ratio. ^c Concentration of SiO₂ pillar in the Si/Ti-PMFI zeolite, calculated from the difference of SiO₂ concentration in pillared zeolite from that in 2D MFI. ^d Weight percentage of SiO₂ units that constitute of pillars in the Si/Ti-PMFI zeolite. ^e Weight percentage of TiO₂ units that constitute of pillars in the pillared zeolite. ^f Molar ratio of Si/Ti in the pillars, evaluated by (weight of SiO₂ in pillars/molecular weight of SiO₂)/(weight of TiO₂ in pillars/molecular weight of TiO₂). ^g Molar ratio of TEOS/TBOT in the mixed solvent for pillaring of 2D MFI.

those of 2D MFI suggests that neither TBOT nor TEOS influences micropores of MFI zeolite. It has been reported that both of these molecules are too bulky to enter the 10-membered rings of MFI zeolite.^{47,48} The presence of significant mesoporosity in 20Si/Ti-PMFI and 40Si/Ti-PMFI zeolites demonstrates that both TEOS and TBOT can enter the galleries of 2D MFI (P) to form titanosilicate pillars. The higher mesopore volume and external surface area of Si/Ti-PMFI zeolites, obtained by increasing the TEOS/TBOT molar ratio in the mixed solvent, confirm the increase in intercalation efficiency of titanosilicate pillars into 2D MFI (P) when the viscosity of the mixed solvent is reduced in the synthesis.

Composition of pillars in Si/Ti-PMFI zeolites

The Si, Al, and Ti compositions of the Si/Ti-PMFI and 2D MFI zeolites were studied, and the results are included in Table 2. By direct comparison of elemental compositions between Si/Ti-PMFI and 2D MFI zeolites, the compositions of the pillars introduced were evaluated. As shown in Table 2, 2D MFI zeolite has a Si/Al ratio of 49, corresponding to 16.3 mmol SiO₂ units per gram of 2D MFI. The Si/Al of ∞Si/Ti-PMFI was increased to 63 (equivalent to 21 mmol SiO₂ per gram of zeolite). The increase of Si/Al ratio from 49 to 63 corresponds to 4.7 mmol SiO₂ per gram of 2D MFI incorporated as pillars in ∞Si/Ti-PMFI. The weight fraction of SiO₂ pillars can then be evaluated, which constitutes of 22 wt% of ∞Si/Ti-PMFI. Similar analysis was conducted for Si/Al ratio of 74 in 40Si/Ti-PMFI, 69 in 20Si/Ti-PMFI, 59 in 5Si/Ti-PMFI, and 48 in 0Si/Ti-PMFI zeolite samples, respectively. Table 2 lists that 34 wt%, 29 wt%, 17 wt%, and 0 wt% of the Si/Ti-PMFI samples are SiO₂ pillars from TEOS solvent in the pillaring of 2D MFI (P) zeolite.

The Si/Ti ratio of each sample in Table 2 indicates that 1.1 wt%, 2.2 wt%, 6.0 wt%, and 31.0 wt% of TiO₂ units exist in the 40Si/Ti-PMFI, 20Si/Ti-PMFI, 5Si/Ti-PMFI, and 0Si/Ti-PMFI zeolites, respectively, which were formed from TBOT in the pillaring process. The Si/Ti molar ratio in the pillars of Si/Ti-PMFI samples (if we assume all incorporated Si- and Ti-species formed as pillars), therefore, can be calculated. As shown in Table 2, the weight gains due to Si- and Ti-species from TEOS and

TBOT solvents, respectively, increase with increasing TEOS or TBOT concentration in the mixed solvent. Compared to the TEOS/TBOT ratio, the Si/Ti ratio in the pillars of the resultant Si/Ti-PMFI zeolites is nearly the same or slightly lower. This indicates that TBOT has comparable or slightly higher tendency than TEOS to be intercalated into the multilamellar MFI structure.

Coordination environment of Ti-species in Si/Ti-PMFI zeolites

The DR UV-Vis spectra (Fig. 3) were used to identify the coordination environment of Ti-species in the Si/Ti-PMFI zeolites. For comparison, the DR UV-Vis spectra of P25 titania and TS-1 zeolite are included. The absence of obvious absorption band in ∞Si/Ti-PMFI is consistent with the absence of Ti-species in this zeolite. The framework Ti-species in TS-1 result in an obvious absorption band centered at ~220 nm, which is assigned to the charge transfer from O²⁻ to Ti⁴⁺ in the tetrahedral (T_d) framework Ti-sites.^{39,49} The P25 exhibits an absorption band at ~280 nm that is

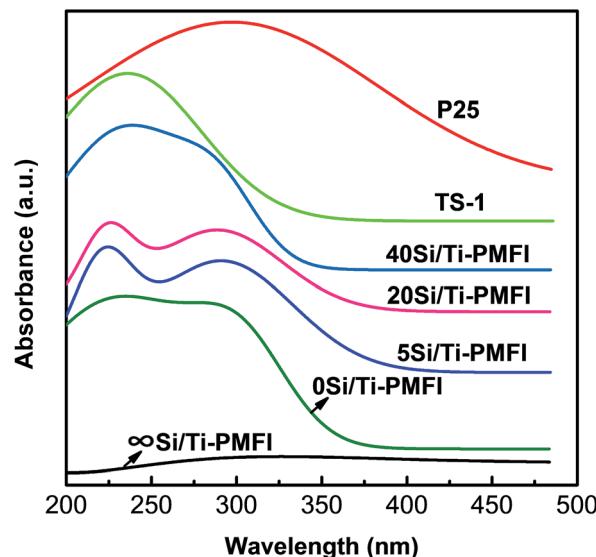


Fig. 3 DR UV-Vis spectra of 0Si/Ti-PMFI, 5Si/Ti-PMFI, 20Si/Ti-PMFI, 40Si/Ti-PMFI, and ∞Si/Ti-PMFI samples, respectively. P25 and TS-1 samples are shown for comparison.



assigned to the charge transfer from O^{2-} to Ti^{4+} in an octahedral (O_h) coordination.^{39,49} Using P25 and TS-1 as bases, the coordination environment of Ti-species in Si/Ti-PMFI zeolites was analyzed.

Fig. 3 shows that the Si/Ti-PMFI samples have both absorption bands of P25 and TS-1, but the relative intensity of these two bands varies. The 0Si/Ti-PMFI sample has comparable absorption bands at ~ 220 nm and ~ 280 nm, but both peaks are blunt. This indicates that Ti-species in 0Si/Ti-PMFI zeolite have both tetrahedral and octahedral coordination, which is consistent with the presence of anatase phase indicated by XRD data above. The tetrahedrally coordinated Ti-species could be due to the formation of titanosilicate on the surface of 2D MFI (P) under assistance of C_{22-6-6} template in 2D MFI (P). The absorption band centered at ~ 220 nm becomes stronger and sharper with the transition of the sample from 0Si/Ti-PMFI to 5Si/Ti-PMFI, 20Si/Ti-PMFI, and 40Si/Ti-PMFI, respectively. Oppositely, the absorption band centered at ~ 280 nm decreases sequentially across these samples. As discussed for 0Si/Ti-PMFI, the presence of both bands indicates that both types of Ti-coordination exist in these zeolites. With an increase in TEOS content in the mixed solvent, more Ti-species are incorporated with tetrahedral coordination in Si/Ti-PMFI zeolites. For 40Si/Ti-PMFI zeolite, the majority of Ti-species are in the framework tetrahedral structure, similar to that of TS-1. The DR UV-Vis spectra suggest that Ti-species in the titanosilicate pillars are very likely crystallized into a MFI zeolite (or TS-1) structure. The tendency for building such a MFI-like pillar structure increases with increasing TEOS (or decreasing TBOT) content in the mixed solvent in the synthesis.

Morphology of Si/Ti-PMFI zeolites

To show the morphology of the Si/Ti-PMFI zeolites, we selected 20Si/Ti-PMFI sample and examined its morphology under TEM. The 2D MFI (P) sample was observed under TEM for comparison. The 2D MFI (P) (Fig. 4(A)) exhibits an ordered arrangement of lamellar MFI nanosheets. The lattice fringe can be clearly seen in the high-resolution TEM (HR-TEM) image of this sample (Fig. 4(B)).

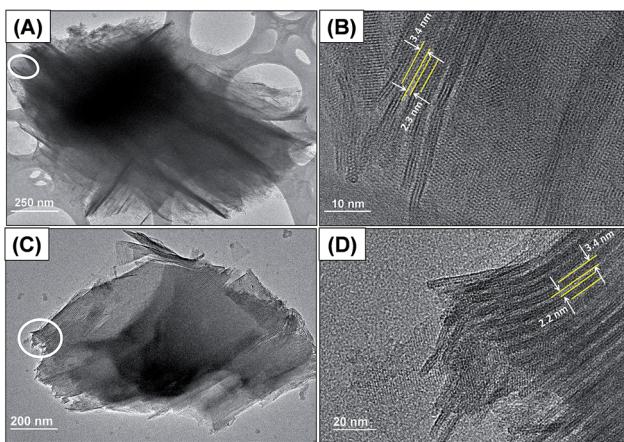


Fig. 4 TEM images of the synthesized 2D MFI (P) (A and B) and 20Si/Ti-PMFI (C and D) samples ((B) and (D) show the high-resolution TEM (HR-TEM) images of the marked areas in (A) and (C), respectively).

The multilamellar stacking of MFI nanosheets is composed of three pentasil sheets that are related to the 1.5 unit-cell dimension along the *b*-axis direction. The thickness of the single MFI zeolite nanosheet is around 3.4 nm, and the distance between neighboring MFI layers is ~ 2.3 nm. Fig. 4(C) illustrates the plate-like particles of 20Si/Ti-PMFI. The HR-TEM image in Fig. 4(D) indicates that the incorporation of titanosilicate pillars into the layered space of 2D MFI (P) did not lead to collapse of the MFI zeolite nanosheets. The distance between the neighboring MFI layers was also ~ 2.3 nm. The TEM observation confirms the interlamellar space between the MFI layers was preserved in the Si/Ti-PMFI zeolites.

Adsorption and photocatalytic performance of Si/Ti-PMFI zeolites

The removal of MO from water was used to evaluate the performance of Si/Ti-PMFI catalysts in adsorptive photocatalysis. The MO degradation rate (η) was indicated by the following equation:

$$\eta = \frac{C_0 - C}{C_0} = \frac{A_0 - A}{A_0} \times 100\%$$

where C_0 and C are the concentrations of MO (g L^{-1}) before catalyst addition and at local reaction time, respectively. A_0 and A are the corresponding absorbencies of MO aqueous solution under both conditions. Fig. 5(A) and (C) show the MO degradation rate *versus* reaction time for the Si/Ti-PMFI catalysts. The degradation of MO over TS-1 and P25 catalysts was also studied and shown in these figures. The influence of acidity from the framework Al-O(H)-Si sites in MFI layers on MO degradation was examined over H^+ -form Si/Ti-PMFI zeolites (Fig. 5(B) and (D)).

As shown in Fig. 5(A), the addition of P25 catalyst into the MO aqueous solution caused $\sim 9\%$ decrease in UV-Vis absorbance (or MO concentration), which suggests that P25 has low external surface area for MO adsorption in the reaction. Upon light irradiation, the MO degradation increased with reaction time, suggesting that P25 is an active catalyst for this reaction. TS-1 catalyst also showed almost similar adsorption of MO from the solution to that of P25, which means that the micropores of TS-1 are too small to allow MO molecules enter into them. The degradation rate of MO over TS-1 is much smaller compared to P25. This is mainly caused by the low number of Ti-sites in TS-1 (Si/Ti ratio = 50) catalyst. In comparison to P25 and TS-1 catalysts, all the Si/Ti-PMFI samples showed obvious adsorption that contributes to the degradation of MO molecules. For example, the 0Si/Ti-PMFI zeolite led to 26% MO degradation due to adsorption and 5.8% degradation after 4 hours of photocatalytic reaction. Similarly, 5Si/Ti-PMFI, 20Si/Ti-PMFI, and 40Si/Ti-PMFI resulted in 44%, 49%, and 47% of MO adsorption and 5.9%, 12.9%, and 8.2% of MO degradation after 4 hours of reaction, respectively. It should be noted that 2 hours of mixing time before the light irradiation has allowed the adsorption to reach the saturation. The 2 hours of mixing time in the present study guaranteed the adsorption event is completed before the photocatalytic event takes place.



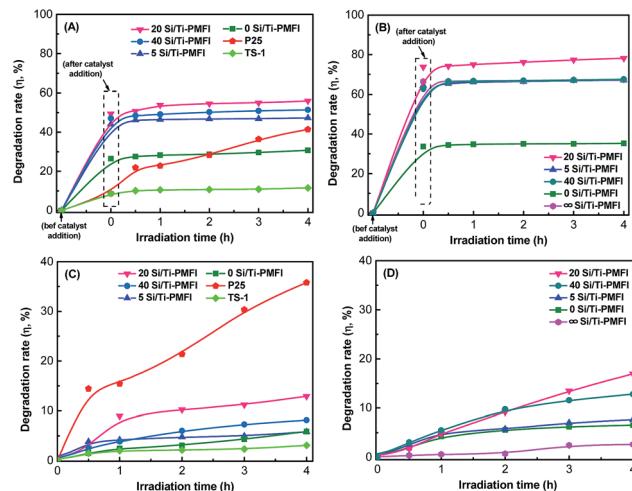


Fig. 5 Methyl orange degradation rate (η) versus irradiation time over 0Si/Ti-PMFI, 5Si/Ti-PMFI, 20Si/Ti-PMFI, and 40Si/Ti-PMFI catalysts in the absence (A) and presence of Al-O(H)-Si sites (B), respectively. (C) and (D) display the photocatalytic contribution to the MO degradation over the catalysts shown in (A) and (B), respectively. Catalytic performances over P25 and TS-1 ((A) and (C)) and ∞ Si/Ti-PMFI ((B) and (D)) were shown for comparison (0.02 g catalyst, room temperature, and 500 rpm stirring speed).

The performance of Si/Ti-PMFI zeolite (in the absence of Al-O(H)-Si acid sites) in photocatalytic degradation of MO molecules is directly related to two factors: the number of Ti-active sites and the accessibility of samples to bulky MO molecules. The composition analysis (Table 2) shows that the concentration of Ti-species in Si/Ti-PMFI zeolites increases with increasing TBOT quantity in the mixed TEOS/TBOT solvent in the synthesis process. The mesoporosity of the Si/Ti-PMFI zeolites (Fig. 2 and Table 1), however, exhibits an opposite trend. The consistency of the MO degradation rate with the mesoporosity of Si/Ti-PMFI zeolites suggests that mesoporosity plays more obvious role in MO degradation over these catalysts. The presence of mesoporosity enhanced the mass transport of bulky MO molecules into or product out of the pillared Si/Ti-PMFI zeolites, and thus facilitated the adsorption of MO molecules in the degradation process. The slightly higher degradation rate of MO over 20Si/Ti-PMFI than that of 40Si/Ti-PMFI should be due to the higher number of Ti-sites in the former sample since both catalysts have similar textural properties. The concentration of Ti sites in the Si/Ti-PMFI zeolites is \sim 20–60 times lower than that of P25, and thus the lower photocatalytic activity was observed in these zeolite-based catalyst materials.

The presence of acid sites in the Si/Ti-PMFI zeolites apparently improved the adsorption capacities of these catalysts in MO degradation reactions. As shown in Fig. 5(B), 33%, 63%, 74%, and 64% of MO degradations due to adsorption were observed in 0Si/Ti-PMFI, 40Si/Ti-PMFI, 20Si/Ti-PMFI, and 5Si/Ti-PMFI, respectively. In comparison to MO adsorption over Si/Ti-PMFI catalysts in the absence of Al-O(H)-Si acid sites, the adsorption was enhanced by \sim 20–40%. The photocatalytic degradation rate also increased (Fig. 5(D)). For example, the MO

degradation reached 6.5%, 12.8%, 16.9% and 7.6%, respectively, over 0Si/Ti-PMFI, 40Si/Ti-PMFI, 20Si/Ti-PMFI, and 5Si/Ti-PMFI catalysts after 4 hours of reaction. For comparison, the MO degradation over ∞ Si/Ti-PMFI zeolite (without Ti-site but with Al-O(H)-Si acid sites) was tested, which showed that 66% degradation was due to catalyst adsorption and had almost no photocatalytic degradation activity (\sim 2%) compared to the Ti-containing Si/Ti-PMFI zeolites. These results indicate that Al-O(H)-Si acid sites promoted adsorption and catalysis capabilities of Si/Ti-PMFI catalysts in MO degradation.

In photocatalytic degradation of organic molecules, it has been reported that pH value of an aqueous solution affects the reaction rate due to generation of the oxidizing species ($\cdot\text{OH}$, $\text{O}_2^{\cdot-}$, HO_2^{\cdot} , etc.) in the photocatalytic reaction systems.^{50,51} An aqueous solution with mild acidity (pH \sim 5) has been reported to enable fast MO photocatalytic degradation over metal oxide catalysts.^{52,53} The Al-O(H)-Si acid sites in Si/Ti-PMFI zeolites form hydronium ions in aqueous MO solution, and thus are expected to influence the progress of the adsorption and reaction as observed on MO degradation in acidic solution. The enhancement in MO adsorption on the proton-form Si/Ti-PMFI might be due to the electrostatic interaction between the catalyst and MO molecules. The adsorption assisted photocatalysis in Si/Ti-PMFI zeolites is expected to be applicable to other types of reactions that deal with bulky molecules such as photocatalytic water or air treatment in environmental remedy.

Conclusions

The titanosilicate pillared MFI (Si/Ti-PMFI) zeolite was prepared by sequential dispersion of the multilamellar MFI precursor (2D MFI (P)) in the mixed TEOS and TBOT solvent, hydrolysis of entrapped solvent in 2D MFI (P), and calcination to form pillared zeolite structure. The Ti-species in the pillars of Si/Ti-PMFI zeolites enabled efficient photocatalytic reactions for MO degradation. The framework Al-O(H)-Si acid sites in MFI layers further accelerated the MO degradation rates by providing mild acidity for the reaction. The co-existence of Ti-species, Al-O(H)-Si acid sites, and mesoporosity in Si/Ti-PMFI zeolites enabled efficient adsorption and reaction of bulky molecules in photocatalysis. The synthesis of Si/Ti-PMFI zeolites diversifies the structural, physicochemical, and catalytic properties of 2D lamellar zeolites. The Si/Ti-PMFI lamellar zeolites are expected to be efficient catalysts for a range of reactions in adsorptive photocatalytic water and/or air treatments for environmental remedies.

Acknowledgements

The authors gratefully acknowledge financial support from the Army Research Laboratory (5284201) and National Science Foundation (NSF-CBET 1264599, 1351384). We acknowledge the support of Maryland NanoCenter and its AIMLab. Laleh Emdadi thanks for the Harry K. Wells Energy Research Fellowship from University of Maryland Energy Research Center (UMERC) to support her research.



Notes and references

1 A. Bhan and E. Iglesia, *Acc. Chem. Res.*, 2008, **41**, 559–567.

2 J. Čejka and S. Mintova, *Catal. Rev.*, 2007, **49**, 457–509.

3 A. Corma, *Chem. Rev.*, 1997, **97**, 2373–2419.

4 C. S. Cundy and P. A. Cox, *Chem. Rev.*, 2003, **103**, 663–701.

5 W. Zhang, J. Xie, W. Hou, Y. Liu, Y. Zhou and J. Wang, *ACS Appl. Mater. Interfaces*, 2016, **8**, 23122–23132.

6 K. Egeblad, C. H. Christensen, M. Kustova and C. H. Christensen, *Chem. Mater.*, 2008, **20**, 946–960.

7 J. Perez-Ramirez, C. H. Christensen, K. Egeblad, C. H. Christensen and J. C. Groen, *Chem. Soc. Rev.*, 2008, **37**, 2530–2542.

8 Y. Tao, H. Kanoh, L. Abrams and K. Kaneko, *Chem. Rev.*, 2006, **106**, 896–910.

9 J. Xie, H. Wen, W. Zhang, Y. Zhou and J. Wang, *CrystEngComm*, 2016, **18**, 1164–1173.

10 L.-H. Chen, X.-Y. Li, J. C. Rooke, Y.-H. Zhang, X.-Y. Yang, Y. Tang, F.-S. Xiao and B.-L. Su, *J. Mater. Chem.*, 2012, **22**, 17381–17403.

11 K. Moller and T. Bein, *Chem. Soc. Rev.*, 2013, **42**, 3689–3707.

12 M. Choi, H. S. Cho, R. Srivastava, C. Venkatesan, D. H. Choi and R. Ryoo, *Nat. Mater.*, 2006, **5**, 718–723.

13 M. Choi, K. Na, J. Kim, Y. Sakamoto, O. Terasaki and R. Ryoo, *Nature*, 2009, **461**, 246–249.

14 W. Fan, M. A. Snyder, S. Kumar, P. S. Lee, W. C. Yoo, A. V. McCormick, R. L. Penn, A. Stein and M. Tsapatsis, *Nat. Mater.*, 2008, **7**, 984–991.

15 W. Fan, P. Wu, S. Namba and T. Tatsumi, *Angew. Chem., Int. Ed.*, 2004, **43**, 236–240.

16 J. C. Groen, T. Bach, U. Ziese, A. Donk, K. P. de Jong, J. A. Moulijn and J. Perez-Ramirez, *J. Am. Chem. Soc.*, 2005, **127**, 10792–10793.

17 H. Wang and T. J. Pinnavaia, *Angew. Chem., Int. Ed.*, 2006, **45**, 7603–7606.

18 J. O. Barth, A. Jentys, J. Kornatowski and J. A. Lercher, *Chem. Mater.*, 2004, **16**, 724–730.

19 C. T. Kresge, W. J. Roth, K. G. Simmons and J. C. Vartuli, (Mobil Oil Corporation) WO Patent 92/011935, 1992.

20 S. Maheshwari, E. Jordan, S. Kumar, F. S. Bates, R. L. Penn, D. F. Shantz and M. Tsapatsis, *J. Am. Chem. Soc.*, 2008, **130**, 1507–1516.

21 K. Na, M. Choi, W. Park, Y. Sakamoto, O. Terasaki and R. Ryoo, *J. Am. Chem. Soc.*, 2010, **132**, 4169–4177.

22 U. Diaz, *ISRN Chem. Eng.*, 2012, **2012**, 35.

23 M. V. Opanasenko, W. J. Roth and J. Čejka, *Catal. Sci. Technol.*, 2016, **6**, 2467–2484.

24 W. J. Roth, B. Gil, W. Makowski, B. Marszalek and P. Eliasova, *Chem. Soc. Rev.*, 2016, **45**, 3400–3438.

25 W. J. Roth, P. Nachtigall, R. E. Morris and J. Čejka, *Chem. Rev.*, 2014, **114**, 4807–4837.

26 A. Corma, U. Diaz, M. E. Domine and V. Fornés, *Angew. Chem., Int. Ed.*, 2000, **39**, 1499–1501.

27 A. Corma, U. Diaz, M. E. Domine and V. Fornés, *J. Am. Chem. Soc.*, 2000, **122**, 2804–2809.

28 L. Schreyeck, P. Caullet, J. C. Mougenel, J. L. Guth and B. Marler, *Microporous Mater.*, 1996, **6**, 259–271.

29 T. Moteki, W. Chaikittisilp, A. Shimojima and T. Okubo, *J. Am. Chem. Soc.*, 2008, **130**, 15780–15781.

30 P. Chlubná, W. J. Roth, H. F. Greer, W. Zhou, O. Shvets, A. Zukal, J. Čejka and R. E. Morris, *Chem. Mater.*, 2013, **25**, 542–547.

31 W. J. Roth, O. V. Shvets, M. Shamzhy, P. Chlubná, M. Kubů, P. Nachtigall and J. Čejka, *J. Am. Chem. Soc.*, 2011, **133**, 6130–6133.

32 J.-O. Barth, A. Jentys, E. F. Iliopoulos, I. A. Vasalos and J. A. Lercher, *J. Catal.*, 2004, **227**, 117–129.

33 J.-O. Barth, J. Kornatowski and J. A. Lercher, *J. Mater. Chem.*, 2002, **12**, 369–373.

34 J. Kornatowski, J. O. Barth and J. A. Lercher, in *Studies in Surface Science and Catalysis*, ed. S. Abdelhamid and J. Mietek, Elsevier, 2005, vol. 156, pp. 349–356.

35 A. Corma, V. Fornés, S. B. Pergher, T. L. M. Maesen and J. G. Buglass, *Nature*, 1998, **396**, 353–356.

36 M. A. Cambor, C. Corell, A. Corma, M.-J. Díaz-Cabañas, S. Nicolopoulos, J. M. González-Calbet and M. Vallet-Regí, *Chem. Mater.*, 1996, **8**, 2415–2417.

37 A. Corma, U. Diaz, V. Fornés, J. L. Jorda, M. Domine and F. Rey, *Chem. Commun.*, 1999, 779–780.

38 J. Kim, J. Chun and R. Ryoo, *Chem. Commun.*, 2015, **51**, 13102–13105.

39 F. Jin, C.-C. Chang, C.-W. Yang, J.-F. Lee, L.-Y. Jang and S. Cheng, *J. Mater. Chem. A*, 2015, **3**, 8715–8724.

40 F. Jin, S. Huang, S. Cheng, Y. Wu, C.-C. Chang and Y.-W. Huang, *Catal. Sci. Technol.*, 2015, **5**, 3007–3016.

41 J. Přech, P. Eliášová, D. Aldhayan and M. Kubů, *Catal. Today*, 2015, **243**, 134–140.

42 J. Prech, R. E. Morris and J. Čejka, *Catal. Sci. Technol.*, 2016, **6**, 2775–2786.

43 L. Emdadi, Y. Wu, G. Zhu, C.-C. Chang, W. Fan, T. Pham, R. F. Lobo and D. Liu, *Chem. Mater.*, 2014, **26**, 1345–1355.

44 D. Liu, A. Bhan, M. Tsapatsis and S. Al Hashimi, *ACS Catal.*, 2011, **1**, 7–17.

45 A. Thangaraj, M. J. Eapen, S. Sivasanker and P. Ratnasamy, *Zeolites*, 1992, **12**, 943–950.

46 L. Grajciar, O. Bludský, W. J. Roth and P. Nachtigall, *Catal. Today*, 2013, **204**, 15–21.

47 S. Kikuchi, R. Kojima, H. Ma, J. Bai and M. Ichikawa, *J. Catal.*, 2006, **242**, 349–356.

48 S. Zheng, H. R. Heydenrych, A. Jentys and J. A. Lercher, *J. Phys. Chem. B*, 2002, **106**, 9552–9558.

49 P. Wu and T. Tatsumi, *Catal. Surv. Asia*, 2004, **8**, 137–148.

50 C.-Y. Chang, Y.-H. Hsieh, K.-Y. Cheng, L.-L. Hsieh, T.-C. Cheng and K.-S. Yao, *Water Sci. Technol.*, 2008, **58**, 873–879.

51 R. G. Zepp, B. C. Faust and J. Hoigne, *Environ. Sci. Technol.*, 1992, **26**, 313–319.

52 F. Zhang, J. Zhao, T. Shen, H. Hidaka, E. Pelizzetti and N. Serpone, *Appl. Catal. B*, 1998, **15**, 147–156.

53 A. Zyoud, A. Zu'bi, M. H. S. Helal, D. Park, G. Campet and H. S. Hilal, *J. Environ. Health Sci. Eng.*, 2015, **13**, 1–10.

