



Cite this: *Nanoscale*, 2017, 9, 11574

Ag–CuO–ZnO metal–semiconductor multiconcentric nanotubes for achieving superior and perdurable photodegradation†

Kaichen Xu,^{‡a,b} Jiagen Wu,^{‡b} Chuan Fu Tan,^a Ghim Wei Ho,^{‡a} Ang Wei*^b and Minghui Hong^{‡a}

Solar energy represents a robust and natural form of resource for environment remediation *via* photocatalytic pollutant degradation with minimum associated costs. However, due to the complexity of the photodegradation process, it has been a long-standing challenge to develop reliable photocatalytic systems with low recombination rates, excellent recyclability, and high utilization rates of solar energy, especially in the visible light range. In this work, a ternary hetero-nanostructured Ag–CuO–ZnO nanotube (NT) composite is fabricated *via* facile and low-temperature chemical and photochemical deposition methods. Under visible light irradiation, the as-synthesized ZnO NT based ternary composite exhibits a greater enhancement (~300%) of photocatalytic activity than its counterpart, Ag–CuO–ZnO nanorods (NRs), in pollutant degradation. The enhanced photocatalytic capability is primarily attributed to the intensified visible light harvesting, efficient charge carrier separation and much larger surface area. Furthermore, our as-synthesised hybrid ternary Ag–CuO–ZnO NT composite demonstrates much higher photostability and retains ~98% of degradation efficiency even after 20 usage cycles, which can be mainly ascribed to the more stable polar planes of ZnO NTs than those of ZnO NRs. These results afford a new route to construct ternary heterostructured composites with perdurable performance in sewage treatment and photocorrosion suppression.

Received 9th May 2017.

Accepted 8th July 2017

DOI: 10.1039/c7nr03279j

rsc.li/nanoscale

1. Introduction

The extensive utilization of fossil fuels, deterioration of the environment and global warming advocate increasing the demand to seek sustainable green strategies based on energy conversion.^{1,2} Fortunately, natural resources, such as wind,³ water wave⁴ and solar energy,^{5,6} endow human beings with abundant pathways to access renewable and clean resources. In particular, in order to address the primary issue of water pollution, solar energy coupled with novel semiconductor photocatalysts has attracted considerable research interest *via* mimicking natural photosynthesis.^{7,8} Relying on the photo-generation of electron–hole pairs in light-harvesting systems, a

photodegradation reaction takes place, which is capable of eliminating organic contaminants in waste streams.^{9,10} In recent years, one-dimensional (1D) nanostructure-based photocatalysts have attracted tremendous research attention owing to their high aspect ratio and excellent carrier transport performance.^{11–15} Among the variety of 1D metal–oxide based photocatalysts, zinc oxide (ZnO) nanorods (NRs), as an n-type single crystal semiconductor, have been envisioned as a promising and efficient candidate in environmental remediation systems owing to its fascinating features, including non-toxicity, high electron mobility, strong oxidation ability, environmentally friendly property and good photocatalytic capability.^{16–18}

Nevertheless, there are several intrinsic drawbacks of ZnO NRs, restricting their practical application in water purification. For instance, the wide bandgap (~3.37 eV) of ZnO results in its sole absorption of UV light, which accounts for a limited portion (~4%) of the solar spectrum.¹⁹ Furthermore, the electron–hole separation efficiency of ZnO is very poor, leading to the degradation of its photocatalytic efficiency.²⁰ In addition, due to the photo-generated holes, the high rate of photocorrosion in nanosized ZnO materials gives rise to their poor photo-stability in aqueous solution.²¹ These critical

^aDepartment of Electrical and Computer Engineering, National University of Singapore, 4 Engineering Drive 3, 117576, Singapore. E-mail: elehmh@nus.edu.sg

^bKey Laboratory for Organic Electronics and Information Displays & Institute of Advanced Materials (IAM), National Synergistic Innovation Center for Advanced Materials (SICAM), Nanjing University of Posts & Telecommunications, 9 Wenyuan Road, Nanjing 210023, China. E-mail: iamawei@njupt.edu.cn

†Electronic supplementary information (ESI) available. See DOI: 10.1039/c7nr03279j

‡These authors contributed equally to this work.



issues primarily inhibit the photocatalytic performance of ZnO in environmental remediation. Therefore, in order to boost the photocatalytic capability of ZnO-based photocatalysts, tremendous effort has been devoted to extending their optical absorption range, boosting the separation efficiency of the photo-generated carriers and improving the photo-stability.^{8,18,22,23} One of the promising strategies is to couple noble metals, such as gold (Au) or silver (Ag), with ZnO nanostructures.^{18,24–26} The metal nanoparticles can facilitate light–matter interactions owing to the effective surface plasmon resonance, contributing to the enhanced absorption of visible light.^{27,28} Furthermore, the unique semiconductor–metal interfaces allow the formation of a Schottky barrier, which promotes the charge separation efficiency.^{29,30} Another intriguing method is to employ binary p–n heterostructures composed of narrow- and wide-bandgap ZnO semiconductors. p-Type narrow-bandgap semiconductors, such as copper oxide (CuO) or cuprous oxide (Cu₂O), can not only optimize the visible light activation, but also enable the opposite propagation of positive and negative charge carriers across the hetero-interface, giving rise to a high charge-separation efficiency.^{31,32}

However, the perdurable property of the ZnO-based photocatalysts has not been well realized, which is of paramount significance for their practical application in the removal of pollutants, especially for multiple uses, which would need photodegradation for a much longer time. Such highly recyclable and photo-stable photocatalytic systems require the composites to be provided with a strong anti-photocorrosion characteristic. Although various studies have been performed to overcome the limitation of photocorrosion *via* surface modification with a passive layer, such as carbon,³³ reduced graphene oxide,³⁴ graphene¹⁷ and polyaniline,³⁵ most of these studies conducted recycling photostability tests with fewer than five cycles.^{36,37} As a matter of fact, the performance of ZnO NRs-based systems suffers from significant degradation during the long and repeated photocatalytic process.²¹ This phenomenon can be ascribed to two main factors. First, the strong oxidative capability of photogenerated active oxygen species can oxidize carbon or organic composites, leading to the suppression of the photocatalytic stability.²¹ Second, conventional studies focus on the amelioration of ZnO NR-based photocatalysts. However, the high surface energy of (0001)-Zn and (000-1)-O for ZnO NRs makes them unstable during the photodegradation process.³⁸ In contrast, their counterpart ZnO nanotubes (NTs) have great potential to overcome the aforementioned issue owing to the lower surface energy (J m⁻²) of the polar surface (10–10).^{38,39} Moreover, the efficient carrier extraction and the larger surface area of ZnO NTs empower them to be promising backbones for practical photocatalysts.^{15,39,40}

In this work, a new type of 1D ternary photosynthetic composite for efficient and perdurable photodegradation is developed *via* facile low-temperature chemical and photochemical deposition approaches based on a unique metal–semiconductor system. In such ternary hybrids, vertically aligned ZnO NTs are employed as the building block. CuO and Ag

nanoparticles are then decorated onto the surface of ZnO NTs. The as-fabricated ternary Ag–CuO–ZnO NT arrays demonstrated enhanced photocatalytic performance (~300%) to decompose methyl orange (MO) and Rhodamine B (RhB) in water driven by visible light, as compared with that of ternary Ag–CuO–ZnO NRs arrays, binary Ag–ZnO (NTs & NRs) and CuO–ZnO (NTs & NRs) composites. This is ascribed to the synergistic effects of the intensified visible light harvesting, high charge-separation efficiency and large surface area of Ag–CuO–ZnO NTs systems. Furthermore, the hybrid structures of ternary ZnO NT-based composites exhibit excellent recyclability due to the strong anti-photocorrosion property of the ZnO NTs as well as the formation of CuO films as a protective layer. Such perdurable photodegradation capability of photocatalysts enables them to serve as a potential means for decomposing persistent organic species in environmental remediation.

2. Experimental details

The fabrication of ternary Ag–CuO–ZnO NT array composites consists of three main steps: the preparation of ZnO NT arrays and then photodeposition of CuO and Ag nanoparticles as illustrated in Fig. 1. The details of the fabrication procedures are as follows.

2.1. Preparation of ternary photocatalysts

2.1.1. Preparation of ZnO NTs arrays. To fabricate ZnO NTs arrays, a seed layer of ZnO is firstly made by a sol–gel approach. Briefly, anhydrous ethanol (40 ml), ethanol amine (0.5 M) and zinc acetate-2-hydrate (0.5 M) are mixed homogeneously. The solution is then spin-coated (300 rpm for 5 s and 2000 rpm for 20 s, respectively) on the glass surface, followed by annealing in air at 300 °C for 3 h.¹⁸ After the sample is naturally cooled to room temperature, the glass substrate is suspended in zinc nitrate hexahydrate (0.1 M) aqueous solution with hexamethylamine (0.1 M) at 95 °C for 2 h. It should be noted that CH₃COOH is added to tune the pH value of the solution to ~6.0. The substrate is then washed with deionized water and immersed in sodium hydrate aqueous solution (0.3 M, 50 mL) at 85 °C for 1 h. At last, the samples are cleaned with deionized water and dried at 120 °C for 0.5 h.³⁸

2.1.2. Fabrication of the binary CuO–ZnO NT composite. CuO nanoparticles are then decorated onto the ZnO NTs by the method of photochemical deposition. The ZnO NT-deposited glass substrate is immersed in a mixed methanol solution (50 mL) containing cupric nitrate hexahydrate (4 mM) and monarkite hexahydrate (4 mM). Subsequently, the samples are illuminated with UV light for 2 h at room temperature (wavelength 365 nm, power 48 W, working distance 0.2 m).¹⁸ The fabrication of binary CuO–ZnO NR arrays is also carried out by the same approach.

2.1.3. Preparation of the ternary Ag–CuO–ZnO NT composite. Ag nanoparticles are then deposited onto CuO–ZnO NTs by photochemical deposition. To be specific, silver nitrate (0.008 g) and monarkite hexahydrate (0.01 g) are dissolved in



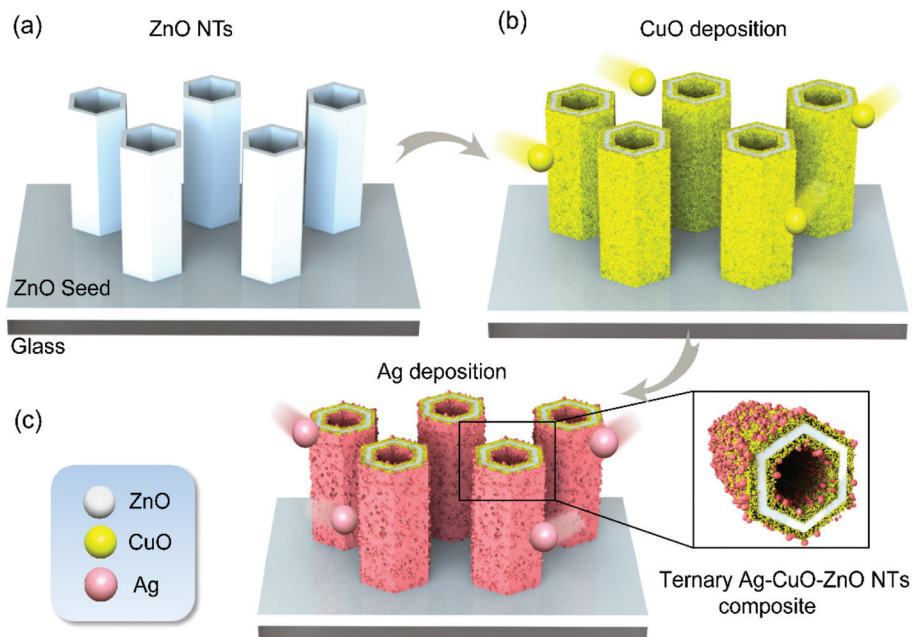


Fig. 1 Schematic diagram of the fabricated ternary Ag–CuO–ZnO NT composite. (a) Preparation of the ZnO NT array. (b) Deposition of CuO nanoparticles onto the ZnO NTs array. (c) Decoration of Ag nanoparticles onto the binary CuO–ZnO composite.

methanol (50 mL). Then, the as-fabricated CuO–ZnO NT substrate is immersed in the mixed methanol solution, which is illuminated with UV light (wavelength 365 nm, power 48 W, working distance 0.2 m) at room temperature for 2 h. Finally, the as-fabricated products are annealed in ambient air (400 °C for 3 h).¹⁸ The preparation of the ternary Ag–CuO–ZnO NR composite is also performed by the same procedure. Meanwhile, for comparison, the binary composites of CuO–ZnO NTs and NRs and Ag–ZnO NTs and NRs are also fabricated.

2.2. Characterization

The crystal structures of the hybrid ternary composites are studied by X-ray diffraction (XRD, Siemens D5005) analysis with CuK α radiation. The XPS spectra of the products are characterized by using a PerkinElmer PHI 5000 C ESCA system provided with a hemispherical electron energy analyzer. The surface morphology of the samples is revealed by scanning electron microscopy (SEM, S4800). Transmission electron microscopy (TEM) and high-resolution electron microscopy (HRTEM) images as well as selected area electron diffraction (SAED) patterns are obtained with a Tecnai G2 F20 S-TWIN. The UV-Vis absorption spectra are measured by using a UV-2401PC spectrometer. A Brunauer–Emmett–Teller (BET) approach is used to calculate the specific surface area tested by a specific surface area and pore size analyzer (2800P V-Sorb). The time-dependent photocurrent is measured on a three-electrode cell using Pt as the counter electrode and Ag/AgCl as the reference electrode at a fixed bias of 0.6 V and Na₂SO₄ (0.1 M) is applied as the electrolyte. A 300 W xenon arc lamp system (intensity: 100 mW cm⁻²) is used as the

irradiation source. Electrochemical impedance spectroscopy (EIS) tests are performed on a CHI-660D workstation (CH Instruments).

2.3. Photocatalytic activity

Rhodamine B (RhB) and methyl orange (MO) are two common pollutants in waste streams, which are chosen to compare the photocatalytic characteristics of Ag–CuO–ZnO NTs and Ag–CuO–ZnO NRs. Taking MO as an example, the samples are grown on a glass substrate with the same effective area (2.0 cm \times 2.0 cm) and are immersed in a MO solution (30 mL) at a concentration of 2×10^{-5} M. The mass of the photocatalysts for all photodegradation steps is in the range of 25 to 30 mg. A visible light source from 420 nm to 720 nm is achieved by applying a xenon lamp equipped with a 420 nm cut-off filter. Before the photodegradation, the MO solution is retained in the dark (30 min) to reach the adsorption–desorption equilibrium. Visible light with an intensity of ~ 50 mW cm⁻² is then focused at the sample surface. The UV-Vis spectrophotometer is applied every 10 min to determine the concentration of MO. After the measurements, irradiation is resumed. To make a comparison, the photocatalytic efficiency of the bare ZnO NTs and NRs, binary CuO–ZnO NTs and NRs as well as binary Ag–ZnO NTs and NRs is also characterized.

To demonstrate the recyclability of ternary ZnO-based composites, the as-fabricated products are placed in a MO solution (30 mL) at a concentration of 2×10^{-5} M driven by the irradiation of visible light for 20 usage cycles. For each round, the MO solution is catalyzed by Ag–CuO–ZnO NTs and NRs for 60 min and then its absorption peak at a wavelength of ~ 460 nm is measured. In the next cycle, the new solution takes



the place of the old one and then the same procedure is carried out.

3. Results and discussion

To reveal the surface morphology of the ternary Ag–CuO–ZnO photocatalysts, the as-synthesised nanostructures are characterized by SEM. As can be observed in Fig. 2a, obviously, the wurtzite ZnO forms a tubular structure with an average tube wall thickness of 8 nm, an external diameter of around 50 nm and a length of around 200 nm. It should be noted that the external diameter and length of ZnO NTs are similar to those of ZnO NRs (Fig. S1†). Furthermore, the vertically aligned ZnO NTs array is perpendicular to the glass substrate in the *c*-axis direction with a smooth surface (inset of Fig. 2a). Noticeably, the pH value plays a significant role in influencing the morphology of ZnO NTs. The fabrication procedure with the lower pH values leads to a non-continuous ZnO NTs array, while the higher pH values result in failure to form tubular structures.³⁸ Employing the as-fabricated ZnO NT array as the building block, CuO nanoparticles are then photodeposited onto the surface of ZnO NTs to allow the formation of the binary CuO–ZnO NT composite driven by UV light irradiation (Fig. 2b). Under the sustained UV irradiation, more CuO particles are obtained, leading to a thicker and relatively rougher CuO layer covered on the surfaces of ZnO NTs (inset of Fig. 2b). After the decoration of Ag nanoparticles, grainy materials are observed on the surface of CuO–ZnO NTs (Fig. 2c and c, inset), indicating the formation of ternary Ag–CuO–ZnO NT composites. To make a comparison and exhibit the superior photocatalytic performance of ZnO NTs-based composites, ZnO NR-based compounds are also synthesized using similar procedures and their surface morphology is shown in Fig. S1.†

The morphology of the ternary ZnO NT-based composite is further elucidated by TEM analyses. Fig. 3a depicts a characteristic TEM image of hetero-nanostructured Ag–CuO–ZnO NTs. As can be seen, ZnO NTs exhibit 1D structures with a well-defined hollow morphology, and serve as the building block

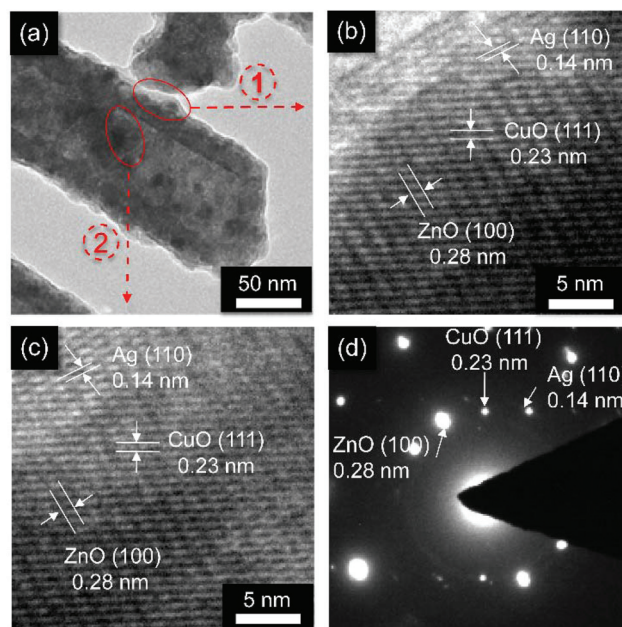


Fig. 3 (a) TEM images of the ternary Ag–CuO–ZnO NT composite. (b) and (c) HRTEM images of the areas indicated by red circles in (a), corresponding to the outer and inner surfaces, respectively. (d) The SAED pattern of the nanostructures illustrated in (a).

for the formation of core–shell architectures. Fig. 3b and c show high-resolution TEM (HRTEM) images of the ZnO NT core and CuO and Ag shells in the inner and outer surfaces. The lattice spacing of around 0.28 nm, 0.23 nm and 0.14 nm represents the (100) plane of ZnO, the (111) plane of CuO and the (110) plane of Ag, respectively. The corresponding SAED patterns further confirm the successful preparation of Ag–CuO–ZnO NTs (Fig. 3d). These results indicate that vertically aligned ZnO NTs are obtained with the CuO and Ag nanoparticles existing on the outer and inner surfaces.

The crystallographic structures of the ternary composites are investigated by XRD analyses. It is seen from Fig. 4a that the diffraction peaks of wurtzite ZnO (JCPDS no. 36-1451) can

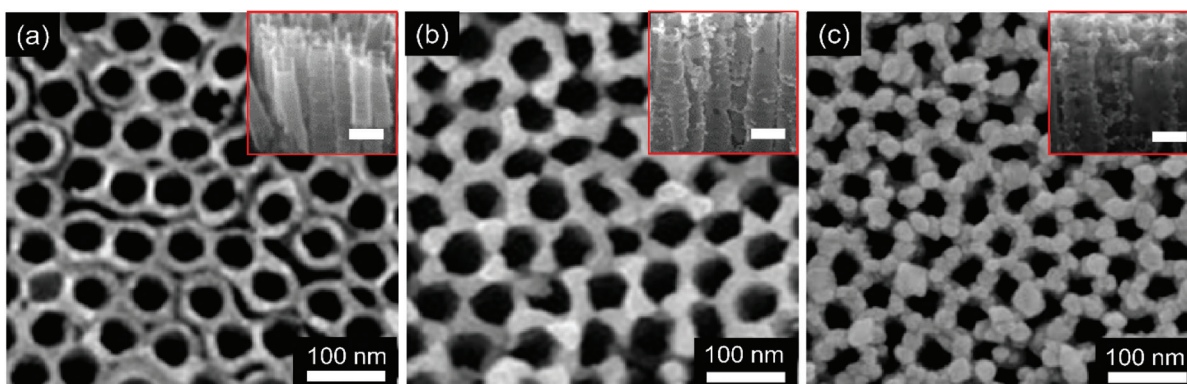


Fig. 2 SEM images of ZnO NT-based composites. (a) ZnO NTs, (b) the binary CuO–ZnO NT composite and (c) the ternary Ag–CuO–ZnO NT composite. The insets show cross-section images of the corresponding products. The scale bars of the insets are 50 nm.



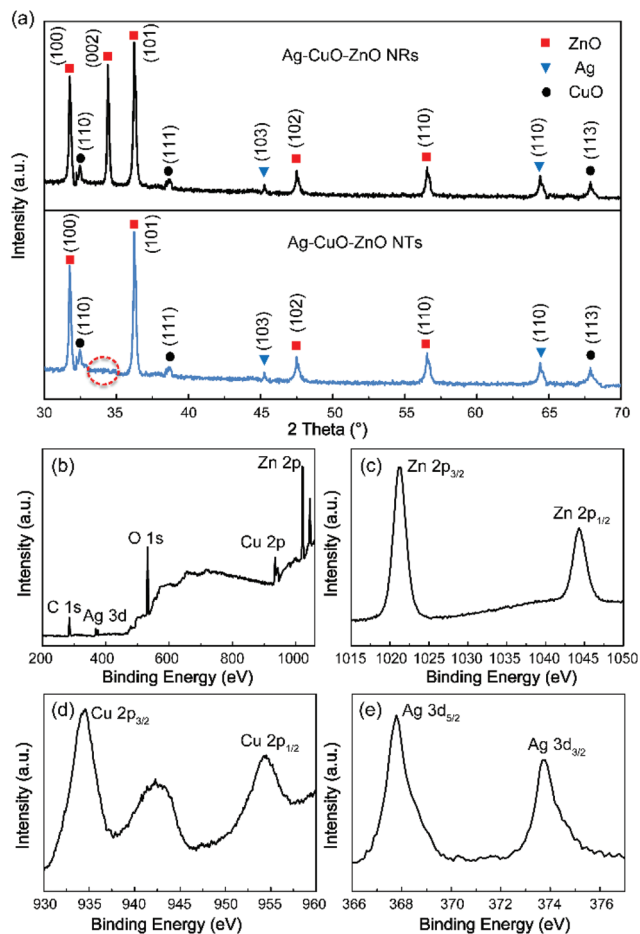


Fig. 4 (a) XRD patterns of the ternary Ag–CuO–ZnO NRs and Ag–CuO–ZnO NTs. XPS spectra of Ag–CuO–ZnO NTs. (b) Survey, (c) Zn 2p, (d) Cu 2p, (e) Ag 3d.

be measured from both ternary composites based on ZnO NRs (top graph) and ZnO NTs (bottom graph). This observation also indicates that the structures of ZnO NRs and NTs are retained well after the photodeposition of CuO and Ag nanoparticles, which correspond to the SEM and TEM results as illustrated above. Notably, the (002) peak intensity of ZnO NTs becomes much weaker in comparison with that of ZnO NRs (indicated by the red circle), implying the elimination of the (00l) planes of ZnO NTs.³⁸ In addition, three diffraction peaks (110), (111), (113) (JCPDS no. 48-1548) of CuO and two diffraction peaks (103) and (110) (JSPDS no. 41-1402) of Ag are observed, revealing that the as-fabricated ternary Ag–CuO–ZnO composites have excellent crystalline structures. Furthermore, XPS spectra are recorded to identify the chemical constituents and valence states of the ternary Ag–CuO–ZnO NT composite. Fig. 4b exhibits the typical XPS survey spectra, which demonstrate the high purity of the hybrid nanostructures, including the elements Zn, Cu, O and Ag. The peaks at 1044.6 eV and 1021.5 eV correspond to asymmetric peaks of Zn 2p_{1/2} (1044.6 eV) and Zn 2p_{3/2} (1021.5 eV), respectively (Fig. 4c).³⁶ The two typical peaks of Cu 2p_{3/2} (933.8 eV) and Cu 2p_{1/2} (953.9 eV) as

well as the satellite peak at 943 eV confirm the oxidation states of Cu²⁺ rather than Cu¹⁺ (Fig. 4d).⁴¹ The binding energies of 374 eV and 368 eV are in accordance with the Ag 3d_{3/2} and Ag 3d_{5/2} with a splitting energy of 6 eV, indicating the existence of metallic Ag (Fig. 4e).²⁴ The XRD and XPS analyses above further identify the coexistence of ZnO, CuO, and Ag in the as-synthesised photocatalysts.

The photocatalytic capability of the ZnO-based compound is evaluated towards the reduction of MO, a hazardous dye applied in the textile industry. Before gaining insight into the ternary catalysts, the photocatalytic activities of bare ZnO NTs (NRs), binary CuO–ZnO NTs (NRs) and binary Ag–ZnO NTs (NRs) driven by visible light irradiation (420–720 nm) are firstly investigated (Fig. S4†). It is found that bare ZnO NTs or NRs exhibit minimum degradation efficiency as compared to their corresponding binary composites, which is attributed to the poor efficiency of utilization of the visible light because of the wide bandgap and low charge-separation efficiency. Noticeably, the photocatalytic performance of bare ZnO NTs is slightly higher in comparison with that of bare ZnO NRs. This is primarily ascribed to the larger surface area of ZnO NTs with more active sites. The decoration of CuO or Ag nanoparticles onto the surface of ZnO NRs or NTs is capable of moderately improving the photocatalytic efficiency. Such observation is primarily due to the fact that the introduction of CuO or Ag allows the formation of the heterojunction or the Schottky barrier, resulting in higher charge-separation efficiency.^{29–31} Particularly, the ternary Ag–CuO–ZnO NT composite demonstrates optimal photocatalytic performance. To be specific, Fig. 5a and b show the absorption spectra of MO at different time intervals when degraded by Ag–CuO–ZnO NTs and Ag–CuO–ZnO NRs, respectively. After a reaction for 1 h under visible light irradiation, there is almost no absorption peak observed at 464 nm for the ternary catalyst of Ag–CuO–ZnO NTs, indicating the complete degradation of the dyes. However, the absorption intensity for MO catalysed by Ag–CuO–ZnO NRs only decreases from ~30% to ~16%. The time-dependent concentration of MO relative to its initial concentration (C/C_0) without the catalyst and in the presence of the ternary ZnO-based composites is depicted in Fig. 5c. Importantly, the degradation of ternary Ag–CuO–ZnO NTs is much faster than that of Ag–CuO–ZnO NRs driven by visible light irradiation, even in a dark environment. Fig. 5d exhibits the kinetic behaviors of MO degradation by the ternary ZnO-based composites.

The pseudo-first-order kinetics of the ternary ZnO-based photocatalysts are investigated *via* employing the pseudo-first-order model as follows:⁴²

$$\ln\left(\frac{C_0}{C_t}\right) = kt \quad (1)$$

where C_0 and C_t represent the concentrations of MO at time 0 and t , respectively, and k is the reaction rate constant (min^{-1}). It is observed that ternary Ag–CuO–ZnO NTs show a high rate constant k of $\sim 0.3 \text{ min}^{-1}$, which is around three times larger than that of ternary Ag–CuO–ZnO NRs ($k \sim 0.1 \text{ min}^{-1}$). It



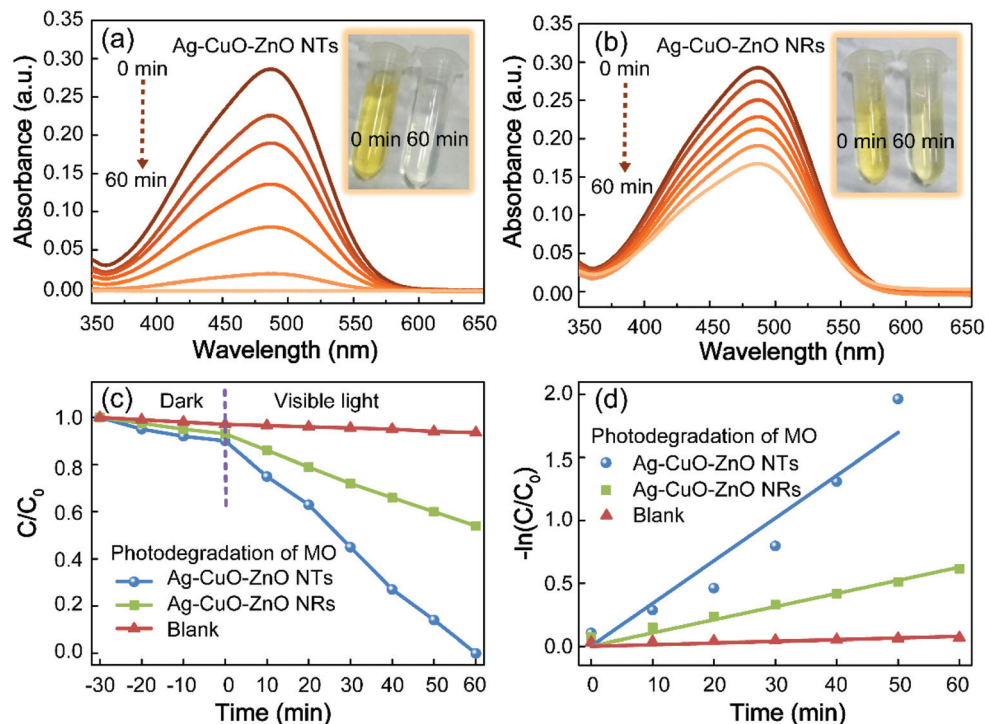


Fig. 5 Photocatalytic performance of the ternary ZnO-based composites to degrade MO. UV-Vis absorption spectra of MO solution degraded by the ternary (a) Ag–CuO–ZnO NT and (b) Ag–CuO–ZnO NR composites. The insets in (a) and (b) show photographic images of MO degraded by Ag–CuO–ZnO NTs and Ag–CuO–ZnO NRs before and after visible light irradiation (60 min), respectively. (c) Time-dependent photodegradation efficiency of MO without photocatalysts and in the presence of ternary composites based on the ZnO NTs and NRs. (d) The corresponding graph of $-\ln(C_t/C_0)$ versus the irradiation time.

should be noted that without any photocatalysts, the degradation of MO is quite slow at a rate constant k of $\sim 0.01 \text{ min}^{-1}$. In addition, extensive experiments on degrading the pollutants of RhB are carried out. The degradation efficiency of ternary composites based on ZnO NTs and NRs also demonstrates a similar trend to that of MO molecules, indicating that our ternary composites can be potentially applied in the reduction of various pollutants (Fig. S6†).

In order to untangle the origin of the enhanced photocatalytic performance of the ternary Ag–CuO–ZnO NTs towards the reduction of MO in comparison with that of bare ZnO, binary ZnO-based composites as well as ternary Ag–CuO–ZnO NRs, UV-Vis absorbance spectra of the composites are measured (Fig. 6). It is clearly seen that because of its wide bandgap, the ZnO NR array only shows a response to UV light ($< \sim 380 \text{ nm}$). When ZnO NTs are formed, the absorption is slightly increased in the visible range. Such an increase can be ascribed to the larger optical path lengths of photons, leading to the higher probability of absorption, which, in turn, contributes to larger scale production of electron–hole pairs and higher photodegradation efficiency than that of ZnO NRs.⁴³ After the decoration of CuO nanoparticles, the heterojunction structures are formed and the absorption for both binary CuO–ZnO NTs and NRs is enhanced, which is attributed to the narrow bandgap of CuO ($\sim 1.7 \text{ eV}$).³¹ As for the absorption spectra of ternary composites, there is a typical surface

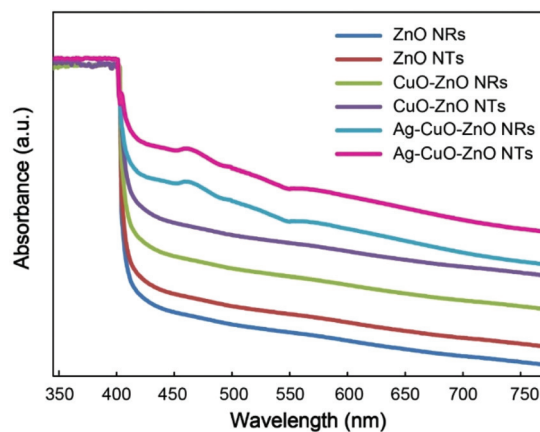


Fig. 6 UV-Vis absorption spectra of the bare ZnO NTs and NRs, binary CuO–ZnO NTs and NRs, ternary Ag–CuO–ZnO NTs and NRs.

plasmon resonance (SPR) peak at around 440 nm, further confirming that Ag nanoparticles are successfully deposited onto the binary CuO–ZnO composites. The characteristic SPR band extends the absorption of ternary Ag–CuO–ZnO NTs and NRs in the visible range, which promotes the generation of charge carriers to boost the photocatalytic performance.^{44,45} Furthermore, the BET surface area is also studied to reveal the causes of enhanced photocatalytic efficiency of Ag–CuO–ZnO



Table 1 Surface area, pore volume and pore size of the as-fabricated ternary Ag–CuO–ZnO NT and Ag–CuO–ZnO NR composites

Materials	BET surface area ^a (m ² g ⁻¹)	Single point total pore volume ^b (cm ³ g ⁻¹)	Average pore diameter ^c (nm)
Ag–CuO–ZnO NTs	19.85	0.35	14.57
Ag–CuO–ZnO NRs	1.77	0.01	7.04

^aThe BET surface area is achieved from the linear part of the BET graph. ^bThe total pore volume of the pores ($P/P_0 = 0.99$). ^cThe average adsorption pore width (4 V A⁻¹ by BET).²³

NTs. It can be found from Table 1 that Ag–CuO–ZnO NTs are endowed with a surface area of 19.85 m² g⁻¹, which is nearly 11 times larger than that of Ag–CuO–ZnO NRs. The larger surface area affords a variety of active sites for the photochemical reaction, contributing to the diffusion and transportation of dye pollutants and higher photocatalytic efficiency.^{46,47} To further reveal the advantages of the ternary Ag–CuO–ZnO NTs over the Ag–CuO–ZnO NRs in photocatalytic activity, photoelectrochemical analysis is also performed. It is found that the photocurrent density of the Ag–CuO–ZnO NTs is much higher than that of the Ag–CuO–ZnO NRs, indicating more efficient separation of photogenerated charge carriers on the Ag–CuO–ZnO NTs (Fig. S8(a)†).^{17,25,30} Furthermore, it can be observed from EIS Nyquist plots that the ternary Ag–CuO–ZnO NTs exhibit smaller semicircles at higher frequencies in comparison with the ternary Ag–CuO–ZnO NRs, which reveals that more efficient interfacial transportation as well as separation

of photogenerated electron–hole pairs are achieved over the Ag–CuO–ZnO NT electrode (Fig. S8(b)†).^{17,25,30} These results further demonstrate that the enhanced photocatalytic capability of ternary Ag–CuO–ZnO NTs is primarily ascribed to the intensified visible light harvesting, much larger surface area and efficient charge carrier separation.

The proposed mechanisms of the intensive photodegradation efficiency for the ternary Ag–CuO–ZnO NTs are illustrated in Fig. 7. Different from the ternary Ag–CuO–ZnO NRs, photocatalytic reactions of which only take place at the outer surface, those on the ternary Ag–CuO–ZnO NTs are able to occur simultaneously on the inner and outer surfaces. Under visible light irradiation, the photocatalytic processes involve two steps, which principally depend on the charge transfer reactions to create radical species for the decomposition of pollutants. To be specific, owing to the narrow bandgap of CuO, electrons are stimulated by visible light and transferred from the valance band (VB) of CuO to the conduction band (CB), while the holes are retained in the VB.³¹ Then, ZnO is employed as a photoelectronic acceptor along with the photo-generated electrons in the CB of CuO being promptly propagated to the CB of ZnO. The photo-induced electrons can be captured by oxygen, allowing the formation of superoxide radical anions ($O_2 + e^- \rightarrow O_2^{\cdot-}$). This results in the removal of organic pollutants in the sewage or further produces oxygen species *via* various reactions with H⁺. Simultaneously, the holes transfer in the reverse direction, which can oxidize H₂O into hydroxyl radicals ($H_2O + h^+ \rightarrow OH^{\cdot}$). Such highly oxidized hydroxyl radicals have chemical reactions with organic species to primarily generate CO₂ as well as H₂O (step 1).⁴⁸

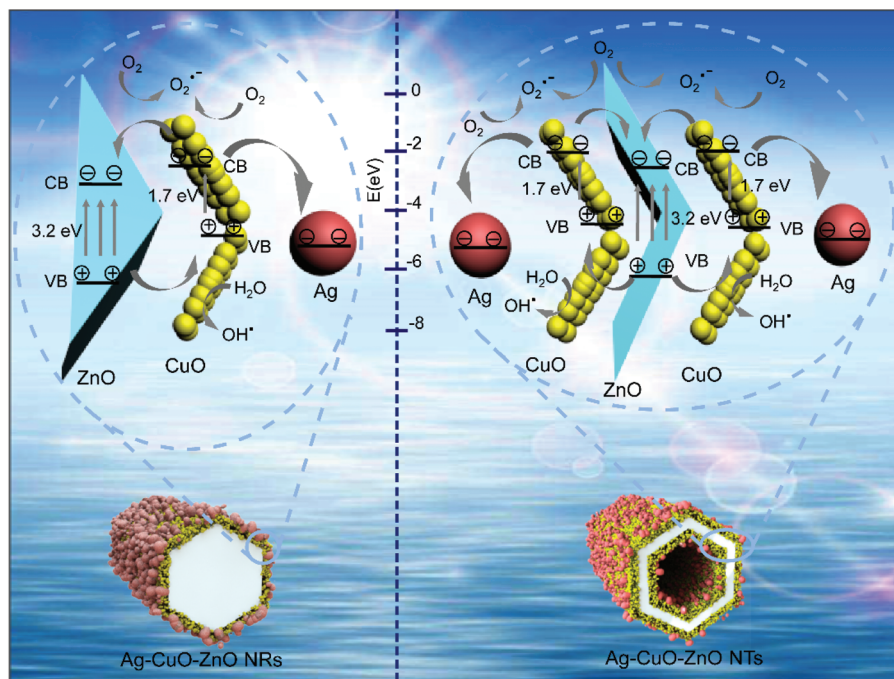


Fig. 7 Proposed mechanisms of the photocatalytic performance of the ternary Ag–CuO–ZnO NRs (left) and Ag–CuO–ZnO NTs (right) composites towards the decomposition of pollutants. The two graphs in the bottom represent the cross-section of Ag–CuO–ZnO NRs and NTs, respectively.



When binary CuO–ZnO NTs are functionalized with Ag nanoparticles, the energy level of the CB for CuO is larger than the newly generated Fermi energy level, which is ascribed to the difference of work functions between CuO (~ 5.5 eV) and Ag (~ 4.3 eV).²⁶ This results in a portion of the photogeneration of electrons being moved from CB of CuO to Ag. Then, these electrons in the Ag component can be captured by the chemisorbed O_2 to form the superoxide radicals ($O_2^{\cdot -}$), leading to the reaction with organic species (step 2).²⁷ Therefore, the optimized photocatalytic efficiency of the ternary Ag–CuO–ZnO NTs is attributed to the synergistic effect of three functional components with the vertically aligned ZnO NT array as the backbone structures.

The highly stabilized photocatalysts, without photocorrosion in aqueous solution, are of paramount importance for practical wastewater remediation. To demonstrate the photostability of the as-fabricated ternary ZnO NTs and NR-based photocatalysts, twenty successive recycling measurements for the degradation of MO are carried out (Fig. 8). It is observed

that the photodegradation performance of the ternary Ag–CuO–ZnO NT composite almost remains unchanged, while the photocatalytic activity of Ag–CuO–ZnO NRs decreases dramatically from 46% to 5% after twenty recycle tests. Three possible factors are responsible for the superior photostability of ternary ZnO NT-based composites as compared to that of ternary ZnO NR-based catalysts. First, the ZnO NTs are composed of only (10–10) surfaces with the equivalent O^{2-} and Zn^{2+} , which have lower surface energy. However, the ZnO NRs consist of (0001)-O and (0001)-Zn planes with non-equivalent O^{2-} and Zn^{2+} . As the reactions of photodegradation principally involve holes (h^+) and surface oxygen of ZnO, the ZnO NRs with higher surface energy are prone to be consumed by holes and thus demonstrate much weaker photocatalytic performance after more than ten recycling cycles.³⁴ Second, the protective layer of CuO only occurs in the outer surface of Ag–CuO–ZnO NRs, while the coverage of the CuO shells appears in the inner and outer surfaces of Ag–CuO–ZnO NTs, leading to more effective protection of the surface oxygen in ZnO exposed in solution.²¹ Third, the more efficient propagation of holes from ZnO NTs to CuO also hinders them from attacking the surface oxygen of ZnO NTs.³¹

Furthermore, XRD patterns of the samples also indicate that Ag–CuO–ZnO NTs keep their crystal structures even after photocatalytic degradation for 20 cycles, while all the diffraction peaks of Ag–CuO–ZnO NRs decrease rapidly and even disappear (Fig. S11†). This phenomenon indicates that the framework of crystal structures of ZnO NRs collapses after the prolonged photodegradation, making the Ag and CuO nanoparticles dissolve in the pollutant solution. In contrast, the photocorrosion of ZnO NTs can be efficiently inhibited, which is capable of preserving the intact structures of photocatalysts after the super-long photodegradation. These results reveal that the ternary Ag–CuO–ZnO NTs exhibit perdurable and effective photocatalytic capability, which not only affords excellent potential for practical environmental remediation, but also opens up new opportunities for other applications, such as recyclable surface enhanced Raman scattering (SERS) substrates.^{49,50}

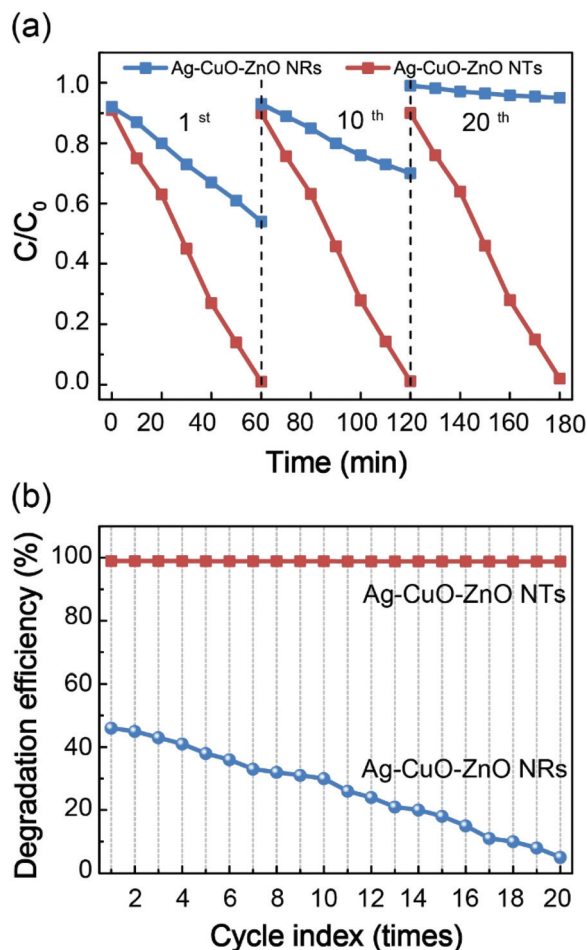


Fig. 8 Stability demonstration. (a) Cycling runs (1st, 10th and 20th) of the photocatalytic performance of the ternary Ag–CuO–ZnO NTs (NRs) for the degradation of MO solution driven by visible light illumination. (b) Degradation efficiency of the MO solution catalyzed by the ternary Ag–CuO–ZnO NTs (NRs) within 20 cycles.

4. Conclusions

In summary, the ternary Ag–CuO–ZnO NT composite is successfully synthesized using the oriented 1D ZnO NTs as the building block *via* facile chemical deposition and photochemical deposition approaches. It is found that Ag–CuO–ZnO NTs demonstrate much higher photocatalytic performance in pollutant decomposition than Ag–CuO–ZnO NRs. Such excellent performance of ZnO NT-based photocatalysts is primarily ascribed to the intensive visible light harvesting, efficient electron–hole separation and high specific surface area. Meanwhile, the Ag–CuO–ZnO NTs demonstrate ultra-stable photocatalytic performance towards pollutant degradation in comparison with Ag–CuO–ZnO NRs. This is mainly attributed to the unique intrinsic crystal structures of ZnO NTs as well as



the functionalized CuO as the protective layers. This work paves a new way to construct ternary ZnO NT-based heterostructured photocatalysts with the excellent performance of superior photodegradation and photocorrosion suppression for practical sewage treatment and other potential applications.

Acknowledgements

This work was funded by an A*STAR, SERC 2014 Public Sector Research Funding (PSF) grant (SERC Project No. 1421200080).

Notes and references

- X. Liu, J. Iocozzia, Y. Wang, X. Cui, Y. Chen, S. Zhao, Z. Li and Z. Lin, *Energy Environ. Sci.*, 2017, **10**, 402–434.
- Z. L. Wang, *Faraday Discuss.*, 2015, **176**, 447–458.
- X. S. Meng, G. Zhu and Z. L. Wang, *ACS Appl. Mater. Interfaces*, 2014, **6**, 8011–8016.
- T. Jiang, L. M. Zhang, X. Chen, C. B. Han, W. Tang, C. Zhang, L. Xu and Z. L. Wang, *ACS Nano*, 2015, **9**, 12562–12572.
- M. Q. Yang, N. Zhang, M. Pagliaro and Y. J. Xu, *Chem. Soc. Rev.*, 2014, **43**, 8240–8254.
- M. Wang, J. Iocozzia, L. Sun, C. Lin and Z. Lin, *Energy Environ. Sci.*, 2014, **7**, 2182–2202.
- N. Zhang, R. Ciriminna, M. Pagliaro and Y. J. Xu, *Chem. Soc. Rev.*, 2014, **43**, 5276–5287.
- C. F. Tan, W. L. Ong and G. W. Ho, *ACS Nano*, 2015, **9**, 7661–7670.
- D. Hong, W. Zang, X. Guo, Y. Fu, H. He, J. Sun, L. Xing, B. Liu and X. Xue, *ACS Appl. Mater. Interfaces*, 2016, **8**, 21302–21314.
- N. Zhang, M. Q. Yang, S. Liu, Y. Sun and Y. J. Xu, *Chem. Rev.*, 2015, **115**, 10307–10377.
- L. Zhu, M. Hong and G. W. Ho, *Nano Energy*, 2015, **11**, 28–37.
- F. X. Xiao, J. Miao, H. B. Tao, S. F. Hung, H. Y. Wang, H. B. Yang, J. Chen, R. Chen and B. Liu, *Small*, 2015, **11**, 2115–2131.
- Y. Wang, L. Liu, L. Xu, X. Cao, X. Li, Y. Huang, C. Meng, Z. Wang and W. Zhu, *Nanoscale*, 2014, **6**, 6790–6797.
- D. Deng, S. T. Martin and S. Ramanathan, *Nanoscale*, 2010, **2**, 2685–2691.
- Y. J. Liu, L. Sun, J. G. Wu, T. Fang, R. Cai and A. Wei, *Mater. Sci. Eng., B*, 2015, **194**, 9–13.
- Y. L. Chen, L. C. Kuo, M. L. Tseng, H. M. Chen, C. K. Chen, H. J. Huang, R. S. Liu and D. P. Tsai, *Opt. Express*, 2013, **21**, 7240–7249.
- C. Han, Z. Chen, N. Zhang, J. C. Colmenares and Y. J. Xu, *Adv. Funct. Mater.*, 2015, **25**, 221–229.
- J. G. Wu, T. Fang, R. Cai, S. Y. Li, Y. Wang, C. E. Zhao and A. Wei, *RSC Adv.*, 2016, **6**, 4145–4150.
- C. Tian, Q. Zhang, A. Wu, M. Jiang, Z. Liang, B. Jiang and H. Fu, *Chem. Commun.*, 2012, **48**, 2858–2860.
- A. van Dijken, E. A. Meulenkaamp, D. Vanmaekelbergh and A. Meijerink, *J. Phys. Chem. B*, 2000, **104**, 1715–1723.
- K. M. Lee, C. W. Lai, K. S. Ngai and J. C. Juan, *Water Res.*, 2016, **88**, 428–448.
- L. Zhu, M. Hong and G. W. Ho, *Sci. Rep.*, 2015, **5**, 11609–11619.
- Z. Chen, N. Zhang and Y. J. Xu, *CrystEngComm*, 2013, **15**, 3022–3030.
- Q. Deng, X. Duan, D. H. Ng, H. Tang, Y. Yang, M. Kong, Z. Wu, W. Cai and G. Wang, *ACS Appl. Mater. Interfaces*, 2012, **4**, 6030–6037.
- N. Zhang, S. Xie, B. Weng and Y. J. Xu, *J. Mater. Chem. A*, 2016, **4**, 18804–18814.
- Y. Sun, Y. Sun, T. Zhang, G. Chen, F. Zhang, D. Liu, W. Cai, Y. Li, X. Yang and C. Li, *Nanoscale*, 2016, **8**, 10774–10782.
- S. T. Kochuveedu, Y. H. Jang and D. H. Kim, *Chem. Soc. Rev.*, 2013, **42**, 8467–8493.
- N. Zhang, C. Han, Y. J. Xu, J. J. Foley IV, D. Zhang, J. Codrington, S. K. Gray and Y. Sun, *Nat. Photonics*, 2016, **10**, 473–482.
- Y. Wang, H. B. Fang, Y. Z. Zheng, R. Ye, X. Tao and J. F. Chen, *Nanoscale*, 2015, **7**, 19118–19128.
- C. Han, Q. Quan, H. M. Chen, Y. Sun and Y. J. Xu, *Small*, 2017, **13**, 1602947.
- J. Yu, S. Zhuang, X. Xu, W. Zhu, B. Feng and J. Hu, *J. Mater. Chem. A*, 2015, **3**, 1199–1207.
- J. Bai, Y. Li, R. Wang, K. Huang, Q. Zeng, J. Li and B. Zhou, *J. Mater. Chem. A*, 2015, **3**, 22996–23002.
- C. Han, M. Q. Yang, B. Weng and Y. J. Xu, *Phys. Chem. Chem. Phys.*, 2014, **16**, 16891–16903.
- B. Weng, M. Q. Yang, N. Zhang and Y. J. Xu, *J. Mater. Chem. A*, 2014, **2**, 9380–9389.
- S. Ameen, M. S. Akhtar, Y. S. Kim, O. B. Yang and H. S. Shin, *Colloid Polym. Sci.*, 2010, **289**, 415–421.
- P. Y. Kuang, Y. Z. Su, K. Xiao, Z. Q. Liu, N. Li, H. J. Wang and J. Zhang, *ACS Appl. Mater. Interfaces*, 2015, **7**, 16387–16394.
- P. Zhang, B. Li, Z. Zhao, C. Yu, C. Hu, S. Wu and J. Qiu, *ACS Appl. Mater. Interfaces*, 2014, **6**, 8560–8566.
- D. Chu, Y. Masuda, T. Ohji and K. Kato, *Langmuir*, 2010, **26**, 2811–2815.
- A. Wei, X. W. Sun, C. Xu, Z. L. Dong, Y. Yang, S. T. Tan and W. Huang, *Nanotechnology*, 2006, **17**, 1740–1744.
- X. Qi, G. She, Y. Liu, L. Mu and W. Shi, *Chem. Commun.*, 2012, **48**, 242–244.
- G. D. Moon, J. B. Joo, I. Lee and Y. Yin, *Nanoscale*, 2014, **6**, 12002–12008.
- J. M. Herrmann, H. Tahiri, Y. Ait-Ichou, G. Lassaletta, A. Gonzalez-Elipe and A. Fernandez, *Appl. Catal., B*, 1997, **13**, 219–228.
- I. Alessandri and J. R. Lombardi, *Chem. Rev.*, 2016, **116**, 14921–14981.
- C. Chen, Y. Zheng, Y. Zhan, X. Lin, Q. Zheng and K. Wei, *Dalton Trans.*, 2011, **40**, 9566–9570.



- 45 S. A. Ansari, M. M. Khan, M. O. Ansari, J. Lee and M. H. Cho, *J. Phys. Chem. C*, 2013, **117**, 27023–27030.
- 46 Y. Hong, C. Tian, B. Jiang, A. Wu, Q. Zhang, G. Tian and H. Fu, *J. Mater. Chem. A*, 2013, **1**, 5700–5708.
- 47 D. Liu, Y. Lv, M. Zhang, Y. Liu, Y. Zhu, R. Zong and Y. Zhu, *J. Mater. Chem. A*, 2014, **2**, 15377–15388.
- 48 L. Zheng, S. Han, H. Liu, P. Yu and X. Fang, *Small*, 2016, **12**, 1527–1536.
- 49 K. C. Xu, C. T. Zhang, R. Zhou, R. Ji and M. H. Hong, *Opt. Express*, 2016, **24**, 10352–10358.
- 50 K. C. Xu, C. T. Zhang, T. H. Lu, P. Wang, R. Zhou, R. Ji and M. H. Hong, *Opto-Electron. Eng.*, 2017, **44**, 185–191.

