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# Synthesis and temperature-dependent studies of a perovskite-like manganese formate framework templated with protonated acetamidine†

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We report the synthesis, crystal structure, thermal, dielectric, phonon and magnetic properties of the  $[\text{CH}_3\text{C}(\text{NH}_2)_2][\text{Mn}(\text{HCOO})_3]$  (AceMn) compound. Our results show that this compound crystallizes in the perovskite-like orthorhombic structure, space group *Imma*. It undergoes a structural phase transition at 304 K into a monoclinic structure, space group *P2<sub>1</sub>/n*. X-ray diffraction, dielectric, IR and Raman studies show that the ordering of the acetamidinium cations triggers the phase transition. Low-temperature magnetic studies show that this compound exhibits weak ferromagnetic properties below 9.0 K.

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## Introduction

Metal formate frameworks templated by ammonium or protonated amines constitute a class of dense metal organic framework (MOF) compounds that have received a lot of attention in recent years. In this context, rare-earth formate frameworks were found to be promising luminescent and non-linear optical materials<sup>1,2</sup> whereas heterometallic and mixed-valence formates, with general formulas  $[\text{amineH}^+][\text{M}^{\text{III}}\text{M}^{\text{II}}(\text{HCOO})_6]$  and  $[\text{amineH}^+][\text{Na}_{0.5}\text{M}_{0.5}^{\text{III}}(\text{HCOO})_3]$  (amine = dimethylamine, ethylamine, diethylamine;  $\text{M}^{\text{III}}$  = Fe, Cr;  $\text{M}^{\text{II}}$  = Mg, Zn, Cu, Ni, Fe, Co, Mn) were shown to exhibit luminescence, ferroelectric, magnetic or even multiferroic properties.<sup>3–5</sup> Multiferroic properties were also reported for the  $[\text{NH}_2(\text{CH}_2)_4\text{NH}_2][\text{Mn}(\text{HCOO})_6]$  compound.<sup>6</sup> The most interesting sub-group of formate frameworks constitutes, however, MOFs with the general formula  $[\text{amineH}^+][\text{M}(\text{HCOO})_3]$  ( $\text{M}$  = divalent metal cation). The first

members of this family of formates with methylammonium, dimethylammonium ( $\text{DMA}^+$ ), ethylammonium and azetidinium cations were discovered by Wang *et al.* in 2004 and these compounds were shown to exhibit weak ferromagnetic properties.<sup>7</sup> In 2009, the first report was published on the discovery of multiferroic properties in  $[(\text{CH}_3)_2\text{NH}_2][\text{M}(\text{HCOO})_3]$  compounds ( $\text{M}$  = Mn, Fe, Ni, Co).<sup>8</sup> This discovery promoted broad interest in the studies of these formates and the search for novel frameworks templated by various cations.<sup>9–24</sup> In this context, azetidinium analogues were shown to exhibit giant dielectric anomalies<sup>9</sup> whereas guanidinium ( $\text{Gua}^+$ ) analogues showed interesting magnetic properties.<sup>10,11</sup> Furthermore, the coexistence of ferroelectric and magnetic orders was discovered in metal formate frameworks templated by ammonium, methylammonium, hydrazinium and methylhydrazinium cations, and  $\text{M}^{\text{II}}$  = Mn, Fe, Co, Ni.<sup>12–16</sup> It is also worth adding that multiferroic properties were also predicted theoretically for guanidinium ( $\text{M}^{\text{II}}$  = Cr, Cu) and ethylammonium ( $\text{M}^{\text{II}}$  = Mn) analogues.<sup>17,18</sup> Apart from magnetic and dielectric properties, some of these divalent metal formates also exhibit negative thermal expansion and negative linear compressibility properties.<sup>19,20</sup>

Almost all known divalent metal formates templated by ammonium or protonated amines undergo temperature-induced structural phase transitions associated with ordering of the organic cations and distortion of the metal formate framework.<sup>25,26</sup> The size and shape of the organic cation as well as its ability to form hydrogen bonds (HBs) with the oxygen atoms of the framework determine the crystal structures and physicochemical properties.<sup>25,26</sup> Therefore, much effort has been made to synthesise metal formate frameworks with various protonated amines. Until now, 11 cations were

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† Electronic supplementary information (ESI) available: Tables S1–S3: crystallographic data for AceMn, comparison of structural parameters for AceMn and related compounds, IR and Raman wavenumbers of AceMn and suggested assignments. Fig. S1–S7: XRD patterns, DSC traces, view of the structure, dielectric loss spectra, the IR and Raman spectra at different temperatures, temperature dependence of wavenumbers and FWHM values for selected Raman modes. CCDC 1530423 and 1530424. For ESI and crystallographic data in CIF or other electronic format see DOI: 10.1039/c7dt01251a



employed, *i.e.* apart from the above-mentioned 7 cations, formates with hydroxylammonium,<sup>27</sup> formamidinium (FMD<sup>+</sup>),<sup>28,29</sup> imidazolium,<sup>30</sup> and tetramethylammonium<sup>31</sup> are also known.

The acetamidinium cation (CH<sub>3</sub>C(NH<sub>2</sub>)<sub>2</sub><sup>+</sup>, Ace<sup>+</sup>) has not yet been used in the synthesis of formate frameworks. However, it has been recently successfully used in the synthesis of the switchable dielectric material [CH<sub>3</sub>C(NH<sub>2</sub>)<sub>2</sub>]<sub>2</sub>[KFe(CN)<sub>6</sub>].<sup>32</sup> Since its size (2.77 Å)<sup>33</sup> is similar to the size of DMA<sup>+</sup> (2.72 Å)<sup>33,34</sup> and Gua<sup>+</sup> (2.78 Å)<sup>34</sup> that were previously used as a template ion in the metal formate frameworks, we have decided to try the synthesis of novel formate frameworks containing this cation. We will show that this synthesis was successful for a manganese analogue and that the novel MOF undergoes a temperature-induced structural phase transition.

## Experimental details

### Materials and instrumentation

MnCl<sub>2</sub> (99%, Sigma-Aldrich), methanol (99.8%, Sigma-Aldrich), acetamidine hydrochloride (95%, Sigma-Aldrich), sodium formate (99%, Sigma-Aldrich) and formic acid (98%, Fluka) were commercially available and used without further purification. Heat capacity was measured using a Mettler Toledo DSC-1 calorimeter with a high resolution of 0.4 μW. Nitrogen was used as a purging gas and the heating and cooling rate was 5 K min<sup>-1</sup>. The excess heat capacity associated with the phase transition was evaluated by subtraction from the data the baseline representing variation in the absence of the phase transitions. A powder XRD pattern was obtained on an X'Pert PRO X-ray diffraction system equipped with a PIXcel ultrafast line detector, a focusing mirror, and Soller slits for CuKα<sub>1</sub> radiation (λ = 1.54056 Å). The complex dielectric permittivity was measured using a broadband impedance Novocontrol Alpha analyser. The sample was investigated isothermally in the frequency range 10<sup>-1</sup> to 10<sup>6</sup> Hz and the measurements were taken every 1 K over the temperature range from 230 to 350 K. Since the obtained single crystals were not big enough to perform single crystal dielectric measurements, a pellet made of a well-dried sample was measured instead. The pellet with a diameter of 6 mm and a thickness of 0.4 mm was located between copper electrodes. The sample was located in the sample chamber filled with dry nitrogen. The temperature values, with a stability ±0.2 K, were changed by a flow of vaporized liquid nitrogen heated and controlled by a Novocontrol heating system. The complex dielectric function is given by ε''(ω) = ε' - iε'' = -i/(ωZ\*(ω)C<sub>0</sub>), where Z\* is the measured complex impedance, C<sub>0</sub> is the geometrical capacitance of the sample and ω = 2πf is the angular frequency. All measurements including the calibration routine, temperature stabilization, and data acquisition were automatically controlled. The magnetic properties of a large number of freely oriented single crystals of AceMn (about 40 mg in total) were measured using a commercial SQUID (superconducting quantum interference device) magnetometer in the temperature range 2–300 K and in external magnetic fields up to

3 kOe. The AC magnetic susceptibility was measured below 20 K in zero magnetic field applied using the probing AC magnetic field of an amplitude of 3 Oe and frequencies ranging between 1 and 1000 Hz. The background coming from a weakly diamagnetic sample holder was found in the temperature range studied to be negligible in comparison to the total signal measured, so its subtraction was omitted. No demagnetization corrections were made to the data reported here either. The low-temperature specific heat of the largest single crystal available (about 2 mg) was measured using a Quantum Design PPMS (Physical Property Measurement System) platform utilizing the thermal relaxation method. The heat capacity of the calorimeter (sapphire plate) and adhesive (vacuum grease Apiezon N) was measured separately and subtracted from the total signal. Temperature-dependent Raman spectra were obtained using a Renishaw inVia Raman spectrometer equipped with a confocal DM 2500 Leica optical microscope, a thermoelectrically cooled CCD as a detector, an argon laser operating at 488 nm and a Linkam cryostat cell. Temperature-dependent IR spectra were obtained using a Nicolet iN10 stand-alone infrared microscope and a Linkam cryostat cell. The spectral resolution of the Raman and IR spectra was 2 cm<sup>-1</sup>.

### Synthesis of the sample

In order to grow single crystals of AceMn, 8 mL of methanol solution containing 3 mmol of acetamidine hydrochloride was mixed with 8 mL of methanol solution containing 3 mmol of sodium formate. The mixture was stirred for half an hour. After filtering off the white precipitate, 0.7 mL of HCOOH was added to the clear solution and the mixture was placed at the bottom of a glass tube (9 mm inner diameter). Into this solution, 15 mL of methanol solution containing 1 mmol of MnCl<sub>2</sub> was gently added. The tube was sealed and kept undisturbed. Light pink crystals were harvested after 3 days. Comparison of their powder XRD pattern with the calculated ones based on the single-crystal data at 330 and 100 K (see Fig. S1 in the ESI†) confirmed the phase purity of the bulk sample.

### Single crystal X-ray diffraction

The X-ray intensity data for the AceMn crystal were collected using graphite monochromatic MoKα radiation on a four-circle κ geometry KUMA KM-4 diffractometer with a two-dimensional area CCD detector at 100(1) and 330(1) K. The ω-scan technique with Δω = 1.0° for each image was used for data collection. The unit cell parameters were refined by least-squares methods on the basis of all measured reflections. One image was used as a standard after every 40 images for monitoring of the crystal stability and data collection, and no correction on the relative intensity variations was necessary. Data collections were made using the CrysAlis CCD program.<sup>35</sup> Integration, scaling of the reflections, correction for Lorentz and polarisation effects and absorption corrections were performed using the CrysAlis Red program.<sup>35</sup> The structure was solved by direct methods using SHELXS97,<sup>36</sup> and refined using the SHELXL-2014 programs.<sup>37</sup> The hydrogen atoms of amine



groups were located in difference Fourier maps and were refined for the structure at 100(1) K, whereas for the structure at 330(1) K due to the disorder the H atoms were introduced geometrically. The hydrogen atoms joined to carbon atoms were generated geometrically (C–H 0.96 Å) and refined as a riding model. The final difference Fourier maps showed no peaks of chemical significance. Details of the data collection parameters, crystallographic data and final agreement parameters are shown in Table S1.† Visualisation of the structure was made with the Diamond 3.0 program.<sup>38</sup>

## Results and discussion

### DSC

The DSC measurements show the presence of one heat anomaly for AceMn at 304 K upon heating and at 303 K upon cooling (Fig. S2†). Fig. 1 shows that the heat capacity anomaly is asymmetric. The associated change in enthalpy  $\Delta H$  and entropy  $\Delta S$  was estimated to be  $\sim 604$  J mol<sup>-1</sup> and  $\sim 2.2$  J mol<sup>-1</sup> K<sup>-1</sup>, respectively (Fig. 1). For an order–disorder transition,  $\Delta S = R \ln(N)$ , where  $R$  is the gas constant and  $N$  is the ratio of the number of configurations in the disordered and ordered phase. Since according to the X-ray diffraction data discussed in the next paragraph  $N = 4$ ,  $\Delta S$  is expected to be 11.5 J mol<sup>-1</sup> K<sup>-1</sup>. The experimental value is lower than expected. It is worth noting that significantly smaller than expected  $\Delta S$  values were often reported for many metal formate frameworks templated by protonated amines, especially those that showed dielectric relaxation behaviour.<sup>15,29,39,40</sup> For instance, the change of entropy associated with the loss of the trigonal disorder in formamidinium manganese formate, [NH<sub>2</sub>CHNH<sub>2</sub>][Mn(HCOO)<sub>3</sub>], which undergoes a second-order phase transition, was found to be only 0.97 J mol<sup>-1</sup> K<sup>-1</sup>, instead of the expected value 9.1 J mol<sup>-1</sup> K<sup>-1</sup>.<sup>24</sup> Since some residual entropy is always left over if a phase transition has some relaxor character,<sup>41</sup> the small value of  $\Delta S$  for AceMn can be most likely explained in the same way as reported for related formate frameworks. That is, this behaviour

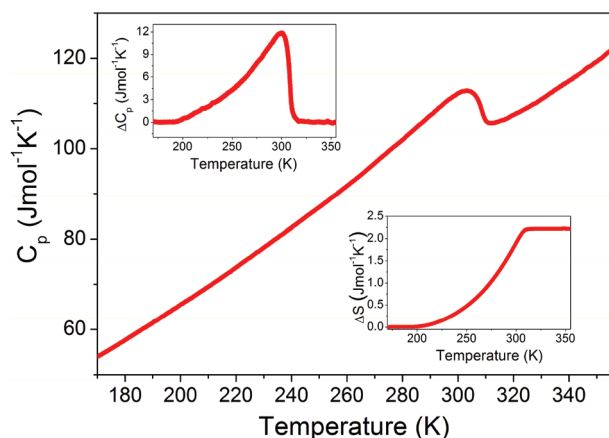


Fig. 1 The heat capacity of AceMn measured in a heating mode. The insets show the change in  $C_p$  and  $S$  related to the phase transition.

indicates most likely that much residual entropy is left over below  $T_c$ , implying the relaxor character of the phase transition.

### Single crystal X-ray diffraction

The structures of AceMn at 100 and 330 K are provided in Fig. 2a and b, respectively. Selected bond lengths and angles for both modifications are summarized in Table 1.

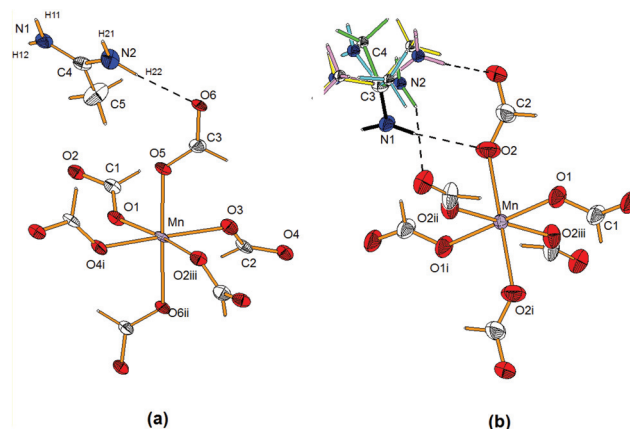


Fig. 2 View of the structure of (a) the low-temperature (LT) monoclinic and (b) high-temperature (HT) orthorhombic modifications of AceMn showing anisotropic displacement parameters at 50 and 30% probability level for the LT and HT phase, respectively. Dashed lines represent N–H...O HBs. For the orthorhombic phase the green, blue, yellow and light-purple colours represent the four possible orientations of the Ace<sup>+</sup> cation (C3 and amine N1 atoms are ordered). The symmetry code is the same as in Table 1.

Table 1 Selected geometrical parameters (Å, °)

100 K			
Mn–O1	2.211(2)	Mn–O2 <sup>iii</sup>	2.214(2)
Mn–O3	2.168(2)	Mn–O4 <sup>i</sup>	2.173(2)
Mn–O5	2.156(2)	Mn–O6 <sup>ii</sup>	2.175(2)
O3–Mn–O1	101.88(7)	O5–Mn–O1	93.34(8)
O5–Mn–O3	90.83(8)	O5–Mn–O4 <sup>i</sup>	87.32(8)
O3–Mn–O6 <sup>ii</sup>	88.81(8)	O1–Mn–O4 <sup>i</sup>	84.39(7)
O1–Mn–O6 <sup>ii</sup>	89.03(8)	O3–Mn–O2 <sup>iii</sup>	87.18(7)
O5–Mn–O2 <sup>iii</sup>	87.02(8)	O1–Mn–O4 <sup>i</sup>	84.39(7)
O1–Mn–O2 <sup>iii</sup>	170.93(7)	O3–Mn–O4 <sup>i</sup>	173.56(7)
O5–Mn–O6 <sup>ii</sup>	177.62(7)	Mn...Mn <sup>iii</sup>	6.1386(11)
Mn...Mn <sup>ii</sup>	5.9665(8)	Mn <sup>ii</sup> ...Mn...Mn <sup>iii</sup>	89.882(7)
Mn...Mn <sup>iv</sup>	6.3824(11)	Mn <sup>iii</sup> ...Mn...Mn <sup>iv</sup>	90.888(7)
Mn <sup>ii</sup> ...Mn...Mn <sup>iv</sup>	87.658(7)		
Symmetry codes: (i) $x - 1/2, -y + 1/2, z - 1/2$ ; (ii) $-x + 1/2, y - 1/2, -z + 3/2$ ; (iii) $x - 1/2, -y + 1/2, z + 1/2$ ; (iv) $x + 1/2, -y + 1/2, z - 1/2$ .			
330 K			
Mn–O1 (×2)	2.166(2)	Mn–O2 (×4)	2.183(2)
O1–Mn–O2	90.25(5)	O1–Mn–O2 <sup>i</sup>	89.75(5)
O1–Mn–O2 <sup>ii</sup>	90.25(5)	O1–Mn–O2 <sup>iii</sup>	89.75(5)
O1–Mn–O1 <sup>i</sup>	180.00(4)	O2–Mn–O2 <sup>i</sup>	180.00(6)
O2 <sup>ii</sup> –Mn–O2 <sup>iii</sup>	180.00(5)	Mn...Mn <sup>v</sup>	6.2869(9)
Mn...Mn <sup>iv</sup>	6.0445(10)	Mn <sup>iv</sup> ...Mn...Mn <sup>vi</sup>	90.00(1)
Mn...Mn <sup>vi</sup>	6.2869(9)		
Mn <sup>iv</sup> ...Mn...Mn <sup>v</sup>	90.00(1)		
Mn <sup>v</sup> ...Mn...Mn <sup>vi</sup>	88.173(1)		
Symmetry codes: (i) $-x, -y + 1, -z + 1$ ; (ii) $-x, y, z$ ; (iii) $x, -y + 1, -z + 1$ ; (iv) $-x, -y + 1/2, z$ ; (v) $-x + 1/2, -y + 1, z - 1/2$ ; (vi) $-x - 1/2, -y + 1, z - 1/2$ .			



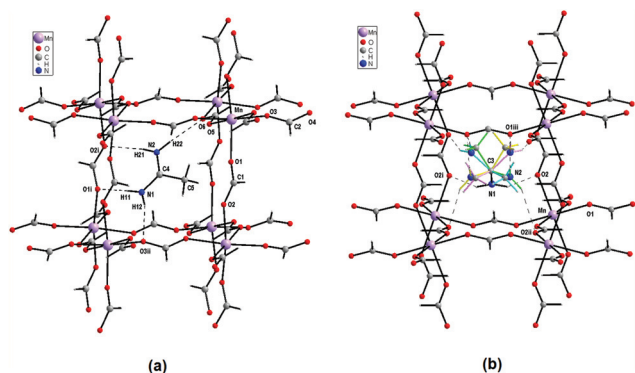
At low temperature, the AceMn structure is monoclinic (space group  $P2_1/n$ ) whereas at high temperature it is orthorhombic (space group  $Imma$ ). In both phases the  $Mn^{II}$  ion is octahedrally coordinated by formate ligands that bridge neighbouring coordinating spheres in the *anti-anti* mode similar to many reported compounds with formate as the bridge ligand,<sup>7,42</sup> thus forming the three-dimensional pseudo-perovskite anionic Mn-formate framework of  $\{[Mn(HCOO)_3]_n\}^-$  with the topology of  $4^{12} \cdot 6^3 \cdot 7,43$ . In the monoclinic phase the  $Mn^{II}$  ion lies in the general position and the  $MnO_6$  octahedron is distorted, the Mn–O bond lengths are in the range of 2.156(2)–2.214(2) Å, the *cis* O–Mn–O angles are in the range of 84.39(7)–101.88(7)° while the *trans* O–Mn–O angles lie in the range of 170.93(7)–177.62(7)°. In the orthorhombic phase the  $Mn^{II}$  ion lies on the two-fold axis and the  $MnO_6$  octahedron has  $mm2$  symmetry and exhibits 4 + 2 shortened octahedral conformation geometry with four long Mn–O bonds of 2.183(2) Å and two short apical Mn–O bonds of 2.166(2) Å. The *cis* O–Mn–O angles are less deviated from 90° and the *trans* O–Mn–O angles are equal to 180° (Table 1). The 3D anionic Mn-formate framework in both modifications has cavities occupied by the template  $Ace^+$  cations (Fig. 3a and b).

In the LT monoclinic phase the cavity has three edges (Mn–OCO–Mn) with distances of 5.9665(8) (×4), 6.1386(11) (×4) and 6.3824(11) Å (×4); the shortest linkages are along the *b* direction, whereas the longer ones are along the *ac* plane. These distances are the Mn…Mn through the bridging formate anions. The angles between the edges slightly deviate from 90.0° (Table 1). In the orthorhombic phase the cavity has eight long edges that lie in the *ac* plane (6.2869(9) Å) and four slightly shorter ones (6.0445(9) Å) that are parallel to the *b* direction. The cavity volume calculated by the PLATON software<sup>44</sup> is equal to ~84 Å<sup>3</sup> for the monoclinic LT phase and ~88 Å<sup>3</sup> for the orthorhombic HT phase. In the cavity of the LT modification, the planar  $Ace^+$  cation (except for the H atoms of the  $CH_3$  group) has its molecular plane nearly parallel to the

two body-diagonal lines but normal to the third one that interacts by four conventional N–H…O HBs with the anionic Mn-formate framework (Fig. 3a). The HB geometries with the N…O distances of 2.910(4)–3.045(3) Å and the N–H…O angles of 164–175° (Table 2) indicate different HB strengths. In the orthorhombic phase the cavity has about 5% greater volume than that of the LT phase. As a result, the  $Ace^+$  cation is held inside the cavity by weaker N–H…O HBs with the N…O distances and N–H…O angles of 3.111(18)–3.198(2) Å and 137–159°, respectively (Table 2). These HBs are too weak to overcome thermally induced motions and the  $Ace^+$  cation is disordered (except for the amine N1 group and the C3 atom) over four symmetrically equivalent positions (Fig. 3b).

A view of the crystal structure of the anionic pseudo-perovskite Mn-formate framework together with the  $Ace^+$  counter-ions in the cavities of both low- (monoclinic) and high-temperature (orthorhombic) phases is provided in Fig. S3a and S3b,<sup>†</sup> respectively. The anionic pseudo-perovskite cavities with the edges of Mn–OCO–Mn exhibit  $C_{2h}$  and  $D_{2h}$  symmetry at low and high temperatures, respectively.

It is worth noting that although AceMn has a different orthorhombic structure (space group  $Imma$ ) from the related GuaMn and formamidinium analogues with  $M^{II} = Fe, Co$  and  $Mg$  (space group  $Pnna$ ),<sup>9,28,45,46</sup> the metrics of the unit cell as well as the topology of the frameworks are very similar. In particular, all these compounds possess pseudo-tetragonal cavities in the metal formate framework with the shortest distances between  $M^{II}$  ions parallel to the *b* direction (5.8703, 5.989 and 6.0445 Å) and the longer ones of 6.0661, 6.219 and 6.2869 Å lying in the *ac* plane for FMDFe, GuaMn and AceMn, respectively (see Table S2<sup>†</sup>). These cavities accommodate counter-ions and they are less distorted for AceMn (Mn–Mn–Mn angles are 88.173–91.827°) compared to GuaMn and FMDFe, for which the M–M–M angles are 86.493–93.507° and 86.668–93.507°, respectively (Table S2<sup>†</sup>). Comparison of these compounds also shows that the size of the unit cell is the smallest for formamidinium analogues and it increases by 3.3% when going from GuaMn to AceMn. It is worth adding



**Fig. 3** View of the N–H…O hydrogen bonded  $Ace^+$  cation inside the pseudo-perovskite cavity for (a) monoclinic and (b) orthorhombic phases. For the orthorhombic phase, the green, blue, yellow and light purple represent the four possible orientations of  $Ace^+$  cation (C3 and amine N1 atoms are ordered). The symmetry code is the same as in Table 1.

**Table 2** Hydrogen bond geometry in AceMn (Å, °)

100(1) K				
D–H…A	D–H	H…A	D…A	D–H…A
N1–H11…O1 <sup>i</sup>	0.87(3)	2.13(3)	2.989(4)	169(3)
N1–H12…O3 <sup>ii</sup>	0.77(3)	2.30(3)	3.045(3)	164(3)
N2–H21…O2 <sup>i</sup>	0.85(3)	2.08(3)	2.910(4)	167(3)
N2–H22…O6	0.89(3)	2.05(3)	2.939(4)	175(3)
Symmetry codes: (i) $-x + 1/2, y + 1/2, -z + 1/2$ ; (ii) $x, y, z - 1$ .				
330(1) K <sup>a</sup>				
D–H…A	D–H	H…A	D…A	D–H…A
N1–H1N…O2	0.85(4)	2.52(3)	3.198(2)	137(2)
N1–H1N…O2 <sup>j</sup>	0.85(4)	2.52(3)	3.198(2)	137(2)
N2–H2A…O2 <sup>ii</sup>	0.86	2.34	3.131(19)	153
N2–H2B…O1 <sup>iii</sup>	0.86	2.29	3.111(18)	159
Symmetry codes: (i) $-x, y, z$ ; (ii) $x - 1/2, y, -z + 1/2$ ; (iii) $-x, -y + 1, -z$ .				

<sup>a</sup> The two last HBs involving the N2 atom are disordered over four equivalent positions.



here that the effective size of the  $\text{Ace}^+$  cation was estimated as very close (2.77 Å)<sup>33</sup> to the size of  $\text{Gua}^+$  (2.78 Å).<sup>34</sup> A significantly larger unit cell volume and the distances between  $\text{Mn}^{\text{II}}$  for  $\text{AceMn}$  compared to  $\text{GuaMn}$  indicate, however, that the ionic size of  $\text{Ace}^+$  is larger compared to  $\text{Gua}^+$ . This conclusion is also consistent with the structural data reported for isostructural  $[\text{CH}_3\text{C}(\text{NH}_2)_2]_2[\text{KFe}(\text{CN})_6]$  and  $[\text{C}(\text{NH}_2)_3]_2[\text{KFe}(\text{CN})_6]$  compounds that also revealed 5.0% larger unit cell volume in the former case.<sup>32</sup> Table S2† also shows that the N...O distances in  $\text{AceMn}$  (3.111–3.198 Å at 330 K and 2.910–3.045 Å at 100 K) are longer than in  $\text{GuaMn}$  (2.953–2.991 Å at 293 K),  $\text{FMDFe}$  (2.8883 and 2.9340 Å at 294 K) and  $\text{FMDMn}$  (2.8876 Å at 355 K, and 2.8570 and 2.8984 Å at 110 K). This behavior is consistent with increasing ionic size when going from  $\text{FMD}^+$  to  $\text{Gua}^+$  and  $\text{Ace}^+$ . Weaker HBs for  $\text{AceMn}$  compared to  $\text{FMDMn}$  explain its lower phase transition temperature (303 K) compared to  $\text{FMDMn}$  (334 K).<sup>29</sup> It is worth adding that although HBs are weaker for  $\text{GuaMn}$  than for  $\text{FMDMn}$ , this compound remains ordered up to the decomposition temperature.<sup>45</sup> This behavior can be attributed to the fact that each  $\text{Gua}^+$  cation forms six HBs whereas  $\text{AceMn}$  and  $\text{FMDMn}$  form only four such bonds.

### Dielectric studies

The structural analysis reveals the ordering of the  $\text{Ace}^+$  cations with decreasing temperature. Such a process is usually accompanied by a dielectric response. We have, therefore, measured the temperature-variable complex dielectric constant ( $\epsilon$ ) ( $\epsilon = \epsilon' - i\epsilon''$ , where  $\epsilon'$  is the real part and  $\epsilon''$  is the imaginary part) of  $\text{AceMn}$  at frequencies from 1 Hz to 1 MHz. As shown in Fig. 4a the real part of dielectric permittivity takes relatively small values of around 4 in the vicinity of the phase transition. Similar behaviour was previously reported for other perovskite-like MOFs such as tetrapropylammonium metal dicyanamides

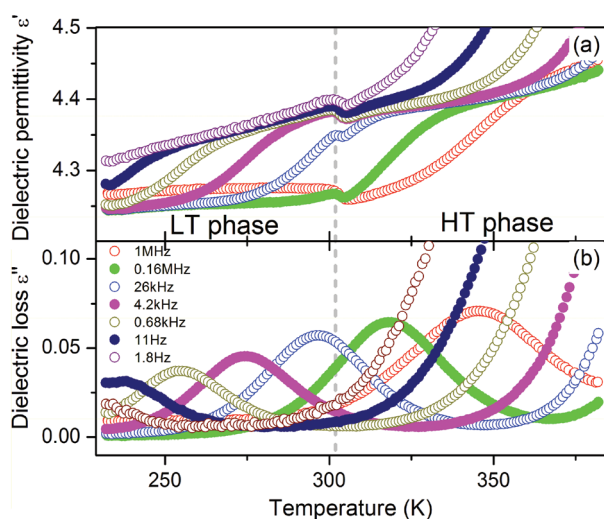


Fig. 4 (a) Dielectric permittivity and (b) dielectric loss spectra as a function of temperature in  $\text{AceMn}$ . The representative curves are plotted in frequency decades between 1.8 Hz and 1 MHz. The dashed line corresponds to the structural phase transition temperature.

and trimethylammonium metal azides.<sup>47,48</sup> Generally, the observed tendency is that  $\epsilon'$  increases monotonically with increasing temperature at all frequencies. There is a clear anomaly at around 304 K in the plot of  $\epsilon'$  vs. temperature. It is natural to associate this anomaly with the order–disorder phase transition in the investigated material. Additionally, in the whole investigated temperature range  $\epsilon'$  shows a regular frequency dispersion. At the same time  $\epsilon''$  exhibits strong frequency-dependent changes with no distinct anomalies owing to the structural phase transition (see Fig. 4b). The peak maximum shifts towards higher temperatures with increasing frequency, implying a relaxor nature of the observed electric ordering. In the investigated host–guest compound, the dielectric permittivity  $\epsilon'$  is generally determined by the possessing dipole moment of the  $\text{Ace}^+$  cation. The dynamic movement of this molecule modifies the internal electric field and leads to dielectric permittivity changes. The ordering of the  $\text{Ace}^+$  cations leads to a change of the orientation of the internal dipole moment of these cations and in consequence modifies the shape of  $\epsilon'$ , especially at the phase transition temperature.

In order to identify the nature of the relaxation process occurring in the material, the frequency-dependent dielectric permittivity of  $\text{AceMn}$  for several isotherms was analyzed. The isothermal dielectric spectra of  $\text{AceMn}$  at temperatures above and below the structural phase transition at 304 K are presented in Fig. 5. Close inspection of these spectra, especially for the  $\epsilon''$  as a function of frequency, reveals the occurrence of two dipolar relaxation processes. In the investigated temperature and frequency ranges, the faster relaxation process is well visible in both phases, whereas the slower one can be noticed in the experimental frequency window just above the phase transition temperature. Both the observed dipolar dielectric responses of  $\text{AceMn}$  deviate from the classical Debye behavior. The dielectric loss spectra were fitted (see Fig. S4†) by the sum

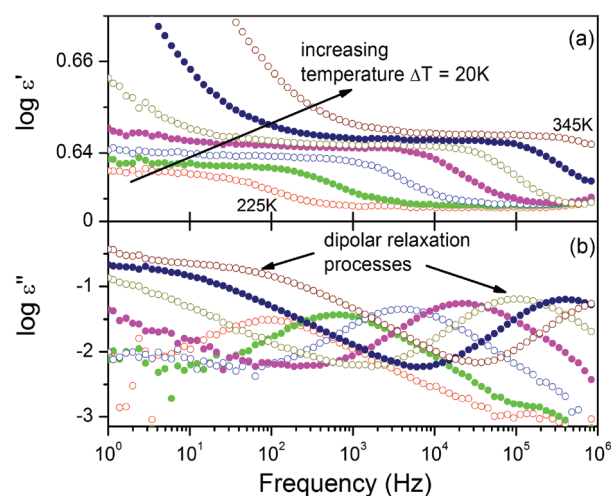


Fig. 5 Frequency dependence of (a) the dielectric permittivity ( $\epsilon'$ ) and (b) loss spectra ( $\epsilon''$ ) for selected temperatures. The arrow indicates the direction of shift of the relaxation process on heating. The dipolar relaxation processes were depicted in  $\log \epsilon''$  spectra.



of the conductivity and the two Havriliak–Negami (HN) equations:

$$\varepsilon''(\omega) = \text{Im} \left( \varepsilon_{\infty} + \sum_i \frac{\Delta\varepsilon_i}{(1 + (i\omega\tau_i)^{\alpha_i})^{\beta_i}} + \frac{\sigma_{\text{DC}}}{\varepsilon_0\omega} \right) \quad (1)$$

where  $\varepsilon_{\infty}$  is the high-frequency dielectric constant,  $\Delta\varepsilon_i$  is the dielectric strength,  $\tau_i$  is the dielectric relaxation time,  $\alpha$  and  $\beta$  are the shape parameters and  $\sigma_{\text{DC}}$  is the dc conductivity;  $i$  means the faster or the slower process, respectively. For the low temperature data (from 225 K up to 300 K), where the faster process is present, only the single HN function was used to parametrize the spectra.

In order to describe the relaxation dynamics of the observed dipolar processes, we compare the temperature dependent behavior of its dielectric relaxation times, estimated from the fits by eqn (1), as a function of  $1000/T$  (see Fig. 6). This relaxation map mirrors the dynamic properties of the system and the central feature of the investigated structure.

Closer inspection of the slower process reveals that the thermal activation of their relaxation time  $\tau_s$  spans through almost four decades in frequency. In the high-temperature (HT) phase,  $\tau_s$  exhibits almost linear dependence on the inverse temperature. Therefore, the relaxation times in the first approximation can be modelled using the Arrhenius relationship:

$$\tau = \tau_0 \exp \left( \frac{E_a}{k_B T} \right) \quad (2)$$

where  $\tau_0$ ,  $k_B$  and  $E_a$  are the relaxation time in the high temperature limit, the Boltzmann constant and the activation energy, respectively. It is evident that the experimental data fitted to the Arrhenius law is fulfilled with the characteristic time

$\log_{10} \tau_0 = -17.9$  s ( $\tau_0 = 1.25 \times 10^{-18}$  s). The physical nature and mechanism of this relaxation process in the examined material are attributed to the reorientational motion of  $\text{Ace}^+$  cations in the structure. From the structural data, it is known that the  $\text{Ace}^+$  cation is held inside the cavity by HBs, which are too weak in the HT phase to overcome thermally induced motions leading to the  $\text{Ace}^+$  cation disorder. The rotation dynamics of the  $\text{Ace}^+$  ions is mostly controlled by the confinement effect, including HB interactions. The deformation of the framework during the phase transition results in different confinement for the  $\text{Ace}^+$  cations and in consequence to its ordering.

The activation energy and dielectric relaxation, attributed to the active rotation of the guest cations, strongly depend on the chemical components. The  $E_a$  values in amine-templated frameworks may vary from tens of meV up to several eV.<sup>48–52</sup> The relatively high value of  $E_a$  in the HT phase ( $E_a = 1.1$  eV) indicates that it is not easy to force the  $\text{Ace}^+$  cation to oscillate with the applied electric field. The freedom of motions in the disordered phase definitely makes these oscillations difficult. Based on the data reported for other compounds,<sup>49,52</sup> we expect that in the LT (ordered) phase of  $\text{AceMn}$  the  $E_a$  should significantly decrease. It is also worth noting that the value of the relaxation time attributed to the  $\text{Ace}^+$  motions ( $\tau_0 = 1.25 \times 10^{-18}$  s) is relatively small. For instance, the relaxation time found in dimethylammonium analogues is of the order of  $10^{-13}$ – $10^{-14}$  s whereas for  $[\text{CH}_3\text{NH}_2\text{NH}_2][\text{Zn}(\text{HCOO})_3]$  and  $[\text{CH}_3\text{CH}_2\text{NH}_3][\text{Mg}(\text{HCOO})_3]$  the relaxation time is smaller, *i.e.*  $3.9 \times 10^{-16}$  s and  $5.3 \times 10^{-16}$  s, respectively.<sup>5a,19a,53,54</sup> The mentioned compounds differ significantly with respect to the HB network. In particular, each  $\text{DMA}^+$  and  $\text{CH}_3\text{CH}_2\text{NH}_3^+$  cation forms two<sup>5a,24a</sup> and three<sup>19a</sup> HBs, respectively, four H-atoms in the  $\text{CH}_3\text{NH}_2\text{NH}_2^+$  cation are able to form HBs but only those involving the central  $\text{NH}_2$  group are relatively strong<sup>16</sup> whereas all four H-atoms of the  $\text{Ace}^+$  cations in  $\text{AceMn}$  are involved in relatively strong HBs. This comparison suggests, therefore, that the relaxation time is dominated by the number and strength of HBs.

The faster, also dipolar relaxation process, occurring in the whole investigated temperature range, exhibits an activation type of dynamics with a lower magnitude of activation energy, *i.e.*,  $E_a = 0.6$  eV and  $\log_{10} \tau_0 = -15.2$  s ( $\tau_0 = 6.31 \times 10^{-16}$  s). This  $E_a$  value is about two times smaller in comparison with that of the  $\text{Ace}^+$  relaxation process. Such behavior, together with the smaller relaxation time, indicates that the intermolecular stiffness of the cage-like anionic framework is weaker. It is worth noting that the symmetry changes and the associated framework deformation at the phase transition have no significant impact on the activation energy of the framework relaxation. This fact clearly confirms that the  $\text{Ace}^+$  cation ordering is responsible for the observed order–disorder phase transition.

### Magnetic properties

Fig. 7 displays the results of magnetic property measurements performed for  $\text{AceMn}$ . At high temperatures the compound shows paramagnetic behavior with a linear temperature depen-

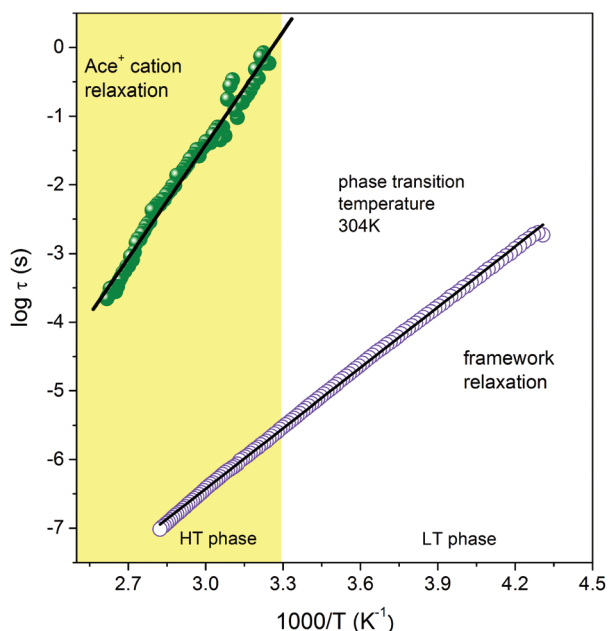
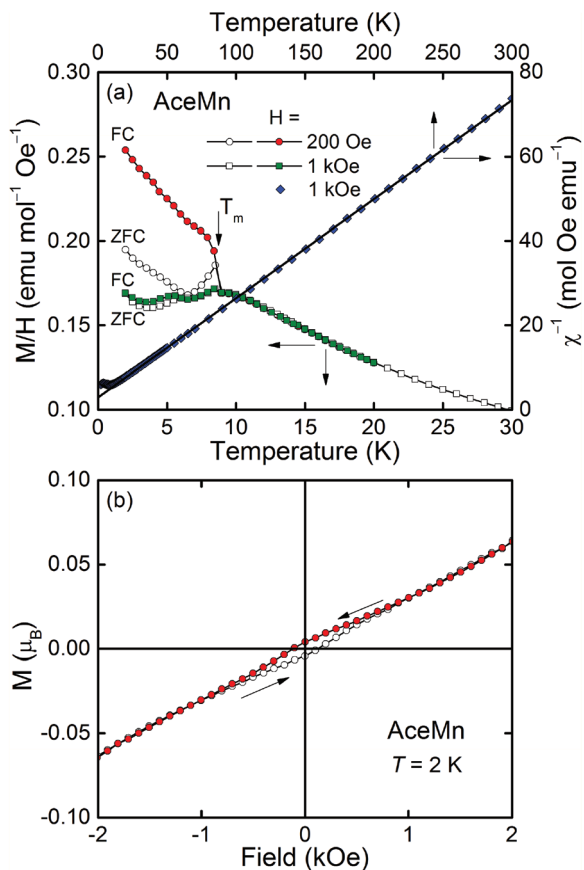


Fig. 6 Relaxation times of two dipolar processes as a function of  $1000/T$ .





**Fig. 7** (a) Left and bottom axes: The temperature variation of magnetization  $M$  of AceMn divided by the applied magnetic field  $H$  and measured in zero-field-cooling (ZFC) and field-cooling (FC) regimes; solid lines serve as guides for the eye and the arrow marks the ordering temperature  $T_m$ . Right and top axes: The inverse magnetic susceptibility  $\chi^{-1}$  vs.  $T$ ; the solid line is a fit of the Curie–Weiss law to the experimental data. (b)  $M$  measured at the lowest temperature upon increasing and decreasing field (open and closed symbols, respectively); solid lines and arrows serve as guides for the eye.

dence of the inverse magnetic susceptibility  $\chi^{-1}$  (see the right and top axes in Fig. 7a). The experimental  $\chi^{-1}(T)$  data can be described by the conventional Curie–Weiss law:

$$\chi(T) = \frac{C}{T - \theta_p} \quad (3)$$

where  $C$  is the Curie constant and  $\theta_p$  is the paramagnetic Curie–Weiss temperature. Least squares fits of eqn (3) to the experimental data yielded the parameters  $C = 4.25(1)$  emu mol $^{-1}$  Oe $^{-1}$  and  $\theta_p = -12.3(2)$  K. The effective magnetic moment  $\mu_{\text{eff}}$  derived from the Curie constant as  $\mu_{\text{eff}} = \sqrt{8C}$  is equal to  $5.83(1)\mu_B$ , which is close to  $5.92\mu_B$  calculated for free Mn $^{2+}$  ions within the Russel–Sounders coupling scheme, *i.e.* for the  ${}^6S_{5/2}$  term of the electron configuration  $3d^5$  ( $S = 5/2$ ,  $L = 0$ ,  $J = 5/2$  and  $g = 2$ ). This finding fully corroborates the results of the structural studies of AceMn, showing the presence of Mn $^{2+}$  in the crystal structure. The negative value of  $\theta_p$  points at the predominantly antiferromagnetic character of the interactions between the magnetic moments.

A distinct anomaly observed in  $M(T)/H$  at low temperatures (see the left and bottom axes in Fig. 7a) indicates an onset of long-range ordering of the magnetic moments of Mn below  $T_m = 9.0(5)$  K. The cusp-like shape of this anomaly in  $H = 1$  kOe and the hardly visible difference between the curves measured in that field in zero-field-cooling (ZFC) and field-cooling (FC) regimes indicate the antiferromagnetic character of the ordering, being in line with the negative value of the paramagnetic Curie–Weiss temperature. However, distinct bifurcation of  $M(T)$  associated with some tendency of  $M$  to saturation visible in the FC curve measured in  $H = 200$  Oe suggests the presence of a small ferromagnetic component in the antiferromagnetic lattice, *i.e.* weak ferromagnetism. The origin of the small hump visible at about 5 K remains unclear.

The field dependence of the magnetization of AceMn measured at 2 K is displayed in Fig. 7b. At high fields the magnetization increases linearly with increasing magnetic field (as expected for antiferromagnets) achieving at the highest fields studied the values far below the saturation value  $M_S = 5\mu_B$  calculated for spin-only Mn $^{2+}$ . At low fields (below about 800 Oe) a magnetic hysteresis loop is noticeable with the coercivity field  $H_c = 130$  Oe and the remnant magnetization  $M_R = 0.004\mu_B$ , being in line with the postulated weak ferromagnetism. The ferromagnetic component results most probably from the canting of the antiferromagnetically ordered moments by a small angle. It is a consequence of an antisymmetric exchange interaction or single-ion anisotropy,<sup>55,56</sup> which is very likely in the compound studied. The canting angle  $\alpha$  can be estimated *via* the relationship  $\sin \alpha = M_R/M_S$  as about 0.05 degree.

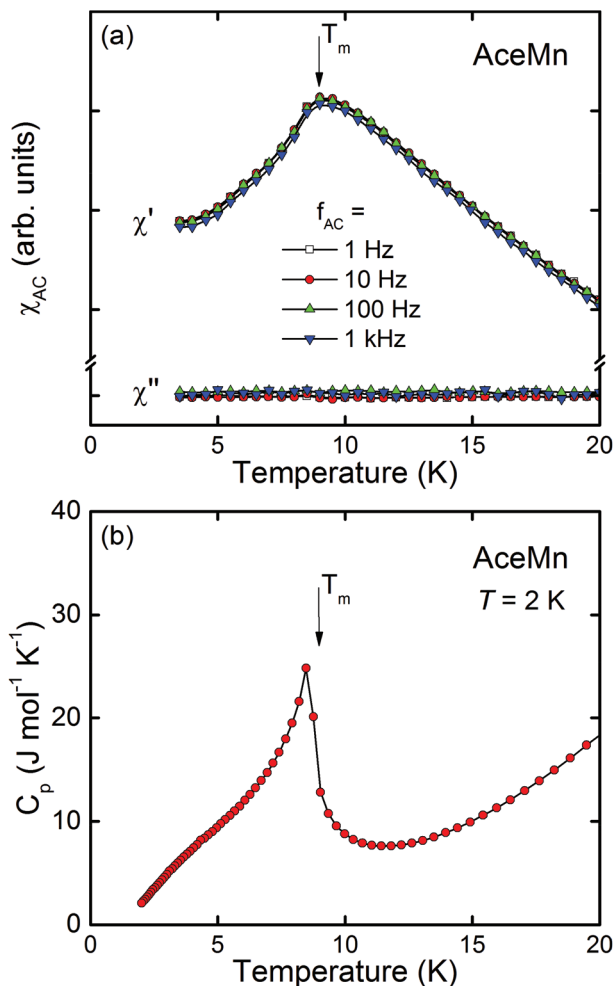
In order to verify the long-range character of the magnetic ordering of AceMn we measured the low-temperature AC magnetic susceptibility and specific heat of the compound (Fig. 8). A distinct cusp visible at  $T_m$  in the real part of the susceptibility and associated with the absence of any clear anomaly in its imaginary part (*cf.* Fig. 8a) confirms the predominantly antiferromagnetic character of the ordering of AceMn. The lack of any frequency dependence of either  $\chi'$  or  $\chi''$  allows us to safely exclude a spin-glass behavior. A sharp lambda-shaped peak observed in the temperature dependence of the specific heat (Fig. 8b) manifests a second-order phase transition, hence being another confirmation of the bulk, long-range character of the magnetic ordering at  $T_m$ .

The magnetic behavior of the compound AceMn is similar to that reported for other perovskite-like Mn-bearing MOFs, *e.g.*  $[(\text{CH}_3)_2\text{NH}_2]\text{Mn}(\text{HCOO})_3$ ,  $[\text{NH}_2\text{NH}_3]\text{Mn}(\text{HCOO})_3$  and  $[(\text{CH}_3)_2\text{ND}_2][\text{Mn}(\text{HCOO})_3]$ ,<sup>14a,24a,57</sup> which have been found to order magnetically at very similar temperatures, *i.e.* 8.5 K, 7.9 K and 8.5 K, respectively. AceMn is most probably another example of a weak ferromagnet.

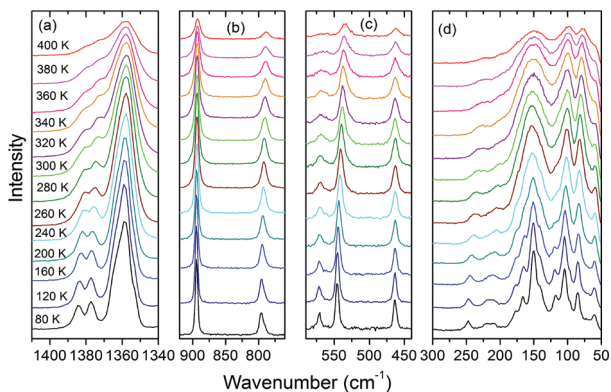
### Raman scattering and IR studies

Temperature-dependent Raman and IR spectra of AceMn are presented in Fig. 9, 10 and Fig. S5, S6.† The observed Raman and IR modes and their assignment are listed in Table S3.† We also present the temperature dependence of wavenumbers and

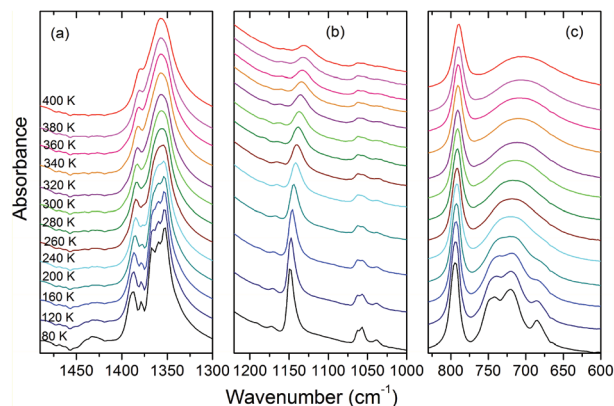




**Fig. 8** (a) Real ( $\chi'$ ) and imaginary ( $\chi''$ ) part of AC magnetic susceptibility of AceMn measured in zero applied magnetic field as a function of temperature. (b) Low-temperature specific heat  $C_p$  of AceMn measured in  $H = 0$ . Solid lines serve as guides for the eye and the arrows mark the ordering temperature  $T_m$ .



**Fig. 9** Details of the Raman spectra of AceMn corresponding to the spectral ranges 1410–1340, 920–760, 590–440 and 300–50  $\text{cm}^{-1}$ .



**Fig. 10** Details of the IR spectra of AceMn corresponding to the spectral ranges 1490–1300, 1220–1000 and 830–600  $\text{cm}^{-1}$ .

full width at half maximum (FWHM) for a few selected Raman-active modes in order to better see the changes induced by the phase transition (Fig. S7†).

The assignment of modes corresponding to the internal vibrations of  $\text{HCOO}^-$  is straightforward since these vibrations give rise to characteristic bands observed in narrow wavenumber ranges for different metal formate frameworks.<sup>4,6,29,58</sup> The internal vibrations of acetamidinium cations should be observed in a very broad wavenumber range, *i.e.*, from about 400 to 3400  $\text{cm}^{-1}$ .<sup>59</sup> Based on the literature data for formamidinium metal formates,<sup>29,46,60</sup> as well as acetamidinium salts and theoretical studies,<sup>59</sup> we assign the characteristics bands at 3150–3450, 2860–3100, 1700–1720, 1430–1430, 1157–1170, 1130–1149, 893, 685–742, 534–570 and 462  $\text{cm}^{-1}$  to  $\nu\text{NH}_2$ ,  $\nu\text{CH}_3$ ,  $\delta\text{NH}_2$ ,  $\delta\text{CH}_3$ ,  $\nu_{\text{as}}\text{CCN}$ ,  $\rho\text{NH}_2$ ,  $\nu_{\text{s}}\text{CCN}$ ,  $\omega\text{NH}_2$ ,  $\tau\text{NH}_2$  and  $\delta\text{NCN}$  modes, respectively. Lattice modes are observed in the Raman spectrum below 250  $\text{cm}^{-1}$  and similarly as in other formates, the major contribution to the observed bands comes from the librational motions of  $\text{HCOO}^-$  ions.<sup>58</sup>

Temperature-dependent Raman and IR spectra show weak changes upon cooling from 400 K to 300 K. They show, however, a few characteristic features upon cooling below 300 K. Firstly, the observed bands exhibit clear narrowing. This behaviour is most pronounced for the lattice modes (Fig. 9d) and the modes related to the vibrations of the  $\text{NH}_2$  groups. For instance, the FWHM of the IR bands at 3136 and 1130  $\text{cm}^{-1}$  changes from 289 and 31.3  $\text{cm}^{-1}$  at 400 K to 49.3 and 9.7  $\text{cm}^{-1}$ , respectively, at 80 K. This behaviour indicates that the phase transition is associated with ordering of the  $\text{Ace}^+$  cations. Secondly, a number of modes split at low temperatures. In particular, a single Raman (IR) band corresponding to the  $\nu_5\text{HCOO}^-$  mode near 1360  $\text{cm}^{-1}$  splits into three components. This behavior proves that the phase transition is also associated with distortion of the manganese formate framework. Thus IR and Raman data are consistent with X-ray diffraction that revealed the presence of two unique  $\text{HCOO}^-$  ions with equal C–O bonds in the HT phase and three such ions with all C–O bonds different in the LT phase. It is worth adding that the observed splitting of the  $\nu_5$  modes is





less pronounced than in the related  $[\text{NH}_2\text{CHNH}_2][\text{Mn}(\text{HCOO})_3]$  compound,<sup>29</sup> indicating smaller distortion of the framework in the former case. Thirdly, the  $\nu\text{NH}_2$  bands exhibit a very pronounced shift towards lower wavenumbers upon cooling (up to  $62\text{ cm}^{-1}$ , see Fig. S5–S7†). This behavior points to a strong increase of the HB strength upon cooling.

## Conclusions

We report the synthesis of the first metal formate framework templated by  $\text{Ace}^+$  cations. In contrast to the related formamidinium manganese formate crystallizing in the trigonal structure,<sup>29</sup>  $\text{AceMn}$  crystallizes in the orthorhombic structure, space group *Imma*, with disordered  $\text{Ace}^+$  cations. Similar orthorhombic structures were, however, reported for guanidinium ( $\text{M}^{\text{II}} = \text{Mn, Fe, Co, Ni, Zn}$ ) and formamidinium ( $\text{M}^{\text{II}} = \text{Mg, Co, Fe}$ ) analogues.<sup>10,28,46</sup>  $\text{AceMn}$  undergoes a temperature-induced phase transition at 304 K into the monoclinic  $P2_1/n$  structure. X-ray diffraction, DSC, dielectric, Raman and IR data prove that the phase transition is associated with the ordering of the  $\text{Ace}^+$  cations and distortion of the manganese formate framework. IR and Raman data provide evidence for large strengthening of HBs upon cooling. This compound also exhibits weak ferromagnetic properties below 9.0 K.

## References

- (a) A. Rossin, G. Giambastiani, M. Peruzzini and R. Sessoli, *Inorg. Chem.*, 2012, **51**, 6962–6968; (b) X. Ma, J. Tian, H. Y. Yang, K. Zhao and X. Li, *J. Solid State Chem.*, 2013, **201**, 172–177.
- M. Ptak, M. Mączka, A. Gągor, P. Głuchowski and L. Macalik, *J. Mater. Chem. C*, 2016, **4**, 1019–1028.
- L. Cañadillas-Delgado, O. Fabelo, J. A. Rodríguez-Velamazán, M. H. Lemée Cailleau, S. A. Mason, E. Pardo, F. Lloret, J. P. Zhao, X. H. Bu, V. Simonet, C. V. Colin and J. Rodríguez-Carvajal, *J. Am. Chem. Soc.*, 2012, **134**, 19772–19781.
- (a) A. Ciupa, M. Mączka, A. Gągor, A. Sieradzki, J. Trzmiel, A. Pikul and M. Ptak, *Dalton Trans.*, 2015, **44**, 8846–8854; (b) A. Ciupa, M. Mączka, A. Gągor, A. Pikul and M. Ptak, *Dalton Trans.*, 2015, **44**, 13234–13241; (c) M. Mączka, A. Ciupa, A. Gągor, A. Sieradzki, A. Pikul and M. Ptak, *J. Mater. Chem. C*, 2016, **4**, 1186–1193.
- (a) M. Mączka, B. Bondzior, P. Dereń, A. Sieradzki, J. Trzmiel, A. Pietraszko and J. Hanuza, *Dalton Trans.*, 2015, **44**, 6871–6879; (b) M. Ptak, M. Mączka, A. Gągor, A. Sieradzki, B. Bondzior, P. Dereń and S. Pawlus, *Phys. Chem. Chem. Phys.*, 2016, **18**, 29629–29640.
- M. Mączka, A. Gągor, N. L. M. Costa, W. Paraguassu, A. Sieradzki and A. Pikul, *J. Mater. Chem. C*, 2016, **4**, 3185–3194.
- (a) Z.-M. Wang, B. Zhang, H. Fujiwara, H. Kobayashi and M. Kurmoo, *Chem. Commun.*, 2004, 416–417; (b) X. Y. Wang, L. Gan, S. W. Zhang and S. Gao, *Inorg. Chem.*, 2004, **43**, 4615–4625.
- P. Jain, V. Ramachandran, R. J. Clark, H. D. Zhou, B. H. Toby, N. S. Dalal, H. W. Kroto and A. K. J. Cheetham, *J. Am. Chem. Soc.*, 2009, **131**, 13625–13627.
- B. Zhou, Y. Imai, A. Kobayashi, Z.-M. Wang and H. Kobayashi, *Angew. Chem., Int. Ed.*, 2011, **50**, 11441–11445.
- K.-L. Hu, M. Kurmoo, Z. Wang and S. Gao, *Chem. – Eur. J.*, 2009, **15**, 12050–12064.
- Z. Wang, K. Hu, S. Gao and H. Kobayashi, *Adv. Mater.*, 2010, **22**, 1526–1523.
- G. C. Xu, W. Zhang, X. M. Ma, Y. H. Hen, L. Zhang, H. L. Cai, Z. M. Wang, R. G. Xiong and S. J. Gao, *J. Am. Chem. Soc.*, 2011, **133**, 14948–14951.
- L. C. Gómez-Aguirre, B. Pato-Doldán, J. Mira, S. Castro-García, M. A. Señaris-Rodríguez, M. Sánchez-Andújar, J. Singleton and V. S. Zapf, *J. Am. Chem. Soc.*, 2016, **138**, 1122–1125.
- (a) S. Chen, R. Shang, K.-L. Hu, Z.-M. Wang and S. Gao, *Inorg. Chem. Front.*, 2014, **1**, 83–98; (b) S. Chen, R. Shang, B.-W. Wang, Z.-M. Wang and S. Gao, *Angew. Chem., Int. Ed.*, 2015, **54**, 11093–11096.
- M. Mączka, K. Pasińska, M. Ptak, W. Paraguassu, T. Almeida da Silva, A. Sieradzki and A. Pikul, *Phys. Chem. Chem. Phys.*, 2016, **18**, 31653–31663.
- M. Mączka, A. Gągor, M. Ptak, W. Paraguassu, T. A. da Silva, A. Sieradzki and A. Pikul, *Chem. Mater.*, 2017, **29**, 2264–2275.
- (a) A. Stroppa, P. Barone, P. Jain, J. M. Perez-Mato and S. Picozzi, *Adv. Mater.*, 2013, **25**, 2284–2290; (b) D. Di Sante, A. Stroppa, P. Jain and S. Picozzi, *J. Am. Chem. Soc.*, 2013, **135**, 18126–18130; (c) Y. Tian, A. Stroppa, Y. S. Chai, P. Barone, M. Perez-Mato, S. Picozzi and Y. Sun, *Phys. Status Solidi RRL*, 2015, **9**, 62–67.
- S. Ghosh, D. Di Sante and A. Stroppa, *J. Phys. Chem. Lett.*, 2015, **6**, 4553–4559.
- (a) R. Shang, G.-C. Xu, Z. M. Wang and S. Gao, *Chem. – Eur. J.*, 2014, **20**, 1146–1158; (b) I. E. Colins, J. A. Hill, A. B. Cairns, R. I. Cooper, A. L. Thompson, J. E. Parker, C. C. Tang and A. L. Goodwin, *Dalton Trans.*, 2016, **45**, 4169–4178.
- W. Li, M. R. Probert, M. Kosa, T. D. Bennett, A. Thirumurugan, R. P. Burwood, M. Parinello, J. A. K. Howard and A. K. Cheetham, *J. Am. Chem. Soc.*, 2012, **134**, 11940–11943.
- M. Sánchez-Andújar, S. Presedo, S. Yáñez-Vilar, S. Castro-García, J. Shamir and M. A. Señaris-Rodríguez, *Inorg. Chem.*, 2010, **49**, 1510–1516.
- Y. Tian, A. Stroppa, Y. Chai, L. Yan, S. Wang, P. Barone, S. Picozzi and Y. Sun, *Sci. Rep.*, 2014, **4**, 6062.
- M. Mączka, A. Pietraszko, B. Macalik and K. Hermanowicz, *Inorg. Chem.*, 2014, **53**, 787–794.
- (a) M. Mączka, A. Gągor, B. Macalik, A. Pikul, M. Ptak and J. Hanuza, *Inorg. Chem.*, 2014, **53**, 457–467; (b) M. Mączka, A. Sieradzki, B. Bondzior, P. Dereń, J. Hanuza and K. Hermanowicz, *J. Mater. Chem. C*, 2015, **3**, 9337–9345.



- 25 W.-J. Xu, Z.-Y. Du, W.-X. Zhang and X.-M. Chen, *CrystEngComm*, 2016, **18**, 7915–7928.
- 26 K. Asadi and M. A. van der Veen, *Eur. J. Inorg. Chem.*, 2016, 4332–4344.
- 27 B. Liu, R. Shang, K.-L. Hu, Z.-M. Wang and S. Gao, *Inorg. Chem.*, 2012, **51**, 13363–13372.
- 28 A. Rossin, M. R. Chierotti, G. Giambastiani, R. Gobetto and M. Peruzzini, *CrystEngComm*, 2012, **14**, 4454–4460.
- 29 M. Mączka, A. Ciupa, A. Gaĝor, A. Sieradzki, A. Pikul, B. Macalik and M. Drozd, *Inorg. Chem.*, 2014, **53**, 5260–5268.
- 30 (a) P. Pato-Doldán, L. C. Gómez-Aguirre, J. M. Bermúdez-García, M. Sánchez-Andújar, A. Fondado, J. Mira, S. Castro-García and M. A. Señaris-Rodríguez, *RSC Adv.*, 2013, **3**, 22404–22411; (b) M. Mączka, N. L. M. Costa, A. Gaĝor, W. Paraguassu, A. Sieradzki and J. Hanuza, *Phys. Chem. Chem. Phys.*, 2016, **18**, 13993–14000.
- 31 C.-Y. Han, M.-M. Liu and Q.-Q. Dang, *Acta Crystallogr., Sect. E: Struct. Rep. Online*, 2013, **69**, m541.
- 32 W. J. Xu, K. P. Xie, Z. F. Xiao, W. X. Zhang and X. M. Chen, *Cryst. Growth Des.*, 2016, **76**, 7212–7217.
- 33 W. Travis, E. N. K. Glover, H. Bronstein, D. O. Scanlon and N. G. Palgrave, *Chem. Sci.*, 2016, **7**, 4548–4556.
- 34 (a) G. Kieslich, S. Sun and A. K. Cheetham, *Chem. Sci.*, 2014, **5**, 4712; (b) G. Kieslich, S. Sun and A. K. Cheetham, *Chem. Sci.*, 2015, **6**, 3430.
- 35 *CrysAlis CCD and CrysAlis Red, Version 1.171.38.41*, Rigaku Oxford Diffraction, 2015.
- 36 G. M. Sheldrick, *SHELXS97, Programs for Crystal Structures Solution and Refinement*, University of Göttingen, Göttingen, Germany, 1997.
- 37 G. M. Sheldrick, *Acta Crystallogr., Sect. C: Cryst. Struct. Commun.*, 2015, **71**, 3–8.
- 38 K. Brandenburg and H. Putz, *DIAMOND Version 3.0*, Crystal Impact GbR, Bonn, Germany, 2006.
- 39 M. Ptak, M. Mączka, A. Gaĝor, A. Sieradzki, A. Stroppa, D. Di Sante, J. M. Perez-Mato and L. Macalik, *Dalton Trans.*, 2016, **45**, 2574–2583.
- 40 F. F. Wang, C. Chen, Y. Zhang, H. Y. Ye, Q. Ye and D. W. Fu, *J. Mater. Chem. C*, 2015, **3**, 6350–6358.
- 41 R. Samantaray, R. J. Clark, E. S. Choi and N. S. Dalal, *J. Am. Chem. Soc.*, 2012, **134**, 15953–15962.
- 42 A. Cornia, A. Caneschi, P. Dapporto, A. C. Faberetti, D. Gatteschi, W. Malevasi, C. Sangregorio and R. Sessoli, *Angew. Chem., Int. Ed.*, 1999, **38**, 1780–1781.
- 43 (a) S. R. Batten and R. Robson, *Angew. Chem., Int. Ed. Engl.*, 1988, **37**, 1460–1494; (b) O. M. Yaghi, M. O’Keefe, N. W. Ockwig, H. K. Chae, M. Eddaoudi and J. Kim, *Nature*, 2003, **423**, 705–714.
- 44 A. L. Spek, *PLATON, A Multipurpose Crystallographic Tool*, Utrecht University, 2001.
- 45 W. Li, A. Thirumurugan, P. T. Barton, Z. Lin, S. Henke, H. H. M. Yeung, M. T. Wharmby, E. G. Bithell, C. J. Howard and A. K. Cheetham, *J. Am. Chem. Soc.*, 2014, **136**, 7801–7804.
- 46 A. Ciupa, M. Mączka, A. Gaĝor, A. Pikul, E. Kucharska, J. Hanuza and A. Sieradzki, *Polyhedron*, 2015, **85**, 137–143.
- 47 J. M. Bermúdez-García, M. Sánchez-Andújar, S. Yáñez-Vilar, S. Castro-García, R. Artiaga, J. López-Beceiro, L. Botana, A. Alegria and M. A. Señaris-Rodríguez, *J. Mater. Chem. C*, 2016, **4**, 4889–4898.
- 48 Z.-Y. Du, Y.-Z. Sun, S.-L. Chen, B. Huang, Y.-J. Su, T. T. Xu, W.-X. Zhang and X.-M. Chen, *Chem. Commun.*, 2015, **51**, 15641–15644.
- 49 Z.-Y. Du, T. T. Xu, B. Huang, Y.-J. Su, W. Xue, C.-T. He, W.-X. Zhang and X.-M. Chen, *Angew. Chem., Int. Ed.*, 2015, **54**, 914–918.
- 50 G.-Z. Zou, H. Gao, J.-L. Liu, S. P. Zhao, Z.-F. Tian and X.-M. Ren, *RSC Adv.*, 2013, **3**, 23596–23603.
- 51 A. Sieradzki, S. Pawlus, S. N. Tripathy, A. Gaĝor, A. Ciupa, M. Mączka and M. Paluch, *Phys. Chem. Chem. Phys.*, 2016, **18**, 8462–8467.
- 52 A. Sieradzki, S. Pawlus, S. N. Tripathy, A. Gaĝor, M. Ptak, M. Paluch and M. Mączka, *Dalton Trans.*, 2017, **46**, 3681–3687.
- 53 T. Besara, P. Jain, N. S. Dalal, P. L. Kuhns, A. P. Reyes, H. W. Kroto and A. K. Cheetham, *Proc. Natl. Acad. Sci. U. S. A.*, 2011, **108**, 6828–6832.
- 54 M. Šimėnas, S. Balčiūnas, M. Trzebiatowska, M. Ptak, M. Mączka, G. Völkel, A. Pöppel and J. Banys, *J. Mater. Chem. C*, 2017, **5**, 4526–4536.
- 55 I. Dzyaloshinsky, *J. Phys. Chem. Solids*, 1958, **4**, 241–255.
- 56 T. Moriya, *Phys. Rev.*, 1960, **120**, 91–97.
- 57 W. Wang, L.-Q. Yan, J.-Z. Cong, Y.-L. Zhao, F. Wang, S.-P. Shen, T. Zou, D. Zhang, S.-G. Wang, X.-F. Han and Y. Sun, *Sci. Rep.*, 2013, **3**, 2024.
- 58 (a) M. Mączka, M. Ptak and L. Macalik, *Vib. Spectrosc.*, 2014, **71**, 98–104; (b) M. Mączka, W. Zierkiewicz, D. Michalska and J. Hanuza, *Spectrochim. Acta, Part A*, 2014, **128**, 674–680; (c) M. Mączka, K. Szymborska-Małek, A. Ciupa and J. Hanuza, *Vib. Spectrosc.*, 2015, **77**, 17–24.
- 59 (a) M. Davies and E. A. Parsons, *Z. Phys. Chem.*, 1959, **20**, 34–41; (b) E. Leon, B. Amekraz, J. Tortajada, J. P. Morizur, A. I. González, O. Mó and M. Yáñez, *J. Phys. Chem. A*, 1997, **101**, 2489–2495.
- 60 E. Kucharska, J. Hanuza, A. Ciupa, M. Mączka and L. Macalik, *Vib. Spectrosc.*, 2014, **75**, 45–50.

