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Reconciling the valence state with magnetism in mixed-valent polyoxometalates: the case of a $\{VO_2F_2@V_{22}O_{54}\}$ cluster†

Piotr Kozłowski,^{*a} Almudena Notario-Estévez,^b Coen de Graaf,^{bc} Xavier López ^b and Kirill Yu. Monakhov ^{*d}

The electronic structure and magnetism of the recently synthesised mixed-valent $\{V^VO_2F_2@V_{22}^{IV}VO_{54}\}$ (**1**) host–guest polyoxovanadate envisaged as a potential building block of a molecule-based quantum computer are analysed using density functional theory (DFT) and effective Hamiltonian calculations. The form of the t – J like effective Hamiltonian has been inspired by the acquired DFT data, and the valence state used in DFT calculations has been suggested by the fits to the experimental magnetic data with the effective Hamiltonian. This self-consistent approach breaks through the magnetochemical limitations of vanadium-oxo cluster **1**, giving results fully concordant with the experiment and allowing us to determine the valence state of **1**, which contrary to other members of this host–guest family appears to feature 9 valence electrons.

The use of polyoxometalates¹ (POMs) in the next-generation information technology^{2–4} requires control of their electronic charge and spin states at the molecular-to-nanoscale level.⁵ One of the fields currently being pursued in POM science is the development of molecular spin qubits based on reduced metal-oxo cluster units with magnetic ground states.^{6–10} Besides the exploration of the possibilities for a long quantum coherence lifetime¹¹ and scalability to a number of qubits, the development of magnetochemical methodologies to gain all-necessary insights into the potential single molecule-based quantum system before its attachment at a solid surface is mandatory.

Mixed-valent polyoxovanadates^{12,13} (POVs) feature a unique interplay between molecular charge on one side and the redox ($3d^1\text{-}V^{IV}/3d^0\text{-}V^V$) and magnetic states on the other, and as such they emerge as promising candidates for engineering molecular

qubit functionality. In sufficiently symmetric mixed-valent POVs some of the valence electrons are (partially) delocalised over the cage, giving rise to magnetic effects induced by hopping of the electrons.¹⁴ In particular, a spherical POV magnetically functionalised in the void with a metal-based species through weak host–guest interactions represents an interesting prototype of a supramolecular quantum system where the distinct magnetochemical effects of the host and their operation by external stimuli will be implemented by the guest entity. The distant predecessor of such an *endo*-functionalised POV spin capsule is a synthetically accessible, all-inorganic $\{V^VO_2F_2@V_{22}^{IV}VO_{54}\}$ (**1**) self-assembly with isotropic quantum spin 1/2 vanadyl $\{V^{IV}O\}^{2+}$ groups.‡ The difluorovanadate guest anion has been shown to cause drastic effects on the experimental magnetic characteristics of the vanadium-oxo host-shell.¹⁵ For such POMs it is, however, difficult to find magneto-structural relations, such as for dimers,¹⁶ and to determine precisely the valence state which is crucial for the design, modelling and optimisation of potential POM-based spin qubits. Modelling the electronic structure of POMs as **1** with delocalised electrons is challenging since the dynamics of hopping electrons must be taken into account, which leads to more complex and often computationally inaccessible effective Hamiltonians. Even if a simplified semi-classical version of a Hamiltonian is used it is often problematic to perform calculations for a full-size POM.¹⁴

Herein we have described for the first time the results of the effective Hamiltonian calculations of an aforementioned large, mixed-valent POV (**1**) taking into account the dynamics of the delocalised electrons. These calculations could be efficiently performed due to the use of a massively parallel computing and evolutionary algorithm. The particular form of the effective Hamiltonian was inspired by the results of DFT calculations that were introduced to shed light on the geometric and electronic peculiarities of this polyoxoanion. The self-consistent use of these two complementary theoretical techniques allows us to determine the valence state and microscopic parameters that lead to the full agreement with the experimentally measured magnetic properties of **1**.

Due to similar electrochemical properties, it has been hypothesised¹⁵ that the valence state of **1** is the same as those

^a Faculty of Physics, Adam Mickiewicz University in Poznań, ul. Umultowska 85, 61-614 Poznań, Poland. E-mail: kozl@amu.edu.pl

^b Departament de Química Física i Inorgànica, Universitat Rovira i Virgili, Marcel·lí Domingo 1, 43007 Tarragona, Spain

^c ICREA, Passeig Lluís Companys 23, 08010, Barcelona, Spain

^d Institut für Anorganische Chemie, RWTH Aachen University, Landoltweg 1, 52074 Aachen, Germany. E-mail: kirill.monakhov@ac.rwth-aachen.de

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of other $\{X@V_{22}O_{54}\}$ family members ($X = SCN^-$, ClO_4^-) and features 8 valence electrons. This means that one gets a POV anion, $[V^V O_2 F_2 @ H V^IV V^V O_{54}]^{6-}$ (**1a**), with a guest monoanion, $V^V O_2 F_2^-$, and one proton to maintain the total charge balance. The total number of unpaired electrons cannot be determined from the susceptibility measured at high temperature since the χT curve is not saturating at room temperature. Direct redox titration was impossible due to the low solubility of the compound.¹⁵ Moreover, the simplified calculations made for 8 valence electrons did not give good fits to the magnetic field dependence of magnetisation at low temperature.¹⁵ Thus, it cannot be excluded that the valence state of **1a** is different from that of SCN^- - and ClO_4^- -based family members and that the limited success of modelling **1a** in ref. 15 is caused not only by the use of an over-simplified model but also by the wrong valence state considered. Therefore, in this paper we additionally apply our modelling to **1** in two other valence states which are the closest to **1a**: $[V^V O_2 F_2 @ V^IV V^V O_{54}]^{6-}$ (**1b**) and $[V^V O_2 F_2 @ H_2 V^IV V^V O_{54}]^{6-}$ (**1c**).

DFT calculations were performed for the hydrogen-free species based on **1a–1c** (see technical details in the ESI†). The idealised maximal D_{2d} symmetry of the empty $\{V_{22}O_{54}\}$ cage decreases to C_{2v} by inclusion of the guest $VO_2F_2^-$ moiety. The optimised **1a–1c** structures have slightly different geometrical parameters, but in all the cases we identified several large imaginary vibrational frequencies, revealing that the C_{2v} structures are not real minima. In general, V^IV/V^V mixed-valent POVs feature small gaps between the highest singly occupied and the lowest unoccupied molecular orbitals (SOMO–LUMO gaps). The C_{2v} structures reveal narrow gaps ranging from 0.98 eV (**1a**) to 0.48 eV (**1c**) (Fig. 1). Removing the C_{2v} symmetry restrictions, we re-optimised

the structures with associated stabilisation energies of 0.14 eV (**1a**), 0.12 eV (**1b**) and 0.11 eV (**1c**). The resulting absolute (C_1) minima present significantly larger SOMO–LUMO gaps of 1.13 eV (**1a**) and 0.73 eV (**1c**). Such variations and the $C_{2v} \rightarrow C_1$ energy stabilisations denote a pseudo-Jahn–Teller (PJT) effect previously described theoretically for other POMs.¹⁷ The energies and occupations of relevant MOs in Fig. 1 illustrate facts associated with the PJT effect. Furthermore, molecular reduction **1b** \rightarrow **1a** \rightarrow **1c** gives a concomitant destabilisation of the occupied MOs of ~ 0.15 eV per unit charge. As such, species **1a** and **1c** have the highest doubly occupied MOs at -7.93 and -7.75 eV and the highest SOMOs at -5.53 and -5.38 eV, respectively.

The atomic spin densities (ASDs) of **1c** are nearly identical for the C_{2v} and C_1 minima (Table S1 and Fig. S1 in the ESI†). The apical V centres have ASDs close to 1.0 (*i.e.* V^IV). Neighbouring each apical site are two fully-oxidised V^V sites (ASDs 0.04–0.17). The rest of the metal centres are partially populated by unpaired electrons (ASDs 0.34–0.64) in the C_{2v} and C_1 minima, sharing the remaining 7 unpaired electrons of the molecule. Most unpaired electrons are delocalised over the major part of the V_{22} framework (16 mixed-valent V atoms, see Table 1). The removal of one electron from the high symmetry structure (**1c** \rightarrow **1a**) leads to electron density loss from the central V_8 ring, but if the C_1 structure is considered, in addition to electron loss from the central ring, the two apical electrons partially delocalize over the outer V_6 rings, and the apical V^IV centres become $V^{IV/V}$. The MOs show that pure apical-like orbitals in **1c** (Fig. S2, ESI† right) delocalize over a neighbouring site in **1a**, thus reducing the spin on apical sites to half (Fig. S2, ESI† left), partially explaining that both V_6 rings gain electron density. Remarkably, the central V_8

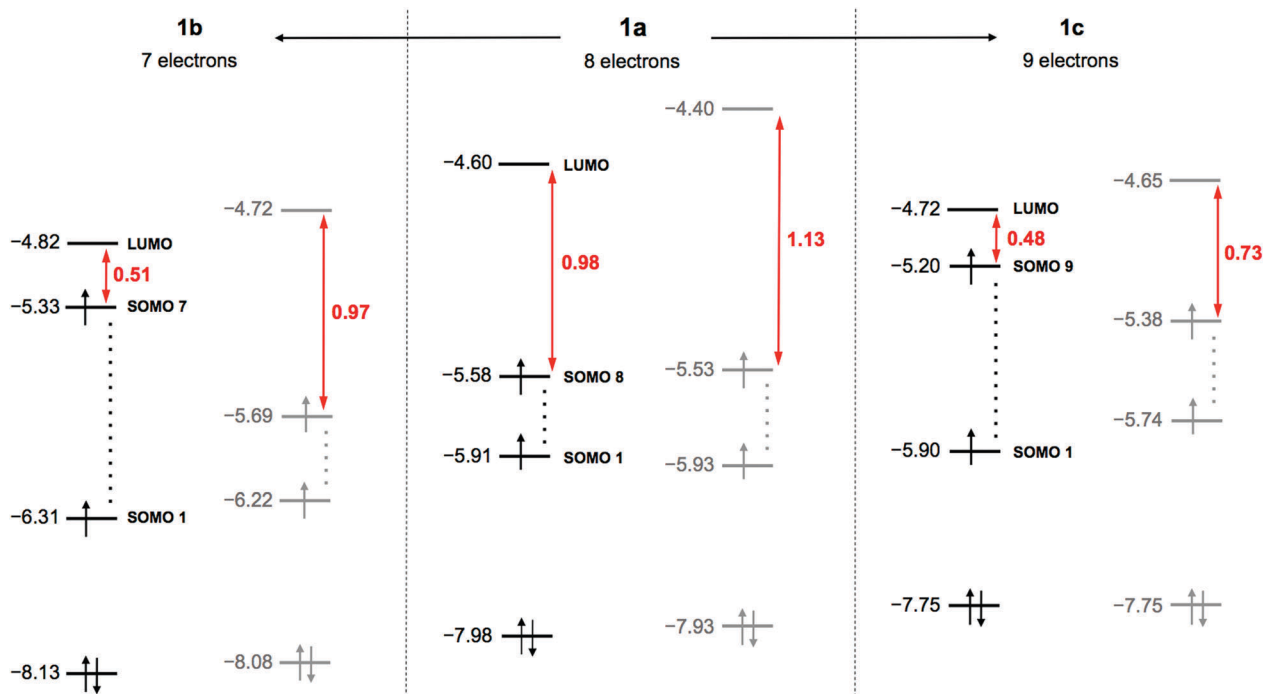


Fig. 1 Electronic structure of **1**, highlighting orbital energies for **1a–1c** in the C_{2v} (black) and C_1 (grey) forms, and SOMO–LUMO energy gaps (red). Values are given in eV.



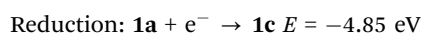
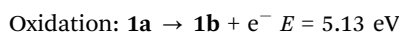
Table 1 Numbers of V^{IV} , V^V and $V^{IV/V}$ centres in structures **1a–1c** with high (C_{2v}) and low (C_1) symmetry, estimated from DFT calculations

		1b (7e)	1a (8e)	1c (9e)
C_{2v}	V^{IV}	2	2	2
	V^V	8	4	4
	$V^{IV/V}$	12	16	16
C_1	V^{IV}	0	0	2
	V^V	8	6	4
	$V^{IV/V}$	14	16	16

ring has four V^V centres. Comparing this electronic distribution with that of the less stable C_{2v} form of **1a** shows that the apical sites would be V^{IV} and two V^V sites would appear in each V_6 ring. No V^V atoms are found in the central ring. Another electron removal (**1a** to **1b**) in the C_{2v} structure results in a net loss of one electron in both outer V_6 rings combined with an increment of one electron in the central V_8 ring, the apical centres remaining as V^{IV} . The electron distribution of its C_1 form is identical to that of **1a** except for one electron removed from the V_6 ring situated in the proximity of the internal oxygens. Table S1 in the ESI† summarizes the relevant data discussed here.

The internal fragment orientation has some effect on the electron distribution. In the C_{2v} forms, **1a–1c** invariably feature one electron more in the hemisphere closer to F and the C_1 form of **1b** also concentrates more unpaired electrons in the F-hemisphere, likely due to the electrostatic effect of the $VO_2F_2^-$ unit. However, in the more electron-rich species **1a** and **1c** in C_1 , both hemispheres carry the same spin density. Apparently, structural distortion distributes the electron density in a more equitable way.

Species **1a–1c** can be electrochemically interconverted. Taking **1a** as the resting species, **1b** and **1c** can be obtained by oxidation or reduction with computed energies:



These theoretical values correspond to 0.37 and 0.65 eV vs. Ag/AgCl, respectively, in qualitative agreement with the measured values of 0.171 and 0.557 V.¹⁵ Considering that the reported resting potential for species **1a** is close to 0.3 V vs. Ag/AgCl, reduction and oxidation can be considered to be highly accessible in terms of energy, and all three species **1a–1c** could coexist in some ratio in the same medium. From the perspective of DFT calculations, species **1c** can be considered to be stable: (i) This is because of its highly negative reduction energy. Taking as the reference the well-known oxidised POM structure $[P_2W_{17}MoO_{62}]^{6-}$,¹⁸ the presently computed reduction of **1a** to produce **1c** is 0.25 eV more exothermic. This fact suggests that, in comparison, species **1c** can be obtained at even lower potentials. (ii) The stability of **1c** is also due to the very low energy of the highest SOMO, -5.38 eV. Although this absolute value is not self-explanatory, it can be compared to that of another member of the POV family as an indicator of the relative stability of the system. For the previously reported stable species $[V_{18}O_{42}(I)]^{5-}$,^{19,20} a member of the same

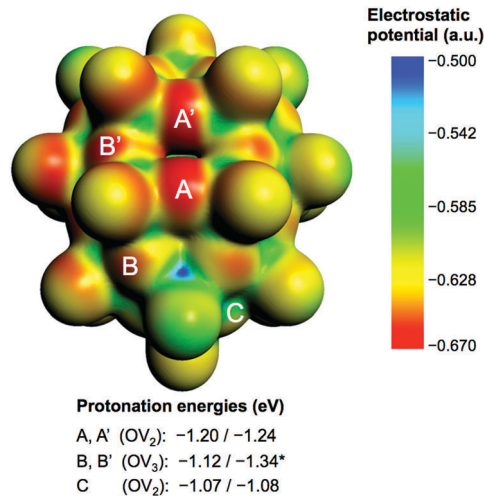


Fig. 2 Nucleophilicity of **1c**. Computed MEP and protonation energies at bridging oxygens (OV_x), labelled A–C by their location in the cage and the number, x , of linked V atoms. Energy ranges arise from the non-equivalence of oxygens of the same type due to the internal VO_2F_2 fragment. B-type positions are the most affected (*) by the relative orientation of the internal ion. The same molecular orientation is adopted in all figures.

family of compounds, we computed the HOMO as high as -4.79 eV (unpublished value). Hence, these computed electronic structures point out considerable electrochemical stability for **1c**.

Fig. 2 shows the molecular electrostatic potential (MEP) of the hydrogen-free species **1c** (see **1a** and **1b** in Fig S3 in the ESI†) and relevant protonation energies at bridging oxygens (red), with computed values in the -1.07 to -1.34 eV range (-24.7 to -30.9 kcal mol⁻¹). The most basic oxygen is B-type (OV_3), as expected.²¹ The energy range associated with the most nucleophilic positions, A and B (8 in total), suggests that attached protons might be itinerant within this set of oxygen atoms. See text and Fig. S4 in the ESI† for more details about the protonation of **1c**.

To account for the magnetic properties of **1** and to exploit fully the results of DFT calculations concerning the distribution of valence electrons, a semi-classical t - J model has been used:¹⁴

$$H = \sum_{\langle i,j \rangle} J_{ij} \delta_i \delta_j S_i \cdot S_j + \sum_{i,j=1}^N t_{ij} \delta_i (1 - \delta_j) - g\mu_B B \sum_{i=1}^N \delta_i S_i^z \quad \delta_i = 0, 1 \quad \sum_{i=1}^N \delta_i = n \quad (1)$$

where δ_i are occupation numbers, n stands for a number of unpaired electrons and N is a number of vanadium sites that can host a valence electron. Thus, this Hamiltonian depends not only on spin states $\{S_i\}$, but also on spin distribution $\{\delta_i\}$. The values of n and N in (1) depend on the considered species and their symmetry (see Table 1 and Table S1 in the ESI†). The first term in Hamiltonian (1) describes superexchange interactions between unpaired electrons/spins ($S = 1/2$). The second term accounts “classically” for electron hopping, and the last term describes the Zeeman energy in magnetic field B with



gyromagnetic ratio $g = 1.965$ and Bohr magneton μ_B . The use of the fully quantum t - J model^{22,23} is impossible because of the large size of Hamiltonian (1).

Superexchange interactions between 3d electrons at vanadium sites in **1** are mediated by three different oxo bridges: single, single-shared and double²⁴ (see Fig. 3, left). Thus, in the simplest model, at least three different exchange couplings (J_i) and also three different electron transfer couplings (t_i) must be introduced. It is possible to differentiate couplings further on the basis of distances, angles, and the neighbourhood, which gives rise to three coupling schemes (see details in the ESI†). In the idealised symmetry of **1** (C_{2v}), due to the inclusion of the guest anion, the number of parameters in all the coupling schemes should be approximately double. For C_1 symmetry the number of parameters should be even larger. Therefore, we initially assume the symmetry of the empty cage (D_{2d}).

For **1a** (C_{2v}) and **1c** (C_{2v} and C_1), the distribution of formal oxidation states for V ions and exchange paths are presented in Fig. 3, right. It can be seen that this is a D_{2d} distribution – that of the empty cage. Unfortunately, all three coupling schemes give very poor fits ($n = 8$, $N = 18$ and $\delta_{17} = \delta_{18} = 1$ in eqn (1)) to the magnetic data for **1a** (C_{2v}), as shown in Fig. S6 and Table S3 in the ESI.† Especially difficult is a proper modelling of the magnetisation. Even using the fully quantum t - J model it is impossible to model properly the magnetisation profile at low temperatures. System **1a** has an even number of electrons, and hence, the ground state, which largely determines low temperature magnetism, possesses an integer total spin moment. This stands in contradiction to the experiment which suggests that magnetisation saturates at the value corresponding to the fractional total spin with $S_{\text{tot}} = 3/2$. Similarly, the same poor results can be expected for **1a** in C_1 symmetry. The coupling scheme does not significantly influence the quality of the fit (see Table S3 in the ESI†). Since the same holds for **1b** and **1c**, in what follows we restrict our considerations only to simpler scheme I.

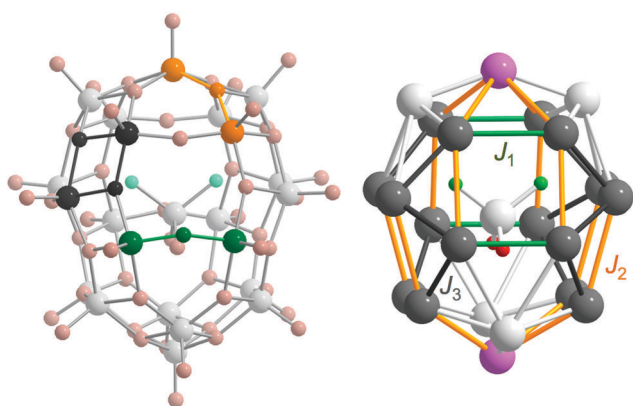


Fig. 3 Magnetic structure of **1**. Left: Example of vanadium ions connected by three types of oxygen bridges: double (black), single (green), single-shared (orange). Remaining atoms: V (grey), O (light red), and F (light green). Right: Formal oxidation states of vanadium ions and exchange paths for **1a** (C_{2v}) and **1c** (C_{2v} and C_1) structures. Colour code (spheres): V^V (purple), V^V (white), V^{IV}/V^V (grey), O (red), and F (green). Coloured sticks represent superexchange paths. Colouring of the exchange paths corresponds to coupling scheme I: J_1 (green), J_2 (orange), J_3 (black). Cage oxygens are omitted.

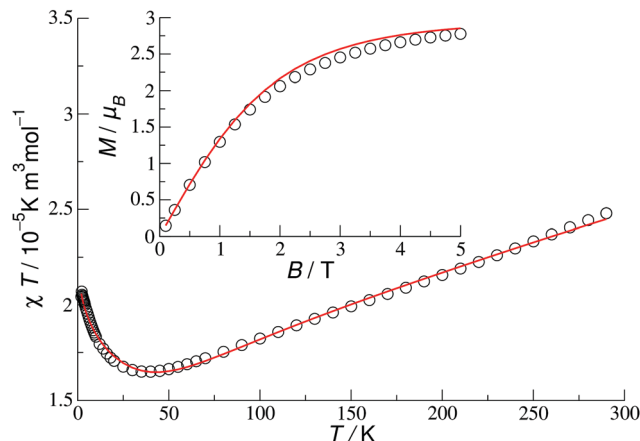


Fig. 4 Molar susceptibility ($B = 0.1$ T) and magnetisation ($T = 2$ K) for **1c** in coupling scheme I with asymmetric couplings J_1 and t_1 (Table 2). Circles indicate experimental data and solid lines indicate the best theoretical fit.

For **1b** (C_{2v}), the fit ($n = 7$, $N = 14$ and $\delta_{13} = \delta_{14} = 1$ in eqn (1)) is still of poor quality (Fig. S7 and Table S3 in the ESI†); however, here the main problem is with a proper modelling of susceptibility at low temperatures. In C_1 symmetry the quality of the fit ($n = 7$, $N = 14$ in eqn (1)) improves, but it is impossible to model properly the minimum of the experimental susceptibility.

Finally, the best fit ($n = 9$, $N = 18$ and $\delta_{17} = \delta_{18} = 1$ in eqn (1)) has been obtained for **1c**. Since the distribution of the formal oxidation states and the couplings follow the D_{2d} symmetry of the cage, it is not possible to reproduce precisely the distribution of the valence electrons in the ground state found using DFT (see Table S1, ESI†). The simplest way to break the symmetry (from D_{2d} to C_{2v}) is to assume that couplings J_1 and t_1 have different values in the F- and O-hemispheres. The best fit for such a setup (Fig. 4) has been obtained for two different parameter sets. However, only for one of them (Table 2) the probabilities to find an electron at a given V^{IV}/V^V site in the $S_{\text{tot}} = 3/2$ ground state resemble remarkably well the DFT atomic spin densities scaled to the number of unpaired electrons (Table 3). Yet, this distribution changes quickly at finite temperatures (see Fig. S8 in the ESI†).

The values in Table 2 should be considered as examples since they depend on the details of the model, *e.g.* on the way the asymmetry of couplings is introduced, and in this sense they are not unique.‡ It is important, however, that for other electronic states we were not able to obtain equally good fits.

In summary, using two complementary theoretical methods, DFT and effective Hamiltonian calculations, the magnetism and valence state of a host-guest POV $\{V^V O_2 F_2 @ V_{22}^{IV/V} O_{54}\}$ have been elucidated. We thus provide an example of the systematic magnetochemical treatment of a large POM with localised and

Table 2 Optimal Hamiltonian parameters for **1c** in K

F-Hemisphere	J_1	581	t_1	1456
O-Hemisphere	J_1	385	t_1	2421
Both hemispheres	J_2	-185	t_2	3205
	J_3	609	t_3	1162



Table 3 Probability to find an electron (in the ground state) at V^{IV}/V^V sites from DFT and t - J Hamiltonian calculations for **1c** in C_{2v} symmetry

Position in the cage	Rescaled ASD	t - J Hamiltonian
Upper ring – F-hemisphere	0.51	0.50
Inner ring	0.50–0.51	0.50
Lower ring – O-hemisphere	0.29	0.25

delocalised electrons. Here, however, contrary to common practice,^{14,25,26} the DFT technique is not used to estimate directly microscopic parameters. Our self-consistent method can indeed be applied to other nanosized, mixed-valent POMs and to mixtures of species with different valence states, which due to the easiness of the redox process cannot be excluded. The results of the application of our method to other members of the $\{X@V_{22}O_{54}\}$ family and to mixtures will be published elsewhere.

Conflicts of interest

There are no conflicts of interest to declare.

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Notes and references

‡ Details of the synthesis and magnetic measurements of the compound containing the title polyoxoanion can be found in our previous work (see ref. 15).

§ In all solutions which agree with DFT results there are large anti-ferromagnetic and smaller ferromagnetic exchange couplings in accord with a previous result.¹⁵ Moreover, such values correspond well to the experimental results: no saturation of χT at high temperature and a minimum at lower temperature (Fig. 4). Interpretation of transfer integrals is more difficult as due to their classical nature they can also comprise other effects (e.g. inter-site repulsion). However, a positive sign suggests a tendency of itinerant electrons to group together to minimize the border between occupied and unoccupied sites. For some coupling schemes also negative values appear, though they are not dominating. This is different from those in similar compounds¹⁴ where rather negative values dominate and requires further investigation.

- U. Kortz, A. Müller, J. van Slageren, J. Schnack, N. S. Dalal and M. Dressel, *Coord. Chem. Rev.*, 2009, **253**, 2315–2327.
- L. Vilà-Nadal, S. G. Mitchell, S. Markov, C. Busche, V. Georgiev, A. Asenov and L. Cronin, *Chem. – Eur. J.*, 2013, **19**, 16502–16511.
- C. Busche, L. Vilà-Nadal, J. Yan, H. N. Miras, D.-L. Long, V. P. Georgiev, A. Asenov, R. H. Pedersen, N. Gadegaard, M. M. Mirza, D. J. Paul, J. M. Pobllet and L. Cronin, *Nature*, 2014, **515**, 545–549.

- A. Balliou, G. Papadimitropoulos, G. Skoulatakis, S. Kennou, D. Davazoglou, S. Gardelis and N. Glezos, *ACS Appl. Mater. Interfaces*, 2016, **8**, 7212–7220.
- K. Y. Monakhov, M. Moors and P. Kögerler, *Advances in Inorganic Chemistry*, in *Polyoxometalate Chemistry*, ed. R. van Eldik and L. Cronin, Academic Press, Elsevier, Amsterdam, 2017, vol. 69, pp. 251–286.
- J. Lehmann, A. Gaita-Ariño, E. Coronado and D. Loss, *Nat. Nanotechnol.*, 2007, **2**, 312–317.
- J. R. Lehmann, A. Gaita-Ariño, E. Coronado and D. Loss, *J. Mater. Chem.*, 2009, **19**, 1672–1677.
- J. M. Clemente-Juan, E. Coronado and A. Gaita-Ariño, *Chem. Soc. Rev.*, 2012, **41**, 7464.
- S. Cardona-Serra, J. M. Clemente-Juan, E. Coronado, A. Gaita-Ariño, N. Suaud, O. Svoboda, R. Bastardis, N. Guihéry and J. J. Palacios, *Chem. – Eur. J.*, 2015, **21**, 763–769.
- A. Pali, B. Tsukerblat, J. M. Clemente-Juan and E. Coronado, *J. Phys. Chem. C*, 2016, **120**, 16994–17005.
- A. Ardavan, O. Rival, J. Morton, S. Blundell, A. Tyryshkin, G. Timco and R. Winpenny, *Phys. Rev. Lett.*, 2007, **98**, 57201.
- D. Gatteschi, L. Pardi, A. L. Barra and A. Müller, *Mol. Eng.*, 1993, **3**, 157–169.
- K. Y. Monakhov, W. Bensch and P. Kögerler, *Chem. Soc. Rev.*, 2015, **44**, 8443–8483.
- C. J. Calzado, J. M. Clemente-Juan, E. Coronado, A. Gaita-Ariño and N. Suaud, *Inorg. Chem.*, 2008, **47**, 5889–5901.
- K. Y. Monakhov, O. Linnenberg, P. Kozłowski, J. van Leusen, C. Besson, T. Secker, A. Ellern, X. López, J. M. Pobllet and P. Kögerler, *Chem. – Eur. J.*, 2015, **21**, 2387–2397.
- W. Plass, *Angew. Chem., Int. Ed. Engl.*, 1996, **35**, 627–631.
- L. Yan, X. López, J. J. Carbó, R. Sniatynsky, D. C. Duncan and J. M. Pobllet, *J. Am. Chem. Soc.*, 2008, **130**, 8223–8233.
- P. A. Aparicio, X. López and J. M. Pobllet, *J. Mol. Eng. Mater.*, 2014, **2**, 1440004.
- A. Müller, R. Sessoli, E. Krickemeyer, H. Bögge, J. Meyer, D. Gatteschi, L. Pardi, J. Westphal, K. Hovemeier, R. Rohlfing, J. Döring, F. Hellweg, C. Beugholt and M. Schmidtman, *Inorg. Chem.*, 1997, **36**, 5239–5250.
- O. Linnenberg, M. Moors, A. Solé-Daura, X. López, C. Bäumer, E. Kentzinger, W. Pyckhout-Hintzen and K. Y. Monakhov, *J. Phys. Chem. C*, 2017, **121**, 10419–10429.
- J. Y. Kempf, M. M. Rohmer, J. M. Pobllet, C. Bo and M. Benard, *J. Am. Chem. Soc.*, 1992, **114**, 1136–1146.
- J. M. Clemente-Juan, J. J. Borrás-Almenar, E. Coronado, A. V. Pali and B. S. Tsukerblat, *Inorg. Chem.*, 2009, **48**, 4557–4568.
- J. J. Borrás-Almenar, S. Cardona-Serra, J. M. Clemente-Juan, E. Coronado, A. V. Pali and B. S. Tsukerblat, *J. Comput. Chem.*, 2010, **31**, 1321–1332.
- O. Linnenberg, P. Kozłowski, C. Besson, J. van Leusen, U. Englert and K. Y. Monakhov, *Cryst. Growth Des.*, 2017, **17**, 2342–2350.
- N. A. G. Bandeira, O. Sadeghi, T. J. Woods, Y.-Z. Zhang, J. Schnack, K. Dunbar, M. Nyman and C. Bo, *J. Phys. Chem. A*, 2017, **121**, 1310–1318.
- T. Gupta and G. Rajaraman, *Chem. Commun.*, 2016, **52**, 8972–9008.

