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# Crystal engineering of a family of hybrid ultramicroporous materials based upon interpenetration and dichromate linkers†

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A new family of 2-fold interpenetrated primitive cubic (pcu) networks of formula  $[M(L)_2(Cr_2O_7)]_n$  ( $M = Co^{2+}$ ,  $Ni^{2+}$ ,  $Cu^{2+}$  and  $Zn^{2+}$ ;  $L = 4,4'$ -azopyridine), DICRO-3-M-i, has been synthesised and their structures, permanent porosity and gas sorption properties were comprehensively characterised. Molecular simulations indicate that  $CO_2$  molecules occupy both of the two distinct ultramicropores that run through this isostructural series. The orientation of the  $Cr_2O_7^{2-}$  pillars is thought to contribute to high isosteric enthalpy of adsorption ( $Q_{st}$ ) towards  $CO_2$  and temperature programmed desorption experiments reveal that DICRO-3-Ni-i selectively adsorbs  $CO_2$  from gas mixtures that simulate flue gas. Performance in this context is among the highest for physisorbents measured to date and these materials are readily regenerated at 50 °C.

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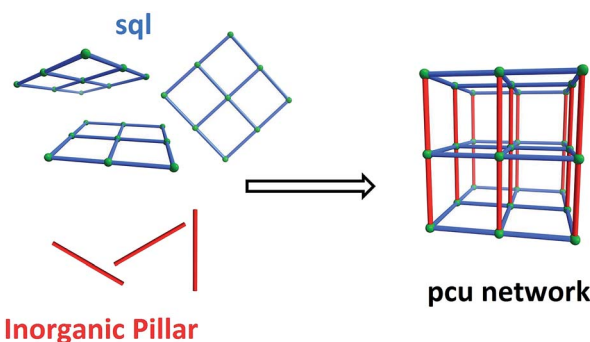
## Introduction

Crystal engineering<sup>1</sup> exploits structural information at the molecular level to design new crystal structures and, in turn, assert control over physicochemical properties. To date, crystal engineering has evolved to the point that it offers rational design strategies for the synthesis of new families of crystalline materials that can afford control over important functional properties such as porosity, chirality and solubility. With respect to porosity, the “node-and-linker” strategy<sup>2</sup> has enabled the rapid development of porous Metal–Organic Materials (MOMs),<sup>3</sup> including coordination networks known as porous coordination polymers (PCPs)<sup>4</sup> and/or metal–organic frameworks (MOFs).<sup>5</sup>

We have recently undertaken systematic examination of an underexplored subclass of porous materials that combines two key features, which together can enhance selectivity towards

certain sorbates and stability towards humidity: incorporation of strong electrostatics thanks to the use of inorganic linker anions that expose electronegative atoms to pore walls; the use of short organic linker ligands or interpenetration to generate ultramicropores (<0.7 nm). The resulting materials, Hybrid Ultramicroporous Materials, HUMs,<sup>6,7</sup> can exhibit strong and selective adsorption sites for important sorbates such as  $CO_2$ . Indeed, HUMs have set a series of benchmarks for selectivity ( $S$ ) to  $CO_2$  over competing less polarisable gases like  $N_2$  ( $S_{CN}$ )<sup>7–9</sup> and vapours such as  $H_2O$  ( $S_{CW}$ ).<sup>6b</sup>

An important feature of HUMs is that they are inherently modular, which makes them amenable to fine-tuning through substitution of one or more of their components. To date, we have developed two related platforms of HUMs using the same simple design strategy, *i.e.* the use of inorganic linker anions to



Scheme 1 Self-assembly of pillared pcu topology networks from sql networks cross-linked by linear inorganic anions that serve as pillars.

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pillar square grid lattice (**sql**) coordination networks (Scheme 1). The resulting networks are based upon the following linker anions: hexafluorometallate anions (*e.g.*  $\text{SiF}_6^{2-}$ , **SIFSIX**) that pillar in linear fashion to generate primitive cubic (**pcu**) networks<sup>6b,7</sup> (this platform also includes analogues which exhibit supermicropores, *i.e.*  $>0.7$  but  $<2$  nm);<sup>8</sup> **mmo** topology networks that incorporate angular pillars which enable self-penetration of the **sql** nets ( $\text{CrO}_4^{2-}$ ,  $\text{MoO}_4^{2-}$ ,  $\text{WO}_4^{2-}$ ).<sup>9</sup> We have also reported the first porous **pcu** network sustained by  $\text{Cr}_2\text{O}_7^{2-}$  (**DICRO**) linear linkers, **DICRO-2-Ni-i**<sup>6a</sup> ( $2 = 1,2$ -bis(4-pyridyl) acetylene), but a hitherto unprecedented angled mode of interpenetration mitigated against strong gas sorption performance towards  $\text{CO}_2$ .<sup>6a</sup> The first two platforms seem to validate the effect upon gas sorption performance when combining ultramicropores and high electrostatics in the same material. Specifically, their modular nature enabled systematic fine-tuning of both pore size and pore chemistry to optimise physorption performance. These observations prompted us to study other families of HUMs.

Two crystal engineering approaches exist for the formation of networks that exhibit ultramicropores: incorporation of short linkers ( $<0.7$  nm); interpenetration of two or more networks with relatively large cavities that afford much reduced pore size following interpenetration. Interpenetration was initially thought of as a hindrance to gas sorption applications.<sup>10</sup> Indeed, the prototypal **pcu** net that is sustained by **DICRO** pillars, **DICRO-1-M-i**,  $1 = 4,4'$ -bipyridine, ( $\text{M}^{2+} = \text{Fe},^{11} \text{Co},^{12} \text{Ni}$  (ref. 13) and  $\text{Cu}$ ;<sup>13</sup> **i** = interpenetrated) eschews porosity because of 2-fold interpenetration. Nevertheless, it is becoming apparent that interpenetration, a well-recognised phenomenon,<sup>4c</sup> can be an effective tool to generate ultramicroporous materials that exhibit high isosteric enthalpy of adsorption ( $Q_{st}$ ) and high selectivity for polarisable gases.<sup>7b</sup>

We report herein a family of 2-fold interpenetrated HUMs based upon **DICRO** pillars:  $[\text{M}(4,4'\text{-azopyridine})_2(\text{Cr}_2\text{O}_7)]_n$ , **DICRO-3-M-i**, ( $3 = 4,4'$ -azopyridine;  $\text{M} = \text{Co}^{2+}, \text{Ni}^{2+}, \text{Cu}^{2+}, \text{Zn}^{2+}$ ). **DICRO-3-M-i** form **pcu** topology nets which exhibit offset interpenetration and their pore walls are lined by inorganic moieties, unlike **DICRO-2-Ni-i**.<sup>6a</sup> The **DICRO-3-M-i** family was characterised by single-crystal and powder X-ray diffraction, thermogravimetric analysis (TGA), FT-IR spectroscopy (FTIR), single component gas sorption measurements ( $\text{CO}_2$  and  $\text{N}_2$ ), molecular simulations of  $\text{CO}_2$  adsorption, and temperature programmed desorption (TPD) experiments coupled with TGA/mass spectrometry (MS). We evaluated the performance of **DICRO-3-Ni-i** in the context of direct capture of  $\text{CO}_2$  from air and five other gas mixtures including simulated flue gas. The stability of **DICRO-3-Ni-i** to atmospheric water vapour was studied *via* standard accelerated stability tests<sup>6b</sup> that simulate long-term storage or “shelf-life”.

## Results and discussion

### Syntheses and characterisation

Single-crystals of **DICRO-3-M-i** ( $\text{M} = \text{Ni}^{2+}, \text{Cu}^{2+}, \text{Zn}^{2+}$ ) were prepared *via* slow diffusion of aqueous  $\text{K}_2\text{Cr}_2\text{O}_7/\text{M}(\text{NO}_3)_2$  into MeCN solutions of ligand **3** at room temperature. Yellow block-

shaped crystals were obtained after one week. Single-crystals of **DICRO-3-Co-i** were synthesised by heating a sealed vial of reagents dissolved in  $\text{H}_2\text{O}$  to  $120^\circ\text{C}$  for 48 h, followed by cooling to  $30^\circ\text{C}$  over 6 h. **DICRO-3-Co-i** formed as small, block-shaped, dark red coloured crystals that were weakly diffracting. As such, the crystallographic data for **DICRO-3-Co-i** is of low quality; however, the space group and crystal packing matches the other three compounds in the series. **DICRO-3-M-i** were also synthesised *via* direct mixing of reagents to afford powder samples suitable for use in gas sorption studies. All four compounds were found to be stable after one week of exposure to ambient conditions. PXRD data are provided in the ESI; Fig. S1–S4.† Thermogravimetric analyses reveal that all four compounds exhibit thermal stability to a least  $200^\circ\text{C}$  (Fig. S5†). Infrared spectra are presented in Fig. S6.†

### Structural analysis

Single-crystal diffraction analysis of **DICRO-3-M-i** revealed that they are isostructural, crystallising in the triclinic space group  $P\bar{1}$ . The hexacoordinate metal centres serve as six-connected (6-c) nodes in **pcu** topology networks.<sup>14</sup> Four pyridyl groups from four different linker ligands comprise the equatorial plane (Fig. 1) which propagates to the expected **sql** network. Average  $\text{M}^{2+}$ -N bond lengths,  $2.15 \text{ \AA}$  (**Co**),  $2.08 \text{ \AA}$  (**Ni**),  $2.02 \text{ \AA}$  (**Cu**) and  $2.16 \text{ \AA}$  (**Zn**), are consistent with the **DICRO-1-M-i** family according to data archived in the Cambridge Structural Database (CSD).<sup>15</sup> The axial positions of the  $\text{M}^{2+}$  centres coordinate to oxygen atoms from  $\text{Cr}_2\text{O}_7^{2-}$  anions, which pillar the **sql** nets to afford the observed **pcu** networks. A survey of the CSD revealed that, excluding polyoxometalates, there are only 23 structures of transition metals coordinated to  $\text{Cr}_2\text{O}_7^{2-}$  anions. Of these, 18 structures involve  $\text{Cr}_2\text{O}_7^{2-}$  ligands that are bridging.  $\text{M}^{2+}$ -O distances of  $2.012(7) \text{ \AA}$  (**Co**),  $2.047(3) \text{ \AA}$  (**Ni**),  $2.274(4) \text{ \AA}$  (**Cu**) and  $2.088(2) \text{ \AA}$  (**Zn**) are within expected ranges. The Cu-O bonds in **DICRO-3-Cu-i** are elongated because of Jahn-Teller distortion. Cr ions exhibit tetrahedral geometry with average O-Cr-O angles of  $109.65(4)^\circ$  (**Co**),  $109.20(17)^\circ$  (**Ni**),  $108.2(2)^\circ$  (**Cu**) and  $109.24(12)^\circ$  (**Zn**). In each compound, the terminal Cr-O bond distances are shorter than the bridging Cr-

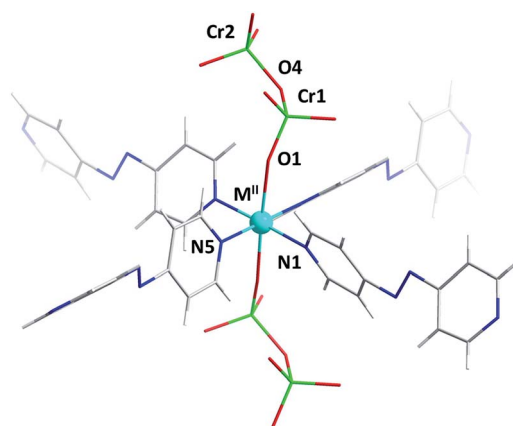


Fig. 1 Coordination environment in the **DICRO-3-M-i** family.



O distances, which is consistent with previously reported structures.<sup>11–13</sup> The Cr(1)–O(4)–Cr(2) angles for **DICRO-3-Ni-i**, **DICRO-3-Cu-i** and **DICRO-3-Zn-i** range between 126.0(2)–127.3(3)° but are more obtuse in **DICRO-3-Co-i** (135.6(5)°). Crystallographic details and selected bond lengths and angles are summarised in Tables 1 and S1–S2.†

The **pcu** networks feature rhombohedral cavities with dimensions of *ca.* 8 × 13 × 13 Å (defined by M···M distances) that enable 2-fold interpenetration in which the node of the second net is “offset” from the cavity of the first net (Fig. 2). This mode of interpenetration is commonly encountered in 2-fold interpenetrated **pcu** networks and is also seen in **DICRO-1-M-i**.<sup>11–13</sup> The offset arrangement could be an artifact of C–H···O interactions between pyridyl groups of one network and the Cr<sub>2</sub>O<sub>7</sub><sup>2-</sup> pillars of the adjacent network (Fig. S7–S8†).

Notably, offset interpenetration results in four crystallographically distinct channels running through the structure parallel to the *a*-axis. However, only two of these channels are large enough to accommodate even small guest molecules. Channels A and B (Fig. 2) exhibit pore diameters of 4.4–5.8 Å and 3.0–3.4 Å, respectively, based upon the closest O···O and C···C distances for channels A and B respectively (minus van der Waals radii). The larger of the two channels is lined by terminal O-atoms from Cr<sub>2</sub>O<sub>7</sub><sup>2-</sup> pillars (O6) that orient towards the inside of the channel. For channel B, the terminal O-atom (O3) does not protrude into the cavity. The location of accessible O-atoms from the dichromate inorganic pillars in these networks could be of significance with respect to gas sorption, as discussed earlier and below. Channel A has features that resemble a member of the **MFSIX** platform, [Cu(L)<sub>2</sub>(SiF<sub>6</sub>)] (L = 1,2-bis(4-pyridyl)acetylene), **SIFSIX-2-Cu-i**,<sup>7b</sup> which displays centred, 2-fold interpenetration and one type of channel. This is because the staggered disposition of the interpenetrated nets results in electronegative atoms (O-atoms in the case of **DICRO**, F-atoms in the case of **SIFSIX**) protruding into the channel walls in such a manner that the electronegative atoms are 7–8 Å apart. These features are discussed in relation to gas sorption and isosteric enthalpy of adsorption in the next section.

### Gas sorption analyses

Single component CO<sub>2</sub> and N<sub>2</sub> sorption measurements were performed on **DICRO-3-Ni-i**, **DICRO-3-Co-i** and **DICRO-3-Cu-i** to determine gas uptake, ideal selectivity and isosteric enthalpy of

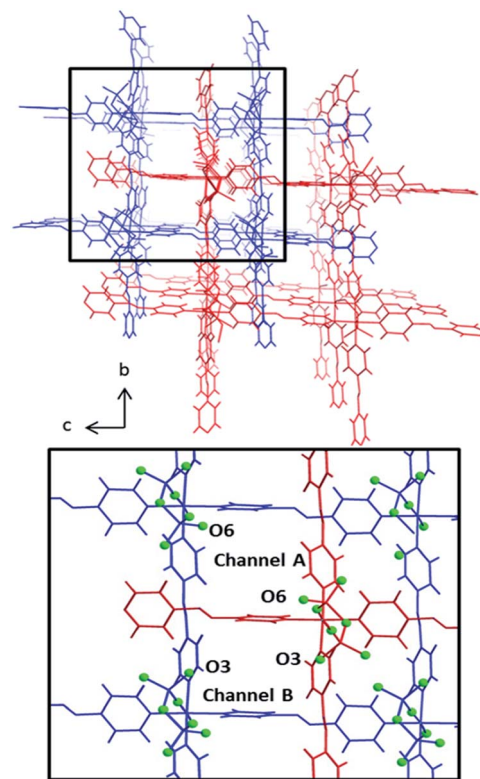


Fig. 2 Top: Packing of **pcu** nets in **DICRO-3-M-i**, frameworks shown in red and blue, viewed down the *a*-axis. Bottom: Inset of top figure. Channel A exhibits a terminal O-atom (O6) protruding into the channel. Channel B has an O-atom (O3) lining the channel wall. O-atoms are coloured light-green and presented in ball-and-stick mode.

adsorption ( $Q_{st}$ ). PXRD patterns reveal that **DICRO-3-Zn-i** loses crystallinity with loss of solvent and it also exhibits low uptake of N<sub>2</sub> at 77 K and CO<sub>2</sub> at 195 K (Fig. S9 and S10†). **DICRO-3-Zn-i** was therefore not studied further in terms of gas sorption. The permanent porosities of **DICRO-3-Co-i**, **DICRO-3-Ni-i** and **DICRO-3-Cu-i** were established by N<sub>2</sub> isotherms at 77 K (Fig. S9†). The apparent BET surface area was calculated to be 142, 388 and 325 m<sup>2</sup> g<sup>-1</sup> for **DICRO-3-Co-i**, **DICRO-3-Ni-i** and **DICRO-3-Cu-i**, respectively. Pore volumes of 0.052, 0.143 and 0.122 cm<sup>3</sup> g<sup>-1</sup> for **DICRO-3-Co-i**, **DICRO-3-Ni-i**, and **DICRO-3-Cu-i** were calculated by assuming liquid filling of N<sub>2</sub> at the  $P/P_0 = 0.95$ . These values are similar to those calculated from the

Table 1 Selected bond lengths and angles for **DICRO-3-M-i**

	<b>DICRO-3-Co-i</b>	<b>DICRO-3-Ni-i</b>	<b>DICRO-3-Cu-i</b>	<b>DICRO-3-Zn-i</b>
M <sup>II</sup> –N1 Å	2.162(7)	2.108(3)	2.036(5)	2.160(3)
M <sup>II</sup> –N5 Å	2.142(9)	2.069(4)	2.020(5)	2.161(2)
M <sup>II</sup> –O1 Å	2.012(7)	2.047(3)	2.274(4)	2.088(2)
Cr1–O4–Cr2°	135.6(5)	125.9(2)	127.3(3)	126.06(16)
M <sup>II</sup> –O1–Cr1	154.9(5)	158.7(2)	167.0(2)	163.25(14)
Torsion angle of 3 (°)	61.7, <sup>a</sup> 57.2 <sup>b</sup>	105.2, <sup>a</sup> 94.8 <sup>b</sup>	62.3, <sup>a</sup> 72.3 <sup>b</sup>	77.3, <sup>a</sup> 64.5 <sup>b</sup>
Rhombohedral cavity dimensions (Å)	8.794, 13.276, 13.314	8.734, 13.155, 13.153	8.993, 13.014, 13.002	8.782, 13.278, 13.270

<sup>a</sup> Torsion angle of C2C3C6C7. <sup>b</sup> Torsion angle of C12C13C16C17.



crystallographic data using *PLATON*<sup>16</sup> for **DICRO-3-Ni-i** and **DICRO-3-Cu-i** (0.158 and 0.151 cm<sup>3</sup> g<sup>-1</sup>, respectively). The lower uptake exhibited by **DICRO-3-Co-i** could be attributed to partial collapse of the network following activation.

CO<sub>2</sub> and N<sub>2</sub> sorption isotherms at 273, 283 and 293 K were collected for **DICRO-3-Co-i**, **DICRO-3-Ni-i** and **DICRO-3-Cu-i**. 293 K isotherms for both gases are presented in Fig. 3 (top), while 273, 283 and 293 K isotherms are given in the ESI (Fig. S11–S13†). A summary of gas sorption data is also provided in Table 2. **DICRO-3-Co-i** and **DICRO-3-Ni-i** both exhibit type I isotherms for CO<sub>2</sub> sorption whereas **DICRO-3-Cu-i** has a distinct step and hysteresis. The step observed in the CO<sub>2</sub> adsorption isotherms for **DICRO-3-Cu-i** varies with temperature, occurring at higher partial pressures and becoming more gradual at higher temperature. The stepped nature of the isotherm observed for **DICRO-3-Cu-i** may be due to either of two structural changes: translation of independent networks with respect to one another upon increased CO<sub>2</sub> sorption, a phenomenon that has been reported previously for similar systems;<sup>17</sup> flexible networks switching between a narrow pore (NP) and a large pore (LP) form, which has also been observed in **pcu** networks.<sup>17b,18</sup> Since, stepped isotherms were not observed for the **Co**, **Ni** and **Zn** analogues it seems unlikely that translation of independent networks is the origin for the stepped behaviour seen in **DICRO-**

**3-Cu-i**. Cu<sup>2+</sup> exhibits Jahn–Teller distortion, which is known to be temperature dependent (the distortion is less pronounced at higher temperatures). The stepped nature of the isotherm could therefore be a consequence of the effect of Jahn–Teller distortion upon the coordination sphere, which would in turn affect the pore size and shape. To verify this hypothesis, PXRD experiments were conducted on ‘as-synthesised’ powder at ambient pressure, at reduced pressure and again after the powder was cycled back to ambient pressure. A phase change is apparent from the PXRD patterns (Fig. S14†) but attempts to collect diffraction data on single-crystals of **DICRO-3-Cu-i** at reduced pressure were unsuccessful.

Selectivity for CO<sub>2</sub> over N<sub>2</sub> (*S*<sub>CN</sub>) for **DICRO-3-Co-i**, **DICRO-3-Ni-i** and **DICRO-3-Cu-i** was calculated from the uptakes of CO<sub>2</sub> at 0.15 bar and N<sub>2</sub> at 0.85 bar at 293 K to be 70.2, 79.3 and 145.9, respectively. The relatively high *S*<sub>CN</sub> for **DICRO-3-Cu-i** is partly a result of its very low N<sub>2</sub> uptake at 293 K. Such *S*<sub>CN</sub> values are comparable to porous MOMs containing amino-functional groups and open-metal sites, which are absent herein.<sup>19</sup>

CO<sub>2</sub> isotherms measured at 273, 283 and 293 K were fitted using the virial equation (Fig. S15, S17, S19†), and the isosteric enthalpy of adsorption (*Q*<sub>st</sub>) was calculated using the Clausius–Clapeyron equation.<sup>20</sup> *Q*<sub>st</sub> values at low loading for **DICRO-3-Co-i**, **DICRO-3-Ni-i** and **DICRO-3-Cu-i** are 35, 37 and 37 kJ mol<sup>-1</sup>, respectively (Fig. S16, S18, S20†). The relatively flat shape of the *Q*<sub>st</sub> curves indicates that the binding energy is relatively independent of loading. As mentioned earlier, **SIFSIX-2-Cu-i** exhibits similar pore size, shape and chemistry to channel A of the **DICRO-3-M-i** family. *Q*<sub>st</sub> values at mid-loading (**SIFSIX-2-Cu-i** = 35 kJ mol<sup>-1</sup>; **DICRO-3-Cu-i** = 35 kJ mol<sup>-1</sup>) and curve shapes are indeed comparable. *Q*<sub>st</sub> values for **DICRO-3-M-i** are also similar or higher than seen for **MOF-5** (34 kJ mol<sup>-1</sup>),<sup>21</sup> **HKUST-1** (35 kJ mol<sup>-1</sup>),<sup>22</sup> **MAF-25** (26 kJ mol<sup>-1</sup>),<sup>23</sup> **InOF-1** (29 kJ mol<sup>-1</sup>)<sup>24</sup> and **NOTT-140** (25 kJ mol<sup>-1</sup>).<sup>25</sup> However **HUMs** with **mmo** topology exhibit narrower pores and much higher *Q*<sub>st</sub> values at low loading: **MOFOUR-1-Ni** (56 kJ mol<sup>-1</sup>),<sup>9b</sup> **CROFOUR-1-Ni** (50 kJ mol<sup>-1</sup>)<sup>9b</sup> and **WOFOUR-1-Ni** (65 kJ mol<sup>-1</sup>).<sup>9a</sup> Comparison can also be made with **DICRO-2-Ni-i**, for which the pore diameter (3.6 Å) is intermediate to those of channels A and B of **DICRO-3-M-i**. The Cr<sub>2</sub>O<sub>7</sub><sup>2-</sup> pillar is directed away from the accessible channel in **DICRO-2-Ni-i** but towards the interior of the channel in **DICRO-3-M-i**. The larger *Q*<sub>st</sub> values of **DICRO-3-M-i** vs. **DICRO-2-Ni-i** (30.5 kJ mol<sup>-1</sup>)<sup>6a</sup> support the importance of pore size and pore chemistry seen for the **MFSIX** platform.<sup>7b</sup> While interpenetration necessarily reduces pore dimensions, likely enhancing binding interactions between sorbate and sorbent, interpenetration may also afford non-uniform pores and significantly change pore structure. Whereas uniform pore binding is likely to be generally preferred for enhanced selectivity, different sorbates could require very different pore structures to optimise the energetics of binding. Therefore, generalisation about the influence of interpenetration upon separations is inappropriate since interpenetrated networks may in some circumstances be more able to create uniform pores than their non-interpenetrated analogues.

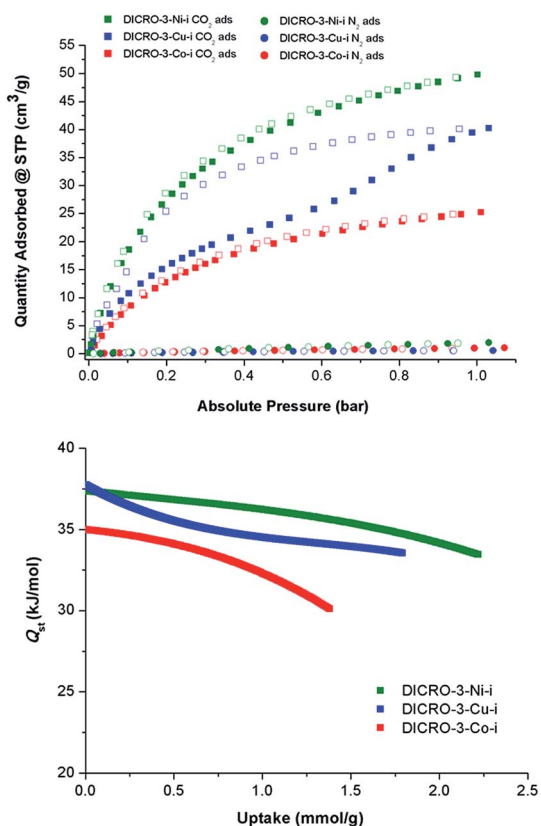


Fig. 3 Top: Single component gas sorption isotherms for **DICRO-3-M-i** at 293 K (Co = red, Ni = green, Cu = blue). Solid symbols signify adsorption while empty symbols represent desorption measurements. Bottom: Isosteric enthalpy of adsorption (*Q*<sub>st</sub>) of **DICRO-3-M-i** compounds towards CO<sub>2</sub>.



Table 2 Summary of gas sorption data for the DICRO-3-M-i family

	Calculated void space <sup>a</sup>	Calculated micropore volume (cm <sup>3</sup> g <sup>-1</sup> )	CO <sub>2</sub> /N <sub>2</sub> selectivity at <sup>b</sup> 293 K	CO <sub>2</sub>			N <sub>2</sub>		
				Micropore volume <sup>c</sup> (195 K)	Q <sub>st</sub> at low loading (kJ mol <sup>-1</sup> )	Uptake (cm <sup>3</sup> g <sup>-1</sup> ) at 293 K 0.1/1 bar	Micropore volume <sup>d</sup> (77 K)	BET exp. (m <sup>2</sup> g <sup>-1</sup> ) at 77 K	Uptake (cm <sup>3</sup> g <sup>-1</sup> ) at 293 K, 1 bar
<b>DICRO-3-Co-i</b>	28.2%	0.201	70.2	0.054	35	8.58/25.20	0.052	142	1.01
<b>DICRO-3-Ni-i</b>	22.6%	0.158	79.3	0.109	37	18.59/49.76	0.143	388	1.95
<b>DICRO-3-Cu-i</b>	21.9%	0.151	145.9	0.072	37	10.75/40.26	0.122	325	0.51

<sup>a</sup> The accessible void space and micropore volume were calculated with the program *PLATON* using a probe radius of 1.2 Å. <sup>b</sup> Preliminary selectivity values were calculated as a ratio of gas uptakes at a specified pressure and 293 K from single component N<sub>2</sub> and CO<sub>2</sub> data. Langmuir curves were fitted to CO<sub>2</sub> isotherms and used to calculate uptake at pressure of 0.85 bar, linear curves were fitted to N<sub>2</sub> data and uptake calculated at 0.15 bar. <sup>c</sup> (DR method) [cm<sup>3</sup> g<sup>-1</sup>]. <sup>d</sup> (*t*-Plot) exp. [cm<sup>3</sup> g<sup>-1</sup>].

### Molecular simulations

Molecular simulations of CO<sub>2</sub> adsorption for **DICRO-3-Ni-i** and **DICRO-3-Cu-i** indicate that guest CO<sub>2</sub> molecules are adsorbed into both channels A and B and that, at CO<sub>2</sub> saturation, CO<sub>2</sub> molecules interact with exposed oxygen atoms of the Cr<sub>2</sub>O<sub>7</sub><sup>2-</sup> pillar (Fig. 4). The tight fit within the channels and the presence of exposed oxygen atoms from Cr<sub>2</sub>O<sub>7</sub><sup>2-</sup> pillars, contributes to the relatively high zero loading CO<sub>2</sub> Q<sub>st</sub> for these HUMs. Simulations of CO<sub>2</sub> adsorption at 273, 283 and 293 K were performed for **DICRO-3-Ni-i** and **DICRO-3-Cu-i**. For **DICRO-3-Ni-i** isotherms are in good agreement with the experimental data (Fig. S21†). However, for **DICRO-3-Cu-i** simulated CO<sub>2</sub> isotherms correspond well with experimental data only in the low and high-pressure regions of the isotherms for the three temperatures, *i.e.*, before and after the inflection point in the experimental isotherms (Fig. S22†). The simulated Q<sub>st</sub> values are also in good agreement with experiment at 32.8 (*vs.* 37) and 33.5 kJ mol<sup>-1</sup> (*vs.* 37) for **DICRO-3-Ni-i** and **DICRO-3-Cu-i**, respectively (Fig. S23, S24†). The simulations for **DICRO-3-Cu-i** do not show the inflection point from the experimental CO<sub>2</sub> isotherms because the HUM is treated as a rigid network in the

simulations. The simulations indicate that the X-ray crystal structure for **DICRO-3-Cu-i** is representative of the structure occurring after the step (at higher pressure) in the experimental isotherms.

### Accelerated stability and temperature programmed desorption

Because of its relatively high CO<sub>2</sub> uptake and Q<sub>st</sub>, **DICRO-3-Ni-i** was selected for testing regarding its stability and ability to uptake CO<sub>2</sub> directly from the atmosphere, as well as under simulated flue gas conditions. The stability of **DICRO-3-Ni-i** to atmospheric water vapour was evaluated *via* accelerated stability testing,<sup>6b</sup> which simulates long-term storage and shelf-life (see ESI for full details and Fig S27–S29†). The surface area of **DICRO-3-Ni-i** was found to decrease upon increased exposure. An as-synthesised sample exhibited a BET surface area of 388 m<sup>2</sup> g<sup>-1</sup>, while surface areas for 1 day, 7 days and 16 days were 289 m<sup>2</sup> g<sup>-1</sup> (–25%), 255 m<sup>2</sup> g<sup>-1</sup> (–34%) and 173 m<sup>2</sup> g<sup>-1</sup> (–55%), respectively. The stability of **DICRO-3-Ni-i** is comparable to **HKUST-1** (–28% surface area following 1 day exposure; –33% after 7 days) but superior to **Mg-MOF-74** (–74% surface area following 1 day exposure).<sup>6b</sup> The performance of **DICRO-3-Ni-i** for capturing CO<sub>2</sub> directly from air and under conditions resembling flue gas streams was also evaluated. TGA/MS was used to determine quantitative uptakes of specific adsorbents from these gas streams (full details given in ESI). DAC measurements on **DICRO-3-Ni-i** revealed a CO<sub>2</sub> uptake of ~0.97 L CO<sub>2</sub> per kg (~2 mg CO<sub>2</sub> per g solid) when exposed to the laboratory atmosphere for 12 h (see Fig. S30† for schematic diagram of DAC/TPD apparatus; all TPD and uptake plots are located in Fig. S32–S40 and data tabulated in Table S9†). This equates to a S<sub>CW</sub> CO<sub>2</sub> : H<sub>2</sub>O ratio of >2% : <97% in terms of mass uptake for each component, which is above that measured for **HKUST-1** (1% : 99%) but lower than that of **SIFSIX-3-Ni** (8% : 92%) and **Mg-MOF-74** (4% : 96%).<sup>6b</sup> Under simulated flue gas conditions (moist CO<sub>2</sub>, 0.15 atm), **DICRO-3-Ni-i** exhibits high uptake of CO<sub>2</sub>, ~9.77 L CO<sub>2</sub> per kg (19.2 mg CO<sub>2</sub> per g solid) (Fig. 5). This equates to a S<sub>CW</sub> CO<sub>2</sub> : H<sub>2</sub>O ratio of 53% : 47% which is higher than that of **HKUST-1** (8.5% : >91%); slightly higher than to that of **Mg-MOF-74** (51% : 49%) and second only to **SIFSIX-3-Ni** (62% : 38%) when tested under the

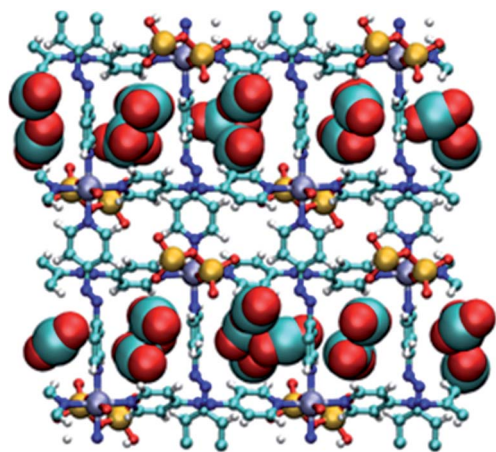


Fig. 4 Simulated packing of CO<sub>2</sub> guest molecules in **DICRO-3-M-i**. View down *a*-axis. Light blue (carbon), red (oxygen), blue (nitrogen), yellow (chromium), purple (M<sup>2+</sup>), white (hydrogen).



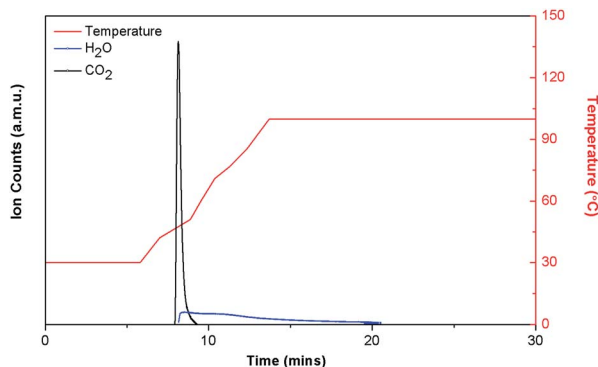


Fig. 5 TPD plot for DICRO-3-Ni-i exposed to moist 0.15 atm CO<sub>2</sub>. Uptake equal to 19.2 mg CO<sub>2</sub> per g, ~52.7% CO<sub>2</sub>, ~47.3% H<sub>2</sub>O.

same conditions.<sup>6b</sup> TPD experiments revealed that desorption occurs below 50 °C for DICRO-3-Ni-i, which is relatively easier than both HKUST-1 and Mg-MOF-74.

## Conclusions

A family of 3D Hybrid Ultramicroporous Materials (HUMs), DICRO-3-M-i, with four members has been synthesised using a crystal engineering approach that exploits Cr<sub>2</sub>O<sub>7</sub><sup>2-</sup> anions to pillar square lattice networks (sql) constructed from 4,4'-azopyridine organic linkers (3) and M<sup>2+</sup> metal centres. Single component gas sorption measurements for three of these compounds, DICRO-3-Co-i, DICRO-3-Ni-i and DICRO-3-Cu-i, were collected and reveal permanent microporosity, selective CO<sub>2</sub>/N<sub>2</sub> uptake and relatively high Q<sub>st</sub> values towards CO<sub>2</sub>. Temperature programmed desorption experiments conducted on DICRO-3-Ni-i revealed excellent CO<sub>2</sub> uptake and S<sub>CN</sub> when exposed to flue gas conditions. Structural analysis, gas sorption experiments and molecular simulations from this study reaffirm that HUMs can afford favourable sorbate-sorbent interactions. Specifically, for the DICRO-3-M-i family, Cr<sub>2</sub>O<sub>7</sub><sup>2-</sup> anions orient in such a manner that terminal O-atoms line the walls of the accessible 1D ultramicroporous channels, resulting in high Q<sub>st</sub> and enhanced CO<sub>2</sub> sorption performance.

To conclude, whereas we have not yet elucidated the origin of the flexibility in this family of materials, there are four take-home messages that could aid the custom-design of porous materials with enhanced gas sorption performance: the combination of ultramicropores and strong electrostatics in HUMs necessarily leads to tight binding sites for polarisable guests such as CO<sub>2</sub>; interpenetration can be used to facilitate ultramicropores, but may also generate non-uniform pores; the presence of groups with strong electrostatics alone is not sufficient, but rather these groups must be favourably positioned to achieve optimal guest binding; binding energies of 35–40 kJ mol<sup>-1</sup> enable facile regeneration of these sorbents.

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## Notes and references

- (a) G. R. Desiraju, *The Design of Organic Solids*, Elsevier, Amsterdam, 1989; (b) B. Moulton and M. J. Zaworotko, *Chem. Rev.*, 2001, **101**, 1629.
- (a) A. F. Wells, *Three Dimensional Nets and Polyhedra*, Wiley, New York, 1977; (b) B. F. Hoskins and R. Robson, *J. Am. Chem. Soc.*, 1990, **112**, 1546; (c) B. F. Hoskins and R. Robson, *J. Am. Chem. Soc.*, 1989, **111**, 5962.
- J. J. Perry IV, J. A. Perman and M. J. Zaworotko, *Chem. Soc. Rev.*, 2009, **38**, 1400.
- (a) S. R. Batten, S. M. Neville and D. R. Turner, *Coordination Polymers Design, Analysis and Application Introduction*, RSC Publishing, Cambridge, UK, 2009; (b) S. Kitagawa, R. Kitaura and S. Noro, *Angew. Chem., Int. Ed.*, 2004, **43**, 2334; (c) S. R. Batten and R. Robson, *Angew. Chem., Int. Ed.*, 1998, **37**, 1460.
- (a) *Metal-Organic Frameworks: Applications from Catalysis to Gas Storage*, ed. D. Farrusseng, Wiley-VCH Verlag GmbH & Co, Weinheim, 2011; (b) *Metal-Organic Frameworks: Design and Application*, ed. L. R. MacGillivray, John Wiley & Sons, Hoboken, USA, 2010; (c) *Functional Metal-Organic Frameworks: Gas storage, separation and catalysis*, ed. M. Schröder, Springer-Verlag, Berlin, 2009.
- (a) H. S. Scott, A. Bajpai, K. J. Chen, T. Pham, B. Space, J. J. Perry IV and M. J. Zaworotko, *Chem. Commun.*, 2015, **51**, 14832; (b) A. Kumar, D. G. Madden, M. Lusi, K. J. Chen, E. A. Daniels, T. Curtin, J. J. Perry IV and M. J. Zaworotko, *Angew. Chem., Int. Ed.*, 2015, **54**, 14372.
- (a) S. K. Elsaidi, M. H. Mohamed, H. T. Schaefer, A. Kumar, M. Lusi, T. Pham, K. A. Forrest, B. Space, W. Xu, G. J. Halder, J. Liu, M. J. Zaworotko and P. K. Thallapally, *Chem. Commun.*, 2015, **51**, 15530; (b) P. Nugent, Y. Belmabkhout, S. D. Burd, A. J. Cairns, R. Luebke, K. Forrest, T. Pham, S. Q. Ma, B. Space, L. Wojtas, M. Eddaoudi and M. J. Zaworotko, *Nature*, 2013, **495**, 80; (c) O. Shekhah, Y. Belmabkhout, K. Adil, P. M. Bhatt, A. J. Cairns and M. Eddaoudi, *Chem. Commun.*, 2015, **51**, 13595; (d) O. Shekhah, Y. Belmabkhout, Z. J. Chen, V. Guillerm, A. Cairns, K. Adil and M. Eddaoudi, *Nat. Commun.*, 2014, **5**, 1.
- (a) S. D. Burd, S. Q. Ma, J. A. Perman, B. J. Sikora, R. Q. Snurr, P. K. Thallapally, J. Tian, L. Wojtas and M. J. Zaworotko, *J. Am. Chem. Soc.*, 2012, **134**, 3663; (b) S. Noro, S. Kitagawa, M. Kondo and K. Seki, *Angew. Chem., Int. Ed.*, 2000, **39**, 2082; (c) P. Nugent, V. Rhodus, T. Pham, B. Tudor, K. Forrest, L. Wojtas, B. Space and M. J. Zaworotko, *Chem. Commun.*, 2013, **49**, 1606.



- 9 (a) M. H. Mohamed, S. K. Elsaidi, T. Pham, K. A. Forrest, B. Tudor, L. Wojtas, B. Space and M. J. Zaworotko, *Chem. Commun.*, 2013, **49**, 9809; (b) M. H. Mohamed, S. K. Elsaidi, L. Wojtas, T. Pham, K. A. Forrest, B. Tudor, B. Space and M. J. Zaworotko, *J. Am. Chem. Soc.*, 2012, **134**, 19556.
- 10 N. L. Rosi, M. Eddaoudi, J. Kim, M. O'Keeffe and O. M. Yaghi, *Angew. Chem., Int. Ed.*, 2002, **41**, 284.
- 11 X. Y. Chen, B. Zhao, P. Cheng, B. Ding, D. Z. Liao, S. P. Yan and Z. H. Zhiang, *Eur. J. Inorg. Chem.*, 2004, 562.
- 12 Y. Hayashi, T. Tagami, H. Mano and A. Uehara, *Chem. Lett.*, 2001, **30**, 562.
- 13 A. L. Kopf, P. A. Maggard, C. L. Stern and K. R. Poeppelmeier, *Acta Crystallogr., Sect. C: Cryst. Struct. Commun.*, 2005, **61**, M165.
- 14 Topology determined using the computer program TOPOS: V. A. Blatov, A. P. Shevchenko and D. M. Proserpio, *Cryst. Growth Des.*, 2014, **14**, 3576.
- 15 F. H. Allen, *Acta Crystallogr., Sect. B: Struct. Sci.*, 2002, **58**, 380.
- 16 A. L. Spek, *J. Appl. Crystallogr.*, 2003, **36**, 7.
- 17 (a) K. L. Mulfort, O. K. Farha, C. D. Malliakas, M. G. Kanatzidis and J. T. Hupp, *Chem.–Eur. J.*, 2010, **16**, 276; (b) S. Kitagawa and K. Uemura, *Chem. Soc. Rev.*, 2005, **34**, 109.
- 18 T. Devic, P. Horcajada, C. Serre, F. Salles, G. Maurin, B. Moulin, D. Heurtaux, G. Clet, A. Vimont, J. M. Greneche, B. Le Ouay, F. Moreau, E. Magnier, Y. Filinchuk, J. Marrot, J. C. Lavalley, M. Daturi and G. Ferey, *J. Am. Chem. Soc.*, 2010, **132**, 1127.
- 19 (a) B. Arstad, H. Fjellvag, K. O. Kongshaug, O. Swang and R. Blom, *Adsorption*, 2008, **14**, 755; (b) K. J. Chen, R. B. Lin, P. Q. Liao, C. T. He, J. B. Lin, W. Xue, Y. B. Zhang, J. P. Zhang and X. M. Chen, *Cryst. Growth Des.*, 2013, **13**, 2118; (c) R. Vaidhyanathan, S. S. Iremonger, K. W. Dawson and G. K. H. Shimizu, *Chem. Commun.*, 2009, **45**, 5230.
- 20 F. Rouquerol, J. Rouquerol, K. S. W. Sing, P. Llewellyn and G. Maurin, *Adsorption by Powders and Porous Solids: Principles, Methodology and Applications*, Academic Press, Amsterdam, 2nd edn, 2014.
- 21 Z. X. Zhao, Z. Li and Y. S. Lin, *Ind. Eng. Chem. Res.*, 2009, **48**, 10015.
- 22 Q. Y. Yang, C. Y. Xue, C. L. Zhong and J. F. Chen, *AIChE J.*, 2007, **53**, 2832.
- 23 J. B. Lin, J. P. Zhang and X. M. Chen, *J. Am. Chem. Soc.*, 2010, **132**, 6654.
- 24 J. J. Qian, F. L. Jiang, D. Q. Yuan, M. Y. Wu, S. Q. Zhang, L. J. Zhang and M. C. Hong, *Chem. Commun.*, 2012, **48**, 9696.
- 25 C. R. Tan, S. H. Yang, N. R. Champness, X. A. Lin, A. J. Blake, W. Lewis and M. Schröder, *Chem. Commun.*, 2011, **47**, 4487.

