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The dual capture of As^V and As^{III} by UiO-66 and analogues†

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UiO-66 and analogues were successfully tailored to chemoselectively capture As^V oxyanions at the hydroxylated node and neutral As^{III} species with the thiolated organic linkers. More efficient and faster uptake can be achieved with increasing defect densities, increasing pore aperture sizes, and decreasing particle sizes.

Introduction

Metal organic frameworks (MOFs) are highly porous and crystalline coordination polymers that can be synthesized from a wide variety of metal-cluster nodes and a diverse selection of multitopic organic linkers. As a result, they are highly tailorable and different combinations of functionalities can be incorporated into the nodes and the linkers in an orthogonal manner.[1–5] Not surprisingly, these materials have recently garnered increasing interest in capture-and-release studies, where the metal cluster nodes were tailored to capture/release phosphonate-based substrates[6,7] or the organic linkers were used to sequester toxic heavy metals.[8–10] These precedents led us to propose that MOFs can be used as a model platform to demonstrate the capture of toxic anionic arsenates (As^V) and neutral arsenites (As^{III}), both of which exist in ground water (pH 6–8.5),[11] in a complementary fashion (Fig. 1); the node can be used for binding anionic As^V and the linkers can be functionalized to capture neutral As^{III}. Such a design can serve as a versatile strategy for developing materials that efficiently capture multiple pollutants or toxic agents that exist as diverse species in certain environments.

Given their high chemical stability and hydrophilicity,[12–17] we deemed MOFs with hexazirconium oxo hydroxo ($Zr_6O_4(OH)_4$) cluster nodes, such as UiO-66,[16–22] to be suitable model targets for modifications to capture both As^V and As^{III} from aqueous media. We predicted strong interactions between the nodes of UiO-66 and $[As^VO_4H_{3-n}]^{n-}$ oxyanions (Fig. 2 and ESI,† Fig. S1), based on the observed strong coordination of the $Zr_6O_4(OH)_4$ cluster nodes[6,7,24–27] to phosphonates and phosphates, which are isostructural and have similar Brønsted basicity as arsenates.[6,7,23–27] In addition, the incorporation of thiol-containing BDC ligands (*i.e.*, 2,5-dimercaptoterephthalic acid) into UiO-66 (ref. 9) should facilitate binding to neutral $[As^{III}(OH)_3]_n$ species,[11] akin to the known arsenophilicity of sulfur-containing enzymes and thiol-rich chelators.[28–31] Herein, we report the successful use of HCl-UiO-66-$(SH)_2$, a UiO derivative with thiolated linkers and nodes that are "capped" with weakly binding ligands, to efficiently capture both As^{III} and As^V from aqueous media (Fig. 1). The missing-linker sites on the $Zr_6(O)_4(OH)_4$ nodes can serve as excellent binders for As^V oxyanions while the thiolated linkers can selectively coordinate As^{III} for dual-capture purposes. The efficiency and capacity of this dual-binding feature is best

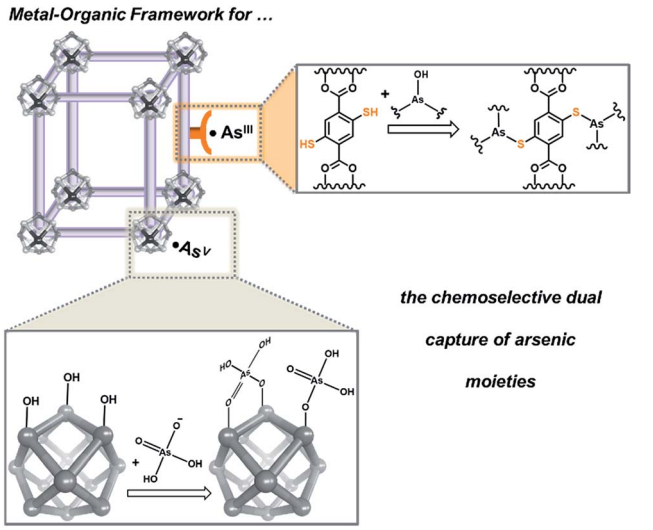

Fig. 1 A schematic representation that suggests how MOFs can be tailored to coordinate anionic As^V moieties at the node while binding neutral As^{III} with the linkers.

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† Electronic supplementary information (ESI) available: Includes synthesis and preparation of UiO-66 and analogues along with characterization data (PXRD, TGA analysis, N_2 isotherms, DRIFTS, XPS analysis), Lagergren kinetic data fittings, leaching studies, recycling studies, and other relevant observations. See DOI: 10.1039/c6sc00490c

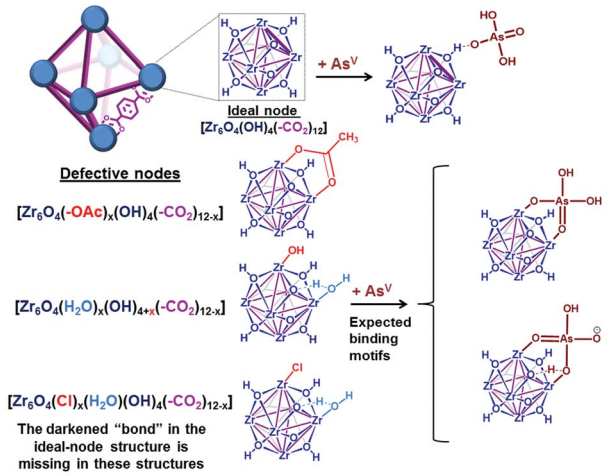

Fig. 2 Representative views (see ESI,† Fig. S19 for other possible binding motifs) of a unit cell of UiO-66, with either 12-coordinated ideal nodes $[Zr_6O_4(OH)_4(-CO_2)_{12}]$ or imperfect nodes $[Zr_6O_4(OH)_x(-CO_2)_y]$, which result from missing linkers. Each purple "bond" indicates a coordinating carboxylate from the terephthalate linker.

realized when the binding sites are made easily accessible, either by enlarging the pore aperture size of the MOF or by reducing the particle size of the MOF nanocrystals.

Results and discussion

As a model platform, UiO-66 is highly attractive given its excellent synthetic tunability: functionalized derivatives of *p*-benzene dicarboxylate (BDC) can be easily incorporated into the framework either through *de novo* synthesis[17,32] or post-synthetically.[33,34] In addition, the degree of coordinative unsaturation of the nodes can be tuned with the use of organic[18,19,35] or inorganic[17] acid-modulators. In the current study, we select nearly defect-free AcOH-UiO-66$|_{11/12}$ (*i.e.*, AcOH-"capped" UiO-66; see Fig. 2, top structure in the lower left corner) as a control sample. This material was prepared from $ZrCl_4$ and H_2BDC in dimethylformamide (DMF) and in the presence of acetic acid as a modulator. It comprises well-defined octahedral particles with a Brunauer–Emmett–Teller (BET) surface area of ~1150 $m^2\ g^{-1}$, and a powder X-ray diffraction (PXRD) pattern identical to that of crystalline UiO-66 (ESI,† Fig. S2–S4). Thermogravimetric analysis (TGA) data for this material suggested a formula unit of $Zr_6O_4(OH)_4(BDC)_{5.5}$, alternatively referred to as $Zr_6O_4(OH)_4(-CO_2)_{11/12}$, which suggests that only a small amount of defects (*i.e.*, missing-linker sites) are present (ESI,† Fig. S7). (The "$(-CO_2)_{11/12}$" notation is used to indicate 1 missing carboxylate per node or 0.5 missing BDC linker per formula unit). As mentioned above, these missing-linker sites are known to bind well to phosphonates[6] and vanadates,[36] and should also be susceptible toward arsenate binding (Fig. 2).

To elucidate the ability of the $Zr_6O_4(OH)_4$ node to bind As^V, we additionally synthesized HCl-UiO-66$|_{(12-x)/12}$, a series of UiO-66 materials where the amount of missing linker (x) was systematically varied using the HCl-modulator strategy.[17] These materials were also prepared from $ZrCl_4$ and H_2BDC in DMF, but with different molar ratios and with HCl as the modulator (see details in ESI,† Section S2 and Table S1). Notably, we successfully obtained HCl-UiO-66$|_{9/12}$, with ~3 available missing-linker sites per node, presumably being "weakly capped" by either HO^-, Cl^-, or H_2O (see Fig. 2, last two structures in the lower left corner).

As^V uptake by UiO-66 derivatives

As^V-adsorption experiments were conducted at pH ~7,[37] to simulate the middle range of ground water pH, by exposing samples of the MOFs (10 mg each) to separate 50 ppm solutions (30 mL portions[38]) of $Na_2HAsO_4\cdot 7H_2O$ as the As^V source. The amount of As^V in the supernatant is monitored using inductively coupled plasma optical emission spectroscopy (ICP-OES) and the per-node uptake of As^V by the MOF at time t can then be calculated. As expected, AcOH-UiO-66$|_{11/12}$ showed good uptake of As^V (Fig. 3, green profile) from the $Na_2HAsO_4\cdot 7H_2O$ test solution. The near-stoichiometric $As^V : Zr_6$ uptake ratio (1.1 : 1 $As^V : Zr_6$) at 6 h strongly suggests a preferential binding of As^V to the missing-linker sites on the Zr_6 nodes. Assuming that missing-linker sites are the most easily accessible, half of these sites (0.5 : 1 $As^V : Zr_6$), presumably those that are close to the surface of the MOF nanocrystals, would be saturated within the first 30 min (Fig. 3, green profile). The rate of adsorption slows down as the remaining, internal sites (*i.e.*, deeper inside the MOF nanocrystals) are saturated over the next few hours, consistent with a diffusion-limited behavior. That the 24 h uptake ratio (1.2 : 1 $As^V : Zr_6$) slightly exceeds the estimated available binding site suggests the possible involvement of secondary binding pathways, such as formation of As oligomers[11] on the node (see ESI,† Fig. S21a for an illustration), linker displacement by the incoming As^V species (see below and ESI,† Section S3 for additional discussions),[39] and/or anion exchange with the bridging hydroxyl sites of the nodes,[40,41] although delineating these pathways is beyond the scope of this manuscript.

Consistent with our hypothesis that missing-linker sites can also bind arsenates well, HCl-UiO-66$|_{9/12}$, which has 3 missing-linker sites per node, captured As^V substantially

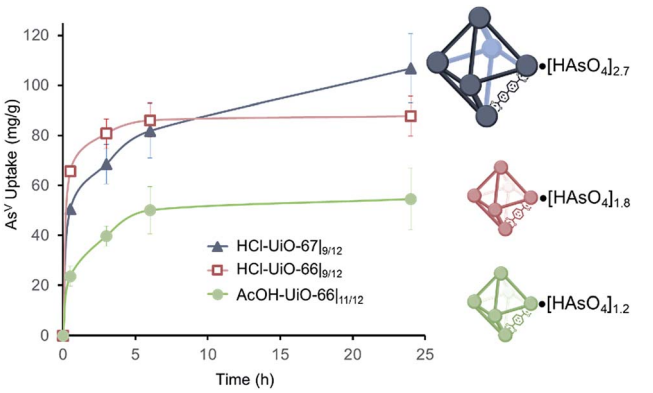

Fig. 3 As^V-uptake profiles for MOF samples (10 mg) that have been exposed to As^V-containing solutions (30 mL; 50 ppm initial concentration). The total amounts of bound As^V per node is indicated on the right. Each data point is an average of 3 to 4 different experiments.

faster and more (1.4 : 1 As^V : Zr_6 or ~46% of the missing-linker sites after 30 minutes) than the nearly defect-free AcOH-UiO-66$|_{11/12}$ (Fig. 3, *cf.* red and green profiles). Interestingly, the uptake ratio for HCl-UiO-66$|_{9/12}$ in our standard 50 ppm initial As-exposure experiment levels out relatively quickly and does not change after 6 h: the As^V uptake ratio at this time point, as well as after 24 h of exposure, only amounts to ~60% of the available missing-linker sites (1.8 : 1 As^V : Zr_6) even though there is still excess As^V in solution.[38] These data suggests that the As^V uptake in HCl-UiO-66$|_{9/12}$ is probably dominated by an equilibrium-driven process, as demonstrated for organophosphorus[6] and selenates[42] uptakes in UiO-type MOFs. This is indeed the case: as the As-exposure concentration is increased to 100 ppm, all of the available missing-linker sites (3 : 1 As^V : Zr_6) for HCl-UiO-66$|_{9/12}$ can be filled after 24 h (Fig. 4).

The combination of equilibrium-driven and secondary uptake mechanisms lead to quite different behaviors for AcOH-UiO-66$|_{11/12}$ and HCl-UiO-66$|_{9/12}$ as the initial As-exposure concentration was varied from 5 to 100 ppm (Fig. 4). While the As^V adsorption profiles for HCl-UiO-66$|_{9/12}$ at different time points all increased in the same manner (see also ESI,† Fig. S10d), the As^V : Zr_6 uptake ratios never exceeded the per-node number of missing-linker sites, presumably due to sterics.[43] In contrast, the As^V adsorption profiles for AcOH-UiO-66$|_{11/12}$ varied significantly depending on the timing of the measurements: at 0.5 h the As^V : Zr_6 uptake ratios did not vary much beyond 53% of the available missing-linker sites (0.20–0.53 : 1 As^V : Zr_6) while that at 24 h varied over a very large 34–235% range (0.34–2.35 : 1 As^V : Zr_6) (see also ESI,† Fig. S10c as well as the accompanying discussion that follows Fig. S10).

The aforementioned large discrepancy in behaviors accentuates the differences in As-uptake mechanisms between HCl-UiO-66$|_{9/12}$ and AcOH-UiO-66$|_{11/12}$. While the As-uptake by the former is presumably based on filling up missing-linker sites, that for the latter changes between short and long exposure times. We speculate that the As-uptake mechanism for AcOH-UiO-66$|_{11/12}$ during the short exposure is also based on filling up missing-linker sites but that for the long exposure is dominated by the secondary binding pathways mentioned above. We note in passing that the large number of defect sites, and presumably larger pores (ESI,† Fig. S5), in HCl-UiO-66$|_{9/12}$ are highly advantageous for capturing purposes: the adsorption profiles measured at 3, 6, and 24 h are quite similar (ESI,† Fig. S10d), suggesting that most of the removal occurs during the first 3 h. Under our experimental conditions (Fig. 4), this means that >90% of the As^V oxyanions from a 5 ppm solution can be removed after 0.5 h (0.29 : 1 As^V : Zr_6; see also ESI,† Fig. S10f), and complete removal (0.31 : 1 As^V : Zr_6) occurred after 3 h.

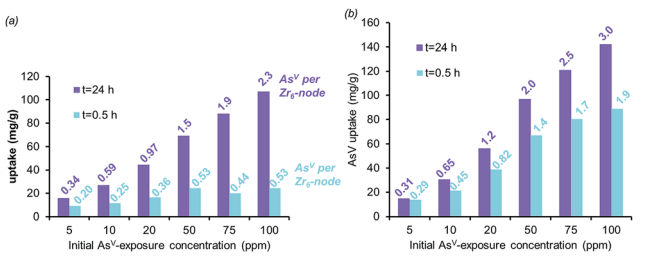

Fig. 4 The As^V-adsorption isotherms for AcOH-UiO-66$|_{11/12}$ (a) and HCl-UiO-66$|_{9/12}$ (b) at short (0.5 h) and long (24 h) exposure time, plotted as bar graphs with the As^V : Zr_6 ratios indicated at the top of each bar. Experimental conditions: batch exposure of a sample of MOF (10 mg) to the appropriate As^V-containing solution (30 mL).

TEM-based EDS analyses of AcOH-UiO-66$|_{11/12}$ and HCl-UiO-66$|_{9/12}$ samples that have been exposed to 100 ppm As^V solutions for 24 h are also consistent with the aforementioned uptake contrast. While no visible morphological changes can be observed, the latter sample clearly showed a much higher As^V uptake based on the relative As/Zr signal ratios (ESI,† Fig. S18). The PXRD data for As^V-exposed materials are identical to the data for the corresponding as-synthesized materials (ESI,† Fig. S1), indicating that the crystallinities of the MOF samples are mostly retained even after significant As^V uptake and prolonged (24 h) shaking. Interestingly, while ICP-OES analysis of the supernatants from the batch-adsorption experiments shows no evidence of Zr^{IV} ions, concurrent analyses of these samples by ESI-MS and high-resolution water-suppression ^{1}H NMR spectroscopy reveals the presence of some H_2BDC linker. Although these data support the occurrence of the aforementioned linker-displacement secondary binding mechanism, and thus possible partial degradation of the initial MOF structure, a quantitative assessment is not possible at the present time (see ESI,† Section S3 for further discussion). Additionally, it is worthwhile to note that the As concentrations that we explored for the uptake experiments herein are much higher than those that exists in natural water sources (1 ppb to 3 ppm),[44] which could accelerate secondary linker-displacement mechanisms such as those mentioned above.

The importance of site accessibility is also reflected in the faster initial-uptake behavior by HCl-UiO-66$|_{9/12}$, which have larger pores (ESI,† Fig. S5) and whose particles are about four times smaller than those of AcOH-UiO-66$|_{11/12}$ (ESI,† Fig. S6). The larger pores of HCl-UiO-66$|_{9/12}$, in comparison to AcOH-UiO-66$|_{11/12}$, can be attributed a combination of higher number of missing linkers and smaller "capping" ligands (HO^-, Cl^-, or H_2O). In the aqueous uptake experiments, these weak-binding ligands can also be more easily displaced by the incoming As^V moieties in contrast to the chelating acetate "capping" ligand for AcOH-UiO-66$|_{11/12}$. In addition to the increase in accessibility that comes with larger pores, samples with smaller particles should have higher external surface area (ESI,† Table S2) that also enables faster As^V chemisorption. Partially supporting this conjecture is the similar initial uptake rates for all three HCl-UiO-66$|_{x/12}$ samples (ESI,† Fig S13a and b and Table S3), which have nearly identical average particle sizes (ESI,† Fig S6).

As^V uptake by UiO-67 derivative

The aforementioned data prompted us to hypothesize that enlarging the pore aperture of UiO-66, through the use of a longer linker, would enhance diffusion and facilitate accessibility to binding sites that are deeper inside a MOF crystal, as

demonstrated by Li and coworkers[45] for the capture of As^V by mesoporous ZIF-8. Thus, we synthesized HCl-UiO-67|$_{9/12}$, a UiO-66 analogue with the longer biphenyl-4,4′-dicarboxylate linker and a similar number of missing linkers as our HCl-UiO-66|$_{9/12}$ sample (based on TGA estimation of missing-linker sites[46]), again using HCl as a modulator. Fortuitously, this sample has similar particle sizes as HCl-UiO-66|$_{9/12}$ (~280 nm *vs.* ~250 nm, see ESI,† Fig. S6), allowing us to compare their As^V uptake behaviors without the need to account for the effects of particle size differences.

Similar to HCl-UiO-66|$_{9/12}$, HCl-UiO-67|$_{9/12}$ displayed a high initial As^V uptake ~56% (1.7 : 1 As^V : Zr_6) within 30 min of exposure to the As^V testing solution (Fig. 3, dark blue curve). However, after 6 h, the As^V uptake for HCl-UiO-67|$_{9/12}$ has risen above that of HCl-UiO-66|$_{9/12}$ (70% binding sites *vs.* 60%). The uptake continues to rise, albeit at a slow rate, over the next 18 h, presumably due to the gradual diffusion of As^V into the internal binding sites of the HCl-UiO-67|$_{9/12}$ particles, filling 90% of binding sites (2.7 : 1 As^V : Zr_6). While the uptake clearly has not reached equilibrium at 24 h, this capacity is very close to that of the expected 3 As^V oxyanions per Zr_6 node. As in the cases for AcOH-UiO-66|$_{11/12}$ and HCl-UiO-66|$_{9/12}$, the PXRD pattern of the As^V-exposed HCl-UiO-67|$_{9/12}$ does not differ from that the corresponding as-synthesized materials (ESI,† Fig. S3a), which is consistent with a retention of some sample crystallinity. Together, these data clearly indicate that the larger pores in HCl-UiO-67|$_{9/12}$ (ESI,† Table S2) can definitely facilitate uptake by sites that are deeper inside a MOF nanocrystal (see further discussion below).

It is worth noting that the initial uptake profiles (over the first 30 minutes) of all five of our MOF samples discussed thus far fit well to the pseudo-first-order Lagergren kinetic model while the total uptake profiles (over a 24 h period) fit best to the pseudo-second-order Lagergren kinetic model (ESI,† Fig. S12 and S13). These results are consistent with the adsorption process being governed initially by the chemisorption of As^V to the readily accessible binding sites near the surfaces of the nanocrystals and becoming diffusion-limited over time as As^V anions migrate into the MOF nanoparticles. Among the HCl-UiO-66|$_{x/12}$ samples, this diffusion-limited behavior becomes most apparent after 3 h, with the sample having the most missing linkers displaying the highest equilibrium capacity. Presumably, the samples with more missing linkers will also have larger pores that facilitate diffusion (see ESI,† Fig. S13 and its caption for further discussion).

As mentioned earlier, phosphonates,[6,7] and vanadates,[36] which have similar structures to arsenates, have been reported to bind strongly to the missing-linker sites on the $Zr_6(O)_4(OH)_4$ node of UiO-type MOFs through Zr–O–M motifs (M = V, P). As such, we also expect arsenates to displace any weakly bound monotopic "capping" ligand (acetates, chlorides, and/or water) at the missing-linker sites (Fig. 2, bottom section) in our HCl-UiO-type MOF crystals and form strong Zr–O–M bonds through the so-called anion-exchange mechanism.[39,42] Analyses of the diffuse-reflectance infrared Fourier-transformed spectroscopy (DRIFTS) data revealed a broad peak (800–900 cm^{-1}) indicative of the adsorbed As^V. The blue-shifted shoulder peak

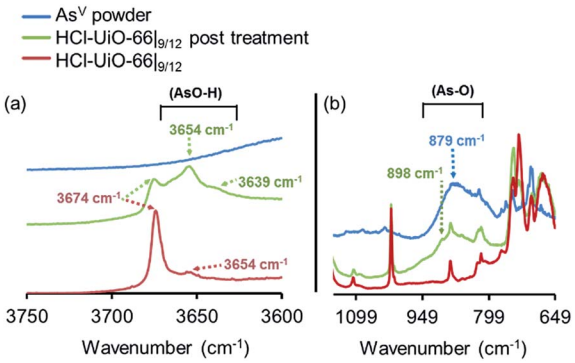

Fig. 5 DRIFTS spectra of HCl-UiO-66|$_{9/12}$ sample before and after As^V treatment, showing the presence of –As–O–H bonds after exposure. Spectra for powder $Na_2HAsO_4 \cdot 7H_2O$ are included as reference. Data for AcOH-UiO-66|$_{11/12}$ can be found in the ESI,† Fig. S14.

at 898 cm^{-1} could be due to formation of As–OZr stretch as observed for zirconium arsenate crystals.[47] The As3d XPS spectrum of As^V-treated HCl-UiO-67|$_{9/12}$ reflected a ~0.5 eV blue shift in binding energy in comparison to that for powder $Na_2HAsO_4 \cdot 7H_2O$ (Fig. 5a), in agreement with formation of As–O–Zr species[41,47,48] (see ESI,† Fig. S13–S16 and their captions for further discussion). Interestingly, there is a new peak (3654 cm^{-1}) in the bridging hydroxide region of the DRIFTS spectra for the As^V-treated MOF samples (Fig. 5a and ESI,† Fig. S14b), which we attribute to a combination of As(OH) and AsO···H···OZr species, similar to the PO···H···OZr species reported by Deria *et al.*[26] Together, these data leads us to believe that the arsenate oxyanions are coordinated to the $Zr_6(O)_4(OH)_4$ node (possible binding motifs shown in Fig. 2 and ESI,† Fig. S14).

As^{III} and As^V uptake by thiolated UiO-66 derivatives

To demonstrate that neutral As^{III} species can also be captured by the UiO-66 platform using thiolated ligand sites, we prepared UiO-66$(SH)_2$ using both AcOH and HCl modulators (see ESI,† Section S2 for experimental details), which afforded thiolated MOFs with the same amount of defects, $Zr_6O_4(OH)_4(-CO_2)_{10.5/12}$ (see ESI,† Section S5). When exposed to aqueous solutions of As^{III},[48] both AcOH-UiO-66$(SH)_2$ and HCl-UiO-66$(SH)_2$ showed much higher uptake compared to the non-thiolated controls (Fig. 6a), confirming our hypothesis that thiol can be used to chemoselectively capture As^{III}. Further supporting the importance of the thiol functionalities is the observation that HCl-UiO-66$(OH)_2$, the hydroxylated analogue of HCl-UiO-66$(SH)_2$, does not uptake any significant amount of As^{III} under the same condition (Fig. 6a). While HCl-UiO-66$(SH)_2$, with higher total pore volume and higher micropore surface area (see ESI,† Table S2), is expected to have a higher As^{III} capacity, the uptake amount (40 mg g^{-1}, 1 : 1 As^{III} : Zr_6) is twice that obtained for AcOH-UiO-66$(SH)_2$. We attribute this to the non-negligible size of the acetate "capping" ligand in the latter, which can significantly narrow the surface apertures of the MOF crystals and reduce the amount of uptake by subsurface sites. As mentioned above, the chelating nature of the acetate ligand made it less likely to be displaced by HO^- or H_2O ligand during the uptake

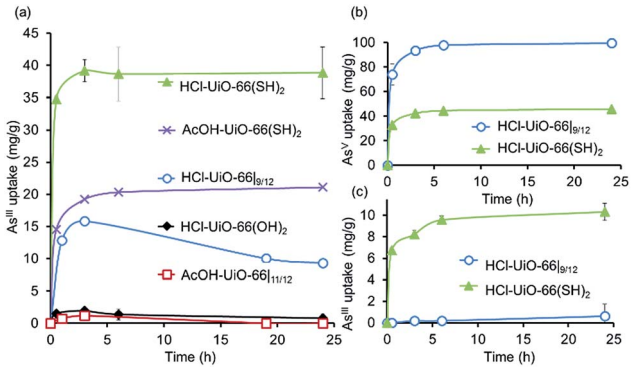

Fig. 6 (a) As^{III} uptake profiles[49] by thiolated UiO-66 samples and non-functionalized analogues confirming the important role of the soft thiol ligands in capturing As^{III}. While the AcOH-UiO-66$|_{11/12}$ and HCl-UiO-66$|_{9/12}$ controls show some initial uptake of As^{III}, this appears to be semi-reversible, not unexpectedly if we consider the weaker binding nature of the soft As^{III} ion to the hard missing-linker sites of the nodes, especially under slightly acidic conditions. (b and c) In sequential exposures to As^{V} (b) and As^{III} (c), HCl-UiO-66$(SH)_2$ shows good uptakes for both As^{V} and As^{III} compared to HCl-UiO-66$|_{9/12}$, which only binds As^{V}. See ESI† for data concerning the reverse exposure order. Experimental conditions: batch exposure of a sample of MOF (10 mg) to the appropriate As-containing solution (30 mL; 50 ppm initial concentration).

experiments, making this mechanism of pore-aperture-narrowing more significant for the "acetate-capped" materials. Notably, sequential exposure of both HCl-UiO-66$(SH)_2$ and HCl-UiO-66$|_{9/12}$ to As^{V} (Fig. 6b) and then to As^{III} (Fig. 6c), showed a stark contrast of the two materials: the latter only bind As^{V} while the former can still bind a notable amount of As^{III} (10 mg g^{-1} at 6 h, 0.26 : 1 As^{III} : Zr_6) after significant As^{V} uptake (40 mg g^{-1} at 6 h, 1 : 1 As^{V} : Zr_6). This 1 : 1 As^{V} : Zr_6 stoichiometry corresponds to 67% of the defect sites on HCl-UiO-66$(SH)_2$ and is comparable to that of AcOH-UiO-66$|_{11/12}$ (*cf.* Fig. 6b and 3). Consistent with its low level of defects, exposing HCl-UiO-66$(SH)_2$ to As^{V} solution (30 mL of 50 ppm As^{V}) for 24 h does not appear to degrade it (<1% BDC-SH linker seen in solution *via* ICP-OES sulfur analysis; see ESI,† Section S3).

Interestingly, sequential exposure of HCl-UiO-66$(SH)_2$ to As^{III} and then to As^{V} solutions (ESI,† Fig. S9) shows a reverse ordering of uptake capacities (40 mg g^{-1} at 6 h of the first exposure plus 10 mg g^{-1} of As^{V} at 6 h of the second exposure). These data suggest that while HCl-UiO-66$(SH)_2$ has capability for binding both As^{III} and As^{V}, the total capacity may again be limited by sterics. Full uptake of the first species, regardless of the oxidation state, invariably results in a narrowed pore that lowers the accessibility of the binding sites deeper inside the nanocrystals. Such a case would result in lower uptakes of the second species than available binding sites.

The reversibility of As^{III} and As^{V} binding

To identify some possible post-adsorption regeneration strategies for these promising MOF sorbents, we examined the reversibility of As binding in these materials (see additional discussion in ESI,† Section S9). Given that $[As^{III}(OH)_3]_n$ species are known to bind to sulfur-containing enzymes and thiol-rich chelators through the exchange of the hydroxyl group with RSH moieties, we hypothesized that the $(BDC\text{-}S)_xAs(OH)_{3-x}$ species present in As^{III}-treated HCl-UiO-66$(SH)_2$ is best decomposed into soluble As^{III} moieties and intact MOF *via* competitive ligand exchange with excess soluble thiols (eqn (1)). Indeed, about 30% of As^{III} can be removed from As^{III}-treated UiO-66$(SH)_2$ within 3 h when treated with an ~0.5 M solution of thiophenols (~90 equiv in excess) at 50 °C under stirring. No As was removed without stirring or heating, consistent with a thermodynamically driven equilibrium that is limited by slow diffusion through the crystal. Also consistent with this hypothesis are the observations that treatments with a smaller excess of thiophenols and the use of less acidic alkylthiols did not work as well (ESI,† Table S5). Notably, Zr^{IV} ions and BDC-$(SH)_2$ linkers were not observed in the treatment solution, suggesting that HCl-UiO-66$(SH)_2$ is completely stable under these conditions. The stability of this MOF to regeneration is highly beneficial as it can be reused in applications that aim to remove As^{III}, which is the more toxic and prevalent form of arsenic in anaerobic ground water streams.[11]

$$RSH\ (xs) + BDC\text{-}S\text{-}As^{III} \rightleftharpoons RS\text{-}As^{III} + BDC\text{-}SH \qquad (1)$$

The desorption of As^{V} from As^{V}-treated AcOH-UiO-66$|_{11/12}$ and HCl-UiO-66$|_{9/12}$ is slightly more problematic given their strong chelation to the nodes (Fig. 1), which can only be disrupted by treatment with either a strong acid or base (eqn (2a) and (b)); however such treatments may also facilitate some decomposition of the MOF.[50] Indeed, while subjecting As^{V}-treated HCl-UiO-66$|_{9/12}$ to either a 3.3 M HCl (~16 000 equiv per node) or a 3.3 M NaOH (>1000 equiv per node) solution led to the desorption of a significant amount of As^{V}, a small amount of Zr^{IV} ions was also released (ESI,† Table S4). From the limited set of data that we obtained to date, we are optimistic that this loss of Zr^{IV} ions can be minimized with proper optimization of exposure time and acid (or base) concentrations.

$$HCl\ (xs) + Zr_6O_4(OH)_4(\text{-}CO_2)_{x/12}(HAsO_4^-)_{12-x} \rightleftharpoons Zr_6O_4(OH)_4(\text{-}CO_2)_{x/12}(Cl)_{12-x}(H_2O)_{12-x} + H_3AsO_4 \qquad (2a)$$

$$NaOH\ (xs) + Zr_6O_4(OH)_4(\text{-}CO_2)_{x/12}(HAsO_4^-)_{12-x} \rightleftharpoons Zr_6O_4(OH)_4(\text{-}CO_2)_{x/12}(OH)_{12-x}(H_2O)_{12-x} + Na_yH_{3-y}AsO_4 \qquad (2b)$$

We note that exposing As^{V}-treated HCl-UiO-66$|_{9/12}$ to solutions with pH 2, 7, or 12 did not lead to any noticeable As release until pH 12 (ESI,† Fig. S22). This observation is in agreement with a previous report by Wang *et al.* where UiO-66 exhibits the lowest As^{V} uptake at pH 11 in the 2–11 pH range that was studied.[39] Interestingly, exposing As^{V}-treated AcOH-UiO-66$|_{11/12}$ to the same series of pH solutions led to a more noticeable desorption of As^{V} that increases proportionally with the basicity of the solution (ESI,† Fig. S22). This process may be attributed

to the removal of the weakly bound As^V that originally adsorbed onto AcOH-UiO-66$|_{11/12}$ through secondary binding mechanisms (see discussion above).

Conclusions

In summary, we have successfully demonstrated the respective use of the nodes and linkers in a series of UiO MOFs to chemoselectively capture anionic As^V and neutral As^{III}. Missing-linker sites on the $Zr_6(O)_4(OH)_4$ nodes are excellent binders for As^V oxyanions, thiolated linkers can selectively coordinate As^{III}, and both of these recognition motifs can be incorporated into the same framework for dual-capture purposes. Our results also suggest that the full capacity of this dual-binding feature is best realized when the binding sites internal to the MOF crystals are made easily accessible, either by enlarging the pore aperture size or by reducing the particle sizes. Notably, the binding of both As^V and As^{III} appear to be reversible with the proper treatments, suggesting that the dual-capture strategy can be incorporated into the design of regenerable/reusable adsorbents capable of efficiently capturing multiple pollutants or toxic agents that exist as diverse species in aqueous environments.

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