



CrossMark  
click for updates

Cite this: *RSC Adv.*, 2016, 6, 98267

# Recent advancements in chemical looping water splitting for the production of hydrogen

G. Voitic\* and V. Hacker

Chemical looping water splitting or chemical looping hydrogen is a very promising technology for the production of hydrogen. In recent years extensive research has enabled remarkable leaps towards a successful integration of the chemical looping technology into a future hydrogen infrastructure. Progress has been reported with iron based oxygen carriers for stable hydrogen production capacity over consecutive cycles without significant signs of degradation. The high stability improvements were achieved by adding alien metal oxides or by integrating the active component into a mineral structure which offers excellent resistance towards thermal stress. Prototype systems from small  $\mu$ -systems up to 50 kW have been operated with promising results. The chemical looping water splitting process was broadened in terms of its application area and utilization of feedstocks using a variety of renewable and fossil resources. The three-reactor system was clearly advantageous due to its flexibility, heat integration capabilities and possibility to produce separate pure streams of hydrogen, CO<sub>2</sub> and N<sub>2</sub>. However two-reactor and single fixed-bed reactor systems were successfully operated as well. This review aims to survey the recently presented literature in detail and systematically summarize the gathered data.

Received 23rd August 2016  
Accepted 29th September 2016

DOI: 10.1039/c6ra21180a

[www.rsc.org/advances](http://www.rsc.org/advances)

## 1. Introduction

Over the last decades the global energy demand has risen significantly. This means that large amounts of fossil resources

are consumed, leading to the emission of huge quantities of CO<sub>2</sub>.<sup>1</sup> There is a strong consensus in the scientific community that these anthropogenic carbon emissions are jointly responsible for the global temperature rise.<sup>2</sup> To counteract the trend of continuing temperature increase, clear targets have been presented to the public and to political leaders. One very important task is the reduction of CO<sub>2</sub> emissions from fossil fuels by switching to clean and renewable energy sources, to limit the

Graz University of Technology, Institute of Chemical Engineering and Environmental Technology, NAWI Graz, Inffeldgasse 25/C/II, 8010 Graz, Austria. E-mail: gernot.voitic@tugraz.at; viktor.hacker@tugraz.at



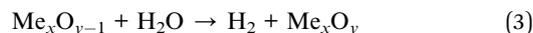
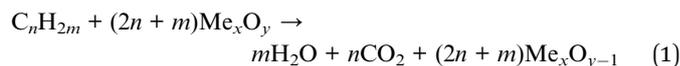
*Gernot Voitic is a PhD student at the Institute of Chemical Engineering and Environmental Technology at Graz University of Technology under supervision of Professor Viktor Hacker. He graduated in Chemistry in 2011 on polymer membranes for hydrogen purification. His research interests are hydrogen production, chemical looping technology and catalytic and non-catalytic gas reactions.*



*Viktor Hacker received the MSc in mechanical engineering, the PhD in electrotechnical engineering and 2004 the *venia docendi* in the field of electrochemical energy technologies from Graz University of Technology. As Associate Professor he is Head of the Fuel Cell Research Group and Deputy Head of the Institute of Chemical Engineering at Graz University of Technology, has long-time experience in the coordination and realization of national and international research projects in the field of fuel cells and hydrogen production and acts as Austrian Representative in the International Energy Agency (IEA), Implementing Agreement on Advanced Fuel Cells. Prof. Hacker published more than hundred papers on fuel cells and hydrogen production.*



average global temperature increase to 2 °C.<sup>3,4</sup> In this context hydrogen is considered as a promising clean fuel for future generations. However, hydrogen is a secondary energy source, meaning that its environmental impact is defined by the resources and their production and purification processes. Currently hydrogen is mainly produced in large scale reforming plants with fossil hydrocarbons or coal as feed, which results in significant CO<sub>2</sub> emissions.<sup>5</sup> There are two main technological routes, which enable the production of clean hydrogen. The first option is the application of technologies which utilize CO<sub>2</sub>-neutral resources or energy for instance electrolysis from wind and solar or reforming of renewable hydrocarbons.<sup>6</sup> The second option is production based on fossil fuels in combination with carbon capture and storage systems.<sup>7</sup> Both pathways are pursued using several different technologies, which are already in operation as prototypes or on an industrial scale. However, they usually suffer from higher hydrogen production costs compared to conventional plants, which hinders market penetration.<sup>6,8</sup> Chemical looping hydrogen systems represent a promising technology that can offer both the use of renewable, CO<sub>2</sub> neutral resources as well as efficient CO<sub>2</sub> sequestration capabilities.<sup>7</sup> Historically, the chemical looping process for the generation of hydrogen was introduced as the “steam-iron” process in 1903 by Howard Lane and his co-workers.<sup>9–17</sup> Nowadays, it is still referred to as a steam-iron process, although it is usually described as a modification of the chemical looping combustion process CLC and is thus named “chemical looping hydrogen” CLH or “chemical looping water splitting” CLWS. Generally, chemical looping technology (Fig. 1) uses a solid oxygen carrier, which supplies oxygen for the combustion of a fuel and reduction of the solid material (eqn (1)). This omits the utilization of air as oxygen supply and thus enables the fuel conversion to a pure stream of CO<sub>2</sub> and H<sub>2</sub>O, which can be easily separated by condensation. The solid material is re-oxidized with air to regenerate the depleted oxygen (eqn (2)). The addition of a third reaction step enables the production of pure hydrogen by re-oxidizing the oxygen carrier with water (eqn (3)). An air oxidation might still be executed to close the looping cycle, but it reduces the hydrogen production rate and thereby the maximum efficiency of an integrated system as discussed by Hacker *et al.* in relation to different fuels.<sup>18,19</sup>



This state-of-the-art review focuses on very recent research data presented in literature on the subject of hydrogen production by means of chemical looping technology. The publications studied were divided between the main sections (i) oxygen carrier development, (ii) reactor design, (iii) process development, (iv) process simulation and (v) kinetic studies. An overview of the evaluated data is presented in tables for each section.

## 2. Oxygen carrier development

The oxygen carrier functions as a transport intermediate to deliver pure oxygen without nitrogen dilution from the oxidizing stream, air or water, to the fuel. The utilization of a suitable oxygen carrier is of great importance in chemical looping systems for hydrogen production. Thus, a great deal of literature deals with improvements mainly implemented by means of two different approaches, (i) optimizing the composition or (ii) optimizing the particle structure. A summary of the utilised oxygen carrier is presented in Table 1. Key properties for industrial applications, which must be met by suitable oxygen carriers are:<sup>20–30</sup>

- A high oxygen exchange capacity.
- Favourable thermodynamics and reactivity regarding the selected fuel for the reduction reactions.
- A low risk of carbon formation.
- Structural stability under continuous reduction and oxidation cycles.
- High reactivity in the oxidation reactions.
- Good fluidization properties and resistance to attrition, which is important in fluidized reactor systems.
- Feasible material and production costs.
- Environmental friendly characteristics.

Various metals *viz.* Fe, W, Sn, Ni, Cu, Mn and Ce have been tested as the main active component. However, the more recent work almost exclusively investigated iron based oxygen carrier.<sup>31</sup> As pure metals do not fulfil the listed requirements, inert components such as ZrO<sub>2</sub>, TiO<sub>2</sub>, SiO<sub>2</sub>, Al<sub>2</sub>O<sub>3</sub> as additives are used to improve the mechanical and chemical properties.<sup>32</sup> Additionally, the active component can be incorporated into a mineral structure, for instance a perovskites or a spinel structure, which are known for their mechanical resistance or two active metals are combined as a bimetallic oxygen carrier to generate synergetic effects of their components.<sup>33</sup>

### Zirconia as additive

Liu *et al.*<sup>34</sup> achieved a stable hydrogen yield of 90% based on the theoretical value over 20 cycles by adding up to 30 wt% of ZrO<sub>2</sub>. The structural analysis revealed partial sintering of iron

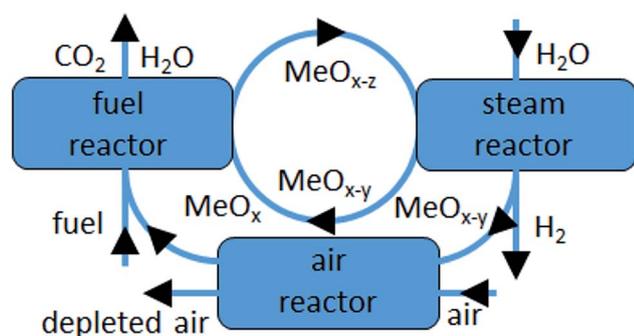


Fig. 1 Process schematic of the chemical looping hydrogen production process with an alternative air oxidation.



Table 1 Summary of prepared and utilized oxygen carrier

Active component	Additives/support	Preparation method	Experimental device	Composition	Reference
Fe	NiFe <sub>2</sub> O <sub>4</sub> , CoFe <sub>2</sub> O <sub>4</sub> , ZrO <sub>2</sub>	Incipient wetness method	Fixed bed	Fe/Ni = 1.9/1 wt%, Fe/Co = 2.1/1 wt%	29
Fe	CeO <sub>2</sub>	Chemical precipitation		Ce <sub>1-x</sub> Fe <sub>x</sub> O <sub>2-δ</sub> , x = 0, 0.1, 0.2, 0.3, 0.4, 0.5, 0.6, 1	31
Fe	TiO <sub>2</sub> , La <sub>0.8</sub> Si <sub>0.2</sub> , FeO <sub>3</sub> , Al <sub>2</sub> O <sub>3</sub> , YSZ	Solid state reaction	TGA	Fe <sub>2</sub> O <sub>3</sub> = 60 wt%	33
Fe	ZrO <sub>2</sub>	Co-precipitation	TGA, packed bed, fluidized bed	Fe <sub>2</sub> O <sub>3</sub> = 83.7–82.2 wt%, Fe <sub>2</sub> O <sub>3</sub> 58–58.2 wt%	34
Fe	ZrO <sub>2</sub>	Co-precipitation	TGA + fixed bed	Fe <sub>2</sub> O <sub>3</sub> /ZrO <sub>2</sub> 20/80 wt%	35 and 36
Fe	ZrO <sub>2</sub> , CeO <sub>2</sub> , Zr <sub>0.8</sub> Y <sub>0.2</sub> O <sub>2-δ</sub> (YSZ), Ce <sub>0.8</sub> Gd <sub>0.2</sub> O <sub>2-δ</sub> (GDC)	Solid state reaction	TGA	Fe <sub>2</sub> O <sub>3</sub> /support = 30/70 wt%	37
Ce	ZrO <sub>2</sub>	Colloidal crystal	TPR	Ce/Zr = 80/20 wt%	38 and 39
Fe	Ce <sub>0.8</sub> Zr <sub>0.2</sub> O <sub>2</sub>	template method	Fixed bed	Fe/Ce = 1	40
Fe	CeO <sub>2</sub>	Co-precipitation	TPR	Fe/Ce = 95/5, 90/10, 80/20, 60/40, 50/50 wt%	41
				Fe <sub>2</sub> O <sub>3</sub> wt%    Al <sub>2</sub> O <sub>3</sub> wt%    MoO <sub>3</sub> wt%    CeO <sub>2</sub> wt%	
				100    0    0    0	
				98    1.90    0.10    0	
				98    1.75    0.25    0	42
				98    1.50    0.50    0	
				98    1.75    0    0.25	
				97.79    1.75    0.20    0.25	
Fe	Support: Al <sub>2</sub> O <sub>3</sub> , additives: MoO <sub>3</sub> , CeO <sub>2</sub>	Citrate sol-gel method	Fixed bed		
Fe	Ce, Zr, Mo	Urea hydrolysis, impregnation	TPR, TGA		43 and 44



Table 1 (Contd.)

Active component	Additives/support	Preparation method	Experimental device	Composition	Reference																																																		
				<table border="1"> <thead> <tr> <th>Sample name</th> <th>Fe wt%</th> <th>Ce wt%</th> <th>Zr wt%</th> <th>Mo wt%</th> </tr> </thead> <tbody> <tr><td>FCZ</td><td>56.1</td><td>10.1</td><td>4.49</td><td></td></tr> <tr><td>FCZ-1Mo</td><td>58.0</td><td>11.0</td><td>4.50</td><td>0.96</td></tr> <tr><td>FCZ-2Mo</td><td>58.0</td><td>10.0</td><td>4.50</td><td>1.50</td></tr> <tr><td>FCZ-3Mo</td><td>57.0</td><td>10.0</td><td>4.50</td><td>3.00</td></tr> <tr><td>FCZ-4Mo</td><td>57.0</td><td>10.0</td><td>4.50</td><td>4.10</td></tr> <tr><td>FCZ-5Mo</td><td>56.0</td><td>11.0</td><td>4.40</td><td>5.50</td></tr> </tbody> </table>	Sample name	Fe wt%	Ce wt%	Zr wt%	Mo wt%	FCZ	56.1	10.1	4.49		FCZ-1Mo	58.0	11.0	4.50	0.96	FCZ-2Mo	58.0	10.0	4.50	1.50	FCZ-3Mo	57.0	10.0	4.50	3.00	FCZ-4Mo	57.0	10.0	4.50	4.10	FCZ-5Mo	56.0	11.0	4.40	5.50																
Sample name	Fe wt%	Ce wt%	Zr wt%	Mo wt%																																																			
FCZ	56.1	10.1	4.49																																																				
FCZ-1Mo	58.0	11.0	4.50	0.96																																																			
FCZ-2Mo	58.0	10.0	4.50	1.50																																																			
FCZ-3Mo	57.0	10.0	4.50	3.00																																																			
FCZ-4Mo	57.0	10.0	4.50	4.10																																																			
FCZ-5Mo	56.0	11.0	4.40	5.50																																																			
Fe	Mo	Hydrothermal synthesis	TPR	<table border="1"> <tbody> <tr><td>0 Mo</td><td>56.1</td><td>10.1</td><td>4.49</td><td></td></tr> <tr><td>1 Mo</td><td>52.6</td><td>9.72</td><td>4.43</td><td>0.98</td></tr> <tr><td>2 Mo</td><td>50.8</td><td>9.61</td><td>4.28</td><td>1.91</td></tr> <tr><td>3 Mo</td><td>49.4</td><td>9.16</td><td>4.22</td><td>2.85</td></tr> <tr><td>4 Mo</td><td>48.8</td><td>9.20</td><td>4.18</td><td>3.78</td></tr> <tr><td>5 Mo</td><td>48.4</td><td>8.94</td><td>4.12</td><td>4.71</td></tr> </tbody> </table> <p>Fe<sub>2</sub>O<sub>3</sub>, Fe<sub>2</sub>O<sub>3</sub>/Mo = 95/5, 92/8, 90/10 mol%</p>	0 Mo	56.1	10.1	4.49		1 Mo	52.6	9.72	4.43	0.98	2 Mo	50.8	9.61	4.28	1.91	3 Mo	49.4	9.16	4.22	2.85	4 Mo	48.8	9.20	4.18	3.78	5 Mo	48.4	8.94	4.12	4.71	45																				
0 Mo	56.1	10.1	4.49																																																				
1 Mo	52.6	9.72	4.43	0.98																																																			
2 Mo	50.8	9.61	4.28	1.91																																																			
3 Mo	49.4	9.16	4.22	2.85																																																			
4 Mo	48.8	9.20	4.18	3.78																																																			
5 Mo	48.4	8.94	4.12	4.71																																																			
Fe	Al <sub>2</sub> O <sub>3</sub> , SiO <sub>2</sub> , CaO	Co-precipitation urea hydrolysis	TPR	<table border="1"> <thead> <tr> <th>Sample name</th> <th>Fe<sub>2</sub>O<sub>3</sub> wt%</th> <th>Al<sub>2</sub>O<sub>3</sub> wt%</th> <th>SiO<sub>2</sub> wt%</th> <th>CaO wt%</th> </tr> </thead> <tbody> <tr><td>F1</td><td>100.0</td><td></td><td></td><td></td></tr> <tr><td>F2</td><td>95.0</td><td>5.0</td><td></td><td></td></tr> <tr><td>F3</td><td>92.5</td><td>5.0</td><td></td><td>2.5</td></tr> <tr><td>F4</td><td>90.0</td><td>5.0</td><td>7.5</td><td>5.0</td></tr> <tr><td>F5</td><td>88.5</td><td>5.0</td><td>7.5</td><td></td></tr> <tr><td>F6</td><td>90.0</td><td>5.0</td><td>7.5</td><td>2.5</td></tr> <tr><td>F7</td><td>85.0</td><td>5.0</td><td>7.5</td><td>7.5</td></tr> <tr><td>F8</td><td>85.0</td><td></td><td>5.0</td><td>2.5</td></tr> <tr><td>F9</td><td>85.0</td><td></td><td></td><td>5</td></tr> </tbody> </table>	Sample name	Fe <sub>2</sub> O <sub>3</sub> wt%	Al <sub>2</sub> O <sub>3</sub> wt%	SiO <sub>2</sub> wt%	CaO wt%	F1	100.0				F2	95.0	5.0			F3	92.5	5.0		2.5	F4	90.0	5.0	7.5	5.0	F5	88.5	5.0	7.5		F6	90.0	5.0	7.5	2.5	F7	85.0	5.0	7.5	7.5	F8	85.0		5.0	2.5	F9	85.0			5	46
Sample name	Fe <sub>2</sub> O <sub>3</sub> wt%	Al <sub>2</sub> O <sub>3</sub> wt%	SiO <sub>2</sub> wt%	CaO wt%																																																			
F1	100.0																																																						
F2	95.0	5.0																																																					
F3	92.5	5.0		2.5																																																			
F4	90.0	5.0	7.5	5.0																																																			
F5	88.5	5.0	7.5																																																				
F6	90.0	5.0	7.5	2.5																																																			
F7	85.0	5.0	7.5	7.5																																																			
F8	85.0		5.0	2.5																																																			
F9	85.0			5																																																			
Fe	CaO	Mechanical mixing	TGA, TPR, fluidized bed	Fe <sub>2</sub> O <sub>3</sub> /CaO = 50/50, 42.7/57.3, 33.3/66.7 mol%	47																																																		
Fe, Ca <sub>2</sub> Fe <sub>2</sub> O <sub>5</sub>	ZrO <sub>2</sub>	Pechini method	TGA, fixed bed	Fe <sub>2</sub> O <sub>3</sub> /ZrO <sub>2</sub> = 60/40 wt%	48																																																		
Fe	Al <sub>2</sub> O <sub>3</sub> , Na <sub>2</sub> O, MgO	Co-precipitation	TGA, fixed bed	<table border="1"> <thead> <tr> <th>Sample name</th> <th>Fe<sub>2</sub>O<sub>3</sub> wt%</th> <th>Al<sub>2</sub>O<sub>3</sub> wt%</th> <th>Na<sub>2</sub>O wt%</th> <th>MgO wt%</th> <th>other wt%</th> </tr> </thead> <tbody> <tr><td>NAFO</td><td>64.3</td><td>20.7</td><td>12.2</td><td>0</td><td>2.8</td></tr> <tr><td>NMAFO</td><td>51.7</td><td>25.2</td><td>10.7</td><td>10.2</td><td>2.1</td></tr> </tbody> </table>	Sample name	Fe <sub>2</sub> O <sub>3</sub> wt%	Al <sub>2</sub> O <sub>3</sub> wt%	Na <sub>2</sub> O wt%	MgO wt%	other wt%	NAFO	64.3	20.7	12.2	0	2.8	NMAFO	51.7	25.2	10.7	10.2	2.1	53																																
Sample name	Fe <sub>2</sub> O <sub>3</sub> wt%	Al <sub>2</sub> O <sub>3</sub> wt%	Na <sub>2</sub> O wt%	MgO wt%	other wt%																																																		
NAFO	64.3	20.7	12.2	0	2.8																																																		
NMAFO	51.7	25.2	10.7	10.2	2.1																																																		
Fe	TiO <sub>2</sub> ilmenite	Ball milled tablet pressed	TGA, fixed bed	Fe <sub>2</sub> TiO <sub>5</sub>	57																																																		





Table 1 (Contd.)

Active component	Additives/support	Preparation method	Experimental device	Composition	Reference																												
Fe	TiO <sub>2</sub> , SiO <sub>2</sub> , Al <sub>2</sub> O <sub>3</sub> , CaO, MgO, Na <sub>2</sub> O, MnO, Cr <sub>2</sub> O <sub>3</sub> , K <sub>2</sub> O	Fractioning, calcination	TGA, fixed bed	<table border="1"> <thead> <tr> <th>Fe<sub>2</sub>O<sub>3</sub> wt%</th> <th>TiO<sub>2</sub> wt%</th> <th>SiO<sub>2</sub> wt%</th> <th>Al<sub>2</sub>O<sub>3</sub> wt%</th> <th>CaO wt%</th> <th>MgO wt%</th> <th>Na<sub>2</sub>O wt%</th> <th>MnO wt%</th> <th>Cr<sub>2</sub>O<sub>3</sub> wt%</th> <th>K<sub>2</sub>O wt%</th> <th>rest wt%</th> </tr> </thead> <tbody> <tr> <td>44.8</td> <td>29.1</td> <td>11.3</td> <td>8.3</td> <td>1.34</td> <td>2.83</td> <td>1.33</td> <td>0.10</td> <td>0.11</td> <td>0.13</td> <td>0.66</td> </tr> </tbody> </table>	Fe <sub>2</sub> O <sub>3</sub> wt%	TiO <sub>2</sub> wt%	SiO <sub>2</sub> wt%	Al <sub>2</sub> O <sub>3</sub> wt%	CaO wt%	MgO wt%	Na <sub>2</sub> O wt%	MnO wt%	Cr <sub>2</sub> O <sub>3</sub> wt%	K <sub>2</sub> O wt%	rest wt%	44.8	29.1	11.3	8.3	1.34	2.83	1.33	0.10	0.11	0.13	0.66	58 and 87						
				Fe <sub>2</sub> O <sub>3</sub> wt%	TiO <sub>2</sub> wt%	SiO <sub>2</sub> wt%	Al <sub>2</sub> O <sub>3</sub> wt%	CaO wt%	MgO wt%	Na <sub>2</sub> O wt%	MnO wt%	Cr <sub>2</sub> O <sub>3</sub> wt%	K <sub>2</sub> O wt%	rest wt%																			
				44.8	29.1	11.3	8.3	1.34	2.83	1.33	0.10	0.11	0.13	0.66																			
CVRD - Iron ore	94.79	0.00	4.31	0.90	0.00	0.00	0.00	0.00	0.00	0.00																							
Fe	MAC - Iron ore			93.96	0.00	3.96	2.08	0.00	0.00	0.00	0.00	0.00	0.00																				
				YD - Iron ore	93.19	0.00	5.20	1.61	0.00	0.00	0.00	0.00	0.00	0.00																			
				Fe <sub>2</sub> O <sub>3</sub> /SiO <sub>2</sub> /Al <sub>2</sub> O <sub>3</sub> /other = 83.25/7.06/5.32/4.37 wt%, KNO <sub>3</sub> loading = 0, 3, 6, 10 wt%	60																												
Mixed Fe-Cu	Ni, La <sub>2</sub> O <sub>3</sub> , KNO <sub>3</sub> as promotor Al <sub>2</sub> O <sub>3</sub>	Co-precipitation, incipient wetness impregnation	TPR, fluidized bed	20 °C samples prepared, samples tested	61																												
Co-W	Ni, La <sub>2</sub> O <sub>3</sub>	Co-precipitation, impregnation	TGA	<table border="1"> <thead> <tr> <th>Sample name</th> <th>Fe<sub>2</sub>O<sub>3</sub> wt%</th> <th>Cu<sub>2</sub>O wt%</th> <th>Al<sub>2</sub>O<sub>3</sub> wt%</th> </tr> </thead> <tbody> <tr> <td>Fe18-IWI</td> <td>8</td> <td></td> <td>82</td> </tr> <tr> <td>Fe62-CoPr</td> <td>62</td> <td></td> <td>38</td> </tr> <tr> <td>Cu17-IWI</td> <td></td> <td>17</td> <td>83</td> </tr> <tr> <td>Cu60-CoPr</td> <td></td> <td>60</td> <td>40</td> </tr> <tr> <td>Fe16Cu15-IWI</td> <td>16</td> <td>15</td> <td>69</td> </tr> <tr> <td>Fe17Cu15-CoPr</td> <td>17</td> <td>15</td> <td>68</td> </tr> </tbody> </table>	Sample name	Fe <sub>2</sub> O <sub>3</sub> wt%	Cu <sub>2</sub> O wt%	Al <sub>2</sub> O <sub>3</sub> wt%	Fe18-IWI	8		82	Fe62-CoPr	62		38	Cu17-IWI		17	83	Cu60-CoPr		60	40	Fe16Cu15-IWI	16	15	69	Fe17Cu15-CoPr	17	15	68	63
				Sample name	Fe <sub>2</sub> O <sub>3</sub> wt%	Cu <sub>2</sub> O wt%	Al <sub>2</sub> O <sub>3</sub> wt%																										
				Fe18-IWI	8		82																										
				Fe62-CoPr	62		38																										
				Cu17-IWI		17	83																										
Cu60-CoPr		60	40																														
Fe16Cu15-IWI	16	15	69																														
Fe17Cu15-CoPr	17	15	68																														
Fe	Perovskite, LaNiO <sub>3</sub> , CeO <sub>2</sub> /LaNiO <sub>3</sub>	Citric acid complexation, impregnation	TPR, fixed bed	CoWO <sub>4</sub> /Ni = 10 wt% Ni, CoWO <sub>4</sub> /La <sub>2</sub> O <sub>3</sub> = 10 wt% La	64																												
				Fe <sub>2</sub> O <sub>3</sub> /LaNiO <sub>3</sub> = 15/85 wt%, Fe <sub>2</sub> O <sub>3</sub> /CeO <sub>2</sub> /LaNiO <sub>3</sub> = 15/5/80 wt%	64																												
Fe	CeO <sub>2</sub> , Ca <sub>0.8</sub> Sr <sub>0.2</sub> Ni <sub>0.2</sub> O <sub>3</sub> , MgAl <sub>2</sub> O <sub>4</sub>	Citric acid, co-precipitation, solid state reaction	TGA	Fe <sub>2</sub> O <sub>3</sub> /CeO <sub>2</sub> = 30/70 mol% solid state reaction, Fe <sub>2</sub> O <sub>3</sub> /MgAl <sub>2</sub> O <sub>4</sub> = 30/70 wt% solid state reaction, Fe <sub>2</sub> O <sub>3</sub> /CeO <sub>2</sub> = 30/70 mol% co-precipitation, Fe <sub>2</sub> O <sub>3</sub> /CeO <sub>2</sub> = 40/60 mol% co-precipitation, Fe <sub>2</sub> O <sub>3</sub> /Ca <sub>0.8</sub> Sr <sub>0.2</sub> Ti <sub>0.8</sub> Ni <sub>0.2</sub> O <sub>3</sub> = 30/70 mol% citric acid	65																												
Fe	LaFeO <sub>3</sub>	3DOM-LaFeO <sub>3</sub> : polystyrene colloidal crystal templating method, nano-LaFeO <sub>3</sub> : combustion method	TPR, fixed bed	3DOM-LaFeO <sub>3</sub> , nano-LaFeO <sub>3</sub>	66																												
Fe		LaFe <sub>1-x</sub> Co <sub>x</sub> O <sub>3</sub> ; x = 0.1, 0.3, 0.5, 0.7, 1.0	TPR, fixed bed	LaFe <sub>1-x</sub> Co <sub>x</sub> O <sub>3</sub> ; x = 0.1, 0.3, 0.5, 0.7, 1.0	67																												



Table 1 (Contd.)

Active component	Additives/support	Preparation method	Experimental device	Composition	Reference
Fe	LaFeO <sub>3</sub> , Co doping La <sub>0.98</sub> Str <sub>0.02</sub> FeO <sub>3-δ</sub> (LSF)	Combustion method	Fixed bed	Fe <sub>2</sub> O <sub>3</sub> /LSF = 25/40 wt%	71
Fe	CoFe <sub>2</sub> O <sub>4</sub> spinel	Co-precipitation	Fixed bed	CoFe <sub>2</sub> O <sub>4</sub>	77
Fe	CoFe <sub>2</sub> O <sub>4</sub> spinel	Co-precipitation	Fixed bed	CoFe <sub>2</sub> O <sub>4</sub>	78
Fe	Spinel type, NiFe <sub>2</sub> O <sub>4</sub> , Fe <sub>3</sub> O <sub>4</sub>	Co-precipitation	TPR, fixed bed	NiFe <sub>2</sub> O <sub>4</sub> , Fe <sub>3</sub> O <sub>4</sub>	79
Fe	Co, and Mn	Co-precipitation	Fixed bed, TPR	M <sub>1.6-x</sub> M <sub>x</sub> Fe <sub>2.4</sub> O <sub>y</sub> , Co : Mn = 1 : 0, 0.9 : 0.1, 0.5 : 0.5 and 0 : 1	80
Fe	Spinel, NiFeAlO <sub>4</sub>	Mechanical mixing	TGA fixed bed	NiFeAlO <sub>4</sub>	81
Fe	BHA, barium hexaaluminate	One-pot synthesis	Fixed bed	40 wt% Fe-nanoparticles = 16, 22, 54 nm	82
Fe	MgAl <sub>2</sub> O <sub>4</sub>	Freeze-granulation	Fluidized bed	Fe <sub>2</sub> O <sub>3</sub> /MgAl <sub>2</sub> O <sub>4</sub> = 60/40 wt%	84
Fe	Al <sub>2</sub> O <sub>3</sub>	Co-precipitation	Fluidized bed	Fe <sub>2</sub> O <sub>3</sub> /Al <sub>2</sub> O <sub>3</sub> = 66.7/33.3 wt%	95
Fe	Al <sub>2</sub> O <sub>3</sub>	Mechanical mixing	Fixed bed	Fe <sub>2</sub> O <sub>3</sub> /Al <sub>2</sub> O <sub>3</sub> = 60/40 wt%	97
Fe	ZrO <sub>2</sub>	Co-precipitation	Fluidized bed	Fe <sub>2</sub> O <sub>3</sub> /ZrO <sub>2</sub> = 20/80 wt%	98
Fe	CeO <sub>2</sub> /Al <sub>2</sub> O <sub>3</sub>	Co-precipitation	Fixed bed	Fe <sub>2</sub> O <sub>3</sub> /CeO <sub>2</sub> /Al <sub>2</sub> O <sub>3</sub> = 98/0.25/1.75 wt%	101 and 102
Fe	Co-precipitation	Co-precipitation	TPR fixed bed	Fe <sub>3</sub> O <sub>4</sub>	103
Fe	Hematite: thermal treatment, magnetite: thermal decomposition, goethite: co-precipitation	Hematite: thermal treatment, magnetite: thermal decomposition, goethite: co-precipitation	TPR	Goethite (γ-FeOOH), hematite (α-Fe <sub>2</sub> O <sub>3</sub> ), magnetite (Fe <sub>3</sub> O <sub>4</sub> )	104
Fe	CeO <sub>2</sub> /Al <sub>2</sub> O <sub>3</sub> , NiO/NiAl <sub>2</sub> O <sub>4</sub>	Co-precipitation, mechanical mixing	Fixed bed	Fe <sub>2</sub> O <sub>3</sub> /CeO <sub>2</sub> /Al <sub>2</sub> O <sub>3</sub> = 98/0.25/1.75 wt%, NiO/NiAl <sub>2</sub> O <sub>4</sub> = 10 wt% excess, mixture = 67.5 wt% OC, 7.5 wt% catalyst, 25 wt% silica	105–107
Fe	CeO <sub>2</sub> /Al <sub>2</sub> O <sub>3</sub>	Co-precipitation	Fixed bed	Fe <sub>2</sub> O <sub>3</sub> /CeO <sub>2</sub> /Al <sub>2</sub> O <sub>3</sub> = 90.17/4.79/5.04 wt%	108
Fe	Fe <sub>2</sub> O <sub>3</sub> /Al <sub>2</sub> O <sub>3</sub>	Mechanical mixing	Fixed bed	Fe <sub>2</sub> O <sub>3</sub> /Al <sub>2</sub> O <sub>3</sub> = 90/10 wt%	109 and 110
Fe	Al <sub>2</sub> O <sub>3</sub>	Mechanical mixing	Fixed bed	Fe <sub>2</sub> O <sub>3</sub> /Al <sub>2</sub> O <sub>3</sub> = 59.4/40.3 wt%	134
Fe	Al <sub>2</sub> O <sub>3</sub>	Mechanical mixing	TGA	Fe <sub>2</sub> O <sub>3</sub> /Al <sub>2</sub> O <sub>3</sub> = 50/50 wt%	136
Fe	ZrO <sub>2</sub>	Co-precipitation	TGA	Fe <sub>2</sub> O <sub>3</sub> /ZrO <sub>2</sub> = 20/80 wt%	138
Fe	Al <sub>2</sub> O <sub>3</sub>	Impregnation method	TGA	Fe <sub>2</sub> O <sub>3</sub> /Al <sub>2</sub> O <sub>3</sub> = 45/55 wt%, Fe <sub>2</sub> O <sub>3</sub> /Al <sub>2</sub> O <sub>3</sub> = 25/75 wt%	142 and 143
Fe, Fe-Cu	Al <sub>2</sub> O <sub>3</sub> , TiO <sub>2</sub>	Mechanical mixing	TGA	Fe <sub>2</sub> O <sub>3</sub> /CuO/Al <sub>2</sub> O <sub>3</sub> = 60/20/20 wt%, Fe <sub>2</sub> O <sub>3</sub> /TiO <sub>2</sub> = 80/20 wt%	144
Fe	CeO <sub>2</sub> /Al <sub>2</sub> O <sub>3</sub>	Citrate method	Fixed bed	Fe <sub>2</sub> O <sub>3</sub> /CeO <sub>2</sub> /Al <sub>2</sub> O <sub>3</sub> = 98/0.5/1.5 wt%	147

particles but the pore structure was maintained and no chemical interaction between the active redox component and the support, which leads to the unreactive component Fe–Zr–O, was observed. Cho *et al.*<sup>35</sup> analysed the structural changes of an oxygen carrier with a 20 wt% loading of Fe<sub>2</sub>O<sub>3</sub> on ZrO<sub>2</sub> at 750 °C. After 10 redox cycles the iron particles were homogeneously distributed with only slight signs of sintering as long as a deep reduction was avoided and reduction was stopped at wustite. However, after deep reductions to Fe, diffusion of iron particles to the outer regions and significant growth of iron grains were found. Therefore, samples reduced to wustite performed better in terms of hydrogen yield. The oxygen carriers were successfully applied in a fixed bed reactor at 800 °C using methane as the reducing feed.<sup>36</sup> The redox reaction results were used to determine the design guideline for a 1 MW<sub>th</sub> chemical looping plant with a bed inventory of 170–220 kg MW<sub>th</sub><sup>-1</sup>. The stability was further enhanced by using an yttrium-stabilized zirconia (YSZ) defect support, which provides strong support and inhibits iron sintering.<sup>35</sup> The iron diffusion, which leads to uneven distribution of active metal and support, can be avoided by preventing particle sintering. It was found that the stability was increased by an activation process consisting of 10 cycles of reduction including air-oxidations. The formation of large cracks and pores occurred, which directed the iron diffusion towards the newly formed cavities. Thus, a heterogeneous particle distribution was maintained. Samples, which underwent an activation method, provided higher reduction rates using methane as feed compared to reference samples without the prior treatment. Kosaka *et al.*<sup>37</sup> reported similar improvements of redox behaviour with iron oxide supported by yttrium-stabilized zirconia (YSZ) compared to Fe<sub>2</sub>O<sub>3</sub>/ZrO<sub>2</sub> reference samples. A positive correlation between the reduction rate constant and the oxide ion conductivity of YSZ due to Y-doping was observed in the reduction step and the oxidation reaction. Improved reaction rates compared to ZrO<sub>2</sub> were reported, when using samples with CeO<sub>2</sub> and Ce<sub>0.8</sub>Gd<sub>0.2</sub>O<sub>2-δ</sub> (GDC) support. In contrast to the enhanced ion conductivity of YSZ, CeO<sub>2</sub> and GDC increase the electronic conductivity and accelerate the reduction kinetics of FeO to Fe significantly (Fig. 2).<sup>33</sup> This suggests that oxygen vacancy formation by cerium plays an important role in the acceleration of the redox reaction.

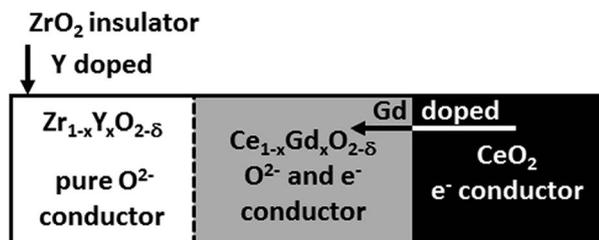


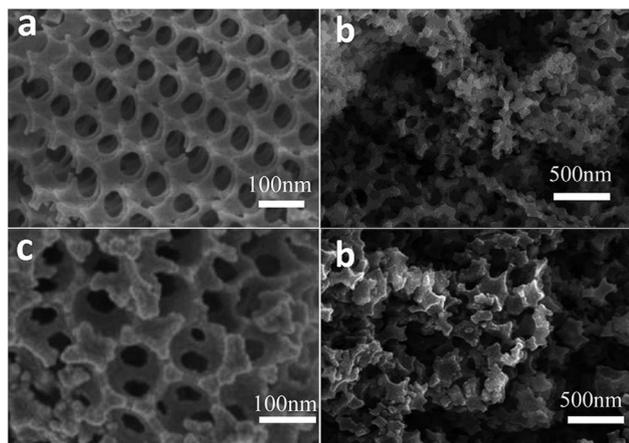
Fig. 2 Physical properties of the support materials utilized by Kosaka *et al.*<sup>37</sup> Reprinted and adapted from F. Kosaka, H. Hatano, Y. Oshima, and J. Otomo, Iron oxide redox reaction with oxide ion conducting supports for hydrogen production and storage systems, *Chem. Eng. Sci.*, 2015, 123, 380–387, Copyright (2014), with permission from Elsevier.

The GDC support, which contained both attributes, the ion- and electric conductivity, possessed the highest performance in redox experiments at 550–800 °C with hydrogen as reducing feed followed by a steam oxidation.

### CeO<sub>2</sub> as active component

The stabilizing effects of a ZrO<sub>2</sub> support with CeO<sub>2</sub> was investigated by Zheng *et al.*<sup>38</sup> Samples with a metal ratio of Ce/Zr = 8/2 and with pore sizes ranging from 100–200 nm were prepared using the colloidal crystal method. The pore sizes were controlled by the sizes of the PMMA template spheres. A nonporous sample prepared by co-precipitation was used as a reference sample. H<sub>2</sub>-TPR measurements revealed a single reduction peak (at around 650 °C) for the macro-porous samples, whereas the nonporous oxygen carrier showed two separated reduction peaks, which can be related firstly to the exchange of bulk oxygen (600 °C) and secondly to the lattice oxygen (900 °C). The merging of the two reduction peaks suggested an improved mobility of lattice oxygen in the macro porous samples, which enhanced its reducibility. Cyclic experiments were performed in a fixed bed quartz reactor at 800 °C using methane as reducing agent. The macro porous samples indicated an average methane conversion ranging from 46–50% and a H<sub>2</sub>/CO ratio approaching the stoichiometric value. The hydrogen consumption increased with declining pore sizes. The methane conversion of the nonporous sample was 4%-points lower with a H<sub>2</sub>/CO ratio of two, which suggested methane cracking induced by a deficiency of oxygen, and thus an increased risk of carbon formation. A similar trend was observed in steam oxidations. A declining pore size raised the hydrogen production rate and the amount of hydrogen. Stability measurements over 10 cycles revealed a clearly visible continuous decrease of produced hydrogen. SEM images showed that the three-dimensionally ordered macro-porous structure had collapsed, however the samples maintained a sufficient porosity. The influence of different calcination temperatures on redox performance and structural traits of the proposed macro-porous CeO<sub>2</sub>–ZrO<sub>2</sub> oxygen carrier samples were studied.<sup>39</sup> Increasing calcination temperatures from 450 to 850 °C changed the structure due to the growth of larger grains. Lower calcination temperatures resulted in higher structured samples (Fig. 3), which yielded better CH<sub>4</sub> conversions and H<sub>2</sub>/CO ratios. A similar trend for the steam oxidation was observed, a decreasing hydrogen production rate with increasing calcination temperature. The results suggest that the structural properties of CeZrO<sub>2</sub> strongly influenced the mobility of lattice oxygen, thus effecting conversion efficiency during the redox reactions. All samples revealed a loss of produced amount of hydrogen after 10 consecutive cycles. However, a high calcination temperature led to a complete loss of the initial macro-structure but preserved the microscale porosity (Fig. 3). Lower calcination temperatures stabilized the three dimensional structures, even if the samples were used at high reaction temperatures. Nevertheless, stability experiments indicated a gradual loss of the skeletal structure, which left the long term cycle stability disputable.





**Fig. 3** SEM images of  $\text{CeO}_2/\text{ZrO}_2$  oxygen carriers. Fresh samples after a calcination step at different temperatures (a) 450 °C, (c) 750 °C. After 10 redox cycles, the influence of different calcination temperatures (b) 450 °C, (d) 750 °C on the structural changes can be observed.<sup>39</sup> Reprinted and adapted from Y. Zheng, Y. Wei, K. Li, X. Zhu, H. Wang, and Y. Wang, Chemical-looping steam methane reforming over macroporous  $\text{CeO}_2\text{-ZrO}_2$  solid solution: effect of calcination temperature, *Int. J. Hydrogen Energy*, 2014, **39**, 13361–13368, Copyright (2014), with permission from Elsevier.

### $\text{CeO}_2$ as additive

Synergetic effects of a mixture of  $\text{CeO}_2$  and  $\text{Fe}_2\text{O}_3$  were investigated by Zhu *et al.*<sup>31,40</sup>  $\text{CeO}_2$  was used as support material with iron oxide dispersed on the surface.  $\text{Fe}^{3+}$  was partially imbedded in the  $\text{CeO}_2$  lattice structure, thus changing the chemical properties of the particles. An enhanced reducibility of the mixed Ce–Fe oxide was observed in  $\text{H}_2$ -TPR measurements and CL-SMR at temperatures of 800–900 °C with a higher lattice oxygen mobility compared to single oxides. An increasing iron content raised the amount of released hydrogen in the oxidation step due to the high oxygen intake capability. The sample with the molar ratio of  $\text{Ce}/\text{Fe} = 1$  was selected for 10 consecutive cycles with fixed reduction and oxidation reaction times at 850 °C. A steady decrease of the  $\text{CH}_4$  conversion from roughly 63% to 55% was observed, while the amount of produced syngas and the  $\text{H}_2/\text{CO}$  ratio remained unchanged. The oxidations showed only a slight decline in the produced hydrogen with no carbonaceous impurities. During the methane reduction step, the formation of  $\text{CeFeO}_3$  occurred by means of a solid–solid reaction, which was re-oxidized to  $\text{CeO}_2$  and  $\text{Fe}_3\text{O}_4$  during the steam oxidation step. However, a Ce–Fe–O phase equilibrium was preserved with constant composition during the redox process, thus increasing the oxygen mobility by the formation of abundant oxygen vacancies. Gu *et al.*<sup>41</sup> reported a significant improved reducibility of surface and bulk oxygen in iron oxide supported with  $\text{CeO}_2$  prepared using the co-precipitation method. Samples produced by mechanical mixing exhibited only a slightly better performance. Thus the formation of a solid solution of  $\text{CeFeO}_3$  and the structural change, which were strongly related to the synthesis method, were key to enhancing the redox performance. The increasing Ce content increased its specific surface which reached

a maximum at the composition of  $\text{Fe}/\text{Ce} = 80/20$  and decreased afterwards. Small Ce-particles with a high dispersity and strong chemical interaction were formed and stabilized the structure after several redox cycles. The improved stability of iron oxide particles by adding Ce was also reported by Romero *et al.*<sup>42</sup> Small amounts of  $\text{CeO}_2$  were added to iron oxide supported on  $\text{Al}_2\text{O}_3$  with the final composition of  $\text{Fe}_2\text{O}_3/\text{Al}_2\text{O}_3/\text{CeO}_2$  98/1.75/0.25%. The sample did not suffer from any deactivation during 17 cycles at 500 °C when fully oxidized, but the cycle time slightly increased.

### $\text{MoO}_3$ as additive

When replacing  $\text{CeO}_2$  with  $\text{MoO}_3$  with a content of 0.25%, a better steam oxidation behaviour was observed, however a reduced cycle stability, especially within the first 8 cycles led, to a decreasing amount of produced hydrogen.<sup>42</sup> The combined addition of  $\text{CeO}_2$  and  $\text{MoO}_3$  did not improve either the redox behaviour or stability. Datta *et al.*<sup>43,44</sup> impregnated iron oxide samples supported on 20 wt%  $\text{Ce}_{0.5}\text{Zr}_{0.5}\text{O}_2$  with different molybdenum concentrations to achieve a final Mo loading of 0.96–5.5 wt%. The formation of a spinel-type ferrite  $\text{Mo}_x\text{Fe}_{3-x}\text{O}_4$  was reported, which inhibited the contact between the iron particles and impeded the deep reduction from  $\text{FeO}$  to  $\text{Fe}$ . This hindered sintering and deactivation of the active component. The increasing Mo content led to a higher temperature for the reduction with hydrogen and a higher air-oxidation temperature. However, decreasing reduction activation energy with increasing molybdenum content was observed. A significant improvement in stability was reported for Mo-impregnated samples compared to a reference  $\text{Fe}_2\text{O}_3\text{-Ce}_{0.5}\text{Zr}_{0.5}\text{O}_2$  sample. A Mo content of  $\geq 1.5$  wt% preserved a hydrogen production amount of at least 90% from the theoretical value after 100 cycles at 750 °C. A further extended stability test with 200 cycles revealed a loss of 10% from the initial amount of produced hydrogen within the first 40 cycles followed by a stable production afterwards. The trade-off for higher stability was the amount of produced hydrogen per cycle, which was only 30–34% of the amount theoretically possible. Wang *et al.*<sup>45</sup> reported a decrease in the activation energy by adding small amounts of Mo <10 mol%. TPR experiments revealed a shift of the reduction peak of iron towards lower temperatures from 610 °C to 572 °C. A third broad peak at 758 °C was visible, which was related to the reduction of  $\text{Mo}^0$  and to the formation of  $\text{Fe}_2\text{Mo}_3\text{O}_8$ . An increased production rate and higher amount of hydrogen during steam oxidation were measured. A shift to lower oxidation temperatures was observed from 400 °C to 350 °C, with the sample consisting of 8 mol% Mo revealing the best performance.

### $\text{CaO}$ as additive

Improved stability of iron based oxygen carriers by the combination with  $\text{CaO}$  was investigated by some work groups.<sup>46–48</sup> Datta<sup>46</sup> added 2.5 wt%  $\text{CaO}$  to  $\text{Fe}_2\text{O}_3$  samples supported with either  $\text{SiO}_2$  (7.5 wt%) or  $\text{Al}_2\text{O}_3$  (5 wt%). The combination of  $\text{SiO}_2$  and  $\text{CaO}$  enabled a stable hydrogen yield of 95% of its initial value over 100 cycles as long as the reaction temperature



remained below 750 °C. A significant stability loss was observed at higher temperatures caused by the agglomeration of SiO<sub>2</sub>. The sample with Al<sub>2</sub>O<sub>3</sub> and CaO stabilized the oxygen carrier to a hydrogen amount of 70–80% of its initial value after 100 cycles. Datta suggested that the formation of CaO formed a dispersed ferrite structure, which inhibited sintering. Ismail *et al.*<sup>47</sup> prepared mixtures of Fe<sub>2</sub>O<sub>3</sub> with 50, 57.3 and 66.7 mol% of CaO. After calcination CaO, CaFe<sub>2</sub>O<sub>4</sub>, Ca<sub>2</sub>Fe<sub>2</sub>O<sub>5</sub>, CaO and Fe<sub>2</sub>O<sub>3</sub> were identified in the different samples depending on the mol% of CaO. Deviations from the expected equilibrium compositions were detected as a result of inhomogeneous mixing and slow solid-state diffusion during the calcination process. The calcium ferrite samples were successfully reduced with CO and H<sub>2</sub>, forming the stable component Ca<sub>2</sub>Fe<sub>2</sub>O<sub>5</sub>, and efficiently re-oxidized with CO<sub>2</sub>. Equilibrium partial pressure values suggest that the oxidation of Ca<sub>2</sub>Fe<sub>2</sub>O<sub>5</sub> enables a higher steam conversion to hydrogen compared to the unmodified iron to wustite oxidation. The mixed metal ferrites achieved a stable gas conversion during consecutive redox reactions over 20 cycles, whereas the pure iron sample, which started with a high amount of exchanged-oxygen/sample-mass, indicated a loss of approximately 90% within 10 cycles. Deep reduction experiments resulted in the formation of segregated phases of Fe and CaO, hence a significant degradation of the cycle stability. Fluidized redox experiments indicated high attrition losses of the calcium ferrite samples of 4–6% wt. Thus, the mechanical strength needs to be improved for industrial applications. Chan *et al.*<sup>48</sup> observed high steam conversion efficiencies with an equilibrium composition of 75/25 mol% H<sub>2</sub>/H<sub>2</sub>O in a single oxidation step using a Ca<sub>2</sub>Fe<sub>2</sub>O<sub>5</sub> sample. A complete recovery to the initial stage of Ca<sub>2</sub>Fe<sub>2</sub>O<sub>5</sub> was achieved without the requirement of an air oxidation. The amount of released hydrogen was 88.9% compared to a Fe<sub>2</sub>O<sub>3</sub>/ZrO<sub>2</sub> reference sample. However, Ca<sub>2</sub>Fe<sub>2</sub>O<sub>5</sub> suffered from a slow reduction rate, which was about twice as long as pure iron oxide caused by an unfavourable reduction equilibrium. To overcome the trade-off and utilize the higher steam oxidation efficiency a combination of two oxygen carriers in two reactors was suggested. During reduction the feed initially reduces the Ca<sub>2</sub>Fe<sub>2</sub>O<sub>5</sub> containing bed. The off-gas can still convert a pure iron oxide bed and enables complete conversions to CO<sub>2</sub> for efficient carbon sequestration. During oxidation the flow direction changes and passes first through the iron bed and finally through the CaO + Fe bed. Thus, the higher steam conversion capabilities are utilized.

### Al<sub>2</sub>O<sub>3</sub> as additive

It is well known that the addition of Al<sub>2</sub>O<sub>3</sub> can stabilize the cycle stability of iron based oxygen carriers. However sample studies of deeply reduced iron oxide supported by Al<sub>2</sub>O<sub>3</sub> have shown that the formation of unreactive hercynite FeAl<sub>2</sub>O<sub>4</sub> occurred.<sup>49–52</sup> Liu *et al.*<sup>53</sup> investigated possibilities for inhibiting the formation of hercynite and preserving the initial reduction rates by adding Na<sub>2</sub>O and MgO. These additives were expected to prevent the formation of FeAl<sub>2</sub>O<sub>4</sub> by forming the stable compounds MgAl<sub>2</sub>O<sub>4</sub> and NaAlO<sub>2</sub> instead, which also act as support materials. The two oxides were prepared by precipitation with the

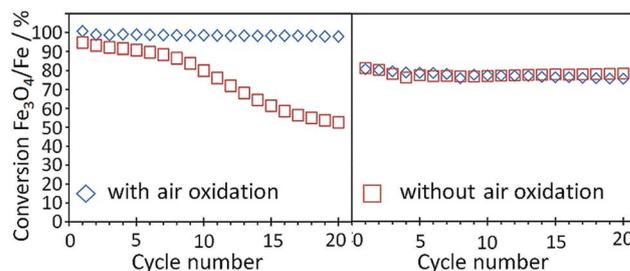


Fig. 4 The conversion degree of Fe<sub>3</sub>O<sub>4</sub> to Fe as a function of cycle numbers with 0.3 g of NAFO (left) and NMAFO (right) in a packed bed reactor at 850 °C with or without an air oxidation step.<sup>53</sup> Reproduced from W. Liu, M. Ismail, M. T. Dunstan, W. Hu, Z. Zhang, P. S. Fennell, S. A. Scott, and J. S. Dennis, Inhibiting the interaction between FeO and Al<sub>2</sub>O<sub>3</sub> during chemical looping production of hydrogen, *RSC Adv.*, 2015, 5, 1759–1771, Copyright (2015), with permission from the Royal Society of Chemistry.

compounds Fe, Al and Na (NAFO) and Fe, Mg, Al and Na (NMAFO). When cycled under conditions suitable for producing hydrogen at 1123 K, both oxygen carriers showed stable performance over 20 cycles, demonstrating approximately 100% and 80% of their theoretical hydrogen capacities, as long as an air oxidation was included in each step. Cycle experiments without air oxidation revealed a conversion loss of roughly 50% for NAFO, while the NMAFO sample retained its stability regardless of the oxidation procedure. The gradual formation of NaFeO<sub>2</sub> led to free Al<sub>2</sub>O<sub>3</sub> and the formation of hercynite in consecutive cycles. However, in the presence of Mg<sup>2+</sup> cations, Al<sub>2</sub>O<sub>3</sub> formed the stable compound of MgAl<sub>2</sub>O<sub>4</sub> thus stabilized the NMAFO sample (Fig. 4).

### Natural minerals as oxygen carrier

The application of oxygen carriers from natural deposits in chemical looping has attracted the attention of several research groups in the past.<sup>54–56</sup> Ku *et al.*<sup>57</sup> considered the utilization of natural-ilmenite (Fe<sub>2</sub>TiO<sub>5</sub>) as an inexpensive and abundant oxygen carrier. In a segmented reduction the sample was gradually reduced to Fe<sub>2</sub>TiO<sub>4</sub> and FeTiO<sub>3</sub> and finally Fe and TiO<sub>2</sub>. The consecutive air oxidation led to the formation of a sample containing only Fe<sub>2</sub>O<sub>3</sub> and TiO<sub>2</sub>. Hydrogen generation by steam oxidation was feasible at 900 °C and Fe<sub>3</sub>O<sub>4</sub> and FeTiO<sub>3</sub> were found in the oxidized sample. The stability tests showed no significant changes at 900 °C after 100 cycles with an oxygen storage capacity of approximately 33%. Sample analysis revealed agglomerated structures of Fe<sub>2</sub>O<sub>3</sub> but due to the presence of cracks and pores, the reactivity was maintained. Xiao *et al.*<sup>58</sup> compared ilmenite with different natural iron ores in a reducing feed using different CO/N<sub>2</sub> ratios. The ilmenite samples had the lowest weight loss during the reduction because of the lower iron oxide content, however a very high resistance to solid carbon and iron carbon formation was observed. Tong *et al.*<sup>59</sup> suggested that the introduction of Fe<sub>2</sub>O<sub>3</sub> in a TiO<sub>2</sub> network leads to the formation of lattice substitution and interstitial defects in the support framework. These defects increase the rate of oxygen anion diffusion. In unsupported iron



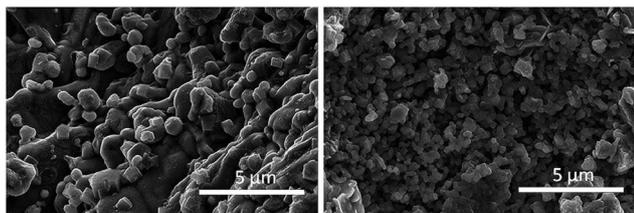


Fig. 5 SEM images of the oxygen carrier after 10 cycles, original iron ore (left) and iron ore loaded with 10 wt%  $\text{KNO}_3$  (right).<sup>60</sup> Reprinted and adapted with permission from W. Liu, L. Shen, H. Gu, and L. Wu, Chemical looping hydrogen generation using potassium-modified iron ore as an oxygen carrier, *Energy Fuels*, 2016, 30, 1756–1763, Copyright (2016), American Chemical Society.

oxides the outward iron cation diffusion was dominant, whereas in a supported iron oxide, like ilmenite, both the outwards diffusion of iron cations and the inwards diffusion of oxygen anions occurred. Liu *et al.*<sup>60</sup> impregnated natural hematite, which consisted of  $\text{Fe}_2\text{O}_3$  supported with  $\text{SiO}_2$  and  $\text{Al}_2\text{O}_3$ , with a solution of  $\text{KNO}_3$ , to achieve a loading of 10 wt%  $\text{KNO}_3$ . After calcination,  $\text{K}_2\text{Fe}_4\text{O}_7$  and  $\text{K}_2\text{Fe}_{22}\text{O}_{34}$  were found in the samples, which showed signs of sintering and displayed a lower degree of porosity than the unmodified ore. Redox experiments were performed in a fluidized bed reactor at temperatures ranging from 750–900 °C with CO as feed followed by steam oxidations. A loading of 10 wt%  $\text{KNO}_3$  significantly accelerated the reduction reaction and enabled a greater reduction degree of the iron oxide. Thus, higher amounts of hydrogen were obtained in the consecutive oxidations. Additionally by reducing the reduction time and increasing carbon conversion efficiency less coking occurred. The cycle stability was improved efficiently without the loss of produced hydrogen by maintaining a porous structure, whereas the unmodified ore lost 24% of its initial production capacity (Fig. 5).

### Bimetallic oxygen carrier

Chiron *et al.*<sup>61</sup> proposed the use of a bimetallic oxygen carrier to enable a synergetic effect of different redox behaviour of two metal oxides CuO and  $\text{Fe}_2\text{O}_3$ . In contrast to other metal oxides CuO is reduced by methane by releasing heat. Combining CuO with endothermic reacting metal oxides can reduce the external heat demand and lower the heat transfer requirement to the fuel reactor. Samples supported by  $\text{Al}_2\text{O}_3$  consisting of either pure  $\text{Fe}_2\text{O}_3$ , pure CuO or a bimetal with varying compositions were prepared by an incipient wetness impregnation (IWI) and co-precipitation (CoPr) method. Most samples prepared by (CoPr) could not be fluidized since large slugs were observed that filled the entire cross-section of the reactor and an unsteady pressure drop occurred. Samples prepared by IWI were easily fluidized and reached their maximum solid conversion by steam oxidations significantly faster than samples prepared by CoPr. The bimetallic samples revealed a low solid conversion of approximately 10% irrespective of the preparation method. The highest conversion of 27% was measured with pure iron prepared by co-precipitation.

### $\text{CoWO}_4$ as oxygen carrier

$\text{CoWO}_4$  has been reported as a suitable oxygen carrier for hydrogen production with methane as the reducing feed.<sup>62</sup> De los Ríos Castillo *et al.*<sup>63</sup> impregnated cobalt tungstate  $\text{CoWO}_4$  with Ni and  $\text{La}_2\text{O}_3$  to enhance its reactivity towards methane reduction. Reduction experiments at 850 °C with  $\text{CH}_4$  indicated improved reaction rates up to 6 and 2.5 times by the addition of 10 wt% Ni and 10 wt% La respectively. However, the reactivity improvements diminished gradually at increasing reduction temperatures. In the second cycle the unmodified  $\text{CoWO}_4$  sample displayed higher reduction rates compared to the first cycle. This was related to the incomplete re-oxidation of Co by steam and the catalytic activity of the metal species. Nevertheless, samples promoted with Ni and La had improved reactivity but with smaller margins. Similar trends were reported regarding the activation energies. The inclusion of Ni and La significantly reduced the activation energy and thus improved reducibility. Again, the difference between the promoted samples and the reference sample disappeared after the first cycle.

### Perovskites

Perovskite structured oxygen carriers are reported to have a good thermal stability and a high ionic and electronic conductivity.<sup>64,65</sup> Zhao *et al.*<sup>66</sup> studied three-dimensionally ordered macro-porous (3DOM)  $\text{LaFeO}_3$  and nano- $\text{LaFeO}_3$  perovskite oxides synthesis gas and pure hydrogen using  $\text{CH}_4$  as fuel. 3DOM- $\text{LaFeO}_3$  enabled a stable partial oxidation of methane at 850 °C in a fixed bed reactor over 10 cycles. An average methane conversion of 50% and the production of hydrogen by steam oxidation with small amounts of  $\text{CO}_2$  was achieved. The amount of released hydrogen decreased within the first 3 cycles but remained stable afterwards. The nano- $\text{LaFeO}_3$  enabled both a very high methane conversion >90% and a high molar ratio  $\text{H}_2/\text{CO}$  of 9 in the first cycles, which meant a high risk of carbon formation. Within 10 cycles the methane conversion decreased and ended up below the 3DOM- $\text{LaFeO}_3$  sample. The three-dimensional skeletal structure in the fresh sample gradually collapsed within repeated cycles caused by thermal and chemical stress. This led to a decrease in crystallinity and grain sizes, nevertheless a porous structure with little agglomeration was maintained, which preserved a stable cyclability. Zhao *et al.*<sup>67</sup> further enhanced the performance of  $\text{LaFeO}_3$  perovskite oxides by doping with Co, which is expected to increase the vacancies sites, thus improving the redox activity.<sup>68–70</sup> Perovskite-type oxides  $\text{LaFe}_{1-x}\text{Co}_x\text{O}_3$  maintained their crystalline structure despite the doping with Co. An increasing Co content resulted in a higher concentration of oxygen vacancies, in a greater resistance to reduction and in a longer period of time required to reach a stable fuel conversion of >90%. Nevertheless, all sample compositions were completely reducible, indicated by the appearance of  $\text{La}_2\text{O}_3$ ,  $\text{Fe}_2\text{O}_3$ , FeO, CoO and Co. A Co molar ratio of 0.3 and 0.5 yielded the best performance regarding low risk of carbon formation, amount of produced hydrogen and stable  $\text{CH}_4$  conversion. Hence,  $\text{LaFe}_{0.7}\text{Co}_{0.3}\text{O}_3$  was used for consecutive redox cycling



experiments. A steady feed conversion of 86% was achieved over 20 cycles with slight improvements within the first 5 cycles. A hydrogen production potential of  $4 \text{ mmol g}^{-1}$  was obtained with an average purity of 96%. Post experimental analysis revealed degradation caused by sintering but these degradations were compensated by an increased oxygen vacancy concentration. Liang<sup>64</sup> investigated the effects of  $\text{CeO}_2$  on perovskite supported iron oxides ( $\text{Fe}_2\text{O}_3$  15 wt%,  $\text{CeO}_2$  5 wt%,  $\text{LaNiO}_3$ ). The addition of cerium oxide slightly reduced the pore volume and pore diameter by partially filling void zones with  $\text{CeO}_2$ . TPR experiments indicated a shift of the 3 iron oxide reduction peaks to lower temperatures, which suggests the promotion of the reduction reaction. In addition, the appearance of a fourth peak at  $520^\circ\text{C}$  was observed, which corresponded to the reduction of  $\text{CeO}_2$ . Redox experiments were executed in a fixed bed reactor at  $800^\circ\text{C}$ , consisting of a reduction with  $\text{CH}_4$ , a steam oxidation and a final air oxidation. The reference sample without  $\text{CeO}_2$  exhibited a complete methane conversion and a stable hydrogen production. After 57 cycles, a sharp drop in the hydrogen yield was measured, which was a result of the decomposition of the perovskite structure, leading to isolated  $\text{Fe}_3\text{O}_4$  and  $\text{FeO}$ . The sample enhanced by  $\text{CeO}_2$  did not suffer from deactivation over 100 redox cycles, with a stable hydrogen yield slightly lower than the reference sample. The formation of a solid solution of  $\text{CeFeO}_3$  was observed, which increases the oxygen content and improves stability and activity.

He *et al.*<sup>71</sup> used the perovskite  $\text{La}_{0.8}\text{Sr}_{0.2}\text{FeO}_{3-\delta}$  (LSF) as a supplement to iron oxide for a higher steam conversion efficiency during the steam oxidation step.  $\text{La}_{0.8}\text{Sr}_{0.2}\text{FeO}_{3-\delta}$  is harder to reduce but offers a high resistance to carbon deposition. A reactor with a two-layer filling was proposed, consisting of an iron oxide rich phase at the bottom with small amounts of LSF to prevent loss of stability by sintering and a LSF layer at the top (Fig. 6 right). Reduction reactions were executed with  $\text{CH}_4$  supplied from the top, which yielded an average methane conversion of 99% and a syngas production efficiency of 62%. After the reduction the LSF layer consisted of  $\text{Fe}$ ,  $\text{La}_2\text{O}$  and small amounts of  $\text{LaSrFeO}_{4-\delta}$ . The oxidations were performed by

supplying steam from the bottom, which initially oxidized the iron rich layer to iron oxide. The gas mixture, which consisted of 62.3%  $\text{H}_2$  and 37.7% steam, was able to oxidize the upper layer further by forming  $\text{La}_{0.8}\text{Sr}_{0.2}\text{FeO}_{3-\delta}$  resulting in a total steam conversion of 77.2% and a hydrogen purity of 98.5% (Fig. 6 left). The higher steam conversion efficiencies are expected to significantly increase the process efficiency. The material characteristics with a high oxygen defect concentration of the LSF are responsible for the high affinity towards steam oxidation. The total amount of exchanged oxygen of the LSF was little compared to the iron rich layer, but the favourable thermodynamic equilibrium enabled a higher steam conversion efficiency. Galinsky *et al.*<sup>33</sup> reported significant reduction rate improvements by supporting iron oxide on LSF. Experiments were performed with  $\text{H}_2$ ,  $\text{CO}$  and  $\text{CH}_4$  in a TGA system. A faster reduction rate resulting in a higher degree of solid conversion and a high reactivity at lower temperatures were reported. Two material properties were attributed to the performance increases. Firstly, the electronic and ionic mixed conductivity enabled efficient transport of  $\text{O}^{2-}$ -ions in the solid material and a full accessibility of bulk oxygen. Secondly,  $\text{Fe}_2\text{O}_3$  crystallites with small grain sizes of  $100 \mu\text{m}$  were distributed evenly within the LSF support. Compared to  $\text{TiO}_2$  supported samples, a reduction rate acceleration of one magnitude was observed. The LSF samples indicated very high cycle stability with a capacity loss of 4% within 50 cycles.

### Oxygen carrier with spinel structures

Spinel structured ferrites have been widely investigated widely in the last couple of years. Modifying physical and chemical properties by selecting suitable cations offers the prospect of preparing a specific oxygen carrier optimized for the selected application and a structural stability at high reaction temperatures and mechanical stress.<sup>72–75</sup> Aston *et al.*<sup>29</sup> investigated two mixed metal spinels,  $\text{NiFe}_2\text{O}_4$  and  $\text{CoFe}_2\text{O}_4$  prepared by incipient wetness method, with  $\text{ZrO}_2$  as substrate. The Ni ferrite and Co ferrite were expected to possess higher reducibility with  $\text{H}_2$  and  $\text{CO}$  and to be fully re-oxidized by steam.<sup>76</sup> The reductions of the oxygen carriers were performed with syngas until a breakthrough of  $\text{H}_2$  and  $\text{CO}$  was observed. A period of full syngas conversion greater than 99% was visible, followed by a gradual increase of  $\text{H}_2$  and  $\text{CO}$ . The complete oxidation of the two spinel samples with steam took significantly longer than the oxidation of a pure iron reference sample. After an initial high steam conversion an extended transition phase with a low conversion efficiency was observed, which resulted in reaction times that were 3 times and 16 times longer for  $\text{CoFe}_2\text{O}_4$  and  $\text{NiFe}_2\text{O}_4$  respectively. Both samples were finally fully re-oxidized. Compared to pure iron a higher amount of hydrogen with a factor of 7–9 per mass of metal oxide can be generated, while maintaining a full syngas conversion during the reduction. However, in an industrial application the impact on the system efficiency due to the low steam conversion needs to be addressed.

$\text{CoFe}_2\text{O}_4$  spinel structured oxygen carriers using methanol as a reducing agent were investigated by Crocellà *et al.*<sup>77</sup> and

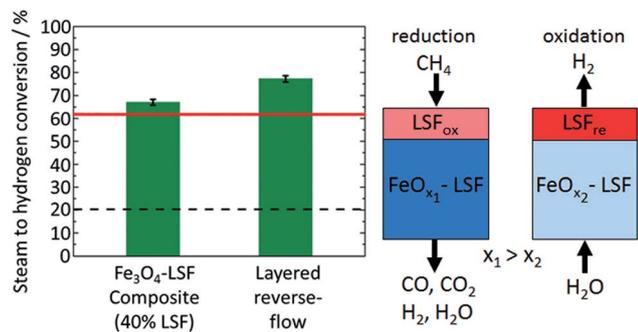


Fig. 6 Left: Steam to hydrogen conversion at  $930^\circ\text{C}$ . The red line displays the thermodynamically maximum conversion. Right: Schematic of the proposed layered reverse-flow redox process.<sup>71</sup> Reproduced from, F. He and F. Li, Perovskite promoted iron oxide for hybrid water-splitting and syngas generation with exceptional conversion, *Energy Environ. Sci.*, 2015, 8, 535–539, Copyright (2015), with permission from the Royal Society of Chemistry.



Cocchi *et al.*<sup>78</sup> Samples were prepared by co-precipitation and calcined at either 450 °C (CF450) or 750 °C (CF750). In the first reduction cycle at 300 °C the CF450 exhibited significantly faster reduction rates compared to CF750 and reached a complete solid conversion, which was attributed to a higher active surface and the presence of larger amounts of surface octahedral sites. In both samples the formation of iron carbide Fe<sub>3</sub>C was detected, which decomposed to Fe and solid carbon and acted as a carbon source during the steam oxidation. The amount of Fe<sub>3</sub>C increased with the reaction time and the formation was intensified at higher temperatures. When comparing the influence of the different calcination temperatures, it was found that the ratio between the Fe<sub>3</sub>C and the CoFe alloy was lower in the CF750 sample. Several consecutive redox cycles indicated that the CF450 sample underwent irreversible structural changes, which resulted in an increase in crystallinity and loss of active octahedral sites, while the CF750 retained its structural origin state. Eventually, both samples possessed similar structural attributes. After the steam oxidations the presence of Fe<sub>3</sub>O<sub>4</sub> was found, which revealed that a full recovery of the oxidized state was not possible, due to kinetic restrictions. Additionally, an incomplete removal of carbon was visible, meaning that an accumulation over several cycles occurred. This oxidation behaviour suggested that a final air oxidation is required to remove solid carbon and to regenerate the oxygen carrier. Trevisanut *et al.*<sup>79</sup> analysed the spinel NiFe<sub>2</sub>O<sub>4</sub> as an oxygen carrier using ethanol as a reducing agent. Samples prepared by co-precipitation were studied in a fixed bed reactor at 450 °C. A high ethanol conversion, close to 100%, and a high activity was reported. The NiFe<sub>2</sub>O<sub>4</sub> oxygen carriers led to a higher amount of fully oxidized components compared to pure iron oxide, which suggests greater accessibility of surface oxygen. The spinel sample enabled an enhanced reduction rate, despite it having similar surface areas to Fe<sub>3</sub>O<sub>4</sub>. Hence, the redox reactivity might be attributed to an increased mobility of bulk oxygen. The presence of separated Ni and NiFe alloy was observed after a steam oxidation at 450 °C, indicating an incomplete oxidation due to slow oxidation kinetics. Hydrogen was obtained with significant amounts of CO<sub>2</sub> and CO, which were related to carbon formation during the reduction. Sample analysis reassured the presence of iron carbides and solid carbon during the reduction. To overcome this incomplete steam oxidation Vozniuk *et al.*<sup>80</sup> prepared binary and ternary ferro-spinels M<sub>0.6-x</sub>M<sub>x</sub><sup>2</sup>Fe<sub>2.4</sub>O<sub>y</sub> using Co and Mn with different atomic ratios from  $x = 0$  to 1. Samples with uniform morphology and particle size distribution were obtained. The integration of the metal cations strongly modified the reducibility towards hydrogen by changing the structural characteristics and the oxygen mobility. Direct ethanol reduction experiments in a fixed bed quartz reactor identified acetates and carbonates as the primary intermediates of the redox reaction if Co modified oxygen carriers were used, while samples with solely Mn as additives exhibited a complete absence of carbonates and a reduced reactivity. An initial complete ethanol conversion was achieved with all oxygen carrier samples. The concentrations of CO<sub>2</sub> and CO suggested that the partial oxidation of ethanol was the main reduction route. Acetone,

which indicated the aldol reaction route and the decomposition of ethanol to acetaldehyde was detected in the beginning and decreased over time. With the ongoing experiments the amount of coke and higher hydrocarbons was increasing due to the limitation of exchangeable oxygen in the oxygen carrier. It was concluded that the modification of Co improves the reducibility of oxygen carriers with ethanol, while Mn enhanced the resistance to coking, but did not completely exclude it. Co<sub>0.3</sub>Mn<sub>0.3</sub>Fe<sub>2.4</sub>O<sub>y</sub> was identified as the most promising candidate because of its high hydrogen yield during reduction and a significantly reduced carbon formation. The complete carbon conversion was not achieved during the oxidation, which still required a final carbon removal step by air. Kuo *et al.*<sup>81</sup> investigated a nickel aluminium spinel ferrite NiFeAlO<sub>4</sub> as oxygen carrier for hydrogen generation. After a calcination temperature of 1300 °C a single cubic spinel phase of NiFeAlO<sub>4</sub> was observed with a crystalline size distribution between 100–300 nm. After a complete reduction Al<sub>2</sub>O<sub>3</sub> and the bimetal FeNi was formed, which possessed a high porosity and efficiently prevented agglomeration by a large triple phase boundary area. The oxygen carrier was submitted to 10 consecutive redox cycles to investigate the cycle stability. The amount of weight loss increased within the first four cycles, possibly by the formation of additional pores by the phase changes and finally stabilized without any signs of degradation. A slight decrease of the redox reaction rate was observed. The steam oxidation yielded hydrogen and a phase mixture of Ni, Fe<sub>3</sub>O<sub>4</sub> and Al<sub>2</sub>O<sub>3</sub>, which was oxidized by air to the original spinel phase. Bhavsar *et al.*<sup>82</sup> focused on preparing a nanostructured oxygen carrier support based on barium hexaaluminate (BHA). A carrier structure with a high porosity and sponge-like nanostructure, which suppressed migration and prevented sintering, was obtained. Depending on the application, a selected active metal oxide can be embedded as nano-particles in the support material. In case of chemical looping hydrogen production nano-sized iron particle were applied (Fig. 7). The carrier showed a high stability at temperatures up to 900 °C without any signs of deactivation. Redox reaction revealed similar reduction rates compared to conventional oxygen carrier supports, however, significantly

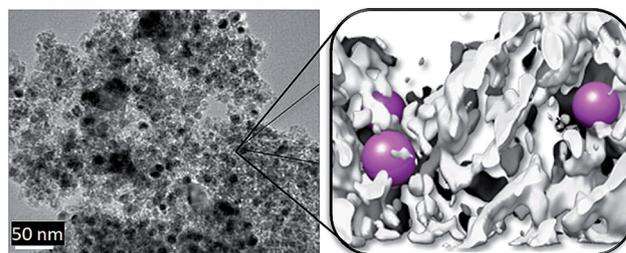


Fig. 7 Left: TEM image of a Ni-BHA nanostructured carrier (Ni 40 wt%). Right: 3D tomographic reconstruction of the carrier nanostructure from a set of TEM slices. The white structure represents the BHA carrier with the embedded metal nanoparticles shown as pink spheres.<sup>82</sup> Reprinted and adapted from S. Bhavsar, M. Najera, R. Solunke, and G. Vesper, Chemical looping: to combustion and beyond, *Catal. Today*, 2014, 228, 96–105, Copyright (2013), with permission from Elsevier.





**Table 2** Overview of chemical looping plants in operation for hydrogen generation

System	Organization	Power range	Gas feed	Solid flow/solid inventory	Reaction temperatures	Dimension	Reference
Three-reactor moving bed system	Ohio State University	25 kW <sub>th</sub>	Syngas, CO/H <sub>2</sub> = 2/1, 15 L min <sup>-1</sup>	150 g min <sup>-1</sup>	900 °C		59 and 91–93
2-Compartment fluidized bed	Chalmers University of Technology, Sweden		Syngas or CO = 0.5–1.25 L min <sup>-1</sup> , steam = 4–5 L min <sup>-1</sup>	180–240 g min <sup>-1</sup>	800–900 °C	System height = 300 mm reactor = 25 × 25 mm	84
Three-reactor fluidized bed system	Key Laboratory of Energy Thermal Conversion and Control, China	50 kW	Feed = 24–31 Nm <sup>3</sup> h <sup>-1</sup>	Solid material = 5 kg	Cold flow mode		88
Interconnected circulating fluidized bed reactor	Aragon Institute of Engineering Research, Spain		H <sub>2</sub> /CH <sub>4</sub>	0.235 kg, 85 wt% Fe <sub>2</sub> O <sub>3</sub> , 15 wt% SiO <sub>2</sub>	500–575 °C	Reactor: steel tube 5.4 cm ID, 6.0 cm OD, height = 41.5 cm Reactor length = 0.05 m, cross section area = 200 × 100 μm	89
μ-System	University of Newcastle, Australia	μ-System	CH <sub>4</sub>		900 °C		95 and 96
Three-reactor counter-current moving bed system	Hydrogen and Fuel Cell Department, Korea	300 W <sub>th</sub>	CH <sub>4</sub> = 0.55 L min <sup>-1</sup> , steam = 0.25 L min <sup>-1</sup>	65–75 g min <sup>-1</sup>	FR = 900 °C, SR = 800 °C		98
Multistage circulating moving bed reactor	Department of Chemical Engineering, Korea		Compressed air	22 kg, 20–100 kg m <sup>-2</sup> s <sup>-1</sup>	From ambient to 850 °C	Fuel reactor: 0.07 (bottom), 0.16 (top) × 0.06 × 1 m <sup>3</sup> , steam reactor = 0.16 × 0.06 × 1.4 m <sup>3</sup> , riser = 0.03 × 0.06 × 3.8 m <sup>3</sup>	99

accelerated oxidation rates were measured. The steam oxidation rates were further improved by decreasing the particle sizes from 54 nm to 16 nm.

### 3. Reactor design

Two reactor designs, the fixed bed reactor and the fluidized or moving bed, were successfully operated and presented in the surveyed literature (summarised in Table 2). The fixed bed reactor system is operated in batch mode by switching the operation gases. The oxygen carrier is kept in place, while the reduction and the oxidation reactions are executed consecutively. This enables a design with a low degree of complexity due to the absence of moving parts. High purity hydrogen is obtained because the carbon containing fuel streams and the hydrogen stream are completely separated. The fluidized or moving bed system circulates the oxygen carrier between separated reactors or reactor sections. The moving solid material increases the system complexity and mechanical strain, however, the higher technological complexity enables several beneficial properties. In contrast to a fixed bed reactor these systems are operated in continuous mode and thus enable an (i) uninterrupted hydrogen production combined with pure separated streams of CO<sub>2</sub> and N<sub>2</sub>.<sup>30,83,84</sup> In each reactor the same reaction, either reduction or oxidation, is performed, which permits (ii) a specific reactor layout regarding materials, design and operation parameters to optimize gas yield and solid conversion.<sup>85–89</sup> (iii) The solid materials, which suffer from cycle stability can be continuously replaced during normal operation modes. (iv) The oxygen carrier can participate in the heat management of the system by delivering heat from the exothermic oxidations to the endothermic reduction and thus reduce the required amount of external heat.<sup>83,90</sup> However attrition losses due to movement of the solids and a lower hydrogen purity caused by the gas slip between the reactors are to be expected.<sup>84,89</sup>

A 25 kW<sub>th</sub> sub-pilot syngas for syngas chemical looping (SCL) was introduced by Ohio State University (Fig. 8 left).<sup>59,91–93</sup> Sridhar *et al.*<sup>91</sup> discussed the adaption of the system to a three-reactor iron based system for the co-generation of pure hydrogen and electricity. The fuel reactor and the steam reactor were operated in a co-current moving bed system and a fluidized bed air reactor, which acted as a riser. The reactors were separated by a system of two ball valves. The solid flow rate was controlled with a rotary disc feeder. Experiments were performed with syngas (CO/H<sub>2</sub> 2/1 flow rate 15 L min<sup>-1</sup>) at 900 °C and a solid flow of 150 g min<sup>-1</sup>. First operations demonstrated that a hydrogen production with an average purity of 94.5% and *in situ* carbon capture is feasible in this system. Tong *et al.*<sup>92</sup> optimized operation parameters and investigated the redox performance of two iron based oxygen carriers with 2 mm and 4.5 mm diameters during a total operation of 300 hours. Both samples provided sufficient high cycle stability, whereas the smaller particles possessed a faster reaction rate, probably due to the higher surface area. Both samples performed similarly well regarding syngas conversion and hydrogen purity. The smaller particles had a lower fluidizing velocity and required

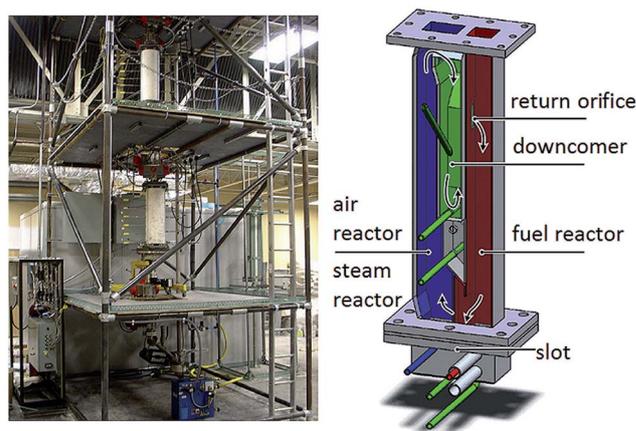


Fig. 8 Left: Photo of the 25 kW<sub>th</sub> sub-pilot SCL unit at The Ohio State University.<sup>59</sup> Right: Schematic representation of the two-compartment fluidized-bed reactor proposed by Rydén *et al.*<sup>84</sup> Reprinted and adapted from A. Tong, S. Bayham, M. V. Kathe, L. Zeng, S. Luo, L.-S. Fan, and W. G. Lowrie, Iron-based syngas chemical looping process and coal-direct chemical looping process development at Ohio State University, *Appl. Energy*, 2014, **113**, 1836–1845, Copyright (2013), with permission from Elsevier. Reprinted and adapted from M. Rydén and M. Arjmand, Continuous hydrogen production via the steam-iron reaction by chemical looping in a circulating fluidized-bed reactor, *Int. J. Hydrogen Energy*, 2012, **37**, 4843–4854, Copyright (2013), with permission from Elsevier.

less air in the rise to be transported. Hence, smaller oxygen carriers are expected to show better flow properties and a higher resistance to attrition. System improvements by reducing gas leakage between the reactors were achieved by replacing mechanical seals<sup>91</sup> with a non-mechanical valve consisting of a standpipe with injected nitrogen.<sup>92</sup> Furthermore, a gas preheater for the synthesis gas feed was installed to prevent carbon deposition at lower temperatures and a dual syngas injection was added to optimize to axial gas concentration profile and increase the syngas conversion from 78.34% to 92.23%. A final hydrogen purity >99% at steady conditions was obtained. The system was also operated using methane with a feed conversion of 99.99% while maintaining a high oxygen carrier conversion of 32.50%.<sup>93</sup> A reduced solid flow rate increased the reduction degree to 48.82%, however the CH<sub>4</sub> conversion dropped to 80.35%. The results suggested that a complete fuel conversion is predefined by a critical solid/gas flow ratio. Flow ratios that exceed this ratio result in a non-reacting zone in the top of the moving bed. The counter-current moving bed used in this system is expected to increase the oxygen carrier conversion by up to 5 times compared to a co-current fluidized bed systems with 11%.<sup>59,94</sup> In an experimental investigation lasting 2 hours, an oxygen carrier conversion in the reducer reactor of 35.54%, related to the theoretical value of 39.6%, was achieved, while maintaining a syngas conversion of >99%.<sup>91</sup> Based on previous results with a syngas feed, the 25 kW<sub>th</sub> sub-pilot unit was utilized for the coal direct chemical looping process (CDCL) in continuous operation lasting 200 hours. A coal conversion ranging from 90–99% with an average conversion of 99.7% was obtained. These



experiments were not combined with the steam oxidizer, hence no conclusion can be drawn on the possible hydrogen purity, however, combustor gas analysis for carbon sequestration indicated a CO<sub>2</sub> purity of 99.7% with CO 0.2% and CH<sub>4</sub> 0.1% as the main impurities.<sup>59</sup>

Rydén *et al.*<sup>84</sup> examined a two-compartment fluidized bed reactor for hydrogen production at temperatures between 800–900 °C (Fig. 8 right). The system with 300 mm in height consisted of a steam reactor (25 × 25 mm) which transported the oxygen carrier upwards to the down-comer, where the particles passed into a J-type loop-seal and were transferred to the fuel reactor through an orifice. The fuel reactor was operated as a bubbling bed. At the bottom, the oxygen carrier returned past a U-type slot to the steam reactor and closed the circle. Experiments were executed with syngas or CO as feed (0.5–1.25 L min<sup>-1</sup>) and steam (4.0–5.0 L min<sup>-1</sup>) as oxidant. An iron based oxygen carrier (60 wt% Fe<sub>2</sub>O<sub>3</sub> and 40 wt% MgAl<sub>2</sub>O<sub>4</sub>) in the diameter range of 90–250 μm was applied. The operation started with a full conversion of the fuel due to the presence of Fe<sub>2</sub>O<sub>3</sub>. After the initial reduction to Fe<sub>3</sub>O<sub>4</sub>, CO an incomplete conversion was observed due to reduction stage of Fe<sub>3</sub>O<sub>4</sub> to FeO. Particles reduced to FeO were transferred to the steam reactor. A stable operation without problems of de-fluidization was maintained for up to 4 hours with a gas conversion between 75–80%, which increased with a decreasing fuel flow. These results were slightly higher than expected by thermodynamic predictions, probably caused by gas leaking from the fuel reactor into the steam reactor. In the steam reactor the FeO particles were oxidizing with a hydrogen production rate of 0.5 L min<sup>-1</sup>, which matched the theoretically value as long as the fuel gas flow was kept at low values. The impurities CO<sub>2</sub> 19% and CO 5% were detected in the hydrogen as a results of gas leakage.

Moghtaderi<sup>95,96</sup> developed a miniaturized chemical looping steam reformer CLSR in micro-reactor assembly, which is suitable for on-board mobile fuel enrichment with hydrogen (Fig. 9 right bottom). The miniaturized system is expected to show improvements compared to conventional designs with regard to (i) volumetric productivity, (ii) safety issues, (iii) operation costs, (iv) scalability, (v) pressure drop and (vi) surface-to-volume-ratio. The micro-reactor assembly used in the experiments consisted of a single stainless steel plate with a zigzag shaped engraving (0.05 m length, cross section area: 200 × 100 μm) (Fig. 9 right top). Fe<sub>3</sub>O<sub>4</sub> on alumina support with a layer thickness of 1 μm and a particle size of 20–40 μm was utilized as an oxygen carrier. Reduction experiments at 900 °C with methane resulted in a rapid reduction to FeO and the formation of solid carbon by catalytic decomposition of methane. The consecutive steam oxidations started with an initially fast oxidation rate, which decreased significantly at a solid conversion of 80%. The complete oxidation took approximately 4 times longer than the reduction. A series of 4 reactors was proposed, each consisting of one μ-CLSR unit as described before, to equalize the different reaction times and compensate for the longer oxidations. The proposed operation procedure consisted of consecutively executed reductions and simultaneously performed oxidations in the 4 reactors. The experimental results suggested that, within the timeframe of 48

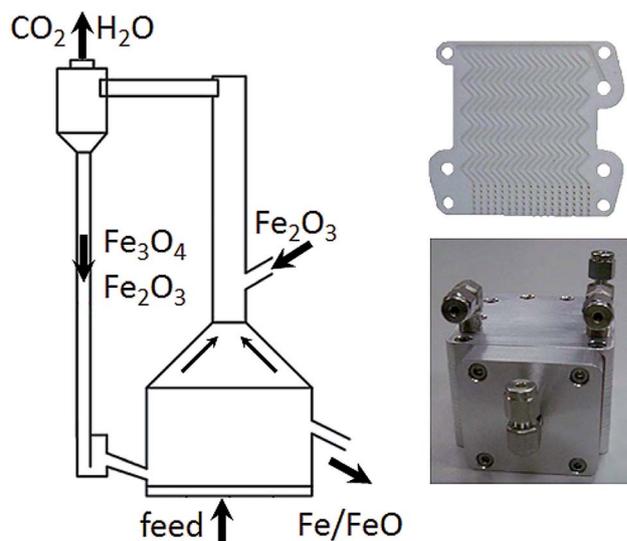


Fig. 9 Left: Schematic representation of the two-staged fluidized-bed system by Xue *et al.*<sup>88</sup> A single reactor plate with the zigzag shaped pattern (right top) utilized in the CLSR micro-reactor (right bottom) proposed by Moghtaderi.<sup>95</sup> Reprinted and adapted with permission from Z. Xue, S. Chen, D. Wang, and W. Xiang, Design and fluid dynamic analysis of a three-fluidized-bed reactor system for chemical-looping hydrogen generation, *Ind. Eng. Chem. Res.*, **51**, 4267–4278, Copyright (2012) American Chemical Society. Reprinted and adapted with permission from B. Moghtaderi, Hydrogen enrichment of fuels using a novel miniaturised chemical looping steam reformer, *Chem. Eng. Res. Des.*, **2012**, **90**, 19–25, Copyright (2011) The Institution of Chemical Engineers, Published by Elsevier B.V.

seconds, a solid conversion degree of 30% in all participating μ-reactors was achieved followed by a complete re-oxidation yielding a hydrogen product stream with 96% purity. Thus, the layout of 2 × 4 reactors enables a continuous production of hydrogen in a μ-CLSR system.

Xue *et al.*<sup>88</sup> developed a three-reactor fluidized-bed system in the range of 50 kW. The system was constructed and operated as a cold-flow model. A fluidized fuel reactor was proposed as a key component, which consisted of a base bubbling bed at low velocity and an upper fluidized bed at high velocity (Fig. 9).<sup>87,88,97</sup> Fe<sub>3</sub>O<sub>4</sub> particles entering near the bottom bubbling bed react with the fresh fuel gas and leave the reactor in reduced state at the bottom. The reduction of Fe<sub>3</sub>O<sub>4</sub> takes significantly longer than the reduction of Fe<sub>2</sub>O<sub>3</sub>, thus a sufficiently large bubbling bed guarantees the required residence time in the bottom part to ensure a desired solid conversion. The partially reacted fuel gas moves through the upper reactor section, where Fe<sub>2</sub>O<sub>3</sub> particles from the oxidizer are supplied. The Fe<sub>2</sub>O<sub>3</sub> particles are partially reduced by completely converting the reduction gas to H<sub>2</sub>O and CO<sub>2</sub>. The fuel reactor design was successfully operated for 600 minutes using aqueous bio-oil.<sup>87</sup> Xue *et al.*<sup>88</sup> integrated the reactor into a 50 kW three-fluidized-bed reactor system to investigate operation parameters and solid circulation characteristics. The main design values were an operation temperature of 900 °C, methane as fuel, a particle size of 100–305 μm, a solid inventory of 5 kg and gas flows between 24–31 Nm<sup>3</sup> h<sup>-1</sup>. Cold-flow operations were performed with He/air in all three



reactors with glass beads as particles. By varying the gas flows and changing the fluidizing conditions, the solid circulation rate was manipulated. It was found that the solid circulation rate influenced the gas leaks between the reactors, which is of high importance regarding hydrogen purity and CO<sub>2</sub> emissions. The cold flow operations were performed in long-term operation tests with stable pressure conditions.

Cho *et al.*<sup>98</sup> constructed a 300 W<sub>th</sub> three-reactor chemical looping system to produce high purity hydrogen out of methane. Steady solid circulation operation, using Fe<sub>2</sub>O<sub>3</sub> supported with ZrO<sub>2</sub> with a diameter of 125–300 μm, was executed for 13 hours.<sup>36</sup> The solids showed no agglomeration and only small signs of thermal sintering, which suggested a high durability. An average methane conversion of 94.15% at 900 °C was achieved. Fuel flow analysis indicated that a high methane conversion is possible as long as a moving bed regime is maintained. A transition to a fluidized bed reduced the methane conversion to below 90%. Hydrogen with a purity of 99.95% was produced at 800 °C with minor CO contaminations, when reducing the oxygen carrier to FeO. CH<sub>4</sub> was not detected, which suggested that gas leakage into the steam reactor was negligible. The H<sub>2</sub> production rate was approximately 63% below the theoretical value due to a non-uniform flow pattern. The formation of carbon was investigated under the deep reduction of the oxygen carrier by lowering the solid flow. The higher amount of completely reduced iron catalysed the deposition of solid carbon. A higher amount of produced hydrogen with a significantly reduced purity was observed, as a result of the oxidation of carbon with steam.

Hong *et al.*<sup>99</sup> studied the solid flow characteristics of zirconia beads (mean diameter 186 μm) in a multi stage moving bed system. The gas flows into the loops seals were varied to investigate the changes in the flow pattern, the solid mass flux

and the solid mean residence time at ambient temperatures. In the moving bed regime the solid mass flux and the pressure drop increased with a rising gas flow in the loop seal. A stable pressure profile was maintained. The higher gas flow correlated with a growing bed height but lowered the mean residence time of the solid particles because of an enlarged solid mass flux. A further increase of the gas flow changed the flow pattern to a bubbling flow, and finally to a slug flow characteristic. The flow patterns had little effect on the solid mass flux but resulted in significant pressure deviations. As long as a moving bed regime was maintained the solid flux pattern remained similar at different temperatures, however higher gas velocities and a higher solid flow rate were measured. Lee *et al.*<sup>100</sup> concluded that a riser is required as a 4<sup>th</sup> module to increase the stability of the system. The selected operation conditions of gas velocities and solid circulation rate enabled a stable pressure profile and prevented gas leakage between the reactors.

Herguido *et al.*<sup>89</sup> introduced a single reactor with two interconnected beds, a slow reduction bed with H<sub>2</sub>/CH<sub>4</sub>, and a second fast bed for steam oxidation (Fig. 10d). Oxygen carrier with particle size of 150–200 μm mixed with silica sand were applied. The two beds are connected with two slits at the bottom and the top. Redox experiments were performed at 550 °C. Three designs of interconnections between the two beds in respect to gas leakage were evaluated, a rectangular slit, a J-connection and an orifice connection (Fig. 10a–c). The orifice connection with a hole-diameter of 0.3 cm and an open area of 0.4 cm<sup>2</sup> showed a more superior performance than the other design variations. The gas leakage was successfully minimized by reducing the gas velocity difference in the two fluidized beds.

## 4. Process development

Chemical looping technologies for hydrogen generation are usually operated with a reducing synthesis gas provided by external processes, *e.g.* from coal gasification or methane reforming. However, recent work has shown that the process can be easily modified to handle a variety of feedstocks, including unconventional resources, either by direct reduction or by the incorporation of the chemical looping system with syngas conversion processes. Herguido *et al.*<sup>89</sup> investigated the applicability of the chemical looping process for the separation of a mixture of hydrogen and methane. A reactor with two interconnected fluidized beds and an iron based oxygen carrier were used for the fuel processing. Reactions were executed in a temperature range of 500–550 °C and with hydrogen fractions between 0.35–0.65. The operation conditions enabled a reduction of iron oxide with hydrogen but left methane and other impurities inert. An average hydrogen conversion of 20% was obtained with little influence of the varied parameters (gas velocity, relative velocity and hydrogen content), which suggested that the separation performance is mainly limited by thermodynamic restrictions. The separations system was tested for 44 h and enabled the production of pure hydrogen without carbon contaminations. Small amounts of solid carbon were found in the reactor after the operation, thus coke remained

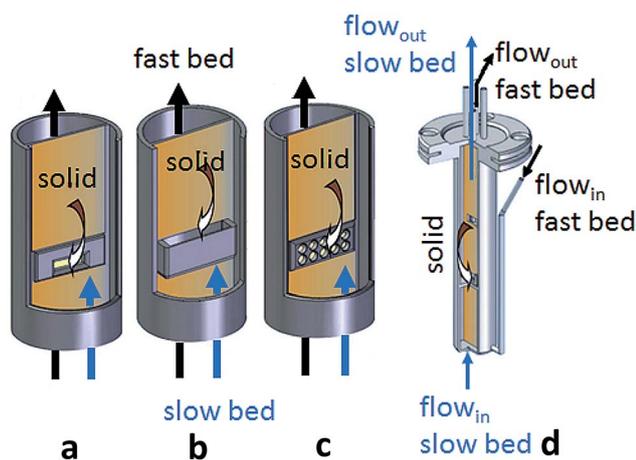


Fig. 10 Schematic of different connecting devices (a) rectangular connection, (b) J-connection, and (c) orifice connection. Schematic representation of the fluidized-bed system (d) by Herguido *et al.*<sup>89</sup> Reprinted and adapted from J. Herguido, J. Peña, and E. Carazo, Experimental assessment of hydrogen separation from H<sub>2</sub>/H<sub>4</sub> mixtures by the "steam-iron process" in an interconnected circulating fluidized bed reactor, *Int. J. Hydrogen Energy*, 2014, 39, 14050–14060, Copyright (2014), with permission from Elsevier.



inert during the experiments at the selected oxidation conditions.

### Alcohols as feedstock

The same group investigated ethanol<sup>101</sup> and methanol<sup>102</sup> as feed in a fixed bed reactor. Fe<sub>2</sub>O<sub>3</sub> supported with CeO<sub>2</sub> and Al<sub>2</sub>O<sub>3</sub> was used as the oxygen carrier. Higher temperatures and a higher ethanol partial pressure (0.05–0.15 bar) accelerated the reduction rates. However, an increased reduction temperature led to a slower re-oxidation rate in the consecutive steam oxidation, most likely caused by thermal sintering of the particles. The reductions of the alcohol feeds revealed 3 stages: (i) the reduction of Fe<sub>2</sub>O<sub>3</sub>, (ii) the reduction of Fe<sub>3</sub>O<sub>4</sub> to Fe and (iii) a stable thermal decomposition of ethanol and methanol. A higher reduction degree of the solid led to a higher amount of partially oxidized components (CO and H<sub>2</sub>) and a lesser amount of fully oxidized components (H<sub>2</sub>O and CO<sub>2</sub>). The gas composition converged towards the theoretical values of thermal decomposition products after the full reduction of the oxygen carrier. Solid carbon formation occurred in all three stages and was verified by the detection of structured nanotubes after the experiments. The formation of iron carbide, which decomposed to iron and solid carbon, was promoted at 450 °C after the reduction to metallic iron. A process with short-time looping steps may avoid the carbide formation and thus limit the amount of solid carbon to minimize the carbon contaminations.<sup>103</sup> The methanol reduction showed a significantly higher carbon deficit in the carbon balance, which correlated with higher amounts of coke at lower temperatures.<sup>102</sup> The consecutive oxidations with steam at 500 °C produced hydrogen with little CO and CO<sub>2</sub> contaminations (below 1 ppm) in the case of ethanol as feed and no carbon impurities using methanol (50 ppm detection limit). Hence, coke remained mostly inert on the solid particles. Several consecutive cycles suggested that the advisable reaction temperature for ethanol and methanol is 675 °C and 700 °C respectively. At these temperatures the smallest decay of the oxygen carrier, while providing a sufficient fast reduction rate, was observed. Coke formation occurred over to complete temperature range with both feeds, which demands a decoking step after several cycles to prevent carbon accumulation.<sup>101–103</sup> TPR reduction analysis of three iron oxides, hematite, goethite and magnetite using low concentrations of ethanol in nitrogen was performed by Rosmaninho *et al.*<sup>104</sup> All samples revealed the reduction intermediates Fe<sub>3</sub>O<sub>4</sub> and FeO before the complete reaction to Fe, irrespective of the initial oxidation state. In all cases the ethanol conversion started at a temperature of 300 °C. The complete conversion of ethanol was reached according to the accessibility of surface oxygen. Hence, hematite and goethite enabled a complete ethanol transformation between 350–400 °C, whereas the application of magnetite required a temperature beyond 500 °C to achieve a complete conversion. At 700 °C, a complete reduction of the oxygen carrier to iron was observed. A further temperature increase led to a significant formation of solid carbon and iron carbide, thus 700 °C was identified as the most suitable reaction temperature to achieve a complete reduction and to minimize

the deposition of carbon. The oxidation with steam re-oxidized the oxygen carrier to magnetite with small amounts of hematite and wustite. Consecutive redox cycles revealed a degradation of the oxygen carrier, which led to an incomplete reduction and a reduced amount of produced hydrogen of 70% after 5 cycles.

### Process combination of reforming of hydrocarbons with CLWS

Plou *et al.* and Herrer *et al.* combined the catalytic dry reforming with the chemical looping hydrogen in a single reactor bed.<sup>105–107</sup> As a solid mixture a triple oxide oxygen carrier (Fe<sub>2</sub>O<sub>3</sub> supported with CeO<sub>2</sub> and Al<sub>2</sub>O<sub>3</sub>) and a nickel based catalyst (NiO/NiAl<sub>2</sub>O<sub>4</sub> with an excess of NiO of 10 wt%) was applied with a weight ratio of OC/catalyst of 85/15 wt%. Experiments were performed with different biogas compositions CH<sub>4</sub>/CO<sub>2</sub> 50/50–65/35 at temperatures from 600–750 °C. Three stages were visible in the fixed bed reductions, which were analysed in detail. The reaction started with a high methane conversion directly followed by a conversion minimum. This phase was initially dominated by the reduction of Fe<sub>2</sub>O<sub>3</sub> to Fe<sub>3</sub>O<sub>4</sub> and the full oxidation of the feed to CO<sub>2</sub> and H<sub>2</sub>O. After the consumption of the lattice oxygen of Fe<sub>2</sub>O<sub>3</sub> the reduction of NiO to Ni occurred, which was indicated by the appearance of H<sub>2</sub> and CO. The presence of elemental Ni enabled the catalytic dry reforming of methane to produce a stable fraction of CO and H<sub>2</sub> in combination with the deep reduction of Fe<sub>3</sub>O<sub>4</sub> to iron. Once the oxygen carrier was completely reduced, the solely catalytic dry reforming of biogas was observed. An increasing reaction temperature enhanced the reaction rates and the transition to the stable reforming stage, whereas an increased CH<sub>4</sub> ratio slightly accelerated the transition between the stages. The comparison of experimental data with theoretical data showed good accordance for the dry reforming but a constant deviation regarding the iron oxide reduction reaction, which suggests diffusion limitations hindering the mass transfer. The formation of coke by CH<sub>4</sub> decomposition indicated by a too high amount of measured hydrogen and thus the deactivation of the catalysts by blocking of active catalytic sites was observed. The effect was enhanced with increasing CH<sub>4</sub> fractions. Post reaction analysis reported the presence of graphitic carbon<sup>105,107</sup> and highly amorphous carbon.<sup>106</sup> Coke was only present on the nickel particles, which explained the catalytic degradation. During purging phases with inert gas a mass loss was observed, which increased with higher temperatures. This suggested a solid–solid decoking reaction between the incompletely reduced oxygen carriers and the solid carbon and explained the absence of coke on the oxygen carrier in the post experimental analysis. The oxidations with steam at 500 °C produced hydrogen with a purity of >99% with carbon dioxide as impurity. The main part of the carbon depositions remained inert during the steam oxidation, and thus accumulates over consecutive cycles without a regeneration step.<sup>102,106–108</sup> A combination of a catalytic steam reforming and the chemical looping hydrogen process in one reactor with two adjacent fixed bed was investigated by Nestl *et al.*<sup>109</sup> Investigations of methane reforming were executed, varying temperature, pressure and the S/C-ratio, emphasizing a high conversion of CH<sub>4</sub>, the avoidance



of carbon depositions and a reformat-gases with a sufficient reduction potential. The results suggested that high  $\text{CH}_4$  conversion of >99% is possible, however higher system pressures require a significantly greater S/C ratios, and thus yield synthesis gas with increased water content and a lower reduction potential. It was concluded that the reforming in combination with the reduction step of an iron based oxygen carrier should be performed at ambient pressure. The oxidations with steam were executed at elevated pressure up to 8–11 bar to produce hydrogen with a purity >99%. Small amounts of  $\text{CO}_2$  were detected, which implied the formation of solid carbon during the reduction, although not predicted by the simulations.

### Pressurised oxidations

Voitic *et al.*<sup>108,110</sup> analysed the correlations between the increased oxidation pressure and the cycle stability, the conversion efficiency and the structural integrity of an iron based oxygen carrier. Steam oxidations were performed at 750 °C and at different pressure steps up to 55 bar. The results revealed no repercussion of an increased system pressure on the oxygen carrier conversion and on the oxidation rate. A high purity hydrogen was obtained by pressurized steam oxidation in the range of 99.95–99.999% with CO and  $\text{CO}_2$  as impurities. The amount of carbon impurities was not influenced by the increased system pressure, however a correlation with the length of the reduction reaction was observed. The mass balances indicated an incomplete oxidation of solid carbon by steam. This resulted in an accumulation of solid carbon and a gradually decrease of hydrogen purity. Hence, an air regeneration step had to be included to completely remove carbon contaminations and to regenerate the oxygen carrier.<sup>101,102,108</sup> A linear decay of the oxygen carrier stability was visible which was related to thermal sintering of iron by deep reduction. A porous structure was maintained in the particles despite the pressure load, which demonstrates the applicability of the chemical looping water splitting process in a fixed bed configuration for high pressure hydrogen production with a hydrocarbon feedstock.

### Heavy fraction bio-oil as feedstock

Zeng *et al.*<sup>87,97</sup> and Xiao *et al.*<sup>58</sup> investigated the reduction behaviour of a bio-oil heavy fraction from a cotton stalks pyrolysis unit with a water content of 18.54 wt%. Ilmenite was selected as the proper oxygen carrier in a fixed bed redox reaction for its resistance to carbon depositions and reducibility at temperatures >950 °C. Preliminary experiments yielded a carbon conversion in the range of 57–82%, which was enhanced by an increasing temperature with an optimum at 950 °C. Beyond 950 °C a declining carbon conversion was measured, which was related to an increased methane crack reaction and the formation of solid carbon. In several consecutive cycles, the ilmenite sample achieved a stable hydrogen yield of 93–96  $\text{Nm}^3 \text{kg}^{-1}$  within the first 10 cycles. However, after 10 cycles, severe sintering, which led to a lower active surface and average pore diameter, and a drop in released hydrogen was observed. Significant amounts of carbonaceous contaminations were

detected with a hydrogen purity below 85%. The gasification of solid carbon with steam was identified as the source of contaminates. The heavy fraction of the used bio-oil contained a high amount of phenolic compounds, thus increasing the tendency towards carbon deposits. In order to improve the hydrogen purity steam was added during the reduction process. A steam/oil ratio of 2.0 improved the carbon conversion of the fuel to >99% and enhanced the hydrogen purity to >99%. However higher amounts of steam decreased the oxygen carrier reduction degree, thus lowering the amount of produced hydrogen.

### Hydrothermal hydrogen generation

Hydrothermal hydrogen generation by the oxidation of iron enables the operation at very mild temperatures in the range of 100–200 °C. This reduces the complexity of the oxidation process significantly compared to conventional chemical looping water splitting, which usually requires temperatures >600 °C. Feasibly studies were performed by Tsai *et al.*<sup>111</sup> in an autoclave by measuring the pressure profile and gas phase at different reactor temperatures. Reference tests were performed with solely water as reactor filling. The supplement of iron powder led to a constant pressure increase, which was further raised by the amount of iron, the reactor temperature and smaller particle sizes.<sup>111,112</sup> After oxidizing the iron powder the pressure remained at a constant level. The gas phase consisted of pure hydrogen with a low water content of 0.525% at 120 °C. Michiels *et al.*<sup>112</sup> further enhanced the hydrothermal process by using a 1 molar potassium hydroxide solution. The system was put under  $\text{CO}_2$  pressure, which dissolved in the aqueous solution and formed potassium carbonate in the presence of potassium hydroxide. The reactor was heated to temperatures up to 200 °C. Iron in the presence of carbonate was oxidized with water to magnetite,  $\text{H}_2$  and  $\text{CO}_2$  within a reaction time of 16 h. After cooling down, hydrogen was measured in the gas phase with a purity of >99 mol%. Experiments focussing on the role of compressed  $\text{CO}_2$  in the system revealed that the dissolved carbon dioxide was critical for the process by forming carbonate ions to promote the oxidation of iron to hematite. The absence or deficit of  $\text{CO}_2$  resulted in a reduced amount of produced hydrogen. An excess of  $\text{CO}_2$  caused the formation of siderite instead of magnetite, thus a lower amount of hydrogen with significant  $\text{CO}_2$  contaminations was detected. The presence of potassium hydroxide was vital in the system to promote the solubility of  $\text{CO}_2$  and the formation of  $\text{CO}_3^{2-}$ -ions. The absence of potassium carbonate resulted in a lower amount of hydrogen. The system did not suffer from a net-carbonate-ions loss within one complete cycle. Hence, the aqueous solution can be reused in the sequential cycle by adding the amount of consumed water.

## 5. Process simulations

The main research focus in the field of chemical looping hydrogen technology is currently on three-reactor systems. These systems are very flexible in terms of the fuel input, while





The conventional plant suffered from a larger exergy destruction within the combustion engine, meaning that the OSR-CLH system had a higher total plant efficiency of 7.3% points. However, the economic analysis indicated higher capital costs and higher production costs for the chemical looping system of approximately 23% caused by the additional system units and by the oxygen carrier as second material.

Zeng *et al.*<sup>119</sup> simulated a direct coal three-reactor chemical looping plant solely for hydrogen generation using iron oxide. The fuel reactor temperature was operated at 900 °C with a solid conversion of 56.3%. Two pressure conditions 30 atm and 2 atm with a Fe<sub>2</sub>O<sub>3</sub>/coal flow rate ratio of 1.38 enabled a complete coal conversion. A pressure increase beyond 30 atm reduced the coal conversion efficiency. In the steam reactor, a temperature of 700 °C and a steam/Fe ratio of 1.9 ensured a complete oxidation to magnetite while optimizing the steam conversion irrespective of the selected system pressure. The two operation conditions, pressurized and atmospheric, yielded total efficiencies of 79.69% and 71.36% respectively with carbon capture rates of 99%. The operation at atmospheric pressure had a lower efficiency, however, the absence of gas compressors and expanders significantly lowered the plant investment cost and its complexity. A third case, which considered kinetical limitations with a lower carbon conversion of 95% yielded a total efficiency of 77.24%. The lower coal conversion led to carbon impurities in the produced hydrogen. Hence, a higher purification effort and a larger amount of tail-gas had to be taken into account. The formation of pollutants, which originated from the coal feed were considered. Chlorine was converted to HCl, whereas mercury stayed in elemental form in the fuel reactor. Both pollutants exited the system with the CO<sub>2</sub> stream. Sulphur formed Fe<sub>0.877</sub>S in the reducer and was mainly carried over to the steam and air reactor where it was re-oxidized, resulting in both H<sub>2</sub>S and SO<sub>2</sub> being present in the system.

Chen *et al.*<sup>120</sup> simulated the integration of a three-reactor chemical looping hydrogen system with a SOFC/gas turbine system for power generation. A shell gasifier was selected to convert bituminous coal to a synthesis gas. The reduction gas was utilized to convert an iron based oxygen carrier to wustite at 950 °C in the fuel reactor. Pure hydrogen was generated in the steam reactor at 750–800 °C and supplied to a SOFC. The SOFC anode off gas, which contained 10–25% of hydrogen, was burned in a gas turbine. Heat recovery steam generator systems were installed downstream after the fuel reactor and the gas turbine. The combination of a CLH system with a SOFC and the utilization of pure hydrogen was expected to show advantages regarding (i) better electrochemical kinetics, (ii) the complete prevention of solid carbon formation, (iii) a better heat distribution by avoiding endothermic reforming reactions and (iv) the complete sequestration of carbon from the fuel. The plant analysis yielded an electrical efficiency of 43.53% with complete CO<sub>2</sub> sequestration. A higher system pressure was found to be beneficial for raising the total efficiency, while increasing reaction temperatures in the chemical looping plant had only limited effects on the overall plant efficiency. The variation of the steam reactor temperature raised the power output of the gas turbine but lowered the power output of the steam turbine,

hence the two effects annulled each other. The analysis of the three-reactor chemical looping hydrogen system combined with a SOFC by Ozcan *et al.*<sup>121</sup> yielded a total energy efficiency of 45.39% for full power conversion and 56.92% for combined power/hydrogen production (ratio of 1.5/1). The two cases resulted in total exergy efficiencies of 34.98% and 45.05% respectively. The three-reactor chemical looping components alone contained an energy and exergy efficiency of 65.37% and 57.9% respectively. The highest amount of exergy destruction occurred in these elements due to the high process heat and the low hydrogen yield. Operating conditions required to maintain a complete conversion of syngas and the full re-oxidation of Fe/FeO to Fe<sub>3</sub>O<sub>4</sub> were determined. Hence, the ratio of steam to feed played a substantial role in terms of efficiency. Another important parameter was the ratio of utilized/stored hydrogen. The system efficiency was raised by roughly 10% points by increasing the amount of deliverable hydrogen to 40% of the total power output at the cost of electric efficiency.<sup>116,119,121</sup>

### Two-reactor chemical looping systems

Chen *et al.*<sup>90</sup> combined an iron oxide based two-reactor cyclic water-splitting system with a nickel oxide based chemical looping combustion unit. Syngas, produced by coal gasification in a shell gasifier, was used as a feed. The oxygen carrier was cycled between magnetite and wustite, thus syngas was not fully converted in the Fe fuel reactor. This lean gas reduced the nickel oxide in a second fuel reactor. The Ni-looping cycle was closed by oxidation with air. A pure hydrogen stream was generated in the iron cycle, while the nickel cycle yielded a pure carbon dioxide stream. Both streams were expanded separately for power generation and heat recovery. Two cases were investigated, the first using a supplementary firing unit to increase the turbine inlet temperature of the oxygen deplete air gas stream. Hence, carbon emissions were produced with a carbon capture efficiency of 69.54%. The second cases omitted the supplementary firing unit and enabled a complete carbon sequestration. The total equivalent efficiency for both cases were 70.75% and 57.9% respectively. Operation parameters were varied to optimize the electrical power efficiency and the hydrogen yield for co-generation. A flexible production ratio of hydrogen and electricity was achieved by varying the iron oxide solid flow rate. The hydrogen production efficiency was optimized by raising the steam flow and the temperatures in the steam reactor and air-reactor. The addition of an inert support to the nickel oxide improved the heat integration in the system and the hydrogen efficiency. Generally, a trade-off between hydrogen production efficiency and total efficiency was visible, hence a raised hydrogen yield was accompanied by a reduction in power efficiency. Both efficiencies were elevated with higher iron fuel reactor temperatures but at the cost of dropping carbon capture rates.

### Evaluation of operation parameter

Kathe *et al.*<sup>94</sup> analysed CH<sub>4</sub> as fuel in a three-reactor chemical looping hydrogen system using an iron oxide as oxygen carrier supported by Al<sub>2</sub>O<sub>3</sub>. Sensitivity analyses were executed to



maximize the hydrogen production efficiency. Initial thermodynamic simulations explained the benefits of a counter-current moving bed of reactants and solid material compared to a co-current fluidized system. This operation mode reached the thermodynamic constraints of redox cycling more efficiently, thus lowering the required steam/solid ratio to reach a defined solid conversion. Additionally, a higher fuel/solid ratio was possible, while maintaining a complete fuel conversion. An increase of support material increased the reactor outlet temperature and the solid conversion at a fixed solid/CH<sub>4</sub> ratio, which enabled a full conversion of methane without the risk of coking. A value of 50 wt% Al<sub>2</sub>O<sub>3</sub> was found to be optimum. The sensitivity analysis of Kang *et al.*<sup>122</sup> based on a linear empirical kinetic model suggested a content of 20 wt% of Fe<sub>2</sub>O<sub>3</sub> on ZrO<sub>2</sub> as suitable content. The reaction temperatures in the steam and fuel reactor reactors were key to minimize the bed material input. In both reactors, an increase in the temperature reduced the required solid inventory to an optimum temperature value. Beyond the optimum, a contrary behaviour was observed in the steam reactor, which was related to the exothermic nature of the reaction and the unfavourable equilibrium. Khan *et al.*<sup>123</sup> reported an increasing CO<sub>2</sub> and H<sub>2</sub> yield by raising the flow rates of steam, fuel and air until the stoichiometric values were reached. A Fe-oxygen carrier with 70 wt% MgAl<sub>2</sub>O<sub>4</sub> as support yielded the highest amounts of product gases, whereas higher values reduced the gas output. Higher solid mass flow rates enabled an enhanced H<sub>2</sub> and CO<sub>2</sub> yield until an optimum. Kathe *et al.*<sup>94</sup> analysed the steam conversion and outlet temperature, based on the oxygen carrier composition by adjusting the steam/solid ratio, steam conversion and outlet temperature. A re-oxidation to 11% of solid conversion yielded the highest amount of heat released at a high steam conversion efficiency. To complete the looping cycle, re-oxidation with air and a split of the solid flow was used, which bypasses the steam reactor and is directly transferred to the air reactor. The operation parameter air flow/fuel and split ratio of solid to oxidizer/bypassing-oxidizer were specified to facilitate auto-thermal operation conditions. The final plant layout for the simulation used a solid stream split of 0.15, which bypassed the steam reactor and was directly fed into the air reactor. In addition, 9% of the natural gas feed was directly injected into the air reactor in order to deliver the required heat duty. Hence, the process enabled a carbon capture rate of only 90%. A hydrogen production rate of 25 711 kg h<sup>-1</sup>, while utilizing 6196 kg h<sup>-1</sup> natural gas and a solid conversion of 60%, was calculated and utilized in several case studies. The highest system efficiencies were obtained by performing the process at a system pressure of 10 atm and exploiting a comprehensive energy recovery, which included gas expanders and a heat recovery steam generation system (HRSG). A cold gas efficiency of 77.6% and a thermal efficiency of 75.1% was obtained compared to 72.1% and 69.7% of the reference case (a conventional SMR system). The chemical looping system required 8% less natural gas and had a lower cooling demand and water requirement than the SMR case. Sanfilippo *et al.*<sup>85</sup> reported that an increase in the solid circulation rate required an external electric power input, but could further enhance the

hydrogen production capacity and increase the total efficiency. The simulation results indicated a total system efficiency in the range of 78–79%, while maintaining a full carbon capture rate. The system variation can alter the process design and capabilities significantly. For instance, an increase in thermal efficiency was obtained by reducing carbon capture capabilities, while omitting the heat recovery units and lowering the operation pressure reduced the system complexity and the capital system costs at the expense of a lower thermal efficiency.<sup>116</sup>

Zhang *et al.*<sup>124,125</sup> performed thermodynamic and exergetic analyses on a two-reactor chemical looping hydrogen system and different oxygen carriers using methane as feed. The hydrogen produced in the steam reactor was burned with pure oxygen from an air separation unit. The generated steam was used for power generation in a steam turbine. The residual heat after the turbine is re-fed to supply the endothermic reaction in the fuel reactor. In a second system, the power generation unit consisted of a high-pressure and a low-pressure steam turbine. The single turbine and double turbine systems yielded a thermal efficiency of 55.5% and 59.8% respectively, which were 0.9% and 5.2% points higher than conventional combined cycles. The authors assumed a carbon capturing without any additional energy penalties by generation of a stream of CO<sub>2</sub> and H<sub>2</sub>O and by the reduction between Fe<sub>3</sub>O<sub>4</sub> and FeO in the fuel reactor. Hence, the efficiency was further raised by 7–9% points. The work does not discuss how the pure outlet stream in the fuel reactor is produced by the reduction of magnetite with CH<sub>4</sub>. The two reactor chemical looping system was integrated into a coal based methanol production system.<sup>126</sup> Heat from the coal gasification, which supplied syngas for the methanol synthesis, was used in the chemical looping system for hydrogen supply. The reduction in the looping fuel reactor was performed with natural gas. The exergy analysis indicated that utilizing the coal gasification heat reduced the exergy destruction compared to the conventional operation of combusting parts of the natural gas. Metals suitable for use as oxygen carrier such as Ni, Zn and Fe enable a steam oxidation at low equilibrium temperatures and thus assist in the conversion of chemical energy with very high second law efficiency. Energy savings are made possible by minimizing the exergy destruction and the lowered consumption of natural gas.<sup>118,125,126</sup>

### Unconventional feedstocks

Cormos *et al.*<sup>127</sup> investigated the replacement of coal with biomass (sawdust) as feedstock. The external gasifier still required a high amount of coal (70%) in the feed because the commercial availability of industrial scale gasifier solely for biomass is limited. The iron based chemical looping systems achieved the highest electric efficiency and a carbon capture rate >99% with the lowest carbon capture energy penalty. The reference systems based on gas liquid absorption achieved an energy efficiency 5.7% points lower than the chemical looping plant. Yan *et al.*<sup>128</sup> added a SOFC unit to the biomass-fed three-reactor chemical looping hydrogen system using coal and wheat straw. Post combustion of the anode off-gas, a steam turbine for power generation and CO<sub>2</sub> sequestration were considered. The



amount of hydrogen and the total thermal efficiency were raised by increasing the gasification temperature, which improved the carbon conversions efficiency. Upon reaching a complete biomass conversion no further positive temperature effects were visible. The same behaviour was found regarding the S/C ratio in the gasification unit. Below the ratio for a complete biomass conversion an increased amount of steam indicated positive effects on the plant performance, while an excess of steam lowered the total efficiency. The increased oxygen carrier circulation rate raised the amount of produced hydrogen per mass biomass. However, upon reaching the point of full oxidation of CO to CO<sub>2</sub>, only hydrogen was consumed for the reduction reaction in the fuel reactor, which lowered the plant performance. The system evaluation yielded a total energy efficiency of 39.9% and total exergy efficiency of 37.6% with a carbon capture rate of 96%. The comparison of a three-reactor system and a two-reactor system using the same feed gas conveyed the main difference in the behaviour of the fuel reactor temperature.<sup>129</sup> Higher temperatures in a three-reactor system increased both the total thermal efficiency as well as the amount of hydrogen per mass feed. The endothermic reduction reaction was thermodynamically promoted at higher temperatures and enabled a greater reduction degree of the oxygen carrier. In the two-reactor system the lack of an air oxidation to meet the required heat supply had to be substituted by combustion of biomass. Thus the thermal efficiency and the amount of hydrogen per mass feed decreased. Both systems enabled a carbon capture rate of >95%. The two-reactor system had a higher cold gas efficiency of 60% compared to 54% and a higher hydrogen production capability of 0.76 Nm<sup>3</sup> kg<sup>-1</sup> compared to 0.66 Nm<sup>3</sup> kg<sup>-1</sup>. A calcium based sorption enhanced looping system used as reference had a higher cold gas efficiency of 72% with 0.73 Nm<sup>3</sup> kg<sup>-1</sup> of hydrogen but a lower carbon capture efficiency of 92%.

The application of bioethanol (12% ethanol and 88% water) from fermentation processes was examined by Cormos.<sup>130</sup> High system efficiencies were achieved in two system variations, (i) an upstream reforming reactor with a syngas based chemical looping and (ii) a direct bioethanol chemical looping system. The total energy efficiencies were 59.78% with an electric power output of 13% and 63.66% with an electric power output of 7%. The carbon capture rate was >99% in both plant simulations. The conventional reforming system with a gas-liquid absorption as a benchmark yielded efficiencies in the range of 53–58%, while enabling a carbon capture efficiency of >99%.

Gopaul *et al.*<sup>131</sup> investigated the conversion of unconventional biomass. Three types of biomass were evaluated, assuming the absence of sulphur and nitrogen components. Poultry litter was chosen over wood pellets and oak pellets because of the higher hydrogen yield. Tar products formed during the biomass gasification process were not separated but fed into the chemical looping fuel reactor. Reduced iron and remaining tars were separated from the gaseous products and fully oxidized in the steam reactor. Both reactors yielded a hydrogen rich synthesis gas (H<sub>2</sub>, CO<sub>2</sub>, CO, CH<sub>4</sub>, H<sub>2</sub>O), which was utilized in a reformer at 500 °C to oxidize the residual methane and maximize the hydrogen output. A hydrogen yield

of 1.60 kmol kmol<sub>feed</sub><sup>-1</sup> diluted with 34% CO<sub>2</sub> and 2.3% H<sub>2</sub>O with a final hydrogen purity of 63% was obtained. A reference water-gas-shift system generated 0.97 kmol kmol<sub>feed</sub><sup>-1</sup>. Sanz *et al.*<sup>132</sup> calculated the hydrogen production cost using a two-reactor chemical looping system. A syngas stream with a mass flow of 27 820 kg h<sup>-1</sup> produced by biomass gasification was assumed as feed. The feed stream contained high amounts of steam and a low reducing potential, thus the stream was cooled-down to condense the water before entering the fuel reactor. Two reactors were assumed in reduction mode per each reactor in oxidation mode, with a total of 12 reactors for handling the used feed stream. The system calculations were based on thermodynamic considerations of the Baur-Glaessner diagram. The reduction off gas was burned to provide heat for the system. Under the consideration of equipment costs, amortization and operation costs a hydrogen production of 4786 t per year with production costs of 2.13 € kg<sup>-1</sup> hydrogen were obtained. Comparative data based on a conventional water-gas-shift system resulted in production costs of 1.76 € kg<sup>-1</sup>. The techno-economic evaluation indicated higher investment, operation and maintenance costs (O&M) for chemical looping hydrogen systems compared to state-of-the art technology. The high degree of technological complexity and the small number of large-scale applications are the two main cost factors. However, under the consideration of costs per energy output the direct biomass chemical looping system has the lowest costs due to the high system efficiency and low carbon capture energy penalty.<sup>127,132</sup>

Edrisi *et al.*<sup>83,133</sup> proposed a three-reactor chemical looping technology to produce pure streams of hydrogen and nitrogen as feed for an ammonia production plant and a third stream of pure carbon dioxide for sequestration. In the fuel reactor an iron based oxygen carrier completely converted CH<sub>4</sub> to CO<sub>2</sub> at 726 °C. In the steam reactor a pure hydrogen stream was produced by partially oxidizing iron with steam at 723 °C. In the air reactor, a pure nitrogen stream was generated at 880 °C by fully consuming oxygen in the air stream and completely oxidizing the solid material. The fuel and air fed and the oxygen carrier fed were varied to optimize the operation conditions with regard to the product streams. The ratio OC/CH<sub>4</sub> of 4.0 and the ratio air/CH<sub>4</sub> of 3.206 were found to enable the production of pure streams, while maximizing the reduction degree of the solid material. MgAl<sub>2</sub>O<sub>4</sub> was added as support material with a ratio of MgAl<sub>2</sub>O<sub>4</sub>/CH<sub>4</sub> of 6 to improve the heat integration of the system.<sup>133</sup> The simulation yielded a hydrogen production efficiency of 80.2%.<sup>133</sup>

## 6. Kinetic studies

Several papers have been published on the kinetic characterization of the reduction and oxidation of iron based oxygen carriers. Besides pure iron oxygen, supported samples and the interactions between the active components and the supporting components were studied. The kinetic data and redox models gained is important information for modelling large-scale chemical looping plants and process optimizations. The presented data is summarized in Table 3.





Table 3 Overview of presented kinetic data

Sample	Sample properties	Reaction parameters	Reaction model	Rate constants reduction	Rate constants oxidation	$E_a$ reduction	$E_a$ oxidation	Reference
$\text{Fe}_2\text{O}_3/\text{ZrO}_2$	Particle size = 150–300 $\mu\text{m}$ , porosity 50%, BET 2.2 $\text{m}^2 \text{g}^{-1}$	550–800 °C, $\text{H}_2$ 2 vol%, $\text{H}_2\text{O}$ 5 vol%		Step 1 = $7.8 \times 10^{-3} \text{ s}^{-1}$ , step 2 = $1.4 \times 10^{-3} \text{ s}^{-1}$ , step 3 = $2.7 \times 10^{-4} \text{ s}^{-1}$	$4.4 \times 10^{-4} \text{ s}^{-1}$			37
$\text{Fe}_2\text{O}_3/\text{CeO}_2$	Particle size = 150–300 $\mu\text{m}$ , porosity 32%, BET 1.0 $\text{m}^2 \text{g}^{-1}$	550–800 °C, $\text{H}_2$ 2 vol%, $\text{H}_2\text{O}$ 5 vol%		Step 1 = $4.1 \times 10^{-3} \text{ s}^{-1}$ , step 2 = $1.4 \times 10^{-3} \text{ s}^{-1}$ , step 3 = $7.0 \times 10^{-4} \text{ s}^{-1}$	$9.6 \times 10^{-4} \text{ s}^{-1}$			37
$\text{Fe}_2\text{O}_3/\text{YSZ}$	Particle size = 150–300 $\mu\text{m}$ , porosity 50%, BET 1.5 $\text{m}^2 \text{g}^{-1}$	550–800 °C, $\text{H}_2$ 2 vol%, $\text{H}_2\text{O}$ 5 vol%		Step 1 = $2.7 \times 10^{-2} \text{ s}^{-1}$ , step 2 = $2.8 \times 10^{-3} \text{ s}^{-1}$ , step 3 = $4.1 \times 10^{-4} \text{ s}^{-1}$	$1.2 \times 10^{-3} \text{ s}^{-1}$			37
$\text{Fe}_2\text{O}_3/\text{GDC}$	Particle size = 150–300 $\mu\text{m}$ , porosity 38%, BET 1.0 $\text{m}^2 \text{g}^{-1}$	550–800 °C, $\text{H}_2$ 2 vol%, $\text{H}_2\text{O}$ 5 vol%		Step 1 = $1.9 \times 10^{-2} \text{ s}^{-1}$ , step 2 = $2.3 \times 10^{-3} \text{ s}^{-1}$ , step 3 = $9.1 \times 10^{-4} \text{ s}^{-1}$	$1.5 \times 10^{-3} \text{ s}^{-1}$			37
$\text{Fe}_2\text{O}_3/$ $\text{Ce}_{0.5}\text{Zr}_{0.5}\text{O}_2$ 80/ 20 wt%, Mo 1–5 wt%	Surface area $\text{m}^2 \text{g}^{-1}$ , 0 Mo = 68.67, 1 Mo = 55.54, 2 Mo = 56.67, 3 Mo = 56.66, 4 Mo = 50.56, 5 Mo = 48.44	650, 750, 800 °C, 5–15 °C $\text{min}^{-1}$ , $\text{H}_2$				Peak 1/peak 2 kJ $\text{mol}^{-1}$ , 0 Mo = 110.6/69.2, 1 Mo = 107.4/67.1, 2 Mo = 108.5/63.6, 3 Mo = 100.3/70.4, 4 Mo = 99.6/60.4, 5 Mo = 87.3/55.4		43
$\text{Fe}_2\text{O}_3/$ $\text{Ce}_{0.5}\text{Zr}_{0.5}\text{O}_2$ 80/ 20 wt%, Mo 1–5 wt%	Surface area $\text{m}^2 \text{g}^{-1}$ , FCZ = 68.7, FCZ-1 Mo = 60.2, FCZ-2 Mo = 53.5, FCZ-3 Mo = 65.4, FCZ-4 Mo = 66.2, FCZ-5 Mo = 60.3	750, 800 °C, $\text{H}_2$				Peak 1/peak 2 kJ $\text{mol}^{-1}$ , FCZ = 110.6/69.2, FCZ-1 Mo = 54.9/46.5, FCZ-2 Mo = 63.2/54.8, FCZ-3 Mo = 51.4/48.4, FCZ-4 Mo = 87.6/55.7, FCZ-5 Mo = 94.6/41.6		44
$\text{Fe}_2\text{O}_3/\text{Al}_2\text{O}_3/$ $\text{SiO}_2/\text{CaO}$	Surface area $\text{m}^2 \text{g}^{-1}$ , $F_1$ ( $\text{Fe}_2\text{O}_3$ ) = 32.2, $F_2$ (Fe-5Al) = 48.8, $F_3$ (Fe-5Al-5Ca) = 48.3, $F_4$ (Fe-5Al-5Ca) = 74.1, $F_5$ (Fe-5Al-7.5Si) = 93.1, $F_6$ (Fe-7.5Si-2.5Ca) = 74.8, $F_7$ (Fe-7.5Si-7.5Ca) =	650, 750, 800 °C, $\text{H}_2$				Peak 1/peak 2 kJ $\text{mol}^{-1}$ , $F_1$ = 104.4/37.0, $F_2$ = 116.3/111.9, $F_3$ = 125.3/72., $F_4$ = 89.3/96.9, $F_5$ = 96.4/207.3, $F_6$ = 114.8/137.5, $F_7$ = 78.9/270.5, $F_8$ = 101.6/332.5, $F_9$ = 93.7/332.6		46



Table 3 (Contd.)

Sample	Sample properties	Reaction parameters	Reaction model	Rate constants reduction	Rate constants oxidation	$E_a$ reduction	$E_a$ oxidation	Reference
Cu-Fe	79.3, $F_8$ (Fe-5Al-7.5Si-2.5Ca) = 95.6, $F_9$ (Fe-5Al-5Si-5Ca) = 77.6 Particle size: $Fe_{16}Cu_{15}$ -IWI = 115 $\mu\text{m}$ , $Fe_{17}Cu_{15}$ -CoPr = 143 $\mu\text{m}$	530–825 °C, $H_2O$ = 5.5 vol%	IWI = Avrami Erofeev (A2), CoPr = nuclei growth model				IWI = $46 \pm 4$ kJ $\text{mol}^{-1}$ , CoPr = $51 \pm 5$ kJ $\text{mol}^{-1}$	61
$CoWO_4$ , $CoWO_4$ -10 wt% Ni, $CoWO_4$ -10 wt% La		850–950 °C, $CH_4$ = 2–8 vol%		1 <sup>st</sup> cycle, CoW = $0.2\text{--}1.6 \times 10^{-3}$ $\text{min}^{-1}$ , CoW-Ni = $1.2\text{--}3.2 \times 10^{-3}$ $\text{min}^{-1}$ , CoW-La = $0.5\text{--}1.7 \times 10^{-3}$ $\text{min}^{-1}$ , 2 <sup>nd</sup> cycle CoW = $0.7\text{--}1.5 \times 10^{-3}$ $\text{min}^{-1}$ , CoW-Ni = $1.3\text{--}3.0 \times 10^{-3}$ $\text{min}^{-1}$ , CoW-La = $1.1\text{--}2.9 \times 10^{-3}$ $\text{min}^{-1}$			1 <sup>st</sup> cycle CoW = 221 kJ $\text{mol}^{-1}$ , CoW-Ni = 134 kJ $\text{mol}^{-1}$ , CoW-La = 140 kJ $\text{mol}^{-1}$ , 2 <sup>nd</sup> cycle CoW = 151 kJ $\text{mol}^{-1}$ , CoW-Ni = 95 kJ $\text{mol}^{-1}$ , CoW-La = 111 kJ $\text{mol}^{-1}$	63
$Fe_2O_3$	Particle size = 450–900 $\mu\text{m}$	1073 K, CO		Reaction front reaction rate $Fe_2O_3/Fe_3O_4$ = 5.280, $\text{mol m}^{-3} \text{s}^{-1}$ , $Fe_3O_4/FeO$ = 3.329 $\text{mol m}^{-3} \text{s}^{-1}$ , $FeO/Fe$ = 4.379 $\text{mol m}^{-3} \text{s}^{-1}$				135
$Fe_2O_3$ , 50 wt% $Al_2O_3$	Particle size = 100–200 $\mu\text{m}$	973–1173 K, CO	Pure $Fe_2O_3$ : 2D phase boundary, $Fe_2O_3/Al_2O_3$ diffusion controlled				Pure $Fe_2O_3$ = $33.3 \pm 0.8$ kJ $\text{mol}^{-1}$ , $Fe_2O_3/Al_2O_3$ = $41.1 \pm 2.0$ kJ $\text{mol}^{-1}$	136
$Fe_2O_3$ -FeO, FeO-Fe	Density = 3.80 g $\text{cm}^{-3}$ , bulk density = 2.46 g $\text{cm}^{-3}$ , BET surface area = 2.6 $\text{m}^2 \text{g}^{-1}$ , particle size = 125–300 $\mu\text{m}$	$CH_4$ = 10–55 vol%, 825–900 °C, $H_2$ = 10–25 vol%, 750–900 °C, CO = 10–25 vol%, 750–900 °C	Phase boundary controlled, contracting sphere $R_3$ contracting infinite slab $R_1$				$CH_4$ : $Fe_2O_3$ -FeO = 251 kJ $\text{mol}^{-1}$ , FeO-Fe = 230 kJ $\text{mol}^{-1}$ , $H_2$ : $Fe_2O_3$ -Fe = 59 kJ $\text{mol}^{-1}$ , CO: $Fe_2O_3$ -Fe = 37 kJ $\text{mol}^{-1}$	138
$Fe$ - $Fe_3O_4$ , $Fe$ - $Fe_2O_3$	Density = 3.80 g $\text{cm}^{-3}$ , bulk density = 2.46 g $\text{cm}^{-3}$ , BET surface area = 2.6 $\text{m}^2 \text{g}^{-1}$ , particle size = 125–300 $\mu\text{m}$	$H_2O$ = 5, 10, 15, 20 vol%, air = 3, 6, 9, 12 vol%, 600–900 °C	Phase boundary controlled, contracting sphere $R_3$				$H_2O$ : $Fe$ - $Fe_3O_4$ = 27 kJ $\text{mol}^{-1}$ , air: $Fe$ - $Fe_2O_3$ = 7 kJ $\text{mol}^{-1}$	138
$Fe_2O_3$ - $Fe_3O_4$ , natural ore	Mean particle ~ 92.5 $\mu\text{m}$ , surface	800–950 °C, $CH_4$ = 33 vol%	Phase boundary					215 kJ $\text{mol}^{-1}$



Table 3 (Contd.)

Sample	Sample properties	Reaction parameters	Reaction model	Rate constants reduction	Rate constants oxidation	$E_a$ reduction	$E_a$ oxidation	Reference
$\text{Fe}_2\text{O}_3/\text{Fe}_3\text{O}_4/\text{FeO}$	area $\sim 5.74 \text{ m}^2 \text{ g}^{-1}$ Particle size = 100–300 $\mu\text{m}$	700–950 °C, $\text{H}_2$ 5, 10, 20 vol%				Double-step: $R_1 = 6.15 \pm 0.2 \text{ kJ mol}^{-1}$ , $R_2 = 56.9 \pm 1.1 \text{ kJ mol}^{-1}$ , single-step = $64.15 \pm 0.5 \text{ kJ mol}^{-1}$		140
$\text{Fe}_2\text{O}_3/\text{Fe}_3\text{O}_4/\text{FeO}$	Particle size = 60–100 $\mu\text{m}$	750–825 °C, $\text{CH}_4 = 15, 20, 25 \text{ vol}\%$	Multi-step kinetics: first order, nucleation growth			$R_1 = 34.4 \pm 0.5 \text{ kJ mol}^{-1}$ , $R_2 = 39.3 \pm 1.6 \text{ kJ mol}^{-1}$		141
$\text{Fe}_{4.5}/\text{Al}$	Particle size = 75–150 $\mu\text{m}$	973–1073 K, $\text{CH}_4 = 0.5 \text{ vol}\%$	$\text{Fe}_2\text{O}_3/\text{Fe}_3\text{O}_4 =$ nuclei growth D2, D3, $\text{Fe}_3\text{O}_4/\text{FeAl}_2\text{O}_4 =$ Jander eqn D3			$\text{Fe}_2\text{O}_3/\text{Fe}_3\text{O}_4 = 24.96 \text{ kJ mol}^{-1}$ , $\text{Fe}_3\text{O}_4/\text{FeAl}_2\text{O}_4 = 171.24 \text{ kJ mol}^{-1}$		142
$\text{Fe}_{2.5}/\text{Al}$	Particle size = 75–150 $\mu\text{m}$	973–1073 K, $\text{CH}_4 = 0.5 \text{ vol}\%$	$\text{Fe}_2\text{O}_3/\text{Fe}_3\text{O}_4 =$ nuclei growth D2, D3, $\text{Fe}_3\text{O}_4/\text{FeAl}_2\text{O}_4 =$ Jander eqn D3			$\text{Fe}_2\text{O}_3/\text{Fe}_3\text{O}_4 = 24.77 \text{ kJ mol}^{-1}$ , $\text{Fe}_3\text{O}_4/\text{FeAl}_2\text{O}_4 = 209.16 \text{ kJ mol}^{-1}$		142
$\text{Fe}_2\text{O}_3$ on 20% $\text{TiO}_2$	Particle size < 250 $\mu\text{m}$	600–950 °C, $\text{H}_2$	First order reaction $F_1$ phase boundary controlled $R_3$			33.8 $\text{kJ mol}^{-1}$		144
$\text{Fe}_2\text{O}_3/\text{CuO}$ on $\text{Al}_2\text{O}_3$ 60/20/20	Particle size = < 250 $\mu\text{m}$	600–950 °C, $\text{H}_2$	First order reaction $F_1$ 3D diffusion Jander equation			42.6 $\text{kJ mol}^{-1}$		144
$\text{FeOOH}$ , $\text{Fe}_2\text{O}_3$ , $\text{Fe}_3\text{O}_4$	Particle size = 62–125 $\mu\text{m}$ , surface area = $93.5 \text{ m}^2 \text{ g}^{-1}$ , $\text{Fe}_2\text{O}_3 = 7.5 \text{ m}^2 \text{ g}^{-1}$ , $\text{Fe}_3\text{O}_4 = 1.7 \text{ m}^2 \text{ g}^{-1}$	850 °C, $\text{H}_2 = 10 \text{ vol}\%$ , $\text{O}_2 = 10 \text{ vol}\%$		$\text{FeOOH} = 0.0740 \text{ min}^{-1}$ , $\text{Fe}_2\text{O}_3 = 0.0435 \text{ min}^{-1}$ , $\text{Fe}_3\text{O}_4 = 0.1274 \text{ min}^{-1}$	$\text{FeOOH} = 0.0857 \text{ min}^{-1}$ , $\text{Fe}_2\text{O}_3 = 0.0450 \text{ min}^{-1}$ , $\text{Fe}_3\text{O}_4 = 0.0894 \text{ min}^{-1}$		$\text{FeOOH} = 21.5 \text{ kJ mol}^{-1}$ , $\text{Fe}_2\text{O}_3 = 48.9 \text{ kJ mol}^{-1}$ , $\text{Fe}_3\text{O}_4 = 11.0 \text{ kJ mol}^{-1}$	145
$\text{Fe}_{0.933}\text{O}$ , 0.05 < X < 0.2	Particle size = 300–425 $\mu\text{m}$ , 150–212 $\mu\text{m}$	1073 K, CO = 50 vol%	Linear correlation			$\text{H}_2 = 62 \pm 9 \text{ kJ mol}^{-1}$ , CO = $83 \pm 14 \text{ kJ mol}^{-1}$		146
$\text{Fe}_{0.933}\text{O}$ , 0.2 < X < 1	Particle size = 300–425 $\mu\text{m}$ , 150–212 $\mu\text{m}$	1073 K, CO = 50 vol%	Random pore model			$\text{H}_2 = 62 \pm 19 \text{ kJ mol}^{-1}$ , CO = $64 \pm 8 \text{ kJ mol}^{-1}$		146
$\text{Fe}_2\text{O}_3/\text{Al}_2\text{O}_3/\text{CeO}$		450–600 °C, $\text{H}_2/\text{CH}_4$ mixture	Johnson-Mehl-Avrami-Kolmogorov, Levenberg-Marquardt algorithm			Differential kinetics = $75 \text{ kJ mol}^{-1}$ , integral kinetics = $48 \pm 4 \text{ kJ mol}^{-1}$	Differential kinetics = $0 \text{ kJ mol}^{-1}$ , integral kinetics = $0 \text{ kJ mol}^{-1}$	147
Fe		973–1473 K, $\text{H}_2\text{O}_g = 4.9, 9.4, 17.2 \text{ vol}\%$ , $\text{CO}_2 = 4, 7.7, 14.3 \text{ vol}\%$				$\text{H}_2\text{O} = 47.3 \text{ kJ mol}^{-1}$ , $\text{CO}_2 = 32.8 \text{ kJ mol}^{-1}$		148

Zhu and Hua *et al.*<sup>134–136</sup> analysed the reduction stage distribution in a fixed bed reactor under fuel breakthrough operation. Reductions were performed at 900 °C with a mixture of CO and CO<sub>2</sub> in a tubular reactor divided into 10 separated segments by quartz cotton. The oxygen carrier analyses indicated an uneven solid conversion caused by the reaction front with their different reduction kinetics moving through the reactor (Fig. 12). The first two layers contained Fe and FeO with a mass percentage of 85 wt% and 7 wt% respectively, while FeO and Fe<sub>3</sub>O<sub>4</sub> were found in the remaining 8 layers. In the consecutive steam oxidation the segments 1–6 contributed to 90% of the produced hydrogen. Hertel *et al.*<sup>137</sup> identified the two zones with a stable conversion which corresponded to the reduction zones of Fe<sub>3</sub>O<sub>4</sub>/FeO and FeO/Fe described by the Baur-Glaessner diagram. Stable zones were separated by transition periods, which are induced by limited gas–solid reaction rates. The simulation results indicated that the travel time of the reaction zones is 3–4 orders of magnitude of the gas phase movement. Zhu *et al.*<sup>134</sup> separated each reduction step by adapting the CO/CO<sub>2</sub> feed-ratio according to the phase equilibrium. The reaction rates of the reductions Fe<sub>2</sub>O<sub>3</sub>/Fe<sub>3</sub>O<sub>4</sub> and FeO/Fe possessed similar characteristics with a rapid increase to a stable maximum. Both steps ended with a fast decline until the conversion rate dropped. Fe<sub>2</sub>O<sub>3</sub> was completely reduced, while 36% of FeO remained unreacted after step 3. The second step Fe<sub>3</sub>O<sub>4</sub>/FeO had the slowest reduction rate, approximately one order of magnitude lower than the other two, with a continuous linear decrease and a reaction stop at a solid conversion of roughly 38%, meaning that it should be the focus of material optimization. The reaction front velocities were in the order Fe<sub>2</sub>O<sub>3</sub>/Fe<sub>3</sub>O<sub>4</sub> > FeO/Fe > Fe<sub>3</sub>O<sub>4</sub>/FeO. Reduction simulations of Fe<sub>2</sub>O<sub>3</sub> with CO in a temperature range of 973–1173 K were executed with the phase boundary kinetic model.<sup>136</sup> However, the addition of 50 wt% Al<sub>2</sub>O<sub>3</sub> revealed a transition from a phase boundary to a diffusion controlled kinetic model to better match the experimental results. An explanation was provided by SEM analysis. Larger pores were found in the reduced pure sample, whereas the supported oxygen carrier

possessed grain clusters with reduced porosity, which are expected to hamper the gas diffusion. Kang *et al.*<sup>138</sup> performed an extensive isothermal kinetic investigation on iron oxide supported with 80 wt% ZrO<sub>2</sub> at temperatures of 740–900 °C. The reduction with methane was separated into two steps Fe<sub>2</sub>O<sub>3</sub> → FeO and FeO → Fe, while the reduction with H<sub>2</sub> and CO was considered a one-stage reaction. According to the Hancock and Sharp method the phase-boundary controlled mechanism was suitable for describing all reduction and oxidation reactions except the second step of the methane reduction, which showed a linear behaviour. Nasr *et al.*<sup>139</sup> studied the reduction kinetic of a natural iron ore with CH<sub>4</sub>. Reactions were performed in a TGA in the temperature range of 800–950 °C. Based on the Hancock and Sharp method, a phase boundary controlled mechanism was used to describe the solid conversion. The impurities in the natural ore, mainly SiO<sub>2</sub>, CaO, MgO and Al<sub>2</sub>O<sub>3</sub> were expected to lower the activation energy which resulted in a value of 215 kJ mol<sup>-1</sup> for the reduction reaction.

Monazam *et al.*<sup>140</sup> analysed the reduction kinetics of hematite particles to wüstite with 20% hydrogen in a thermogravimetric analysis system (Fig. 13). Changing values were obtained for the activation energies, depending on the solid conversion initially decreased from 23 to 8 kJ mol<sup>-1</sup> (0.025 < solid conversion < 0.1) followed by an increase to 47 kJ mol<sup>-1</sup> (0.1 < solid conversion < 0.33) and finally declined to 30 kJ mol<sup>-1</sup> (0.33 < solid conversion < 1), suggesting a double-step reduction kinetic. In a temperature range of 700–800 °C a 1D nucleation growth model including an induction period was used to describe the reaction process. At temperatures beyond 900 °C the induction period diminished and the single-step nuclei growth reaction became dominant. A similar approach was used for the reduction kinetics of hematite with CH<sub>4</sub> (15–35 vol%).<sup>141</sup> However, the kinetic models did not properly describe the reaction. The analysis of the activation energy corresponding to the solid reduction degree indicated a descending trend from 56 to 36 kJ mol<sup>-1</sup> with the conversion progressing from 0.1 to 0.5. Hence, a multi-step parallel model with a superposition of a first order reaction and a nucleation model was applied. Activation energies for the two reaction steps of 34.4 ± 0.5 kJ mol<sup>-1</sup> and 39.3 ± 1.6 kJ mol<sup>-1</sup> were obtained. Zhang *et al.*<sup>142,143</sup>

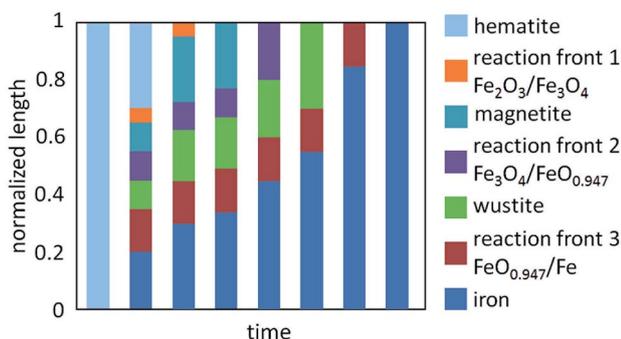


Fig. 12 Schematic temporal progression of the reaction fronts in a fixed bed at 800 °C.<sup>135</sup> Reprinted and adapted with permission from X. Hua, W. Wang, and Y. Hu, Analysis of reduction stage of chemical looping packed bed reactor based on the reaction front distribution, *J. Mater. Cycles Waste Manag.*, 2014, 16, 583–590, Copyright (2014), Springer Japan.

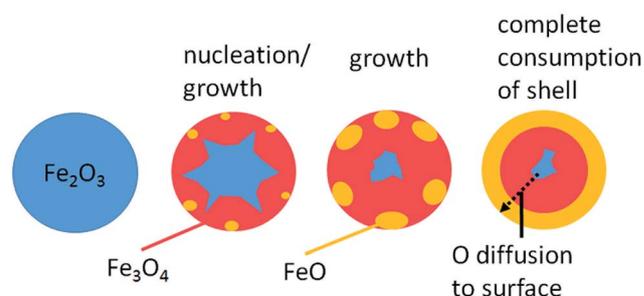


Fig. 13 Schematic representation of the progression of the reduction of Fe<sub>2</sub>O<sub>3</sub> to FeO.<sup>140</sup> Reprinted and adapted with permission from E. R. Monazam, R. W. Breault, and R. Siriwardane, Kinetics of hematite to wüstite by hydrogen for chemical looping combustion, *Energy Fuels*, 2014, 28, 5406–5414, Copyright (2014), American Chemical Society.



analysed the reduction kinetics of 25 wt% and 45 wt% Fe<sub>2</sub>O<sub>3</sub> on Al<sub>2</sub>O<sub>3</sub> with a low concentrated CH<sub>4</sub> feed of 0.5%. The reduction step of Fe<sub>2</sub>O<sub>3</sub> → Fe<sub>3</sub>O<sub>4</sub>, with a phase change of rhombohedral to cubic, was described with the 2D growth of nuclei up to solid conversions of 0.7. At a higher solid conversion, a 3D nuclei growth Avrami Erofeev mechanism yielded better fitting results. The reduction for magnetite to FeAl<sub>2</sub>O<sub>4</sub> was significantly slower and described by a diffusion-controlled reaction mechanism. The 3D diffusion Jander reaction mechanism fitted best and indicated a strong correlation between the iron content of the sample and the reduction reactivity. Ksepko *et al.*<sup>144</sup> compared the reduction of Fe<sub>2</sub>O<sub>3</sub> supported on 20 wt% TiO<sub>2</sub> with a bimetallic sample of Fe<sub>2</sub>O<sub>3</sub>/CuO supported on Al<sub>2</sub>O<sub>3</sub> (60/20/20 wt%) with hydrogen in the temperature range of 600–950 °C. The reaction model of a first order reaction and a phase boundary controlled mechanism was found to be suitable for the Fe<sub>2</sub>O<sub>3</sub> sample, whereas the 3D diffusion Jander equation and the first order reaction yielded better results for the reduction of the bimetallic oxygen carrier. Jeong *et al.*<sup>145</sup> investigated the redox kinetics of iron oxides with three different main iron phases FeOOH, Fe<sub>2</sub>O<sub>3</sub> and Fe<sub>3</sub>O<sub>4</sub> with initial surface area of 93.5, 7.5 and 1.7 m<sup>2</sup> g<sup>-1</sup>. Hence, the transformation to bulk Fe<sub>3</sub>O<sub>4</sub> with a homogenizing of the different grain sizes was observed. The Jander equation was used to determine the kinetic parameter of the redox reaction. It was suggested that the reduction and oxidation of magnetite led to the highest reaction rate and lowest activation energy. Liu *et al.*<sup>146</sup> investigated the reduction kinetic of wustite particles in a fluidized bed using H<sub>2</sub> and CO. The solid reduction and the formation of metallic iron were divided into three separated conversion stages. The first stage with a solid conversion of 0 < X < 0.05 was identified as the partial depletion of oxygen in wustite. In the second stage 0.05 < X < 0.2 surface nucleation and the formation of isolated Fe grains took place. The reaction mechanism was controlled by dissociative adsorption and described by a linear reaction model. The final stage 0.2 < X < 1 was identified as the thickening of the iron particle with a final formation of a uniform Fe product layer and the changing of the reaction rate to a diffusion limiting system. The random pore model was able to describe the solid conversion reactions. The three stages were also measured when mixtures of H<sub>2</sub> and CO were applied resulting in a superposition of their non-interacting reduction rates. The hydrogen reduction rate was significantly faster than the CO reduction rate but the overall reduction rate was dominated by the slow diffusion of oxygen atoms in the solid particle.

Plou *et al.*<sup>147</sup> incorporated the influence of thermodynamic equilibriums into differential kinetic measurements to reduce deviations during the initial steam oxidation phase and the transition phases. The Johnson–Mehl–Avrami–Kolmogorov model, which provided the differential kinetic data was amalgamated into an integral data fit method based on the Levenberg–Marquardt method by minimizing the sum of square deviation. The simulations were able to take the equilibrium compositions and the moving reaction fronts in the fixed bed reactor into account and yielded improved fitting results. The calculations suggested a decrease of the oxidation activity after consecutive cycles, meaning that the oxidation reaction was

primarily responsible for the degradation of the oxygen carrier. Stehle *et al.*<sup>148</sup> performed steam oxidation experiments with a pristine iron rod. At low steam concentrations in the oxidizing feed, a linear correlation with the produced hydrogen was observed, however higher steam concentrations did not further improve the hydrogen production rate. A rapid increase in the production rate in the range of 1075–1175 K was found, which suggested a transition from the kinetical limited reaction to a diffusion-limited regime. A detailed investigation of the iron oxide layer after the oxidation indicated an average layer thickness of 5 μm, which deviated significantly in its width after consecutive cycles.<sup>149</sup> The oxide layer possessed a stoichiometric ratio of Fe<sub>3</sub>O<sub>4</sub> at the outermost point. A linear decrease to an O/Fe ratio of 0.8 within 80% of the layer width was detected, followed by a sharp bend to zero. 70% of the surface thickness was able to participate in the oxygen exchange reaction with a continuous gradient of oxygen atom concentration, which explained the dominant diffusion limitation. The oxidation of an iron powder with steam in a fluidized bed environment by Singh *et al.*<sup>150</sup> showed two bends representing the transition of Fe/FeO and FeO/Fe<sub>3</sub>O<sub>4</sub>. The experimental validation results revealed some deviations from the thermodynamic simulations due to non-ideal mixing of the bed material. At a reaction temperature of 660 °C the influence of kinetic hindrance was clearly visible resulting in a low conversion. A temperature increase to 960 °C diminished the kinetic effects.

## 7. Conclusions

Hydrogen production by chemical looping is a very promising technology, where significant scientific progress has been made over the last years. In the area of oxygen carrier development, the focus was on iron as the active component. New synthesis methods and material composition have shown that mixtures with high melting support materials or the incorporation of iron into thermally stable structures, such as perovskites, can successfully enable long-term cycle stability and a satisfying material strength. The addition of different metal oxides, which act as promoters, enhanced the redox properties and permitted the fabrication of specialized and selective oxygen carrier design for specific process conditions and feed stocks. Two- and three reactor moving bed systems in a power range up to 50 kW have been successfully operated. The experiments revealed stable conditions, a high fuel conversion and a continuous production of hydrogen with high purities. Although, some reactor and system developments using fixed bed technology were presented, most research was related to interconnected fluidized beds. Emphasis has been put on the modification of the chemical looping hydrogen process utilizing different feedstocks. Experimental studies and simulations have shown that the process can handle a variety of renewable resources, different fossil fuels and unconventional materials. The operation parameters were optimized to maximize the hydrogen yield. Energy and exergy analyses showed high efficiencies and excellent capabilities for carbon capturing and thus proved the competitiveness of these systems compared to well-established technologies. However, despite the calculated high system



efficiencies in system simulations, the chemical looping hydrogen technology still requires prototype plants to be tested in large scale applications in long term operation, which would then demonstrate the advantages under real application scenarios. Furthermore, the oxygen carriers, which indicated satisfying performance in the lab environment, still need to demonstrate their applicability under authentic conditions.

## Acknowledgements

This work is funded by the IEA Research Cooperation on behalf of the Austrian Federal Ministry of Transport, Innovation and Technology.

## References

- 1 P. Nejat, F. Jomehzadeh, M. Mahdi and M. Gohari, *Renewable Sustainable Energy Rev.*, 2015, **43**, 843–862.
- 2 J. Cook, D. Nuccitelli, S. A. Green, M. Richardson, B. Winkler, R. Painting, R. Way, P. Jacobs and A. Skuce, *Environ. Res. Lett.*, 2013, **8**, 1–7.
- 3 R. K. Pachauri and L. A. Meyer, *Climate Change Synthesis Report*, IPCC, 2014, p. 151.
- 4 Adoption of the Paris Agreement Proposal by the President, <https://unfccc.int/resource/docs/2015/cop21/eng/l09r01.pdf>, accessed August 2016.
- 5 J. D. Holladay, J. Hu, D. L. King and Y. Wang, *Catal. Today*, 2009, **139**, 244–260.
- 6 S. K. Ngoh and D. Njomo, *Renewable Sustainable Energy Rev.*, 2012, **16**, 6782–6792.
- 7 J. Adanez, A. Abad, F. Garcia-Labiano, P. Gayan and L. F. De Diego, *Prog. Energy Combust. Sci.*, 2012, **38**, 215–282.
- 8 R. Chaubey, S. Sahu, O. O. James and S. Maity, *Renewable Sustainable Energy Rev.*, 2013, **23**, 443–462.
- 9 H. H. Gastineau and H. Lane, *GB pat.*, 190310356, 2014.
- 10 H. Lane, *US pat.*, 1028366, 1912.
- 11 H. Lane, *US pat.*, 1078686, 1913.
- 12 A. Messerschmitt, *US pat.*, 971206, 1910.
- 13 F. Bergius, *US pat.*, 1059818, 1913.
- 14 C. E. Parsons, *US pat.*, 1658939, 1926.
- 15 W. H. Marshall, *US pat.*, 2182747, 1939.
- 16 L. Fan, *Chemical Looping Systems for Fossil Energy Conversions*, John Wiley & Sons, Inc., Hoboken, New Jersey, 2010.
- 17 A. Thursfield, A. Murugan, R. Franca and I. S. Metcalfe, *Energy Environ. Sci.*, 2012, **5**, 7421–7459.
- 18 V. Hacker, *J. Power Sources*, 2003, **118**, 311–314.
- 19 S. D. Fraser, M. Monsberger and V. Hacker, *J. Power Sources*, 2006, **161**, 420–431.
- 20 D. D. Miller and R. Siriwardane, *Energy Fuels*, 2013, **27**, 4087–4096.
- 21 B. Dou, Y. Song, C. Wang, H. Chen and Y. Xu, *Renewable Sustainable Energy Rev.*, 2013, **30**, 950–960.
- 22 E. Lorente, J. A. Peña and J. Herguido, *Int. J. Hydrogen Energy*, 2008, **33**, 615–626.
- 23 F. Li, H. R. Kim, D. Sridhar, F. Wang, L. Zeng, J. Chen and L.-S. Fan, *Energy Fuels*, 2009, **23**, 4182–4189.
- 24 M. Ortiz, L. F. De Diego, A. Abad, F. García-Labiano, P. Gayan and J. Adanez, *Energy Fuels*, 2012, **26**, 791–800.
- 25 X.-L. Liu, X.-J. Yin and H. Zhang, *Energy Fuels*, 2014, **28**, 6066–6076.
- 26 R. D. Solunke, *Ind. Eng. Chem. Res.*, 2010, **49**, 11037–11044.
- 27 P. Gupta, L. G. Velazquez-vargas and L. Fan, *Energy Fuels*, 2007, **21**, 2900–2908.
- 28 M. Thaler, V. Hacker, M. Anilkumar, J. Albering, J. O. Besenhard, H. Schröttner and M. Schmied, *Int. J. Hydrogen Energy*, 2006, **31**, 2025–2031.
- 29 V. J. Aston, B. W. Evanko and A. W. Weimer, *Int. J. Hydrogen Energy*, 2013, **38**, 9085–9086.
- 30 W. C. Cho, D. Y. Lee, M. W. Seo, S. D. Kim, K. S. Kang, K. K. Bae, C. H. Kim, S. U. K. Jeong and C. S. Park, *Appl. Energy*, 2014, **113**, 1667–1674.
- 31 X. Zhu, Y. Wei, H. Wang and K. Li, *Int. J. Hydrogen Energy*, 2013, **38**, 4492–4501.
- 32 H. Fang, L. Haibin and Z. Zengli, *Int. J. Chem. Eng.*, 2009, 1–16.
- 33 N. L. Galinsky, Y. Huang and F. Li, *Sustainable Chem. Eng.*, 2013, **1**, 364–373.
- 34 W. Liu, J. S. Dennis and S. A. Scott, *Ind. Eng. Chem. Res.*, 2012, **51**, 16597–16609.
- 35 W. C. Cho, C. G. Kim, S. U. Jeong, C. S. Park, K. S. Kang, D. Y. Lee and S. D. Kim, *Ind. Eng. Chem. Res.*, 2015, **54**, 3091–3100.
- 36 W. C. Cho, M. W. Seo, S. D. Kim, K. S. Kang, K. K. Bae, H. Kim, S. U. Jeong and C. S. Park, *Int. J. Hydrogen Energy*, 2012, **37**, 16852–16863.
- 37 F. Kosaka, H. Hatano, Y. Oshima and J. Otomo, *Chem. Eng. Sci.*, 2015, **123**, 380–387.
- 38 Y. Zheng, X. Zhu, H. Wang, K. Li, Y. Wang and Y. Wei, *J. Rare Earths*, 2014, **32**, 842–848.
- 39 Y. Zheng, Y. Wei, K. Li, X. Zhu, H. Wang and Y. Wang, *Int. J. Hydrogen Energy*, 2014, **39**, 13361–13368.
- 40 X. Zhu, K. Li, Y. Wei, H. Wang and L. Sun, *Energy Fuels*, 2014, **28**, 754–760.
- 41 Z. Gu, K. Li, S. Qing, X. Zhu, Y. Wei, Y. Li and H. Wang, *RSC Adv.*, 2014, **4**, 47191–47199.
- 42 E. Romero, R. Soto, P. Durán, J. Herguido and J. A. Peña, *Int. J. Hydrogen Energy*, 2011, **37**, 6978–6984.
- 43 P. Datta, L. K. Rihko-Struckmann and K. Sundmacher, *Mater. Chem. Phys.*, 2011, **129**, 1089–1095.
- 44 P. Datta, L. K. Rihko-Struckmann and K. Sundmacher, *Fuel Process. Technol.*, 2014, **128**, 36–42.
- 45 H. Wang, J. Zhang, F. Wen and J. Bai, *RSC Adv.*, 2013, **3**, 10341–10348.
- 46 P. Datta, *Mater. Res. Bull.*, 2013, **48**, 4008–4015.
- 47 M. Ismail, W. Liu, M. T. Dunstan and S. A. Scott, *Int. J. Hydrogen Energy*, 2015, **41**, 4073–4084.
- 48 M. S. C. Chan, W. Liu, M. Ismail, Y. Yang, S. A. Scott and J. S. Dennis, *Chem. Eng. J.*, 2016, **296**, 406–411.
- 49 P. R. Kidambi, J. P. E. Cleeton, S. A. Scott, J. S. Dennis and C. D. Bohn, *Energy Fuels*, 2012, **26**, 603–617.
- 50 A. M. Kierzkowska, C. D. Bohn, S. A. Scott, J. P. Cleeton, J. S. Dennis and C. R. Müller, *Ind. Eng. Chem. Res.*, 2010, **49**, 5383–5391.



- 51 C. D. Bohn, J. P. Cleeton, C. R. Müller, S. A. Scott and J. S. Dennis, *Energy Fuels*, 2010, **24**, 4025–4033.
- 52 M. Ishida, K. Takeshita, K. Suzuki and T. Ohba, *Energy Fuels*, 2005, **19**, 2514–2518.
- 53 W. Liu, M. Ismail, M. T. Dunstan, W. Hu, Z. Zhang, P. S. Fennell, S. A. Scott and J. S. Dennis, *RSC Adv.*, 2015, **5**, 1759–1771.
- 54 M. Mufti, E. Jerndal, H. Leion, T. Mattisson and A. Lyngfelt, *Chem. Eng. Res. Des.*, 2010, **88**, 1505–1514.
- 55 F. Liu, Y. Zhang, L. Chen, D. Qian, J. K. Neathery, S. Kozo and K. Liu, *Energy Fuels*, 2013, **27**, 5987–5995.
- 56 H. Leion, A. Lyngfelt, M. Johansson, E. Jerndal and T. Mattisson, *Chem. Eng. Res. Des.*, 2008, **86**, 1017–1026.
- 57 Y. Ku, Y.-C. Liu, P.-C. Chiu, Y.-L. Kuo and Y.-H. Tseng, *Ceram. Int.*, 2014, **40**, 4599–4605.
- 58 R. Xiao, S. Zhang, S. Peng, D. Shen and K. Liu, *Int. J. Hydrogen Energy*, 2014, **39**, 19955–19969.
- 59 A. Tong, S. Bayham, M. V. Kathe, L. Zeng, S. Luo, L.-S. Fan and W. G. Lowrie, *Appl. Energy*, 2014, **113**, 1836–1845.
- 60 W. Liu, L. Shen, H. Gu and L. Wu, *Energy Fuels*, 2016, **30**, 1756–1763.
- 61 F.-X. Chiron and G. S. Patience, *Int. J. Hydrogen Energy*, 2012, **37**, 10526–10538.
- 62 T. D. L. R. Castillo, L. C. Palacios, S. A. D. L. Ríos, D. D. Vigil, J. S. Gutiérrez, A. L. Ortiz and V. Collins-Martínez, *J. New Mater. Electrochem. Syst.*, 2012, **12**, 55–61.
- 63 T. D. L. R. Castillo, J. S. Gutiérrez, A. L. Ortiz and V. Collins-Martínez, *Int. J. Hydrogen Energy*, 2013, **38**, 12519–12526.
- 64 H. Liang, *Int. J. Hydrogen Energy*, 2015, **40**, 13338–13343.
- 65 N. L. Galinsky, A. Shafiefarhood, Y. Chen, L. Neal and F. Li, *Appl. Catal., B*, 2015, **164**, 371–379.
- 66 K. Zhao, F. He, Z. Huang, A. Zheng, H. Li and Z. Zhao, *Int. J. Hydrogen Energy*, 2014, **39**, 3243–3252.
- 67 K. Zhao, F. He, Z. Huang, G. Wei, A. Zheng, H. Li and Z. Zhao, *Appl. Energy*, 2016, **168**, 193–203.
- 68 I. Popescu, Y. Wu, P. Granger and I. Marcu, *Appl. Catal., A*, 2014, **485**, 20–27.
- 69 C. Feng, S. Ruan, J. Li, B. Zou, J. Luo, W. Chen, W. Dong and F. Wu, *Sens. Actuators, B*, 2011, **155**, 232–238.
- 70 W. Haron, T. Thaweechai and W. Wattanathana, *Energy Procedia*, 2013, **34**, 791–800.
- 71 F. He and F. Li, *Energy Environ. Sci.*, 2015, **8**, 535–539.
- 72 T. A. S. Ferreira, J. C. Waerenborgh, M. H. R. M. Mendonça, M. R. Nunes and F. M. Costa, *Solid State Sci.*, 2003, **5**, 383–392.
- 73 D. S. Mathew and R. Juang, *Chem. Eng. J.*, 2007, **129**, 51–65.
- 74 E. Manova, B. Kunev, D. Paneva, I. Mitov and L. Petrov, *Chem. Mater.*, 2004, **16**, 5689–5696.
- 75 Y. Qu, H. Yang, N. Yang, Y. Fan, H. Zhu and G. Zou, *Mater. Lett.*, 2006, **60**, 3548–3552.
- 76 K. Svoboda, G. Slowinski, J. Rogut and D. Baxter, *Energy Convers. Manage.*, 2007, **48**, 3063–3073.
- 77 V. Crocellà, F. Cavani, G. Cerrato, S. Cocchi, M. Comito, G. Magnacca and C. Morterra, *J. Phys. Chem. C*, 2012, **116**, 14998–15009.
- 78 S. Cocchi, M. Mari, F. Cavani and J.-M. M. Millet, *Appl. Catal., B*, 2014, **152–153**, 250–261.
- 79 C. Trevisanut, M. Mari, J.-M. M. Millet and F. Cavani, *Int. J. Hydrogen Energy*, 2015, **40**, 5264–5271.
- 80 O. Vozniuk, S. Agnoli, L. Artiglia, A. Vassoi, N. Tanchoux, F. Di Renzo, G. Granozzi and F. Cavani, *Green Chem.*, 2016, **18**, 1038–1050.
- 81 Y. Kuo, W. Huang, W. Hsu, Y. Tseng and Y. Ku, *Aerosol Air Qual. Res.*, 2015, **15**, 2700–2708.
- 82 S. Bhavsar, M. Najera, R. Solunke and G. Vesper, *Catal. Today*, 2014, **228**, 96–105.
- 83 A. Edrisi, Z. Mansoori and B. Dabir, *Int. J. Hydrogen Energy*, 2014, **39**, 8271–8282.
- 84 M. Rydén and M. Arjmand, *Int. J. Hydrogen Energy*, 2012, **37**, 4843–4854.
- 85 D. Sanfilippo, *Catal. Today*, 2016, **272**, 58–68.
- 86 M. Sorgenfrei and G. Tsatsaronis, *Appl. Energy*, 2014, **113**, 1958–1964.
- 87 D. Zeng, R. Xiao, Z. Huang, J. Zeng and H. Zhang, *Int. J. Hydrogen Energy*, 2016, **41**, 6676–6684.
- 88 Z. Xue, S. Chen, D. Wang and W. Xiang, *Ind. Eng. Chem. Res.*, 2012, **51**, 4267–4278.
- 89 J. Herguido, J. Peña and E. Carazo, *Int. J. Hydrogen Energy*, 2014, **39**, 14050–14060.
- 90 S. Chen, Z. Xue, D. Wang and W. Xiang, *Int. J. Hydrogen Energy*, 2012, **37**, 8204–8216.
- 91 D. Sridhar, A. Tong, H. Kim, L. Zeng, F. Li, L.-S. Fan and W. G. Lowrie, *Energy Fuels*, 2012, **26**, 2292–2302.
- 92 A. Tong, D. Sridhar, Z. Sun, H. R. Kim, L. Zeng, F. Wang, D. Wang, M. V. Kathe, S. Luo, Y. Sun and L. S. Fan, *Fuel*, 2013, **103**, 495–505.
- 93 A. Tong, L. Zeng, M. V. Kathe, D. Sridhar, L.-S. Fan and W. G. Lowrie, *Energy Fuels*, 2013, **27**, 4119–4128.
- 94 M. V. Kathe, A. Empfield, J. Na, E. Blair and L. Fan, *Appl. Energy*, 2016, **165**, 183–201.
- 95 B. Moghtaderi, *Chem. Eng. Res. Des.*, 2012, **90**, 19–25.
- 96 B. Moghtaderi and E. Doroodchi, *Int. J. Hydrogen Energy*, 2012, **37**, 15164–15169.
- 97 D.-W. Zeng, R. Xiao, S. Zhang and H.-Y. Zhang, *Fuel Process. Technol.*, 2015, **139**, 1–7.
- 98 W. C. Cho, D. Y. Lee, M. W. Seo, S. D. Kim, K. Kang, K. K. Bae, H. Kim, S. Jeong and C. S. Park, *Appl. Energy*, 2014, **113**, 1667–1674.
- 99 Y. S. Hong, K. S. Kang, C. S. Park, D. Kim, J. W. Bae, J. W. Nam, Y. Lee and D. H. Lee, *Int. J. Hydrogen Energy*, 2013, **38**, 6052–6058.
- 100 D. Lee, M. W. Seo, T. D. B. Nguyen, W. C. Cho and S. D. Kim, *Int. J. Hydrogen Energy*, 2014, **39**, 14546–14556.
- 101 E. Hormilleja, P. Durán, J. Plou, J. Herguido and J. A. Peña, *Int. J. Hydrogen Energy*, 2014, **39**, 5267–5273.
- 102 R. Campo, P. Durán, J. Plou, J. Herguido and J. A. Peña, *J. Power Sources*, 2013, **242**, 520–526.
- 103 C. Trevisanut, F. Bosselet, F. Cavani and J. M. M. Millet, *Catal.: Sci. Technol.*, 2015, **3**, 1280–1289.
- 104 M. G. Rosmaninho, F. C. C. Moura, L. R. Souza, R. K. Nogueira, G. M. Gomes, J. S. Nascimento, M. C. Pereira, J. D. Fabris, J. D. Ardisson, M. S. Nazzarro, K. Sapag, M. H. Araújo and R. M. Lago, *Appl. Catal., B*, 2012, **115–116**, 45–52.



- 105 J. Plou, P. Durán, J. Herguido and J. A. Peña, *Fuel*, 2014, **118**, 100–106.
- 106 J. Plou, P. Durán, J. Herguido and J. A. Peña, *Fuel*, 2015, **140**, 470–476.
- 107 M. Herrero, J. Plou, P. Durán, J. Herguido and J. A. Peña, *Int. J. Hydrogen Energy*, 2015, **40**, 5244–5250.
- 108 G. Voitic, S. Nestl, K. Malli, J. Wagner, B. Bitschnau, F. Mautner and V. Hacker, *RSC Adv.*, 2016, **6**, 53533–53541.
- 109 S. Nestl, G. Voitic, M. Lammer, B. Marius, J. Wagner and V. Hacker, *J. Power Sources*, 2015, **280**, 57–65.
- 110 G. Voitic, S. Nestl, M. Lammer, J. Wagner and V. Hacker, *Appl. Energy*, 2015, **157**, 399–407.
- 111 Y. Tsai, L. Liu and D. Chen, *RSC Adv.*, 2016, **6**, 8930–8934.
- 112 K. Michiels, J. Spooren and V. Meynen, *Fuel*, 2015, **160**, 205–216.
- 113 A.-M. Cormos and C.-C. Cormos, *Int. J. Hydrogen Energy*, 2014, **39**, 2067–2077.
- 114 C. Cormos, A. Cormos and L. Petrescu, *Chemical Engineering Transactions*, 2014, **39**, 247–252.
- 115 C.-C. Cormos, *Int. J. Greenhouse Gas Control*, 2012, **37**, 13371–13386.
- 116 C.-C. Cormos, A.-M. Cormos and L. Petrescu, *Chem. Eng. Res. Des.*, 2014, **92**, 741–751.
- 117 S. Mukherjee, P. Kumar, A. Hosseini, A. Yang and P. Fennell, *Energy Fuels*, 2014, **28**, 1028–1040.
- 118 Q. Yang, Y. Qian, Y. Wang, H. Zhou and S. Yang, *Ind. Eng. Chem. Res.*, 2015, **54**, 6156–6164.
- 119 L. Zeng, F. He, F. Li, L.-S. Fan and W. G. Lowrie, *Energy Fuels*, 2012, **26**, 3680–3690.
- 120 S. Chen, Z. Xue, D. Wang and W. Xiang, *J. Power Sources*, 2012, **215**, 89–98.
- 121 H. Ozcan and I. Dincer, *Energy Convers. Manage.*, 2014, **85**, 477–487.
- 122 K.-S. Kang, C.-H. Kim, K.-K. Bae, W.-C. Cho, S.-U. Jeong, S.-H. Kim and C.-S. Park, *Int. J. Hydrogen Energy*, 2012, **37**, 3251–3260.
- 123 M. N. Khan and T. Shamim, *Appl. Energy*, 2016, **162**, 1186–1194.
- 124 X. Zhang, S. Li, H. Hong and H. Jin, *Energy Convers. Manage.*, 2014, **85**, 701–708.
- 125 X. Zhang and H. Jin, *Appl. Energy*, 2013, **112**, 800–805.
- 126 X. Zhang, S. Li and H. Jin, *Energies*, 2014, **7**, 7166–7177.
- 127 C.-C. Cormos, *Fuel Process. Technol.*, 2015, **137**, 16–23.
- 128 L. Yan, G. Yue and B. He, *Bioresour. Technol.*, 2016, **205**, 133–141.
- 129 L. Yan, B. He, X. Pei, C. Wang, Z. Duan, J. Song and X. Li, *Int. J. Hydrogen Energy*, 2014, **39**, 17540–17553.
- 130 C.-C. Cormos, *Int. J. Hydrogen Energy*, 2014, **39**, 5597–5606.
- 131 S. G. Gopaul, A. Dutta and R. Clemmer, *Int. J. Hydrogen Energy*, 2014, **39**, 5804–5817.
- 132 A. Sanz, D. Nieva and J. Dufour, *Int. J. Hydrogen Energy*, 2015, **40**, 5074–5080.
- 133 A. Edrisi, Z. Mansoori, B. Dabir and A. Shahnazari, *Int. J. Hydrogen Energy*, 2014, **39**, 10380–10391.
- 134 J. Zhu, W. Wang, X. N. Hua, F. Wang, Z. Xia and Z. Deng, *Int. J. Hydrogen Energy*, 2015, **40**, 12097–12107.
- 135 X. Hua, W. Wang and Y. Hu, *J. Mater. Cycles Waste Manage.*, 2014, **16**, 583–590.
- 136 X. Hua, W. Wang and F. Wang, *Front. Environ. Sci. Eng.*, 2015, **9**, 1130–1138.
- 137 C. Hertel, P. Heidebrecht and K. Sundmacher, *Int. J. Hydrogen Energy*, 2012, **37**, 2195–2203.
- 138 K.-S. Kang, C.-H. Kim, K.-K. Bae, W.-C. Cho, S.-U. Jeong, Y.-J. Lee and C.-S. Park, *Chem. Eng. Res. Des.*, 2014, **92**, 2584–2597.
- 139 S. Nasr and K. P. Plucknett, *Energy Fuels*, 2014, **28**, 1387–1395.
- 140 E. R. Monazam, R. W. Breault and R. Siriwardane, *Energy Fuels*, 2014, **28**, 5406–5414.
- 141 E. R. Monazam, R. W. Breault, R. Siriwardane, G. Richards and S. Carpenter, *Chem. Eng. J.*, 2013, **232**, 478–487.
- 142 Y. Zhang, E. Doroodchi and B. Moghtaderi, *Energy Fuels*, 2015, **29**, 337–345.
- 143 Y. Zhang, E. Doroodchi and B. Moghtaderi, *Appl. Energy*, 2014, **113**, 1916–1923.
- 144 E. Ksepko, M. Sciazko and P. Babinski, *Appl. Energy*, 2013, **115**, 374–383.
- 145 M. H. Jeong, D. H. Lee and J. W. Bae, *Int. J. Hydrogen Energy*, 2015, **40**, 2613–2620.
- 146 W. Liu, J. Y. Lim, M. A. Saucedo, A. N. Hayhurst, S. A. Scott and J. S. Dennis, *Chem. Eng. Sci.*, 2014, **120**, 149–166.
- 147 J. Plou, P. Duran, J. Herguido and J. A. Peña, *Int. J. Hydrogen Energy*, 2012, **37**, 6995–7004.
- 148 R. C. Stehle, M. M. Bobek and D. W. Hahn, *Int. J. Hydrogen Energy*, 2015, **40**, 1675–1689.
- 149 M. M. Bobek, R. C. Stehle and D. W. Hahn, *Materials*, 2012, **5**, 2003–2014.
- 150 A. Singh, F. Al-Raqom, J. Klausner and J. Petrasch, *Int. J. Hydrogen Energy*, 2012, **37**, 7442–7450.

