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Magnetic properties and giant reversible magnetocaloric effect in GdCoC_2

Lingjian Meng,^{ab} Chi Xu,^c Ye Yuan,^c Yang Qi,^b Shengqiang Zhou^c and Lingwei Li^{*ab}

The crystal structure, magnetic properties and magnetocaloric effect (MCE) of GdCoC_2 have been studied. The compound crystallizes in an orthorhombic CeNiC_2 -type structure which belongs to the $Amm2$ space group. A giant reversible MCE is observed in GdCoC_2 accompanied by a second-order paramagnetic to ferromagnetic (PM–FM) phase transition around the Curie temperature ~ 15 K. For the magnetic field change of 0–5 T, the maximum values of the magnetic entropy change ($-\Delta S_M^{\text{max}}$), relative cooling power (RCP), and refrigerant capacity (RC) are $28.4 \text{ J kg}^{-1} \text{ K}^{-1}$, 566 J kg^{-1} and 369 J kg^{-1} , respectively. The present results indicate that GdCoC_2 is a promising candidate for low temperature magnetic refrigeration.

1. Introduction

Materials with large/giant magnetocaloric effect (MCE) have gained more and more interest due to their applications for magnetic refrigeration. Compared with traditional gas refrigeration, magnetic refrigeration has significant advantages in conversion efficiency, low noise and environmental protection.^{1–5} The MCE manifests as ΔS_M (isothermal magnetic entropy change) and ΔT_{ad} (adiabatic temperature change). Finding special materials with a large MCE value is considered to be the most important work of magnetic refrigeration, since the MCE is an inherent characteristic of magnetic materials. In recent years, rare-earth based compounds have been widely investigated with respect to their MCE properties.^{6–19} And some of them exhibit large/giant MCE at 10–25 K, which is around the boiling point of hydrogen.^{6–15} Considering the problems in application and storage of liquid hydrogen at room temperature, these materials are eligible to be applied for magnetic refrigeration for hydrogen liquefaction.

In the present study, we have investigated the MCE in GdCoC_2 which belongs to an existing ternary system RETC_2 (RE stands for heavy rare earth element and T stands for transition metal Co and Ni). Schafer *et al.* have reported that all the RECoC_2 compounds order ferromagnetically, whereas, compounds of RENiC_2 are antiferromagnetic.^{20–26} Up to the present, only the MCE in TbCoC_2 belonging to this series compounds has been reported and the maximum magnetic entropy change ($-\Delta S_M^{\text{max}}$) is $15.3 \text{ J kg}^{-1} \text{ K}^{-1}$ for the magnetic field change of 0–5 T.¹⁵ For the presently studied GdCoC_2 , the

values of $-\Delta S_M^{\text{max}}$ are $16.0 \text{ J kg}^{-1} \text{ K}^{-1}$, $28.4 \text{ J kg}^{-1} \text{ K}^{-1}$, and $32.9 \text{ J kg}^{-1} \text{ K}^{-1}$ for magnetic field changes of 0–2 T, 0–5 T and 0–7 T, respectively.

2. Experimental details

The polycrystalline GdCoC_2 sample was synthesized by the method of arc-melting. First, stoichiometric amounts of high-purity components were weight and 3% extra carbon was added to compensate the loss during arc-melting. The sample was turned over and melted for four times under an argon atmosphere to ensure good homogeneity. Then the sample was annealed at 900°C for six days in evacuated quartz tubes and rapidly quenched in ice water. The sample was characterized by X-ray powder diffraction (XRD) measurement by Rigaku D/MAX 2550 using $\text{Cu K}\alpha$ radiation. The magnetization measurements were done with a commercial superconducting quantum interference device (SQUID) based vibrating sample magnetometer (VSM) (Quantum Design Company, USA).

3. Results and discussion

The XRD pattern for GdCoC_2 together with the Rietveld refinement profiles which were analyzed by MAUD software are shown in Fig. 1. The inset of Fig. 1 shows the crystal structure diagram obtained by using the CRYSTALMAKER software package. The factors of Rietveld refinement which were calculated by MAUD software are $R_{\text{wp}} (\%) = 8.99$, $R_B (\%) = 6.69$, $R_{\text{exp}} (\%) = 12.0$. The results reveal that the sample is crystallized in single phase with orthorhombic CeNiC_2 -type structure ($Amm2$ space group) and no impurity phases can be detected. According to the Rietveld refinement profiles, the refined lattice parameters a , b , and c are calculated by Bragg equation and found to be 3.621, 4.506, and 6.062 \AA , respectively, which are close to the standard values.

^aKey Laboratory of Electromagnetic Processing of Materials (Ministry of Education), Northeastern University, Shenyang 110819, China. E-mail: lingwei@epm.neu.edu.cn

^bInstitute of Materials Physics and Chemistry, School of Materials Science and Engineering, Northeastern University, Shenyang 110819, China

^cHelmholtz-Zentrum Dresden-Rossendorf, Institute of Ion Beam Physics and Materials Research, Bautzner Landstrasse 400, 01328 Dresden, Germany

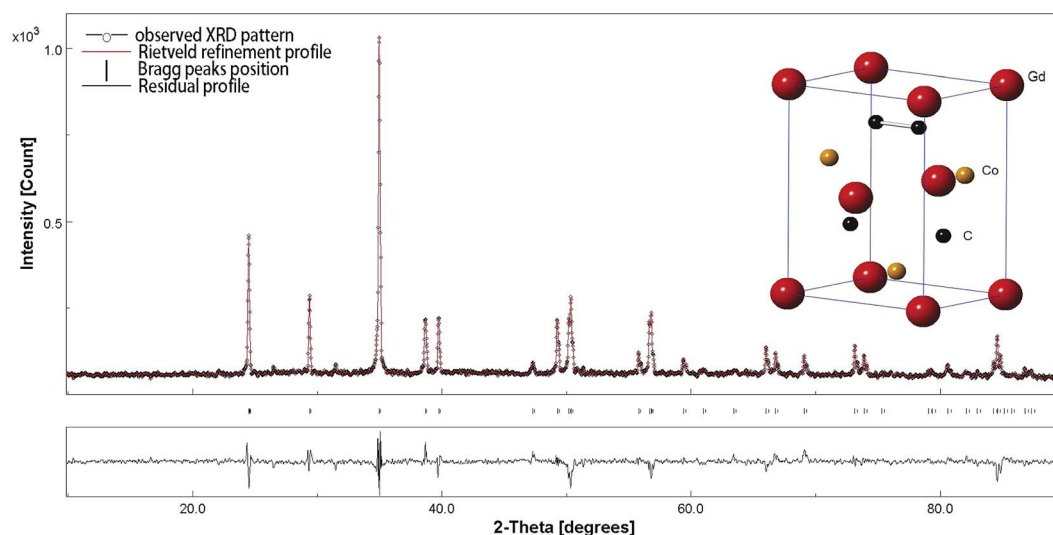


Fig. 1 XRD pattern for GdCoC₂ together with the Rietveld refinement profiles, Bragg peak positions, Rietveld profiles. The inset shows the crystal structure diagram.

The zero field cooling (ZFC) and field cooling (FC) magnetization (M) at magnetic field (H) of 0.2 T is shown in the inset of Fig. 2. A typical paramagnetic to ferromagnetic (PM-FM) phase transition can be observed at the Curie temperature (T_C) \sim 15 K (defined as the minimum value of dM/dT curve). No obvious difference can be observed between ZFC and FC curves, which is a typical behavior for the second order phase transition materials. The temperature dependence of magnetization (M) and the inverse susceptibility ($1/\chi$) under magnetic field (H) of 1 T are shown in Fig. 2. At high temperature, the $1/\chi$ - T curve shows Curie-Weiss behavior. The evaluated value of the effective magnetic moment ($8.0 \mu_B/\text{f.u.}$) for GdCoC₂ is close to that of the free ion value of Gd³⁺ ($7.94 \mu_B$). A set of magnetization isotherm (M - H) was measured at temperatures from 3 K to 45 K under magnetic field up to 7 T, and several M - H curves with increasing field are shown in Fig. 3. Among them, the hysteresis cycles at 3 K, 13 K, 15 K, 20 K, and 30 K

are shown in the figure as well. No obvious hysteresis is observed and the magnetization rapidly changes with varying temperature around T_C , i.e. a large reversible MCE may appear around 15 K. The inset of Fig. 3 shows the M^2 versus H/M curves (also named as Arrot plot) from 3 K to 45 K. Based on the Banerjee criterion,²⁷ the present GdCoC₂ undergoes a second-order magnetic phase transition since neither negative slope nor inflection can be observed in the whole temperature range.

The isothermal magnetic-entropy change ΔS_M is obtained from magnetization isotherms by integrating the Maxwell relation

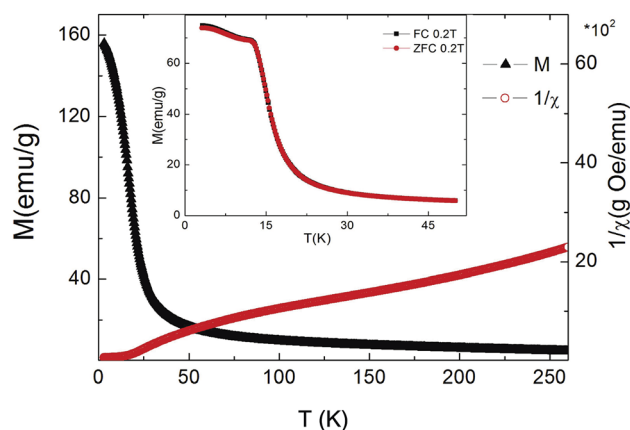


Fig. 2 Temperature dependence of magnetization (left scale) and the inverse susceptibility (right scale) for GdCoC₂. The inset shows the temperature dependence of zero field cooling (ZFC) and field cooling (FC) magnetization (M) under magnetic fields (H) = 0.2 T.

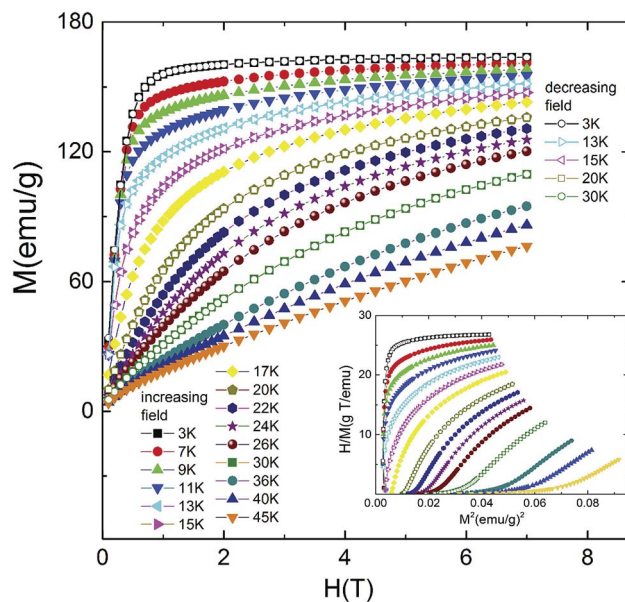


Fig. 3 Magnetic field (H) dependence of the magnetization (M) for GdCoC₂ at different temperatures. The inset shows the Arrot plots curves (M^2 versus H/M).



$$\Delta S_M(T, \Delta H) = \int_0^{H_{\max}} \left(\frac{\partial M}{\partial T} \right)_H dH \approx \frac{1}{\Delta T} \int_0^{H_{\max}} [M(T_{i+1}, H) - M(T_i, H)] dH,$$

The temperature-dependent $-\Delta S_M$ under magnetic field changes up to 0–7 T is shown in Fig. 4. $-\Delta S_M$ increases with the increasing value of ΔH . The maxima of $-\Delta S_M$ appears around 17 K, which is close to the paramagnetic to ferromagnetic (PM–FM) phase transition. The values of $-\Delta S_M^{\max}$ are equal to 9.3 J kg^{−1} K^{−1}, 20.9 J kg^{−1} K^{−1}, 28.4 J kg^{−1} K^{−1}, 32.9 J kg^{−1} K^{−1} under the field changes of 0–1 T, 0–3 T, 0–5 T and 0–7 T, respectively. These values are around two times of those for TbCoC₂ and larger than those of most of recently reported giant MCE materials in the same temperature range, indicating that the presently studied GdCoC₂ compound belongs to a class of giant MCE materials. The origin of giant MCE in GdCoC₂ is probably related to the large saturation moment, especially under low magnetic field as well as its field and temperature sensitive magnetic phase transition. Additionally, the large value of $-\Delta S_M^{\max}$ of 16.0 J kg^{−1} K^{−1} is reached under the field changes of 0–2 T with a quite wide temperature range for present GdCoC₂, which is quite beneficial to application.

Franco *et al.* revealed a universal behavior of the field dependence of ΔS_M for the materials with a second-order transition, *i.e.* in ref. 28, the ΔS_M versus T curves under different magnetic fields can collapse into a universal curve; therefore the influence of different magnetic fields can be ignored. All the curves are normalized to their respective maximum value as $\Delta S_M/\Delta S_M^{\max}$ and the axis of temperature is rescaled to θ below and above T_C ,

$$\theta = \begin{cases} -(T - T_C)/(T_{r1} - T_C); & T \leq T_C \\ (T - T_C)/(T_{r2} - T_C); & T > T_C \end{cases},$$

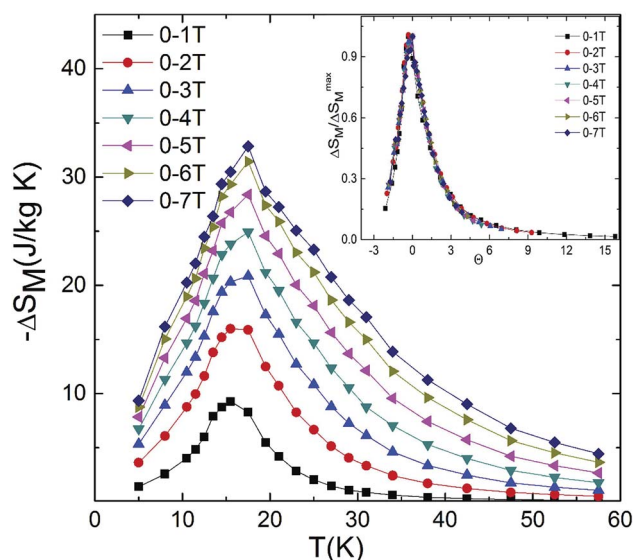


Fig. 4 Temperature-dependent magnetic entropy change $-\Delta S_M$ for GdCoC₂ in various magnetic fields changes. The inset shows the universal curves.

Table 1 The transition temperature (T_M), $-\Delta S_M^{\max}$ and relative cooling power (RCP) under 2 T and 5 T for GdCoC₂ and other materials with large

	T_M (K)	$-\Delta S_M^{\max}$ (J kg ^{−1} K ^{−1})		RCP (J kg ^{−1})		Reference
		2 T	5 T	2 T	5 T	
GdCoC ₂	15	16.0	28.4	216	566	Present
Eu ₈ Ga ₁₆ Ge ₃₀	11	4.5	11.5	80	400	7
GdNiBC	15	9.3	19.8	188	474	8
HoAgAl	18	3.8	10.3	99	344	10
ErFeSi	22	14.2	23.1	~171	~486	29
GdCo ₂ B ₂	25	9.3	17.1	167	462	14
TbCoC ₂	28	7	15.3	~115	~375	15

T_{r1} and T_{r2} are the temperatures of the two reference points of each curve that corresponds to 0.6 $-\Delta S_M^{\max}$. The curves of normalized entropy change $\Delta S_M/\Delta S_M^{\max}$ versus the rescaled temperature θ under different magnetic fields are shown in the inset of Fig. 4. All the curves under different magnetic fields collapse onto a single line for GdCoC₂, which further confirms that the present GdCoC₂ undergoes a second order magnetic phase transition.

The relative cooling power (RCP) and refrigerant capacity (RC) are important factors to evaluate refrigeration materials. The values of RCP can be calculated as

$$\text{RCP} = -\Delta S_M^{\max} \times \delta T_{\text{FWHM}}$$

δT_{FWHM} is the full width at half maximum and $-\Delta S_M^{\max}$ is the maximum of magnetic entropy change, respectively.⁴ The values of RCP for GdCoC₂ are 216, 566, 769 J kg^{−1} for $\Delta H = 0-2$, $0-5$ and $0-7$ T, respectively. The values of RC can be calculated as

$$\text{RC} = \int_{T_1}^{T_2} |\Delta S_M| dT,$$

T_1 and T_2 are the temperatures of half-maximum value of $-\Delta S_M$ peak.⁴ The value of RC for GdCoC₂ are equal to 160, 369, 514 J kg^{−1} for $\Delta H = 0-2$, $0-5$ and $0-7$ T, respectively. For a comparison, the transition temperature (T_M), together with the values of $-\Delta S_M^{\max}$, RCP and RC with the field change from 0 to 2 T and 5 T for GdCoC₂ and some recently reported large/giant MCE materials with T_M around 15 K are shown in Table 1. These values for GdCoC₂ are obviously larger than most of those reported materials in the similar working temperature range.

4. Conclusions

In summary, a single phase GdCoC₂ compound is synthesized, and its crystal structure, magnetic properties and magnetocaloric properties have been investigated. The compound undergoes a second-order paramagnetic to ferromagnetic (PM–FM) transition at the Curie temperature $T_C \sim 15$ K. Accompanied with the transition, giant reversible MCE is observed. The values of $-\Delta S_M^{\max}$ for GdCoC₂ reach 16.0, 28.4, and 32.9 J kg^{−1} K^{−1} for the field changes of 0–2 T, 0–5 T and 0–7 T, respectively. The corresponding values of RCP (RC) are 216 (160), 566 (369),



and 769 (514) J kg⁻¹. For the reason that the compound has large MCE and small hysteresis in a wide temperature range, it could be considered as potential refrigerant material for low temperature magnetic refrigeration.

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References

- 1 E. Bruck, *J. Phys. D: Appl. Phys.*, 2005, **38**, R381.
- 2 K. A. Gschneidner, V. K. Pecharsky and A. O. Tsokol, *Rep. Prog. Phys.*, 2005, **68**, 1479.
- 3 B. G. Shen, J. R. Sun, F. X. Hu, H. W. Zhang and Z. H. Cheng, *Adv. Mater.*, 2009, **21**, 4545.
- 4 L. W. Li, *Chin. Phys. B*, 2016, **25**, 037502.
- 5 A. Smith, C. R. H. Bahl, R. Bjørk, K. Engelbrecht, K. K. Nielsen and N. Pryds, *Adv. Energy Mater.*, 2012, **2**, 1288.
- 6 A. I. Vinokur and D. C. Fredrickson, *Inorg. Chem.*, 2016, **55**, 6148.
- 7 A. Chaturvedi, S. Stefanoski, M. H. Phan, G. S. Nolas and H. Srikanth, *Appl. Phys. Lett.*, 2011, **99**, 162513.
- 8 L. W. Li, M. Kadonaga, D. X. Huo, Z. H. Qian, T. Namiki and K. Nishimura, *Appl. Phys. Lett.*, 2012, **101**, 122401.
- 9 Y. Zhang, Y. Yang, X. Xu, L. Hou, Z. Ren, X. Li and G. Wilde, *J. Phys. D: Appl. Phys.*, 2016, **49**, 145002.
- 10 Y. Zhang, B. Yang and G. Wilde, *J. Alloys Compd.*, 2015, **619**, 12.
- 11 K. P. Shinde, S. H. Jang, J. W. Kim, D. S. Kim, M. Ranot and K. C. Chung, *Dalton Trans.*, 2015, 20386.
- 12 T. Paramanik, T. Samanta, R. Ranganathan and I. Das, *RSC Adv.*, 2015, **5**, 47860.
- 13 B. Sattibabu, A. K. Bhatnagar, K. Vinod, S. Rayaprol, A. Mani, V. Siruguri and D. Das, *RSC Adv.*, 2016, **6**, 48636.
- 14 L. W. Li, K. Nishimura and H. Yamane, *Appl. Phys. Lett.*, 2009, **94**, 102509.
- 15 B. Li, W. J. Hu, X. G. Liu and F. Yang, *Appl. Phys. Lett.*, 2008, **92**, 242508.
- 16 S. Yin, V. Sharma, A. McDannald, F. A. Reboredo and M. Jain, *RSC Adv.*, 2016, **6**, 9475.
- 17 Y. Zhang, L. Hou, Z. Ren, X. Li and G. Wilde, *J. Alloys Compd.*, 2016, **565**, 635.
- 18 B. Uthaman, K. S. Anand, R. K. Rajan, H. H. Kyaw, S. Thomas, S. Al-Harhi, K. G. Suresh and M. R. Varma, *RSC Adv.*, 2015, **5**, 86144.
- 19 Y. K. Zhang, X. Xu, Y. Yang, L. Hou, Z. Ren, X. Li and G. Wilde, *J. Alloys Compd.*, 2016, **667**, 130.
- 20 H. Amanai, H. Onodera, M. Ohashi, S. Matsuo, H. Yamauchi, Y. Yamaguchi and N. Sato, *J. Magn. Magn. Mater.*, 1995, **148**, 413.
- 21 H. Onodera, N. Uchida, M. Ohashi, H. Yamauchi, Y. Yamaguchi and N. Sato, *J. Magn. Magn. Mater.*, 1994, **137**, 35.
- 22 P. A. Kotsanidis, J. K. Yakinthos and W. Schafer, *J. Alloys Compd.*, 1996, **242**, 90.
- 23 W. Schafer, G. Will, P. A. Kotsanidis and J. K. Yakinthos, *J. Magn. Magn. Mater.*, 1990, **88**, 13.
- 24 W. Schafer, W. Kockelmann, G. Will, J. K. Yakinthos and P. A. Kotsanidis, *J. Alloys Compd.*, 1997, **250**, 565.
- 25 M. Reehuis, J. Rodriguez-Carvajal, M. E. Danebrock and W. Jeitschko, *J. Magn. Magn. Mater.*, 1995, **151**, 273.
- 26 W. Schafer, W. Kockelman, G. Will, P. A. Kotsanidis, J. K. Yakinthos and J. Linhart, *J. Magn. Magn. Mater.*, 1994, **132**, 243.
- 27 B. K. Banerjee, *Phys. Lett.*, 1964, **12**, 16.
- 28 V. Franco, A. Conde, V. Provenzano and R. D. Shull, *J. Magn. Magn. Mater.*, 2010, **322**, 218.
- 29 H. Zhang, B. G. Shen, Z. Y. Xu and J. Shen, *Appl. Phys. Lett.*, 2013, **102**, 092401.

