**Electrochemical and electromechanical properties of superior-performance hybrid polymer actuators exhibiting synergistic effects due to manganese oxide and multi-walled carbon nanotubes on various ionic liquids†**

Naohiro Terasawa* and Kinji Asaka

The electrochemical and electromechanical properties of poly(vinylidene fluoride-co-hexafluoropropylene) actuators were investigated using a non-activated multi-walled carbon nanotube (MWCNT)–ionic liquid (IL) gel electrode containing manganese oxide (MnO₂) on various ILs. The MnO₂/MWCNT/IL actuators surpassed the strain and maximum generated stress performance observed for the only-MWCNT and only-SWCNT actuators on all ILs. These actuators are different from only-MWCNT and only-SWCNT actuators, which act solely as electrostatic double-layer capacitor (EDLC) units. The frequency dependence of the displacement responses of the MnO₂-xH₂O/MWCNT/IL polymer actuators was successfully simulated using a double-layer charging kinetic model. Two parameters for the simulation were determined: the strain at the low frequency limit and the time constant. Simulations of the electromechanical response of the MnO₂-xH₂O/MWCNT/IL actuators predicted strains at low frequencies as well as at the associated time constant, confirming that the model is applicable not only to EDLC-based actuator systems but also to the fabricated EDLC/FC system. These results suggest that flexible, robust films, fabricated from metal oxides and MWCNTs that function synergistically, are potential candidates for actuator materials that can be applied in wearable and energy conversion devices.

**Introduction**

Recently, there has been significant interest in soft materials that can directly transform electrical energy into mechanical work, since such materials have many potential applications in tactile and optical displays, prosthetics, medical devices, robotics, and microelectromechanical systems. An especially promising application is the use of low-voltage electroactive polymer actuators with fast response times as artificial muscle-like actuators in various practical biomedical devices. Previously, our group reported a dry actuator based on a gelatinous, room-temperature ionic liquid (IL) containing single-walled carbon nanotubes (SWCNTs), known as a ‘bucky-gel’. The bimorph configuration of this device consisted of a polymer-supported IL electrolyte layer held between two polymer-supported ‘bucky-gel’ electrode layers. Because of this design, the device is suitable for rapid operations and has a significant usable lifespan when operated in air at low applied voltages. ILs have a number of properties that make them quite applicable in quick-response actuators as well as in highly electrochemically stable devices, including low volatility, high ionic conductivity, and wide potential windows. However, our research has found that the specific IL, nanocarbon, and polymer used in such devices can affect the resulting electromechanical and electrochemical characteristics of the actuator.

Electrochemical capacitors (ECs) can be placed in two broad categories. The first category covers faradaic capacitors (FC) employing electrochemically active compounds such as metal oxides as electrodes. They are capable of directly storing charge throughout the charging and discharging processes. The second category includes electrostatic double-layer capacitors (EDLC) employing electrochemically inactive substances such as carbon particles as electrode materials. No electrochemical reaction occurs on the electrode surface during the charging and discharging of these devices, and therefore, only physical charge accumulates at the electrode/electrolyte interface. Regardless of the EC category, the device should have a significant surface area along with a suitable pore-size distribution, and should employ high-conductivity electrode materials to obtain the highest possible capacitance values. A third possible EC is the hybrid capacitor, in which both the EDLC and FC mechanisms occur simultaneously, with one being dominant.
From the above discussion, it is evident that FCs, also known as pseudo capacitors, differ from EDLCs. The application of a potential to an FC generates rapid, reversible faradaic (or redox) reactions on the electrode surface. Conductive polymers and various metal oxides, including RuO$_2$, MnO$_2$, and Co$_3$O$_4$, are known to allow these redox reactions. During this process, charge moves across the double layer in a process that mimics battery charging and discharging, such that a faradaic current passes through the capacitor cell.

To the best of our knowledge, no non-activated multi-walled carbon nanotube (MWCNT) polymer actuators showing improved performance over SWCNT polymer actuators have been developed. SWCNTs are specially prepared compounds that are very expensive. MWCNTs, on the other hand, are quite inexpensive and are commonly used in battery electrodes. Carbon nanotubes (CNTs) have been recognized as potential inexpensive and are commonly used in battery electrodes. That are very expensive. MWCNTs, on the other hand, are quite inexpensive and are commonly used in battery electrodes. Carbon nanotubes (CNTs) have been recognized as potential expensive and are commonly used in battery electrodes. Carbon nanotubes (CNTs) have been recognized as potential inexpensive and are commonly used in battery electrodes.

In the present work, we fabricated a hybrid (electrostatic double-layer/faradaic capacitor) polymer actuator that exploits the synergistic effects resulting from the combination of MWCNTs and MnO$_2$. The electrochemical and electromechanical characteristics of this device were subsequently compared with those of polymer actuators fabricated using only SWCNTs or MWCNTs with various ILs (Fig. 1). The frequency dependence of the displacement responses of the MnO$_2$/MWCNT polymer actuators was simulated using a double-layer charging kinetic model with various ILs.

**Experimental**

**Materials**

MWCNTs (NC3150, Nanocyl Inc.) and SWCNTs (high-purity HiPco™, Unidym Inc.) were used as received. The MWCNTs had an average diameter of 9.5 nm and average length of ~1 μm, whereas the SWCNTs had an average diameter of ~0.8–1.2 nm and average length of ~0.1–1 μm. MnO$_2$·xH$_2$O was synthesized according to the literature. The ILs, based on 1-ethyl-3-methylimidazolium (EMI) cations and tetrafluoroborate, trifluoroacetate, triflate, bis[(fluorosulfonyl)imide, or bis[(trifluoromethanesulfonyl)imide anions (EMI[BF$_4$] (Fluka), EMI [CF$_3$COO] (Fluka), EMI[CF$_3$SO$_3$] (IoLiTec Ionic Liquids Technologies), EMI[FSI] (Dai-ichi Kogyo Seiyaku), and EMI[CF$_3$SO$_3$] (Fluka), respectively) were also used as received; their chemical structures are shown in Fig. 1. The IL 1-ethyl-3-methylimidazolium trifluoromethylltrifluoroborate (EMI[CF$_3$BF$_3$], Fig. 1) was synthesized according to the literature method. Other reagents, poly(vinylidene fluoride-co-hexafluoropropylene) (PVdF(HFP); Kynar Flex 2801, Arkema Chemicals Inc.), methyl pentanone (MP; Aldrich), propylene carbonate (PC; Aldrich), and dimethylacetamide (DMAc; Kishida Chemical Co. Ltd.) were used as received.

**Preparation of the actuator films**

The configuration of the bucky-gel actuator is illustrated in Fig. 1. The polymer-supported bucky-gel electrode layer was characteristic in the past, we considered that the frequency dependence of the actuator strain could be qualitatively modelled using an electrochemical equivalent circuit model based on the lumped resistance and capacitive of the electrode layer and the lumped resistance of the electrolyte layer.

**Preparation of the actuator films**

The configuration of the bucky-gel actuator is illustrated in Fig. 1. The polymer-supported bucky-gel electrode layer was characteristic.
typically composed of 16.7 wt% MWCNTs, 16.7 wt% MnO$_2$·xH$_2$O, 40 wt% IL, and 26.6 wt% PVdF(HFP), and was prepared as follows. The MWCNTs (50 mg), MnO$_2$·xH$_2$O (50 mg), IL (120 mg), and PVdF(HFP) (80 mg) were dispersed in DMAC (9 mL) using ultrasonication for more than 5 h. A gelatinous mixture was then obtained. The electrode layer was fabricated by casting 1.6 mL of the electrode mixture in a Te mould (2.5 × 2.5 cm$^2$) and evaporating the solvent. The solvent was then completely removed in vacuo at 80 °C. The thickness of the obtained electrode film was 70–80 μm. Each gel electrolyte layer was then fabricated by combining an IL and PVdF(HFP) (0.5 mmol/100 mg) with a mixture of MP (1 mL) and PC (250 mg), and casting a portion of the solution (0.3 mL) in a Teflon mould (2.5 × 2.5 cm$^2$), followed by solvent evaporation, and complete removal of the solvent in vacuo at 80 °C. The thickness of the obtained gel electrolyte film was 20–30 μm. The actuator film was fabricated by hot-pressing the electrode and electrolyte layers with the same IL. The typical thickness of the actuator film was 150–175 μm, which is smaller than the sum of the two electrode and one electrolyte layer thicknesses, because the thickness of each layer was decreased by hot-pressing.

**Displacement measurements**

The actuator experiments were conducted by applying a triangular voltage to a 10 × 1 mm$^2$ actuator strip clipped by two gold disk electrodes. The displacement, at a point 5 mm away (free length) from the fixed point, was continuously monitored from one side of the actuator strip with a laser displacement meter (Keyence, LC2100/2220). A potentiogalvanostat (Hokuto Denko, HA-151) with a waveform generator (Yokogawa Electric, FC 200) were used to activate the bucky-gel actuator. The electrical parameters were simultaneously measured. The measured displacement $\delta$ was transformed into the strain difference between the two bucky-gel electrode layers ($\varepsilon$) by using the following equation, on the assumption that the cross sections are plane planar at any position along the actuator, i.e. there is no distortion of the cross section:

$$\varepsilon = 2d\delta/(L^2 + \delta^2),$$

(1)

where $L$ is the free length and $d$ is the thickness of the actuator strip.

**Characterization of the electrode and electrolyte**

Fourier-transform infrared (FT-IR) spectra were collected using the attenuated total reflection (ATR) technique with a Cary 640 FT-IR spectrometer (Agilent Technologies). X-Ray diffraction (XRD) analysis was carried out to elucidate the crystal structure of the electrodes using a Rigaku X-ray diffractometer with Cu Kα radiation. X-Ray photoelectron spectroscopy (XPS) analysis was carried out to evaluate the chemical state of the electrodes using a PHI 5000 VersaProbe XPS instrument equipped with an ULVAC-PHI source using monochromated Al Kα radiation. The morphologies of the electrode films were observed by scanning electron microscopy (SEM) with a JEOL JSM-6510 instrument. Young’s moduli for the electrodes were estimated from the stress–strain curves, which were measured using a Seiko TMA/SS 6000 thermal stress–strain instrument.

The specific capacitance of each polymer-supported bucky-gel electrode ($\phi$ 7 mm) was estimated by cyclic voltammetry (CV), which was measured using a two-electrode configuration with a potentiostat (Hokuto Denko, HSV-100). The electrical conductivities of the electrodes were evaluated using the four-probe DC current method, where a linear sweep wave of current was applied from the outer probe electrodes, and the voltage was measured by the inner probe electrodes. Current–voltage curves were obtained using a potentiogalvanostat (Hokuto Denko, HA-151) with a waveform generator (Yokogawa Electric, FC 200). The conductivity of the gel electrolyte layer was assessed by impedance measurement, which was performed using a Solartron 1250 Impedance Analyser.

**Results and discussion**

**Chemical structure, morphology, and mechanical properties of the electrodes**

The FT-IR spectrum of the MnO$_2$/MWCNT/EMI[B$_4$F$_4$]/PVdF(HFP) (MnO$_2$ : MWCNT = 1 : 1) electrode is shown in Fig. 2(a). The spectrum exhibits peaks characteristic of PVdF(HFP), as expected: at 879 and 1053 cm$^{-1}$ for the –CF$_2$ stretching mode, and at 748, 1170, and 1402 cm$^{-1}$ owing to the presence of the α-phase in the PVdF(HFP). In addition, the FT-IR spectrum displays a peak at 839 cm$^{-1}$, which is associated with the β-phase of PVdF(HFP). These results imply that this electrode is...
composed of a mixture of α- and β-phases of PVdF(HFP). The peaks at 1572, 3122, and 3162 cm\(^{-1}\) are attributed to the presence of [EMI]\(^+\). The spectra for the other IL-based polymer-supported bucky-gel electrodes display similar features.

The XRD results obtained for the electrodes were in good agreement with the FT-IR results. The XRD pattern for the MnO\(_2\)/MWCNT/EMI[B\(_4\)F\(_4\)]/PVdF(HFP) (MnO\(_2\) : MWCNT = 1 : 1) electrode is shown in Fig. 2(b). The electrode exhibits characteristic peaks for PVdF(HFP), with major crystalline peaks at 2\(θ\) = 31.1° and 20.1° which correspond to the (110) crystalline planes of the α-phase.\(^{25}\) The carbon materials show typical peaks at 2\(θ\) = 22.2°, 24.6° and 45.4°.\(^{26}\) Suggesting its amorphous nature, the MnO\(_2\) film shows weak peaks at 2\(θ\) = 37.4° and 65.5°. The X-ray patterns of the other IL-based polymer-supported bucky-gel electrodes show the same features.

The electrode films were characterized by XPS. Fig. S1\(\dagger\) presents a comparison of the XPS spectra for MnO\(_2\)/MWCNT/IL/PVdF(HFP) electrodes with different MnO\(_2\) : MWCNT ratios (1 : 1, 0 : 1) and ILs (EMI[B\(_4\)F\(_4\)], EMI[CF\(_3\)SO\(_3\)]). In Fig. S1(a), clear peaks for Mn\(2p_{1/2}\) (655 eV) and Mn\(2p_{3/2}\) (642 eV) are observed for (A) (MnO\(_2\) : MWCNT = 1 : 1), whereas no peaks due to Mn are observed for (B) (MnO\(_2\) : MWCNT = 0 : 1). The XPS spectra of the other IL-based polymer-supported bucky-gel electrodes show the same features. Fig. S1(b)\(\dagger\) shows a pair of S2p\(_{1/2}\) and S2p\(_{3/2}\) peaks for the EMI[CF\(_3\)SO\(_3\)]-based electrode (A), whereas no corresponding peaks are observed for the EMI[B\(_4\)F\(_4\)]-based sample (B). In the O1s region (Fig. S1(c)\(\dagger\) (A-C)), peaks are individually observed for the CF\(_3\)SO\(_3\)- groups in MnO\(_2\)/MWCNT/EMI[CF\(_3\)SO\(_3\)]/PVdF(HFP) (MnO\(_2\) : MWCNT = 0 : 1) and the MnO\(_2\) in MnO\(_2\)/MWCNT/EMI[B\(_4\)F\(_4\)]/PVdF(HFP) (MnO\(_2\) : MWCNT = 1 : 1). The O1s peaks due to both MnO\(_2\) and CF\(_3\)SO\(_3\) are observed for MnO\(_2\)/MWCNT/EMI[CF\(_3\)SO\(_3\)]/PVdF(HFP) (MnO\(_2\) : MWCNT = 1 : 1). Fig. 3 presents the SEM micrographs (magnification 30 000\(\times\)) of polymer-supported bucky-gel electrode layers with MnO\(_2\) : MWCNT = 0 : 1 (composition = 20 wt% MWCNTs, 48 wt% EMI[B\(_4\)F\(_4\)], and 32 wt% PVdF(HFP)) (Fig. 3(a)) and MnO\(_2\) : MWCNT = 1 : 1 (composition = 16.7 wt% MWCNTs, 16.7 wt% MnO\(_2\)-xH\(_2\)O, 40 wt% EMI[B\(_4\)F\(_4\)], and 26.6 wt% PVdF(HFP)) (Fig. 3(b)). MnO\(_2\)-xH\(_2\)O particles can be seen near the MWCNTs in Fig. 3(b). The entangled MWCNTs (average diameter: 9.5 nm) form a network of open mesopores (Fig. 3(a)), and the bulk of the MnO\(_2\)-xH\(_2\)O (average diameter: <0.5 μm) is distributed through this network of open mesopores (Fig. 3(b)).

Fig. S2\(\dagger\) shows the Young’s moduli of the polymer-supported bucky-gel actuators containing MnO\(_2\)/MWCNT. For the six MnO\(_2\)/MWCNT/IL polymer gel electrodes, as well as the SWCNT/IL variants ([IL = EMI[B\(_4\)F\(_4\)], EMI[CF\(_3\)BF\(_3\)], EMI[CF\(_3\)COO], EMI[CF\(_3\)SO\(_3\)], EMI[FSI], or EMI[TFSI]], the Young’s modulus values are 80–120, 90–115, 80–110, 75–100, 85–110, 80–130, and 90–175 MPa, respectively. These results also support the entanglement of the MWCNTs to form a network of open mesopores, similar to the SWCNTs.

In this work, the use of MWCNTs is proposed as a means of improving the electrical conductivity of MnO\(_2\) electrodes used to build capacitors. The results presented herein demonstrate that MWCNTs are effective at increasing the capacitance of such devices and improving the electrochemical properties of MnO\(_2\) electrodes. This enhancement occurs because the use of MWCNT nanoparticles allows ions ready access to the MnO\(_2\) particles (Fig. S3\(\dagger\)). These results are considered to be a result of the synergistic effect between the MWCNTs and the metal oxide.\(^{29}\)

Electrochemical properties of the electrodes

Fig. 4 shows the specific capacitance \(C\) (where the specific capacitance of the nanocarbon, \(C_{\text{nаночастица}} = C_{\text{CNT}}/\text{[weight of nanocarbon]}\)) of the polymer-supported bucky-gel electrodes as a function of the CNT content and type of IL. The specific capacitance values of the MnO\(_2\)/MWCNT electrodes are larger than those of the electrodes containing only MWCNTs (referred to as only-MWCNT), and are proportional to the amount of MnO\(_2\) used. The specific capacitances for the MnO\(_2\)/MWCNT/EMI[B\(_4\)F\(_4\)] or [CF\(_3\)SO\(_3\)] [MnO\(_2\) : MWCNT = 0.25, 0.5, and 1.0] and MnO\(_2\)/MWCNT/EMI[CF\(_3\)BF\(_3\)] , EMI[CF\(_3\)COO], EMI[FSI], and EMI[TFSI] [MnO\(_2\) : MWCNT = 0.5 and 1.0] electrodes are larger than the those for the corresponding only-SWCNT electrodes. The specific capacitance values of the MnO\(_2\)/MWCNT/IL [MnO\(_2\) : MWCNT = 1.0] electrodes follow the order [B\(_4\)F\(_4\)] > [CF\(_3\)COO] and [CF\(_3\)SO\(_3\)] > [CF\(_3\)BF\(_3\)], [FSI], and [TFSI], and thus...

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**Fig. 3** SEM images (magnification 30 000\(\times\)) of (a) MWCNT/EMI[B\(_4\)F\(_4\)]/PVdF(HFP) (MnO\(_2\) : MWCNT = 0 : 1) electrode (composition = 20 wt% MWCNTs, 48 wt% EMI[B\(_4\)F\(_4\)], and 32 wt% PVdF(HFP)), and (b) MnO\(_2\)/MWCNT/EMI[B\(_4\)F\(_4\)]/PVdF(HFP) (MnO\(_2\) : MWCNT = 1 : 1) electrode (composition = 16.7 wt% MWCNTs, 16.7 wt% MnO\(_2\)-xH\(_2\)O, 40 wt% EMI[B\(_4\)F\(_4\)], and 26.6 wt% PVdF(HFP)).

**Fig. 4** Specific capacitance, \(C\), of MnO\(_2\)/MWCNT in polymer-supported IL gel electrodes for various MnO\(_2\) : MWCNT ratios (applied triangular voltage = ±0.5 V, sweep rate = 1 mV s\(^{-1}\)).
depend on the IL species in the electrode layer. The MnO2/MWCNT electrodes containing EMI[BF4] have the largest specific capacitance values of all systems described. These higher values are believed to result from a more optimized pore size distribution in the EMI[BF4] electrodes.

It is believed that the specific capacitance of the MnO2/MWCNT electrode results from both EDLC and FC mechanisms, with the FC mechanism providing the largest contribution. This combination of mechanisms differs from the capacitance mechanism in the electrodes made solely with MWCNTs or SWCNTs. Furthermore, this combined mechanism is considered to result from the synergistic effect between the MWCNTs and metal oxide. The energy storage mechanism associated with these oxides is based primarily on fast faradic redox reactions that occur between the oxide and the electrolyte that give rise to pseudo capacitance. Therefore, electrodes with these oxides can store more energy than the only-MWCNT electrodes used in typical specific capacitors. Consequently, these results may be largely attributed to the pseudo capacitance of the MnO2 species, which can be oxidized and reduced reversibly through the reaction presented in eqn (2).

\[
\text{MnO}_2 + \delta\text{EMI}^+ + \delta\text{e}^- = \text{MnO}_2 - \delta\text{EMI}_{3\delta}, \quad (0 < \delta < 2) \quad (2)
\]

Fig. S4† shows the electrical conductivities of various polymer-supported nanocarbon/IL gel electrode layers containing MnO2 as a function of the CNT content and type of IL. For the six MnO2/MWCNT/IL polymer gel electrodes, as well as the SWCNT/IL variants [IL = EMI[BF4], EMI[CF3COO], EMI[CF3SO3], EMI[FSI], or EMI[TFSI]], the electrical conductivity values are 7–11, 10–17, 9–11, 12–17, 11–17, 12–14, and 13–19 S cm\(^{-1}\), respectively. Therefore, the electrical conductivity depends on the composition of the IL species in the electrode layer.

Electromechanical properties and performance of the actuators

Fig. 5 shows the strain as a function of the applied triangular voltage (±2 V) for actuators containing different polymer-supported CNT/EMI[CF3SO3] bucky-gel electrodes. The strain is dependent on the measured frequency, as the CNTs dispersed in the electrode layer are fully charged at low frequencies (0.05–0.005 Hz). In contrast, at higher frequencies, there is not enough time for the dispersed CNTs to become fully charged. The strains in the other IL-based polymer-supported bucky-gel actuators show the same trends. The strains of the MnO2/MWCNT actuators (over 100–0.005 Hz) are greater than that of the only-MWCNT actuator. Furthermore, the strain of the MnO2 : MWCNT = 0.5 and 1.0 actuators (over the frequency range of 1–0.005 Hz) is larger than that of the only-SWCNT actuator. At low frequencies (0.05–0.005 Hz), the specific capacitance of the MnO2/MWCNT electrode exceeds that of the only-MWCNT electrode. It appears that the specific capacitance values of the MnO2/MWCNT electrodes originate from both EDLC and FC mechanisms, with the greatest contribution coming from the FC mechanism. This differs from the only-MWCNT or only-SWCNT electrodes.

The maximum strain values for the MnO2 : MWCNT = 0.5 and 1.0, only-MWCNT, and only-SWCNT actuators (0.56–0.95%, 0.84–1.85%, 0.09–0.26%, and 0.20–0.43%, respectively) are shown in Table 1. With respect to the IL species, the maximum strains for the MnO2 : MWCNT = 0.5 actuators follow the order \([\text{CF3SO3}] > [\text{BF4}] > [\text{CF3COO}] > [\text{TFSI}] > [\text{CF3BF3}] > [\text{FSI}]\), and the order for the MnO2 : MWCNT = 1.0 devices is \([\text{CF3SO3}] > [\text{BF4}] > [\text{CF3COO}] > [\text{TFSI}] > [\text{CF3BF3}] > [\text{FSI}]\). For all the metal oxide–carbon composite actuators, regardless of the IL, the strains are greater than those of the only-MWCNT actuators. Furthermore, the strains of the MnO2 : MWCNT = 0.5 and 1.0 actuators are greater than those of the corresponding only-SWCNT actuators. The maximum strains of the MnO2 : MWCNT = 1.0 actuators in EMI[BF4], EMI[CF3BF3], EMI[CF3COO], EMI[CF3SO3], EMI[TFSI], and EMI[CF3BF3] are larger by factors of 4, 2.3, 6.7, 6.2, 3.3, and 3.5 compared to the SWCNT actuators in the equivalent IL. The maximum strain values of the MnO2/MWCNT electrodes containing MnO2 are larger than those of the only-MWCNT and only-SWCNT actuators, and similarly, the specific capacitance values of the MnO2/MWCNT electrodes are larger than those of the only-MWCNT and only-SWCNT electrodes. Most likely, the specific capacitances of the MnO2/MWCNT/IL electrodes are the result of both EDLC and FC effects, with the largest capacitance contribution coming from

<table>
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<th>IL</th>
<th>MnO2/MWCNT = 0.5</th>
<th>MnO2/MWCNT = 1.0</th>
<th>MWCNT</th>
<th>SWCNT</th>
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<td>1.85</td>
<td>0.13</td>
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</tr>
<tr>
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<tr>
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<td>0.77</td>
<td>1.23</td>
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the FC mechanism. It appears that an additional factor other than the FC effect contributes to the strain.

Fig. 6 compares the maximum generated stresses of the polymer-supported bucky-gel actuators containing MnO2/MWCNTs with the SWCNT-based actuators. The maximum generated stress (σ) during actuation was calculated from the maximum strain (εmax) and Young’s modulus (Y), according to Hooke’s law, Y = εmax. The maximum stresses generated by the MnO2/MWCNT actuators exceed those of the only-MWCNT actuators. In addition, the maximum stresses generated by the MnO2 : MWCNT = 0.5 and 1.0 actuators are greater than those of the only-SWCNT actuators. In terms of the IL species, the maximum generated stresses for the MnO2 : MWCNT = 1.0 actuators follow the order [CF3COO] > [BF4] > [TFSI] > [CF3SO3] > [CF3BF3] > [FSI]. More interestingly, the σ values for the MnO2 : MWCNT = 1.0 actuators are ca. 1.7 ([BF4]), ca. 2.0 ([CF3BF3]), ca. 5.0 ([CF3COO]), ca. 3.2 ([CF3SO3]), ca. 1.6 ([FSI]), and ca. 2.5 ([TFSI]) times larger than those of the only-SWCNT actuators.

These results can be attributed to the large capacitances of these devices. At low frequencies, the strain of the MnO2/MWCNT/IL actuator is likely the primary contributor to the pseudo capacitance of the actuator. In addition, the MnO2/MWCNT polymer actuators fabricated in this study surpass the performance of the only-MWCNT and only-SWCNT actuators. Consequently, these results demonstrate that MnO2/MWCNT actuators prepared using common nanocarbon materials can generate sufficient strain for real-world applications (such as tactile displays), without the need for specialized SWCNTs.

**Actuation mechanism**

We previously proposed a mechanism for the bending of a SWCNT actuator.3 In this mechanism, the application of a voltage between the two electrode layers leads to transfer of cations and anions in the gel electrolyte layer to the cathode and anode layers, respectively, and thus forms an electric double layer with negatively and positively charged nanotubes. Such ion transport results in the swelling of the cathode layer and shrinkage of the anode layer; hence, the actuator bends towards the anode. In the present work, the capacitances of the MnO2/MWCNT/IL devices are 22–196 F g−1, while those of the MWCNT/IL electrodes are 10–19 F g−1. We, therefore, propose that the contribution of the EDLC mechanism is minor at low frequencies, while that of the pseudo capacitance is predominant. As noted, in hybrid capacitors such as these, both the EDLC and FC mechanisms occur simultaneously, although one will have a more pronounced effect (Fig. 7).

Fig. S5† shows a CV for a cell with the EMI[CF3BF3] electrolyte sandwiched between non-MWCNT-containing gel electrode layers (MnO2·xH2O/EMI[CF3BF3]/PVdF(HFP)). The CV exhibits a capacitive wave at this sweep rate = 40 mV s−1 (0.005 Hz); similar CVs were obtained for the other IL-based cell systems. The capacitances of the EMI[BF4], EMI[CF3BF3], EMI[CF3COO], EMI[CF3SO3], EMI[FSI], and EMI[TFSI]-based electrodes (without MWCNTs) are 0.565, 0.043, 0.852, 1.315, 0.020, and 0.035 F g−1, respectively, which are smaller than the specific capacitance values obtained for the only-SWCNT or MnO2/MWCNT electrodes. Furthermore, the electrical conductivities of the EMI[BF4], EMI[CF3BF3], EMI[CF3COO], EMI[CF3SO3], EMI[FSI], and EMI[TFSI]-based electrodes (without MWCNTs) are 3.5, 1.2, 8.1, 2.4, 0.7, and 1.0 mS cm−1, respectively, i.e. ca. 1/100000 of the electrical conductivity values of the MnO2/MWCNT electrodes. Therefore, MWCNTs are required to obtain the desired high specific capacitance and electrical conductivity.

Fig. 8 compares the strain calculated from the peak-to-peak values of the displacement as a function of the applied triangular voltage (±2 V) for the polymer-supported CNT/EMI [CF3BF3] gel actuators with the polymer-supported non-MWCNT/EMI[CF3BF3] gel actuators containing MnO2. The strain of the non-CNT/IL gel actuator containing MnO2 is less than those of the only-MWCNT and MnO2/MWCNT actuators. The strains in the other polymer-supported nanocarbon/IL gel actuators show the same trends. Therefore, the actuator should contain both MWCNTs and MnO2 to demonstrate performance that surpasses that of the SWCNT polymer actuator.
Electromechanical simulation of the actuators

In a previous study, the voltage–current and voltage–displacement characteristics of a bucky-gel actuator were investigated by applying a voltage with a triangular waveform at various frequencies. From the resulting data, we found that the frequency dependence of the displacement response of carbon black (CB)/vapour-grown carbon fibre (VGCF)/IL polymer actuators could be successfully simulated using an electrochemical kinetic model similar to that used for SWCNT-based actuators. In this manner, the strain at the low-frequency limit and the time constant of the response could be predicted.

In the present study, the frequency dependence of the displacement responses of the MnO2·xH2O/MWCNT actuators was successfully simulated using a double-layer charging kinetic model. This model, which takes into account the oxidation and reduction reactions of the metal oxide, is similar to the model used previously for SWCNT-based and CB/VGCF/IL actuators. This newer model also allows the prediction of the strain at the low-frequency limit and the time constant of the response.

The time constant is represented by the equivalent circuit in a series combination of the ionic resistance $R_i$ specific capacitance $C$, and electrode resistance $R_e$. Eqn (S1) and (S2) show that strain is related to the time constant. Inspection of the simulated results (solid curves in Fig. S6†) indicates that there is a strain dependence on the frequency for the MnO2/MWCNT (MnO2 : MWCNT = 1.0) gel actuator containing various ILs (Fig. 9). In the case of MnO2, at high frequencies (0.1–10 Hz), since the time constant ($C(R + R_e)$) trend is EMI[CF3SO3] > EMI[CF3COO] > EMI[BF4] > EMI[CF3BF3] > EMI[TFSI] > EMI[FSI], the actuator strain trend is EMI[CF3SO3] > EMI[CF3COO] > EMI[BF4] > EMI[CF3BF3] > EMI[TFSI] > EMI[FSI].

Conclusions

The electrochemical and electromechanical properties of actuators prepared using MnO2/MWCNT/IL electrodes formed without ultrasonication were compared with those of actuators prepared using solely MWCNTs or SWCNTs on various ILs.

Several aspects of these new actuators were elucidated during this work, which are as follows. The electrode of this actuator system represents an EC resulting from EDLC and FC mechanisms, and the capacitance of the metal oxide MnO2/MWCNT/IL electrode primarily results from the FC mechanism such that, at low frequencies, the strain of the MnO2/MWCNT/IL actuator is the dominant contributor to the FC. The primary effect of the MWCNTs in this device appears to be the increase of the electroconductivity of the MnO2/MWCNT/IL actuator. The process by which these MnO2/MWCNT/IL actuators function was found to be different from that of the devices produced using solely MWCNTs or SWCNTs that incorporate only the EDLC mechanism. In addition, both MO2 and MWCNTs are required to produce actuators capable of showing significant strain that surpass the performance of SWCNT-based polymer actuators, and are thus suitable for practical applications.

The variation of the displacement with frequency for each MnO2·xH2O/MWCNT actuator was determined based on simulations employing a double-layer charging kinetic model that incorporated the oxidation and reduction reactions of the metal oxide, and was also applicable to SWCNT-based actuators on all ILs.

The frequency dependence of the displacement response of the MnO2/MWCNT actuators can be successfully simulated by the double-layer charging kinetic model. The strain at the limit of low frequency is related to the electromechanical mechanism in the MnO2/MWCNT actuator. At high frequencies (0.1–10 Hz), the time constant depends on the IL species; therefore, the strain is considered to be related to the time constant.

Based on the data acquired in this study, flexible and robust films operating via the synergistic effects of a MnO2 combined with MWCNTs show promise as electrode materials for synthesizing wearable and energy conversion devices. This same concept can also be used with other electrochemical materials containing metal oxides for applications in energy conversion systems, allowing for an increased range of applications.
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Notes and references