

Cite this: *Nanoscale*, 2016, 8, 183Received 7th October 2015,  
Accepted 28th November 2015

DOI: 10.1039/c5nr06937h

www.rsc.org/nanoscale

## Resistive pressure sensors based on freestanding membranes of gold nanoparticles†

Hendrik Schlicke, Matthias Rebber, Svenja Kunze and Tobias Vossmeier\*

In this communication the application of gold nanoparticle membranes as ambient pressure sensors with electromechanical signal transduction is demonstrated. The devices were fabricated by sealing microstructured cavities with membranes of 1,6-hexanedithiol cross-linked gold nanoparticles, which were electrically contacted by metal electrodes deposited on both sides of the cavities. Variations of the external pressure resulted in a deflection of the membranes and, thus, increased the average interparticle distances. Therefore, the pressure change could easily be detected by simply monitoring the resistance of the membranes.

Due to their tunable electronic, mechanical and optical properties, freestanding membranes of gold nanoparticles (GNPs) capped with monofunctional ligands, interlinked with bi- or multifunctional molecular cross-linkers or embedded in polymer matrices are promising candidates for the application as functional materials in micro- and nanoelectromechanical systems (MEMS/NEMS). In contrast to conventional silicon based MEMS/NEMS requiring elaborate multi-step lithographic fabrication schemes, noble metal nanoparticle composites can be fabricated and deposited by cost-effective procedures such as spin-coating, ink-jet printing,<sup>1,2</sup> stamping<sup>3</sup> and are compatible with a broad variety of rigid or flexible substrate materials.

Recently, a series of MEMS/NEMS applications was demonstrated by several research groups. Kanjanaboos *et al.* demonstrated the fabrication of microscale drumhead resonators from freestanding monolayers of monothiol-capped GNPs with resonance frequencies in the MHz range.<sup>4</sup> Tsukruk and co-workers reported the fabrication of Golay type IR microimagers based on nanocomposite GNP/polymer membranes deposited on microcavities.<sup>5</sup> The group also presented the application of freestanding membranes of polymer-encapsulated GNPs as

pressure gauges.<sup>6</sup> However, optical readout techniques of the membrane deflection were applied in all cases. More recently, we reported on the first electrostatic actuator based on a free-standing membrane of alkanedithiol (ADT) cross-linked GNPs.<sup>7</sup>

GNP membranes cross-linked using dithiols offer the ability to adjust the conductivity over several orders of magnitude.<sup>8,9</sup> Using shorter-chain ADTs, fairly good conductivities can be achieved. Analogously to thin films of monothiol-capped GNPs deposited on flexible substrates,<sup>10</sup> the charge transport through assemblies of ADT cross-linked GNPs is sensitive to strain, directly affecting the interparticle distances and impeding tunnel currents. Based on this behavior, the applicability of substrate supported dithiol cross-linked GNP films as sensitive strain gauges with gauge factors  $G_s$  of 10–20 was demonstrated.<sup>11,12</sup>

Previously, freestanding membranes of highly ordered GNP monolayers were fabricated and their remarkable elastic properties were probed by AFM indentation measurements.<sup>13,14</sup> Si *et al.* reported the fabrication of freestanding monolayer sheets from Au@Ag nanocubes, which could be lithographically milled into nanoribbons or folded into 3d origami.<sup>15</sup> In a recent study<sup>16</sup> we reported on the fabrication of freestanding 1,9-nonanedithiol (9DT) cross-linked GNP membranes on circular apertures (~100  $\mu\text{m}$  diameter) and the characterization of their viscoelastic properties by AFM micro-bulge tests.<sup>17</sup> The Young's modulus was measured in the low GPa range and it was demonstrated that these membranes withstand pressure loadings of several kPa, corresponding to a biaxial stress of tens of MPa.<sup>16</sup> These mechanical properties together with their electric conductivity make crosslinked gold nanoparticle membranes interesting functional materials for MEMS/NEMS applications.

In this communication we present the fabrication of resistive pressure gauges employing a 1,6-hexanedithiol (6DT) cross-linked GNP membrane as strain sensitive transducer. To the best of our knowledge, this is the first report on sensors utilizing the unique electromechanical properties of a free-standing nanoparticle membrane to enable a direct and simple resistive signal readout.

Institute of Physical Chemistry, University of Hamburg, Grindelallee 117, 20146 Hamburg, Germany. E-mail: tobias.vossmeier@chemie.uni-hamburg.de;  
Fax: +49 40 42838 3452; Tel: +49 40 42838 7069

† Electronic supplementary information (ESI) available. See DOI: 10.1039/c5nr06937h

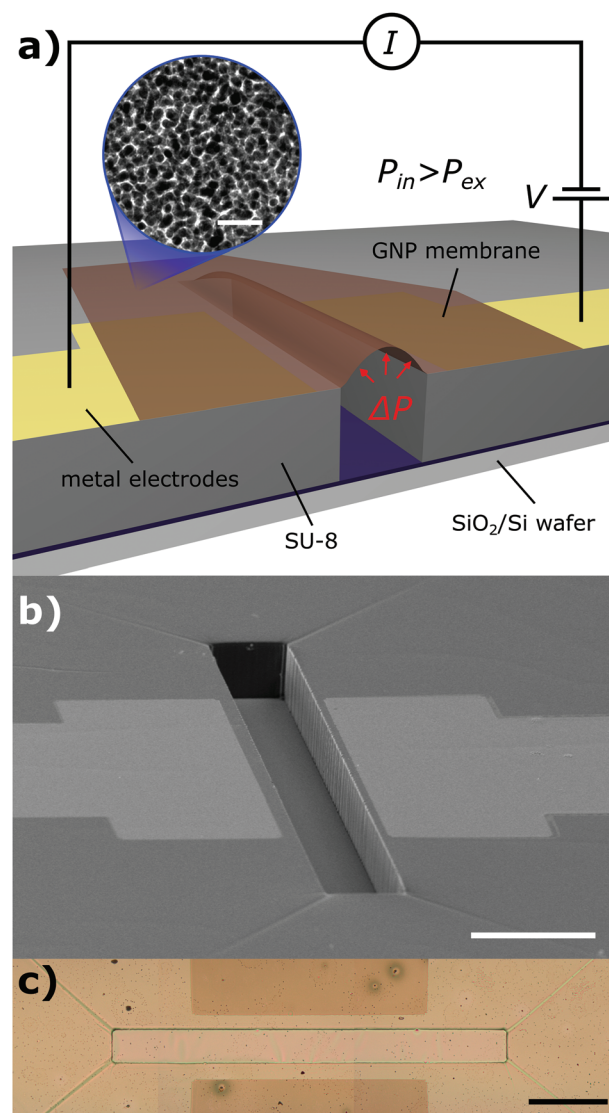


For fabricating the sensor devices, GNP membranes were deposited onto 3d microstructures featuring rectangular cavities, typically of  $\sim 40\ \mu\text{m}$  in width and  $\sim 500\ \mu\text{m}$  in length. The microstructures were prepared from SU-8, a negative tone photoresist commonly used for the fabrication of MEMS structures,<sup>18,19</sup> using standard photolithography on thermally oxidized silicon wafers (oxide thickness 300 nm). The depth of the cavities ( $\sim 40\ \mu\text{m}$ ) could be trimmed by varying the resist layer thickness. In proximity to the long sides of the cavities, gold or platinum electrodes ( $\sim 40\ \text{nm}$  thickness) were deposited for measuring the resistance of the freestanding membrane sections. A schematic showing the device architecture is presented in Fig. 1a. Fig. 1b shows a SEM image of the microcavity prepared in SU-8 resist with proximal metal electrodes, before depositing the nanoparticle membrane. A detailed description of the microstructure fabrication process is provided in the ESI†

6DT cross-linked GNP membranes were fabricated by spin-coating a heptanoic solution of GNPs (diameter  $(3.5 \pm 0.7)\ \text{nm}$ ) and a methanolic solution of 6DT alternately onto a glass substrate, as reported earlier.<sup>20</sup> The resulting GNP films showed a typical conductivity of around  $0.1\ \text{S cm}^{-1}$ .<sup>7</sup> Thicknesses of the films were measured by atomic force microscopy (AFM).<sup>20</sup> A TEM image revealing the granular structure of the 6DT cross-linked GNP membrane is shown as inset in Fig. 1a. See the ESI† for a representative absorbance spectrum, a current-voltage ( $I$ - $V$ ) measurement, and an AFM scan of the substrate supported GNP film used for device fabrication.

Following deposition, the cross-linked GNP films were transferred from their initial substrates by floating them on demineralized water.<sup>20</sup> After an incubation period of up to two days at ambient conditions, the membranes could easily be lifted off their glass substrates by carefully immersing the latter into water. Doing this, the membranes detached from the substrates and remained floating at the liquid-air interface. Subsequently, the electrode microstructures were used to skim the membranes from the water surface. While the samples were allowed to dry, the membranes settled to the 3d structures and remained freestanding over the rectangular cavities. The substrates were then fixed onto custom-designed printed circuit boards, contacted and transferred to a custom-built pressure cell. A Keithley 2601A source measure unit was employed to source a constant voltage of 1 V across the device and to measure the resulting current. The pressure cell was equipped with a cascade of valves, suitable for applying pressure transients in a range of  $\pm 10\ \text{kPa}$ , relative to ambient pressure. As reference sensor a digital pressure gauge (Sensor-technics HDIM100DBF8P5) was connected to the cell. Details of the setup can be found in the ESI.†

Fig. 1c shows the optical micrograph of a device with a 6DT cross-linked GNP membrane (55 nm thickness) spanning the rectangular microfabricated cavity. Also the gold electrodes ( $\sim 300\ \mu\text{m}$  width,  $\sim 80\ \mu\text{m}$  distance), covered by the membrane are clearly recognized. An initial resistance of 259 k $\Omega$  was determined for this device, corresponding to a sheet resistance of  $R_s = 0.95\ \text{M}\Omega$ . Compared to conventional metal foil strain



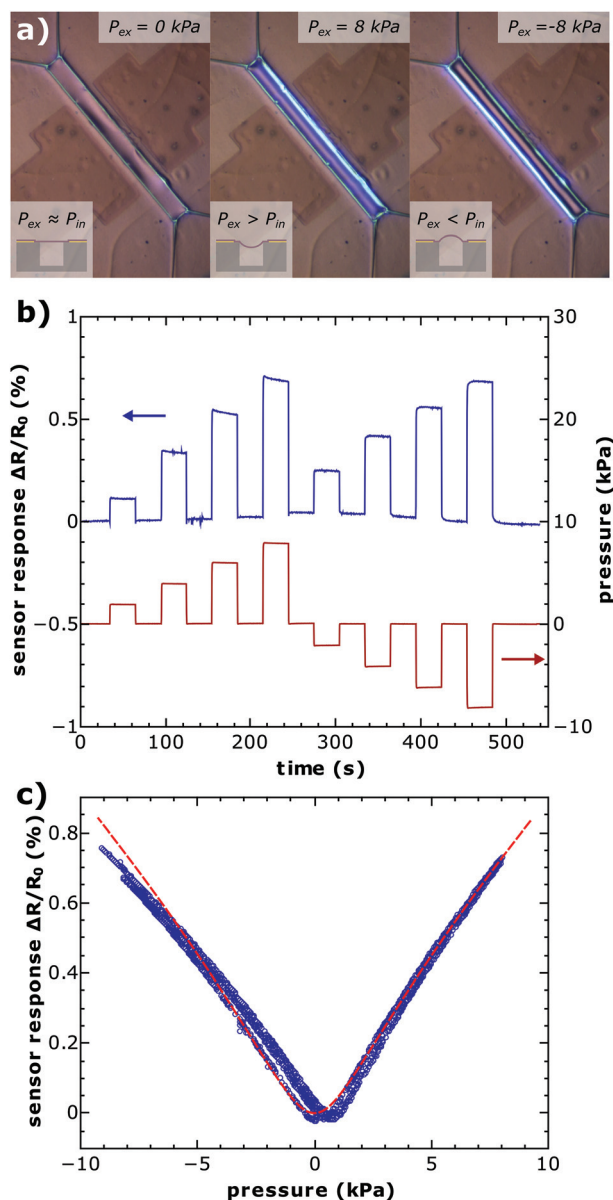
**Fig. 1** (a) Schematic showing a cross-sectional view of a GNP membrane based pressure sensor. The GNP membrane is sealing the cavity microstructured into SU-8 photoresist. Electrodes deposited onto the SU-8 layer are used to monitor the membrane's resistance under applied pressure loading. The inset shows a TEM image of a 1,6-hexanedithiol cross-linked GNP membrane (scale bar: 25 nm). (b) SEM image of a microcavity with proximal metal electrodes prior to membrane deposition. Scale bar: 100  $\mu\text{m}$ . (c) Optical micrograph of a 6DT cross-linked GNP membrane deposited onto a 3d electrode microstructure. Scale bar: 100  $\mu\text{m}$ .

gauges the significantly higher resistance of GNP based transducers allows for operating the devices under lower power dissipation.<sup>10</sup> Additionally, it is straightforward to adjust the sheet resistance for specific device requirements because the resistivity of cross-linked GNP films can be tuned over several orders of magnitude by varying the size<sup>8,9,20</sup> and structure<sup>21</sup> of the cross-linker.

The application of positive pressure transients of up to 8 kPa resulted in pressure differences between the cavity and the



exterior, causing an inward deflection of the freestanding membrane. Movements of the membrane could clearly be observed using a microscope camera (Fig. 2a, middle). A movie



**Fig. 2** (a) Optical micrographs of a pressure gauge fabricated from a 6DT cross-linked GNP membrane placed in a pressure cell under an external pressure loading of (left) 0 kPa, (center) 8 kPa and (right) –8 kPa (relative to the ambient pressure). The deflections of the membrane inwards and outwards the rectangular microcavity are observable by the reflections of the non-centered incident microscope illumination on opposite sides of the bulge. (b) Pressure transients (0 to  $\pm 8$  kPa) applied to the device (red) and resulting resistance changes relative to the baseline resistance  $R_0$  (blue). (c) Transfer function of the device relating the applied external pressure to the measured resistance change. An estimate of the transfer function, which is based on the membrane's mechanical properties, the dimensions and geometry of the device, and a gauge factor  $G_s$  of 7, is depicted as dashed red line. Details to the underlying calculation are provided in the ESI.†

showing the periodical deflections of the nanoparticle membrane induced by the repeated pressure loadings shown in Fig. 2b can be found in the ESI.† The deflection of the membrane was accompanied by an increase in resistance as shown by the response transients in Fig. 2b. When negative external pressures of up to –8 kPa (relative to ambient) were applied again pressure differences between the cavity and the exterior were established. In this case the freestanding membrane showed outward deflections while remaining attached to the SU-8 layer surrounding the cavity (Fig. 2a, right). As expected, the device responded with a similar increase in resistance (up to  $\sim 0.7\%$ ) for positive and negative applied pressures (see Fig. 2b), especially in the higher pressure range, resulting from similar strain experienced by the membrane in both cases, inward and outward deflection. It is to note that the device remained functional over the course of the study of several weeks.

Fig. 2c shows the sensor's transfer curves obtained by repeatedly sweeping the external pressure and continuously monitoring the current at an applied voltage of 1 V. Corresponding resistance and pressure time series can be found in the ESI.† The sensor displayed an almost linear response for the positive and negative external pressure branch. A slight offset of the response curve's minimum is observed and is attributed to a slight difference between the initial cavity pressure and the ambient pressure.

The response of the pressure sensor was estimated by taking into account the mechanical properties of 6DT cross-linked GNP membranes obtained by AFM bulge tests (ESI†).<sup>7</sup> Using this data and the device geometry, an estimate of the bulge height of the membrane in response to the applied pressure loading as well as the resulting strain could be calculated. Taking into account a constant gauge factor  $G_s$  of 7 and a residual membrane stress of 6 MPa (values in the MPa range are common for freestanding GNP membranes<sup>7,16</sup>), resulted in estimates of the sensor responses which are, within the given pressure range of  $\pm 8$  kPa, very similar to the experimental data. This is clearly seen in Fig. 2c, which shows the calculated transfer function (dashed red line) in very good agreement with the experimental response curve. A detailed description of the calculation as well as simulations regarding the influences of different device parameters on the sensor's response characteristics are provided in the ESI.†

In summary, we presented the first prototype of a resistive pressure sensor based on a freestanding membrane of gold nanoparticles. A sensor response of up to  $\sim 0.7\%$  was obtained for pressure changes in a range of  $\pm 8$  kPa. The measured sensor responses were found in good agreement with a simple approximate model relating the applied external pressures to the membrane strain, and assuming a gauge factor  $G_s$  of 7, similar as observed for substrate supported strain gauges based on 9DT cross-linked GNP coatings.<sup>11</sup> According to previous findings reported for strain gauges based on GNP assemblies, it should be possible to enhance the sensitivity of the presented pressure sensor by the following adjustments: first, it was demonstrated that the resistive strain sensitivity of sub-





strate-supported GNP films increased with decreasing thickness and was highest for monolayer films. This finding was attributed to the 2d confinement of conduction paths in the case of the monolayer film.<sup>22</sup> Additionally, as shown by our model calculation provided in the ESI,† decreasing the membrane thickness increases the device's sensitivity because thinner membranes are bulged more heavily than thicker ones at a given pressure. Second, an increase of the nanoparticle size is assumed to increase the sensitivity, because the gauge factor of nanoparticle based resistive strain gauges scales with the particle diameter, as deduced from a simple geometric model.<sup>10</sup> Third, the order of the particles in a freestanding membrane is expected to influence the strain sensitivity. Compared to a highly ordered GNP membrane a disordered membrane responds to stress more easily by local rearrangements, microcrack formation and local relaxation rather than by homogeneous changes in the interparticle distances.<sup>23</sup> Taken together, ultimate sensitivities should be achievable by employing membranes consisting of a single, highly ordered monolayer of relatively large GNPs. This, however, will require reducing the device's dimensions in order to avoid collapse of the membrane and to enable electrode aspect ratios allowing for practical resistance measurements. Additionally, we note that by applying a more elaborate capacitive readout of the membrane deflection, it should be possible to determine the direction of the pressure variation with respect to the internal cavity pressure.

## Acknowledgements

The work of H. S. is supported by a scholarship of the Joachim Herz Foundation. T. V. acknowledges financial support by the German Research Foundation (DFG), grant number VO698/3-1. Furthermore we thank Stefan Werner for transmission electron microscopy measurements.

## References

- 1 S. Fuller, E. Wilhelm and J. Jacobson, *J. Microelectromech. Syst.*, 2002, **11**, 54–60.
- 2 E. S. Park, Y. Chen, T.-J. K. Liu and V. Subramanian, *Nano Lett.*, 2013, **13**, 5355–5360.
- 3 B. Kowalczyk, M. M. Apodaca, H. Nakanishi, S. K. Smoukov and B. a. Grzybowski, *Small*, 2009, **5**, 1970–1973.
- 4 P. Kanjanaboos, X.-M. Lin, J. E. Sader, S. M. Rupich, H. M. Jaeger and J. R. Guest, *Nano Lett.*, 2013, **13**, 2158–2162.
- 5 C. Jiang, M. E. McConney, S. Singamaneni, E. Merrick, Y. Chen, J. Zhao, L. Zhang and V. V. Tsukruk, *Chem. Mater.*, 2006, **18**, 2632–2634.
- 6 C. Jiang, S. Markutsya, Y. Pikus and V. V. Tsukruk, *Nat. Mater.*, 2004, **3**, 721–728.
- 7 H. Schlicke, D. Battista, S. Kunze, C. J. Schröter, M. Eich and T. Vossmeier, *ACS Appl. Mater. Interfaces*, 2015, **7**, 15123–15128.
- 8 Y. Joseph, I. Besnard, M. Rosenberger, B. Guse, H.-G. Nothofer, J. M. Wessels, U. Wild, A. Knop-Gericke, D. Su, R. Schlögl, A. Yasuda and T. Vossmeier, *J. Phys. Chem. B*, 2003, **107**, 7406–7413.
- 9 M. Brust, D. J. Schiffrin, D. Bethell and C. J. Kiely, *Adv. Mater.*, 1995, **7**, 795–797.
- 10 J. Herrmann, K. H. Müller, T. Reda, G. R. Baxter, B. Raguse, G. J. J. B. De Groot, R. Chai, M. Roberts and L. Wiczorek, *Appl. Phys. Lett.*, 2007, **91**, 183105.
- 11 T. Vossmeier, C. Stolte, M. Ijeh, A. Kornowski and H. Weller, *Adv. Funct. Mater.*, 2008, **18**, 1611–1616.
- 12 M. Segev-Bar and H. Haick, *ACS Nano*, 2013, **7**, 8366–8378.
- 13 W. Cheng, M. J. Campolongo, J. J. Cha, S. J. Tan, C. C. Umbach, D. A. Muller and D. Luo, *Nat. Mater.*, 2009, **8**, 519–525.
- 14 K. E. Mueggenburg, X.-M. Lin, R. H. Goldsmith and H. M. Jaeger, *Nat. Mater.*, 2007, **6**, 656–660.
- 15 K. J. Si, D. Sikdar, Y. Chen, F. Eftekhari, Z. Xu, Y. Tang, W. Xiong, P. Guo, S. Zhang, Y. Lu, Q. Bao, W. Zhu, M. Premaratne and W. Cheng, *ACS Nano*, 2014, **8**, 11086–11093.
- 16 H. Schlicke, E. W. Leib, A. Petrov, J. H. Schröder and T. Vossmeier, *J. Phys. Chem. C*, 2014, **118**, 4386–4395.
- 17 A. Turchanin, A. Beyer, C. T. Nottbohm, X. Zhang, R. Stosch, A. Sologubenko, J. Mayer, P. Hinze, T. Weimann and A. Götzhäuser, *Adv. Mater.*, 2009, **21**, 1233–1237.
- 18 E. H. Conradie and D. F. Moore, *J. Micromech. Microeng.*, 2002, **12**, 368–374.
- 19 H. Lorenz, M. Despont, N. Fahrni, N. LaBianca, P. Renaud and P. Vettiger, *J. Micromech. Microeng.*, 1997, **7**, 121–124.
- 20 H. Schlicke, J. H. Schröder, M. Trebbin, A. Petrov, M. Ijeh, H. Weller and T. Vossmeier, *Nanotechnology*, 2011, **22**, 305303.
- 21 J. M. Wessels, H.-G. Nothofer, W. E. Ford, F. von Wrochem, F. Scholz, T. Vossmeier, A. Schroedter, H. Weller and A. Yasuda, *J. Am. Chem. Soc.*, 2004, **126**, 3349–3356.
- 22 C. Farcau, H. Moreira, B. Viallet, J. Grisolia, D. Ciuculescu-Pradines, C. Amiens and L. Ressler, *J. Phys. Chem. C*, 2011, **115**, 14494–14499.
- 23 N. Olichwer, E. W. Leib, A. H. Halfar, A. Petrov and T. Vossmeier, *ACS Appl. Mater. Interfaces*, 2012, **4**, 6151–6161.

