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Spin–orbit effects on the ^{119}Sn magnetic-shielding tensor in solids: a ZORA/DFT investigation†

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Periodic-boundary and cluster calculations of the magnetic-shielding tensors of ^{119}Sn sites in various co-ordination and stereochemical environments are reported. The results indicate a significant difference between the predicted NMR chemical shifts for tin(II) sites that exhibit stereochemically-active lone pairs and tin(IV) sites that do not have stereochemically-active lone pairs. The predicted magnetic shieldings determined either with the cluster model treated with the ZORA/Scalar Hamiltonian or with the GIPAW formalism are dependent on the oxidation state and the co-ordination geometry of the tin atom. The inclusion of relativistic effects at the spin–orbit level removes systematic differences in computed magnetic-shielding parameters between tin sites of differing stereochemistries, and brings computed NMR shielding parameters into significant agreement with experimentally-determined chemical-shift principal values. Slight improvement in agreement with experiment is noted in calculations using hybrid exchange–correlation functionals.

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1. Introduction

Solid-state nuclear magnetic resonance (NMR) spectroscopy is an important tool for studying ^{119}Sn -containing materials because experimental parameters, particularly the magnetic-shielding tensor, are quite sensitive to structural features.^{1–3} Experimental ^{119}Sn NMR parameters^{3–7} depend on the local co-ordination geometry and share important similarities with those of ^{207}Pb .^{8–10} For instance, tin(II) species generally exhibit hemidirected co-ordination chemistry^{11,12} with characteristically wide chemical-shift spans ($\Omega = |\delta_{33} - \delta_{11}|$) of 600–1200 ppm.^{5,6,13} In contrast, tin(IV) species generally exhibit holodirected co-ordination chemistry^{11,14,15} with spans under 400 ppm.^{4,5} The differences in magnetic-shielding parameters between the two structural motifs reflect the fact that crystal structures of tin(II) compounds usually have large void spaces to accommodate the lone pair of electrons, whereas the co-ordination geometry around tin(IV) sites is more nearly spherically symmetric. A recent computational study of ^{207}Pb NMR parameters illustrates the differences among NMR parameters of lead nuclei at sites of

different stereochemistry and how these differences are enhanced by relativistic effects, particularly by spin–orbit (SO) coupling.¹⁶ A similar result is expected for ^{119}Sn .

Quantum-chemical calculations provide the connection between NMR parameters and structure.^{17,18} For NMR-active nuclei in period 6, such as ^{195}Pt , ^{199}Hg , and ^{207}Pb , one must use relativistic theory, including SO coupling, for accurate predictions of the principal components of the magnetic-shielding tensors, and thus parameters such as the isotropic chemical shift.^{19–31} For nuclei in period 5 (^{119}Sn , ^{125}Te , ^{113}Cd), there remains some ambiguity in the literature regarding the impact of relativistic effects on calculated NMR parameters.^{22,32–37} Benchmark studies are important in determining the efficacy of various computational protocols.

There are suggestions in the literature that relativistic effects, particularly SO coupling, are important in calculating magnetic shielding of ^{119}Sn nuclei.^{38–40} For example, Malkin *et al.*⁴¹ recently proposed that the ^{119}Sn magnetic-shielding scale should be increased by around 1000 ppm, a result obtained from the four-component (4c) relativistic theory. Strong correlations between experiment and theory have been suggested for various tin-containing molecules at the non-relativistic DFT level as well.^{33,42,43} Non-relativistic calculations are usually justified by the possible cancellation of relativistic effects when calculated ^{119}Sn magnetic shieldings are converted to the chemical-shift scale (relative to a reference compound such as $\text{Sn}(\text{CH}_3)_4$).^{22,34} In other studies, it is suggested that SO coupling cannot be ignored for accurate predictions of chemical shifts.^{37,44} Bagno *et al.*³⁷ have discussed applying the zeroth-order regular

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approximation (ZORA) to ^{119}Sn magnetic shielding and spin-spin coupling constants of tin(IV)-containing species. They have shown that inclusion of SO coupling is important when other heavy atoms such as bromine or iodine are directly bonded to tin.³⁷ When tin is coordinated to lighter atoms, chemical shifts and spin-spin couplings calculated with inclusion of SO coupling and without it are quite similar, suggesting that inclusion of such terms is not significant.³⁷ Predicted isotropic ^{119}Sn chemical shifts in tin-containing solids calculated with plane-wave DFT techniques employing relativistic pseudopotentials at the ZORA level (without the SO component) are in good agreement with isotropic chemical shifts, although the calculated values do deviate by up to 200 ppm from experiment in some cases.⁴⁵

The role of SO coupling on calculated magnetic-shielding tensors is highly system-dependent.¹⁶ In this study, we present calculations of the principal components of ^{119}Sn magnetic-shielding tensors for tin sites with different oxidation states and different co-ordination environments. In all cases, we compare the tensor elements, rather than isotropic shifts.^{29,31} Comparisons of various theoretical approaches are presented, with or without SO coupling, using the ZORA-DFT methodology. Calculations are performed in plane-wave and cluster-based frameworks to demonstrate the differences in these two approaches. The utility of cluster-based calculations for the predictions of magnetic-shielding tensors of nuclei in solids has been established in several studies.^{29,31,46–55} The effect of hybrid functionals on computed magnetic shielding is also assessed. Our motivation is to understand the impact of various theoretical considerations on the quality of predicted values of magnetic shielding for ^{119}Sn -containing systems.

2. Computational details

Twelve tin-containing solids with known X-ray or neutron diffraction structures and with known principal components of the chemical-shift tensor determined by solid-state NMR spectroscopy have been investigated. The tin(II)-containing solids are SnO ,^{15,56} SnHPO_4 ,^{13,57} SnHPO_3 ,^{13,57} SnC_2O_4 ,^{13,58} SnSO_4 ,^{13,59} and BaSnF_4 .^{13,60} The tin(IV)-containing solids are SnO_2 ,^{5,12} Ca_2SnO_4 ,^{4,61} SnS_2 ,^{7,62} Pb_2SnO_4 ,^{63,64} $\text{Na}_6\text{Sn}_2\text{S}_7$,^{7,65} and Sr_2SnO_4 .^{4,66} The ^{119}Sn magnetic-shielding tensors in these solids were calculated using (1) a cluster-based approach and (2) a periodic approach, as discussed below.

Calculations on cluster models were performed using the Amsterdam Density Functional (ADF) program package.^{67–69} All clusters were expanded around the ^{119}Sn site up to the third co-ordination shell. The all-electron (AE) TZ2P basis set was employed for the NMR-active nucleus (^{119}Sn) and the first co-ordination shell around the NMR-active nucleus, whereas the remainder of the cluster was treated with the smaller AE TZP basis set. Two example clusters, including a schematic of the partitioning of the basis sets, are illustrated in Fig. 1. Calculations were carried out using the PBE⁷⁰ or PBE0⁷¹ density functionals. Relativistic effects were incorporated using the

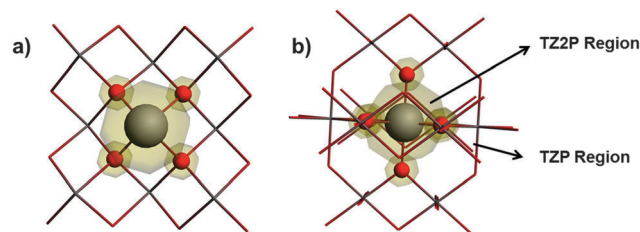


Fig. 1 Cluster models for (a) SnO and (b) SnO_2 . The central ball-and-stick region, representing the NMR-active ^{119}Sn center and the first co-ordination shell, are treated with the TZ2P basis set. The outer co-ordination shells are treated with the smaller TZP basis set.

ZORA Hamiltonian at the scalar (ZORA/SC) or the spin-orbit (ZORA/SO) level.^{72–75} Magnetic-shielding tensors were calculated using the GIAO^{76,77} formalism as implemented in ADF2014.^{19,78–80} A linear-dependence threshold of 10^{-4} was applied for the cluster calculations employing the PBE functional. For calculations employing the PBE0 functional, a more stringent threshold parameter was necessary for numerical problems associated with the linear dependence of the basis functions; therefore the threshold parameter was increased to 5×10^{-3} in these calculations. Effects of this procedure on the calculated magnetic-shielding tensor are given in the ESI† in Table S1 and Fig. S1.

For materials containing hydrogen atoms, a preliminary geometry optimization was run at the scalar (SC) level where the positions of the light atoms were allowed to relax while the heavy atoms remained fixed at their experimental coordinates. See the ESI† for details of the coordinates for each material. The terminal atoms of the clusters were treated with valence modification of terminal atoms using bond valence theory^{81–84} or VM-TA/BV.³¹ In this scheme, the valence of a terminal atom is modified by altering the nuclear charge, Z_{mod} , by the following relation:

$$Z_{\text{mod}} = Z_{\text{nuc}} + \Delta S \quad (1)$$

In eqn (1), Z_{nuc} is the actual formal charge on the terminal atom, whereas ΔS denotes the missing bond strength of the terminal atom in the cluster compared to the bond strength of the same atom in the periodic solid. ΔS is calculated by the following relation:

$$\Delta S = V - \sum_i \exp\left(\frac{R_{i0} - R_i}{b_i}\right) \quad (2)$$

In eqn (2), V is the unaltered valence of the terminal atom. The last term in eqn (2) is the bond-valence relation where R_i is the bond length between two atoms in a pair containing the terminal atom, and R_{i0} and b_i are the fitted bond-valence parameters. For a more detailed discussion of VM-TA/BV, readers are referred to ref. 31.

Calculations employing periodic-boundary conditions (PBCs) were performed using the CASTEP module of Materials Studio 7.0.^{84,85} These calculations were performed at the PBE level with core orbitals replaced by ultrasoft pseudopotentials generated on the fly and with a plane-wave cutoff energy of 600 eV.⁸⁵ Convergence of the computed magnetic-shielding parameters was tested by running several additional calculations



with higher cutoff energies and finer k -point grids for integration over the Brillouin zone. Calculations of the ^{119}Sn magnetic-shielding tensors employed the GIPAW method of Pickard and Mauri.⁸⁶ Relativistic effects were included at the ZORA/SC level through the pseudopotential approximation of Yates and co-workers.⁸⁷

3. Results and discussion

3.1. Spin-orbit effects on calculated ^{119}Sn magnetic-shielding tensors

All calculations discussed in this subsection were performed with the PBE density functional. Fig. 2 shows the correlations between the principal components of calculated magnetic-shielding tensors and the principal components of experimental chemical-shift tensors. Calculations were performed with the periodic PBE/GIPAW method (Fig. 2a), at the PBE/ZORA/SC level (Fig. 2b), and at the PBE/ZORA/SO level (Fig. 2c). Table 1 presents the parameters of the linear best-fit lines for tin(II) and tin(IV)-containing solids.

For tin(II)-containing solids analyzed as a separate subset (Table 1), the correlation between calculated magnetic shielding at the SC level and experimental chemical-shift values deviates significantly from ideal agreement. The deviation from the ideal case (slope = -1.00) is 29% and 23% for the linear best-fit lines obtained using periodic PBE/GIPAW and PBE/ZORA/SC methods, respectively. The extrapolated shielding of the reference compound, σ_{ref} , given by the intercept of the best-fit line, is 3019 ppm by the PBE/GIPAW method. The PBE/ZORA/SC value of σ_{ref} is 2745 ppm. For the subset of tin(IV)-containing solids, the PBE/GIPAW and PBE/ZORA/SC methods are much closer to the ideal value of -1.00 . However, the predicted reference shieldings predicted by the two methods differ by 531 ppm. These results indicate that the predicted magnetic shieldings determined either with the cluster model treated with the ZORA/SC Hamiltonian or with the GIPAW formalism with scalar-relativistic pseudopotentials are dependent on the oxidation state and the co-ordination geometry of the tin atom in the solid system.

In Table 2, we present the predicted chemical-shift parameters resulting from each method, along with reported experimental

Table 1 Linear-regression parameters for the linear relations between calculated magnetic shielding and experimental chemical shifts of ^{119}Sn -containing solids

Method	Slope	σ_{ref} (ppm)	R^2
Tin(II)-containing solids			
PBE/GIPAW	-0.71 ± 0.04	3019 ± 38	0.95
PBE/ZORA/SC	-0.77 ± 0.06	2745 ± 60	0.91
PBE/ZORA/SO	-0.99 ± 0.03	2849 ± 34	0.98
Tin(IV)-containing solids			
PBE/GIPAW	-1.08 ± 0.10	2869 ± 55	0.89
PBE/ZORA/SC	-1.00 ± 0.07	2338 ± 41	0.92
PBE/ZORA/SO	-0.99 ± 0.06	2875 ± 37	0.94
All Systems			
PBE/GIPAW	-0.77 ± 0.04	3001 ± 36	0.90
PBE/ZORA/SC	-0.92 ± 0.07	2499 ± 52	0.85
PBE/ZORA/SO	-0.98 ± 0.03	2867 ± 22	0.97

values for the twelve tin-containing solids. Calculated magnetic-shielding parameters have been converted to the chemical-shift scale using the predicted σ_{ref} from the linear best-fit correlations for all systems given in Table 1. The residuals between the experimental and calculated principal components of the chemical-shift tensors are given in Table 1, which is a measure of the overall quality of performance of each computational methodology (Table 2).

For tin(II)-containing solids, PBE/GIPAW calculations consistently give large deviations from experimental values, with residuals ranging between 229 and 314 ppm. For all tin(II)-containing solids, the δ_{33} component has the largest deviation between experiment and theory, when using the PBE/GIPAW approach. The performance of PBE/ZORA/SC calculations is somewhat better, as the residuals range from 124 to 250 ppm. With both computational protocols, the calculated spans ($\Omega = |\delta_{33} - \delta_{11}|$) are 200–500 ppm smaller than the experimental values. The agreement between experiment and theory is considerably stronger with the PBE/ZORA/SO method for tin(II)-containing solids, with residuals under 100 ppm for five of the six systems. Spans predicted by the PBE/ZORA/SO calculations are in better agreement with experiment than spans obtained by PBE/GIPAW and PBE/ZORA/SC calculations.

For tin(IV)-containing solids, the performance of PBE/GIPAW shows some improvement over its performance in calculations

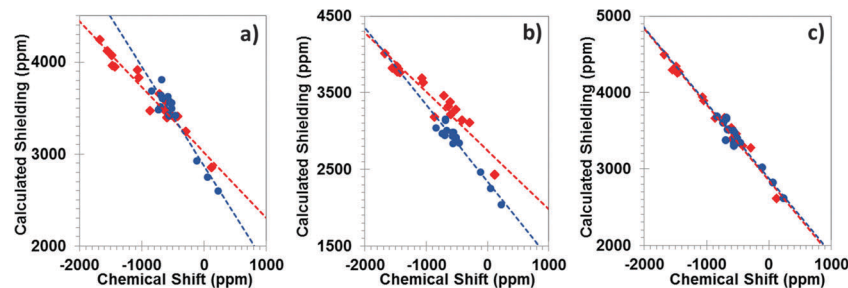


Fig. 2 Correlations between calculated principal components of ^{119}Sn magnetic-shielding tensors and experimental ^{119}Sn chemical-shift tensors for twelve tin-containing solids, as determined with different methodologies. Computed shielding constants were obtained using (a) the PBE/GIPAW method, (b) the PBE/ZORA/SC method, and (c) the PBE/ZORA/SO method. Tin(II) sites are shown in red; tin(IV) sites are shown in blue.



Table 2 Calculated and experimental NMR parameters of ^{119}Sn -containing solids determined with various DFT methods

Compounds	δ_{11} (ppm)	δ_{22} (ppm)	δ_{33} (ppm)	δ_{iso} (ppm)	Ω (ppm)	Residual ^a (ppm)
Tin(II)-containing solids						
SnO	121	121	−867	−208	988	—
PBE/GIPAW	145	145	−472	−61	617	229
PBE/ZORA/SC	45	45	−677	−196	722	126
PBE/ZORA/SO	256	253	−793	−94	1049	117
SnHPO ₄	−606	−712	−1553	−957	947	—
PBE/GIPAW	−564	−655	−1119	−779	555	254
PBE/ZORA/SC	−874	−954	−1323	−1050	449	247
PBE/ZORA/SO	−669	−808	−1429	−969	760	97
SnHPO ₃	−290	−420	−1435	−715	1145	—
PBE/GIPAW	−247	−409	−949	−535	702	282
PBE/ZORA/SC	−602	−638	−1262	−834	660	241
PBE/ZORA/SO	−405	−430	−1402	−745	996	69
SnC ₂ O ₄	−523	−639	−1474	−879	951	—
PBE/GIPAW	−421	−479	−965	−622	544	314
PBE/ZORA/SC	−778	−816	−1266	−953	488	216
PBE/ZORA/SO	−587	−651	−1392	−877	805	61
SnSO ₄	−1047	−1070	−1679	−1265	632	—
PBE/GIPAW	−834	−912	−1239	−995	405	297
PBE/ZORA/SC	−1130	−1183	−1510	−1274	380	127
PBE/ZORA/SO	−1028	−1075	−1630	−1245	602	30
BaSnF ₄	−596	−596	−1486	−893	890	—
PBE/GIPAW	−394	−394	−1073	−620	679	290
PBE/ZORA/SC	−708	−708	−1340	−919	632	124
PBE/ZORA/SO	−520	−520	−1478	−839	958	62
Tin(IV)-containing solids						
SnO ₂	−550	−573	−686	−603	136	—
PBE/GIPAW	−564	−617	−633	−605	69	41
PBE/ZORA/SC	−471	−475	−631	−526	160	79
PBE/ZORA/SO	−605	−630	−785	−673	180	73
Ca ₂ SnO ₄	−459	−512	−664	−545	205	—
PBE/GIPAW	−415	−491	−597	−501	182	48
PBE/ZORA/SC	−334	−389	−495	−406	161	141
PBE/ZORA/SO	−474	−529	−647	−550	173	16
SnS ₂	−730	−730	−835	−765	105	—
PBE/GIPAW	−474	−475	−684	−544	211	226
PBE/ZORA/SC	−456	−456	−527	−479	71	286
PBE/ZORA/SO	−741	−742	−819	−767	77	13
Pb ₂ SnO ₄	−558	−566	−692	−605	134	—
PBE/GIPAW	−410	−419	−512	−447	101	159
PBE/ZORA/SC	−328	−421	−436	−395	108	216
PBE/ZORA/SO	−436	−468	−509	−471	73	139
Na ₆ Sn ₂ S ₇	232	60	−107	62	339	—
PBE/GIPAW	409	255	78	247	331	186
PBE/ZORA/SC	468	257	46	257	423	199
PBE/ZORA/SO	249	41	−150	47	399	29
Sr ₂ SnO ₄	−510	−548	−681	−580	171	—
PBE/GIPAW	−551	−551	−805	−636	253	75
PBE/ZORA/SC	−407	−412	−645	−488	238	100
PBE/ZORA/SO	−536	−539	−801	−625	265	71

$$^a \text{Residual} = \sqrt{\frac{1}{3} \sum_{i=1}^3 (\Delta\delta_{ii}^{\text{calc}} - \Delta\delta_{ii}^{\text{exp}})^2}.$$

of the chemical shifts in tin(II) systems. For example, the accuracies of calculated principal components for SnO₂, Ca₂SnO₄, Pb₂SnO₄ and Sr₂SnO₄ by the PBE/GIPAW method are comparable to results obtained with PBE/ZORA/SO methods. However, the deviation of PBE/GIPAW results from experimental values is not as good for materials like SnS₂ and Na₆Sn₂S₇. In these latter systems, the first co-ordination shell around tin consists of sulfur atoms rather than oxygen atoms. The magnitude of SO effects on ^{119}Sn magnetic shielding is probably increased by the presence of the heavier sulfur atom in the co-ordination environment. In these two cases,

the residuals determined with the PBE/ZORA/SO calculations are 13 and 29 ppm, respectively, whereas residuals by the PBE/ZORA/SC are 286 and 199 ppm.

To understand the effect of SO coupling on the ^{119}Sn magnetic-shielding tensor, we present the differences ($\Delta\sigma_{ii}$) between principal components of magnetic-shielding tensors calculated at the PBE/ZORA/SO level and those calculated at the PBE/ZORA/SC level (Fig. 3). It is evident that SO effects on magnetic-shielding tensors exhibit a strong dependence on the oxidation state of tin. For tin(II)-containing



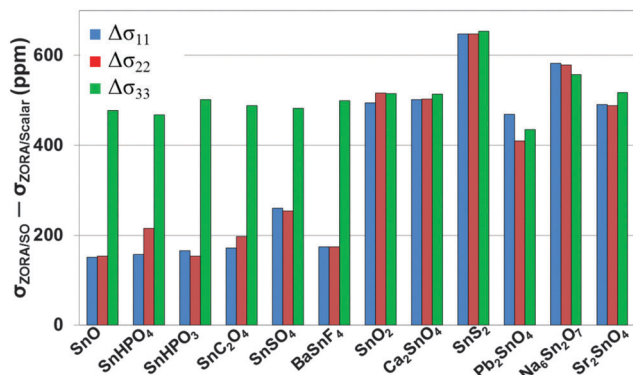


Fig. 3 The differences ($\Delta\sigma_{ii}$) in principal components of ^{119}Sn magnetic-shielding tensors calculated with the ZORA/SO method and the ZORA/SC method. All calculations model the solid-state environment with the cluster-based VMTA/BV approach. Magnetic-shielding calculations use the PBE functional.

systems, the contribution of SO effects on magnetic shielding is largest for the σ_{33} component where $\Delta\sigma_{33}$ values are around 500 ppm. The SO effects are less for σ_{11} and σ_{22} , with $\Delta\sigma_{ii}$ ranging between 154 and 260 ppm. In comparison, the contribution of SO effects on each principal component of the magnetic-shielding tensor is more uniform for tin(IV)-containing systems, with $\Delta\sigma_{ii}$ varying between 435 and 654 ppm. The largest change in any magnetic-shielding tensor between PBE/ZORA/SO and PBE/ZORA/SC results is observed for SnS_2 where $\Delta\sigma_{ii} \sim 650$ ppm. Indeed, among the tin(IV)-containing materials, the residuals of PBE/GIPAW and cluster-based PBE/ZORA/SC results in Table 2 are highest for this compound, due to the presence of significant spin-orbit effects relative to oxygen-co-ordinated tin sites.

The results in Fig. 3 for SO effects on the ^{119}Sn magnetic-shielding tensor show a striking resemblance to the recently-investigated SO effects on co-ordination compounds of lead.¹⁶ In the case of ^{207}Pb -containing solids, SO effects show a similar dependence on the oxidation state (+2 or +4) and co-ordination geometry (hemidirected or holodirected) around the ^{207}Pb nuclei. The magnitudes of the SO effects for ^{119}Sn and ^{207}Pb are quite different, as expected. Overall, the magnitude of SO effects ($\Delta\sigma_{ii}$) for the ^{119}Sn -containing systems varies between 154 and 654 ppm. In comparison, the SO effects on ^{207}Pb magnetic-shielding tensor are generally 2000–3000 ppm. This difference is likely due to the larger nuclear charge on ^{207}Pb , resulting in stronger SO coupling effects on magnetic shielding.

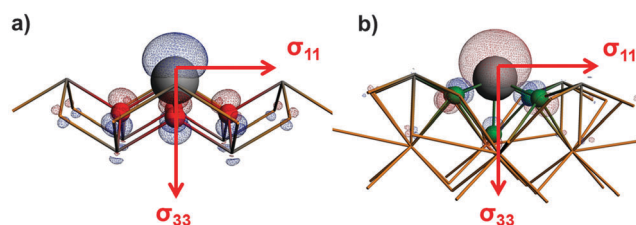


Fig. 4 The orientations of magnetic-shielding tensor axes along with MOs associated with the 'lone pair' on tin(II) for (a) SnO and (b) BaSnF_4 .

Fig. 4 shows the orientations of the axes of the ^{119}Sn magnetic-shielding tensors obtained at the ZORA/SO level of theory for two tin(II) systems in their local frames of reference. For both SnO and BaSnF_4 , the principal axis of the most-shielded component is aligned with the symmetry axis of the molecular orbital (MO) which results mostly from the mixing of the 5s and 5p atomic orbitals of the tin nuclei. Such MOs are often associated with the 'lone-pair' on an atom. In comparison, the σ_{11} and σ_{22} axes are in the plane formed by the tin atoms for both systems. The relationship between the principal axes and the lone-pair of the tin(II) nuclei are analogous to recent findings for hemidirected lead(II) systems.¹⁶

The accuracy of calculated ^{119}Sn NMR parameters has been systematically investigated for a series of isolated tin(IV) molecules by Bagno *et al.*,³⁷ using the ZORA/SC and ZORA/SO methods. The results indicate that the both ZORA/SC and ZORA/SO methods work quite well for predicting chemical shifts when no other heavy atom is bound to tin. In such systems, the predicted SO effects on the isotropic magnetic shielding vary by around 500 ppm and mostly cancel out when magnetic shieldings are converted to the chemical-shift scale. These findings partially agree with the PBE/GIPAW and PBEZORA/SC results for tin(IV)-containing solids. However, for tin(II)-containing solids, the assumption that there are negligible SO effects on the chemical shift is incorrect.

The magnetic shielding (or absolute shielding) of tetramethyltin, σ_{ref} , can be estimated from the intersection of the best-fit correlation lines in Table 1. From the PBE/ZORA/SO method, σ_{ref} is predicted to be 2867 ppm from the correlation obtained for all tin-containing systems. By comparison, a single calculation on tetramethyltin at the same level of theory gives 2852 ppm for σ_{ref} , a discrepancy of only 15 ppm. With the 4c relativistic DFT (with the BP86 functional), σ_{ref} is computed as 3199 ppm.⁴¹ It is clear that PBE/ZORA/SO underestimates σ_{ref} by $\sim 12\%$ compared to the 4c method. The underestimation of absolute shieldings predicted by ZORA calculations has been discussed previously.^{27,88,89} Nevertheless, the current results, as well as previous studies on other heavy nuclei such as ^{207}Pb ^{16,31} and ^{199}Hg ,²⁹ demonstrate that ZORA/SO predictions for the chemical-shift tensor agree with the experimental values within $\sim 2\%$, possibly due to the cancellation of higher-order relativistic effects beyond SO coupling.

Table 3 Comparison of calculated magnetic shieldings (σ_{iso}) for tin-containing molecules using GIPAW^a and ZORA/SC methods

Molecule ^b	σ_{iso} (ppm)	
	PBE/ZORA/SC	PBE/GIPAW
SnF_2	2854	2880
$\text{Sn}(\text{CN})_2$	2053	2111
$\text{Sn}(\text{OH})_2$	2465	2516
SnF_4	3002	3364
$\text{Sn}(\text{CH}_3)_4$	2370	2829
SnH_4	2982	3354

^a For GIPAW calculations the isolated molecular state is approximated by employing large cubic unit cells. ($a = 20$ Å). ^b Geometries are optimized at PBE/ZORA/SC level of theory.

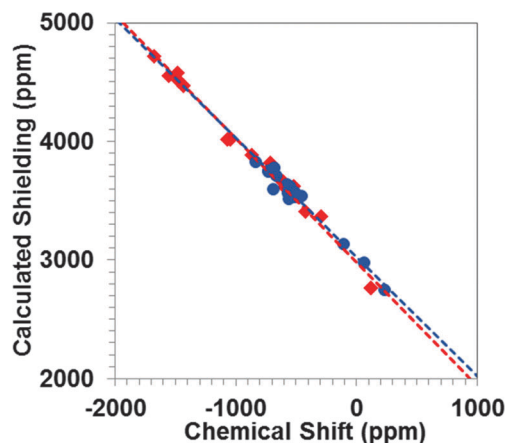


Fig. 5 Correlation between calculated principal components of ^{119}Sn magnetic-shielding tensors and experimental ^{119}Sn chemical-shift tensors for twelve tin-containing solids. Calculations were performed at the PBE0/ZORA/SO level of theory. Tin(II) sites are shown in red; tin(IV) sites are shown in blue.

In the previous investigations of lighter nuclei such as ^{13}C , ^{19}F or ^{29}Si , the performances of the GIPAW method and cluster models for the predictions of magnetic-shielding tensors in solids are similar, provided that sufficiently large clusters are used for the comparison.^{53–55,90} In contrast, the current results show that although PBE/GIPAW and cluster-based PBE/ZORA/SC methods yield similar trends for tin(II)-containing and tin(IV)-containing solids, the two methods yield quite different results for σ_{ref} (Table 1).

To compare the two methods in the absence of solid-state effects, we performed NMR calculations on isolated molecules

(SnF_2 , $\text{Sn}(\text{CN})_2$, $\text{Sn}(\text{OH})_2$, SnF_4 , $\text{Sn}(\text{CH}_3)_4$, and SnH_4) containing ^{119}Sn in oxidation states of +2 or +4. The results in Table 3 show that for the tin(II) species (SnF_2 , $\text{Sn}(\text{CN})_2$, and $\text{Sn}(\text{OH})_2$), the calculated magnetic shieldings determined by the PBE/ZORA/SC and PBE/GIPAW methods are quite similar, with the PBE/GIPAW approach yielding results that are more shielded by 26–58 ppm. On the other hand, the calculated magnetic shieldings of the tin(IV) species (SnF_4 , $\text{Sn}(\text{CH}_3)_4$, and SnH_4) are predicted by the PBE/GIPAW approach to be 362–460 ppm more shielded. Therefore, one should expect that the calculated magnetic shieldings with the PBE/ZORA/SC Hamiltonian and the PBE/GIPAW method deviate from one another for ^{119}Sn nuclei. Moreover, the difference in the calculated magnetic shieldings depends on the electronic structure of the system investigated.

3.2. The performance of hybrid DFT methods when combined with ZORA/SO

In general, hybrid density functionals improve the accuracy of prediction of NMR parameters. A recent study of ^{207}Pb -containing solids indicates that the inclusion of Hartree-Fock (HF) exchange *via* hybrid functionals can have significant effects on calculated principal components of the magnetic-shielding tensor.¹⁶ Therefore, we have investigated the performance of the PBE0 functional (with 25% HF exchange), using cluster models and the ZORA/SO Hamiltonian.

In Fig. 5, we show the correlation between the principal components of the calculated magnetic-shielding tensor at the PBE0/ZORA/SO level of theory and the principal components of the experimental chemical-shift tensor. The slope of the correlation line when all systems are considered is -1.03 ± 0.02 . There is less scatter about the best-fit line ($R^2 = 0.99$) than was

Table 4 Experimental and calculated NMR parameters of ^{119}Sn -containing solids using model clusters and PBE0/ZORA/SO level of theory

Compounds	δ_{11} (ppm)	δ_{22} (ppm)	δ_{33} (ppm)	δ_{iso} (ppm)	Ω (ppm)	Residual (ppm)
Tin(II)-containing solids						
SnO	121	121	−867	−208	988	—
PBE0/ZORA/SO	236	236	−885	−138	1121	94
SnHPO ₄	−606	−712	−1553	−957	947	—
PBE0/ZORA/SO	−657	−813	−1547	−1006	890	66
SnHPO ₃	−290	−420	−1435	−715	1145	—
PBE0/ZORA/SO	−363	−404	−1465	−744	1102	46
SnC ₂ O ₄	−523	−639	−1474	−879	951	—
PBE0/ZORA/SO	−618	−683	−1512	−938	894	65
SnSO ₄	−1047	−1070	−1679	−1265	632	—
PBE0/ZORA/SO	−1011	−1015	−1715	−1247	704	43
BaSnF ₄	−596	−596	−1486	−893	890	—
PBE0/ZORA/SO	−613	−613	−1571	−932	958	51
Tin(IV)-containing solids						
SnO ₂	−550	−573	−686	−603	136	—
PBE0/ZORA/SO	−617	−639	−770	−675	153	73
Ca ₂ SnO ₄	−459	−512	−664	−545	205	—
PBE0/ZORA/SO	−536	−569	−713	−606	177	62
SnS ₂	−730	−730	−835	−765	105	—
PBE0/ZORA/SO	−747	−747	−821	−772	75	16
Pb ₂ SnO ₄	−558	−566	−692	−605	134	—
PBE0/ZORA/SO	−513	−558	−597	−556	85	61
Na ₆ Sn ₂ S ₇	232	60	−107	62	339	—
PBE0/ZORA/SO	252	21	−134	46	387	30
Sr ₂ SnO ₄	−510	−548	−681	−580	171	—
PBE0/ZORA/SO	−526	−531	−772	−610	246	54



obtained at the PBE/ZORA/SO level. In general, the calculated principal components obtained with the PBE0/ZORA/SO method are 100–200 ppm more shielded than the calculated principal components obtained with the PBE/ZORA/SO method. The predicted shielding of the reference compound is found to be 3003 ± 16 ppm, indicating a slightly more shielded value obtained at PBE0/ZORA/SO level of theory than at the PBE/ZORA/SO level.

In Table 4, the predicted principal components of the chemical-shift tensors at the PBE0/ZORA/SO level of theory are tabulated, along with the experimental values. For all tin-containing solids, the calculated residuals between theory and experiment are below 100 ppm and the largest residual (94 ppm) is seen for SnO. In general, the agreement between experiment and theory improves when PBE0 is employed instead of PBE. However, this improvement, for most cases, is quite small. As the NMR calculations employing hybrid functionals are considerably larger than for GGA functionals, the latter may be more cost-effective for calculations of ^{119}Sn magnetic-shielding tensor in similar systems.

4. Conclusion

The effects of SO coupling on the magnetic-shielding or chemical-shift tensors of ^{119}Sn nuclei in various tin compounds are significant. Neglecting the SO coupling in calculations (GIPAW or ZORA/SC) usually results in calculated values that deviate by 20–30% from the experimental values for tin(II) compounds. The deviation is clearly seen for the calculated chemical shifts and spans.

When the ZORA/SO Hamiltonian and cluster models are employed, one obtains correlations between calculated and experimental principal components of chemical-shift tensors that are very close to the ideal relationship. In fact, using this level of theory provides agreement between results for tin(II)- and tin(IV)-containing solids. The residuals between calculated and experimental principal components are below 100 ppm for the majority of tin-containing solids. These results are in contrast to the GIPAW and ZORA/SC results, where large deviations from experiment are present. At the moment, calculations with the SO Hamiltonian are not available in the GIPAW formalism. Inclusion of the SO Hamiltonian in the GIPAW formalism may improve the systematic deviations of ^{119}Sn magnetic-shielding tensors from experiment.

The agreement between calculated and experimental principal components of chemical-shift tensors is improved further by use of the hybrid PBE0 functional. However, this improvement is rather small for most of the cases we have examined.

Direct comparison of the principal components of calculated magnetic-shielding tensors to the principal components of chemical-shift tensors provides a more stringent test of relativistic effects than does the comparison of isotropic values. For example, the effects of SO coupling on the σ_{33} component of tin(II)-containing materials are significantly larger than observed for the other two principal components. This observation suggests

that SO effects may be present for the principal components of the magnetic-shielding tensors of other period 5 nuclei such as ^{113}Cd and ^{125}Te .

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References

- 1 E. D. Becker, *High Resolution NMR: Theory and Chemical Applications*, Academic Press, San Diego, 3rd edn, 2000.
- 2 M. J. Duer, *Introduction to Solid-State NMR Spectroscopy*, Wiley-Blackwell, Oxford, 2nd edn, 2005.
- 3 B. Wrackmeyer, *Annu. Rep. NMR Spectrosc.*, 1999, **38**, 203–264.
- 4 N. J. Clayden, C. M. Dobson and A. Fern, *J. Chem. Soc., Dalton Trans.*, 1989, 843–847.
- 5 C. Cossement, J. Darville, J. M. Gilles, J. B. Nagy, C. Fernandez and J. P. Amoureux, *Magn. Reson. Chem.*, 1992, **30**, 263–270.
- 6 R. K. Harris, S. E. Lawrence, S. W. Oh and V. G. K. Das, *J. Mol. Struct.*, 1995, **347**, 309–319.
- 7 C. Mundus, G. Taillades, A. Pradel and M. Ribes, *Solid State Nucl. Magn. Reson.*, 1996, **7**, 141–146.
- 8 C. Dybowski and G. Neue, *Prog. Nucl. Magn. Reson. Spectrosc.*, 2002, **41**, 153–170.
- 9 F. Fayon, I. Farnan, C. Bessada, J. Coutures, D. Massiot and J. P. Coutures, *J. Am. Chem. Soc.*, 1997, **119**, 6837–6843.
- 10 G. Neue, C. Dybowski, M. L. Smith, M. A. Hepp and D. L. Perry, *Solid State Nucl. Magn. Reson.*, 1996, **6**, 241–250.
- 11 P. G. Harrison, *Organometallic Compounds of Germanium, Tin, and Lead*, Chapman and Hall, London, 1985.
- 12 W. H. Baur, *Acta Crystallogr.*, 1956, **9**, 515–520.
- 13 P. Amornsakchai, D. C. Apperley, R. K. Harris, P. Hodgkinson and P. C. Waterfield, *Solid State Nucl. Magn. Reson.*, 2004, **26**, 160–171.
- 14 J. Burt, W. Levason and G. Reid, *Coord. Chem. Rev.*, 2014, **260**, 65–115.
- 15 J. Pannetier and G. Denes, *Acta Crystallogr., Sect. B: Struct. Crystallogr. Cryst. Chem.*, 1980, **36**, 2763–2765.
- 16 F. Alkan and C. Dybowski, *J. Phys. Chem. A*, 2016, **120**, 161–168.
- 17 J. C. Facelli, *Prog. Nucl. Magn. Reson. Spectrosc.*, 2010, **58**, 176–201.
- 18 C. Bonhomme, C. Gervais, F. Babonneau, C. Coelho, F. Pourpoint, T. Azais, S. E. Ashbrook, J. M. Griffin, J. R. Yates, F. Mauri and C. J. Pickard, *Chem. Rev.*, 2012, **112**, 5733–5779.
- 19 A. Rodriguez-Forte, P. Alemany and T. Ziegler, *J. Phys. Chem. A*, 1999, **103**, 8288–8294.



- 20 S. K. Wolff, T. Ziegler, E. van Lenthe and E. J. Baerends, *J. Chem. Phys.*, 1999, **110**, 7689–7698.
- 21 J. Jokisaari, S. Jarvinen, J. Autschbach and T. Ziegler, *J. Phys. Chem. A*, 2002, **106**, 9313–9318.
- 22 J. Autschbach and S. Zheng, *Annu. Rep. NMR Spectrosc.*, 2009, **67**, 1–95.
- 23 L. A. Truflandier and J. Autschbach, *J. Am. Chem. Soc.*, 2010, **132**, 3472–3483.
- 24 B. J. Greer, V. K. Michaelis, M. J. Katz, D. B. Leznoff, G. Schreckenbach and S. Kroeker, *Chem. – Eur. J.*, 2011, **17**, 3609–3618.
- 25 J. Roukala, A. F. Maldonado, J. Vaara, G. A. Aucar and P. Lantto, *Phys. Chem. Chem. Phys.*, 2011, **13**, 21016–21025.
- 26 J. Autschbach, *J. Chem. Phys.*, 2012, **136**, 15.
- 27 A. Wodynski, M. Repisky and M. Pecul, *J. Chem. Phys.*, 2012, **137**, 014311.
- 28 J. Vicha, M. Patzschke and R. Marek, *Phys. Chem. Chem. Phys.*, 2013, **15**, 7740–7754.
- 29 F. Alkan and C. Dybowski, *Phys. Chem. Chem. Phys.*, 2014, **16**, 14298–14308.
- 30 J. Autschbach, *Philos. Trans. R. Soc., A*, 2014, **372**, 20120489.
- 31 F. Alkan and C. Dybowski, *Phys. Chem. Chem. Phys.*, 2015, **17**, 25014–25026.
- 32 Y. Ruiz-Morales, G. Schreckenbach and T. Ziegler, *J. Phys. Chem. A*, 1997, **101**, 4121–4127.
- 33 P. Avale, R. K. Harris, P. B. Karadakov and P. J. Wilson, *Phys. Chem. Chem. Phys.*, 2002, **4**, 5925–5932.
- 34 M. Kaupp, M. Buhl and V. G. Malkin, *Calculation of NMR and EPR Parameters: Theory and Applications*, WILEY-VCH Verlag GmbH & Co. KGaA, Weinheim, 2004.
- 35 R. Fukuda and H. Nakatsuji, *J. Chem. Phys.*, 2005, **123**, 044101.
- 36 M. Hada, J. Wan, R. Fukuda and H. Nakatsuji, *J. Comput. Chem.*, 2001, **22**, 1502–1508.
- 37 A. Bagno, G. Casella and G. Saielli, *J. Chem. Theory Comput.*, 2006, **2**, 37–46.
- 38 A. F. Maldonado and G. A. Aucar, *J. Phys. Chem. A*, 2014, **118**, 7863–7875.
- 39 J. I. Melo, A. Maldonado and G. A. Aucar, *Theor. Chem. Acc.*, 2011, **129**, 483–494.
- 40 H. Kaneko, M. Hada, T. Nakajima and H. Nakatsuji, *Chem. Phys. Lett.*, 1996, **261**, 1–6.
- 41 E. Malkin, S. Komorovsky, M. Repisky, T. B. Demissie and K. Ruud, *J. Phys. Chem. Lett.*, 2013, **4**, 459–463.
- 42 L. F. Wang, C. E. Kefalidis, T. Roisnel, S. Sinbandhit, L. Maron, J. F. Carpentier and Y. Sarazin, *Organometallics*, 2014, **34**, 2139–2150.
- 43 R. Vivas-Reyes, F. De Proft, M. Biesemans, R. Willem and P. Geerlings, *J. Phys. Chem. A*, 2002, **106**, 2753–2759.
- 44 L. Broeckert, J. Turek, R. Olejnik, A. Ruzicka, M. Biesemans, P. Geerlings, R. Willem and F. De Proft, *Organometallics*, 2013, **32**, 2121–2134.
- 45 M. R. Mitchell, S. W. Reader, K. E. Johnston, C. J. Pickard, K. R. Whittle and S. E. Ashbrook, *Phys. Chem. Chem. Phys.*, 2011, **13**, 488–497.
- 46 J. A. Tossell, *J. Magn. Reson.*, 1997, **127**, 49–53.
- 47 G. Valerio, A. Goursot, R. Vetrivel, O. Malkina, V. Malkin and D. R. Salahub, *J. Am. Chem. Soc.*, 1998, **120**, 11426–11431.
- 48 G. Valerio and A. Goursot, *J. Phys. Chem. B*, 1999, **103**, 51–58.
- 49 Y. Zhang and E. Oldfield, *J. Phys. Chem. B*, 2004, **108**, 19533–19540.
- 50 D. Stueber, *Concepts Magn. Reson., Part A*, 2006, **28**, 347–368.
- 51 D. H. Brouwer and G. D. Enright, *J. Am. Chem. Soc.*, 2008, **130**, 3095–3105.
- 52 J. Weber and J. Gunne, *Phys. Chem. Chem. Phys.*, 2010, **12**, 583–603.
- 53 S. T. Holmes, R. J. Iuliucci, K. T. Mueller and C. Dybowski, *J. Chem. Theory Comput.*, 2015, **11**, 5229–5241.
- 54 S. T. Holmes, R. J. Iuliucci, K. T. Mueller and C. Dybowski, *J. Chem. Phys.*, 2014, **141**, 164121.
- 55 S. T. Holmes, F. Alkan, R. J. Iuliucci, K. T. Mueller and C. Dybowski, *J. Comput. Chem.*, 2016, **37**, 1704–1710.
- 56 A. W. MacGregor, L. A. O'Dell and R. W. Schurko, *J. Magn. Reson.*, 2011, **208**, 103–113.
- 57 R. C. McDonald and K. Eriks, *Inorg. Chem.*, 1980, **19**, 1237–1241.
- 58 A. Gleizes and J. Galy, *J. Solid State Chem.*, 1979, **30**, 23–33.
- 59 J. D. Donaldson and D. C. Puxley, *Acta Crystallogr., Sect. B: Struct. Crystallogr. Cryst. Chem.*, 1972, **28**, 864–867.
- 60 M. M. Ahmad, Y. Yamane and K. Yamada, *J. Appl. Phys.*, 2009, **106**, 074106.
- 61 H. Yamane, Y. Kaminaga, S. Abe and T. Yamada, *J. Solid State Chem.*, 2008, **181**, 2559–2564.
- 62 R. M. Hazen and L. W. Finger, *Am. Mineral.*, 1978, **63**, 289–292.
- 63 J. Catalano, A. Murphy, Y. Yao, F. Alkan, N. Zurnbulyadis, S. A. Centeno and C. Dybowski, *J. Phys. Chem. A*, 2014, **118**, 7952–7958.
- 64 J. R. Gavarri, J. P. Vigouroux, G. Calvarin and A. W. Hewat, *J. Solid State Chem.*, 1981, **36**, 81–90.
- 65 J. C. Jumas, J. Olivierf, F. Vermotga, M. Ribes, E. Philippo and M. Maurin, *Rev. Chim. Miner.*, 1974, **11**, 13–26.
- 66 W. T. Fu, D. Visser and D. J. W. Ijdo, *J. Solid State Chem.*, 2002, **169**, 208–213.
- 67 G. te Velde, F. M. Bickelhaupt, E. J. Baerends, C. F. Guerra, S. J. A. Van Gisbergen, J. G. Snijders and T. Ziegler, *J. Comput. Chem.*, 2001, **22**, 931–967.
- 68 C. F. Guerra, J. G. Snijders, G. te Velde and E. J. Baerends, *Theor. Chem. Acc.*, 1998, **99**, 391–403.
- 69 ADF2014, SCM, Theoretical Chemistry, Vrije Universiteit, Amsterdam, The Netherlands, <http://www.scm.com>. (accessed September 2, 2014).
- 70 J. P. Perdew, K. Burke and M. Ernzerhof, *Phys. Rev. Lett.*, 1996, **77**, 3865–3868.
- 71 C. Adamo and V. Barone, *J. Chem. Phys.*, 1999, **110**, 6158–6170.
- 72 E. Van lenthe, E. J. Baerends and J. G. Snijders, *J. Chem. Phys.*, 1993, **99**, 4597–4610.
- 73 E. van Lenthe, E. J. Baerends and J. G. Snijders, *J. Chem. Phys.*, 1994, **101**, 9783–9792.
- 74 E. van Lenthe, R. van Leeuwen, E. J. Baerends and J. G. Snijders, *Int. J. Quantum Chem.*, 1996, **57**, 281–293.



- 75 E. van Lenthe, J. G. Snijders and E. J. Baerends, *J. Chem. Phys.*, 1996, **105**, 6505–6516.
- 76 R. Ditchfield, *Mol. Phys.*, 1974, **27**, 789–807.
- 77 K. Wolinski, J. F. Hinton and P. Pulay, *J. Am. Chem. Soc.*, 1990, **112**, 8251–8260.
- 78 M. Krykunov, T. Ziegler and E. van Lenthe, *J. Phys. Chem. A*, 2009, **113**, 11495–11500.
- 79 M. Krykunov, T. Ziegler and E. van Lenthe, *Int. J. Quantum Chem.*, 2009, **109**, 1676–1683.
- 80 J. Autschbach, *Mol. Phys.*, 2013, **111**, 2544–2554.
- 81 I. D. Brown and R. D. Shannon, *Acta Crystallogr., Sect. A: Cryst. Phys., Diffraction, Theor. Gen. Crystallogr.*, 1973, **29**, 266–282.
- 82 I. D. Brown and K. K. Wu, *Acta Crystallogr., Sect. B: Struct. Crystallogr. Cryst. Chem.*, 1976, **32**, 1957–1959.
- 83 I. D. Brown and D. Altermatt, *Acta Crystallogr., Sect. B: Struct. Sci.*, 1985, **41**, 244–247.
- 84 I. D. Brown, *Chem. Rev.*, 2009, **109**, 6858–6919.
- 85 J. R. Yates, C. J. Pickard and F. Mauri, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2007, **76**, 024401.
- 86 C. J. Pickard and F. Mauri, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2001, **63**, 245101.
- 87 J. R. Yates, C. J. Pickard, M. C. Payne and F. Mauri, *J. Chem. Phys.*, 2003, **118**, 5746–5753.
- 88 J. Autschbach, *Theor. Chem. Acc.*, 2004, **112**, 52–57.
- 89 V. Arcisauskaitė, J. I. Melo, L. Hemmingsen and S. P. A. Sauer, *J. Chem. Phys.*, 2011, **135**, 044306.
- 90 A. Sadoc, M. Body, C. Legein, M. Biswal, F. Fayon, X. Rocquefelte and F. Boucher, *Phys. Chem. Chem. Phys.*, 2011, **13**, 18539–18550.

