

CrossMark
click for updatesCite this: *J. Mater. Chem. A*, 2015, 3, 23743Soft X-ray spectroscopic studies of the electronic structure of M:BiVO₄ (M = Mo, W) single crystals†V. Jovic,^{*ab} J. Laverock,^{cd} A. J. E. Rettie,^e J.-S. Zhou,^f C. B. Mullins,^{ef} V. R. Singh,^d B. Lamoureux,^d D. Wilson,^{ab} T.-Y. Su,^d B. Jovic,^g H. Bluhm,^h T. Sönnel^{ab} and K. E. Smith^{abd}

Bismuth vanadate, BiVO₄, is a promising material for use as an anode in photoelectrochemical water splitting. However, its conversion efficiency is limited by poor bulk charge transport, which is via small-polarons. We report here the use of a suite of X-ray spectroscopic probes to determine the electronic structure of 0.3–0.6 at% M:BiVO₄ (M = Mo or W). The results are interpreted in the context of current theories regarding the influence of doping on the existence of inter-band gap small-polaron states and their effect on the conversion efficiency of BiVO₄. Preliminary X-ray absorption and emission measurements reveal that doping widens the band gap from 2.50 to 2.75 eV, whereas the indirect nature of the band gap remains unaffected. X-ray absorption spectroscopy verified that the doping levels did not affect the distorted tetrahedral environment of V⁵⁺ in BiVO₄. For BiVO₄ and W:BiVO₄, V L₃ resonant inelastic X-ray scattering showed energy loss features related to charge transfer from low lying valence metal/oxygen states to unoccupied V e_g conduction band states. A 3.8 eV energy loss feature, coupled with small polaron-like peaks measured in valence band resonant photoelectron spectroscopy of M:BiVO₄, point to the population of inter-band gap V 3d states of e_g symmetry. The data reveals the existence of a band gap state in the absence of an applied bias in M:BiVO₄, linked to small-polaron formation. We tentatively assign it as a deep trap state, which suggests that the improved conversion efficiency of M:BiVO₄ relative to the undoped material is largely due to the increased carrier concentration in spite of increased carrier recombination rates.

Received 2nd October 2015
Accepted 20th October 2015

DOI: 10.1039/c5ta07898a

www.rsc.org/MaterialsA

1. Introduction

Harnessing solar power to generate hydrogen (H₂) through photoelectrochemical (PEC) water splitting is a promising approach towards achieving future energy supply security.^{1,2} There is now a strong incentive to develop metal oxides as alternatives to non-oxide semiconductors (CdTe, CdSe, CdS, Cu(In,Ga)(Se,S)₂, GaInP₂) for use in PEC water splitting due to their commonly non-toxic nature, stability against

photocorrosion and reasonable cost.^{1,3,4} Bismuth vanadate (BiVO₄) is a near ideal candidate in this respect owing to its appropriate band gap (~2.4 eV), well positioned valence band (VB) potential, stability in the presence of a co-catalyst and promising conversion efficiencies.^{2,5–11} A bottleneck for the further use of BiVO₄ in PEC water splitting is its poor bulk conductivity, the result of electron localization ('small-polaron' formation).^{8,12,13}

In BiVO₄ (a d⁰ transition metal (TM) oxide), small-polarons form when charge carriers are introduced by doping; small polarons are vulnerable to be trapped.^{8,14} Instead of flowing freely under an applied bias, the electron is forced to 'hop' slowly through the lattice (via V-sites) by thermal assistance, drastically reducing mobility.^{8,12} BiVO₄ possesses the catalytically favourable monoclinic scheelite (ms) structure, composed of edge-sharing BiO₈ (Bi³⁺) dodecahedra in contact with VO₄ (V⁵⁺) tetrahedra by an apex oxygen atom.^{6,9,10} Distortions of VO₄ and BiO₈ units by an electron lone pair (in Bi 6s/O 2p states at the VB maximum), along with poor wavefunction overlap of V–V neighbours in the *c*-direction, and a large dielectric constant, assist in electron localization.^{15–17} Based on recent theory, this manifests in electrons occupying empty V 3d_{z²} states at the CB minimum, with weak delocalization to the adjacent oxygen

^aSchool of Chemical Sciences, The University of Auckland, Auckland, New Zealand.
E-mail: vjov001@aucklanduni.ac.nz

^bThe MacDiarmid Institute for Advanced Materials and Nanotechnology, New Zealand

^cSchool of Physics, H. H. Wills Physics Laboratory, University of Bristol, Tyndall Avenue, Bristol, BS8 1TL, UK

^dDepartment of Physics, Boston University, Boston, MA 02215, USA

^eMcKetta Department of Chemical Engineering, The University of Texas at Austin, Austin, Texas 78712, USA

^fTexas Materials Institute, The University of Texas at Austin, Austin, Texas 78712, USA

^gDefence Technology Agency, New Zealand Defence Force, Devonport Naval Base, Auckland 0624, New Zealand

^hChemical Science Division, Lawrence Berkeley National Laboratory, Berkeley, CA 94720, USA

† Electronic supplementary information (ESI) available. See DOI: 10.1039/c5ta07898a

atoms (filling antibonding V 3d–O 2p states) and a slight elongation of V–O bonds.¹⁸ The accompanying adverse electron localization diminishes the availability of charge carriers for surface photoreactions. Intriguingly, it is now well established that doping the BiVO₄ lattice with molybdenum (Mo⁶⁺) or tungsten (W⁶⁺), at V⁵⁺ sites in the bulk, can strongly improve the PEC performance.^{11,19–23} In an effort to shed light on the effects of Mo and W doping on the bulk conductivity and PEC activity, Rettie *et al.*, performed a series of comprehensive electronic transport measurements on 0.3–0.6% Mo/W:BiVO₄ single crystals. Despite improvements in PEC activity, results verified that the main conduction mechanism in Mo or W doped BiVO₄ was still small-polaron hopping (SPH) and that the electron drift mobility remained low: $\sim 10^{-4}$ cm² V⁻¹ s⁻¹ at 300 K.^{8,12} Rettie *et al.*, also observed a conductivity anisotropy ratio, σ_{ab}/σ_c , in the *ab*-plane relative to the *c*-plane of approximately 2.3. The discrepancy was structurally rationalized in terms of a next nearest-neighbour hopping path, only available in the *ab* plane.¹² In agreement with the work of Abdi *et al.*, the results suggest that the improved PEC performance seen in many studies is associated with increased charge carrier concentrations on account of Mo⁶⁺ or W⁶⁺ (electron donor) substitution at the V⁵⁺ site.^{7,12,21} Results also suggest that the bulk electronic conductivity remains a key obstacle in the PEC activity of BiVO₄.²¹ However, the formation and nature of deep trap states or small-polaron sites, thought to be responsible for the low conductivity, along with their effects on the conversion efficiency, are open to interpretation.

Our current understanding of the structural aspects associated with electron small-polaron formation in M:BiVO₄ (M = Mo or W) stems from transport data and theoretical calculations. This forms a foundation for interpreting the material's electronic structure and electron localization from an X-ray spectroscopic perspective, as recently pointed out by Kweon *et al.*¹⁸ In this paper, we focused on detailed measurements of the electronic structure of BiVO₄, 0.6 at% Mo:BiVO₄ and 0.3 at% W:BiVO₄ single crystals using X-ray absorption spectroscopy (XAS), X-ray emission spectroscopy (XES), resonant inelastic X-ray scattering (RIXS) and X-ray photoelectron spectroscopy (XPS). Preliminary measurements were aimed at (i) probing the fundamental electronic structure of pure and doped BiVO₄ in an effort to standardise our results and to build on earlier spectroscopic studies, and (ii) at evaluating the influence of Mo or W on the electronic band gap. Next, we focused on the identification of localized electron (small-polaron) sites, which have recently been theoretically predicted to result in the presence of inter-band gap spectroscopic fingerprints.¹⁸ The data is interpreted in the context of current theories aimed at linking the influence of Mo/W doping on the PEC performance of BiVO₄.

2. Experimental

2.1 Materials

High quality single crystals of BiVO₄, 0.6 at% Mo doped BiVO₄ and 0.3 at% W doped BiVO₄ were grown by a floating zone method in an optical image furnace at the University of Texas at Austin.⁸ Details regarding the crystal growth and

characterization along with the results of PEC performance and transport measurements have been comprehensively reported in our earlier publications.^{8,12}

2.2 Spectroscopic methods

Single crystal boules were cleaved *ex situ* to give rectangular plates which revealed mirror-like faces. Soft X-ray spectroscopy (XAS, above-threshold XES and RIXS) data was collected at beamline 8.0.1 of the Advanced Light Source (ALS), Lawrence Berkeley National Laboratory (LBNL). XAS data were collected in the surface sensitive (~ 10 nm) total electron yield (TEY) mode by measuring the sample drain current, and in the bulk sensitive (~ 100 nm) total fluorescence yield (TFY) mode. The energy scales of XAS spectra at the V L- and O K-edges were calibrated with reference to the Ti L- and O K-edge XAS spectra of rutile TiO₂. Spectral resolution was ~ 200 meV at full width at half maximum (FWHM). Agreement between TEY and TFY spectra indicated that BiVO₄ did not charge during measurements, and that the TEY signal was also illustrative of the bulk. O K-edge XES and V L₃ edge RIXS data were calibrated with reference to the L _{α 1,2} and L _{β 1} second order emission lines of a Zn metal, and using the elastic scattering peaks of V L _{α} emission spectra. The XES spectral resolution was 700 meV. For RIXS measurements at the V L₃-edge, the photon resolution was ~ 300 meV. Both XES and RIXS spectra were recorded in a 90° scattering geometry, *i.e.* the angle between the wavevectors of the incident and scattered X-rays. Thus, the incident X-rays were at 45° relative to the surface normal of the sample. This decreases contributions from the elastic feature in the spectra by reducing elastic scattering from the material for linearly polarized light in the scattering plane, as the Brewster angle is 45°.²⁴ The analysis chamber pressure was below 5×10^{-9} Torr during measurements. XPS data were collected at beamline 11.0.2 of the ALS. The UHV chamber base pressure was below 5×10^{-9} Torr for measurements. VB spectra were collected with 275 eV photons and calibrated against the 4f peaks of a metallic gold foil in electrical contact with the sample.

2.3 Data analysis

To remove undesirable high frequency noise components from the intensity signals of noisy spectra, a Finite Impulse Response (FIR) Low Pass (LP) filtering method was implemented.^{25,26} The FIR LP filter was designed by first modelling the intensity signal in the time domain where its duration was set to 1 second. The signal was then transformed to the frequency domain using the Fourier Transform and its spectral components were observed. This is shown in ESI Fig. 1(a)† for an example BiVO₄ V L₃ RIXS spectrum ($h\nu = 515.2$ eV).²⁷ The FIR LP filter used was designed with start and stop cut-off frequencies of 80 and 100 Hz, respectively.²⁵ The filter eliminates all frequency components above 100 Hz, while not modifying, that is, passing, all frequency components between 0 and 80 Hz. The 80–100 Hz frequency range is known as a transition band of a practical filter in which frequencies are progressively attenuated. An inherent feature of a practical FIR filter is that it reduces the number of signal samples during the filtering process. The



number of samples reduced by the filtering process increases as the transition band of an FIR filter is shortened. The non-filtered and filtered intensity signals are shown in the time domain in ESI Fig. 1(b).[†] High frequency noise components were successfully removed, leaving a smooth signal which accentuates desired low frequency characteristics. The designed filter was found to produce the most optimal results out of a number of other FIR LP filters with different cut-off frequencies. Filters based on wavelets offering other advantages over the FIR LP method are described in the ESI.^{†26}

3. Results and discussion

Fig. 1(a) shows V $L_{3,2}$ -edge XAS, TEY (black) and TFY (red), for (a.) BiVO_4 , (b.) 0.6% Mo:BiVO_4 and (c.) 0.3% W:BiVO_4 . Spectra are dominated by two regions; L_3 (excitation from $2p_{3/2}$ core levels to unoccupied CB V 3d states) and L_2 (excitation from $2p_{1/2}$ core levels to unoccupied CB V 3d states). The regions are separated by 6.65 eV due to the spin-orbit coupling of the 2p core-hole, in agreement with the work of Cooper *et al.*¹⁶ The L_3 edge further splits into two main regions; the e_g (516.3 eV) and t_{2g} (518.5 eV) states, separated by 2.14 eV (which is related to the ligand field splitting parameter, $10Dq$).²⁸ Peaks 'I' and 'III', at 515.4 and 517.2 eV, respectively, are triplet states arising from atomic multiplet effects associated with the core-hole following transitions from occupied $2p_{3/2}$ states to unoccupied 3d states (e_g and t_{2g}) of V^{5+} (*i.e.* p-d orbital interactions).²⁸ These features, having wavefunctions with strong ligand character, are mixed through spin-orbit interactions and Coulomb repulsion into the L_3 edge. They provide information regarding charge transfer between oxygen ligands and d^0 metal sites using V L_3 RIXS (as will be discussed).²⁸ The L_2 edge has a similar shape to the L_3 edge; however, as the lifetime broadening at the L_2 edge is shorter, the spectral features are not obvious. Collectively, the spectra provide evidence of a distorted tetrahedral environment

of V^{5+} in the monoclinic scheelite structure of BiVO_4 . The d-states of the dopant distribute near the bottom of the VB (bonding M-O features) and CB (anti-bonding M-O features).²⁹ However, spectral changes with doping are not evident owing to their low concentration in the lattice and similar atomic radii to V (giving only a slight elongation of the M-O bond).²⁹ Furthermore, formation of a small-polaron at V $3d_{z^2}$ states is accompanied by partial electron delocalization to the surrounding oxygen atoms, resulting in a localised distortion due to V-O bond elongation.¹⁸ This is expected to produce a weak spectral feature at energies within the electronic band gap.¹⁸ However, the samples already have a strongly distorted local environment (due to the core-hole). Thus, the small-polaron (a state bound to a weak local distortion), and its spectroscopic fingerprint, will be strongly susceptible to core-hole effects. This makes it difficult to detect simply through XAS or *non-resonant* XES (Fig. 2(a)). Spectroscopic evidence of small-polaron-like features is shown in later text.

Fig. 1(b) shows O K-edge XAS in TEY (black) and TFY (red) modes for (a.) BiVO_4 , (b.) 0.6% Mo:BiVO_4 and (c.) 0.3% W:BiVO_4 . O K-edge XAS involves transitions from occupied 1s core levels to empty O 2p states, thus providing an insight into unoccupied metal states due to hybridization with oxygen valence states.³⁰ Characteristic peaks at 530.8 eV (labelled I) and 532.1 eV (II) are indicative of O 2p-V 3d hybridization and in agreement with the tetrahedral crystal field split V e_g and t_{2g} states.³¹ The t_{2g} feature is a contribution of two peaks (at 531.7 eV and 532.5 eV) owing to the V 3d orbital non-degeneracy.¹⁶ A strong feature at 535.7 eV (III) (particularly obvious in the TFY spectra of each material) is related to Bi 6p states in the bulk.¹⁶ A feature at 537.3 eV can be clearly resolved in the TFY spectra and is prominent in undoped BiVO_4 and in Mo:BiVO_4 but absent in W:BiVO_4 . Its origin will be addressed below, in the context of O K-edge resonant XES. Finally, a broad band extending beyond 538 eV indicates mixing of metal 4sp and O 2p electrons.³¹ As

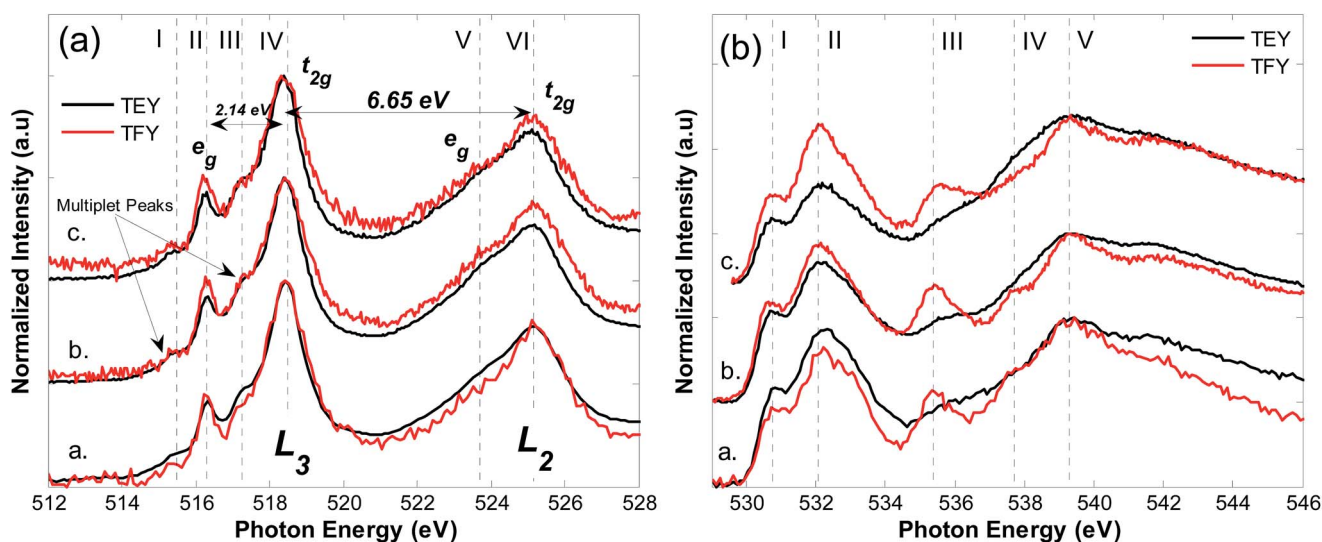


Fig. 1 (a) V $L_{3,2}$ XAS in the TEY (black) and TFY (red) modes for (a.) BiVO_4 , (b.) 0.6% Mo:BiVO_4 and (c.) 0.3% W:BiVO_4 . (b) O K-edge XAS in the TEY (black) and TFY (red) modes for (a.) BiVO_4 , (b.) 0.6% Mo:BiVO_4 and (c.) 0.3% W:BiVO_4 . Spectra are vertically offset for clarity.



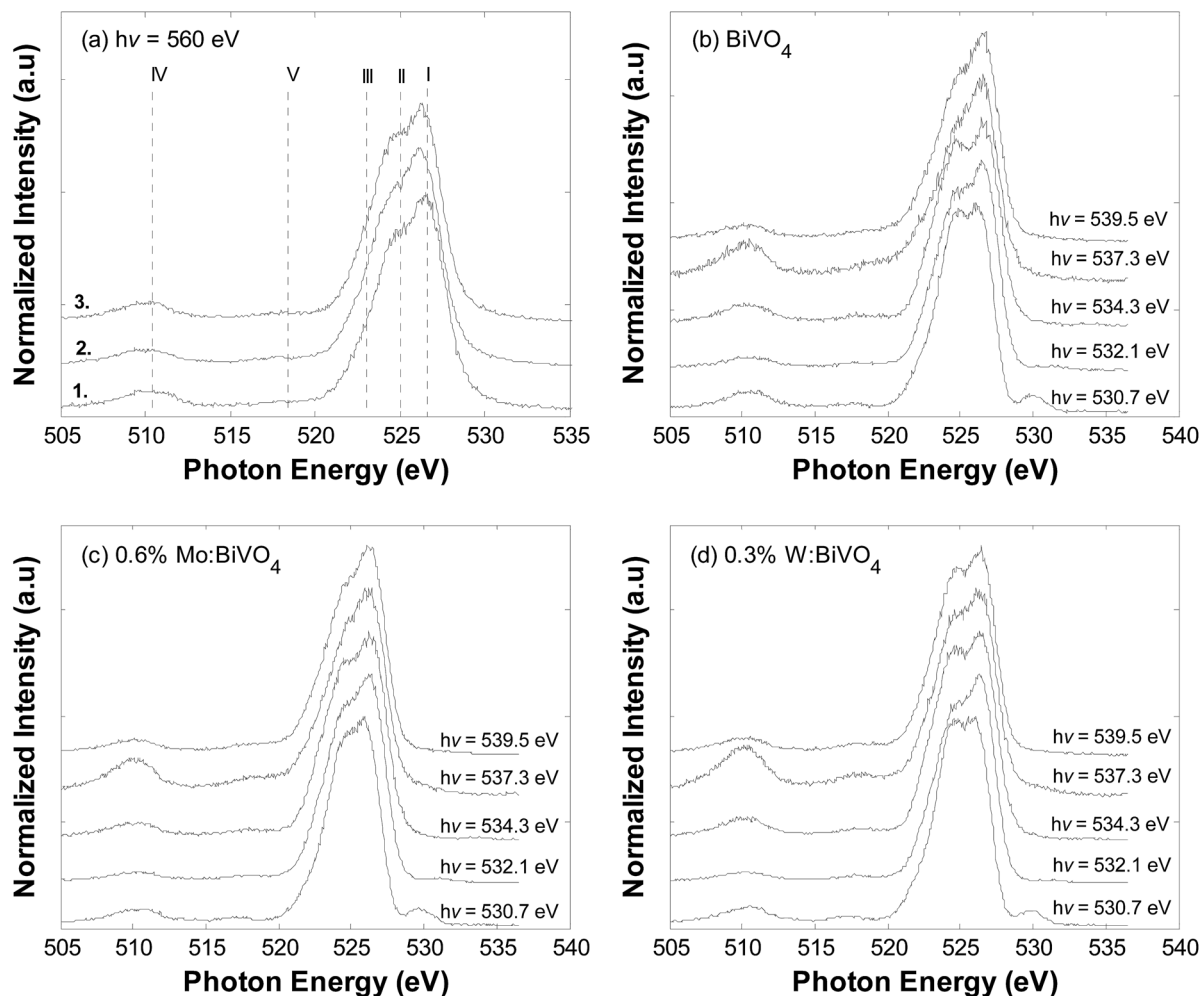


Fig. 2 (a) O K XES spectra for (1.) BiVO_4 , (2.) 0.6% Mo:BiVO_4 and (3.) 0.3% W:BiVO_4 collected with an incident photon energy of 560 eV. (b) BiVO_4 O K-edge RXES. (c) 0.6% Mo:BiVO_4 O K-edge RXES. (d) 0.3% W:BiVO_4 O K-edge RXES.

noted, oxygen states will hybridize with dopant d-states, however, due to low doping levels and strong core-hole effects, spectra do not reflect the presence of Mo and W in the systems.

O K-edge XES (transitions from O 2p states to O 1s core level) was used to probe the materials occupied partial density of states (pDOS). Fig. 2(a) shows XES for (1.) BiVO_4 , (2.) 0.6% Mo:BiVO_4 and (3.) 0.3% W:BiVO_4 collected at an incident photon energy of 560 eV (well above the absorption threshold). A broad feature from 520–530 eV is indicative of a VB dominated by O 2p states, typical of TM oxides. The peak at 526.5 eV (I) signifies emission from O 2p_π states near the VB maximum and experiences a ~0.2 eV low energy shift after 0.6% Mo doping. A similar observation is seen with 0.3% W doping and suggests weak broadening of the bulk band gap energy with doping, verified below. Emission from hybridized O sp²/V 3d states near the VB centre is indicated by a feature at 524.7 eV (II).¹⁶ Spectral broadening near the bottom of the VB is due to emission from hybridized O sp²–Bi 6p states at 523 eV (III).¹⁶ A low energy feature at 510.5 eV (IV) is associated with the V L_{α1,2} emission line (transitions from V 3d to V 2p states). Similarly, a small peak at 518 eV (10 eV below the VB maximum) is likely related to

V L_β emission. This is based on the similar energetic separation (~6.65 eV) as the spin-orbit splitting of the V L₃ and L₂ absorption features in Fig. 1(a). Previously, this feature was also proposed to be associated with Bi 6s states due to the suppression of the V L_β emission by the more intense O K_α emission line (O 2p → O 1s).^{15,32} Due to the 3+ charge of Bi in BiO_8 , the 6s states have an electron lone pair which interacts with O 2p states. This results in an occupied antibonding state with minority 6s character at the VB maximum and majority 6s character at the minimum.^{13,15} Bi 6s–O 2p coupling, and the resulting lone pair of electrons, distorts the VO_4 and BiO_8 units. This weakens the antibonding destabilization.¹⁵ The distortion may also result in the poor wavefunction overlap of V sites in the c-direction, possibly aiding in electron localization (this is further addressed in the text below).¹⁶

By tuning the photon energy to that of an XAS feature, resonant XES (RXES) can be used to provide site specific DOS for the associated chemical environment. Fig. 2 also shows O K-edge RXES data for (b) BiVO_4 , (c) Mo:BiVO_4 and (d) W:BiVO_4 excited at select energies along the O K-edge absorption edge. All spectra show emission at; (i) 526.5 eV (from O 2p_π states); (ii)



524.7 eV (from O sp^2/V 3d states); (iii) 523 eV from O sp^2/Bi 6p hybridized states, (iv) 510 eV due to emission from V L_{α} states and; (v) weak Bi 6s/O 2p states at 518 eV. Resonance with the e_g peak (530.7 eV) results in quasi-elastic scattering (at the same energy) which is quenched by fluorescence signals at higher resonance energies. At resonance energies near the onset of the O K-edge absorption (*i.e.* where electrons are promoted to unoccupied valence states near the CB minimum) emission from O sp^2/V 3d states is enhanced. At higher energies, emission features from O 2p states predominate. Resonance at 537.3 eV (with a feature in the bulk sensitive TFY plots of $BiVO_4$ and 0.6% Mo: $BiVO_4$) gives a distinct enhancement of the V L_{α} emission feature at 510 eV. This is coupled with an enhancement V L_{α} feature at 518 eV and overall a broadening of the VB. Thus, we associate this feature with emission from hybridized V–O sp^2 states.

We next consider the effects of doping on the size of the $BiVO_4$ bulk band gap. Fig. 3 compares O K-edge XAS TFY (red) with non-resonant O K-edge XES (black) for $BiVO_4$, 0.6% Mo: $BiVO_4$ and 0.3% W: $BiVO_4$. Corresponding first derivative plots are shown in the right panels of Fig. 3. XES data were

collected at $h\nu = 550$ eV, which is sufficiently high above the absorption threshold to extract data regarding the high energy VB onset. At higher energies, closer to the ionisation potential, other processes contribute (*i.e.* double excitations, ionisation, *etc.*) and overestimate the energetic onset. With respect to XAS, the O K-edge is 'less complex' than the V L-edge due to core-hole screening from band edge electronic states of the metal. Screening also lowers core-hole effects (perturbations to the system which shift spectra to lower energies), resulting in a more direct picture of the energetic onset at the conduction band. At the O K-edge, the core-hole shift is equal for all crystals studied here (~ 1 eV).³³ Thus, any evolutions with doping are robust irrespective of the shift. On this basis, the bulk band gap energy can be determined from the separation between the first derivative minima and maxima of the XES and XAS spectra, respectively. In undoped $BiVO_4$, the band gap is determined to be 2.50 ± 0.1 eV (corresponding to transitions from occupied O 2p states at the VB maximum to unoccupied V 3d states at the CB minimum).¹⁷ This is in agreement with recent values found by X-ray spectroscopy.³⁴ A band gap of 2.75 ± 0.1 eV was found for 0.6% Mo: $BiVO_4$. The 0.25 eV broadening with Mo doping is

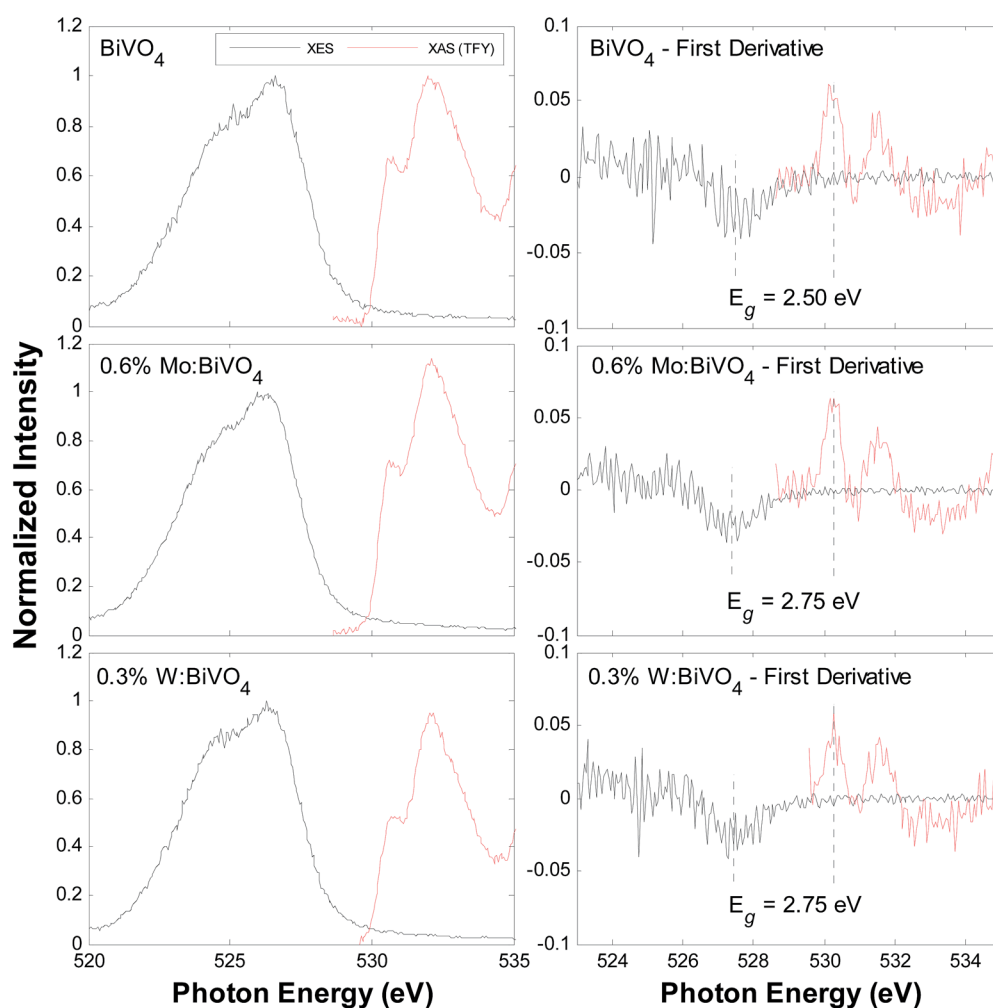


Fig. 3 Left panels; O K-edge XES (black) and O K-edge XAS TFY (red) for $BiVO_4$, 0.6% Mo: $BiVO_4$ and 0.3% W: $BiVO_4$. The XES spectra were taken at $h\nu = 550$ eV. Right panels; first derivative plots for the plots at left.



consistent with a generalized gradient approximation (GGA) predicting an increase in the band gap energy from 2.08 eV to 2.14 eV after 1.04% Mo doping.²⁹ We note that this method underestimates the band gap energy relative to experimental measurements.²⁹ The band gap energy is roughly the same with 0.3% W doping. These experimental values are slightly larger than the 2.40 eV energies found for the crystals using diffuse reflectance UV-vis spectroscopy and incident photon conversion efficiency (IPCE) spectra.⁸ The broadening of the gap is also consistent with an increase in the PEC activity of doped BiVO₄, due to improved transport properties.²² However, this increase in PEC activity is also likely related to changes in the ratio of the effective mass between an electron-hole pair with doping. In pure BiVO₄, the effective mass of charge carriers is thought to be light and equal, which can lead to rapid recombination rates.^{15,29} Doping has been predicted to induce significant variation in the effective masses between electron-hole pairs, particularly along different directions in reciprocal space, which facilitates charge separation and aids in PEC performance.²⁹

Resonant inelastic X-ray scattering, RIXS, can also be used to probe the electronic transitions (direct or indirect) between high and low energy points of the VB and CB, respectively. The direct or indirect nature of the band gap has a strong effect on H₂ production and has been an aspect of debate in

BiVO₄.^{13,29,34,35} Briefly, the VB maximum and the CB minimum are distinguished by particular wavevectors in *k*-space.³⁶ In an indirect band gap (Fig. 4(a)), e[−]-h⁺ angular momenta differ, thus radiative recombination rates are low due to the violation of conservation of momentum.³⁵ Non-radiative recombination occurs at grain boundaries or point defects.³⁷ In a direct band gap, angular momenta are equal and recombination is more dynamic (shown by time resolved microwave conductivity on anatase and rutile TiO₂).^{38,39} In this process, electron excitation to the CB manifests in an intermediate state with a core-hole. Decay of a valence electron to the core-hole conserves momentum between the initial and final states, thus emission occurs from valence states at the same point along the *E*-*k* dispersion as for absorption.^{34,40} The use of RIXS as a probe of vertical transitions in *E*-*k* space was elegantly shown by Cooper *et al.*, on BiVO₄ thin films spin coated on silica substrates.³⁴

Fig. 4 also shows O K-edge RIXS for (b) BiVO₄, (c) 0.6% Mo:BiVO₄ and (d) 0.3% W:BiVO₄ collected at 530.8 eV, 532.6 eV, 534.4 eV (in the vicinity of the O K-edge absorption threshold). The crystals show a shift in the VB maximum with increasing excitation energy (*i.e.* as electrons are promoted to higher energies beyond the onset of the CB). Spectra signified by black lines were collected with incident photon energies of ~530.8 eV (just above the onset of absorption). Here, a core electron moves

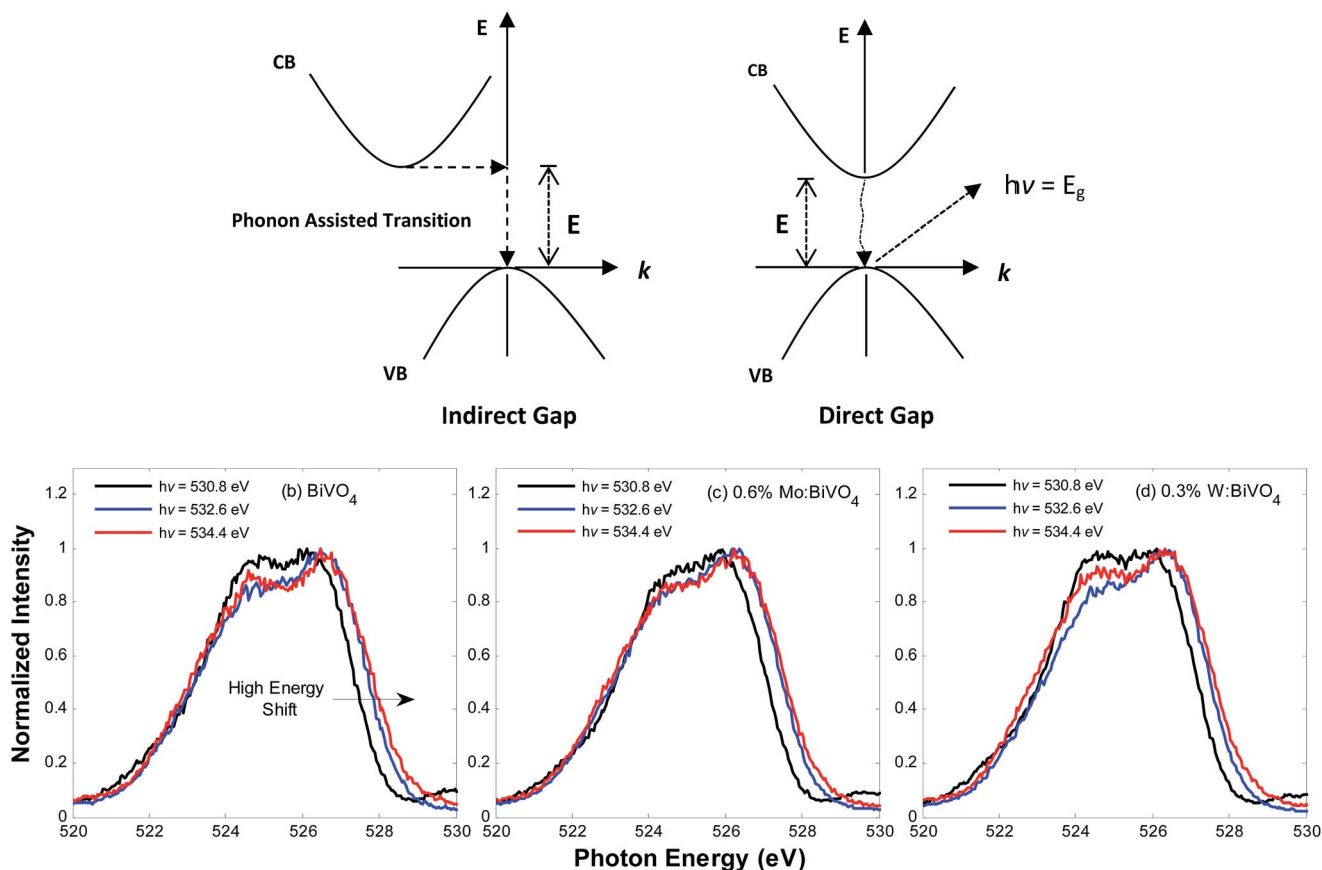


Fig. 4 (a) The indirect (left) and direct (right) electronic band gaps. (b–d) Plots of O K-edge XES spectra collected at three energies near the onset of the O K-edge XAS. The high energy shift of VB onset with increasing photon energy verifies that the band gap of the material is indirect and unaffected by Mo or W doping.



near the CB minimum. The coherent part of the emission comes from valence states at the same momentum (vertical transition in the dispersion relation). With increasing energy, we move away from the CB minimum and see that the spectra shift to higher energies. Thus, emission occurs from states closer to the VB maximum, which happens to be at a different k -point relative to the CB minimum. The data suggests that BiVO_4 has an indirect gap (lower in energy than the direct gap) which is unaffected by doping. The observation of an indirect gap in BiVO_4 is consistent with recent experiments and with theory predicting separations of 2.08–2.17 eV between the VB maxima and CB (found at different k -points along the L and M line).^{17,29,34,41} As stated earlier, the GGA method predicting a marginal increase in the band gap energy with doping, also predicted no effect on the indirect gap with doping.²⁹ The use of RIXS as a probe of the band gap nature should also be approached with some caution. For instance, Walsh *et al.* found that BiVO_4 has a direct band gap between the A points of the Brillouin zone corresponding to an energy of 2.21 eV.¹³ They showed that the A point of the valence band has a strong Bi 6s component, and only very weak V 3d character. The O K-edge RIXS (Fig. 4) shows an indirect gap. Here, the excited states (CB)

are predominantly V 3d states. Thus, it may be that fluorescence from the Bi 6s at the very top of the band, near the A point, is suppressed in our measurements, in favour of fluorescence from the deeper states that mix with vanadium. In that case, it would look indirect to the RIXS measurements performed here, but would formally be a direct band gap.

As RIXS probes excitations related to valence states near the Fermi level, it also provides insight into low energy crystal field excitations and charge transfer events between ligand–metal and metal–metal sites.⁴² In particular, RIXS is a sensitive probe of local perturbations in the system on account of Mo^{6+} or W^{6+} doping at V^{5+} sites. Fig. 5(a) displays the V L_3 -edge RIXS spectra for BiVO_4 , collected at incident photon energies of (a.) 516.4 eV (on the V L_3 e_g peak), (b.) 516.2 eV, (c.) 515.9 eV, (d.) 515.3 eV and (e.) 515.2 eV. These energies correspond to excitation along the V L_3 e_g onset (as shown at the inset of plot (a)). Fig. 5(b) shows the spectra on an energy loss scale (*i.e.* referenced to the incident photon energy). BiVO_4 is a d^0 compound, therefore, the strong signal to noise ratio in the RIXS spectra was anticipated.⁴³ Accordingly, we used the Finite Impulse Response (FIR) Low Pass (LP) filtering method, described in the data analysis section, in an attempt to filter the intensity signal by removing

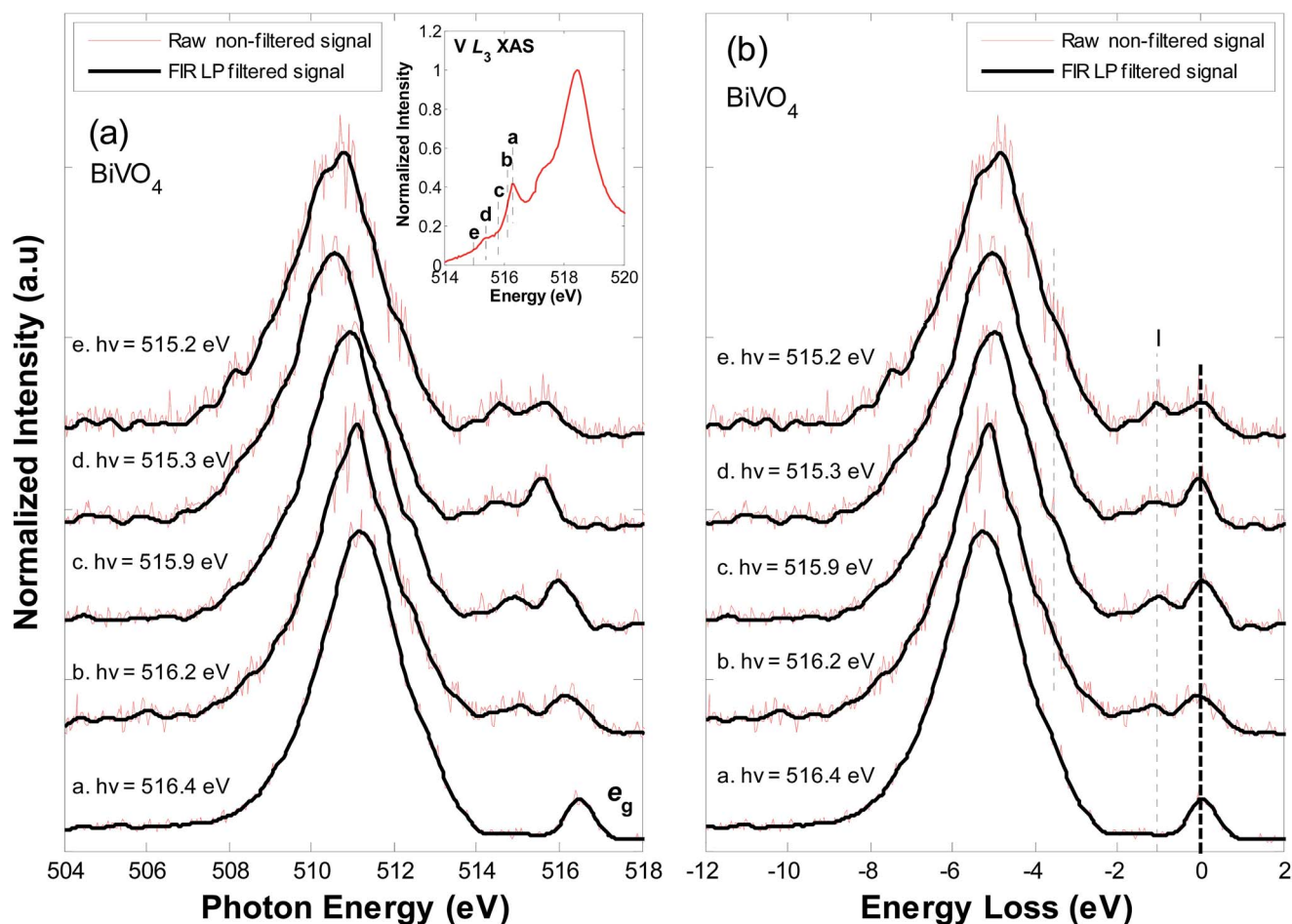


Fig. 5 (a) V L_3 -edge RIXS spectra for BiVO_4 . The spectra have been vertically offset for clarity. The excitation energy for each spectrum is indicated on the plot. (b) The V L_3 -edge RIXS spectra for BiVO_4 on an energy loss (charge transfer) scale. In all plots, the raw, non-filtered, signals are shown in red and the FIR LP filtered signals are shown in black.



noise components.^{25–27} The raw non-filtered signals are shown in red and the FIR LP filtered signals are shown in black.

In V L_3 -edge RIXS (as with other transition metals), three typically different processes are likely to occur, (i) fluorescence from occupied VB states, (ii) charge transfer excitations between hybridized metal–oxygen states and metal–metal sites, (iii) crystal field excitations (transitions within the metal d-manifold).²⁴ Resonance on the e_g peak (516.4 eV) results in a spectrum dominated by fluorescence at ~ 511 eV (emission from hybridized O sp^2/V 3d states) and by elastic scattering of incident radiation. As the incident photon energy is tuned further toward the onset of absorption, the RIXS cross section becomes more Raman-like and less fluorescence-like. This accounts for the apparent shift in the fluorescence feature from 511 eV, in spectrum 'a', to lower energies, in spectra d–f, and also explains the change in shape of the main emission feature. Likewise, resonance near the multiplet state (spectra d–f) enhances signals related to charge transfer processes. Accordingly, the peak broadening between 510–508 eV (-6 eV through -8 eV on the energy loss scale) results from charge transfer excitations from O $2p$ –V 3d sites: O $2p^6 V 3d^n \rightarrow O 2p^5 V 3d^{n+1}$ (a direct RIXS

transition).⁴³ Based on spectral broadening, and on recent theory, the charge transfer most likely corresponds to transitions from low lying V–O states to low lying empty V e_g states.¹⁸ Below 4 eV from the elastic peak, spectra are dominated by scattering features; dd excitations in the metal d-manifold. In particular, an inelastic scattering at ~ 1.0 eV (I) (energy loss) results from transitions within the crystal field (*i.e.* inter-orbital V 3d $e_g \rightarrow e_g$ transitions), in agreement with recent studies.¹⁶ We also address, in the following text, the development of a weak shoulder feature at 3.6–3.7 eV on the energy loss scale.

Fig. 6(a) shows V L_3 -edge RIXS spectra for 0.3% W:BiVO₄, excited at (a.) 516.4 eV, (b.) 515.9 eV, (c.) 515.2 eV and (d.) 514.9 eV (shown at the inset). Fig. 6(b) shows the spectra on an energy loss scale. As for BiVO₄, resonating on the e_g peak results in the characteristic fluorescence signal at 511 eV related to emission from O sp^2/V 3d. Resonance at lower energies results in strong broadening of the main fluorescence feature below 510 eV (-6 eV to -8 eV energy loss) due to O $2p^6 V 3d^n \rightarrow V 3d^{n+1} O 2p^5$ charge transfer processes. Broadening is slightly more prominent in comparison to that in pure BiVO₄. As mentioned earlier, the spectra of BiVO₄ and W:BiVO₄ both show a small

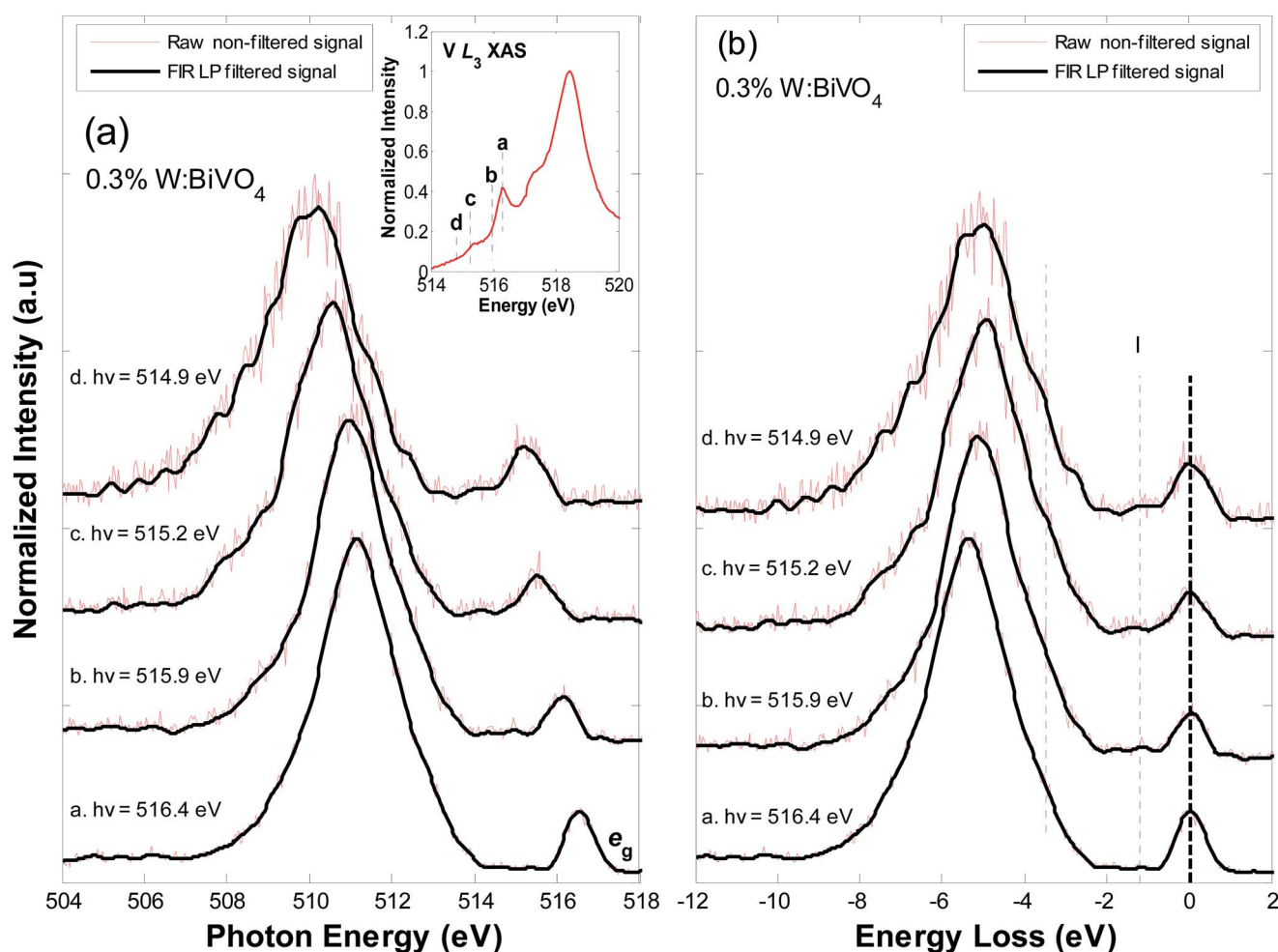


Fig. 6 (a) V L_3 -edge RIXS spectra for 0.3% W:BiVO₄. The spectra have been vertically offset for clarity. The excitation energy for each spectrum is indicated on the plot. (b) The V L_3 -edge RIXS spectra for 0.3% W:BiVO₄ on an energy loss (charge transfer) scale. The raw, non-filtered, signals are shown in red and the FIR LP filtered signals are shown in black.



shoulder peak at ~ 3.8 eV energy loss when resonating at lower energies. In both materials, the lower CB is predominantly V 3d/O 2p in character. Crystal field splitting and compression in the z-direction leaves the V 3d_{z²} states near the CB minimum.¹⁸ As stated in earlier text, DFT/Hartree–Fock theory (which accounts for exchange–correlation potentials and many-body electron interactions) predicts that excess electrons occupy the V 3d_{z²} states, delocalizing to V 3d–O 2p states of the surrounding oxygen ligands and elongating V–O bonds.¹⁸ Polaron formation was expected to result in a *very weak spectral* feature near the conduction band minimum. Based on the theory, transitions from low lying V–O valence states to the polaron band should result in an energy loss feature at ~ 4.8 eV. On this basis, we speculate that the development of feature at ~ 3.8 eV energy loss is possibly related to charge transfer from the lowest lying V–O states to polaron states within the band gap. However, precise energies may vary based on theoretical and empirical accuracy. This, and the V 3d_{z²} nature of the states, is examined in further detail below.

Apart from V L₃ RIXS, empirical signatures of polaron states are likely to be seen by XPS as local perturbations from the core-hole are minimized. Fig. 7(a) shows valence band XPS data for BiVO₄, 0.6% Mo:BiVO₄ and 0.3% W:BiVO₄. The spectra show characteristic contributions from Bi 6s states (10 eV), Bi 6p and V 3d states (8 and 6 eV, respectively) and O 2p states (6 eV below the VB maximum to its onset).¹⁵ A peak appears above the maximum in the doped samples (~ 1.8 eV in binding energy, marked by *). A similar feature was seen by Feng *et al.*, in CeO₂ upon reduction of Ce⁴⁺ to Ce³⁺ and was assigned to localized electrons ('small polarons') occupying a narrow Ce 4f electronic state in the band gap.⁴⁴ To further probe the origin of this feature, we show resonant VB XPS data for 0.3% W:BiVO₄ at excitation energies of 510 eV, 516 eV and 518.5 eV (Fig. 7(b)). These energies correspond to resonance with the V L₃ XAS feature at 516 eV (*i.e.* resonance with the e_g states), the L₃ feature at 518.5 eV (resonance with the t_{2g} states) and off

resonance at 510 eV, as shown in the inset of Fig. 7(b). A distinct increase in the intensity of the feature occurs following resonance at 516 eV (*i.e.* on the e_g peak). The intensity decreases with resonance at 518.5 eV (on the t_{2g} peak). The observations correlate with theory suggesting that the addition of an electron to BiVO₄ creates localized states within the band gap which are V 3d_{z²} in nature.¹⁸ Accordingly, the peak seen here may signify population of low lying V 3d states. As this feature occupies the lower half of the band gap, it may also signify the presence of deeply trapped states formed when BiVO₄ is W doped, as described by Abdi *et al.*²¹ Such an observation would also be consistent with the rapid electron–hole pair recombination rates observed in W:BiVO₄.²¹ Despite having a negative effect on recombination rates, Abdi *et al.* found that the increased carrier density induced by W doping improves the overall photo-response.²¹ This verified that the bulk electronic conductivity is one of the primary setbacks in the PEC performance of BiVO₄. For instance, in our earlier study, we could not detect a measurable photocurrent from undoped BiVO₄ electrodes due to the materials inherently low bulk conductivity. Thus, PEC testing was done with the Mo and W doped samples which showed photocurrents of ~ 0.4 mA cm^{−2} at 1.2 V *versus* RHE (using the (001) faces).⁸ Qualitatively, this observation of improved PEC water splitting performance from the n-type single crystals is in agreement with earlier studies performed using n-type polycrystalline films of Mo/W:BiVO₄.² IPCE spectra were consistent with those of polycrystalline anodes which showed tails extending to 520 nm. However, direct comparisons with polycrystalline anodes is difficult due to the different Mo or W loading percentages and the fact that PEC tests were completed on the exposed (001) faces, whereas polycrystalline anodes have various crystallographic orientations. In this respect, interpretation of PEC activity in single crystal is also not complicated by effects such as grain boundaries. Therefore, single crystals can be considered near model systems for relating transport properties to PEC activity, as reported

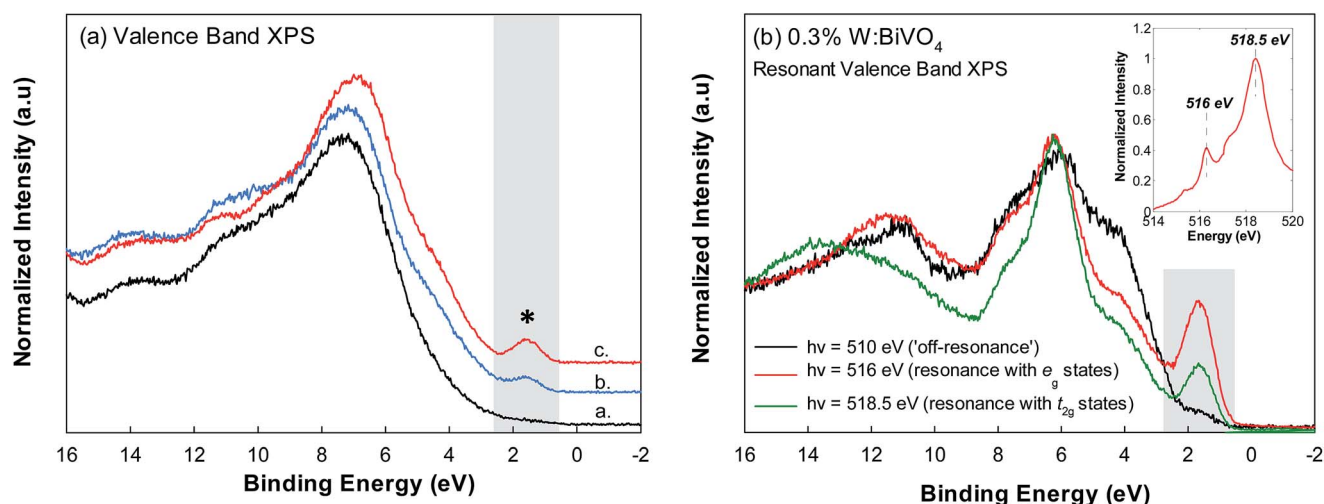


Fig. 7 (a) Valence band XPS spectra for (a.) BiVO₄, (b.) 0.6% Mo:BiVO₄ and (c.) 0.3% W:BiVO₄ collected using an incident photon energy of 275 eV. (b) 0.3% W:BiVO₄ resonant valence band spectra collected at incident photon energies of 510 eV, 516 eV and 518 eV.



earlier.^{8,12} It is also interesting that the 1.8 eV VB feature (Fig. 7(a)) does not scale with the doping percentages, but is most prominent in the 0.3% W doped crystal. This is consistent with our transport measurements, where the Mo and W doped crystals showed the same conductivity despite different doping concentrations.⁸ The result could indicate that W is a more effective dopant, at least in terms of increasing the carrier concentration. It is also surprising that the spectral weight of the feature in Fig. 7(a) is of relatively high intensity, particularly as the maximum doping is 1 at% (*i.e.*, expected 1% V^{4+}). In contrast to RIXS, XPS is highly surface sensitive. Thus, VB spectra may also reflect the presence of native V^{4+} surface sites due to self-doping from oxygen off-stoichiometry initiated by crystal growth. The oxidation state of the doped single crystals was checked by V 2p core level XPS (ESI Fig. 4†). V 2p_{3/2} and 2p_{1/2} spin orbit split (SOS) peaks at 516.9 and 524.0 eV (SOS = 7.1 eV) in a 2 : 1 peak ratio, typical for the V^{5+} oxidation state. Weak shoulder peaks at 515.5 eV signified the expected, but weak, presence of V^{4+} . Thus, self-doping effects are minimal or absent and the VB signal indicating V^{4+} is likely associated with small-polaron and electron trap states within the band gap.

We can again consider the effects of doping on the increased PEC activity of $BiVO_4$.⁷ Our previous study verified that electron mobility in $M:BiVO_4$ is low, but consistent with small-polaron conduction, $\sim 10^{-4}$ cm² V⁻¹ s⁻¹, between 250–400 K.⁸ Furthermore, a conductivity anisotropy ratio of ~ 2.3 was found between the *ab*-plane and *c*-plane.¹² Poor electron mobility was attributed to structural disorder. Specifically, the anisotropy was rationalized in terms of additional ‘Next-Nearest-Neighbour’ (NNN) transfers in the *ab* plane relative to only nearest neighbour hopping along the *c*-axis.^{8,12} The electronic motion was also associated with narrow electronic energy bands near the bottom of the conduction band. In this respect, direct spectroscopic observation of small-polaron-related states in $M:BiVO_4$ may also lend support to recent theory and transport measurements regarding the effects of doping on the conductivity.¹⁸ Specifically, increased charge carrier concentrations induced by doping may result in a higher probability of polaron wavefunction overlap with positive effects on carrier transport. This assists in the migration of photo-generated charge within the $BiVO_4$ bulk, making more charge carriers available for transport to the underlying conducting glass from where they are channelled to the cathode for H₂ production. However, doping levels for such effects may have to be considerably higher than seen here.¹⁸

4. Conclusion

This study used XAS, XES, RIXS and XPS to probe the electronic structure of $M:BiVO_4$ ($M = Mo$ or W), a promising material for use as an anode in PEC water splitting. The data was interpreted in the context of current theories regarding the influence of Mo/W doping on the existence of inter-band gap small-polaron states and their possible effects on the PEC performance of $BiVO_4$. Initially we confirmed that 0.3–0.6 at% Mo and W doping did not affect the indirect nature of the electronic band gap and the catalytically active monoclinic scheelite structure of

$BiVO_4$. Ensuing $M:BiVO_4$ V L₃ RIXS showed excitations from hybridized metal/oxygen states to empty V 3d states, and crystal field excitations within the V 3d-manifold. Furthermore, V L₃ RIXS of pure and $W:BiVO_4$ also showed the first likely sign related to the population of inter-band gap small-polaron states. A small-polaron-like peak was evident in the valence band XPS spectra of $M:BiVO_4$ and intensified when the incident photon energy was resonant with the V L₃ e_g absorption edge. This further pointed to the presence of electrons localized in V 3d_{z²} states of $M:BiVO_4$, in agreement with recent theory. The possible association of this feature with a deep trap state suggests that the improved PEC efficiency of $M:BiVO_4$ relative to undoped $BiVO_4$ is largely due to an increase in charge carrier concentration despite higher charge carrier recombination rates. We hope that this study on the electronic structure of $M:BiVO_4$ will guide further spectroscopic studies related to the nature of detrimental electron localization, therefore pointing to ways of maximizing the materials STH efficiency.

Acknowledgements

The Boston University program is supported by the Department of Energy under Grant No. DE-FG02-98ER45680. The Advanced Light Source is supported by the U.S. Department of Energy under Contract No. DE-AC02-05CH11231. AJER and CBM gratefully acknowledge the U.S. Department of Energy (DOE) Grant DE-FG02-09ER16119 and Welch Foundation Grant F-1436. Additionally, AJER acknowledges the Hemphill-Gilmore Endowed fellowship for financial support. J-SZ was supported by NSF MIRT DMR 1122603. VJ would like to thank the beamline scientists at beamlines 8.0.1 and 11.0.2 of the ALS for their assistance during measurements. Beamline 11.0.2 is also supported by the Director, Office of Science, Office of Basic Energy Sciences, and by the Division of Chemical Sciences, Geosciences, and Biosciences of the US Department of Energy at the Lawrence Berkeley National Laboratory under Contract No. DE-AC02-05CH11231.

References

- 1 X. Lu, S. Xie, H. Yang, Y. Tong and H. Ji, *Chem. Soc. Rev.*, 2014, **43**, 7581–7593.
- 2 Z. Li, W. Luo, M. Zhang, J. Feng and Z. Zou, *Energy Environ. Sci.*, 2013, **6**, 347–370.
- 3 P. Jackson, D. Hariskos, E. Lotter, S. Paetel, R. Wuerz, R. Menner, W. Wischmann and M. Powalla, *Prog. Photovoltaics*, 2011, **19**, 894–897.
- 4 O. Khaselev and J. A. Turner, *Science*, 1998, **280**, 425–427.
- 5 Y. Park, K. J. McDonald and K.-S. Choi, *Chem. Soc. Rev.*, 2013, **42**, 2321–2337.
- 6 F. F. Abdi, L. Han, A. H. M. Smets, M. Zeman, B. Dam and R. van de Krol, *Nat. Commun.*, 2013, **4**, 2195.
- 7 W. Luo, Z. Yang, Z. Li, J. Zhang, J. Liu, Z. Zhao, Z. Wang, S. Yan, T. Yu and Z. Zou, *Energy Environ. Sci.*, 2011, **4**, 4046–4051.



- 8 A. J. E. Rettie, H. C. Lee, L. G. Marshall, J.-F. Lin, C. Capan, J. Lindemuth, J. S. McCloy, J. Zhou, A. J. Bard and C. B. Mullins, *J. Am. Chem. Soc.*, 2013, **135**, 11389–11396.
- 9 M. Li, L. Zhao and L. Guo, *Int. J. Hydrogen Energy*, 2010, **35**, 7127–7133.
- 10 J. A. Seabold and K.-S. Choi, *J. Am. Chem. Soc.*, 2012, **134**, 2186–2192.
- 11 H. S. Park, K. E. Kweon, H. Ye, E. Paek, G. S. Hwang and A. J. Bard, *J. Phys. Chem. C*, 2011, **115**, 17870–17879.
- 12 A. J. E. Rettie, W. D. Chemelewski, J. Lindemuth, J. S. McCloy, L. G. Marshall, J. Zhou, D. Emin and C. B. Mullins, *Appl. Phys. Lett.*, 2015, **106**, 022106.
- 13 A. Walsh, Y. Yan, M. N. Huda, M. M. Al-Jassim and S.-H. Wei, *Chem. Mater.*, 2009, **21**, 547–551.
- 14 P. Nagels, in *The Hall Effect and Its Applications*, ed. C. L. Chien and C. R. Westgate, Springer, US, 1980, ch. 8, pp. 253–280.
- 15 D. J. Payne, M. D. M. Robinson, R. G. Egdell, A. Walsh, J. McNulty, K. E. Smith and L. F. J. Piper, *Appl. Phys. Lett.*, 2011, **98**, 212110.
- 16 J. K. Cooper, S. Gul, F. M. Toma, L. Chen, P.-A. Glans, J. Guo, J. W. Ager, J. Yano and I. D. Sharp, *Chem. Mater.*, 2014, **26**, 5365–5373.
- 17 Z. Zhao, Z. Li and Z. Zou, *Phys. Chem. Chem. Phys.*, 2011, **13**, 4746–4753.
- 18 K. E. Kweon, G. S. Hwang, J. Kim, S. Kim and S. Kim, *Phys. Chem. Chem. Phys.*, 2015, **17**, 256–260.
- 19 H. Ye, J. Lee, J. S. Jang and A. J. Bard, *J. Phys. Chem. C*, 2010, **114**, 13322–13328.
- 20 W. Yao, H. Iwai and J. Ye, *Dalton Trans.*, 2008, 1426–1430.
- 21 F. F. Abdi, T. J. Savenije, M. M. May, B. Dam and R. van de Krol, *J. Phys. Chem. Lett.*, 2013, **4**, 2752–2757.
- 22 S. P. Berglund, A. J. E. Rettie, S. Hoang and C. B. Mullins, *Phys. Chem. Chem. Phys.*, 2012, **14**, 7065–7075.
- 23 Y. Liang, T. Tsubota, L. P. A. Mooij and R. van de Krol, *J. Phys. Chem. C*, 2011, **115**, 17594–17598.
- 24 S. G. Chiuabăian, T. Schmitt, M. Matsubara, A. Kotani, G. Ghiringhelli, C. Dallera, A. Tagliaferri, L. Braicovich, V. Scagnoli, N. B. Brookes, U. Staub and L. Patthey, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2008, **78**, 245102.
- 25 E. C. Ifeachor and B. W. Jarvis, *Digital Signal Processing: A practical Approach*, Addison-Wesley, Wokingham, UK, 1993.
- 26 B. Jovic, *Synchronization techniques for chaotic communication systems*, Springer Science & Business Media, Auckland, New Zealand, 2011.
- 27 R. E. Ziemer, W. H. Tranter and D. R. Fannin, *Signals and Systems: Continuous and Discrete*, New York, US, 3rd edn, 1993.
- 28 F. M. F. de Groot, J. C. Fuggle, B. T. Thole and G. A. Sawatzky, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1990, **41**, 928–937.
- 29 K. Ding, B. Chen, Z. Fang, Y. Zhang and Z. Chen, *Phys. Chem. Chem. Phys.*, 2014, **16**, 13465–13476.
- 30 L. F. J. Piper, A. DeMasi, S. W. Cho, A. R. H. Preston, J. Laverock, K. E. Smith, K. G. West, J. W. Lu and S. A. Wolf, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2010, **82**, 235103.
- 31 J. Laverock, B. Chen, A. R. H. Preston, K. E. Smith, N. R. Wilson, G. Balakrishnan, P. A. Glans and J. H. Guo, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2013, **87**, 125133.
- 32 O. Y. Khyzhun, T. Strunskus, W. Grünert and C. Wöll, *J. Electron Spectrosc. Relat. Phenom.*, 2005, **149**, 45–50.
- 33 B. Chen, J. Laverock, L. F. Piper, A. R. Preston, S. W. Cho, A. DeMasi, K. E. Smith, D. O. Scanlon, G. W. Watson, R. G. Egdell, P. A. Glans and J. H. Guo, *J. Phys.: Condens. Matter*, 2013, **25**, 165501.
- 34 J. K. Cooper, S. Gul, F. M. Toma, L. Chen, Y.-S. Liu, J. Guo, J. W. Ager, J. Yano and I. D. Sharp, *J. Phys. Chem. C*, 2015, **119**, 2969–2974.
- 35 K. Connelly, A. K. Wahab and H. Idriss, *Materials for Renewable and Sustainable Energy*, 2012, **1**, 1–12.
- 36 N. Daude, C. Gout and C. Jouanin, *Phys. Rev. B: Solid State*, 1977, **15**, 3229–3235.
- 37 V. Jovic, Z. N. Al-Azri, W.-T. Chen, D. Sun-Waterhouse, H. Idriss and G. N. Waterhouse, *Top. Catal.*, 2013, **56**, 1139–1151.
- 38 C. Colbeau-Justin, M. Kunst and D. Huguenin, *J. Mater. Sci.*, 2003, **38**, 2429–2437.
- 39 M. Xu, Y. Gao, E. M. Moreno, M. Kunst, M. Muhler, Y. Wang, H. Idriss and C. Wöll, *Phys. Rev. Lett.*, 2011, **106**, 138302.
- 40 Y. Ma, N. Wassdahl, P. Skytt, J. Guo, J. Nordgren, P. D. Johnson, J. E. Rubensson, T. Boske, W. Eberhardt and S. D. Kevan, *Phys. Rev. Lett.*, 1992, **69**, 2598–2601.
- 41 J. Ma and L.-W. Wang, *Appl. Phys. Lett.*, 2014, **105**, 172102.
- 42 L. J. P. Ament, M. van Veenendaal, T. P. Devereaux, J. P. Hill and J. van den Brink, *Rev. Mod. Phys.*, 2011, **83**, 705–767.
- 43 J. Laverock, B. Chen, A. Preston, K. Smith, N. Wilson, G. Balakrishnan, P. Glans and J. Guo, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2013, **87**, 125133.
- 44 Z. A. Feng, F. El Gabaly, X. Ye, Z.-X. Shen and W. C. Chueh, *Nat. Commun.*, 2014, **5**, 4374.

