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Triptycene based 1,2,3-triazole linked network polymers (TNPs): small gas storage and selective CO₂ capture†

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Herein, facile synthesis and characterization of four triazole linked network polymers (TNPs) in high yields is described. These nitrogen rich polymers are derived using a “click” reaction between 2,6,14-triazido triptycene and various di- or triethynyl comonomers. The TNPs are microporous and exhibit high surface area (S_{ABET} up to 1348 m² g⁻¹). Due to the incorporation of the 1,2,3-triazole motif (a CO₂-philic moiety), the TNPs record moderate to high CO₂ uptake (up to 4.45 mmol g⁻¹ at 273 K and 1 bar). The TNPs also show very good CO₂/N₂ (up to 48) and CO₂/CH₄ (8–9) selectivity. While the highest storage capacity has been registered by TNP4 (CO₂ 4.45 mmol g⁻¹ at 273 K and 1 bar, CH₄ 23.8 mg g⁻¹, H₂ 1.8 wt%), the highest CO₂/N₂ and CO₂/CH₄ selectivity is shown by TNP3 which contains additional nitrogen rich building blocks in the form of heteroaromatic pyrazine rings. These results suggest that TNPs are porous materials with potential practical application in gas storage and separation.

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Introduction

Design and syntheses of porous materials have attracted considerable research attention over the past decade.^{1–4} In recent years, research output in this direction has witnessed considerable growth.^{5–13} Depending on the pore size, porous materials may be classified as ultra-microporous^{14–16} microporous,^{12,13} mesoporous^{17–19} or macroporous.^{20,21} Among these, microporous materials have emerged as potential candidates for applications that include, but are not limited to, gas storage and separation. Microporous materials, such as zeolites^{22,23} and metal–organic frameworks (MOFs),^{1–5} contain metal atoms. Alternatively, microporous materials can also be designed in the form of metal-free organic polymers that are highly crosslinked. Acronyms such as COFs, CMPs, CTFs, EOFs, HCPs, MOPs, OCFs, PAFs, PIMs, POFs, POPs, PPFs, PPNs, *etc.* have been coined to describe metal-free porous organic polymers.^{6–13} In general, MOFs have limited industrial applications as porous materials because of the presence of relatively labile coordination bonds that lower their thermal stability.²⁴ On the other hand, in microporous organic polymers (MOPs), light weight elements (such as H, C, N, and O) are interlinked *via* relatively stronger covalent bonds. Therefore these metal-free materials have lower mass densities and higher stabilities (chemical and thermal). In addition, MOPs can be easily tailored with respect to properties such as guest selectivity and pore properties, due

to the availability of a plethora of multifunctional organic molecules that can serve as monomers towards the syntheses of MOPs.

In contemporary research dealing with the development of microporous materials, the design of MOPs for selective CO₂ capture and sequestration has received special attention.^{24–46} This has stemmed from environmental and energy concerns related to CO₂, which is a major greenhouse gas. It is well known that thermal power plants emitting flue gas (containing approximately 15% CO₂) are a major contributor to anthropogenic CO₂. Therefore, in order to contain global climate warming arising out of anthropogenic CO₂ emissions, there is a pressing demand to develop novel materials that can efficiently and selectively adsorb large quantities of CO₂ emitted from fossil-fuel fired power plants (classified as fixed-point CO₂ emission sources). Also it is desirable to separate CO₂ (present as an impurity) from natural gas that contains 80–95% methane.^{47,48} This explains the growing research interest to design MOPs having practical applications such as selective capture and sequestration of CO₂.

As far as the design of polymers for better CO₂ adsorption and gas selectivity is concerned, it has been reported that incorporation of polar groups in the framework of porous organic polymers not only enhances carbon dioxide uptake but also improves its selectivity over methane and nitrogen. This enhancement in CO₂ uptake is attributed to van der Waals interactions (quadrupole–dipole interactions and/or hydrogen bonding) between CO₂ and the polar functional groups that are present in the backbone of MOPs. Additionally, polymeric networks synthesized from electron rich monomers may also exhibit higher CO₂ uptake. This property is attributed to

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favorable Lewis acid–base interactions between the electron deficient carbon atom in CO₂ and the relatively electron rich polymer.^{49,50}

1,2,3-Triazole is a nitrogen rich motif and it can be easily incorporated in the polymer framework by reacting monomers with multiple azide and ethynyl functional groups. Nguyen and co-workers have described the synthesis of porous networks by covalently linking tetrahedral monomers using click chemistry.³³ Triazole molecules (1,2,3- and 1,2,4-triazole) may be considered CO₂-philic, considering the magnitude of their CO₂ binding energies.⁵¹ It is therefore anticipated that polymers having these nitrogen rich moieties in their backbone may be potential candidates for efficient CO₂ sorption and separation. Triptycene, on the other hand, is a robust and structurally rigid motif that has three dimensional orientation of three arene rings fused to a [2.2.2]bicyclooctatriene moiety. Triptycene based porous polymers have often exhibited good gas storage and separation properties.^{52–58} This is often attributed to the “internal molecular free volume (IFV)” that arises from the inefficient packing of rigid three dimensional units of triptycene present in the polymer backbone.^{52,55,57,58}

With these premises in mind, we were interested in designing network polymers that would contain both the 1,2,3-triazole motif (as polar groups) and triptycene motif (as a bulky rigid group). It is anticipated that such polymeric networks would be porous and simultaneously exhibit reasonably high uptake of CO₂, primarily because of the CO₂-philic nature of the polar 1,2,3-triazole moiety and the porosity arising from the inefficient packing of the triptycene units in the polymers.

Herein, we report the synthesis of a series of new triptycene based and 1,2,3-triazole linked network polymers (TNPs) using 2,6,14-triazidotriptycene and various di- and triethynyl substituted comonomers *via* a Cu(I)OAc catalyzed “click” reaction. Complete structural characterization of these polymers and their performance as materials for gas storage as well as selective gas uptake has been described elaborately. TNPs constructed from monomers with three ethynyl functional groups have a relatively higher surface area than TNPs derived from diethynyl comonomers. However, all TNPs exhibit microporosity (pore diameter < 2 nm). Additionally, these microporous TNPs also demonstrate similar or even better carbon dioxide storage ability and selectivity than a wide range of previously reported MOPs.

Experimental section

Materials and methods

Triptycene, aromatic bromides and Cu(I)OAc were obtained from Sigma Aldrich and were utilized without further purification. 2,6,14-Triazidotriptycene was prepared from triptycene using a literature reported protocol.⁵⁹ All the aromatic ethynyl compounds were synthesized from the corresponding aromatic bromides using the Sonogashira cross-coupling reaction.⁶⁰ Chromatographic purifications were performed using silica gel (100–200 mesh). Solid-state ¹³C cross-polarization magic angle spinning (CP-MAS) NMR spectra were obtained using a BRUKER 300 MHz (H-1 frequency) NMR spectrometer at

a mass rate of 8 kHz and CP contact time of 2 ms with a delay time of 3 sec. FTIR analyses were performed using a Shimadzu IR Affinity-1 spectrometer. Elemental analyses were performed using a vario MICRO cube analyser. P-XRD data were recorded using a Rigaku TTRAX III X-ray diffractometer. Porosity analyses were performed using a Quantachrome Autosorb iQ₂ Analyzer using UHP grade adsorbates. In a typical experiment, TNP (60–90 mg) was taken in a 9 mm large cell and attached to the degasser unit and degassed at 120 °C for 12–24 h. The samples were refilled with helium and weighed carefully and then the cells were attached to the analyzer unit. The temperature was maintained using a KGW isotherm bath (provided by Quantachrome) filled with liquid N₂ (77 K), or a temperature controlled bath (298 K and 273 K).

Synthesis of TNP1

A typical experiment for the syntheses of TNPs is described using TNP1 as a representative example. 2,6,14-Triazido-triptycene (189 mg, 0.5 mmol), 2,6,14-triethynyltriptycene (163 mg, 0.50 mmol) and Cu(I)OAc (4 mg, 0.03 mmol) were taken in a Schlenk flask under an inert atmosphere. Degassed DMF (30 mL) was added to the flask with continuous stirring to dissolve the reactants. The flask (containing the reactant) was subsequently heated at 120 °C with vigorous stirring. After a few minutes of stirring, the reaction mixture assumed the form of a light yellow gelatinous precipitate. The reaction mixture was allowed to stir for additional 24 hours with the resultant formation of a large amount of gelatinous precipitate. The reaction mixture was cooled to room temperature and filtered through a glass frit followed by washing with DMF, DMSO, acetone, THF and dichloromethane. The product thus obtained was dried under reduced pressure at 100 °C and crushed with a mortar and pestle to yield a brownish yellow powder.

Yield: 338 mg, 96%; FT-IR (KBr): 3404, 1660, 1464, 1221, 1035 cm⁻¹. Elemental analysis: calculated for C₄₆H₂₅N₉: C 78.51, H 3.58, N 17.91. Found: C 72.57, H 3.67, N 15.13.

Synthesis of TNP2

TNP2 has been prepared following a similar procedure to that described for TNP1 using 2,6,14-triazidotriptycene (189 mg, 0.5 mmol), 4,4'-diethynyl-biphenyl (152 mg, 0.75 mmol) and Cu(I)OAc (4 mg, 0.03 mmol). Unlike TNP1, a large amount of yellow precipitate was observed after 24 hours stirring at 120 °C. After drying at reduced pressure at 120 °C, the final product was obtained as a light yellow powder.

Yield: 300 mg, 88%; FT-IR (KBr): 3400, 1606, 1477, 1226, 1038 cm⁻¹. Elemental analysis: calculated for C₄₄H₂₆N₉: C 77.63, H 3.85, N 18.52. Found: C 71.97, H 4.36, N 15.93.

Synthesis of TNP3

This polymer was synthesized using 2,6,14-triazidotriptycene (189 mg, 0.5 mmol), 2,6-diethynylpyrazine (96 mg, 0.75 mmol) and Cu(I)OAc (4 mg, 0.03 mmol). Unlike other TNPs, formation of a light brown fluffy precipitate was observed within a few minutes of stirring at 120 °C. After 24 hours of stirring, formation of a considerable amount of fluffy precipitate was



observed. After drying at reduced pressure at 120 °C, the final product was obtained as a light brown fluffy powder.

Yield: 242 mg, 85%; FT-IR (KBr): 1569, 1489, 1238, 1033 cm^{-1} . Elemental analysis: calculated for $\text{C}_{32}\text{H}_{17}\text{N}_{12}$: C 67.48, H 3.01, N 29.51. Found: C 62.36, H 3.61, N 25.93.

Synthesis of TNP4

This polymer was synthesized using 2,6,14-triazidotriptycene (189 mg, 0.5 mmol), 1,3,5-triethynylbenzene (75 mg, 0.50 mmol) and $\text{Cu}(\text{I})\text{OAc}$ (4 mg, 0.03 mmol). In this case, a large amount of dark yellow gelatinous precipitation was observed after stirring for 24 hours. After drying at reduced pressure at 120 °C, the final product was obtained as a dark brown solid. The solid product was then crushed in a mortar to yield the final product as a brownish yellow powder.

Yield: 259 mg, 98%; FT-IR (KBr): 3391, 1614, 1486, 1226, 1046 cm^{-1} . Elemental analysis: calculated for $\text{C}_{32}\text{H}_{17}\text{N}_9$: C 72.86, H 3.25, N 23.90. Found: C 65.74, H 4.09, N 20.82.

Results and discussion

Synthesis and characterization of TNPs

The TNPs were synthesized utilizing the well-known $\text{Cu}(\text{I})$ catalyzed click reaction that is known to regioselectively yield the 1,4-disubstituted 1,2,3-triazole derivative. It is well documented in the literature that “standard” azide-alkyne click reactions, carried out in water or water/DMF mixture, utilize $\text{Cu}(\text{I})$ species generated *in situ* via reduction of CuSO_4 with sodium ascorbate. However, in recent years, copper(I) acetate is being used as the source of $\text{Cu}(\text{I})$ instead of the CuSO_4 /sodium ascorbate mixture, since the latter catalytic system either yields insoluble products or reactions are sluggish in nature.⁶¹ Chen and Guan reported click polymerization using CuOAc (source of $\text{Cu}(\text{I})$) in DMF (as a solvent).⁶² Based on this literature report, we have synthesized a series of 1,4-disubstituted 1,2,3-triazole bridged network polymers (TNPs) using 2,6,14-triazidotriptycene and various monomers containing multiple ethynyl functional groups as outlined in Scheme 1.

In a typical reaction for the synthesis of TNPs, 2,6,14-triazidotriptycene,⁵⁹ the corresponding ethynyl monomer (di- or triethynyl) and 6 mol% (with respect to 2,6,14-triazidotriptycene) $\text{Cu}(\text{I})\text{OAc}$ were dissolved in 30 mL of DMF and refluxed for 24 hours under a N_2 atmosphere. In the case of TNP1 and TNP4 that utilized triethynyl monomers, gelatinous precipitation was observed after completion of the reaction. In this case, upon drying the precipitate under vacuum, the final product was obtained as a brownish yellow solid. FESEM images of the product obtained for TNP1 and TNP4 revealed an aggregated sheet-like structure. On the other hand, in the case of reactions leading to the formation of TNP2 and TNP3 (using diethynyl monomers in conjugation with 2,6,14-triazidotriptycene), the corresponding products precipitated as fluffy solids which were distinctly different from the gelatinous precipitates observed in the case of TNP1 and TNP4. As a consequence, for TNP2 and TNP3, the FESEM images revealed the presence of aggregated particles in their morphologies

which was distinctly different from that observed in the case of TNP1 and TNP4 (Fig. 1). Wide-angle X-ray diffraction (WAXD) analyses of TNPs suggested that these polymeric materials were amorphous in nature due to the featureless broad nature of the diffraction pattern in each case (Fig. 1). This amorphous nature of the TNPs was due to the rapid and irreversible formation of the triazole linkage during polymerization.^{12b}

The TNPs thus obtained were structurally characterized using solid state ^{13}C (CP-MAS) NMR, and Fourier transform infrared (FTIR) spectroscopy. The ^{13}C CP-MAS spectrum of TNP4 has been shown as a representative example (Fig. 2). The peak centered at 53 ppm corresponds to the bridgehead carbon of triptycene. The appearance of this peak is consistent with that of literature reported triptycene based monomer or polymers^{54–57,59} thereby confirming the successful incorporation of the triptycene motif in the polymer backbone. The peaks due to the phenyl ring carbons of the triptycene appear in the range of 120–150 ppm. The resonance signal centered at 168 ppm is assigned to a carbon present in the 1,2,3-triazole motif, thereby confirming the formation of 1,2,3-triazole linkages *via* the click reaction. The ^{13}C CP-MAS spectra (Fig. S1†) of the other TNPs match well with those of TNP4, thereby supporting the formation of the desired polymers as shown in Scheme 1.

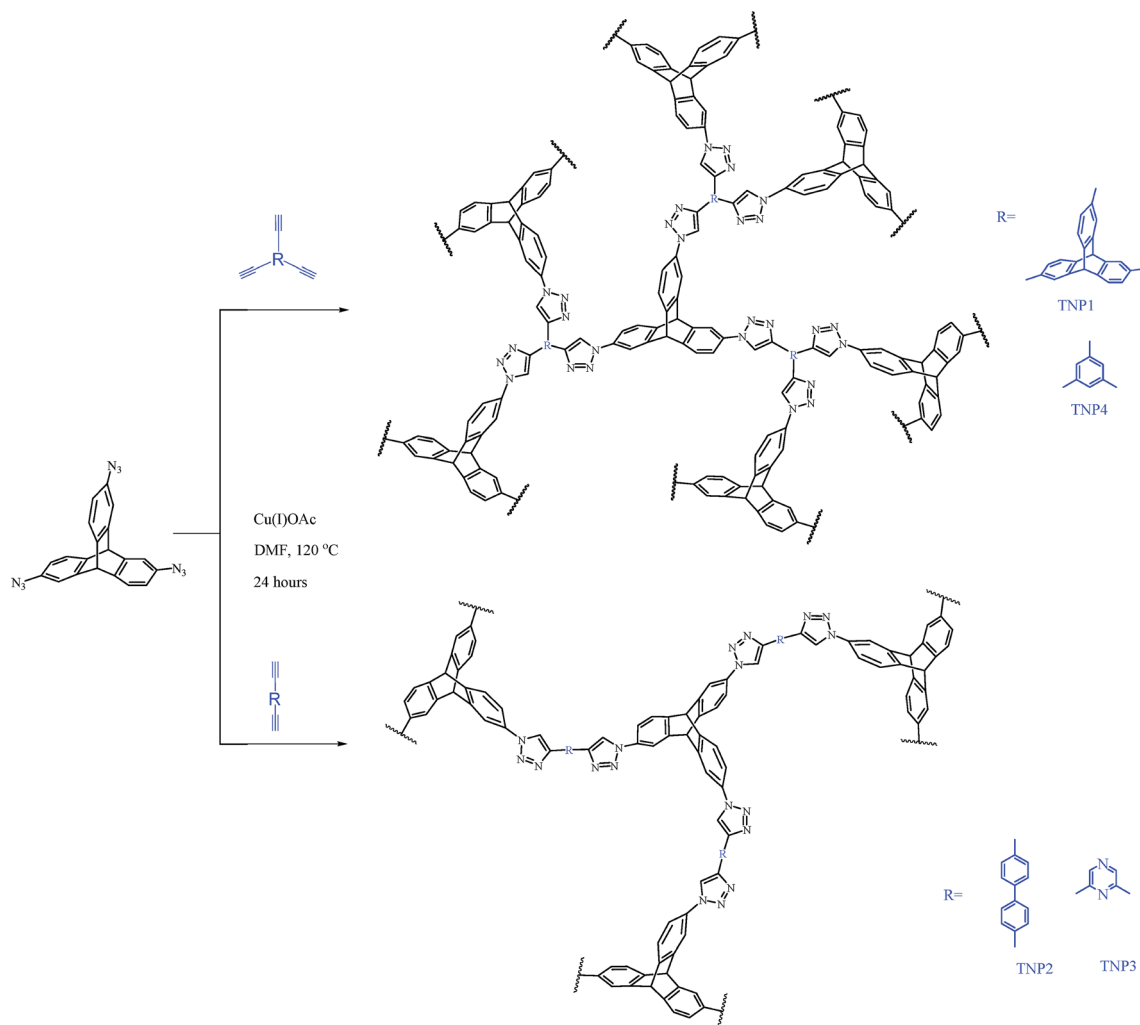
Polymerization has also been confirmed by FTIR spectroscopic data analysis (Fig. S2†). The FTIR spectrum of TNP2 is shown as a representative example in Fig. 3. The characteristic band of the azide functional group present in the triptycene monomer (2096 cm^{-1}) is absent in TNP2. Simultaneously, a band due to the C–H stretching vibration of terminal alkynes (3280 cm^{-1}), observed in the corresponding diethynyl comonomer, is missing in the FTIR spectrum of TNP2. This strongly suggests the click polymerization reaction between monomers containing terminal azide and terminal alkyne groups.

Porosity measurements and gas storage studies

Porous properties of the TNPs were investigated by subjecting these polymers to N_2 adsorption–desorption measurements at 77 K. The isotherms are characterized by a steep increase in slope in the low relative pressure range ($P/P_0 = 0\text{--}0.1$) and this implies uptake of large quantities of N_2 gas. Additionally, the isotherms are reversible in nature. These characteristic features of the obtained adsorption–desorption isotherms (Fig. 4) suggest that these are type-I isotherms and the corresponding TNP polymers reported herein are microporous materials, as per IUPAC classifications.⁶³ A gradual increase in N_2 uptake in the relatively high pressure range ($P/P_0 = 0.1\text{--}0.9$) was also observed in these isotherms. In the case of all TNPs, the adsorption and desorption branches of the hysteresis loop remain almost parallel over an appreciable range of P/P_0 , thereby suggesting that these materials (TNP1–4) show type H4 adsorption–desorption hysteresis. According to IUPAC, “the type H4 loop appears to be associated with narrow slit-like pores, but in this case the type I isotherm character is indicative of microporosity”.⁶³

An additional feature observed in the isotherms of TNP2 and TNP3 is their steep rise while the relative pressure (P/P_0) is

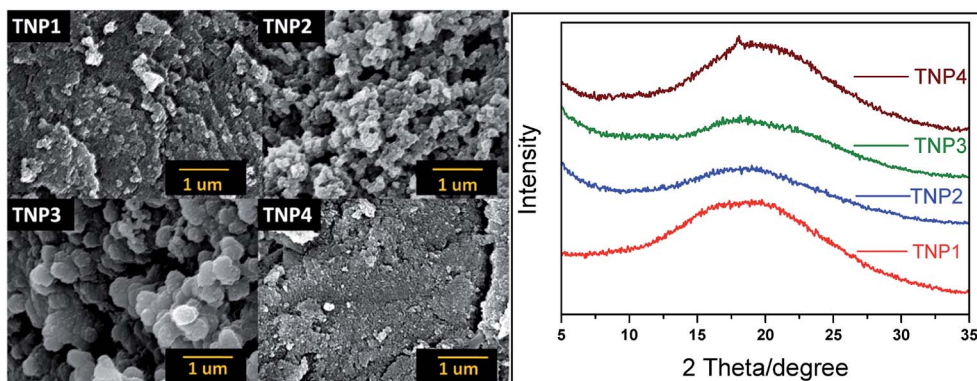




Scheme 1 Synthesis of TNP1–TNP4.

greater than 0.9, which is an indication of the presence of larger pores (macropores) and interparticle voids.^{24,64} Thus isotherms of **TNP2** and **TNP3** are distinctly different from those of **TNP1** and **TNP4**. It may be recalled that in the FESEM micrographs, **TNP2** and **TNP3** exhibited particle-like nature having

interparticle porosity, while **TNP1** and **TNP4** exhibit more aggregated plate-like morphologies suggesting that these have insignificant interparticle porosity. Thus conclusions drawn from the analysis of FESEM micrographs and adsorption-desorption isotherms substantiate each other.

Fig. 1 FESEM micrograph of TNP1–TNP4 (left); WAXD pattern of TNP1–TNP4 recorded at ambient temperature at a scan rate of 3° min^{-1} (right).

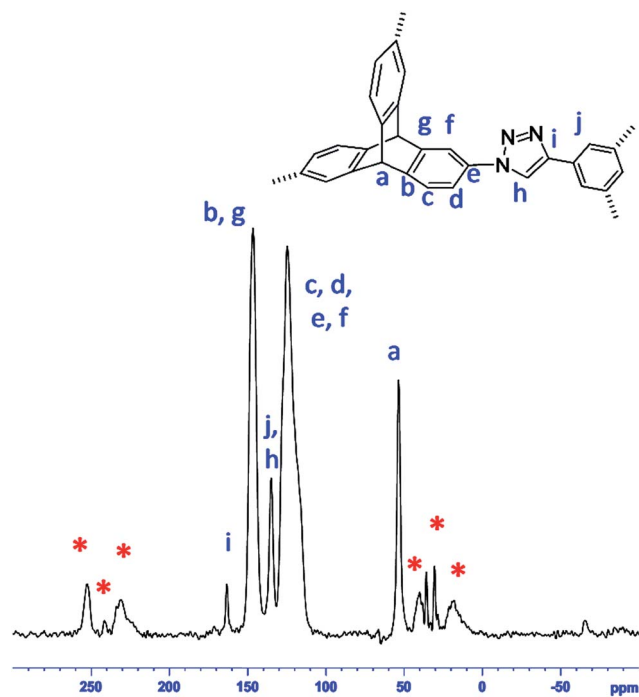


Fig. 2 ^{13}C CP-MAS NMR spectrum of TNP4, asterisks represent the spinning sideband.

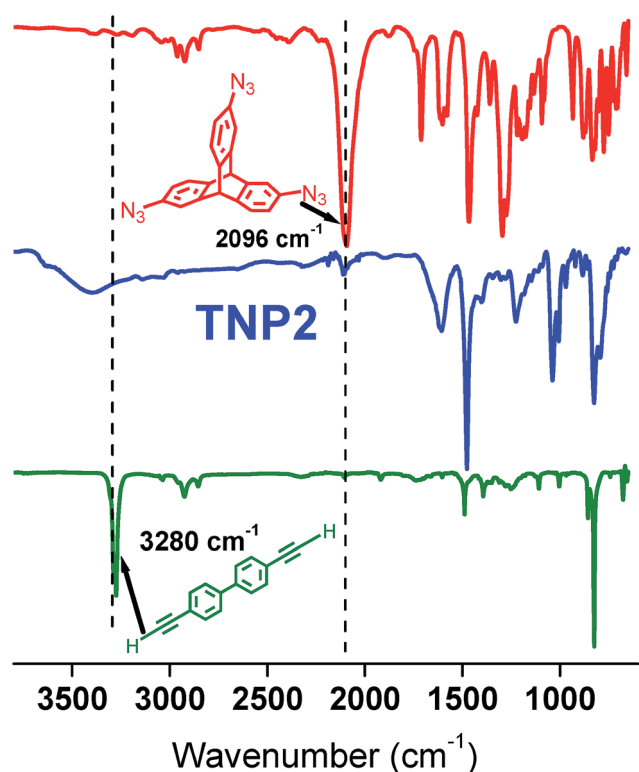


Fig. 3 FT-IR spectrum of 2,6,14-triazidotriptycene (top), TNP2 (middle), and 4,4'-diethynylbiphenyl (bottom).

DFT model based analysis results of pore size distributions (PSDs) of TNPs (calculated from N_2 adsorption data at 77 K) are depicted in Fig. 4b. In the case of TNP1 and TNP4 wherein both

comonomers are trifunctional, the pores are smaller than 2 nm. For these polymers, a narrow peak is observed in the ultra-microporous region (pore width < 0.7 nm) at 0.59 nm and 0.57 nm respectively, thereby suggesting the presence of ultra-micropores in these polymers. The PSD profiles of TNP2 and TNP3 also confirm the presence of micropores. Here a major peak is centered at 1.7 nm and 1.6 nm respectively for TNP2 and TNP3. The total pore volume of TNPs (calculated at $P/P_0 = 0.99$) is in the range of 0.53 to 0.66 $\text{cm}^3 \text{g}^{-1}$ (Table 1). In general, PSD is relatively narrow in the case of polymers (TNP1 and TNP4) that utilize both trifunctional monomers relative to polymers (TNP2 and TNP3) where one of the comonomers is bifunctional. These results clearly show that the pore size distribution of TNPs can be controlled by varying the number of reactive sites in the monomers. The microporous nature of TNPs is attributed to the presence of the rigid three dimensional triptycene motifs in the polymer backbone.

The surface areas (SAs) of TNPs have been obtained using the Brunauer–Emmett–Teller (BET) model within the pressure range $P/P_0 = 0.01$ to 0.1 (Fig. S3†). The SA_{BET} obtained for TNP1, TNP2, TNP3 and TNP4 are 1090 $\text{m}^2 \text{g}^{-1}$, 460 $\text{m}^2 \text{g}^{-1}$, 745 $\text{m}^2 \text{g}^{-1}$ and 1348 $\text{m}^2 \text{g}^{-1}$ respectively. The corresponding Langmuir surface areas are 1475 $\text{m}^2 \text{g}^{-1}$, 756 $\text{m}^2 \text{g}^{-1}$, 1006 $\text{m}^2 \text{g}^{-1}$ and 1723 $\text{m}^2 \text{g}^{-1}$ respectively (Table 1 and Fig. S4†). Polymers TNP1 and TNP4 synthesized from triethynyl comonomers exhibit higher surface areas relative to the other two polymers (TNP2 and TNP3) where a diethynyl comonomer has been utilized for polymerization. This difference in the surface area of network polymers (TNP1 and TNP4) resulting from triethynyl monomers relative to polymers (TNP2 and TNP3) derived from disubstituted monomers is attributed to the higher extent of cross-linking as well as difference in the geometry of the comonomers. Irrespective of the nature of ethynyl monomer used in conjugation with 2,6,14-triazidotriptycene, the resulting TNPs are significantly porous and their corresponding surface areas (SA_{BET}) are comparable to or better than those of a wide variety of microporous polymers reported in the literature such as mPAF (948–1314 $\text{m}^2 \text{g}^{-1}$),²⁷ benzimidazole-linked polymers (BILPs, 599–1306 $\text{m}^2 \text{g}^{-1}$),^{25,65,66} imine-linked PPFs (419–1740 $\text{m}^2 \text{g}^{-1}$),²⁴ nitrogen-doped microporous carbons (263–744 $\text{m}^2 \text{g}^{-1}$),²⁹ imine-linked POFs (466–1521 $\text{m}^2 \text{g}^{-1}$),³¹ nanoporous azo-linked polymers (ALPs, 862–1235 $\text{m}^2 \text{g}^{-1}$),³² “click-based” POPs (1090–1440 $\text{m}^2 \text{g}^{-1}$)³³ and triptycene-based microporous polymers such as benzimidazole networks (600 $\text{m}^2 \text{g}^{-1}$),⁵⁶ porous polymers derived from Tröger's base (TB-MOP, 694 $\text{m}^2 \text{g}^{-1}$),⁵⁸ and microporous polyimides (STPIs, 4–541 $\text{m}^2 \text{g}^{-1}$).⁵⁴ The surface areas and pore properties are summarized in Table 1.

Considering the microporous nature of TNPs and the presence of nitrogen-rich 1,2,3-triazole moieties in the polymer framework, we were interested to investigate the potential of TNPs as porous materials for gas storage applications. It has been reported in the literature that nitrogen enriched microporous organic polymers perform well as materials for storage of small gas molecules in general and particularly CO_2 . Therefore to assess the performance of TNPs in selective gas storage materials, adsorption isotherms for CH_4 , CO_2 , and H_2 were collected under different conditions.



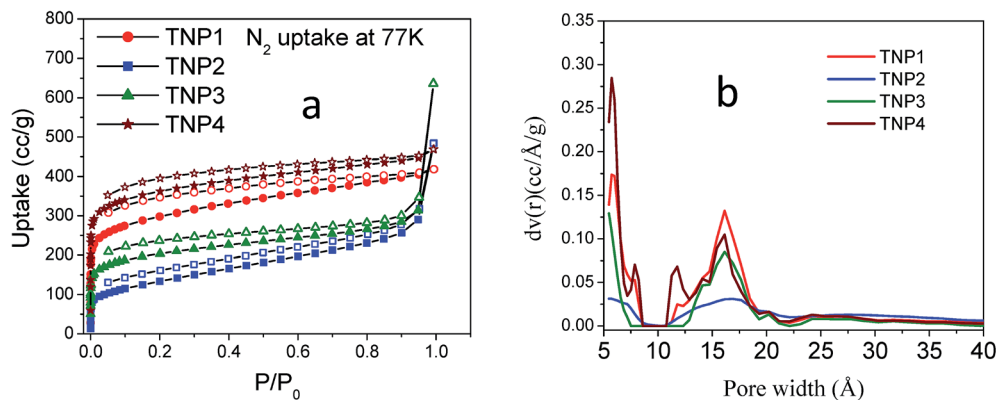


Fig. 4 N_2 adsorption isotherm of TNP1–TNP4 at 77 K (a); pore size distribution of TNP1–TNP4 (b).

The CO_2 sorption isotherms of **TNPs** were collected at 273 and 298 K and pressures up to 1 bar. The corresponding CO_2 isotherms register a steep rise in the initial low pressure region and are completely reversible in the entire region due to the absence of any significant adsorption–desorption hysteresis [Fig. 5a (273 K) and Fig. S5† (298 K)]. The reversibility suggests that interactions between **TNPs** and CO_2 are weak to the extent of regeneration of the polymers without application of heat.³² As observed in the case of N_2 sorption analysis, **TNPs** (1 and 4) derived from triethynyl comonomers show higher uptake of CO_2 relative to **TNPs** (2 and 3) derived from diethynyl comonomers. At 273 K and 1.0 bar, **TNP4** demonstrates the highest CO_2 uptake of 196 mg g^{-1} (4.45 mmol g^{-1}), while **TNP2** shows the lowest value of CO_2 uptake at 70 mg g^{-1} (1.59 mmol g^{-1}). Importantly, the magnitude of CO_2 uptake by these **TNPs** is comparable to or better than those of various literature reported COFs. For example, the uptake of **TNP4** (4.45 mmol g^{-1} at 273 K/1 bar) exceeds this value for –OH functionalized POFs (4.2 mmol g^{-1}),⁶⁷ imine linked ILP (1.97 mmol g^{-1}),²⁹ BLP1 (4.27 mmol g^{-1}),⁶⁵ and tetraphenyladamantane based PAN ($3.36\text{--}4.0 \text{ mmol g}^{-1}$).³⁷ In general, the CO_2 uptake values ($1.59\text{--}4.45 \text{ mmol g}^{-1}$ at 273 K/1 bar) of **TNPs** are comparable with those of various microporous organic polymers such as triazine based TBILP ($2.66\text{--}5.18 \text{ mmol g}^{-1}$),⁴² microporous covalent triazine polymers (MCTP, $3.65\text{--}4.64 \text{ mmol g}^{-1}$),⁴⁰ and carbazolic porous organic frameworks (Cz-POFs, $1.75\text{--}4.77 \text{ mmol g}^{-1}$).³⁸ Furthermore, the CO_2 uptake capacities of these **TNPs** are comparable to those of MOFs such as $Zn_2(C_2O_4)(C_2N_4H_3)_2 \cdot (H_2O)_{0.5}$ (4.35 mmol g^{-1})⁶⁸ and amine-functionalized MOFs such as bio-MOF-

11 (6.0 mmol g^{-1}).⁶⁹ The observed substantial uptake of CO_2 by these triptycene-based **TNPs** can be considered as a consequence of high internal molecular free volume (IMFV) due to the triptycene units and high nitrogen content due to the triazole motifs. Recently, theoretical studies have shown that 1,2,3-triazole molecules have a large CO_2 -binding energy (BE) (20.0 kJ mol^{-1}) possibly due to the high affinity of triazole rings for CO_2 via strong electrostatic interactions.⁵¹

To further understand the interaction of CO_2 with **TNPs**, the isosteric heats of adsorption (Q_{st}) for CO_2 in **TNPs** were estimated from the CO_2 adsorption isotherms at 273 and 298 K. The Q_{st} values for CO_2 at zero coverage (at the onset of adsorption) are in the range of $34.8\text{--}38.5 \text{ kJ mol}^{-1}$. Strong interactions between CO_2 and polar 1,2,3-triazole motifs as well as the narrow distribution in the microporous region may be the reason for the high Q_{st} value observed in the low pressure region. The Q_{st} values ($34.8\text{--}38.5 \text{ kJ mol}^{-1}$) of **TNPs** are higher than those of a variety of previously reported porous organic polymers such as nanoporous azo-linked polymers (ALPs: $27.9\text{--}29.6 \text{ kJ mol}^{-1}$),³² imine-linked porous polymer frameworks (PPFs: $21.8\text{--}29.2 \text{ kJ mol}^{-1}$),²⁴ benzimidazole-linked polymers (BLPs: $26.7\text{--}28.0 \text{ kJ mol}^{-1}$),^{25,65,66} nitrogen-rich networks (PECONFs: $26\text{--}34 \text{ kJ mol}^{-1}$)⁶⁴ nitrogen-rich diaminotriazine-based polymers (APOP, $26.6\text{--}33.3 \text{ kJ mol}^{-1}$),⁷⁰ and carbazolic porous organic frameworks (Cz-POFs, $24.8\text{--}27.8 \text{ kJ mol}^{-1}$).³⁸ However these values are slightly less than those of microporous covalent triazine polymers (MCTPs, $30.6\text{--}40 \text{ kJ mol}^{-1}$).⁴⁰

In addition to CO_2 , **TNPs** were also investigated as materials for the sorption of other gases such as H_2 and CH_4 (Table 2).

Table 1 Reaction yield and pore properties of TNP1–TNP4

Polymer	SA_{BET}^a ($\text{m}^2 \text{g}^{-1}$)	SA_{Lang}^b ($\text{m}^2 \text{g}^{-1}$)	V_{total}^c ($\text{cm}^3 \text{g}^{-1}$)	Yield (%)
TNP1	1090	1475	0.58	96
TNP2	460	756	0.53	88
TNP3	745	1006	0.62	85
TNP4	1348	1723	0.66	98

^a Surface area calculated based on the BET model from the nitrogen adsorption isotherm ($P/P_0 = 0.01\text{--}0.1$). ^b Surface area calculated based on the Langmuir model from the nitrogen adsorption isotherms ($P/P_0 = 0.05\text{--}0.35$). ^c The total pore volume calculated at $P/P_0 = 0.99$.



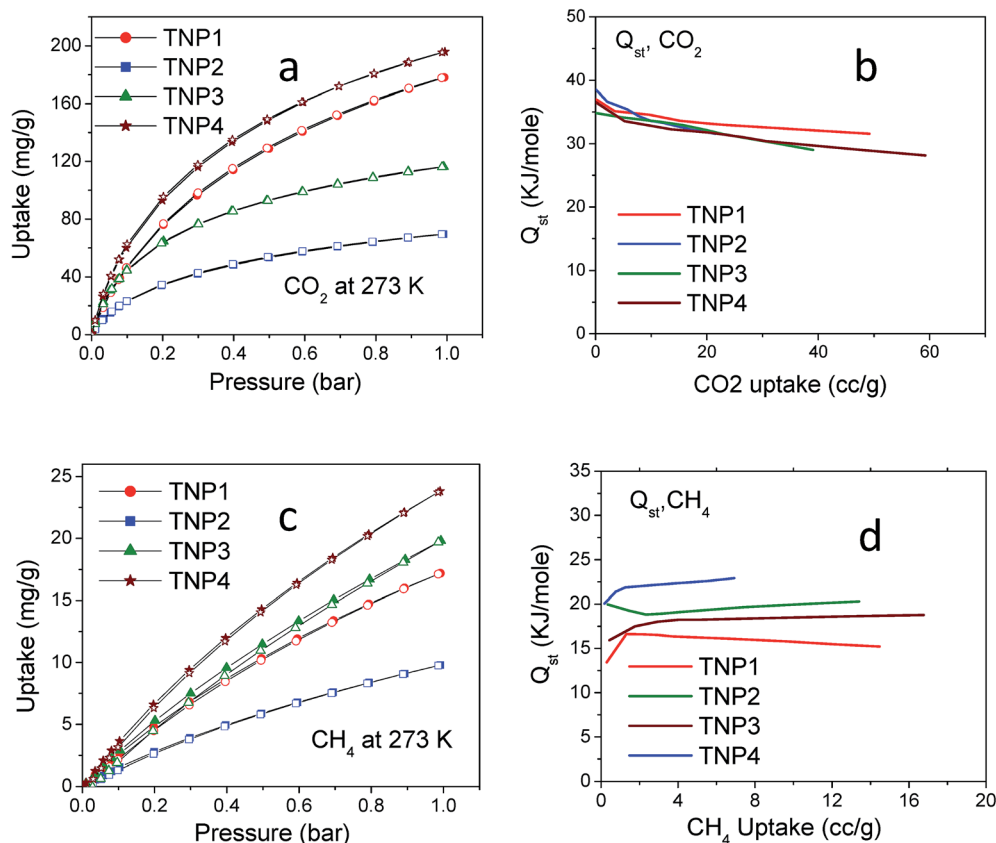


Fig. 5 CO₂ uptake isotherm of TNP1–TNP4 at 273 K (a), isosteric heat of adsorption (Q_{st}) of TNPs for CO₂ (b), methane uptake isotherm of TNP1–TNP4 at 273 K (c) and Q_{st} of TNPs for CH₄ (d). Adsorption (filled) and desorption (empty).

CH₄ gas isotherms of TNPs were collected at 273 K (Fig. 5c) and 298 K (Fig. S5†) up to 1 bar pressure. At 273 K, TNP4 registers the highest methane uptake (23.8 mg g⁻¹) while TNP2 records the lowest (10.0 mg g⁻¹). Irrespective of the extent of methane sorption, all the isotherms are completely reversible. We have also calculated the isosteric heat of adsorption (Q_{st}) for methane in TNPs and values at zero coverage are found to be in the range of 13.2–20.0 kJ mol⁻¹. These magnitudes of Q_{st} are comparable with those of various hetero-functionalized porous organic polymers.^{24,25,27,32}

Similarly, H₂ adsorption isotherms were collected at 77 K for TNPs (Fig. 6). The uptake capacity of H₂ increases in the order of TNP2 < TNP3 < TNP1 < TNP4 which is consistent with the increasing order of surface area ($S_{A_{BET}}$). The H₂ uptakes of TNPs

are in the range of 0.8–1.8 wt% at 77 K and 1 bar, which is comparable with those of various organic porous polymers, such as nanoporous azo-linked polymers (ALPs, 1.39–2.19 wt%)³² and benzimidazole-linked polymers (BILP1, 1.9 wt%),⁶⁵ under identical experimental conditions.

Gas selectivity

After evaluation of the porosity and gas-uptake capacities of these TNPs, we were interested to study their ability to selectively capture CO₂ over CH₄ and N₂. It is well known that in order to reduce environmental pollution due to CO₂ (a greenhouse gas) present in flue gas (N₂/CO₂: 85 : 15), it is necessary to selectively capture CO₂ over N₂. Also it is well

Table 2 H₂, CO₂, CH₄, and N₂ uptakes, isosteric heats of adsorption (Q_{st}) for TNPs, and selectivity for CO₂/N₂ and CO₂/CH₄

Polymer	H ₂ at 1 bar (77 K) (mg g ⁻¹)	CO ₂ at 1 bar (mg g ⁻¹)			CH ₄ at 1 bar (mg g ⁻¹)			N ₂ at 1 bar (mg g ⁻¹)		Selectivity	
		273 K	298 K	Q_{st}	273 K	298 K	Q_{st}	273 K	298 K	CO ₂ /N ₂ 273 K(298 K)	CO ₂ /CH ₄ 273 K(298 K)
TNP1	14	178	99	37.0	17.3	11.5	13.2	3.4	2.1	36(23)	8(5)
TNP2	8	70	43	38.5	10.0	5.2	20	2.3	1.1	40(23)	8(5)
TNP3	11	116	81	34.8	19.8	10.9	16.2	3.8	2.1	48(31)	9(6)
TNP4	18	196	127	36.5	23.8	14.2	20	7.3	3.2	31(27)	8(6)



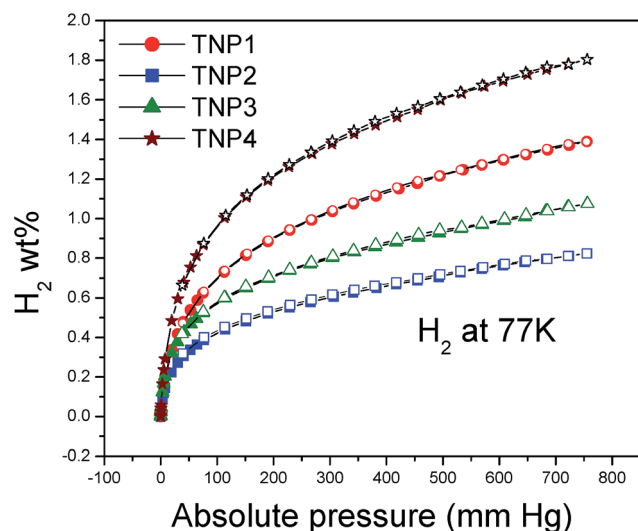


Fig. 6 H_2 uptake isotherms of TNP1–TNP4 at 77 K. Adsorption (filled) and desorption (empty).

accepted that CO_2 is a contaminant in natural gas (CH_4/CO_2 : 95 : 5) and it is necessary to selectively capture CO_2 over methane in order to improve the quality of natural gas as a fuel and also control corrosion in pipelines due to CO_2 .

In order to access the performance of TNPs in CO_2/CH_4 and CO_2/N_2 gas selectivities, it is required to record the single component adsorption isotherms of these gases at 273 K (Fig. 7) and 298 K (Fig. S6†) up to 1 bar pressure. For a given TNP, an initial steeper rise in the CO_2 adsorption isotherm in comparison to that in the case of the corresponding N_2 or CH_4 isotherm was anticipated, considering the higher Q_{st} values of TNPs for CO_2 in zero surface coverage (Table 2) due to the favorable interactions between CO_2 and N_2 centers present in the 1,2,3-triazole moieties. It must be mentioned here that in flue gas, CO_2 partial pressure is typically 0.15 bar at 273 K.²⁵ Thus for a given porous material, it is worth considering its relative uptake of gases (CO_2 vs. N_2 or CH_4) at 0.15 bar while evaluating its performance in selective gas adsorption. In the case of TNPs, at 0.15 bar the adsorption isotherms indicate significantly higher CO_2 uptake relative to N_2 or CH_4 . Considering this fact, these TNPs can be considered as useful materials for selective CO_2 gas adsorption applications under these conditions.

Subsequently, CO_2/N_2 and CO_2/CH_4 gas selectivity of these TNPs was estimated from the initial slope ratios using Henry's law constants for single-component adsorption isotherms (Fig. S7 and S8†). This method has been routinely used to study gas selectivity properties of a wide variety of organic and inorganic hybrid materials.^{25,32}

Therefore, the N_2/CO_2 selectivity of TNPs has been calculated from the isotherms recorded at 273 K and 298 K in the pressure

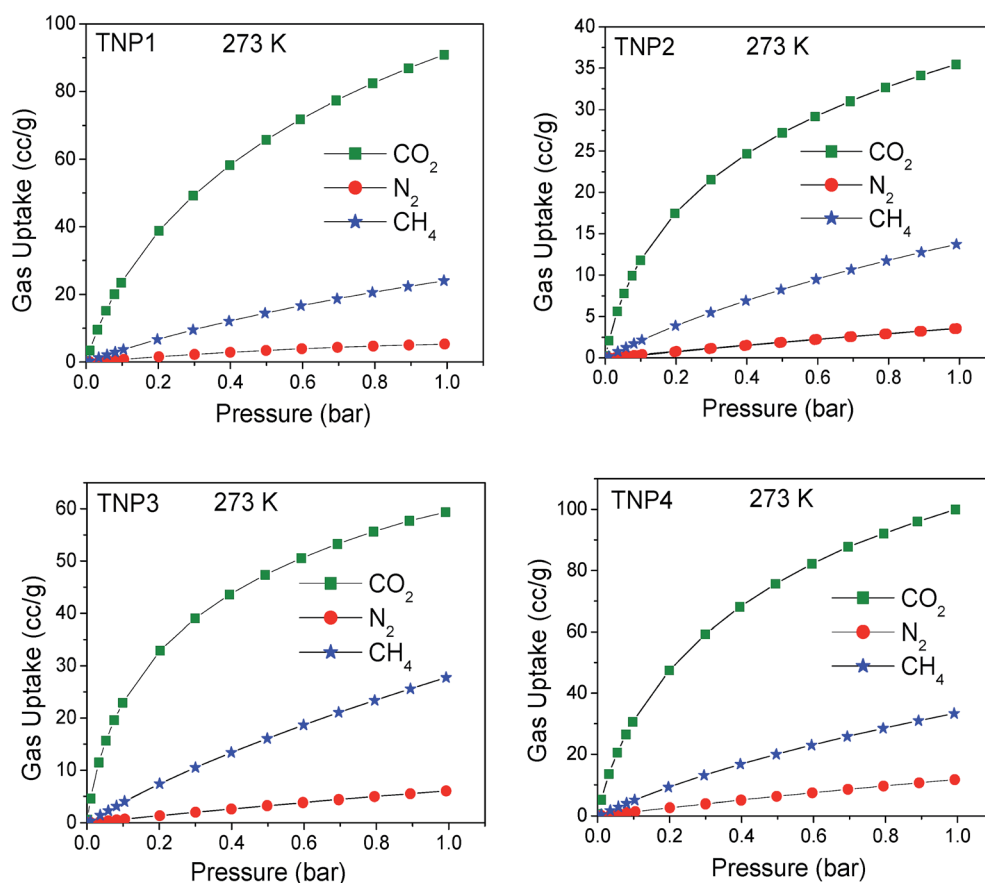


Fig. 7 Gas uptake capacities for TNP1–TNP4 at 273 K. CO_2 (green squares), CH_4 (blue stars), and N_2 (red circles).



range below 0.15 bar, and the results are summarized in Table 2. Overall, the selectivity at 273 K (31–48) is higher than the selectivity at 298 K (23–31) for all **TNPs**. Amongst these four network polymers, **TNP3** shows the highest selectivity for CO₂ over N₂ (48 at 273 K). This may be attributed to the presence of additional pyrazine motifs in **TNP3** that are absent in the other three **TNPs**. This polymer (**TNP3**) offers more nitrogen sites for interaction with CO₂ in comparison to the other **TNPs**. The data also suggest a considerable trade-off between gas storage capacity and gas selectivity amongst the four **TNPs** reported herein. For example, **TNP4** having the highest surface area ($S_{\text{BET}} = 1348 \text{ m}^2 \text{ g}^{-1}$) exhibits less selectivity for CO₂ over N₂ (31 at 273 K) in comparison to **TNP1** ($S_{\text{BET}} = 1090 \text{ m}^2 \text{ g}^{-1}$, CO₂/N₂ = 36 at 273 K) and **TNP2** ($S_{\text{BET}} = 460 \text{ m}^2 \text{ g}^{-1}$, CO₂/N₂ = 40 at 273 K). In general, the **TNP** with a relatively higher surface area exhibits relatively lower CO₂/N₂ selectivity. **TNP3** is an exception since it has higher nitrogen content than other **TNPs**. Thus our results corroborate with the general perception that a higher porosity level is accompanied by a compromise in CO₂/N₂ selectivity.^{25,32} Nevertheless, the CO₂/N₂ selectivities of these 1,2,3-triazole linked network polymers (**TNPs**) are better than those of a wide range of microporous materials reported in the literature.^{32,35–38,41,70}

Additionally, the CO₂/CH₄ selectivity of **TNPs** was also investigated to ascertain their applicability in the purification of natural gas (CH₄/CO₂: 95 : 5) by capture of CO₂. Methane adsorption isotherms were collected at 273 K and 298 K and the selectivity was determined using the initial slope calculation, and the results obtained are depicted in Table 2. CO₂/CH₄ selectivities at 273 K and 1 bar are in the range of 8–9. At a higher temperature (298 K), the CO₂/CH₄ selectivity was found to decrease to 5–6. In general, the **TNPs** exhibit significantly lower CO₂/CH₄ selectivity relative to CO₂/N₂ selectivity. It is reported that due to the higher polarizability of methane relative to N₂, the adsorption potential of methane is much higher than that of N₂ and this may be the explanation for the observed lower CO₂/CH₄ selectivities in comparison to CO₂/N₂ selectivities.³²

Conclusions

In conclusion, facile synthesis and characterization of a series of triptycene based and 1,2,3-triazole laced alternating copolymers has been described. The resulting 1,2,3-triazole linked network polymers (**TNPs**) are microporous materials exhibiting BET surface area up to 1348 m² g⁻¹. We have also shown that incorporation of nitrogen-rich pyrazine moieties results in enhancement of CO₂/N₂ gas selectivity relative to **TNPs** that do not possess this hetero-aromatic structural motif in the polymer backbone. In general, we have shown that integration of triptycene and 1,2,3-triazole motifs in the backbone of organic network polymers results in the yield of materials that demonstrate excellent performance in the storage of small gases at low pressure. Considering the microporosity, large gas uptake and high selectivities of CO₂ over N₂ and CH₄, **TNPs** are potential candidates for practical applications such as CO₂ capture as well as purification of flue and natural gas. Currently,

we are extending our research in the direction of a facile synthesis of novel microporous organic polymers with applications in the clean energy sector by utilizing new multifunctional monomers for tailoring the gas sorption properties.

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