PAPER

TiO$_2$ enhanced ultraviolet detection based on a graphene/Si Schottky diode†

Miao Zhu,[ab] Li Zhang,[a] Xinming Li,[c] Yijia He,[ab] Xiao Li,[ab] Fengmei Guo,[a] Xiaobei Zang,[a] Kunlin Wang,[a] Dan Xie,*[d] Xuanhua Li,[e] Bingqing Wei[ef] and Hongwei Zhu*[ab]

Graphene/Si has been proved to form a quality Schottky junction with high photoelectric conversion efficiency at AM 1.5. However, for the ultraviolet portion of the incident light, the photoelectric performance will degrade significantly due to severe absorption and recombination at the front surface. Herein, to realize enhanced ultraviolet detection with a graphene/Si diode, TiO$_2$ nanoparticles (NPs, 3–5 nm) are synthesized and spin-coated on the graphene surface to improve the photoresponse in the ultraviolet region. According to our results, the conversion efficiency of the graphene/Si diode at 420 nm and 350 nm increases by 72.7% and 100% respectively with TiO$_2$ coating. Then C^{-2}–V measurements of both TiO$_2$ and graphene/Si diode are performed to analyze the electronic band structure of the TiO$_2$/graphene/Si system, based on which we finally present the enhancement mechanism of photodetection using TiO$_2$ NPs.

Introduction

As a highly conductive and transparent material, graphene has been considered as a promising material in energy conversion.[1–4] Solar cells and photodetectors based on graphene/Si Schottky heterojunctions have been widely reported in previous studies as two types of typical photovoltaic devices based on graphene.[5–12] Such devices possess a very simple structure with graphene lying on the surface of an n-type silicon substrate. In such structures, the Richardson constant of Si can be significantly lowered due to the finite density of states of graphene and thus results in a low leakage current, which has been proved using the Landauer transport model.[13] Once the incident light arrives at the graphene/Si interface, carriers are excited and then separated by the built-in electric field to achieve the photoelectrical conversion process. A solar cell with such a structure

was first reported with an initial power conversion efficiency (PCE) of around 1.5%.[14] In the following five years, various approaches have been employed to improve its efficiency, including chemical doping,[15–17] structural design,[18–20] and electrolyte assistance.[21] In a recent study by our group, TiO$_2$ anti-reflection coating and acid doping were performed together to improve the PCE to 14.5% at AM 1.5, nearly ten times higher than the initially reported level.[22]

The approaches applied above are quite valuable to enhance the photoresponse properties of graphene/Si based photodetectors. However, being different from a solar cell, the photodetector mainly focuses on the response of light within a certain wavelength range. The photoelectric conversion from different light wavelengths can hardly be reflected by the investigation of solar cells. For example, ultraviolet will be strongly absorbed by the front surface of the device, resulting in severe recombination and thermal loss. Despite the fact that the PCE of the graphene/Si diode can be improved at AM 1.5, the ultraviolet response maybe still far lower than that in its optimal wavelength region. Thus it is meaningful to enhance the ultraviolet response of the graphene/Si diode to widen its effective working spectrum range.

Introducing a functional layer for light absorption and conversion is a widely used method to improve photovoltaic devices,[23–28] and it is especially suitable to enhance the photoresponse and PCE of ultraviolet for the graphene/Si diode according to the above analysis. TiO$_2$ is a common photocatalytic material with strong absorbance in the ultraviolet range.[29–32] When irradiated by ultraviolet, excitons generated in TiO$_2$ may be used to enhance the response of the graphene/Si diode *via* carrier injection. This is quite different from the case

[a]School of Materials Science and Engineering, State Key Laboratory of New Ceramics and Fine Processing, Tsinghua University, Beijing 100084, China. E-mail: hongweizhu@tsinghua.edu.cn

[b]Center for Nano and Micro Mechanics, Tsinghua University, Beijing 100084, China

[c]National Center for Nanoscience and Technology, Zhongguancun, Beijing 100190, China

[d]Tsinghua National Laboratory for Information Science and Technology (TNList), Institute of Microelectronics, Tsinghua University, Beijing 100084, China. E-mail: xiedan@tsinghua.edu.cn

[e]School of Materials Science and Engineering, Northwestern Polytechnical University, Xi'an 710072, China

[f]Department of Mechanical Engineering, University of Delaware, Newark, DE 19716, USA

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of long-wavelength light which is mainly based on the optical antireflection of TiO$_2$. In this work, the ultraviolet detection of the graphene/Si diode is greatly improved by simply spin-coating a layer of TiO$_2$ nanoparticles (NPs) on the surface of the device. The performances of the device before and after TiO$_2$ coating, including the current density–voltage (J–V) characteristics, responsivity, quantum efficiency, *etc.*, were compared to show the effects of the TiO$_2$ layer. The band diagram of the TiO$_2$/graphene/Si system was then deduced from the C^{-2}–V characteristics and finally a possible enhancement mechanism of the TiO$_2$ layer was presented.

Experimental

Synthesis of TiO$_2$ NPs

TiO$_2$ was synthesized by a nonhydrolytic sol–gel approach described as follows: after stirring a solution of TiCl$_4$ (1 mL), ethanol (5 mL), and benzyl alcohol (35 mL) for 6 h at 80 °C, it was washed with diethyl ether three times. A white TiO$_2$ precipitate was obtained by centrifuging the crude product. The final TiO$_2$ solution was prepared by dispersing it in ethanol.

Preparation of graphene

Multilayer graphene (MLG) was prepared by an atmospheric chemical vapor deposition method. Copper foil (25 μm, 99.8%, purchased from Alfa Aesar Chemical Co. Ltd.) was placed in a tube furnace for use as the substrate. Under the protection of Ar (200 mL min^{-1}) and H$_2$ (30 mL min^{-1}), the furnace was heated to 1000 °C in 50 min and further maintained for 30 min. Methane (20 mL min^{-1}) was then fed in as the carbon source. After 15 min reaction, the copper substrate was quickly moved away from the heating zone and cooled down to room temperature. The copper substrate was then immersed in 0.5 M FeCl$_3$ and graphene was successively transferred to deionized water and rinsed thoroughly to clean the residual FeCl$_3$ and other impurities.

Fabrication of the device

MLG was directly transferred onto an n-type silicon substrate with a SiO$_2$ layer along its surface borders (window area is ~0.1 cm^2). After drying in air, TiO$_2$ NPs dispersed in ethanol (0.2 mg mL^{-1}) were spin-coated on the surface of graphene at 2000 rpm for 30 s, as illustrated in Fig. 1a. Silver paste and the deposited Ti/Au layer were used as the top electrode and back electrode of the device, respectively. A typical photograph of the device (without electrodes) is shown in Fig. 1b.

C^{-2}–V measurement

C^{-2}–V measurement for TiO$_2$ NPs was performed with a Potentiostat (CHI 660D) using a three-electrode system. The Ag/AgCl and platinum electrodes were used as the reference and counter electrodes, respectively. TiO$_2$ NPs were spin-coated on an FTO coated glass to fabricate the working electrode. H$_2$SO$_4$ aqueous solution (0.5 M) was used as the electrolyte. The applied AC potential was 1 kHz with an amplitude of 0.01 V. Electrode potentials were converted to NHE using the
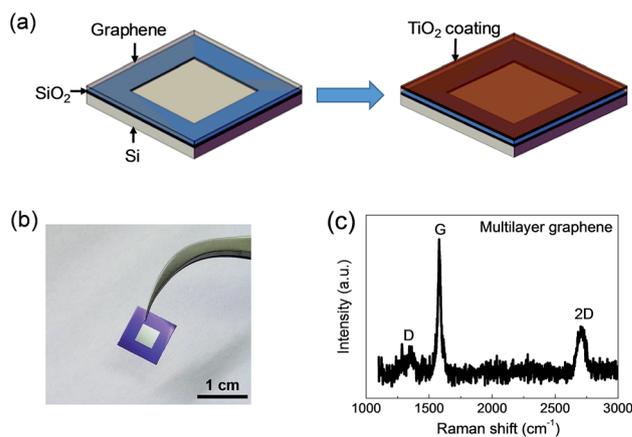

Fig. 1 (a) Schematic of the device fabrication. (b) Photograph of the device. (c) Raman spectrum of the multilayer graphene. $I_{2D}/I_G \approx 0.34$.

relationship $V_{NHE} = V_{Ag/AgCl} + 0.197$ V. For the MLG/Si diode, both the reference and counter electrodes were connected with the top electrode of the device, and the working electrode was connected with its back electrode. The frequency and amplitude of the applied AC voltage are 5 kHz and 0.01 V, respectively.

Characterization

Raman spectra were recorded on a HORIBA LabRAM HR Evolution with a 514 nm laser source. Scanning Electron Microscopy (SEM) was performed on a Zeiss Merlin compact. Transmission Electron Microscopy (TEM) was performed on a JEM-2010 TEM. The UV-Vis spectrum of graphene was measured using an Agilent Cary 5000. Atomic Force Microscopy (AFM) was tested using an Agilent N9451A. The thickness of the TiO$_2$ layer was measured by KLA Tencor Surface Profilometry. The J–V and on/off characteristics were recorded on a Keithley 2601. The X-ray photoelectron spectrum (XPS) was collected on an XPS 250XI.

Results and discussion

In order to enhance the ultraviolet response of the MLG/Si diode, a TiO$_2$ layer was coated on the surface of graphene to utilize the light energy irradiated on the device more efficiently. In the TiO$_2$/MLG/Si sandwich system, graphene functioned not only as the functional layer to form Schottky junctions with both the Si substrate and TiO$_2$ layer, but also acted as the conductive layer through which the carriers transport to the external circuit. Fig. S1† shows that the transmittance of the MLG was about 86.9% (at 550 nm). According to the 2.3% linear optical absorption of monolayer graphene to white light,[33] it could be deduced that as-synthesized MLG was about 6-layered, which was consistent with its Raman spectrum ($I_{2D}/I_G \approx 0.34$, Fig. 1c).[34,35] The metallic impurities in graphene may significantly influence the diode behavior of the MLG/Si device.[13] The XPS study was performed to investigate the metallic impurities in as-prepared graphene samples. The results (Fig. S2†) show a quite low atomic percentage of Fe and Cu, the most possible

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metallic impurities introduced into graphene during preparation and transfer processes.

The morphology of TiO_2 NPs has a great influence on their improvement effects. Poorly dispersed particles and thick bulk agglomeration will severely affect the carrier diffusion and separation, as well as reduce the light energy arriving at the MLG/Si interface. The TEM image (Fig. 2a) reveals that the TiO_2 NPs were well-dispersed with an average size of 3–5 nm. The high resolution TEM (HRTEM) image further shows the lattice fringes of TiO_2 NPs. The inter-planar distance of 0.24 nm and 0.35 nm correspond to the (004) and (101) planes of anatase TiO_2. After being coated on MLG, the TiO_2 NPs were uniformly spread on the MLG surface without an obvious discontinuous region such as cracks or holes, as shown in the SEM image (Fig. 2b).

AFM was further performed to characterize the morphology of the TiO_2 layer, as shown in Fig. 2c. It can be observed that the TiO_2 NPs are self-assembled into spherical agglomerations with larger sizes. Even so, the distribution of the agglomerated TiO_2 NPs still remains uniform in general. The thickness of the TiO_2 layer was measured by Surface Profilometry, and the results (Fig. 2d) showed that the average thickness of the TiO_2 layer was about 100 nm.

The J–V relationship is the most important and widely used characteristic for diode performance. J–V characteristics of the devices were measured under a beam of 420 nm incident light with the intensity of 0.16 W cm^{-2} (Fig. 3a). It can be deduced that the initial photoelectrical conversion efficiency of the device is only 0.44%. After being coated with TiO_2 NPs, the PCE increased to 0.76%, nearly 1.7 times higher than that of the original value. The open circuit voltage (V_{oc}) of the device was nearly stable, but the short circuit current density (J_{sc}) increased apparently, which contributed much to the increase of PCE. Acid doping has been reported to be another efficient way to improve the PCE of the MLG/Si device.[7,9] The TiO_2 coated device was then exposed to HNO_3 (analytical reagent) vapor for about
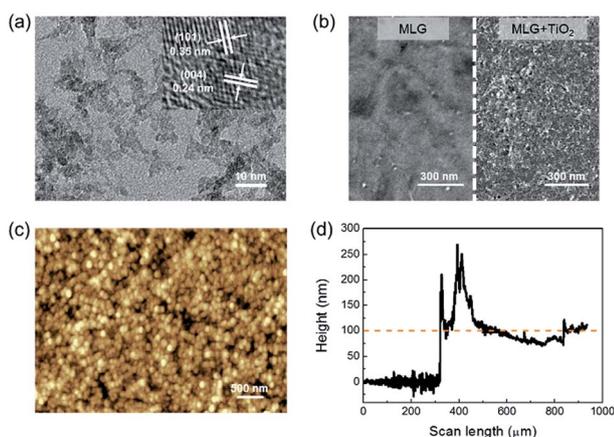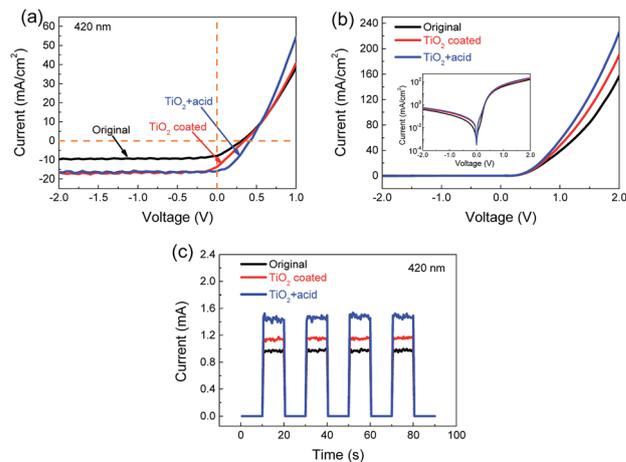

Fig. 3 (a) Light and (b) dark current–voltage characteristics of the device. The inset shows corresponding log-linear plots. (c) On/off characteristics of the device tested at 420 nm incident light.

30 s. The blue curve in Fig. 3a shows that the acid doping still worked efficiently in improving the PCE after being coated with TiO_2 NPs. The probable reason was that the HNO_3 vapor could permeate through the channels formed between TiO_2 NP agglomerations to arrive at the surface of graphene and then cause doping effects. Being different from the only TiO_2 NP coating, it can be seen from Fig. 3a that the increase of V_{oc} and fill factor (FF) contributed the most part of the PCE improvement in the case of acid doping. This indicated that the enhancement mechanism of TiO_2 NPs was different from the doping effect of HNO_3. The obvious increase of J_{sc} after TiO_2 coating was probably because that the TiO_2 NPs could provide extra excited carriers when ultraviolet irradiated on the surface. This was also consistent with the band gap of TiO_2 (about 3.0–3.2 eV, corresponding to the wavelength of 388–413 nm).[36,37] The band gap of nanoparticles was relevant to their sizes and crystal structures.[38,39] The actual excited wavelength limit might fall in a wider range around 413 nm. As an example for the light with higher energy, J–V characteristics of the device at 350 nm incident light (2.21 W cm^{-2}) were also tested (Fig. S3†). The PCE of the device increased by 100% (from 0.065% to 0.13%) after TiO_2 coating, higher than the case at 420 nm. The dark J–V characteristics (Fig. 3b) show that neither TiO_2 coating nor acid doping would significantly affect the rectification behavior of the MLG/Si diode.

Response current is another important parameter of a photodetector because it is usually used directly as the response or output signal of the device. On/off characteristic measurements were performed and the results are shown in Fig. 3c. Thanks to the improvement of PCE, the response current (I_{resp}) at 420 nm increased from 0.97 to 1.15 mA, and finally reached 1.47 mA after acid doping. The TiO_2 coated device shows excellent repeatability of switching behavior (Fig. S4†) in 10 min. The response and recovery time are measured to be 1.30 ms and 2.76 ms, respectively (Fig. S5†), which are comparable to the values of graphene/Si photodetectors reported previously.[5] In addition, the thickness effect of the TiO_2 layer on device

Fig. 2 (a) TEM image of TiO_2 NPs. The mean size of TiO_2 NPs is about 3–5 nm. The inset shows a typical HRTEM image of TiO_2 NPs. (b) SEM images of MLG before (left) and after (right) TiO_2 coating. (c) AFM image of the TiO_2 layer. (d) Height changes from MLG to the TiO_2 layer. The thickness of the TiO_2 layer is about 100 nm.

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performance has been discussed in Fig. S6.† The performance tested under different conditions is summarized in Table 1 to clearly show the improvements and differences.

To investigate the behavior of the TiO_2 layer in a wider spectral range, the quantum efficiency was measured from 300 to 1100 nm. The results (Fig. 4) show that the quantum efficiency in the near-ultraviolet range was obviously improved after TiO_2 coating.

In order to illustrate more clearly, the relative change of quantum efficiency (ΔEQE) was calculated by ($EQE_{TiO_2\ coated}$ − $EQE_{original}$)/$EQE_{original}$ and is shown as the blue curve in Fig. 4, which can be divided into three typical areas (areas I to III) along with the decrease of the light wavelength. In area I, from 1100 nm to point A (570 nm), ΔEQE increased mildly from 0.01 to 0.12. The photon energy of the light in this area was too low to excite the electrons in TiO_2, so the enhancement mechanism was mainly possible to be optical antireflection. In area II, from point A to point B (410 nm), ΔEQE turned out to increase more quickly, indicating that parts of the TiO_2 NPs could be excited by the incident light to generate carriers, which contributed to the increase of ΔEQE. When the photon energy further increased (wavelength < point B), the carriers could be excited abundantly in the TiO_2 layer, resulting in a sharp increase of the ΔEQE. It is worth mentioning that light with too short wavelength may lose most of its energy in the TiO_2 layer and seldom photons can arrive at the MLG/Si interface, thus a peak value of ΔEQE may exist in area III. According to the above analysis, the quantum efficiency of the device in the whole range from 300 to 1100 nm can be improved continuously by TiO_2 coating based on both optical antireflection and carrier injection from the TiO_2 layer.

Similar to the MLG/Si side, the excited electron (e^-)–hole (h^+) pairs in the TiO_2 layer should be separated by a suitable built-in voltage. Thus the investigation of the band structure of the TiO_2/MLG/Si system is critical to support our above analysis. The band structure was investigated by a two-step method based on the C^{-2}–V measurements.

In detail, the band structures of TiO_2/MLG and MLG/Si were first measured individually and then combined to be a whole picture of the sandwich system, assuming that the weak influences between each other could be neglected. The measurements for TiO_2 NPs were based on a three-electrode system. TiO_2 NPs were spin-coated on an FTO coated glass and tested in 0.5 M H_2SO_4 aqueous solution. The results are shown in Fig. 5a. By fitting the linear region of the C^{-2}–V plots, the x-intercept (V_0) was found to be −0.136 V.

Then the experimental carrier concentration and flat-band potential of TiO_2 NPs were calculated from the Mott–Schottky equation:[40]
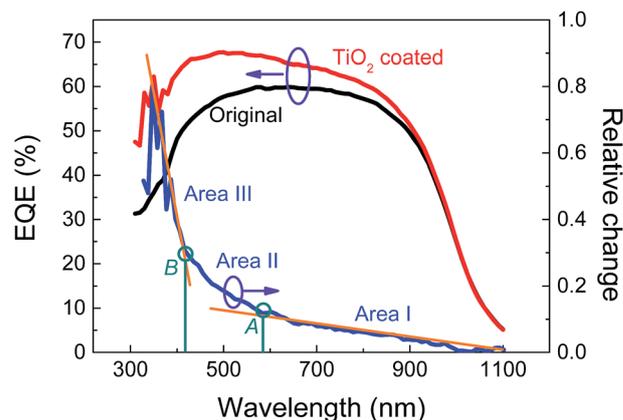

Fig. 4 External quantum efficiency (EQE) of the device before and after TiO_2 coating. The blue curve shows the relative changes.

$$\frac{1}{C^2} = \left(\frac{2}{\varepsilon_s q N_d A^2}\right)\left(V - V_{FB} - \frac{kT}{q}\right) \quad (1)$$

where C, ε_s, q, N_d, A, V, V_{FB}, k, and T are the space-charge capacitance, permittivity of the semiconductor ($48\varepsilon_0$ for TiO_2), electron charge, experimental carrier concentration, effective area, applied bias, flat-band potential, Boltzmann's constant and absolute temperature, respectively. The value of N_{d,TiO_2} was found to be 1.03×10^{20} cm^{-3} and the flat-band potential V_{FB,TiO_2} (vs. NHE) was calculated by $V_{FB,TiO_2} = V_0 - kT/q = -0.162$ V. As the V_{NHE} was about 4.5 V below the vacuum energy level, which was consistent with the work function of graphene (~4.5 eV),[41,42] the V_{FB} (vs. graphene) was numerically identical to the V_{FB} (vs. NHE). Then, the width of the depletion layer was given by

$$W_{sc,\ TiO_2} = \sqrt{\frac{2\varepsilon_s}{q N_d} V_{bi}} \quad (2)$$

The calculated W_{sc,TiO_2} was 2.89 nm. The potential change along the thickness direction of the depletion layer was also confirmed by

$$V(x) = \frac{q N_d}{2\varepsilon_s}(x + W_{sc})^2 \quad (3)$$

Then the band structure of TiO_2/graphene was deduced as shown in the left part of Fig. 6a. To measure the C^{-2}–V characteristics of the MLG/Si side, both the reference electrode and counter electrode were connected with MLG, while the working electrode was connected with the back electrode of the device.

Table 1 Photoelectric performance of the MLG/Si diode tested at 420 nm (0.16 W cm^{-2})

Structure	V_{oc} (mV)	J_{sc} (mA cm^{-2})	FF (%)	PCE_{UV} (%)	I_{resp} (mA)	Responsivity (mA/W)
Original	300	7.11	32.78	0.44	0.97	60.6
TiO_2 coated	300	12.02	33.87	0.76	1.15	71.9
TiO_2 + acid	400	15.12	40.25	1.52	1.47	91.9

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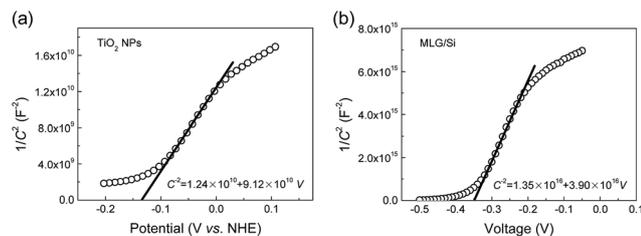
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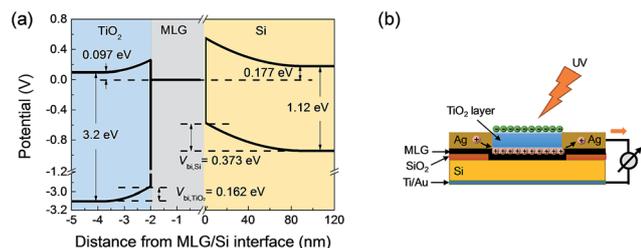**Fig. 5** C^{-2}–V characteristics of (a) TiO$_2$ NPs and (b) MLG/Si Schottky diode.

Fig. 6 (a) Band diagram of the TiO$_2$/MLG/Si system. (b) Schematic of the performance enhancement mechanism by TiO$_2$ coating.

In the same way, $N_{d,Si}$, $V_{FB,Si}$ and $W_{sc,Si}$ were calculated to be 3.08 $\times$ 10^{16} cm^{-3}, −0.373 V and 125.5 nm, respectively. The $V(x)$ in the MLG/Si side was calculated by eqn (3). Therefore, the band diagram of the whole TiO$_2$/MLG/Si system was deduced and is shown in Fig. 6a.

It can be seen that there are two built-in voltages existing in the TiO$_2$/MLG/Si system: one lies at the TiO$_2$/MLG interface and another lies at the MLG/Si interface. Once the incident light arrived at the TiO$_2$ layer, photons with energy higher than the band gap of TiO$_2$ were absorbed to excite the electrons of TiO$_2$ from its valance band to the conduction band, then the photo-generated electron–hole pairs were separated by the built-in voltage. The photo-generated holes (h$^+$) can be transported towards graphene together with the current generated from the MLG/Si side to the external circuit. However, as TiO$_2$ is a wide band gap dielectric material with poor conductivity, the photo-generated electrons (e$^-$) have to remain in the TiO$_2$ layer through capacitive coupling with the holes (Fig. 6b), which is similar to the model presented before for a graphene-quantum dot phototransistor.[43] Despite the fact that the carrier concentration of TiO$_2$ was much high than that of silicon, its enhancement effect might be limited by such mechanisms, because too much electrons accumulated on the surface of the TiO$_2$ layer would enhance their recombination with the holes.

Conclusions

In summary, TiO$_2$ NPs were synthesized to enhance the ultraviolet photoelectrical conversion of the MLG/Si diode. After being coated by TiO$_2$ NPs, the PCE of the device obviously improved (increased by 72.7% and 100% at 420 nm and 350 nm respectively). The EQE characteristics revealed that the enhancement mechanism of the TiO$_2$ layer in the ultraviolet

region was different from the long-wave region. Next, the band diagram of the TiO$_2$/MLG/Si system was established using the Mott–Schottky theory according to its C^{-2}–V characteristics, which confirmed that there existed two reciprocal Schottky junctions in the sandwich system. The enhancement mechanism was finally attributed to the carrier injection from the TiO$_2$ layer to the loop current of graphene/semiconductor based on the electronic band structure.

Acknowledgements

Notes and references

1 F. Bonaccorso, L. Colombo, G. Yu, M. Stoller, V. Tozzini, A. C. Ferrari, R. S. Ruoff and V. Pellegrini, *Science*, 2015, **347**, 1246501.
2 L. Dai, *Acc. Chem. Res.*, 2013, **46**, 31.
3 I. V. Lightcap and P. V. Kamat, *Acc. Chem. Res.*, 2013, **46**, 2235.
4 H. Wang, H. Feng and J. Li, *Small*, 2014, **10**, 2165.
5 X. An, F. Liu, Y. J. Jung and S. Kar, *Nano Lett.*, 2013, **13**, 909.
6 P. Lv, X. Zhang, X. Zhang, W. Deng and J. Jie, *IEEE Electron Device Lett.*, 2013, **34**, 1337.
7 J. Kim, S. S. Joo, K. W. Lee, J. H. Kim, D. H. Shin, S. Kim and S. Choi, *ACS Appl. Mater. Interfaces*, 2014, **6**, 20880.
8 X. Gan, R. Shiue, Y. Gao, I. Meric, T. F. Heinz, K. Shepard, J. Hone, S. Assefa and D. Englund, *Nat. Photonics*, 2013, **7**, 883.
9 X. Wang, Z. Cheng, K. Xu, H. K. Tsang and J. Xu, *Nat. Photonics*, 2013, **7**, 888.
10 A. Pospischil, M. Humer, M. M. Furchi, D. Bachmann, R. Guider, T. Fromherz and T. Mueller, *Nat. Photonics*, 2013, **7**, 892.
11 M. Amirmazlaghani, F. Raissi, O. Habibpour, J. Vukusic and J. Stake, *IEEE J. Quantum Electron.*, 2013, **49**, 589.
12 C. Liu, Y. Chang, T. B. Norris and Z. Zhong, *Nat. Nanotechnol.*, 2014, **9**, 273.
13 D. Sinha and J. U. Lee, *Nano Lett.*, 2014, **14**, 4660.
14 X. Li, H. Zhu, K. Wang, A. Cao, J. Wei, C. Li, Y. Jia, Z. Li, X. Li and D. Wu, *Adv. Mater.*, 2010, **22**, 2743.
15 X. Miao, S. Tonga, M. K. Petterson, K. Berke, A. G. Rinzler, B. R. Appleton and A. F. Hebard, *Nano Lett.*, 2012, **12**, 2745.
16 T. Cui, R. Lv, Z. Huang, S. Chen, Z. Zhang, X. Gan, Y. Jia, X. Li, K. Wang, D. Wu and F. Kang, *J. Mater. Chem. A*, 2013, **1**, 5736.
17 H. Kim, K. Lee, N. McEvoy, C. Yim and G. S. Duesberg, *Nano Lett.*, 2013, **13**, 2182.
18 G. Fan, H. Zhu, K. Wang, J. Wei, X. Li, Q. Shu, N. Guo and D. Wu, *ACS Appl. Mater. Interfaces*, 2011, **3**, 721.

This journal is © The Royal Society of Chemistry 2015

J. Mater. Chem. A, 2015, **3**, 8133–8138 | **8137**

19 L. Yang, X. Yu, M. Xu, H. Chen and D. Yang, *J. Mater. Chem. A*, 2014, **2**, 16877.

20 K. Jiao, X. Wang, Y. Wang and Y. Chen, *J. Mater. Chem. C*, 2014, **2**, 7715.

21 X. Li, X. Zang, X. Li, M. Zhu, Q. Chen, K. Wang, M. Zhong, J. Wei, D. Wu and H. Zhu, *Adv. Energy Mater.*, 2014, **4**, 1400224.

22 E. Shi, H. Li, L. Yang, L. Zhang, Z. Li, P. Li, Y. Shang, S. Wu, X. Li, J. Wei, K. Wang, H. Zhu, D. Wu, Y. Fang and A. Cao, *Nano Lett.*, 2013, **13**, 1776.

23 D. H. Lee, J. Y. Kwon, S. Maldonado, A. Tuteja and A. Boukai, *Nano Lett.*, 2014, **14**, 1961.

24 C. Li, C. Chueh, H. Yip, K. M. O'Malley, W. Chen and A. K.-Y. Jen, *J. Mater. Chem.*, 2012, **22**, 8574.

25 A. Shalav, B. S. Richards and M. A. Green, *Sol. Energy Mater. Sol. Cells*, 2007, **91**, 829.

26 Z. Su, L. Wang, Y. Li, G. Zhang, H. Zhao, H. Yang, Y. Ma, B. Chu and W. Li, *ACS Appl. Mater. Interfaces*, 2013, **5**, 12847.

27 Z. Wu, T. Song, Z. Xia, H. Wei and B. Sun, *Nanotechnology*, 2013, **24**, 484012.

28 M. Xu, X. Zhu, X. Shi, J. Liang, Y. Jin, Z. Wang and L. Liao, *ACS Appl. Mater. Interfaces*, 2013, **5**, 2935.

29 J. Schneider, M. Matsuoka, M. Takeuchi, J. Zhang, Y. Horiuchi, M. Anpo and D. W. Bahnemann, *Chem. Rev.*, 2014, **114**, 9919.

30 S. G. Kumar and L. G. Devi, *J. Phys. Chem. A*, 2011, **115**, 13211.

31 I. Paramasivam, H. Jha, N. Liu and P. Schmuki, *Small*, 2012, **8**, 3073.

32 K. Nakata and A. Fujishima, *J. Photochem. Photobiol., C*, 2012, **13**, 169.

33 G. Pirruccio, L. M. Moreno, G. Lozano and J. G. Rivas, *ACS Nano*, 2013, **7**, 4810.

34 V. T. Nguyen, H. D. Le, V. C. Nguyen, T. T. T. Ngo, D. Q. Le, X. N. Nguyen and N. M. Phan, *Adv. Nat. Sci.: Nanosci. Nanotechnol.*, 2013, **4**, 035012.

35 W. Wu, Z. Liu, L. A. Jauregui, Q. Yu, R. Pillai, H. Cao, J. Bao, Y. P. Chen and S. Pei, *Sens. Actuators, B*, 2010, **150**, 296.

36 S. G. Yenchalwar, V. K. Azhagan and M. V. Shelke, *Phys. Chem. Chem. Phys.*, 2014, **16**, 17786.

37 H. Ariga, T. Taniike, H. Morikawa, M. Tada, B. K. Min, K. Watanabe, Y. Matsumoto, S. Ikeda, K. Saiki and Y. Iwasaw, *J. Am. Chem. Soc.*, 2009, **131**, 14670.

38 D. Segets, J. M. Lucas, R. N. K. Taylor, M. Scheele, H. Zheng, A. P. Alivisatos and W. Peukert, *ACS Nano*, 2012, **6**, 9021.

39 L. Li, J. Hu, W. Yang and A. P. Alivisatos, *Nano Lett.*, 2001, **1**, 349.

40 K. Cheng, C. Huang, G. Pan, P. Chen, T. Lee and T. C. K. Yang, *Mater. Chem. Phys.*, 2008, **108**, 16.

41 J. Kim, J. H. Hwang, J. Suh, S. Tongay, S. Kwon, C. C. Hwang, J. Wu and J. Y. Par, *Appl. Phys. Lett.*, 2013, **103**, 171604.

42 K. Xu, C. Zeng, Q. Zhang, R. Yan, P. Ye, K. Wang, A. C. Seabaugh, H. G. Xing, J. S. Suehle, C. A. Richter, D. J. Gundlach and N. V. Nguyen, *Nano Lett.*, 2013, **13**, 131.

43 G. Konstantatos, M. Badioli, L. Gaudreau, J. Osmond, M. Bernechea, F. P. G. de Arquer, F. Gatti and F. H. L. Koppens, *Nat. Nanotechnol.*, 2012, **7**, 363.