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click for updatesCite this: *J. Mater. Chem. A*, 2015, 3, 6509Room temperature sequential ionic deposition (SID) of Ag<sub>2</sub>S nanoparticles on TiO<sub>2</sub> hierarchical spheres for enhanced catalytic efficiency†Wei Li Ong,<sup>a</sup> Yee-Fun Lim,<sup>b</sup> June Lay Ting Ong<sup>b</sup> and Ghim Wei Ho<sup>\*ab</sup>

Porous TiO<sub>2</sub> hierarchical spheres with high surface area synthesized *via* a solvothermal method were successfully modified with an Ag<sub>2</sub>S co-catalyst by a sequential ionic deposition (SID) method at room temperature. The presence of Ag<sub>2</sub>S facilitated efficient charge separation, thus reducing recombination and enhancing the photocatalytic activity of the photocatalyst. The enhanced photocatalytic performance was demonstrated by water splitting where hydrogen (H<sub>2</sub>) gas was produced at an evolution rate of 708 μmol h<sup>-1</sup> g<sup>-1</sup> and methyl orange was degraded with a rate constant of 0.018 min<sup>-1</sup>. This is the first time that photocatalytic water splitting using a suspension system has been demonstrated on a Ag<sub>2</sub>S/TiO<sub>2</sub> hierarchical heterostructure and the material shows stability in its photocatalytic performance despite being recycled several times. The composite material presents properties which are highly promising for the generation of clean energy and environmental clean up applications.

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## Introduction

There is currently a pressing demand around the globe for photocatalysts capable of generating H<sub>2</sub> as a form of clean and renewable energy source, and also responding to environmental cleaning needs (*e.g.* wastewater treatment). Titanium(IV) dioxide (TiO<sub>2</sub>) has always been a material highly regarded for its photocatalytic properties on top of being cheap, non-toxic and environmentally friendly.<sup>1</sup> Degussa P25 may have demonstrated a decent performance in photocatalysis,<sup>2</sup> but hierarchical structures such as hierarchical spheres have shown an enhanced performance in photocatalytic H<sub>2</sub> production due to their high surface area, and also the ability to entrap incident radiation due to the light scattering effect.<sup>3</sup> Moreover, such hierarchical spheres possess the advantages of short diffusion length for photocatalytic reactions, and high porosity and stability offered by the robust 3D architecture.<sup>4</sup> However, the performance is still far from satisfactory when compared to other photocatalytic materials and the two main reasons are the wide band gap of TiO<sub>2</sub> and high recombination rate of the photogenerated electrons and holes.<sup>5</sup> To enhance the photocatalytic performance, co-catalysts can be loaded on the TiO<sub>2</sub>

hierarchical structures to solve both problems. Besides loading nanoparticles of expensive noble metals,<sup>6</sup> metal-oxides and sulfides, which are equally if not more effective, can also be used as cheaper alternatives.<sup>7,8</sup> In recent years, the development of sulfides as both the core photocatalytic material as well as co-catalysts has attracted increasing amounts of attention.<sup>9–11</sup> Silver sulfide (Ag<sub>2</sub>S), having a direct band gap of 1.0 eV,<sup>12</sup> is a potential candidate as a co-catalyst of TiO<sub>2</sub>. Its conduction band (−0.3 eV) is less anodic than the corresponding TiO<sub>2</sub> band (−0.1 eV), and its valence band (+0.7 eV) is more cathodic than the TiO<sub>2</sub> valence band (+3.1 eV).<sup>13</sup> In addition, α-Ag<sub>2</sub>S possesses negligible toxicity compared to other commonly used narrow band gap materials.<sup>14</sup> Several studies have reported Ag<sub>2</sub>S loading on TiO<sub>2</sub>, however, most of them only demonstrated its use in the photocatalytic degradation of dyes and organic materials,<sup>15,16</sup> or its use in photoelectrochemical water splitting.<sup>17–20</sup> To the best of our knowledge, no paper has reported or investigated its use in photocatalytic water splitting in a photocatalyst suspension system.

One way to synthesize Ag<sub>2</sub>S nanoparticles is *via* the SID method which is a simple fabrication methodology that can be carried out at room temperature and normal atmospheric pressure. The overall process is not only cost-effective, it also does not include any toxic by-product making it environmentally friendly, and the density of nanoparticles deposited can be easily controlled by merely varying the deposition cycles. A typical process involves immersing the host photocatalytic material into silver nitrate (AgNO<sub>3</sub>) and thiourea (SC(NH<sub>2</sub>)<sub>2</sub>) solutions successively for as many cycles as desired to achieve a uniform deposition of Ag<sub>2</sub>S nanoparticles on the host material

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with good heterojunction. The good interface will promote the transfer of charge carriers between the host photocatalytic material and the co-catalysts, hence producing excellent photocatalytic performance.

In this paper,  $\text{Ag}_2\text{S}$  is synthesized as a co-catalyst of  $\text{TiO}_2$  hierarchical spheres using the SID method. The number of loading cycles is varied to investigate the various loading densities and derive the optimal density of  $\text{Ag}_2\text{S}$  nanoparticles for photocatalytic water splitting. The best sample was recycled 5 times to demonstrate its reusability and stability. The performance of the  $\text{Ag}_2\text{S}/\text{TiO}_2$  composite in the degradation of organic pollutants was also demonstrated *via* the photodegradation of methyl orange (MO).

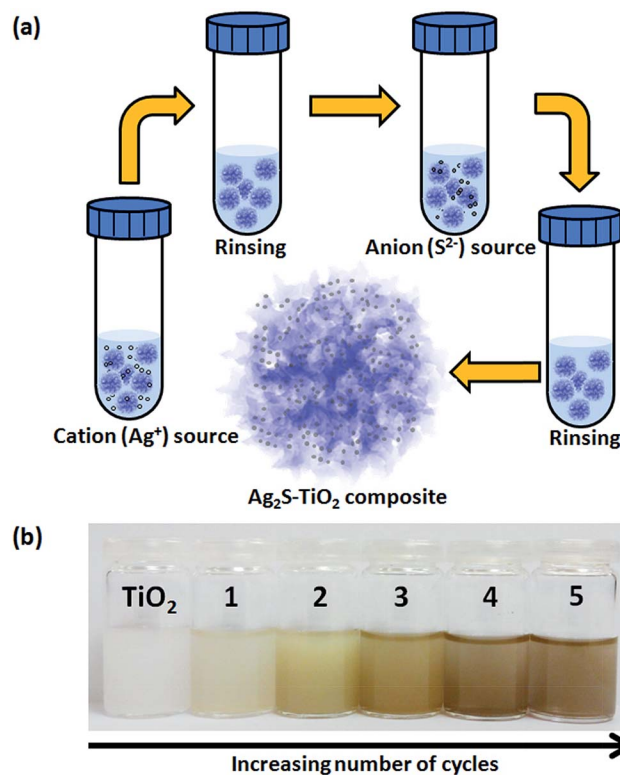
## Experimental

### Synthesis of $\text{TiO}_2$ hierarchical spheres

$\text{TiO}_2$  hierarchical spheres were synthesized by a solvothermal method following Wu *et al.*'s work.<sup>4</sup> 6 ml of dimethylformamide (DMF, Alfa Aesar) was mixed with 20 ml of isopropanol (IPA, HPLC grade, Tokyo Chemical) under stirring. 1 ml of titanium *n*-butoxide (TBT, Sigma Aldrich) was then added to the mixture and stirred. The mixture was transferred to a Teflon-lined autoclave and heated at 200 °C for 3 h. The synthesized product was washed and centrifuged 3 times with ethanol. After drying at 90 °C for 3 h, the  $\text{TiO}_2$  hierarchical spheres were annealed at 450 °C for 2 hours at a ramp rate of 5 °C min<sup>-1</sup> to obtain anatase  $\text{TiO}_2$  hierarchical spheres.

### Loading of $\text{Ag}_2\text{S}$ nanoparticles

The  $\text{Ag}_2\text{S}$  nanoparticles were loaded on the  $\text{TiO}_2$  hierarchical spheres at room temperature using the SID method.  $\text{AgNO}_3$  serves as the precursor for  $\text{Ag}^+$  ions, while  $\text{SC}(\text{NH}_2)_2$  is the precursor for  $\text{S}^{2-}$  ions.  $\text{AgNO}_3$  was dissolved in ethanol to produce a solution of 0.05 M concentration while  $\text{SC}(\text{NH}_2)_2$  was dissolved in methanol to produce a 0.05 M solution. Ethanol and methanol were used instead of water as solvents because of their lower surface tension and high wetting ability which promotes the superior penetration ability of the reacting solution deeply into the pores, resulting in uniform assembly of  $\text{Ag}_2\text{S}$  nanoparticles on the  $\text{TiO}_2$  hierarchical spheres. The  $\text{TiO}_2$  hierarchical spheres were dispersed in the  $\text{AgNO}_3$  solution and magnetically stirred for 1 min as shown in Scheme 1(a). The hierarchical spheres were then centrifuged and washed with ethanol to remove loosely bound and excess precursor solution, before placing into the  $\text{SC}(\text{NH}_2)_2$  solution and stirring for 3 min. The rinsing step also helps to prevent homogeneous precipitation of  $\text{Ag}_2\text{S}$  in the solution. The hierarchical spheres were then centrifuged and washed with methanol. This entire process constitutes 1 cycle of  $\text{Ag}_2\text{S}$  loading. The  $\text{TiO}_2$  hierarchical spheres were loaded with  $\text{Ag}_2\text{S}$  through 1 to 5 loading cycles, and the samples are denoted as  $n\text{-Ag}_2\text{S}/\text{TiO}_2$ , where *n* represents the number of loading cycles. The photocatalyst became darker in colour as the number of cycles increased (Scheme 1(b)). The samples were then dried at 60 °C for 2 hours to complete the loading process.



**Scheme 1** (a) Schematic illustration of the SID process in each cycle of  $\text{Ag}_2\text{S}$  deposition and (b) changes in the color of the composite with an increasing number of cycles.

### Photocatalytic testing (water splitting and MO degradation)

The photocatalytic activity of the composites was investigated by dispersing the photocatalysts in a methanol–water solution and illuminating with simulated solar light. 2 mg of  $n\text{-Ag}_2\text{S}/\text{TiO}_2$  was dispersed in 10 ml of 10% v/v methanol–water solution by sonication in a cylindrical quartz reaction cell (volume 25 ml) which was sealed with a rubber septum. The cell was purged with argon (Ar) gas for 10 min and then irradiated using a 300 W xenon arc lamp (intensity 100 mW cm<sup>-2</sup>) with magnetic stirring. Gas samples were extracted using a 100  $\mu\text{l}$  gas tight syringe and analyzed with a gas chromatograph (GC2010). For the photodegradation of MO, 15 mg of photocatalyst was dispersed in 15 ml of 0.015 mM aqueous MO solution in the cylindrical quartz reaction cell by sonication. The mixture was then magnetically stirred in the dark for 30 min to attain complete adsorption–desorption equilibrium, before being illuminated using a 300 W xenon arc lamp. The concentration of MO was determined using a UV-vis-NIR spectrophotometer and the maximum absorbance peak value at 462.5 nm was used to plot the amount of MO degraded and thus, determine the photodegradation activity of the composite. Photocatalytic experiments under visible irradiation were carried out with a cut-off filter ( $\lambda > 400$  nm). After completion of the photocatalytic experiments, the reaction mixtures were centrifuged to separate the photocatalysts from the reaction mixture and disposed properly.



## Material characterisation

The morphology of the  $n\text{-Ag}_2\text{S}/\text{TiO}_2$  composite was characterized by using a scanning electron microscope (SEM, JEOL FEG JSM 7001F) operating at 15 kV. The crystal structure of the composite was investigated by using a transmission electron microscope (TEM, Philips FEG CM300) operating at 200 kV and an X-ray diffractometer (XRD, D5005 Bruker X-ray diffractometer equipped with graphite-monochromated Cu K $\alpha$  radiation at  $\lambda = 1.541 \text{ \AA}$ ). Elemental composition was studied using energy-dispersive X-ray spectroscopy (EDX, Oxford Instruments) while the valence states of the various elements were determined through X-ray photoelectron spectroscopy (XPS). A UV-vis-NIR spectrophotometer (UV-vis, Shimadzu UV-3600) was used to measure the absorbance characteristics of the composites and MO solution. The gas samples from the photocatalytic experiments were analyzed by gas chromatography (Shimadzu GC2010 TCD) to determine the amount of  $\text{H}_2$  gas produced. Brunauer–Emmett–Teller (BET, Quantachrome Nova

1200) measurements were conducted with nitrogen ( $\text{N}_2$ ) as the adsorbate at liquid nitrogen temperature.

## Results and discussion

The  $\text{TiO}_2$  nanostructures initially obtained by the solvothermal synthesis method were layered protonated titanate hierarchical spheres (LTHSs) constructed from thin curved nanosheets.<sup>4</sup> After annealing at  $450^\circ\text{C}$  in air, the LTHSs were converted to porous titania hierarchical spheres (PTHs) which were assembled from small anatase crystallites as shown in Fig. 1(a) and (b). The disappearance of the nanosheets in PTHs could probably be attributed to the recrystallization of pristine LTHSs into the anatase phase and the sintering process that takes place in the relatively fragile ultrathin nanosheets. The structure of the PTHs could be observed more clearly in TEM (Fig. 1(c) and (d)). The clear lattice fringes observed in Fig. 1(d) indicate the crystallinity of the  $\text{TiO}_2$  hierarchical spheres, and

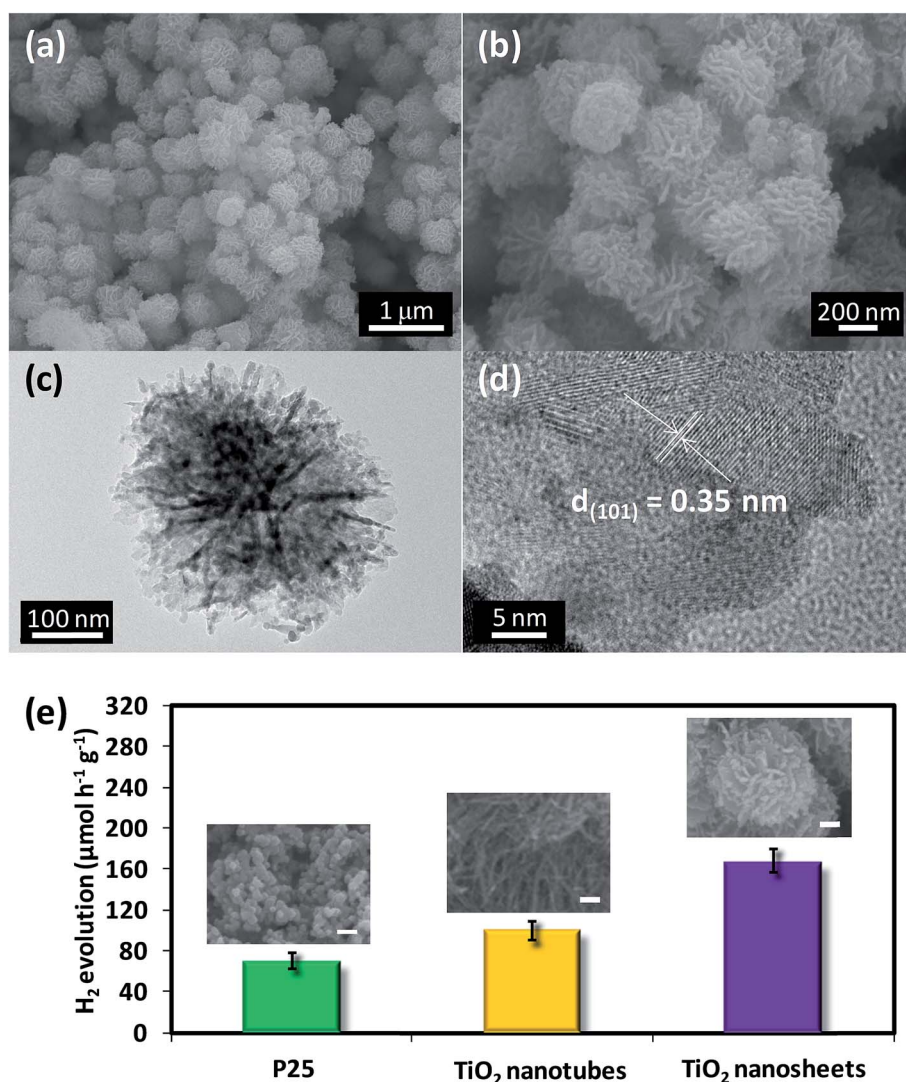


Fig. 1 (a) and (b) SEM images, (c) TEM and (d) HRTEM images of  $\text{TiO}_2$  hierarchical spheres. (e)  $\text{H}_2$  evolution rates of various types of  $\text{TiO}_2$  nanostructures. The insets show the SEM images of the various  $\text{TiO}_2$  nanostructures (scale bar indicates 100 nm).





the lattice spacing of 0.35 nm is in agreement with the values for the (101) lattice planes of tetragonal anatase TiO<sub>2</sub>.<sup>21</sup> The TiO<sub>2</sub> hierarchical spheres exhibit a better H<sub>2</sub> production performance when compared to commercial Degussa P25 and TiO<sub>2</sub> nanotubes (synthesized based on previous work)<sup>22</sup> as shown in Fig. 1(e). The SEM images in the insets of Fig. 1(e) show the morphologies of the various TiO<sub>2</sub> nanostructures. P25 has a H<sub>2</sub> evolution rate of 70  $\mu\text{mol h}^{-1} \text{g}^{-1}$  while TiO<sub>2</sub> nanotubes produced H<sub>2</sub> at a higher rate of 100  $\mu\text{mol h}^{-1} \text{g}^{-1}$ . The as-synthesized TiO<sub>2</sub> hierarchical spheres could produce H<sub>2</sub> at an improved rate of 167  $\mu\text{mol h}^{-1} \text{g}^{-1}$ . BET measurements of the 3 TiO<sub>2</sub> nanostructures revealed specific surface areas of 48, 154 and 71  $\text{m}^2 \text{g}^{-1}$  for the P25, TiO<sub>2</sub> nanotubes and TiO<sub>2</sub> hierarchical spheres respectively. Despite not having the highest surface area, TiO<sub>2</sub> hierarchical spheres still showed the highest H<sub>2</sub> evolution rate due to the faster electron transport rates and slower recombination rates, and also the considerable surface area which facilitates better access and diffusion of liquid and gaseous reactants, which is beneficial for the photocatalytic activity.<sup>23</sup> The hierarchical structure also favors the harvesting of light due to multiple scattering within the framework.<sup>24–26</sup> The photocatalytic performance of the TiO<sub>2</sub> hierarchical spheres can be further improved by loading Ag<sub>2</sub>S as a co-catalyst on it.

Ag<sub>2</sub>S nanoparticles were loaded onto the TiO<sub>2</sub> hierarchical spheres as co-catalysts by the SID method at room temperature. When the TiO<sub>2</sub> hierarchical spheres are dispersed in AgNO<sub>3</sub> solution and magnetically stirred, the Ag<sup>+</sup> ions are adsorbed on the surface of the TiO<sub>2</sub> crystallites. The loosely bound and excess Ag<sup>+</sup> ions were then removed by rinsing in ethanol. The Ag<sup>+</sup> coated TiO<sub>2</sub> hierarchical spheres were subsequently stirred in SC(NH<sub>2</sub>)<sub>2</sub> solution which provides the S<sup>2-</sup> ions to form Ag<sub>2</sub>S nanoparticles. The Ag<sup>+</sup> ions react with the S<sup>2-</sup> ions to form Ag<sub>2</sub>S nanoparticles on the surface of the TiO<sub>2</sub> hierarchical spheres. The density of Ag<sub>2</sub>S nanoparticles deposited on the surface of the TiO<sub>2</sub> hierarchical spheres can be varied by varying the number of loading cycles from 1 to 5. The EDX spectra in Fig. 2(a) shows the presence of Ag and S in 3-Ag<sub>2</sub>S/TiO<sub>2</sub>, indicating that Ag<sub>2</sub>S nanoparticles were successfully deposited. Prominent peaks of Ti L $\alpha_{1,2}$  at 0.45 keV, Ti K $\alpha_{1,2}$  at 4.51 keV and Ti K $\beta_{1,3}$  at 4.93 eV due to the TiO<sub>2</sub> hierarchical spheres were clearly observed in the spectra. Several peaks related to Ag and S were also detected: Ag L<sub>1</sub> at 2.63 keV, Ag L $\alpha_1$  at 2.98 keV, Ag L $\beta_1$  at 3.15 keV, S K $\alpha_{1,2}$  at 2.31 keV and S K $\beta_1$  at 2.46 keV. The Pt peaks originated from the sputtered Pt for enhancement of the image contrast. From the EDX spectra, it can be seen that the SID method is a reliable method to deposit Ag<sub>2</sub>S nanoparticles onto TiO<sub>2</sub> hierarchical spheres. The atomic% of Ag and S on the Ag<sub>2</sub>S/TiO<sub>2</sub> composites is presented in Table 1. The atomic% of Ag and S on the TiO<sub>2</sub> hierarchical spheres was found to increase as the number of loading cycles increases, indicating a higher density of Ag<sub>2</sub>S nanoparticles with more loading cycles. This observation is supported by the TEM images in Fig. 2(b)–(e) which show the distribution and density of Ag<sub>2</sub>S on the LTHSs with loading cycles of 1, 3 and 5 respectively at both low and high (insets of Fig. 2(b)–(e)) magnifications. From the TEM images, the density of Ag<sub>2</sub>S nanoparticles is observed to increase with an increasing number of loading cycles. It should

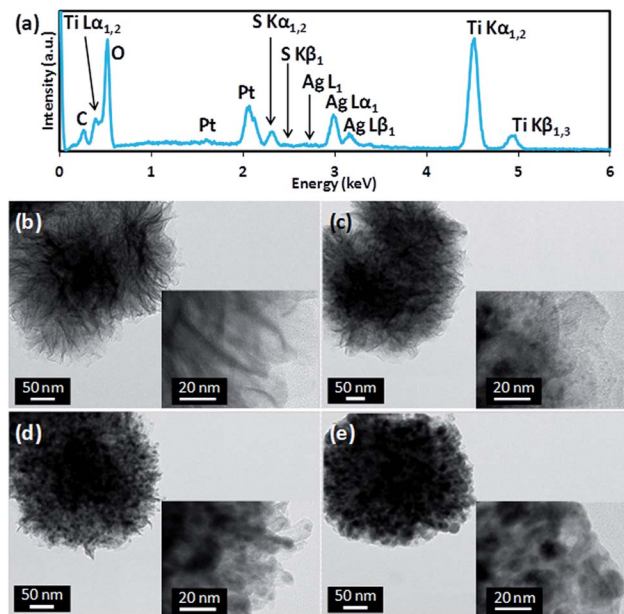


Fig. 2

Fig. 2 (a) EDX spectrum of 3-Ag<sub>2</sub>S/TiO<sub>2</sub>. (b)–(e) Low magnification TEM images of TiO<sub>2</sub>, 1-, 3- and 5-Ag<sub>2</sub>S/TiO<sub>2</sub> respectively. The insets show the high magnification TEM images of the corresponding composite.

be noted that the Ag<sub>2</sub>S nanoparticles were intentionally loaded on LTHSs instead of PTHSs solely for TEM imaging purposes. In doing so, the darker contrast of the Ag<sub>2</sub>S nanoparticles could be seen clearly against the lighter contrast of the nanosheets in the LTHSs. All other characterization and performance measurements were conducted using PTHSs loaded with Ag<sub>2</sub>S.

Fig. 3(a) shows a HRTEM image of 3-Ag<sub>2</sub>S/TiO<sub>2</sub> where the lattice planes of the anatase TiO<sub>2</sub> and the Ag<sub>2</sub>S can be clearly observed, indicating that the composite is highly crystalline and a good interface exists between Ag<sub>2</sub>S and TiO<sub>2</sub>. The diameter of the Ag<sub>2</sub>S nanoparticle was measured to be about 5 nm. A lattice spacing of 0.35 nm was determined by the fast Fourier transform (FFT) pattern (Fig. 3(b)) and could be attributed to the (101) plane of anatase TiO<sub>2</sub><sup>21</sup> while the *d*-spacing of 0.23 nm corresponds to the (−112) plane of Ag<sub>2</sub>S (Fig. 3(c)).<sup>27</sup> The crystalline structure of the composite is also shown by the XRD spectra in Fig. 3(d). The peaks at 25.45, 38.05, 48.25, 54.15, 55.15 and 62.95°, can be indexed to the (101), (004), (200), (105), (211)

Table 1 EDX elemental microanalysis of the *n*-Ag<sub>2</sub>S/TiO<sub>2</sub> composites

Sample name	Atomic (%)	
	Ag (%)	S (%)
1-Ag <sub>2</sub> S/TiO <sub>2</sub>	0.15	0.06
2-Ag <sub>2</sub> S/TiO <sub>2</sub>	0.87	0.41
3-Ag <sub>2</sub> S/TiO <sub>2</sub>	1.37	0.66
4-Ag <sub>2</sub> S/TiO <sub>2</sub>	1.95	0.98
5-Ag <sub>2</sub> S/TiO <sub>2</sub>	2.67	1.23



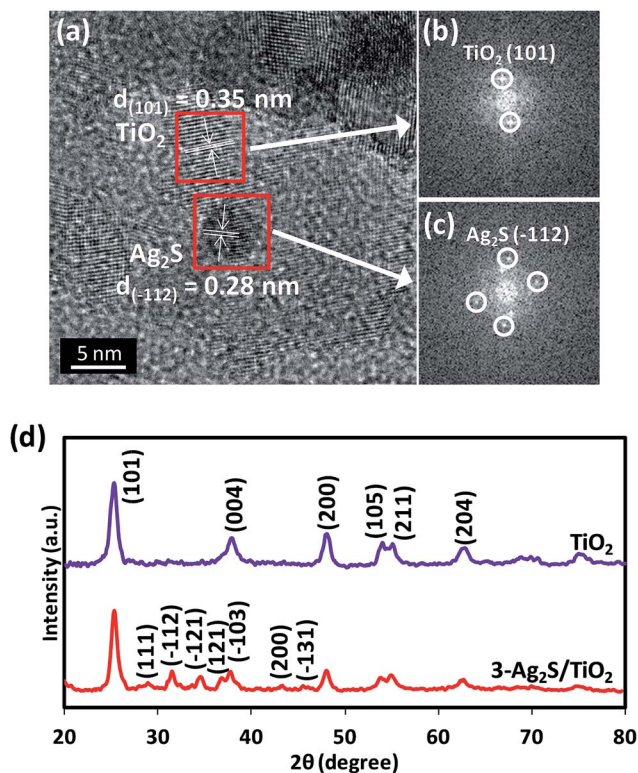


Fig. 3 (a) HRTEM image of 3-Ag<sub>2</sub>S/TiO<sub>2</sub>. FFT of (b) TiO<sub>2</sub> and (c) 3-Ag<sub>2</sub>S/TiO<sub>2</sub>. (d) XRD pattern of TiO<sub>2</sub> and 3-Ag<sub>2</sub>S/TiO<sub>2</sub>.

and (204) crystal planes of the anatase phase of TiO<sub>2</sub> (JCPDS card no. 21-1272) respectively, and these peaks are observed for both TiO<sub>2</sub> and 3-Ag<sub>2</sub>S/TiO<sub>2</sub>. The (111), (−112), (−121), (121), (−103), (200) and (−131) diffraction peaks at 28.97, 31.52, 34.38, 36.81, 37.72, 43.41 and 45.42° respectively can be indexed to monoclinic Ag<sub>2</sub>S (JCPDS card no. 14-0072).

XPS was also carried out to determine the valence states of the elements present in TiO<sub>2</sub> (not shown here) and the 3-Ag<sub>2</sub>S/TiO<sub>2</sub> composite (Fig. 4). Ti and O were detected in TiO<sub>2</sub> and the 3-Ag<sub>2</sub>S/TiO<sub>2</sub> composite as shown in Fig. 4(a) and (b). The Ti 2p<sub>3/2</sub> and 2p<sub>1/2</sub> peaks observed at 458.8 and 464.4 eV respectively indicate that Ti is present as Ti<sup>4+</sup> in the form of TiO<sub>2</sub>. The O 1s peak located at 529.7 eV is attributed to the Ti–O–Ti bond, supporting the presence of TiO<sub>2</sub>.<sup>28</sup> Ag and S were not detected in the XPS spectra of the TiO<sub>2</sub> hierarchical spheres but were present in the 3-Ag<sub>2</sub>S/TiO<sub>2</sub> composite. The Ag 3d<sub>5/2</sub> and 3d<sub>3/2</sub> peaks (Fig. 4(c)) were observed to be located at 368.2 and 374.2 eV respectively. The Ag 3d peaks located at these 2 energy levels indicate the presence of Ag in the state of Ag<sup>+</sup>. No metallic Ag was present in the composite. The S 2p peak (Fig. 4(d)) can be deconvoluted into S 2p<sub>3/2</sub> (159.6 eV) and 2p<sub>1/2</sub> (160.8 eV) which indicates that S is present as S<sup>2−</sup>. From the Ag 3d and S 2p peaks, it can be concluded that Ag and S are present as Ag<sub>2</sub>S.<sup>17</sup>

BET measurements were carried out on the TiO<sub>2</sub> hierarchical spheres and 3-Ag<sub>2</sub>S/TiO<sub>2</sub> to investigate the textural characteristics. The N<sub>2</sub> adsorption–desorption isotherms are shown in Fig. 5(a) and (b), and the insets show the corresponding pore size distribution. The isotherms can be categorized as type IV

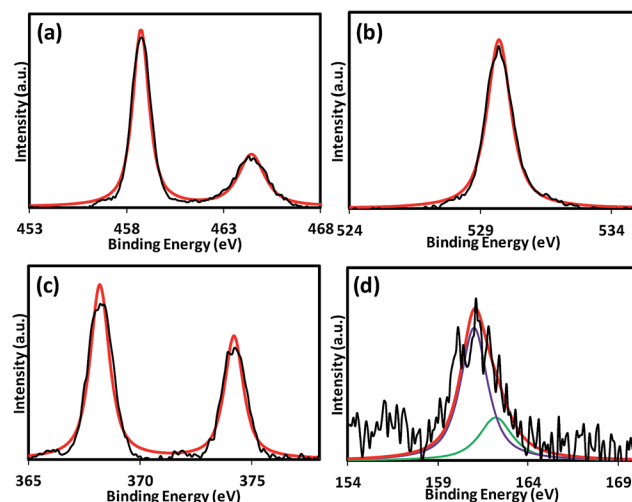


Fig. 4 XPS spectra of 3-Ag<sub>2</sub>S/TiO<sub>2</sub> composite. (a) Ti 2p, (b) O 1s, (c) Ag 3d and (d) S 2p.

with small hysteresis loops observed at a relative pressure of 0.4 to 0.9. The BET specific surface area obtained from the TiO<sub>2</sub> hierarchical spheres is 71 m<sup>2</sup> g<sup>−1</sup> with a total pore volume of 0.28 cm<sup>3</sup> g<sup>−1</sup> while that of 3-Ag<sub>2</sub>S/TiO<sub>2</sub> is 63 m<sup>2</sup> g<sup>−1</sup> with a total pore volume of 0.33 cm<sup>3</sup> g<sup>−1</sup>. The TiO<sub>2</sub> hierarchical spheres have pores with a diameter of 5.78 nm while the 3-Ag<sub>2</sub>S/TiO<sub>2</sub> has a larger pore size at 12.49 nm. The slight decrease in the surface area can be attributed to the deposition of Ag<sub>2</sub>S nanoparticles in the pores of the TiO<sub>2</sub> hierarchical spheres. This suggests that the photocatalytic activity is not affected by the slight decrease in the surface area of the composite, and the enhancement in performance is mainly due to the alignment of the energy levels and charge transfer at the Ag<sub>2</sub>S/TiO<sub>2</sub> interface. The absorption spectra of the *n*-Ag<sub>2</sub>S/TiO<sub>2</sub> composite are shown in Fig. 5(c). The TiO<sub>2</sub> hierarchical spheres exhibited an absorption band edge at 380 nm which is in accordance with its large band gap of 3.2 eV, and the absorption in the visible range was weak. However, after loading Ag<sub>2</sub>S, the absorption in the visible range increased without any observable shift in the band gap. The increase in absorption is attributed to the narrow band gap of Ag<sub>2</sub>S nanoparticles. With an increased absorption in the visible wavelength, more electron–hole pairs will be generated in the composite under UV-visible illumination.

The photocatalytic performance of the composites with different loading cycles for water splitting is shown in Fig. 6(a). After one loading cycle of Ag<sub>2</sub>S, the H<sub>2</sub> evolution rate improved from 167 to 412 μmol h<sup>−1</sup> g<sup>−1</sup>, demonstrating the effectiveness of Ag<sub>2</sub>S as a co-catalyst. The H<sub>2</sub> evolution rate continued to increase with more loading cycles and peaked at 708 μmol h<sup>−1</sup> g<sup>−1</sup> for 3 loading cycles but subsequently decreased for higher loading cycles, plummeting to 330 μmol h<sup>−1</sup> g<sup>−1</sup> for 5 loading cycles. Despite the drop in performance, the H<sub>2</sub> evolution rate is still higher than that of the pristine TiO<sub>2</sub> hierarchical spheres, indicating that the presence of Ag<sub>2</sub>S on TiO<sub>2</sub> as a co-catalyst will aid in improving the photocatalytic performance regardless of the amount. In this case, 3 loading cycles provided the highest enhancement to the H<sub>2</sub> evolution rate, and the H<sub>2</sub> evolution rate



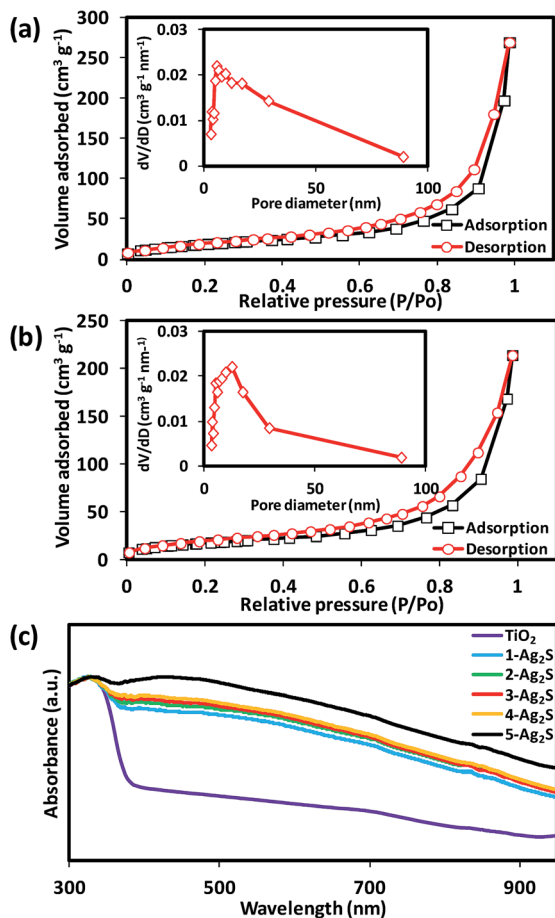
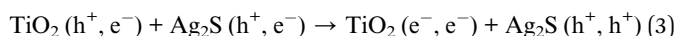
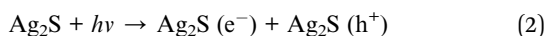
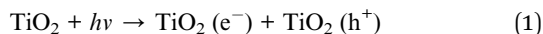


Fig. 5 BET of (a) TiO<sub>2</sub> and (b) 3-Ag<sub>2</sub>S/TiO<sub>2</sub>. The insets show the corresponding pore size distributions obtained from desorption isotherms. (c) Absorbance spectra of *n*-Ag<sub>2</sub>S/TiO<sub>2</sub> composites.

of 708  $\mu\text{mol h}^{-1} \text{g}^{-1}$  was about 4 times better than that of pristine TiO<sub>2</sub> hierarchical spheres. The improved performance is due to the charge transfer between Ag<sub>2</sub>S and TiO<sub>2</sub> as shown in Fig. 6(b) which favors the separation of photo-induced electron-hole pairs in the photosensitized TiO<sub>2</sub>. When the composite is illuminated with UV-visible light, the electrons generated are transferred from the conduction band (CB) of Ag<sub>2</sub>S to the CB of TiO<sub>2</sub>, whereas the holes are transferred from the valence band (VB) of TiO<sub>2</sub> to the VB of Ag<sub>2</sub>S (corresponding to eqn (1)–(3)).



This transfer process is thermodynamically more favorable because both the CB and VB of Ag<sub>2</sub>S lie above that of TiO<sub>2</sub>, and the electron-hole transfer process is faster than its recombination. The holes will react with methanol in a stepwise reaction producing H<sup>+</sup> and several intermediates in the process which were eventually oxidized to CO<sub>2</sub>, while the electrons reduce the H<sup>+</sup> ions to form H<sub>2</sub> according to eqn (4) and (5).<sup>29–31</sup>

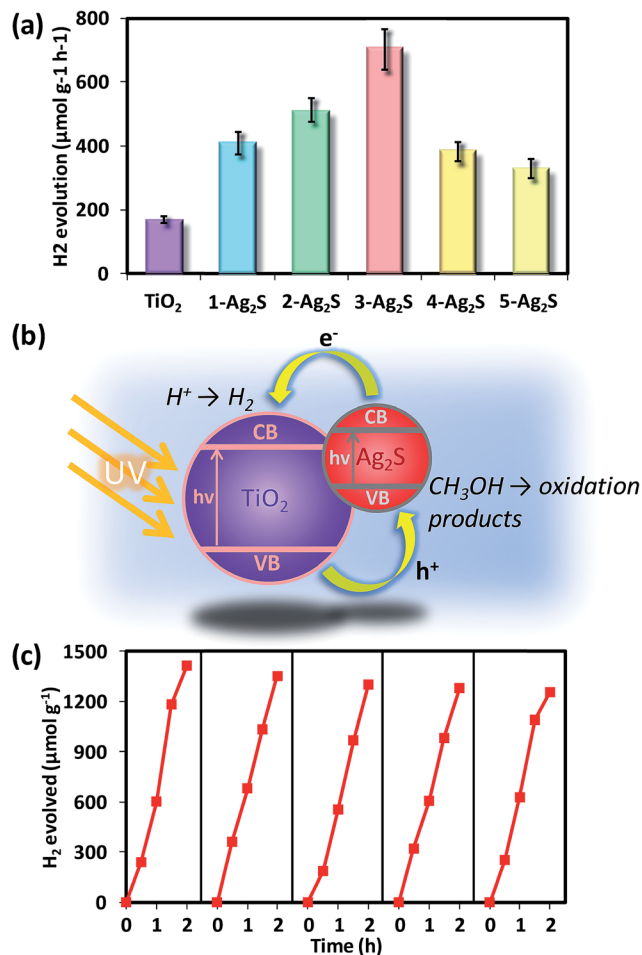
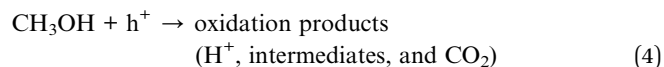


Fig. 6 (a) H<sub>2</sub> evolution rates of *n*-Ag<sub>2</sub>S/TiO<sub>2</sub> composites. (b) Schematic diagram of the charge-transfer processes between Ag<sub>2</sub>S and TiO<sub>2</sub> during water splitting. (c) H<sub>2</sub> production cycling of 3-Ag<sub>2</sub>S/TiO<sub>2</sub>.



When the number of loading cycles was increased to more than 3 times, the density of Ag<sub>2</sub>S nanoparticles populating the surface of the TiO<sub>2</sub> hierarchical spheres was too high causing conglomeration. This can be seen from the TEM image of 5-Ag<sub>2</sub>S/TiO<sub>2</sub> which appears much darker than the as-synthesized TiO<sub>2</sub> hierarchical spheres, 1-Ag<sub>2</sub>S/TiO<sub>2</sub> and 3-Ag<sub>2</sub>S/TiO<sub>2</sub> (Fig. 2). The darker contrast is attributed to the high density of Ag<sub>2</sub>S nanoparticles populating the TiO<sub>2</sub> hierarchical spheres. This will prevent the photo-induced electrons in Ag<sub>2</sub>S from fast injection into TiO<sub>2</sub>, resulting in inefficient separation of the electron-hole pairs and more occurrence of recombination. The high density of Ag<sub>2</sub>S nanoparticles will also block the light from TiO<sub>2</sub>, resulting in a fewer photogenerated electrons and holes for the generation of H<sub>2</sub>. These effects will result in a decline in the H<sub>2</sub> evolution rate.<sup>32</sup> Besides the H<sub>2</sub> evolution rate, another property of the photocatalyst that is of great importance is the stability and consistency of the photocatalytic performance over



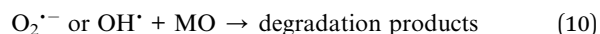
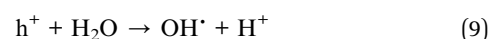
repeated use. To determine this, the 3-Ag<sub>2</sub>S/TiO<sub>2</sub> composite was tested for H<sub>2</sub> evolution repeatedly over 5 cycles (Fig. 6(c)). After each cycle, the photocatalyst was washed with DI water, re-dispersed in a newly prepared 10% v/v methanol-water solution, and purged with Ar gas before irradiating with the xenon arc lamp again. The H<sub>2</sub> evolution rate exhibited only a slight drop from 708 to 628 μmol h<sup>-1</sup> g<sup>-1</sup> after 5 cycles, proving that the photocatalyst is stable and can be recycled over repeated runs.

Besides photocatalytic water splitting, the *n*-Ag<sub>2</sub>S/TiO<sub>2</sub> composites can also be used for photodegradation of MO, which is akin to demonstrating its wastewater treatment capabilities. The degradation kinetics of MO was measured *via* the changes in their concentration, which was calculated from the absorbance peaks at 462.5 nm. From the plot in Fig. 7(a), the TiO<sub>2</sub> hierarchical spheres needed 180 min to fully degrade MO. After loading with Ag<sub>2</sub>S nanoparticles, the time needed to fully degrade MO was reduced to 150 min. The 3-Ag<sub>2</sub>S/TiO<sub>2</sub> composite performed the best and was able to degrade MO

completely in 120 min. The photodegradation activity was further analyzed by studying the pseudo-first order kinetics of the various photocatalysts as shown in Fig. 7(b). This quantitative analysis is derived using the pseudo-first order model<sup>33</sup> as follows:

$$\ln(C_0/C_t) = kt \quad (6)$$

where  $C_0$  and  $C_t$  are the concentrations of MO at time 0 and  $t$  respectively, and  $k$  is the pseudo-first order rate constant. The pseudo-first order rate constants,  $k$ , of the TiO<sub>2</sub> hierarchical spheres and *n*-Ag<sub>2</sub>S/TiO<sub>2</sub> composites are summarized in Table 2. The constant  $k$  of TiO<sub>2</sub> is the lowest at 0.011 min<sup>-1</sup>, but it increases with Ag<sub>2</sub>S loading. The 3-Ag<sub>2</sub>S/TiO<sub>2</sub> composite has the highest  $k$  constant at 0.018 min<sup>-1</sup>, indicating its enhanced dye degradation capability. The degradation results clearly show that the photodegradation of MO improved after loading Ag<sub>2</sub>S on TiO<sub>2</sub> hierarchical spheres, and the 3-Ag<sub>2</sub>S/TiO<sub>2</sub> composite produced the highest rate of photodegradation. This is most likely attributed to the optimal loading of Ag<sub>2</sub>S nanoparticles on TiO<sub>2</sub>, where the light harvesting ability is the best and the recombination rate is the lowest. The enhancement in photodegradation is also due to the charge transfer between Ag<sub>2</sub>S and TiO<sub>2</sub>, and the reduced recombination between the photo-generated electron-hole pairs (Fig. 7(c)). The generated electrons react with dissolved oxygen molecules and produce oxygen peroxide radicals (eqn (7)), while the positively charged holes (h<sup>+</sup>) react with OH<sup>-</sup> derived from H<sub>2</sub>O to form hydroxyl radicals OH<sup>•</sup> (eqn (8) and (9)). O<sub>2</sub><sup>•-</sup> and OH<sup>•</sup> are powerful oxidizing agents capable of degrading most pollutants.<sup>34,35</sup> The MO molecules can be photodegraded by the O<sub>2</sub><sup>•-</sup> and OH<sup>•</sup> radicals to CO<sub>2</sub>, H<sub>2</sub>O, or other mineralization products (eqn (10)).



The Ag<sub>2</sub>S/TiO<sub>2</sub> composite material is also capable of showing photocatalytic activity under visible illumination. 3-Ag<sub>2</sub>S/TiO<sub>2</sub> was used to photodegrade MO under visible light irradiation.

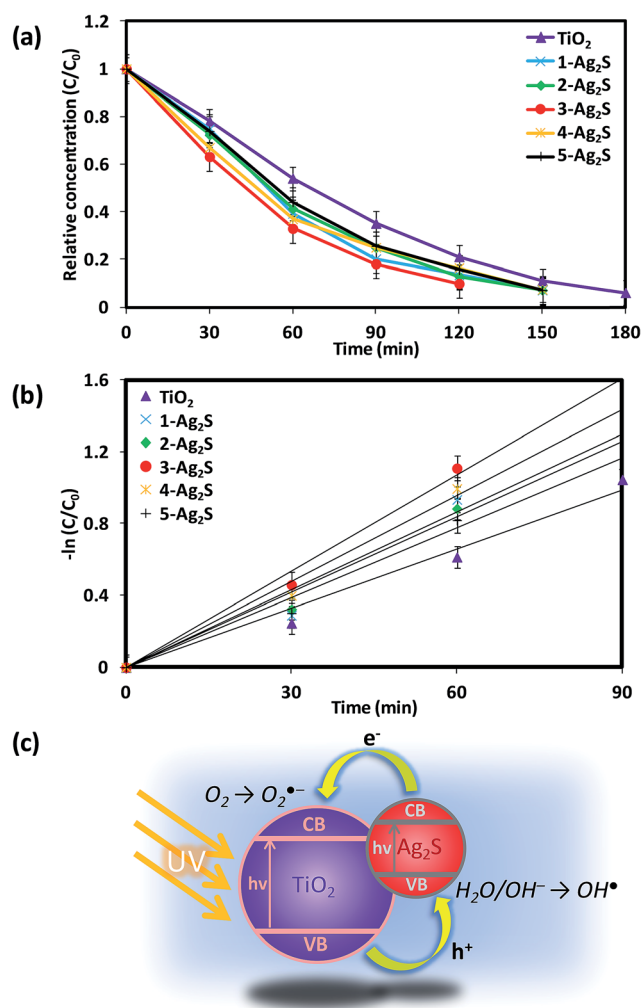


Fig. 7 (a) Degradation kinetics and (b) pseudo-first order kinetics of time evolution MO photodegradation study in the presence of *n*-Ag<sub>2</sub>S/TiO<sub>2</sub> composites. (c) Schematic diagram of the charge-transfer processes between Ag<sub>2</sub>S and TiO<sub>2</sub> during MO photodegradation.

Table 2 Pseudo-first order rate constants  $k$  of the *n*-Ag<sub>2</sub>S/TiO<sub>2</sub> composites

Photocatalysts	Kinetic constants, $k$ (min <sup>-1</sup> )	Correlation coefficient, $R^2$
TiO <sub>2</sub>	0.011	0.981
1-Ag <sub>2</sub> S/TiO <sub>2</sub>	0.014	0.945
2-Ag <sub>2</sub> S/TiO <sub>2</sub>	0.014	0.972
3-Ag <sub>2</sub> S/TiO <sub>2</sub>	0.018	0.988
4-Ag <sub>2</sub> S/TiO <sub>2</sub>	0.016	0.985
5-Ag <sub>2</sub> S/TiO <sub>2</sub>	0.013	0.973



MO was completely degraded after 210 min and from the analysis of the pseudo-first order kinetics, the rate constant  $k$  of the  $\text{Ag}_2\text{S}/\text{TiO}_2$  composite was determined to be  $0.011 \text{ min}^{-1}$  as shown in Fig. S1.† This shows that the  $\text{Ag}_2\text{S}/\text{TiO}_2$  composite material is photoactive towards visible light irradiation, and it is a promising candidate for photocatalytic applications under visible light.

## Conclusions

$\text{Ag}_2\text{S}$  nanoparticles were successfully deposited on  $\text{TiO}_2$  hierarchical spheres at room temperature *via* the SID method. The efficient charge separation induced by  $\text{Ag}_2\text{S}$  helped to reduce recombination, thus enhancing the photocatalytic activity of the photocatalyst. The  $\text{Ag}_2\text{S}$  nanoparticles deposited over 3 loading cycles gave the best performance and could produce  $\text{H}_2$  gas at an evolution rate of  $708 \mu\text{mol h}^{-1} \text{ g}^{-1}$ . The composite material also exhibited stability when it was recycled 5 times for photocatalytic water splitting without any drastic drop in performance. The  $3\text{-Ag}_2\text{S}/\text{TiO}_2$  composite was capable of degrading organic pollutants, and it photodegraded MO fully within 2 hours with a rate constant of  $0.018 \text{ min}^{-1}$ . The photocatalytic activity of the composite towards visible light illumination was also demonstrated.

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## Notes and references

- 1 A. L. Linsebigler, G. Q. Lu and J. T. Yates, *Chem. Rev.*, 1995, **95**, 735.
- 2 G. Wang, L. Xu, J. Zhang, T. Yin and D. Han, *Int. J. Photoenergy*, 2012, **2012**, 9.
- 3 Z. Haider and Y. S. Kang, *ACS Appl. Mater. Interfaces*, 2014, **6**, 10342.
- 4 H. B. Wu, X. W. Lou and H. H. Hng, *Chem.–Eur. J.*, 2012, **18**, 2094.
- 5 M. Ni, M. K. H. Leung, D. Y. C. Leung and K. Sumathy, *Renewable Sustainable Energy Rev.*, 2007, **11**, 401.
- 6 R. Liu and A. Sen, *J. Am. Chem. Soc.*, 2012, **134**, 17505.
- 7 W. J. Foo, C. Zhang and G. W. Ho, *Nanoscale*, 2013, **5**, 759.
- 8 J. Shen, Y. L. Meng and G. Xin, *Rare Met.*, 2011, **30**, 280.
- 9 T. Zhu, C. K. N. Peh, M. H. Hong and G. W. Ho, *Chem.–Eur. J.*, 2014, **20**, 11505.
- 10 J. Hong, Y. Wang, Y. Wang, W. Zhang and R. Xu, *ChemSusChem*, 2013, **6**, 2263.
- 11 H. Sheng, L. Yu, Y. Jian-Hua and Y. Ying, *Nanotechnology for Sustainable Energy*, American Chemical Society, 2013, p. 219.
- 12 Y. Xie, S. H. Heo, Y. N. Kim, S. H. Yoo and S. O. Cho, *Nanotechnology*, 2010, **21**, 015703.
- 13 M. C. Neves, J. M. F. Nogueira, T. Trindade, M. H. Mendonça, M. I. Pereira and O. C. Monteiro, *J. Photochem. Photobiol., A*, 2009, **204**, 168.
- 14 Y. P. Du, B. Xu, T. Fu, M. Cai, F. Li, Y. Zhang and Q. B. Wang, *J. Am. Chem. Soc.*, 2010, **132**, 1470.
- 15 L. Zhu, Z. Meng, G. Trisha and W. C. Oh, *Chin. J. Catal.*, 2012, **33**, 254.
- 16 L. Zhu, Z. D. Meng and W. C. Oh, *J. Nanomater.*, 2012, **2012**, 1.
- 17 M. Gholami, M. Qorbani, O. Moradlou, N. Naseri and A. Z. Moshfegh, *RSC Adv.*, 2014, **4**, 7838.
- 18 W. G. Fan, S. Jewell, Y. Y. She and M. K. H. Leung, *Phys. Chem. Chem. Phys.*, 2014, **16**, 676.
- 19 Z. Shan, D. Clayton, S. Pan, P. S. Archana and A. Gupta, *J. Phys. Chem. B*, 2014, **118**, 14037.
- 20 Y. Xie, S. H. Yoo, C. Chen and S. O. Cho, *Mater. Sci. Eng., B*, 2012, **177**, 106.
- 21 W. G. Yang, F. R. Wan, Q. W. Chen, J. J. Li and D. S. Xu, *J. Mater. Chem.*, 2010, **20**, 2870.
- 22 W. L. Ong, M. Gao and G. W. Ho, *Nanoscale*, 2013, **5**, 11283.
- 23 J. Y. Liao, B. X. Lei, D. B. Kuang and C. Y. Su, *Energy Environ. Sci.*, 2011, **4**, 4079.
- 24 A. Usami, *Chem. Phys. Lett.*, 1997, **277**, 105.
- 25 J. Ferber and J. Luther, *Sol. Energy Mater. Sol. Cells*, 1998, **54**, 265.
- 26 G. Rothenberger, P. Comte and M. Gratzel, *Sol. Energy Mater. Sol. Cells*, 1999, **58**, 321.
- 27 J. L. Wang, H. Feng, K. M. Chen, W. L. Fan and Q. Yang, *Dalton Trans.*, 2014, **43**, 3990.
- 28 Y. F. Gao, Y. Masuda, Z. F. Peng, T. Yonezawa and K. Koumoto, *J. Mater. Chem.*, 2003, **13**, 608.
- 29 L. S. Yoong, F. K. Chong and B. K. Dutta, *Energy*, 2009, **34**, 1652.
- 30 T. Sreethawong and S. Yoshikawa, *Catal. Commun.*, 2005, **6**, 661.
- 31 H. J. Choi and M. Kang, *Int. J. Hydrogen Energy*, 2007, **32**, 3841.
- 32 B. K. Liu, D. J. Wang, Y. Zhang, H. M. Fan, Y. H. Lin, T. F. Jiang and T. F. Xie, *Dalton Trans.*, 2013, **42**, 2232.
- 33 J. M. Herrmann, H. Tahiri, Y. Ait-Ichou, G. Lassaletta, A. R. González-Elipe and A. Fernández, *Appl. Catal., B*, 1997, **13**, 219.
- 34 M. Kaneko and I. Okura, *Photocatalysis: Science and Technology*, Springer, 2002.
- 35 A. J. Bard, R. Parsons and J. Jordan, *Standard Potentials in Aqueous Solution*, Taylor & Francis, 1985.

