



Cite this: *Nanoscale*, 2015, 7, 12673

Understanding filamentary growth in electrochemical metallization memory cells using kinetic Monte Carlo simulations†

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We report on a 2D kinetic Monte Carlo model that describes the resistive switching in electrochemical metallization cells. To simulate the switching process, we consider several different processes on the atomic scale: electron-transfer reactions at the boundaries, ion migration, adsorption/desorption from/to interfaces, surface diffusion and nucleation. These processes result in a growth/dissolution of a metallic filament within an insulating matrix. In addition, the model includes electron tunneling between the growing filament and the counter electrode, which allows for simulating multilevel switching. It is shown that the simulation model can reproduce the reported switching kinetics, switching variability and multi-level capabilities of ECM devices. As a major result, the influence of mechanical stress working on the host matrix due to the filamentary growth is investigated. It is demonstrated that the size and shape of the filament depend on the Young's modulus of the insulating matrix. For high values a wire-like structure evolves, whereas the shape is dendritic if the Young's modulus is negligible.

Received 9th April 2015,
Accepted 19th June 2015

DOI: 10.1039/c5nr02258d

www.rsc.org/nanoscale

Introduction

Resistive switching devices relying on the electrochemical metallization mechanism (ECM) also known as CBRAM are considered likely to be used in future non-volatile memory.^{1,2} Recently, a 16 GB memory test chip based on ECM cells has been demonstrated.^{3,4} The digital information in ECM cells is encoded as different resistance states, which can be programmed by applying appropriate electrical stimuli. Thereby, switching from a high resistive state (HRS) to a low resistive state (LRS) is called SET operation. In contrast, during the RESET operation the cell is switched back to the HRS. Typically, an ECM cell consists of an ion conducting insulating layer sandwiched between an electrochemically active electrode such as Ag or Cu and an inert counter electrode.^{5,6} Thereby, solid electrolytes such as GeS_x,^{7,8} GeSe^{9,10} or AgI,^{11,12} or metal oxides such as Ta₂O₅,¹³ SiO₂,^{14,15} or Al₂O₃¹⁶ can serve as an ion-conducting layer. If a positive voltage is applied to the active electrode, the active electrode is oxidized, and Ag/Cu ions are injected into the insulating material and migrate in the direction of the electric field. At the inert electrode the Ag/Cu ions are reduced and a Ag/Cu filament starts to grow

towards the active electrode until an electronic contact is established. Prior to the filamentary growth a stable nucleus has to be formed.¹² Depending on the current compliance level this electronic contact can be either a tunnelling gap or a galvanic contact.^{17,18} In this way the LRS can be varied over many orders of magnitude.¹⁷ This behaviour has been observed in dynamic voltage sweep experiments for various types of ECM cells using the current compliance of the measurement setup.^{11,15,19–21} For fast switching the response time of this current compliance might be too slow to control the LRS resistance. In this case a transistor in series to the ECM cell (1T1R configuration) can be used to effectively control the maximum current.^{4,16} It has been reported in the literature that the filament is unstable when no galvanic contact is established and the LRS can already be lost while sweeping the voltage to 0 V due to non-equilibrium states, *i.e.* volatile switching.^{22–24} But, it has also been demonstrated that stable resistance states can be obtained at about 100 kΩ,^{4,16} which is higher than the single atomic contact resistance of 12.9 kΩ. By applying a negative voltage the filament is dissolved. Depending on the LRS regime, *i.e.* tunnelling or galvanic contact, the RESET mechanism differs slightly. For the tunnelling regime, the electrochemical processes are reversed and the filament dissolves. When a galvanic contact is present the current is high enough to enable Joule heating and the dissolution becomes thermally assisted.^{24–27}

The shape of the filament has been investigated using scanning electron microscopy (SEM),²⁸ transmission electron

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†Electronic supplementary information (ESI) available. See DOI: 10.1039/c5nr02258d



microscopy (TEM)^{29–31} and conductive atomic force microscope (C-AFM) tomography.³² A dendritic filament structure was observed in a Ag/H₂O/Pt cell,²⁸ whereas the filament was found to be a small solid crystalline Ag wire in a Ag/SiO₂/Pt cell.³¹ Using C-AFM microscopy a conically shaped filament was observed in Cu/Al₂O₃/TiN based ECM cells.³² It has been demonstrated that counter charges are inevitable for the resistive switching to happen.³³ For Cu/SiO₂/Pt³³ and Cu/Ta₂O₅/Pt³⁴ cell moisture, *i.e.* residual water, supplies the relevant counter charges. In contrast, in AgI based ECM cells Ag⁺ ions are already present and no counter charge is required.

To understand the variability of an ECM cell it is crucial to model the processes on the atomic scale. A first 2D Kinetic Monte Carlo (KMC) model was developed by Pan *et al.* in order to describe filamentary growth.³⁵ This model was later applied to describe the switching kinetics of the electroforming and SET switching of ECM cells.³⁶ Recently, a similar 2D KMC model was employed to study the influence of the electrode materials on the switching characteristics²⁷ and the filamentary growth direction has been discussed using 2D KMC models.^{37,38} In all these models, a percolation/resistor model is used in order to describe the LRS state conduction. Thus, multilevel switching over many orders of magnitude cannot be explained using these models. Furthermore, the mechanical stress working on the host matrix due to the filamentary growth has not been included in an ECM model yet.

In this paper, we present a 2D KMC model for ECM switching that has three new features. First, the potential drops at the metal insulator boundaries due to the electron-transfer-reaction are considered. Second, the influence of the mechanical stress on the filamentary growth is investigated. Third, the model includes electron tunnelling between the growing filament and the counter electrode, which allows for simulating multilevel switching. It is shown that the simulation model can reproduce the reported *I-V* characteristics, switching kinetics, switching variability and multilevel capabilities of ECM devices.

Simulation model

A suitable simulation model needs to account for the atomic process, the filamentary growth and the electric properties of the devices. In the proposed model, the ionic processes and the filamentary growth are modelled using a kinetic Monte Carlo approach, whereas the electric properties are described using a continuum model.

Kinetic Monte Carlo model

Fig. 1 illustrates all processes that are described by the KMC model: (a) oxidation, (b) surface migration, (c) desorption, (d) migration within the insulating matrix, (e) adsorption, (f) reduction and (g) nucleation. In the model the structure is discretized into a 2D grid with quadratic tiles, which resemble the atomic positions. Technically speaking this describes a crystalline material and is the classical KMC approach. But, as

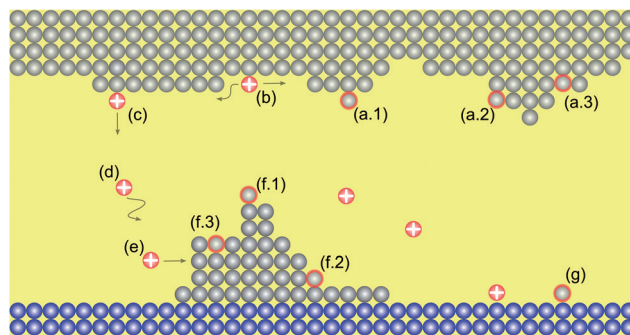


Fig. 1 Illustration of the processes involved in the resistive switching process: oxidation at adatom site (a.1), at kink site (a.2), and hole site (a.3), surface migration (b), desorption (c), bulk migration (d), adsorption, (e) reduction at adatom site (f.1), kink site (f.2) and hole site (f.3), as well as nucleation (g).

amorphous systems have close-range order and the growing filament is very small, this simplification is reasonable for such amorphous systems. The corresponding edge length is identical to the hopping distance *a*. For all simulations the width of the cell is 40 nm and the thickness of the insulating layer is 12.5 nm.

The different processes can be discriminated into four different groups involving motion of ions by hopping, oxidation, reduction reactions, and nucleation. The rate equation for the ion motion is expressed by³⁹

$$\Gamma_{\text{hop}} = \omega_{0,\text{hop}} \exp\left(-\frac{\Delta W_{\text{hop}} - 1/2ze\Delta\phi}{k_{\text{B}}T}\right). \quad (1)$$

Here, ω_0 denotes the vibrational frequency, ΔW_{hop} the migration barrier, *z* the ion charge number, *e* the elementary charge, k_{B} the Boltzmann constant and *T* the local temperature. The potential drop $\Delta\phi$ is defined as the potential difference between two adjacent grid cells in vertical and horizontal directions. Thus, an ion within the insulating matrix (d) can possibly jump up/down or left/right. In the model we discriminate between surface/bulk migration (b)/(d), adsorption to (e) and desorption from (c) an electrode. For these different processes different activation energies are chosen (*cf.* Table 1).

For the reduction processes the rate equation reads

$$\Gamma_{\text{red}} = k_{\text{red}}\omega_{0,\text{red}} \exp\left(-\frac{\Delta W_{\text{red}} - \alpha ze\eta}{k_{\text{B}}T}\right), \quad (2)$$

and for the oxidation

$$\Gamma_{\text{ox}} = k_{\text{ox}}\omega_{0,\text{ox}} \exp\left(-\frac{\Delta W_{\text{ox}} + (1 - \alpha)ze\eta}{k_{\text{B}}T}\right). \quad (3)$$

In eqn (2) and (3) k_{red} and k_{ox} are reaction rate constants, α is the charge transfer coefficient, $\Delta W_{\text{ox}}/\Delta W_{\text{red}}$ the activation energy for oxidation/reduction, and $\omega_{0,\text{red}}$ and $\omega_{0,\text{ox}}$ denote the vibrational frequencies for metal ions and metal atoms, respectively. For oxidation and reduction we discriminate between whether the process happens at an adatom site (a.1/



Table 1 Simulation model parameters

Symbol	Value	Symbol	Value
$\omega_{0,\text{hop}}$	2×10^{13} Hz	$\Delta W_{\text{ox,adatom}}$	0.41 eV
$\omega_{0,\text{red}}$	1×10^{13} Hz	$\Delta W_{\text{ox,kink}}$	0.46 eV
$\omega_{0,\text{ox}}$	2×10^{13} Hz	$\Delta W_{\text{ox,hole}}$	0.58 eV
k_{red}	1	$\Delta W_{\text{red,adatom}}$	0.58 eV
k_{ox}	1	$\Delta W_{\text{red,kink}}$	0.52 eV
α	0.3	$\Delta W_{\text{red,hole}}$	0.45 eV
z	1	$\Delta W_{\text{hop,bulk}}$	0.3 eV
T	300 K	$\Delta W_{\text{hop,surf}}$	0.27 eV
m_{eff}	$0.5m_0$	$\Delta W_{\text{hop,desorption}}$	0.31 eV
ΔW_0	4 eV	$\Delta W_{\text{hop,adsorption}}$	0.25 eV
a	0.25 nm	ΔW_{nuc}	0.6 eV
V_{ref}	2 mV	μ_{ion}	$1 \times 10^{-9} \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$
k_{et}	1	C_1	0.5
ΔW_{et}	0.6 eV	C_2	0.8

(f.1), at a kink site (a.2)/(f.2) or a hole site (a.3)/(f.3). Again, different activation energies are chosen to account for the energetic differences.

As the model geometry is 2D, the 3D information regarding the asymmetry between a small filament and the larger active electrode area cannot be accurately modelled. This issue is resolved by assuming that an atom remains at its position with a certain probability $p = 1 - p_{\text{dissolve}}$. If the atom keeps its position, the oxidized ion is positioned randomly at one of the adjacent grid positions. The dissolution probability depends on the distance between the filament tip and the active electrode x and the filament diameter d_{fil} according to

$$p_{\text{dissolve}} = \frac{C_1}{C_2 + \frac{2x}{d_{\text{fil}}}} \quad (4)$$

The constants C_1 and C_2 are chosen to achieve a realistic dissolution behavior. Initially, the filament consists of the single atomic nucleus, *i.e.* $d_{\text{fil}} = 0.25$ nm, and the tunnelling gap is equal to the ion conducting layer thickness of 12.5 nm. This yields an initial probability of $p \cong 1/200$, which decreases when the filament approaches the counter electrode.

Prior to filamentary growth a stable nucleus has to be formed on the inert electrode.^{13,40,41} As the simulation time is quite long for simulating the nucleation process, it is neglected in the present model and the simulation starts with a stable initial nucleus. Thus, all simulations are performed at high voltage, at which the nucleation process is too fast to be rate-limiting. Nevertheless, an additional formation of a second nucleus is included in the model by allowing for reduction of a metal ion on the inert electrode. This process, however, requires an additional energy ΔW_{nuc} . Thus, the activation energy in this process is given by $\Delta W_{\text{red,IE}} = \Delta W_{\text{red}} + \Delta W_{\text{nuc}}$.

For the KMC simulation the rates for all possible events N are calculated and summed up to the cumulative rate

$$R_i = \sum_{j=1}^i \Gamma_j \quad (5)$$

for a process i , where $i = 1 \dots N$. The event i is randomly chosen according to

$$R_{i-1} < uR < R_i. \quad (6)$$

Here, u is a uniformly distributed random number between 0 and 1 and $R \equiv R_N$. The event i is executed and the time step Δt is updated by

$$\Delta t = -\log(v)/R_N \quad (7)$$

where v is a second uniformly distributed random number between 0 and 1.

Calculation of the electric potential

In order to calculate the KMC transition rates the potential distribution within the cell has to be known. The potential calculation is implemented as a 2D finite element model using COMSOL Multiphysics®. To calculate the electronic and ionic currents within the cell the current continuity equation is solved. Due to the electron transfer reaction at the boundary between ion conductor and electrode a potential drop occurs. To account for this discontinuity two continuity equations are used

$$-\nabla \sigma \nabla \varphi_1 = 0 \quad (8)$$

and

$$-\nabla \sigma_{\text{ion}} \nabla \varphi_2 = 0. \quad (9)$$

Here, eqn (8) is solved for the electric potential φ_1 within the top/bottom electrode and filament area (*cf.* Fig. 2), whereas eqn (9) solves for the electric potential φ_2 within the ion conducting layer.²⁵ In eqn (8) σ denotes the electronic conductivity of the active electrode, inert electrode and filament material. In contrast, σ_{ion} denotes the ionic conductivity of the ion con-

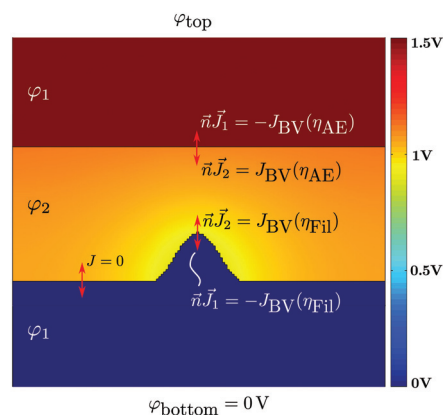


Fig. 2 FEM model for calculation of electric potential φ and currents based on solving the stationary electric continuity equation for electrode (φ_1) and electrolyte domains (φ_2) according to eqn (8) and (9). At the electrolyte/(Ag)-electrode interfaces the Butler-Volmer equation serves as a boundary condition, leading to the reaction overpotentials η (*cf.* eqn (11) and (12)). The colour code illustrates an exemplary potential distribution for $\varphi_{\text{top}} = 1.5$ V.



ducting layer. It depends on the local ion concentration c_{ion} , and the ionic mobility μ_{ion} according to $\sigma_{\text{ion}} = ze c_{\text{ion}} \mu_{\text{ion}}$. The local ion concentration is extracted from the KMC model by averaging the number of ions in a surrounding square with edge length $5a$.

The two continuity eqn (8) and (9) are coupled to each other by affirming that the local current densities at the boundaries are calculated according to the Butler–Volmer equation

$$J_{\text{BV}} = j_0 \left[\exp\left(\frac{(1-\alpha)ze}{k_{\text{B}}T} \eta\right) - \exp\left(-\frac{\alpha ze}{k_{\text{B}}T} \eta\right) \right], \quad (10)$$

which describes the electron-transfer process at the two electrode/ion conducting layer interfaces as shown in Fig. 2. The first term in eqn (10) describes the oxidation process and is related to the oxidation rate eqn (3). In contrast, the second term describes the reduction process, which is expressed as the reduction rate eqn (2) in the KMC model.

In eqn (10) j_0 denotes the exchange current density and obeys an Arrhenius-type law

$$j_0 = ze c_{\text{ion}} k_{\text{et}} \exp\left(-\frac{\Delta W_{\text{et}}}{k_{\text{B}}T}\right) \quad (11)$$

with the activation energy ΔW_{et} and the reaction rate constant k_{et} . The overpotential η at the active electrode/insulator interface is defined by

$$\eta_{\text{ac}} = \varphi_{1,\text{ac}} - \varphi_{2,\text{ac}} - V_{\text{ref}} \quad (12)$$

and at the filament/insulator interface according to

$$\eta_{\text{fil}} = \varphi_{1,\text{fil}} - \varphi_{2,\text{fil}} - V_{\text{ref}}. \quad (13)$$

The reference potential V_{ref} gives the equilibrium potential drop across the corresponding interface, for which the net current through the interface is zero without applied bias, *i.e.* $\eta = 0$. At the inert electrode/ion conducting layer interface the normal current density is set to zero, as nucleation is neglected.

When the filament approaches the active electrode, a significant tunneling current sets in. The corresponding local tunneling current density is calculated using the linear Simmons equation according to

$$J_{\text{Tu}} = \frac{3\sqrt{2m_{\text{eff}}\Delta W_0}}{2x} \left(\frac{e}{h}\right)^2 \exp\left(-\frac{4\pi x}{h} \sqrt{2m_{\text{eff}}\Delta W_0}\right) V. \quad (14)$$

This depends on the effective electron mass m_{eff} , the tunnelling barrier height ΔW_0 , the tunnelling distance x , and Planck's constant h . The potential drop V is determined by $V = \varphi_{1,\text{AE}} - \varphi_{1,\text{fil}}$. As the electrode and filament electric conductivity is significantly higher than the ionic conductivity, the potential φ_1 at the metal/insulator interfaces is almost constant, which is shown exemplarily by the simulated potential distribution for an applied voltage of 1.5 V in Fig. 2. The equation system eqn (8)–(13) is completed by appropriate boundary conditions. Dirichlet boundary conditions are chosen at the top and bottom of the cell, which read $\varphi_{\text{top}} = V_{\text{app}}$ and $\varphi_{\text{bottom}} = 0$ V, respectively. Furthermore, the normal current density is

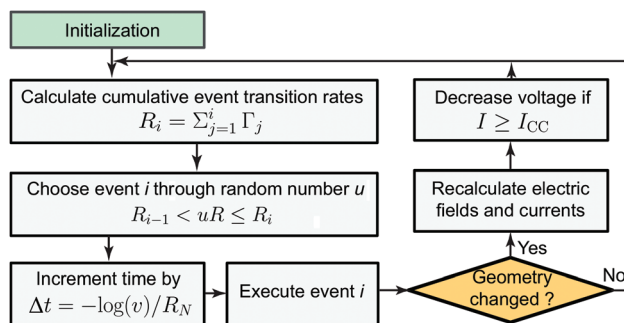


Fig. 3 Flow chart representing the implemented KMC algorithm. Here, u and v are independent random numbers between 0 and 1. The electric fields and currents are recalculated using the FEM model illustrated in Fig. 2.

set to zero on the remaining left and right edges of the cell, *i.e.* $J_{\text{n}} = 0$.

Simulation procedure

The whole simulation process is illustrated in Fig. 3. After an initialization step the first KMC step is executed. This means that all possible transition rates are calculated according to eqn (1)–(3). Then the cumulative event transition rates are determined (eqn (5)) and an event is randomly chosen through a random number u (eqn (6)). Afterwards the time step is adjusted according to eqn (7) and the event i is executed. The electric field/potential is updated according to the equation system eqn (8)–(14), which is implemented in COMSOL Multiphysics. To save computation time an electric field and current calculation is not performed after every step, but only when a geometry change occurs. If the calculated current is bigger than the set compliance current I_{CC} , the applied voltage is reduced so that the current is kept constant. This implementation mimics an ideal current compliance and can be best compared to the current control in a 1T1R configuration. Subsequently, a new KMC step is executed. Note that only the electric field update is calculated using a finite element continuum model, whereas the KMC solver is based on a self-written code.

For the simulations the previously published experimental data of an AgI-based ECM cell are used as reference.^{12,42} Thus, the simulation parameters are chosen to fit these data and are given in Table 1. Furthermore, in AgI a large amount of Ag ion interstitials is already available within the film and a counter charge is not required. Thus, an initial injection of Ag ions and a migration through the complete insulator thickness, which has been taken into account by other groups,⁴³ is not required.

Results

Voltage dependence of filament evolution

To study the filament evolution SET simulations were conducted using voltage pulses of $V_{\text{app}} = 1$ V, 1.5 V, and 2 V. The



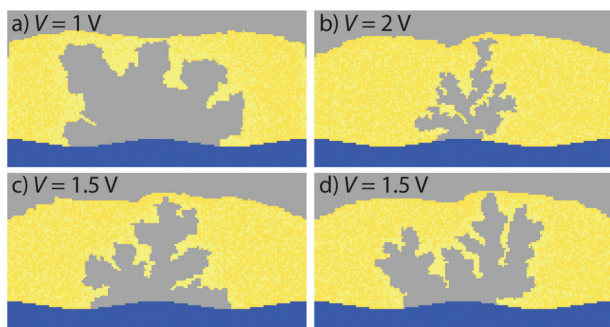


Fig. 4 Simulated filament shapes for cell voltages of (a) 1 V, (b) 2 V, and (c), (d) 1.5 V. For voltages of 1 V and lower a primarily uniform filament growth is observed (a). (c), (d) In the intermediate voltage range a dendritic filament growth is observed as a result of the non-constant ion concentration at the filament surface. If higher voltages are used (b) the growth becomes more anisotropic resulting in a more dendritic structure with a filament diameter of only a few atoms. A movie of the filament evolution for 1 V is given in the ESI.†

simulations are terminated as soon as the device current is larger than a compliance current of $I_{CC} = 100$ nA. Note that the filament evolution is independent of the applied current compliance. The filament would grow a little bit further only for higher current compliances. The simulated filament shapes are shown in Fig. 4. The current compliance has been achieved after $t_{SET} = 447.9$ ns, 116 ns, and 21.3 ns for 1 V (Fig. 4a), 1.5 V (Fig. 4d), and 2 V (Fig. 4b), respectively. In all cases a dendritic-shaped structure rather than a single wire has formed. This is consistent with the experimentally observed dendritic structures in water-based ECM cells.²⁸ Furthermore, the simulation reveals that the filament shape strongly depends on the applied voltage. For 1 V a relatively wide uniform growth is observed. For increasing voltages the growth becomes more anisotropic resulting in a more dendritic structure with a filament diameter of only a few atoms. This behavior can be explained by a different limiting mechanism. For 1 V the electron-transfer reactions limit the growth speed. As the potential at the metal insulator/interfaces is almost constant (*cf.* Fig. 2), the corresponding rates are very similar. Thus, a rather homogeneous growth occurs. At higher voltages ion migration also becomes limiting, *i.e.* the ions cannot be supplied fast enough to the growing filament tip. Thus, a local depletion of Ag ions develops, leading to higher electric fields in this region. Since ion migration is electric field dependent a self-acceleration of the filament growth sets in locally. This results in the observed dendritic structures with small twig diameters.

Multilevel switching

A further simulation study was performed to study the multilevel switching behavior for current compliances of $I_{CC} = 1$ nA, 10 nA, 100 nA, and 1 μ A. For each current compliance level 10 simulations were conducted for an applied voltage of 1.5 V. The resulting medium LRS resistances R_{LRS} are given in Fig. 5a as blue diamonds and the corresponding filament

shapes are shown in Fig. 5b. These results coincide with the calculation results of our previously published analytical model (blue line)⁴⁰ and are thus also consistent with our 1D KMC and numerical model.^{17,42} In the present model it is possible to distinguish between the contributions of tunneling distance and filament diameter in explaining the multilevel switching of ECM cells. On the one hand, the medium tunneling gap decreases with increasing current compliance shown as black squares in Fig. 5a. On the other hand a widening of the filament tip radius (red squares) is observed at higher current compliance. The widening of the filament tip radius can be partly attributed to the fact that the filament grows “into” a retreating active electrode, which in turn enlarges the active tunneling area as shown in Fig. 5b for a filament formed at 1 μ A. The simulation results suggest that the LRS resistance is determined by the microscopic tip topography, *i.e.* effective tunneling area and tunneling gap. Furthermore, the latter appears to dominate at lower current compliances.

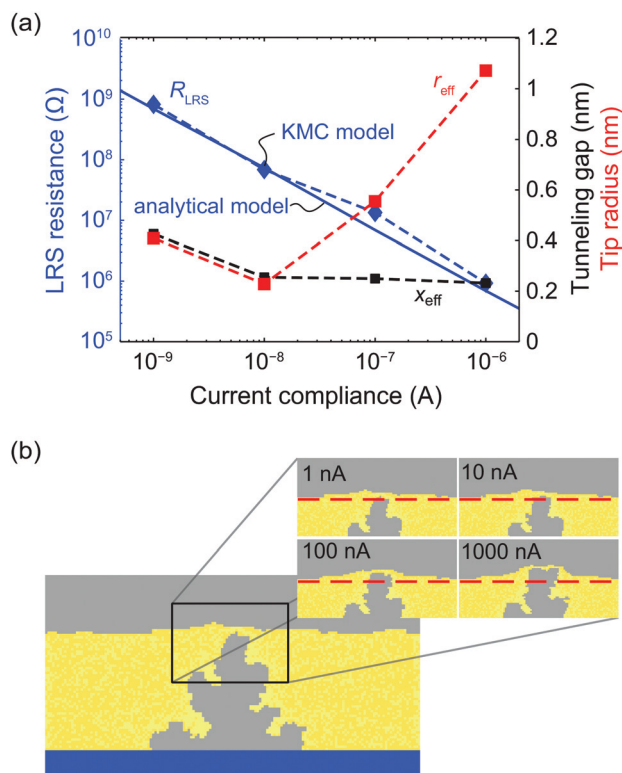


Fig. 5 (a) Simulated LRS resistance as a function of the current compliance for the presented KMC model shown as blue diamonds connected by a dashed line compared to the results of an analytical 1D model depicted as blue solid line.⁴⁰ The corresponding tunnelling gaps x_{eff} and effective filament tip radius r_{eff} are illustrated in black and red, respectively. (b) Filament evolution in multilevel switching using the extended model for four different I_{CC} . When the tunnelling gap decreases, the vertical filament growth velocity and anode dissolution velocity above the filament tip are converging due to the law of mass conservation. Depending on the chosen current compliance different filament tip topologies result. The dashed lines mark the initial active electrode/insulator interface.



Influence of mechanical stress

The previous simulations do not include the mechanical response of the insulating layer on the growing filament. The filament growth can take place in any direction, which reflects the situation in a liquid very well. As mentioned before, the simulated dendritic shape agrees well with the experimentally observed shape in a water-based ECM cell.²⁸ In contrast, for a solid ion conducting layer, *i.e.* a SiO₂-based ECM cell, the filament was identified as a small metal wire.³¹ In fact, it should be expected that the filament growth induces mechanical stress in the host matrix. This should constrain the filament to some finite small size for solid ion conducting materials in general.

To model the evolution of mechanical stress a radial two-spring model as depicted as inset in Fig. 6a is introduced. This model resembles an existing nanotemplate, where the filament grows preferentially, and a surrounding host matrix. The small spring (I) models relaxation processes within the nanotemplate and yields a nonlinear Young's modulus that changes from a lower value $E_{1,0} = 500$ MPa to a $E_{1,\max} = 1$ TPa within the first $l_1 = 5$ nm according to

$$E_1 = E_{1,0} \exp\left(\frac{r}{l_1} \ln\left(\frac{E_{1,\max}}{E_{1,0}}\right)\right) \quad (15)$$

Spring (II) describes the mechanical properties of the bulk host insulator and a constant Young's modulus E_2 is assumed. The Young's modulus is connected to the spring constant k according to $k = EA_k/L$, where A_k denotes the cross section on which the force works, *i.e.* the grid size is a^2 , and L is the cell thickness. The overall radial spring constant is then calculated using $k_{\text{ser}} = k_1 k_2 / (k_1 + k_2)$. The mechanical work ΔW_{mech} that is stored when the spring is compressed by r results from the integration of Hooke's law, which yields

$$\Delta W_{\text{mech}} = e \int_0^d r k_{12}(d') dd' \quad (16)$$

Within the present model the quantity d equals the filament diameter along the vertical axis. Fig. 6a shows the

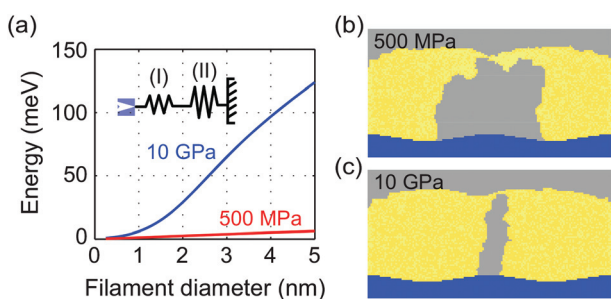


Fig. 6 (a) 1D model of the mechanical stress inside the electrolyte layer depending on the filament diameter. Spring (I) models relaxation processes and yields a nonlinear Young's modulus. Spring (II) describes the mechanical properties of the bulk electrolyte. The corresponding simulated filament shapes at $V_{\text{app}} = 1$ V are shown in (b) and (c) for a Young's modulus of $E_2 = 500$ MP and 10 GPa, respectively.

mechanical work as a function of filament diameter for $E_2 = 10$ GPa (blue) and 500 MPa (red). This mechanical work is added to the reduction rate $\Delta W_{\text{red,eff}} = \Delta W_{\text{red}} + \Delta W_{\text{mech}}$, hence modelling the influence of the mechanical stress of the host matrix imposed on the growing filament. For $E_2 = 500$ MPa the increase in barrier height is very small, but for 10 GPa an increase in the range of about 0.1 eV is calculated. This is in the range of the activation energies of ion reduction. For the reference AgI-based ECM cell it has been shown by X-ray diffraction that the AgI exhibits a zinc blende structure.¹¹ For this type of structure a Young's modulus of 10–40 GPa has been reported.⁴⁴

To study the influence of the additional mechanical stress, SET simulations were performed using 1 V voltage pulses, 100 nA current compliance and different Young's moduli E_2 . The resulting filament shapes are shown in Fig. 6b and c for $E_2 = 500$ MPa and 10 GPa, respectively. In both cases a very homogeneous filament wire results in contrast to the more dendritic structure, when mechanical stress is neglected (*cf.* Fig. 4). Furthermore, the diameter of the grown filament depends on Young's modulus. If the Young's modulus is high, the mechanical stress working on the filament is big resulting in a smaller filament diameter. The simulated filament growth is comparable to a layer-by-layer growth mode, which has been assumed in our previous 1D KMC model.⁴² The influence of mechanical stress has been first considered by Ambrogio *et al.* to explain the asymmetry in SET and RESET current using compact modeling.²⁴ In contrast to the implementation in this study, the ion migration was influenced by mechanical stress rather than the lateral filament growth. In fact, the study of Ambrogio *et al.* predicts further lateral growth due to stress relaxation.

Switching kinetics

In a next step the KMC model is applied to simulate the switching kinetics of a reference AgI-based ECM cell.^{11,12} For this, SET simulations are conducted for pulse voltages $V = 0.5$ V, 0.8 V, 1 V, 1.5 V, and 2 V. Voltages smaller than 0.5 V are not applied as nucleation limits the switching speed in this regime, which is not included in the KMC model. Fifteen iterations are performed for the latter three voltages, five for 0.8 V and a single simulation for 0.5 V. In all simulations the two-spring-model with $E_2 = 10$ GPa is applied to account for the mechanical stress within the insulating matrix. The simulation results are shown in Fig. 7a as open squares compared to the experimental data of the AgI-based ECM cell (red circles). The simulation results are in a reasonably good agreement with the experimental data. The fit is very good between 1 V and 2 V, which is the voltage regime we used for most of the simulations. Note that only one simulation was performed at 0.5 V. Thus, the deviation from experiment might be lower. Consistent with our previous findings^{12,42} one can distinguish between an electron-transfer limited regime and a mixed control regime, in which the electron-transfer reaction and the ion hopping transport limit the switching speed. This is illustrated by the time evolution of the overpotentials during



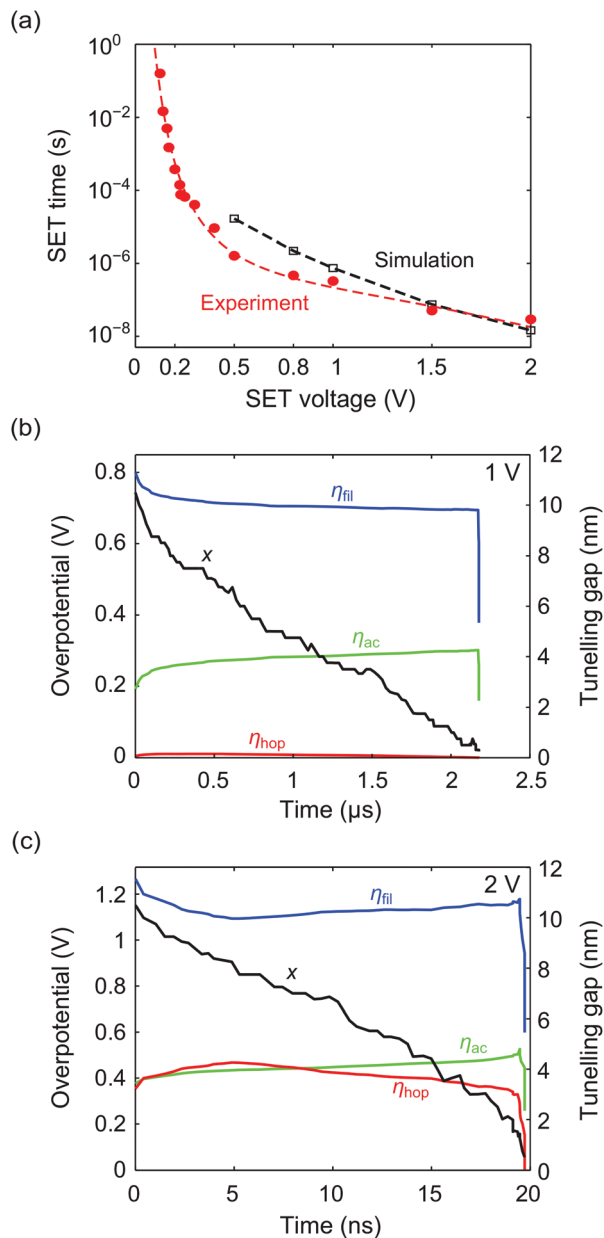


Fig. 7 (a) Simulated mean pulsed SET switching kinetics data (open squares) compared to the experimental data (circles) of an AgI-based ECM cell^{12,42} showing a good agreement. The dashed lines serve as a guide for the eye. (b) and (c) show the time evolution of the overpotentials at the filament/insulator interface (blue), the active electrode/insulator interface (green) and the hopping overpotential (red) for 1 V and 2 V, respectively. The decrease of the tunnelling gap is depicted as solid black line.

filamentary growth for $V = 1$ V and 2 V in Fig. 7b and c, respectively. For $V = 1$ V the hopping overpotential η_{hop} (red solid line) is almost zero and the whole voltage drop is divided between the two electron-transfer overpotentials η_{ac} (green solid line) and η_{fil} (blue solid line). In contrast, a significant portion of the applied voltage is taken up by the hopping overpotential η_{hop} for $V = 2$ V indicating the mixed control regime. As very

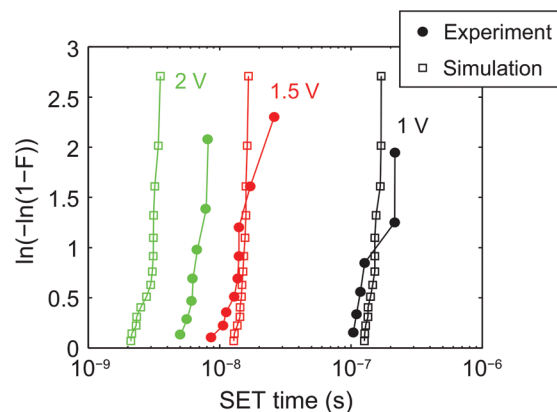


Fig. 8 Weibull plot of the switching kinetics data for $V = 1, 1.5$ and 2 V. The simulation data is depicted as open squares and the experimental data⁴² as filled circles. The switching distributions can be well described using the KMC simulation model.

few voltage levels were simulated due to the long simulation time, the exact transition between those 2 regimes cannot be identified.

Fig. 8 shows the Weibull distribution for the simulations with $V = 1$ V, 1.5 V, and 2 V compared to experimental data from the AgI-based ECM cell.⁴² Both experimental and simulation data show very steep distributions and are in good agreement with each other. A similar result has been obtained for the 1D KMC model published before,⁴² indicating that the different models give very consistent results.

I-V characteristic

Finally, the KMC model is used to simulate the current-voltage behavior of an ECM cell. For this a triangular voltage is applied with a maximum voltage of $V_{\text{max}} = 1.1$ V, a minimum voltage $V_{\text{min}} = -0.4$ V and a sweep rate of $0.1 \text{ V } \mu\text{s}^{-1}$. A very high sweep rate is chosen to save computation time. Furthermore, the current compliance was set to $I_{\text{CC}} = 100$ nA and the two-spring model was applied with $E_2 = 10$ GPa. The resulting *I-V* characteristics are shown in Fig. 9a and the corresponding filament shape for the points (A)–(F) is illustrated in Fig. 9b. Initially, the current is zero and there is only the assumed small nucleus present (A). When a positive voltage is applied, the filament grows (B). A significant tunneling current sets in as soon as the tunneling distance between filament and active electrode becomes very small (C). During the current compliance a slight reshaping of the filament tip topography occurs (D) but the filament growth is strongly suppressed. By reversing the voltage polarity, the current drops as soon as the filament gap opens (E) and finally the complete filament dissolves (F). The filament shape during switching is identified as a very homogeneous wire, whose diameter is determined by the mechanical stress. In contrast, a wider dendritic structure evolved when mechanical stress was neglected as shown in Fig. 4a. The *I-V* characteristics exhibit a quite asymmetric behavior, which is typical for ECM cells. The RESET current is



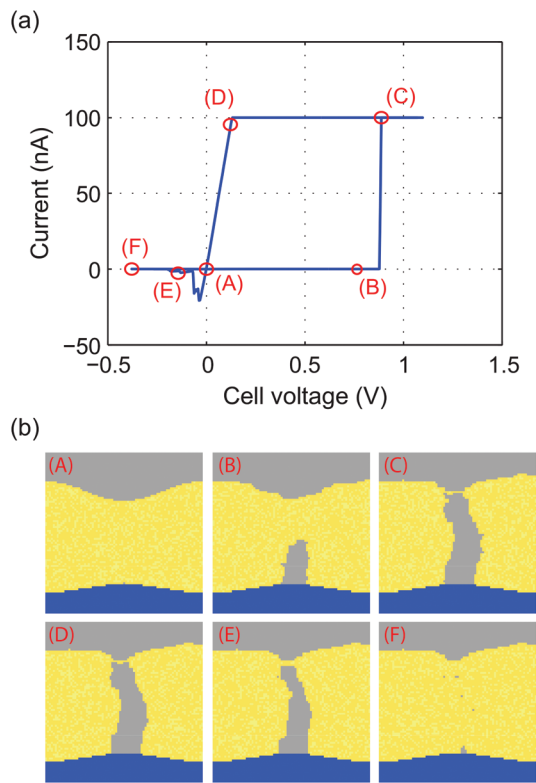


Fig. 9 (a) Simulated quasistatic I - V characteristics using a sweep rate of $0.1 \text{ V } \mu\text{s}^{-1}$ and a current compliance of $I_{CC} = 100 \text{ nA}$. (b) Snapshots of filament evolution in different stages (A)–(F) of the I - V characteristics shown in (a). A movie of the filament evolution is given in the ESI.†

smaller than the SET current and the RESET voltage is lower than the SET voltage. This behavior can be explained based on the findings of a previous study.⁴⁵ The charge transfer coefficient in the present study is $\alpha = 0.3 < 0.5$ and the active electrode area is smaller than the area at the filament tip. For this combination, it has been analytically derived that the observed asymmetry should occur.⁴⁵

Conclusions

We presented a 2D KMC model for the switching behaviour of ECM cells accounting for all observed experimental phenomena: I - V characteristics, multilevel switching, switching kinetics and switching variability. The simulation results are consistent with our previous 1D models. Moreover, the multi-dimensional nature of the model gives insight into the evolution of the filamentary growth and dissolution. In addition, a two-spring model is introduced, which models the mechanical stress that evolves during filamentary growth. Based on our simulation results we conclude the following:

(1) The size and shape of the growing filament depend on the applied voltage, when the filament can grow freely in all directions. This is related to the electrochemical processes determining the switching speed. If the electron-transfer reac-

tion dominates (at low voltages), a rather homogeneous wider dendritic structure results. For high voltages the hopping process becomes more prominent and a very small even more dendritic structure with atomically thin twigs evolves.

(2) If mechanical stress is considered, a wire-like filamentary growth is observed rather than a dendritic one. The filament diameter is then determined by the Young's modulus of the host matrix. Under these conditions the filamentary growth can be described by a layer-by-layer growth mode.

(3) The LRS resistance state during multilevel switching is determined by the tunnelling distance between the filament tip and the active electrode and the filament tip topography, *i.e.* the effective tunnelling area. Combined with the layer-by-layer growth mode, the determining factor can be broken down to the number of atoms within the filament in a 1D model.

(4) The 2D simulations confirm that the switching kinetics are limited by the electron-transfer reaction at a medium voltage range. At high voltages the switching time is controlled by the electron-transfer reaction and the ion hopping process.

Acknowledgements

This work was supported in part by the Samsung Global Research Outreach Program and by the Deutsche Forschungsgemeinschaft (SFB 917).

Notes and references

- 1 J. Jameson, P. Blanchard, C. Cheng, J. Dinh, A. Gallo, V. Gopalakrishnan, C. Gopalan, B. Guichet, S. Hsu, D. Kamalanathan, D. Kim, F. Koushan, M. Kwan, K. Law, D. Lewis, Y. Ma, V. McCaffrey, S. Park, S. Puthentharam, E. Runnion, J. Sanchez, J. Shields, K. Tsai, A. Tysdal, D. Wang, R. Williams, M. Kozicki, J. Wang, V. Gopinath, S. Hollmer and M. Van Buskirk, *IEEE International Electron Devices Meeting (IEDM)*, 2013, 738.
- 2 N. Derhacopian, S. C. Hollmer, N. Gilbert and M. N. Kozicki, *Proc. IEEE*, 2010, **98**, 283.
- 3 R. Fackenthal, M. Kitagawa, W. Otsuka, K. Prall, D. Mills, K. Tsutsui, J. Javanifard, K. Tedrow, T. Tsushima, Y. Shibahara and G. Hush, 2014 *IEEE International Solid-State Circuits Conference Digest of Technical Papers (ISSCC)*, 2014, 338.
- 4 J. Zahurak, K. Miyata, M. Fischer, M. Balakrishnan, S. Chhajed, D. Wells, H. Li, A. Torsi, J. Lim, M. Korber, K. Nakazawa, S. Mayuzumi, M. Honda, S. Sills, S. Yasuda, A. Calderoni, B. Cook, G. Damarla, H. Tran, B. Wang, C. Cardon, K. Karda, J. Okuno, A. Johnson, T. Kunihiro, J. Sumino, M. Tsukamoto, K. Aratani, N. Ramaswamy, W. Otsuka and K. Prall, 2014 *IEEE International Electron Devices Meeting (IEDM)*, 2014, 6.2.
- 5 I. Valov and M. N. Kozicki, *J. Phys. D: Appl. Phys.*, 2013, **46**, 074005.



- 6 R. Waser and M. Aono, *Nat. Mater.*, 2007, **6**, 833.
- 7 G. Palma, E. Vianello, G. Molas, C. Cagli, F. Longnos, J. Guy, M. Reyboz, C. Carabasse, M. Bernard, F. Dahmani, D. Bretegnier, J. Liebault and B. De Salvo, *Jpn. J. Appl. Phys.*, 2013, **52**, UNSP 04CD02/1.
- 8 J. R. Jameson, N. Gilbert, F. Koushan, J. Saenz, J. Wang, S. Hollmer and M. N. Kozicki, *Appl. Phys. Lett.*, 2011, **99**, 063506.
- 9 R. Soni, P. Meuffels, A. Petraru, O. Vavra and H. Kohlstedt, *J. Appl. Phys.*, 2013, **113**, 124504.
- 10 M. N. Kozicki, M. Park and M. Mitkova, *IEEE Trans. Nanotechnol.*, 2005, **4**, 331.
- 11 S. Tappertzhofen, I. Valov and R. Waser, *Nanotechnology*, 2012, **23**, 145703.
- 12 S. Menzel, S. Tappertzhofen, R. Waser and I. Valov, *Phys. Chem. Chem. Phys.*, 2013, **15**, 6945.
- 13 T. Tsuruoka, T. Hasegawa, I. Valov, R. Waser and M. Aono, *AIP Adv.*, 2013, **3**, 32114.
- 14 C. Schindler, G. Staikov and R. Waser, *Appl. Phys. Lett.*, 2009, **94**, 072109.
- 15 Y. Bernard, V. T. Renard, P. Gonon and V. Jousseume, *Microelectron. Eng.*, 2011, **88**, 814.
- 16 A. Belmonte, W. Kim, B. T. Chan, N. Heylen, A. Fantini, M. Houssa, M. Jurczak and L. Goux, *IEEE Trans. Electron Devices*, 2013, **60**, 3690.
- 17 S. Menzel, U. Böttger and R. Waser, *J. Appl. Phys.*, 2012, **111**, 014501.
- 18 I. Valov, I. Sapezanskaia, A. Nayak, T. Tsuruoka, T. Bredow, T. Hasegawa, G. Staikov, M. Aono and R. Waser, *Nat. Mater.*, 2012, **11**, 530.
- 19 T. Liu, Y. Kang, S. El-Helw, T. Potnis and M. Orłowski, *Jpn. J. Appl. Phys.*, 2013, **52**, 084202.
- 20 K. Tsunoda, Y. Fukuzumi, J. R. Jameson, Z. Wang, P. B. Griffin and Y. Nishi, *Appl. Phys. Lett.*, 2007, **90**, 113501.
- 21 C. Schindler, PhD thesis, RWTH, Aachen, 2009.
- 22 J. van den Hurk, E. Linn, H. Zhang, R. Waser and I. Valov, *Nanotechnology*, 2014, **25**, 425202.
- 23 I. Valov, E. Linn, S. Tappertzhofen, S. Schmelzer, J. van den Hurk, F. Lentz and R. Waser, *Nat. Commun.*, 2013, **4**, 1771.
- 24 S. Ambrogio, S. Balatti, S. Choi and D. Ielmini, *Adv. Mater.*, 2014, **26**, 3885.
- 25 S. Menzel, N. Adler, J. van den Hurk, S. Tappertzhofen, I. Valov and R. Waser, 2013 5th IEEE International Memory Workshop (IMW), 2013, 92.
- 26 T. Tsuruoka, K. Terabe, T. Hasegawa and M. Aono, *Nanotechnology*, 2010, **21**, 425205.
- 27 J. Guy, G. Molas, P. Blaise, C. Carabasse, M. Bernard, A. Roule, G. Le Carval, V. Sousa, H. Grampeix, V. Delaye, A. Toffoli, J. Cluzel, P. Brianceau, O. Pollet, V. Balan, S. Barraud, O. Cueto, G. Ghibaudo, F. Clermidy, B. De Salvo and L. Perniola, 2014 IEEE International Electron Devices Meeting (IEDM), 2014, 6.5.
- 28 X. Guo, C. Schindler, S. Menzel and R. Waser, *Appl. Phys. Lett.*, 2007, **91**, 1.
- 29 S. Choi, G. Park, K. Kim, S. Cho, W. Yang, X. Li, J. Moon, K. Lee and K. Kim, *Adv. Mater.*, 2011, **23**, 3272.
- 30 Y. Yang, P. Gao, S. Gaba, T. Chang, X. Pan and W. Lu, *Nat. Commun.*, 2012, **3**, 732.
- 31 Y. Yang, P. Gao, L. Li, X. Pan, S. Tappertzhofen, S. Choi, R. Waser, I. Valov and W. D. Lu, *Nat. Commun.*, 2014, **5**, 4232.
- 32 U. Celano, L. Goux, A. Belmonte, K. Opsomer, A. Franquet, A. Schulze, C. Detavernier, O. Richard, H. Bender, M. Jurczak and W. Vandervorst, *Nano Lett.*, 2014, **14**, 2401.
- 33 S. Tappertzhofen, I. Valov, T. Tsuruoka, T. Hasegawa, R. Waser and M. Aono, *ACS Nano*, 2013, **7**, 6396.
- 34 T. Tsuruoka, K. Terabe, T. Hasegawa, I. Valov, R. Waser and M. Aono, *Adv. Funct. Mater.*, 2012, **22**, 70.
- 35 F. Pan and V. Subramanian, 2010 Device Research Conference, 2010, 255.
- 36 F. Pan, S. Yin and V. Subramanian, *IEEE Electron Device Lett.*, 2011, **32**, 949.
- 37 P. Sun, S. Liu, L. Li and M. Liu, *J. Semicond.*, 2014, **35**, 104007.
- 38 S. Qin, Z. Liu, G. Zhang, J. Zhang, Y. Sun, H. Wu, H. Qian and Z. Yu, *Phys. Chem. Chem. Phys.*, 2015, **17**, 8627.
- 39 J. Maier, *Physical Chemistry of Ionic Materials*, Wiley-VCH, Weinheim, 2004.
- 40 S. Menzel and R. Waser, *Nanoscale*, 2013, **22**, 11003.
- 41 I. Valov and G. Staikov, *J. Solid State Electrochem.*, 2013, **17**, 365.
- 42 S. Menzel, B. Wolf, S. Tappertzhofen, I. Valov, U. Böttger and R. Waser, 2014 6th IEEE International Memory Workshop (IMW), Taipei Taiwan, 2014.
- 43 S. Lv, H. Wang, J. Zhang, J. Liu, L. Sun and Z. Yu, *IEEE Trans. Electron Devices*, 2014, **61**, 3166.
- 44 Y. Li, L. J. Zhang, T. Cui, Y. W. Li, Y. Wang, Y. M. Ma and G. T. Zou, *J. Phys.: Condens. Matter*, 2008, **20**, 195218.
- 45 S. Menzel and R. Waser, *Phys. Status Solidi RRL*, 2014, **8**, 540.

