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# Metallo-dithiaporphyrin pigments for bulk-heterojunction solar cell applications: *ab initio* investigation of structural and optoelectronic properties†

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Metallo-dithiaporphyrin small molecules have been designed by substituting Ru(II) with various transition metals at the same oxidation state ( $M = \text{Mn, Fe, Ni, Cu}$ ) as donor materials for Bulk Heterojunction Organic Solar Cells (BHJ-OSCs). Density functional theory (DFT) and time-dependent DFT (TD-DFT) have been used to study the optoelectronic properties of metallo-dithiaporphyrin at various functionals and basis sets. We discovered that the open-circuit voltage ( $V_{OC}$ ) value increases when Ru(II) in  $\text{Ru}(\text{S}_2\text{TTP})\text{Cl}_2$  ( $\text{S}_2\text{TTP}$  = tetra-*p*-tolylidithiaporphyrin) is substituted. In addition, the light harvesting efficiency (LHE) of nickel, manganese, and iron complexes was found to be similar to that of ruthenium, and the iron complex furthermore presented a comparable charge transfer in the excited state corresponding to the Q-band, compared to  $\text{Ru}(\text{S}_2\text{TTP})\text{Cl}_2$ . Hence  $\text{M}(\text{S}_2\text{TTP})\text{Cl}_2$  ( $M = \text{Mn, Fe, Ni}$ ) appear to be potential low cost candidate donor molecules within a bulk heterojunction solar cell. We further propose suitable engineered acceptor pigments, fitted to provide a good overall solar cell efficiency.

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## Introduction

Nowadays, the most commonly used photovoltaic solar cells are made of mineral semiconductors like silicon. These cells have energy conversion efficiencies of up to 10% for amorphous silicon and 26% for crystalline silicon.<sup>1–3</sup> However, their heavy and brittle nature makes installation expensive, and the energy required to purify and crystallize them increases their energy payback time, reducing their energy return on investment. Thus, it is necessary to find alternative materials that are more cost-effective and energy-efficient.

Organic molecules-based photovoltaic cells offer greater flexibility to optimize their opto-electronic properties. Small molecule Heterojunction-Organic Solar Cells (SMH-OSCs), which consist of donor and acceptor molecules in separate phases but in close contact with the solid semiconductor phases, are particularly interesting.<sup>4–7</sup> The efficiency of these cells depends on various parameters, including the energy gap ( $\Delta\epsilon_{L-H}$ ), oscillator strength ( $f$ ), and open-circuit voltage ( $V_{OC}$ ) of the

donor-acceptor pair, which can be modified through chemical engineering techniques.

To improve the efficiency of the donor-acceptor pair, the donor must have a strong absorption band in the visible range and its chemical potential in the first excited state should be higher than that of the acceptor. This ensures that the electron transfer from donor to acceptor is favourable. In terms of the frozen molecular orbital framework, this can be stated as the acceptor's lowest unoccupied molecular orbital (LUMO) energy being lower than that of the donor for a favourable electron transfer (Scheme 1).

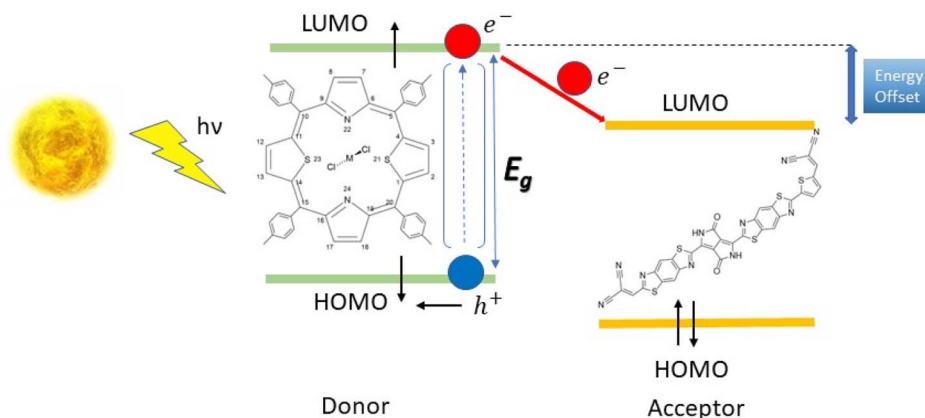
Porphyrins and their related coordination complexes are often brightly coloured and have strong UV-visible absorptions, which make them potential candidates for use in organic solar cells. Previous studies have shown that ruthenium-based dithiaporphyrin complexes can act as efficient donors in organic solar cells.<sup>8–11</sup> However, the scarcity and high cost of ruthenium limit its application in solar cells.<sup>12</sup> The synthesis of heteroporphyrins, especially dithiaporphyrins, has been widely studied,<sup>13–18</sup> and these studies revealed the difficulty of inserting a metal within the molecule because of its ring size, until a recent study of the ruthenium-based dithiaporphyrin complex was conducted by Hang *et al.*<sup>19</sup> The substitution of ruthenium by cheaper and more abundant metals appears desirable, but the aforementioned experimental difficulties in the complexation of dithiaporphyrins may be a strong limitation to their study. With this in mind, we thus considered conducting

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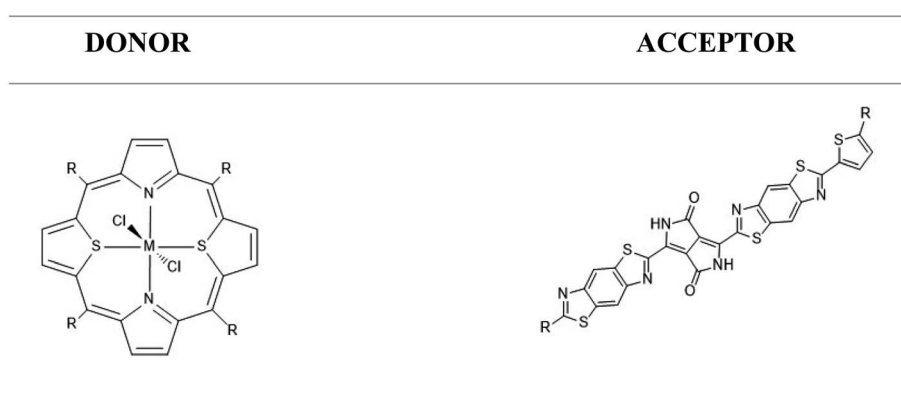
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Scheme 1 Photoinduced electron transfer in a model organic heterojunction solar cell.



Scheme 2 Left: structure of  $M(S_2TMeP)Cl_2$  ( $R = Me$ ) or  $M(S_2TTP)Cl_2$  ( $R = p\text{-tolyl}$ ) complexes, with  $M = Mn, Fe, Ni, Cu$  or  $Ru$ . Right: structure of the acceptor molecules deriving from PDPP2TzT (A1:  $R = H$ ; A2:  $R = C_6H_5$ ; A3:  $R = HC = C(CN)_2$ ; A4:  $R = CHO$ ; A5:  $R = NO_2$ ; A6:  $R = COOH$ ).

a theoretical preliminary work, studying first whether the proposed metal substitution at  $Ru(S_2TTP)Cl_2$  would affect the donor properties of the complex, and second whether a suitable acceptor molecule could be proposed to afford a SMH-OSC.

Our main focus was on transition metals, and iron was our initial choice because it is part of the same group as ruthenium. Thanks to this identical valence configuration, it is more likely to form a stable complex like that of the  $Ru$ .<sup>19</sup> We also considered other elements from the fourth period, based on their cost and natural abundance. As a result, the metals  $Ni$ ,  $Mn$  and  $Cu$  were selected.<sup>‡</sup>

In a first part, we calibrated our theoretical methodology against experimental data obtained for  $Ru(S_2TTP)Cl_2$ . We then discussed the structural parameters of all complexes (sketched in Scheme 2) and their optical properties, among which their computed UV-visible spectra and light harvesting efficiency (LHE). Then, a charge transfer (CT) study gave us more visibility

into the nature of the CT and its orientation within the organometallic pigment.

Finally, we studied a promising acceptor material derived from (poly{3,6-bis(5-hexyldecyl-2-thenyl)-2,5-dihydro-2,5-di(alkyl)pyrrolo[3,4]pyrrolo-1,4-dionethiazole}) (PDPP2TzT) which was synthesized by Janssen and co-workers.<sup>21,22</sup> PDPP2TzT is a very good candidate as an acceptor material thanks to its low energy frontier molecular orbitals ( $-5.63$  eV and  $-4.00$  eV as computed at the B3LYP/DEF2-TZVP level of theory), broad infrared absorption (300–900 nm).

## Methodology and methods

All geometries were optimized using density functional theory DFT<sup>23,24</sup> with several exchange–correlation functionals: B3LYP,<sup>25</sup> CAM-B3LYP,<sup>26</sup>  $\omega$ B97X-D,<sup>27</sup> M05,<sup>28</sup> and M05-2X<sup>29</sup> and various combinations of basis sets and pseudopotentials: either 6-311G(d)<sup>30,31</sup> for dithiaporphyrin and LANL2DZ<sup>32</sup> or LANL2TZ<sup>33</sup> for the metals, or a non-mixed basis-set DEF2-TZVP for all atoms (with appropriate pseudopotentials).<sup>34</sup> All used basis-sets and pseudopotentials were recovered from the Basis Set Exchange library.<sup>35</sup> All DFT calculations were performed with the Gaussian A16 program package.<sup>36</sup>

‡ Cobalt could have been included in this study, but it is known experimentally to form complexes with a quite large magnetic anisotropy due to the orbital magnetic momentum, which require multi reference methods to be correctly modelled. Since we were interested in running DFT calculations here (single reference), we thus preferred to avoid Co.



The performance of an organic chromophore is affected by the energy barrier for the charge injection process. Ionization potential (IP), electron affinity (EA), and reorganization energy ( $\lambda$ ) are the key parameters that determine the energy barrier for the charge injection process of organic molecules.<sup>37</sup>

Electron affinities and vertical ionization potentials have been computed by using eqn (1) and (2).<sup>38</sup>

$$EA = E_0 - E^- \quad (1)$$

$$IP = E^+ - E_0 \quad (2)$$

where IP and EA are the vertical ionization potentials and electron affinities in (eV), respectively;  $E^+$  is the energy in cationic state;  $E_0$  is the energy in neutral state;  $E^-$  is the energy in anionic state, all computed at the equilibrium geometry of the neutral form. Adequate ionization potentials and electron affinity values are needed to ensure an effective transfer of electrons and holes.

The HOMO and LUMO analyses are carried out to explain the molecular characteristics of molecules.  $\Delta\epsilon_{L-H}$  is defined as the electronic gap energy and it is given by the eqn (3)

$$\Delta\epsilon_{L-H} = \epsilon_{LUMO} - \epsilon_{HOMO} \quad (3)$$

To more thoroughly investigate the performance of the organic pigment and acceptor, other key parameters should be studied, namely the open-circuit voltage  $V_{OC}$  which has been calculated from the following expression:<sup>39</sup>

$$V_{OC} = |\epsilon_{HOMO}^D| - |\epsilon_{LUMO}^A| - 0.3 \quad (4)$$

where the labels D and A refer, respectively, to the donor and acceptor molecules. The 0.3 eV value is an empirical factor. An increase in the value of  $V_{OC}$  leads to a high efficiency of the solar cells.

In addition, a favourable energy alignment of HOMO–LUMO orbitals is required (conditions (5) and (6)):<sup>40</sup>

$$\epsilon_{LUMO}^A < \epsilon_{LUMO}^D \quad (5)$$

$$\epsilon_{HOMO}^A < \epsilon_{HOMO}^D \quad (6)$$

Condition (5) describes the injection process, as expressed by the free enthalpy of injection ( $\Delta G_{inj}$ ) as follows:

$$\Delta G_{inj} = \epsilon_{LUMO}^A - \epsilon_{LUMO}^D \quad (7)$$

where  $\Delta G_{inj}$  describes the injection of the excited electron from the pigment's LUMO to the LUMO of the acceptor. Small values of  $\Delta G_{inj}$  imply a thermodynamically favoured injection.<sup>41</sup>

The molecules which exhibit the smallest values of reorganization energies show large mobilities of charge transfer between the donor and acceptor units of devices.<sup>37</sup> The reorganization energies for electrons and holes are determined as follows:

(i) Electron reorganization energy:

$$\lambda_e = [E_0^- - E_-] + [E_0^0 - E_0] \quad (8)$$

(ii) Hole reorganization energy:

$$\lambda_h = [E_0^+ - E_+] + [E_0^0 - E_0] \quad (9)$$

where  $E_0^-/E_0^+$  is the energy of anion/cation at the neutral ground state geometry,  $E_-/E_+$  are the energy of the relaxed anion/cation species,  $E_0^0/E_0^0$  are the ground state energy of the neutral molecule in anionic/cationic geometry, respectively, and  $E_0$  is energy of neutral molecules at the ground state.

Using the ground state optimized geometries, the 35 lowest electronic excitation energies, their vertical transition energy ( $\Delta E_{0-0}$ ) and oscillator strengths ( $f$ ) were computed using the TD-DFT method,<sup>42–44</sup> again with B3LYP and CAM-B3LYP functionals and the DEF2-TZVP basis set. The optical UV-visible absorption spectra were simulated by considering Gaussian functions and a FWHM of 0.1 eV. Besides, Natural Transition Orbital analysis (NTO) was carried out in order to investigate the most important transition for each complex, which allows to study each transition individually and to characterize the associated electron density distortion in a very compact manner (pairs of donor and acceptor orbitals).<sup>45</sup>

### Excited state charge transfer (ESCT)

For push–pull systems like metallo-porphyrins, the UV absorption spectra are dominated by charge transfer excitations and the latter have been analysed in more detail.<sup>46</sup>

Each transition is characterized by excitation energy, between the ground (0) and  $n$ th excited states, ( $\Delta E_{opt} = \Delta E_{0n} = E_n - E_0$ ), and is associated to a transition dipole moment:

$$\mu_{0n} = 0|\hat{\mu}|n \quad (10)$$

Radiative processes are governed by the oscillator strength of the transition, which is related to the transition dipole moment:

$$f_{0n} = \frac{2}{3} \Delta E_{0n} |\mu_{0n}|^2 \quad (11)$$

The CT character of the excitations can be determined from an analysis of the difference of electronic density between the ground and excited states,  $\Delta\rho(\vec{r})$ , following the procedure described by Le Bahers *et al.*<sup>46</sup> Using this method, the distance between the barycentre of the negative and positive values of  $\Delta\rho(\vec{r})$  defines the charge-transfer distance ( $d_{CT}$ ), their integration over the whole space gives the amount of charge transferred ( $q_{CT}$ ), while their product gives the CT dipole moment,

$$\Delta\mu_{CT} = q_{CT} \times d_{CT} \quad (12)$$

It may however be noted that this approach is expected to fail in the case of symmetric molecules, since barycentres of the electron density depletion and relocation areas will collapse. This issue was recently acknowledged by Ciofini *et al.*, who proposed an alternative charge transfer index,  $^A D_{CT}$ . The latter is based on the computation of a partial charge-transfer distance, restricted on the subset of atoms that constitutes the smallest asymmetric fragment of the molecule. By this symmetry



restriction and a careful design of a weight function to piece back together all molecular fragments, it is then possible to evaluate a molecular symmetry-adequate charge distance.<sup>47</sup>

## Results and discussion

### Methodological calibration

We initially looked for an appropriate level of theory for our study. To do so, we first evaluated how efficiently the selected functionals could reproduce the geometry of the starting complex  $\text{Ru}(\text{S}_2\text{TTP})\text{Cl}_2$ . We provide in Table S1 in ESI† the most relevant geometrical parameters for all considered functionals (B3LYP, CAM-B3LYP, M05, M05-2X,  $\omega\text{B97x-D}$ ), used in conjunction with the def2-TZVP basis set (and effective core potential for Ru), as well as the experimental values obtained from X-ray diffraction. In line with experimental data, a singlet multiplicity was considered for  $\text{Ru}(\text{S}_2\text{TTP})\text{Cl}_2$ .

As one may note, all functionals offer a quite comparable reproduction of the experimental structure, small deviations being observed. B3LYP offering the shortest computation time, we thus opted for this functional for the second part of the calibration.

In the second stage, we then evaluated the impact of the basis set, again against the reproduction of the experimental structure. Three basis sets and pseudopotentials were considered here:

- 6-311G(d) for all atoms of the dithiaporphyrin and a double-zeta quality basis set and associated pseudopotential for the Ru atom (LanL2DZ);
- 6-311G(d) for all atoms of the dithiaporphyrin and a triple-zeta quality basis set and associated pseudopotential for the Ru atom (LanL2TZ);

- A triple-zeta basis sets with polarisation functions and adapted pseudopotential for the Ru atom (def2-TZVP).

We report in Table S2 in ESI† the same geometrical parameters as previously. Here again a limited impact of the level of theory is observed, although quite expectedly the deviations are larger when the basis set on Ru is only of double-zeta quality. A more complete analysis on all geometrical parameters reveals that root-mean-square deviations (RMSD) are in all cases quite low, but the best agreement to the experimental structure is reported for the “homogeneous” basis set def2-tzvp (RMSD of 0.03 against 0.07 for the split 6-311G(d)/LanL2DZ or LanL2TZ) (Fig. 1).

At this stage, B3LYP/def2-TZVP thus appears to be a quite efficient level of theory for this study. We then finally evaluated its efficiency in the reproduction of the absorption spectrum of  $\text{Ru}(\text{S}_2\text{TTP})\text{Cl}_2$ . We report in Fig. 2 the spectrum we obtain at the TD-B3LYP/def2-TZVP level of theory, in vacuum (for the first 35 excited states). We also report in inset the experimental electronic absorption spectrum of  $\text{Ru}(\text{S}_2\text{TTP})\text{Cl}_2$ , measured in a  $\text{CH}_2\text{Cl}_2$  solution at room temperature by Hung *et al.*<sup>19</sup> As one may note, computed data meets the experimental one, up to a slight upshift. Indeed, we retrieve a near-IR band with a moderate absorption intensity (maximum at 782 nm against 810 nm experimentally), a low absorption massif around 600 nm (peaks at 586 and 602 nm, experimental peaks at 563 and 594 nm), and a high-intensity Soret band from 472 to 511 nm.

Overall, (TD-DFT)B3LYP/def2-TZVP seems to confirm as an adequate level of theory for our study.

### Effect of the nature of the transition metals on the geometrical and electronic structures

The level of theory being fixed, we then optimized the geometries of the different complexes  $\text{M}(\text{S}_2\text{TTP})\text{Cl}_2$  (Scheme 2) ( $\text{M}(\text{II}) =$

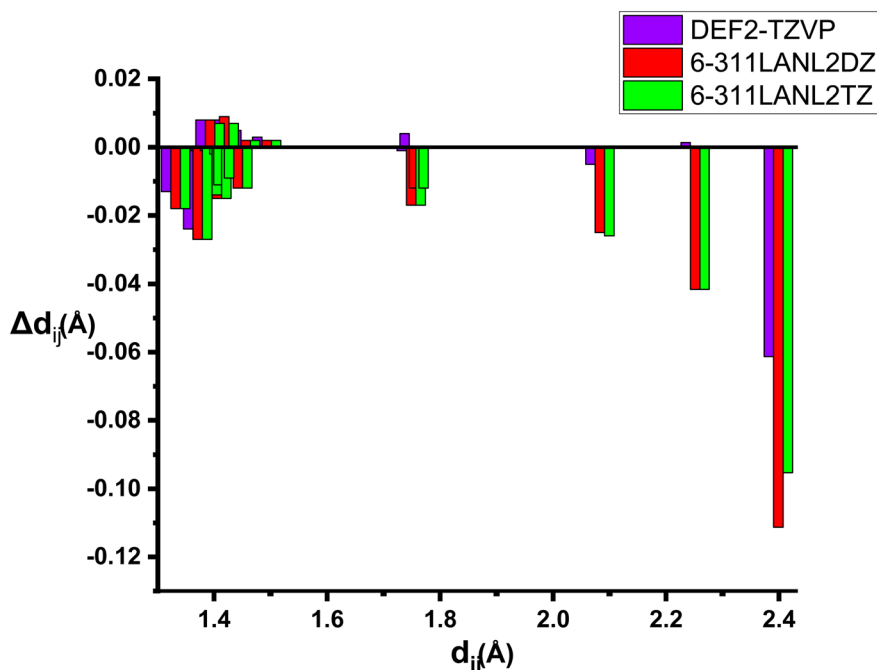


Fig. 1 Bond distances deviations for  $\text{Ru}(\text{S}_2\text{TTP})\text{Cl}_2$  with several basis-sets with B3LYP functional  $\Delta d_{ij} = d_{ij}(\text{experimental}) - d_{ij}(\text{theoretical})$ .



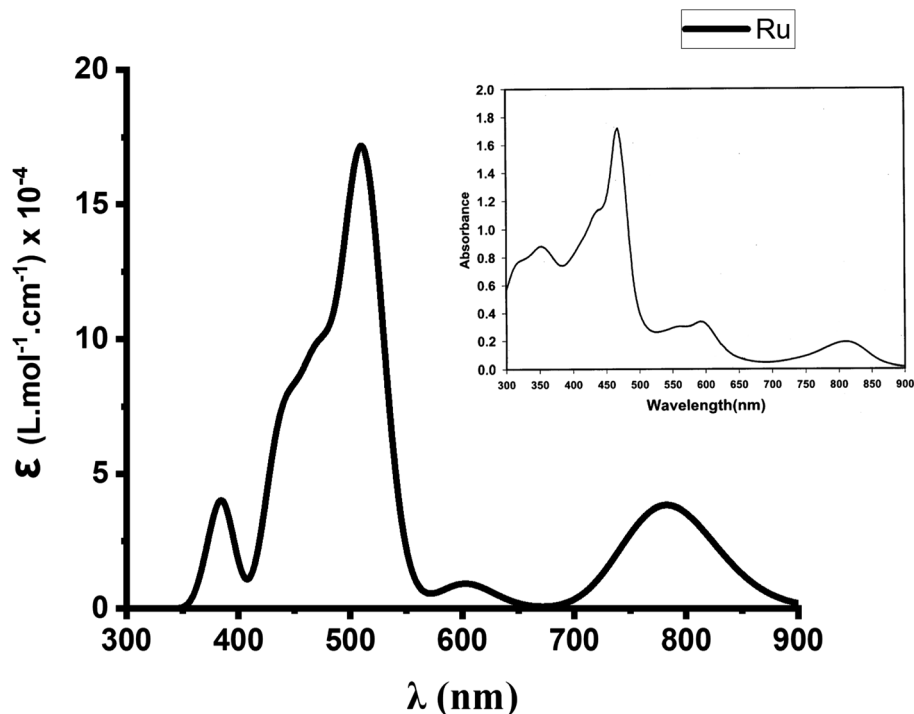


Fig. 2 Experimental (inset) and calculated UV-visible spectra (35 excited states, Gaussian broadening) of  $\text{Ru}(\text{S}_2\text{TTP})\text{Cl}_2$  at the TD-B3LYP/DEF2-TZVP level. UV-vis peak half-width at half height 0.1 eV.

Table 1 Computed bond lengths (in Å) and dihedral angle (in °) for  $\text{M}(\text{S}_2\text{TTP})\text{Cl}_2$  at the B3LYP/DEF2-TZVP level (M = Mn, Fe, Ni, Cu and Ru)

| M                | Parameter                            | Experimental | DFT   |       |       |
|------------------|--------------------------------------|--------------|-------|-------|-------|
| Ru <sup>19</sup> | 2S + 1                               | —            | 1     | 3     | 5     |
|                  | Ru–N                                 | 2.082        | 2.082 | 2.085 | 2.113 |
|                  | Ru–S                                 | 2.251        | 2.247 | 2.280 | 2.296 |
|                  | Ru–Cl                                | 2.399        | 2.458 | 2.396 | 2.394 |
|                  | $\Delta E$ (kcal mol <sup>−1</sup> ) | —            | 0 3.5 | 38.5  |       |
|                  | Dihedral angle of thiophen ring      | 12.7         | 12.4  | 14.5  | 10.0  |
| Mn               | 2S + 1                               | —            | 4     | 6     |       |
|                  | Mn–N                                 | —            | 2.037 | 2.147 |       |
|                  | Mn–S                                 | —            | 2.454 | 2.386 |       |
|                  | Mn–Cl                                | —            | 2.364 | 2.591 |       |
|                  | $\Delta E$ (kcal mol <sup>−1</sup> ) | —            | 16.1  | 0     |       |
|                  | Dihedral angle of thiophen ring      | —            | 11.7  | 12.6  |       |
| Fe               | 2S + 1                               | —            | 1     | 3     | 5     |
|                  | Fe–N                                 | —            | 2.061 | 2.081 | 2.143 |
|                  | Fe–S                                 | —            | 2.181 | 2.184 | 2.335 |
|                  | Fe–Cl                                | —            | 2.407 | 2.656 | 2.506 |
|                  | $\Delta E$ (kcal mol <sup>−1</sup> ) | —            | 9.6   | 11.8  | 0     |
|                  | Dihedral angle of thiophen ring      | —            | 12.1  | 12.3  | 11.3  |
| Ni               | 2S + 1                               | —            | 1     | 3     |       |
|                  | Ni–N                                 | —            | 2.002 | 2.114 |       |
|                  | Ni–S                                 | —            | 2.362 | 2.322 |       |
|                  | Ni–Cl                                | —            | 2.321 | 2.552 |       |
|                  | $\Delta E$ (kcal mol <sup>−1</sup> ) | —            | 22.2  | 0     |       |
|                  | Dihedral angle of thiophen ring      | —            | 11.2  | 11.9  |       |
| Cu               | 2S + 1                               | —            | 2     |       |       |
|                  | Cu–N                                 | —            | 2.059 |       |       |
|                  | Cu–S                                 | —            | 2.360 |       |       |
|                  | Cu–Cl                                | —            | 2.560 |       |       |
|                  | $\Delta E$ (kcal mol <sup>−1</sup> ) | —            | —     |       |       |
|                  | Dihedral angle of thiophen ring      | —            | 11.8  |       |       |

Mn, Fe, Ni, Cu). For each complex, we have considered different spin multiplicities. When considering the Mn(II) complex, which forms a quasi-octahedral geometry, it is anticipated that the valence configuration will be  $4s^0 3d^5$ , resulting in three

distinct electronic configurations determined by the octahedral crystal field:  $(t_{2g})^3(eg)^2$ ,  $(t_{2g})^4(eg)^1$ ,  $(t_{2g})^5(eg)^0$ , hence three spin multiplicities can be proposed following Hund's rule (sextet, quartet or doublet).

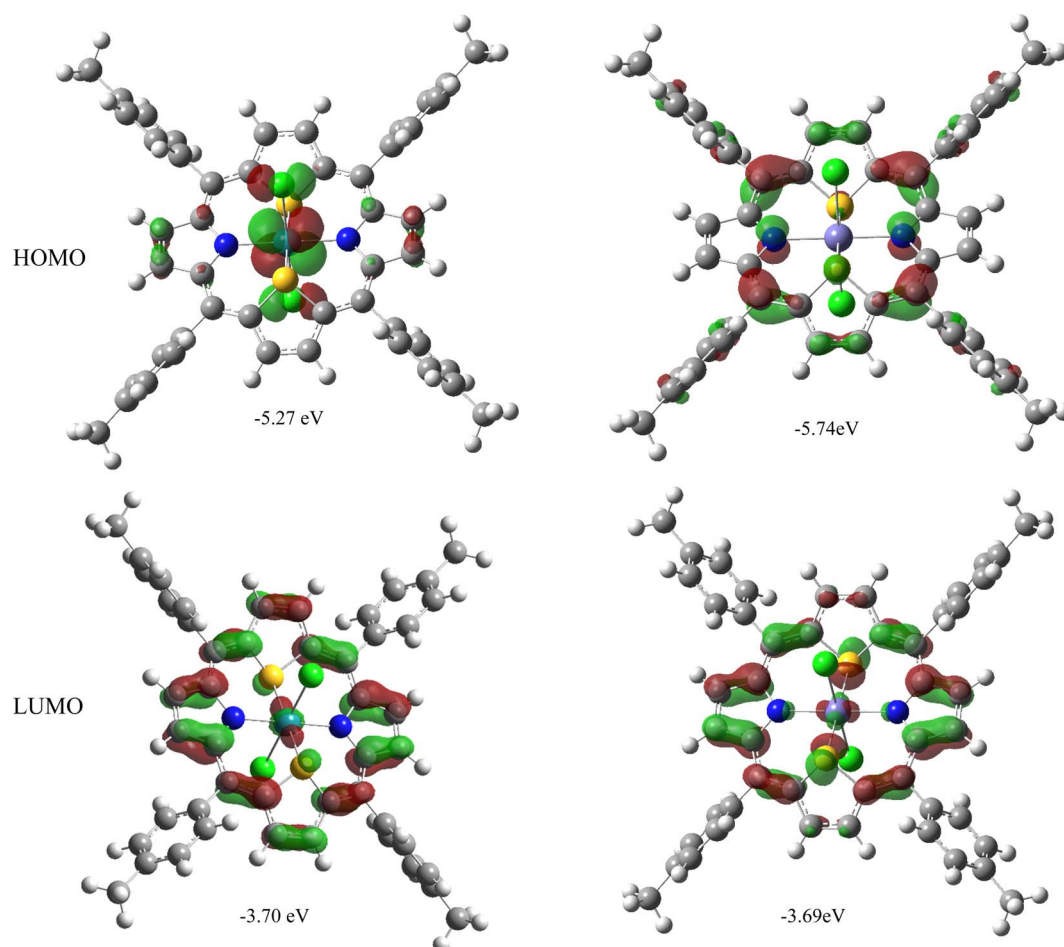
The multiplicity associated with the lowest ground state energy is considered for the next calculations. The geometrical parameters such as bond lengths and dihedral angles characterizing the distortion of the thiophene as well as the energy of the ground state for each multiplicity are summarized in Table 1. Similar information for  $\text{Ru}(\text{S}_2\text{TTP})\text{Cl}_2$  are also given for the sake of comparison.

It may be noted from Table 1 that coordination environments for all metal centres are expectedly similar; noticeably the dihedral angle in the dithiaporphyrin moiety is only marginally impacted by the nature of the metal. This angle being instrumental in the electronic delocalisation within the dithiaporphyrin and between the metal and the dithiaporphyrin,<sup>48</sup> we may at this stage expect the optical properties of the  $\text{M}(\text{S}_2\text{TTP})\text{Cl}_2$  series will be quite comparable.

Additionally, we note that open-shell electronic configurations appear to become the most stable ones for the Mn–Ni series. This was quite expected (energy gaps tending to decrease when going up the periodic table, and the value for Ru being

**Table 2** Frontier molecular orbital energies of  $\text{M}(\text{S}_2\text{TTP})\text{Cl}_2$  and electronic gap energy  $\Delta\epsilon_{\text{L-H}}$  calculated at B3LYP/DEF2-TZVP level

| M              | 2S + 1 | $\epsilon_{\text{HOMO}}$ (eV) | $\epsilon_{\text{LUMO}}$ (eV) | $\Delta\epsilon_{\text{L-H}}$ (eV) |
|----------------|--------|-------------------------------|-------------------------------|------------------------------------|
| Ru             | 1      | −5.27                         | −3.70                         | 1.57                               |
| Mn             | 6      | $\alpha$ −5.59                | −3.50                         | 2.09                               |
|                |        | $\beta$ −5.77                 | −3.72                         | 2.05                               |
| Fe             | 5      | $\alpha$ −5.75                | −3.45                         | 2.30                               |
|                |        | $\beta$ −5.74                 | −3.69                         | 2.05                               |
| Ni             | 3      | $\alpha$ −5.80                | −3.54                         | 2.27                               |
|                |        | $\beta$ −5.78                 | −3.81                         | 1.97                               |
| Cu             | 2      | $\alpha$ −5.78                | −3.54                         | 2.24                               |
|                |        | $\beta$ −5.76                 | −3.87                         | 1.90                               |
| A <sub>1</sub> | 1      | −5.82                         | −3.59                         | 2.23                               |
| A <sub>2</sub> | 1      | −5.72                         | −3.58                         | 2.14                               |
| A <sub>3</sub> | 1      | −5.99                         | −3.97                         | 2.02                               |
| A <sub>4</sub> | 1      | −5.96                         | −3.77                         | 2.19                               |
| A <sub>5</sub> | 1      | −6.03                         | −3.91                         | 2.21                               |
| A <sub>6</sub> | 1      | −5.94                         | −3.71                         | 2.23                               |



**Fig. 3** Representation of HOMO and LUMO for Ru (left) and Fe (right) complexes. Isovalue: 0.04 u.a.



already low). Eventually, the spin multiplicities that relate to the most stable geometries are quintet for Fe, triplet for Ni, doublet for Cu and sextet for Mn.

We provide in Table 2 the computed frontier molecular orbital energies (distinguishing spin up and spin down electrons in the case of open-shell systems) for the  $M(S_2TTP)Cl_2$  complexes and all considered acceptor molecules (*vide infra*).

It may be noted that the electronic gap energies (defined here as the energy difference between the LUMO and HOMO) are larger in all  $M(S_2TTP)Cl_2$  complexes (with  $M = Mn, Fe, Ni, Cu$ ) than in  $Ru(S_2TTP)Cl_2$ , by approximately 0.5 eV. This effect is quite expected in the framework of the Hard and Soft Acids and Bases theory of Pearson:<sup>49</sup> as one goes down the periodic table, chemical hardness (which can be approximated as the electronic gap energy)<sup>50</sup> is expected to drop, as observed here. In fact, if we focus on the iron and ruthenium cases (which are isoelectronic), we observe the variation in the gap energy is seemingly largely due to the HOMO, since LUMOs are found at quite comparable energies. A closer look at these molecular orbitals (MOs), represented in Fig. 3, shows that the HOMOs of Ru and Fe complexes are different ( $\pi$  with no metal contribution in the case of Fe,  $4d + \pi$  for the Ru complex), while the LUMOs are comparable (mostly  $\pi$  with a little contribution from the metal), explaining the observed trends.

### Optical properties

In order to confirm their potential in a heterojunction solar cell, the optical parameters of  $M(S_2TTP)Cl_2$  ( $M = Mn, Fe, Ni, Cu$ ) were then calculated at the TD-DFT level.

We provide in Fig. 4 the calculated absorption spectra for each complex  $M(S_2TTP)Cl_2$ , at the most stable multiplicities and compared to that of  $Ru(S_2TTP)Cl_2$ .

We notice that the Soret band around 500 nm is always present for all complexes and slightly shifts depending on the metal, which is expected since this transition is essentially a ligand-centred  $\pi-\pi^*$  transition and thus is only weakly influenced by the coordination. We also report that all metal-complexes showed a higher molar absorption coefficient ( $\epsilon$ ) value compared to the Ru complex.

Encouraged by the efficient reproduction of the experimental spectrum for the ruthenium complex, we conducted Natural Transition Orbitals (NTO) analysis at the same level of theory in order to investigate electron contributions in excited states of the Soret band for each complex. The obtained results are summarized in Table 3. For  $Ru(S_2TTP)Cl_2$  the principal excited state (ES9) that describes the Soret band ( $\epsilon = 1.8 \times 10^5 \text{ L mol}^{-1} \text{ cm}^{-1}$  and the highest oscillator strength for this complex) can be divided into two main electron density displacements (Scheme S1†). Both principally involve a reorganization of the  $\pi$  electron cloud (as expected for the Soret band), but additionally comprise a ligand-to-metal charge transfer (MLCT), with an electron transfer to a S/Ru and N/Ru anti-bonding accepting orbital (respectively accounting for 0.55 and 0.44 electrons). Similarly, Mn, Fe, Ni and Cu complexes also exhibit a ligand to metal charge transfer in addition to a delocalization within the dithiaporphyrin core.

The principal difference is the slightly higher oscillator strength obtained following the substitution of ruthenium by

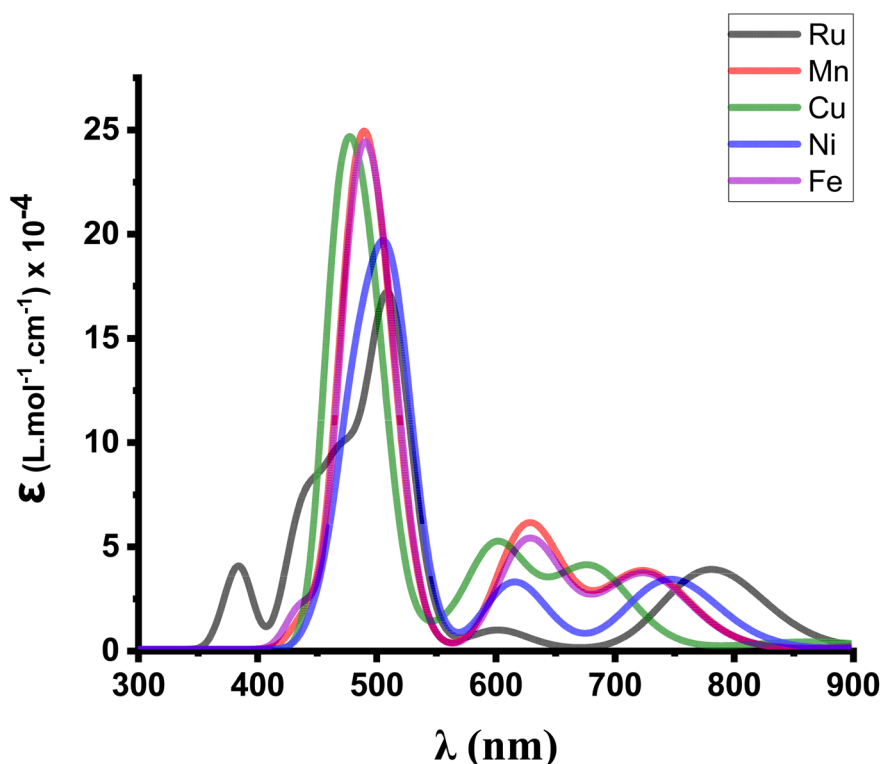


Fig. 4 Modeled UV-visible absorption spectra of  $M(S_2TTP)Cl_2$  ( $M = Mn, Fe, Ni, Cu, Ru$ ) at the TD-B3LYP/DEF2-TZVP level.

**Table 3** NTO analysis for major excited states for  $M(S_2TTP)Cl_2$  at B3LYP/DEF2-TZVP level, Energy of vertical transition  $\Delta E_{0-0}$ , oscillator strength  $f$  and light harvesting efficiency (LHE)

|    | State number | $\lambda$ (nm)/ $\Delta E_{0-0}$ (eV) | Spin     | Transition                                                                       | Occupancy | $f$  | LHE  |
|----|--------------|---------------------------------------|----------|----------------------------------------------------------------------------------|-----------|------|------|
| Ru | 9            | 511, 1/2, 43                          | —        | $\pi \rightarrow \pi^*$<br>+S $\rightarrow$ Ru                                   | 0.55      | 1.23 | 0.94 |
| Mn | 14           | 503.4/2.48                            | $\alpha$ | $\pi \rightarrow \pi^* + (nN \rightarrow d_{z^2}Ru)$                             | 0.44      | 1.00 | 0.9  |
|    |              |                                       |          | $\pi \rightarrow \pi^* + (N \rightarrow Mn)$                                     | 1.09      |      |      |
|    |              |                                       |          | $\pi \rightarrow \pi^* + (S \rightarrow Mn)$                                     | 0.18      |      |      |
|    | 20           | 481.2/2.58                            | $\beta$  | $\pi \rightarrow \pi^* + (Ni \rightarrow S)$                                     | 0.50      | 1.26 | 0.95 |
|    |              |                                       |          | $\pi \rightarrow \pi^* (nN \rightarrow d_{z^2}Ni, nS \rightarrow d_{z^2}Ni)$     | 0.24      |      |      |
|    |              |                                       |          | $\pi \rightarrow \pi^* + (N \rightarrow Mn)$                                     | 1.13      |      |      |
| Fe | 18           | 504.9/2.46                            | $\alpha$ | $\pi \rightarrow \pi^* + (S \rightarrow Mn)$                                     | 0.13      | 1.06 | 0.91 |
|    |              |                                       |          | $\pi \rightarrow \pi^* + (N \rightarrow Mn)$                                     | 0.6       |      |      |
|    |              |                                       |          | $\pi \rightarrow \pi^* + (S \rightarrow Mn)$                                     | 0.12      |      |      |
|    | 22           | 481.5/2.58                            | $\beta$  | $\pi \rightarrow \pi^* + (nN \rightarrow d_{z^2}Fe)$                             | 1.12      | 1.24 | 0.94 |
|    |              |                                       |          | $\pi \rightarrow \pi^*$                                                          | 0.20      |      |      |
|    |              |                                       |          | $\pi \rightarrow \pi^* + (nS \rightarrow d_{z^2}Fe)$                             | 0.54      |      |      |
| Ni | 20           | 512.9/2.42                            | $\alpha$ | $\pi \rightarrow \pi^* + (nN \rightarrow d_{z^2}Fe)$                             | 0.15      | 1.22 | 0.94 |
|    |              |                                       |          | $\pi \rightarrow \pi^*$                                                          | 1.08      |      |      |
|    |              |                                       |          | $\pi \rightarrow \pi^* + (nN \rightarrow d_{z^2}Fe)$                             | 0.14      |      |      |
|    | 25           | 483.3/2.57                            | $\beta$  | $\pi \rightarrow \pi^* (nN \rightarrow d_z \& Fe)$                               | 0.64      | 0.78 | 0.84 |
|    |              |                                       |          | $\pi \rightarrow \pi^* (nS \rightarrow d_{z^2}Fe)$                               | 0.14      |      |      |
|    |              |                                       |          | $\pi \rightarrow \pi^* + (nN \rightarrow d_{z^2}Ni)$                             | 0.96      |      |      |
| Cu | 24           | 496.0/2.5                             | $\alpha$ | $\pi \rightarrow \pi^* + (nS \rightarrow d + z^2Ni)$                             | 0.32      | 1.04 | 0.91 |
|    |              |                                       |          | $\pi \rightarrow \pi^* + (Ni \rightarrow S)$                                     | 0.50      |      |      |
|    |              |                                       |          | $\pi \rightarrow \pi^* (nN \rightarrow d_{z^2}Ni, nS \rightarrow d_{z^2}Ni)$     | 0.24      |      |      |
|    | 27           | 473.2/2.62                            | $\alpha$ | $\pi \rightarrow \pi^* + (N \rightarrow Ni)$                                     | 0.78      | 0.63 | 0.77 |
|    |              |                                       |          | $\pi \rightarrow \pi^* + (S \rightarrow Ni)/(N \rightarrow Ni)$                  | 0.16      |      |      |
|    |              |                                       |          | $\pi \rightarrow \pi^* + (S \rightarrow Ni)$                                     | 0.73      |      |      |
|    | 29           | 468.6/2.65                            | $\beta$  | $\pi \rightarrow \pi^* + (N \rightarrow Ni)$                                     | 0.27      | 0.76 | 0.82 |
|    |              |                                       |          | $\pi \rightarrow \pi^* + (S \rightarrow Mn)$                                     | 0.13      |      |      |
|    |              |                                       |          | $\pi \rightarrow \pi^* (nS \rightarrow d_{z^2}Cu)$                               | 0.67      |      |      |
|    |              |                                       | $\alpha$ | $\pi \rightarrow \pi^* (nN \rightarrow d_{z^2}Cu)$                               | 0.16      | 0.63 | 0.77 |
|    |              |                                       |          | $\pi \rightarrow \pi^* (nN \rightarrow d_{z^2}Cu)$                               | 0.97      |      |      |
|    |              |                                       |          | $\pi \rightarrow \pi^* (nS \rightarrow d_{z^2}Cu)$                               | 0.18      |      |      |
|    |              |                                       | $\beta$  | $\pi \rightarrow \pi^* (nN \rightarrow d_{z^2}Cu)$                               | 0.50      | 0.76 | 0.82 |
|    |              |                                       |          | $\pi \rightarrow \pi^* (Cu \rightarrow S)$                                       | 0.4       |      |      |
|    |              |                                       |          | $\pi \rightarrow \pi^* (nN \rightarrow Cu, Cl \rightarrow Cu, Cu \rightarrow S)$ | 0.57      |      |      |
|    |              |                                       | $\alpha$ | $\pi \rightarrow \pi^* (N \rightarrow Cu)$                                       | 0.40      | 0.76 | 0.82 |
|    |              |                                       |          | $\pi \rightarrow \pi^* (Cl \rightarrow Cu, N \rightarrow Cu)$                    | 0.56      |      |      |
|    |              |                                       |          | $\pi \rightarrow \pi^* (Por \rightarrow Cu)$                                     | 0.31      |      |      |
|    |              |                                       | $\beta$  | $\pi \rightarrow \pi^* + (nN \rightarrow d_{z^2}Cu)$                             | 0.52      | 0.76 | 0.82 |
|    |              |                                       |          | $\pi \rightarrow \pi^* (nN \rightarrow d_{z^2}Cu)$                               | 0.48      |      |      |

iron and manganese, (about +0.01 and +0.03, respectively), giving these new complexes comparable to slightly higher performance than the Ru complex. This small difference slightly improves the LHE value (Table 3). Ni and Cu show oscillator strength values very close to that of Ru and consequently close LHE values, which makes these metals potentially efficient in converting light into electrical energy at a lower cost. However, these optical properties resulting from TD-DFT are not sufficient to define which complex is the most efficient candidate for the design of a solar cell. A detailed CT character analysis should indeed be conducted to evaluate whether these complexes afford significant charge separation in the excited state. This can be achieved by computing the CT dipolar moment (eqn (13)).

### Excited states CT analysis

In this part, we describe the special properties of charge transfer (CT) states. CTs are generally present in molecules in which

electro-donating and electro-accepting moieties are electronically connected *via* a conjugated skeleton. Porphyrins can present a push-pull structure.<sup>51</sup> In push-pull molecules, the most fundamental process is the intramolecular charge transfer (ICT), whereby electron density is transferred from the donor to the acceptor through a  $\pi$ -conjugated linker.<sup>52</sup> Besides the highest LHE values, in order to assign a dye as efficient for PV applications, the excited states should present a large charge transfer, which can be evaluated by using the TD-DFT method.<sup>53</sup> According to the data in Table 4, we have found that all  $M(S_2TTP)Cl_2$  complexes display larger charge transfer in the Soret band than  $Ru(S_2TTP)Cl_2$ , as indicated by the  $q_{CT}$  values. Among all complexes,  $Ni(S_2TTP)Cl_2$  possesses the highest  $q_{CT}$  value, which reveals large charge reorganization between the ground state (GS) and ES. Nevertheless, this alone is not sufficient to characterise efficient CT, as discussed by Le Bahers *et al.*<sup>46</sup> In addition, one should study the electron density



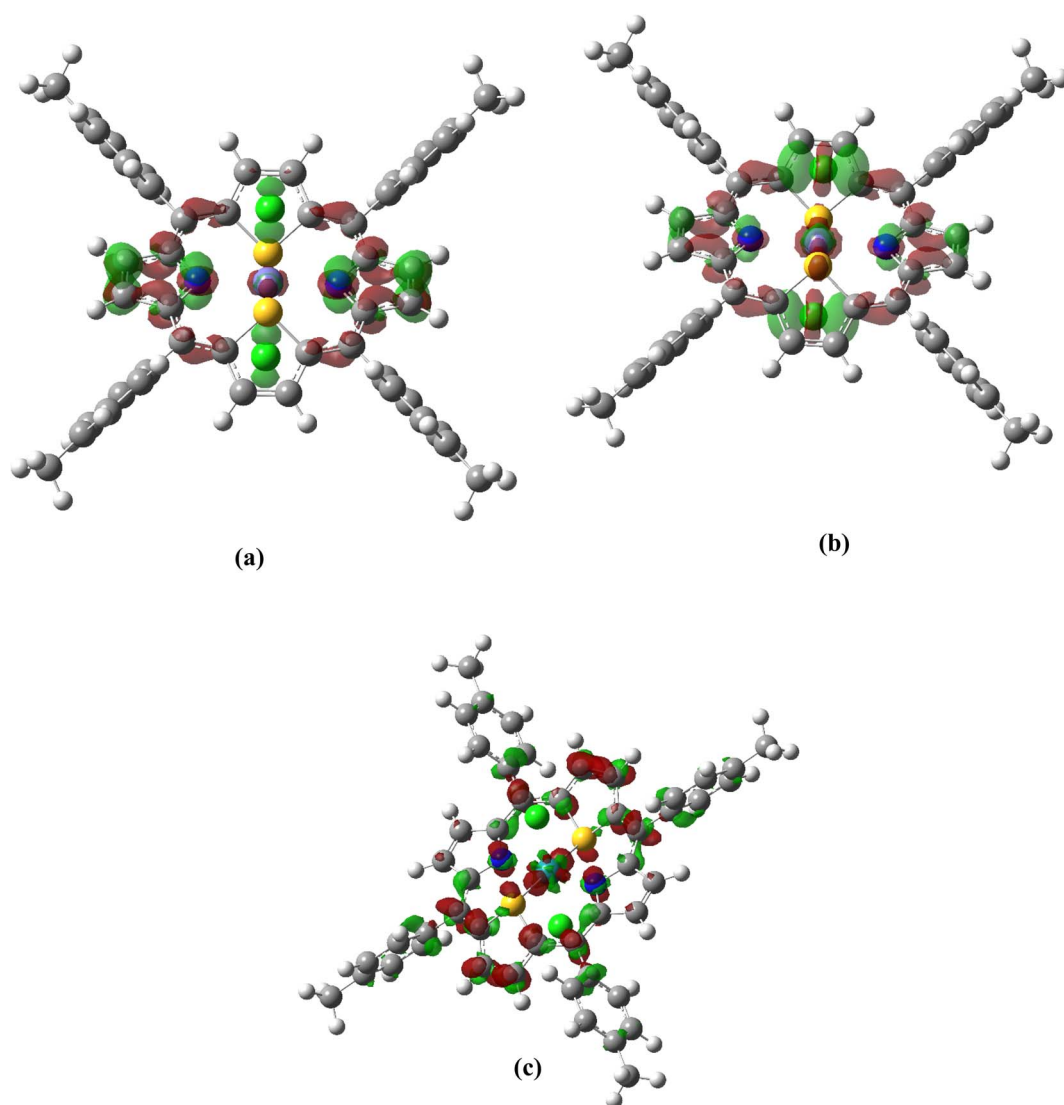
**Table 4** Charge transfer parameters of the lowest-energy dominant excitations of the studied metallo-porphyrin. [ $q_{CT}$  (e),  $d_{CT}$  (Å),  $\Delta\mu_{CT} = q_{CT} \times d_{CT}$  (u.a.)] as calculated at the B3LYP/DEF2-TZVP level

| Compound | 2S + 1 | Number of state | $d_{CT}$ | $\Delta\mu_{CT}$ | $q_{CT}$ |
|----------|--------|-----------------|----------|------------------|----------|
| Ru       | 1      | 9               | 0.002    | 0.003            | 0.277    |
| Cu       | 2      | 24              | 0.007    | 0.026            | 0.806    |
|          |        | 27              | 0.004    | 0.006            | 0.315    |
|          |        | 29              | 0.004    | 0.011            | 0.662    |
| Ni       | 3      | 20              | 0.012    | 0.049            | 0.850    |
|          |        | 25              | 0.011    | 0.050            | 0.938    |
| Fe       | 5      | 18              | 0.002    | 0.006            | 0.655    |
|          |        | 22              | 0.006    | 0.024            | 0.793    |
| Mn       | 6      | 14              | 0.017    | 0.058            | 0.697    |
|          |        | 20              | 0.007    | 0.024            | 0.709    |

**Table 5** Calculated  $^A D_{CT}$  for transitions with highest oscillator strength (in Å)

| Complex | Excited state N° | $^A D_{CT}$ (Å) |
|---------|------------------|-----------------|
| Ru      | 9                | 6.3             |
| Fe      | 18               | 4.8             |
|         | 22               | 4.2             |
| Ni      | 20               | 5.5             |
|         | 25               | 4.3             |
| Mn      | 14               | 4.7             |
|         | 20               | 4.1             |
| Cu      | 24               | 4.5             |
|         | 27               | 4.3             |
|         | 29               | 4.2             |

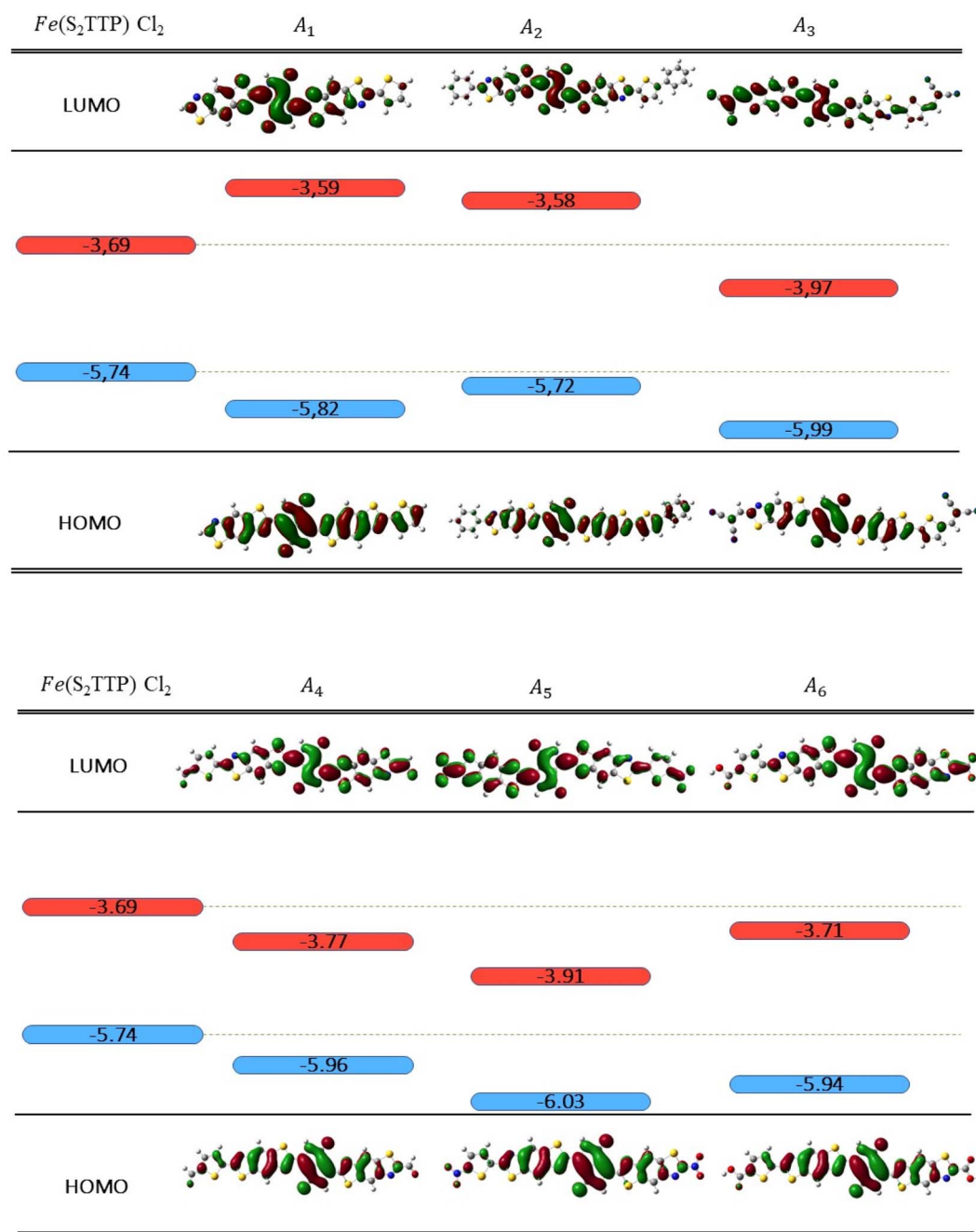
reorganization in space, in order to probe “how far are the electrons reshuffled”. This can in principle be analysed through the charge transfer distance  $d_{CT}$  and/or the CT dipole

**Fig. 5** Excitation-induced TD-DFT/B3LYP/DEF2TZVP total electron density difference [ $\Delta\rho(\vec{r}) = \rho_{\text{excited}}(\vec{r}) - \rho_{\text{ground}}(\vec{r})$ ] for the lowest-energy excited state of: (a) Fe (excited state number 18 with isovalue of 0.002 au), (b) Fe (excited state number 22 with isovalue of 0.002 au) and (c) Ru (excited state number 9 with isovalue of 0.0006 au). (Red: positive value/green: negative value).

momentum, but here an additional difficulty arises: because of the molecular symmetry, most excitations are expected to present very low values of  $d_{CT}$ , even though locally the electron density movement in space can be quite large. Indeed, as can be seen in Table 4, almost all excitations present  $d_{CT}$  values close to 0, while the electron density movements in Fig. 5 are quite significant.

Using the generalized  $^A D_{CT}$  distance proposed by Ciofini *et al.*,<sup>47</sup> we find more relevant charge transfer distances, as listed in Table 5.

The calculated  $^A D_{CT}$  values are consistent for all molecules. We find that, for  $^A D_{CT}$  values ranging from 4.1 to 4.5 Å, the electron density movements are indeed mainly localized on the central ring of dithiaporphyrin surrounding the metal (see Fig. 5 for the case of the Fe complex, see Scheme S1† for all complexes). For higher  $^A D_{CT}$  values, ranging from 4.6 to 6.3 Å, since the accepting NTOs are localized on the dithiaporphyrin ring as well as on the metal, the charge transfer distance increases. In the case of  $Ru(S_2TPP)Cl_2$ , contributions of the peripheral tolyl groups are also observed, which explain the



**Scheme 3** Frontier molecular orbitals energies for  $(Fe(S_2TTP)Cl_2)$ ,  $A_1$ ,  $A_2$ ,  $A_3$ ,  $A_4$ ,  $A_5$  and  $A_6$  at B3LYP/DEF2-TZVP level, in eV. HOMO and LUMO energies are respectively depicted in blue and red (isovalue = 0.02 au).



additional extent of the charge transfer. Overall, all  $M(S_2TTP)Cl_2$  complexes present strong charge transfer in the excited states associated with the Soret band, the largest values being observed in the case of the Ru derivative.

### Study of acceptor tuning

As is well known, exciton excitation, separation, and transfer can be closely correlated with MO energies in the donor and acceptor, particularly with HOMO and LUMO energy levels.<sup>54</sup> Schematically, after the initial excitation of the donor, considered to be in a closed-shell state, relaxation to the first singlet excited state is expected to occur (Kasha's rule). If we assume the MO diagram remains unperturbed by excitation (frozen orbital approximation), this singlet state is constructed by the promotion of one electron from the HOMO to the LUMO. To allow an electron transfer to the acceptor molecule, it is then desirable for the donor LUMO to be located higher in energy than the acceptor LUMO. As shown in Scheme 3, the LUMO of  $A_1$  is too high in energy to allow the transfer of an electron.

In order to tune the energy level of the acceptor's FMOs according to those of the donor, we substituted the hydrogen, located in the  $\alpha$  position of the thiophene and thiazole of  $A_1$  by a phenyl ( $A_2$ ), a dicyanovinyl ( $A_3$ ), a carbonyl ( $A_4$ ), a nitro ( $A_5$ ) and a carboxylic acid ( $A_6$ ) group.

The substitution of R with a phenyl group  $A_2$  affects the delocalization. As seen in Scheme 3, the LUMO is mostly delocalized on the central part of the molecule, but not on the site of the modification, leading to a very slight stabilization of the LUMO at  $A_2$  (0.01 eV). This leads us to exclude  $A_1$  and  $A_2$  as potential acceptors in this work, since their LUMOs remain too high in energy.

On the other hand, the addition of the dicyanovinyl group ( $A_3$ ), which is a strongly attractive group, expectedly stabilizes both LUMO and HOMO, pushing the LUMO at a lower energy than that of  $Fe(S_2TTP)Cl_2$  and all other donor complexes. The same effect is observed for  $A_5$  where a nitro group is added, which makes these two acceptors compatible with all the studied donors. In addition, we notice that the addition of an aldehyde group ( $A_4$ ) stabilizes the LUMO, which makes it compatible with Ru, Fe and Mn complexes, but not enough to form a heterojunction with  $Cu(S_2TTP)Cl_2$  and  $Ni(S_2TTP)$ . Moreover, the low value of  $\Delta G_{inj}$  observed for the  $A_4/Fe$  pair considerably decreases the efficiency of the cell since it is lower than 0.3 eV.<sup>54</sup>

Lastly, the addition of a carboxylic acid group stabilizes the LUMO of the acceptor ( $A_6$ ) but not enough to allow an efficient injection of the electron with any of the selected donors.

In this evaluative part, we have calculated the hole and electron reorganization energies ( $\lambda_h$  and  $\lambda_e$ ) for each  $M(S_2TTP)Cl_2$  with the aim of investigating the hole and electron mobilities and the electron recombination after the injection (Table 6). We have found that Ni and Ru show the lowest  $\lambda_e$  and  $\lambda_h$  values of all donors. According to the Marcus theory<sup>55,56</sup> smaller reorganization energy corresponds to a higher charge hopping rate constant, and hence higher charge mobility. Besides,  $\lambda_e$  of the Fe and Cu complexes is relatively high, while their hole

**Table 6** Computed reorganization energy of electron  $\lambda_e$ , hole reorganization energy  $\lambda_h$ , free energy change of injection  $\Delta G_{inj}$  and open-circuit voltage  $V_{oc}$  (in eV) at B3LYP/DEF2-TZVP

|    | $\lambda_e$ | $\lambda_h$ |
|----|-------------|-------------|
| Cu | 1.08        | 0.78        |
| Ni | 0.28        | 0.26        |
| Fe | 0.94        | 0.35        |
| Mn | 0.35        | 1.18        |
| Ru | 0.28        | 0.21        |

**Table 7** Computed free energy change of injection  $\Delta G_{inj}$  and open-circuit voltage  $V_{oc}$  (in eV) at B3LYP/def2-tzvp for all considered  $M(S_2TTP)Cl_2$ /acceptor pairs

|    | $A_3$    |                  | $A_4$    |                  | $A_5$    |                  | $A_6$    |                  |
|----|----------|------------------|----------|------------------|----------|------------------|----------|------------------|
|    | $V_{oc}$ | $\Delta G_{inj}$ | $V_{oc}$ | $\Delta G_{inj}$ | $V_{oc}$ | $\Delta G_{inj}$ | $V_{oc}$ | $\Delta G_{inj}$ |
| Ru | 1.3      | -0.30            | -1.5     | -0.30            | -1.4     | -0.20            | -1.6     | -0.01            |
| Fe | 1.8      | -0.20            | -2.0     | -0.10            | -1.8     | -0.20            | -2.0     | -0.03            |
| Ni | 1.8      | -0.30            | —        | —                | -1.9     | -0.10            | —        | —                |
| Cu | 1.8      | -0.10            | —        | —                | -1.9     | -0.04            | —        | —                |
| Mn | 1.8      | -0.30            | -2.0     | -0.05            | -1.9     | -0.20            | —        | —                |

reorganization energy, especially for iron, is acceptable with a value of 0.35 eV. Based on these results, we can conclude that Ru and Ni complexes are potential higher mobility materials compared to Fe, Cu and Mn complexes.

We also calculated the open circuit voltage ( $V_{oc}$ ) values and  $\Delta G_{inj}$  for each compatible donor/acceptor pair (based on HOMO/LUMO energy levels) and the obtained results are summarized in Table 7. We notice at first sight that all complexes have a higher  $V_{oc}$  value than Ru which makes recombination phenomenon more likely to be present in the latter. This result is a good confirmation that the substitution of Ru by other metals may be efficient. The highest  $V_{oc}$  are those of the couples  $Fe(S_2TTP)Cl_2/A_6$ ,  $Ru(S_2TTP)Cl_2/A_6$ ,  $Mn(S_2TTP)Cl_2/A_4$ ,  $Fe(S_2TTP)Cl_2/A_4$  and  $Ni(S_2TTP)Cl_2/A_5$ ,  $Cu(S_2TTP)Cl_2/A_5$ . Nevertheless, their injection is very low (lower than 0.3 eV) which makes the injection phenomenon more difficult to be present within the bulk heterojunction solar cell. Assuming that an energy gap of 0.3 eV between the LUMO of the acceptor and the LUMO of the donor is sufficient for efficient charge separation,<sup>54</sup> we conclude that the most promising pairs found in this part are  $Mn(S_2TTP)Cl_2/A_3$ ;  $Ni(S_2TTP)Cl_2/A_3$  and  $Ru(S_2TTP)Cl_2/A_3$ .  $Fe(S_2TTP)Cl_2/A_3$  is also promising considering its high  $V_{oc}$  value, although  $\Delta G_{inj}$  is relatively weak.

## Conclusion

In this work, we investigated the impact of substituting ruthenium with other transition metals in the  $M(S_2TTP)Cl_2$  complex, and characterized the resulting metallo-dithiaphorphyrin series in terms of their optical and electronic properties. Using density functional theory and time-dependent density functional calculations, we determined the most suitable functional and



basis-set for our study. All modeled dithiaporphyrin complexes strongly absorb light in the visible range. Furthermore, we found that Fe and Mn have slightly higher oscillator strengths than Ru, which leads to an increase in the light harvesting efficiency (LHE) value.

In addition to these findings, we calculated the electron reorganization energy  $\lambda_e$ , hole reorganization energy  $\lambda_h$ , free energy change of injection  $\Delta G_{inj}$  and open-circuit voltage  $V_{OC}$  for each complex. Our results showed that Mn(S<sub>2</sub>TTP)Cl<sub>2</sub>/A<sub>3</sub>; Ni(S<sub>2</sub>TTP)Cl<sub>2</sub>/A<sub>3</sub> pairs are the most effective, due to their high LHE value and optimal energy levels of the acceptor's HOMO/LUMO.

Overall, our study provides insights into the design of more efficient donor/acceptor pairs and paves the way for further investigations into the electron injection process in dithiaporphyrin complexes combined with semiconductor surfaces. In this study, we have demonstrated that small molecules based on dithiaporphyrin are potential candidates for applications as donors in BHJ-OSCs and substitution of Ru by cheaper metals improves productivity and lowers cell expenses. Besides, the effective strategy for designing efficient and effective next-generation BHJ-OSC devices is by modifying acceptor units, this is why acceptor A<sub>3</sub> molecule is highly recommended for experimentalists to develop highly efficient solar cell devices. However, the challenge that is presented is to synthesize these complexes experimentally in order to confirm their kinetic and thermodynamic stabilities and evaluate their performance within solar cells offering new opportunities for designing efficient and sustainable energy conversion systems.

## Conflicts of interest

There are no conflicts to declare.

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