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Site-selective photodeposition of Pt on a particulate $\text{Sc-La}_5\text{Ti}_2\text{CuS}_5\text{O}_7$ photocathode: evidence for one-dimensional charge transfer†

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Photodeposition of Pt on the Sc-doped $\text{La}_5\text{Ti}_2\text{CuS}_5\text{O}_7$ (Sc-LTC) photocathode, a visible-light-responsive semiconducting oxysulfide, is accomplished in a solution containing H_2PtCl_6 and $\text{K}_2\text{C}_2\text{O}_4$. Pt particles are selectively deposited on the top surface of rod-like Sc-LTC particles as a result of one-dimensional transfer of photogenerated electrons.

Photoelectrochemical (PEC) H_2 production from water utilizing solar energy has attracted much attention in recent years because of concerns about environmental pollution and energy shortage. A number of semiconductors with visible-light activity have been developed as photoelectrodes for water splitting, because only 4.6% of the solar energy distribution consists of UV light ($\lambda < 400$ nm) whereas visible light ($400 \text{ nm} < \lambda < 800$ nm) constitutes 54.3% of the distribution.^{1–3} The process of PEC water splitting can be divided into three main steps:^{1–4} (1) photon absorption by semiconductors for excitation of electron–hole pairs, (2) migration of minority and majority carriers to, respectively, the photoelectrode/electrolyte interface and the counter electrode/electrolyte interface *via* a backside electrode, and (3) (photo)electrochemical hydrogen and oxygen evolution reactions on the surfaces of the (photo)electrodes. Undesirable charge recombination competes with the above steps. It is therefore essential to improve the charge separation efficiency and charge mobility in a semiconductor photoelectrode in order to suppress charge recombination and improve the efficiency of PEC water splitting.

The diffusion length of minority carriers, which reflects their lifetime and diffusion coefficient, is a critical factor for PEC properties. A semiconductor photoelectrode having a short charge diffusion length has to be small so that excited minority

carriers can reach the surface before recombination. For example, most hematite ($\alpha\text{-Fe}_2\text{O}_3$) photoanodes are prepared at the nanoscale because the hole diffusion length for hematite is 2–4 nm, which is much shorter than that for TiO_2 ($\sim 1 \mu\text{m}$) or WO_3 (~ 100 nm).^{5–7} However, the thinness of the photoelectrode inevitably results in inadequate light absorption. To improve the charge separation efficiency and charge mobility, one idea is to produce functional surface states (*e.g.*, surface microstructure) having an atypical energy potential so that the photogenerated electrons and/or holes are transferred in different directions. Many highly efficient photocatalysts have been reported that are based on this mechanism, such as $\text{NaTaO}_3\text{:La}$,⁸ $\text{Cd}_{0.5}\text{Zn}_{0.5}\text{S}$ ⁹ and anatase/rutile.¹⁰ Another idea is to selectively load cocatalysts on the reduction and/or oxidation site of a photocatalyst, because it is possible that randomly loaded cocatalysts will act as recombination centers for photogenerated electrons and holes.^{11–14} Taking BiVO_4 as a model semiconductor, Li *et al.* reported the enhancement of the photocatalytic or photoelectrochemical water oxidation reaction by controllable deposition of oxidation/reduction cocatalysts on the corresponding oxidation/reduction facets.¹¹

$\text{La}_5\text{Ti}_2\text{CuS}_5\text{O}_7$ (LTC), with an absorption edge wavelength of 650 nm, equal to a band gap of 1.9 eV, was reported by our group as a visible-light-responsive oxysulfide photocatalyst and photocathode for water splitting.^{15–17} The photocatalytic activity for both H_2 and O_2 evolution under sacrificial reagents implies that the valence and conduction band edges of LTC straddle the hydrogen and oxygen evolution redox potentials.^{15,16} It was recently found that the PEC performance of LTC photocathodes was significantly increased by doping with the appropriate amount of Sc (abbreviated as Sc-LTC, hereafter).¹⁷ Thus, we developed a method to load Pt as a hydrogen evolution catalyst on the edge of rod-like Sc-LTC particles selectively *via* PEC reduction in an electrolyte solution containing H_2PtCl_6 and $\text{K}_2\text{C}_2\text{O}_4$. As a result, the hypothetical half-cell solar-to-hydrogen conversion efficiency (HC-STH) near the onset potential of the photocathodic current was improved. Tracking of the Pt nanoparticles loaded on the surface suggested that the Sc-LTC photocathode was capable of one-dimensional charge transfer.

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Sc-LTC powder was prepared by a solid-state reaction (see ESI†). Scanning electron microscopy (SEM) images showed that the produced Sc-LTC comprised highly crystallized rod material with diameters and lengths ranging from 0.7 to 1.4 μm and from 2 to 6 μm (Fig. S1 in the ESI†), respectively, reflecting the crystal structure. Sc-LTC particles were embedded into a Au thin film by the particle transfer (PT) method (Fig. S2 in the ESI†).¹⁸ This Sc-LTC/Au assembly functioned as a photocathode. Pt was loaded on Sc-LTC/Au by PEC reduction in a three-electrode system. The electrolyte solution used for photo-deposition was an aqueous Na_2SO_4 solution (0.1 M, 100 mL) containing $\text{K}_2\text{C}_2\text{O}_4$ (0.1 M) and H_2PtCl_6 (3.5×10^{-6} M for subsequent PEC measurements and 10×10^{-6} M for SEM and EDX measurements). Fig. S3 in the ESI† shows that the Pt deposition process was accomplished in 1 h. For comparison, a sputtering method was employed to deposit Pt particles with an optimized nominal thickness of 1 nm on Sc-LTC/Au.¹⁷ Pt loaded by *in situ* PEC reduction and sputtering will hereafter be abbreviated as Pt-photo and Pt-sputter, respectively. PEC hydrogen evolution reactions were carried out using Pt-photo and Pt-sputter Sc-LTC/Au photocathodes in a fresh aqueous Na_2SO_4 solution (0.1 M, 100 mL) whose pH was adjusted to 10 with a diluted aqueous NaOH solution. The photocathode samples were illuminated with a solar simulator.

Fig. 1(A) shows current–potential curves of Sc-LTC/Au photocathodes with and without Pt as a catalyst for PEC hydrogen evolution under simulated solar illumination (AM 1.5G). Compared with the unmodified sample, the photocathodic current was significantly increased on both Pt-photo and Pt-sputter Sc-LTC/Au photocathodes, and the former showed a higher cathodic current as the electrode potential was varied from 0.0 to 0.9 V vs. RHE. This difference leveled off, especially at potentials more negative than 0 V vs. RHE, the thermodynamic potential for H_2 evolution from water. It is noted that, for a photocathode, a higher photocathodic current at a more positive potential is particularly meaningful for HC-STH and for applications in p/n PEC cells employing another photoanode to accomplish overall water splitting. As shown in Fig. 1(B), the maximum HC-STH was estimated to be 0.07% (at 0.65 V) and 0.05% (at 0.45 V) for Pt-photo and Pt-sputter Sc-LTC/Au photocathodes, respectively.

In previous research, PEC loading of Pt was accomplished by using PtCl_6^{2-} ions as a precursor in the absence of reducing reagents such as $\text{K}_2\text{C}_2\text{O}_4$.^{19–21} However, in our study, Pt could be loaded on Sc-LTC/Au photocathodes slowly and the addition of $\text{K}_2\text{C}_2\text{O}_4$ accelerated the photodeposition. This is presumably due to kinetic reasons rather than thermodynamics, because the potential of the conduction band edge of the Sc-LTC semiconductor is more negative than the hydrogen evolution potential and thus the redox potentials of PtCl_6^{2-} and possible intermediates (see ESI†). $\text{K}_2\text{C}_2\text{O}_4$ was reported to produce strongly reducing $\text{CO}_2^{\bullet-}$ radicals *via* oxidation by electrolysis or photocatalysis.^{22–24} The potential of the $\text{CO}_2^{\bullet-}/\text{CO}_2$ couple was reported to be about -1.96 V vs. SHE. Therefore, $\text{CO}_2^{\bullet-}$ is capable of electron transfer to many metal ions including PtCl_6^{2-} . It is believed that in the present three-electrode PEC reaction system, the $\text{CO}_2^{\bullet-}$ radicals produced on a counter

Fig. 1 Current–potential curves (A) and HC-STH (B) for (a) Sc-LTC/Au, (b) Pt/Sc-LTC/Au-sputter and (c) Pt/Sc-LTC/Au-photo electrodes under solar light irradiation (AM 1.5G). The electrolyte is a 0.1 M aqueous Na_2SO_4 solution (pH = 10).

electrode could reduce PtCl_6^{2-} ions into more reactive intermediates.²³ Such intermediates could be reduced to Pt metal more readily by photogenerated electrons on Sc-LTC photocathodes. Note that, through the same process as that used for Pt loading, other noble metals, such as Rh, Ir, Pd and Ru, were loaded on Sc-LTC as a catalyst; however, these did not lead to as significant an enhancement in PEC H_2 evolution as Pt did.

Tracking the Pt particles deposited on the surface of Sc-LTC can provide information about the reduction sites of this material, namely, the transfer direction of photogenerated electrons. Fig. 2 shows SEM images of Sc-LTC/Au electrodes before and after PEC deposition of Pt. It is found that small Pt particles (most < 10 nm) were deposited on the edge of the Sc-LTC rod upon photoreduction of PtCl_6^{2-} ions, while no such particles were observed on the lateral side of the rod. In addition, Pt particles were selectively loaded on a step when the Sc-LTC rod was broken into half and a step was exposed (Fig. 2d and Fig. S4, ESI†). EDX analysis of Pt was carried out on different sites of a Sc-LTC rod loaded with excess Pt (Fig. 3) to confirm the site-selective photoreduction of Pt nanoparticles. A line scan is plotted over a distance of ca. 1.3 μm to show the distribution of Pt from the middle to top of the Sc-LTC rod. It can be seen from the top right of Fig. 3 that the Pt signals detected on the body of the Sc-LTC rod was within the noise level, whereas a significant increase was observed at the edge of the rod. The Pt concentrations were compared by collecting EDX signals from different parts of the Sc-LTC rod. As tabulated in the table in Fig. 3, the Pt concentration at point a (the edge of a Sc-LTC particle) was higher than the whole area measured, whereas only a negligible Pt signal was detected at point b (the lateral part of

Fig. 2 SEM characterization of Sc-LTC/Au (a and b) and Pt/Sc-LTC/Au (c and d) photoelectrodes. Pt is loaded by photodeposition. All the scale bars are 300 nm.

Fig. 3 EDX analysis of Pt distribution on a Pt/Sc-LTC/Au photoelectrode. Pt is loaded by *in situ* photodeposition. Top right: plot of a line scan. Bottom right: spot and overall content. The scale bar is 1 μm .

the Sc-LTC particle). These observations strongly suggest that, for the rod-like Sc-LTC semiconductor particles, photogenerated electrons transfer along the long axis of the rod (*i.e.*, one-dimensionally) until reaching any transection plane (top or broken parts) where reduction reactions occur.

It may be concerned that the site-selective photodeposition of Pt was due to the selective adsorption of $\text{K}_2\text{C}_2\text{O}_4$ or PtCl_6^{2-} species on different crystal faces of Sc-LTC rods. To exclude such possibilities, photodeposition of Pt and Pd on Sc-LTC was carried out in electrolyte solutions containing PtCl_6^{2-} (without $\text{K}_2\text{C}_2\text{O}_4$), $(\text{NH}_4)_2\text{PdCl}_4$ and $\text{K}_2\text{C}_2\text{O}_4$, respectively. SEM and EDX analyses (Fig. S5 and S6, ESI†) show that, in both cases, Pt and Pd particles were selectively deposited on the top faces of Sc-LTC rods, which confirms that the site-selective photodeposition is caused by the selectivity of charge transfer rather than the selectivity of absorption.

It should be noted that Pt-photo was deposited at the edge of LTC rods although these rods were as long as 6 μm . In the crystal structure of LTC, $\text{TiO}_x\text{S}_{6-x}$ ($x = 4$ or 5) octahedra and CuS_4 tetrahedra are stacked along the *b*-axis (see Fig. S7, ESI†).^{16,25} DFT calculations revealed that the valence band edge of LTC composed of Cu 3d and S 3p hybrid orbitals was localized around CuS_4 tetrahedra and the conduction band edge composed of Ti 3d orbitals was localized around $\text{TiO}_x\text{S}_{6-x}$ ($x = 4$ or 5) octahedra.¹⁶ Such a disassociated structure could be favorable for efficient charge separation. Furthermore, photoexcited

electrons and holes could move through linear chains of $\text{TiO}_x\text{S}_{6-x}$ and CuS_4 , respectively. Accordingly, charge carriers could migrate over a long distance, comparable to the length of the rod-like LTC particles.

In summary, Pt nanoparticles were loaded on Sc-LTC by photodeposition in the presence of H_2PtCl_6 and $\text{K}_2\text{C}_2\text{O}_4$. The nanoparticles were selectively loaded on the edge of the rod-like Sc-LTC particles, taking advantage of the unique charge transport character of LTC. It was experimentally confirmed that the transfer of the photoexcited electrons was restricted one-dimensionally along the long axis of the rod-like Sc-LTC particles. This is because the crystal structure of LTC is made up of one-dimensionally stacked $\text{TiO}_x\text{S}_{6-x}$ octahedra and CuS_4 tetrahedra consisting of the conduction and valence band edges, respectively. Sc-LTC/Au photoelectrodes modified with Pt by the photodeposition method exhibited higher photocurrents and thus higher HC-STH than those modified by Pt sputtering. LTC is a prominent candidate for solar-driven water splitting owing to the excellent onset potential of the photocathodic current in addition to the narrow band gap and the abundance of the constituent elements.

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Notes and references

- 1 K. Maeda and K. Domen, *J. Phys. Chem. C*, 2007, **111**, 7851.
- 2 A. Kudo and Y. Miseki, *Chem. Soc. Rev.*, 2009, **38**, 253.
- 3 T. Hisatomi, J. Kubota and K. Domen, *Chem. Soc. Rev.*, 2014, **43**, 7520.
- 4 Y. Moriya, T. Takata and K. Domen, *Coord. Chem. Rev.*, 2013, **257**, 1957.
- 5 K. Sivula, F. L. Formal and M. Grätzel, *ChemSusChem*, 2011, **4**, 432.
- 6 T. Hisatomi, H. Dotan, M. Stefiik, K. Sivula, A. Rothschild, M. Grätzel and N. Mathews, *Adv. Mater.*, 2012, **24**, 2699.
- 7 Z. Li, W. Luo, M. Zhang, J. Feng and Z. Zou, *Energy Environ. Sci.*, 2013, **6**, 347.
- 8 H. Kato, K. Asakura and A. Kudo, *J. Am. Chem. Soc.*, 2003, **125**, 3082.
- 9 J. Zhang, Q. Xu, Z. Feng, M. Li and C. Li, *Angew. Chem., Int. Ed.*, 2008, **47**, 1766.
- 10 M. Liu, L. Wang, G. Lu, X. Yao and L. Guo, *Energy Environ. Sci.*, 2011, **4**, 1372.
- 11 R. Li, F. Zhang, D. Wang, J. Yang, M. Li, J. Zhu, X. Zhou, H. Han and C. Li, *Nat. Commun.*, 2013, **4**, 1432.
- 12 H. Yan, J. Yang, G. Ma, G. Wu, X. Zong, Z. Lei, J. Shi and C. Li, *J. Catal.*, 2009, **266**, 165.
- 13 G. Liu, J. C. Yu, G. Q. Lu and H. M. Cheng, *Chem. Commun.*, 2011, **47**, 6763.
- 14 T. Ohno, K. Sarukawa and M. Matsumura, *New J. Chem.*, 2002, **26**, 1167.
- 15 M. Katayama, D. Yokoyama, Y. Maeda, Y. Ozaki, M. Tabata, Y. Matsumoto, A. Ishikawa, J. Kubota and K. Domen, *Mater. Sci. Eng., B*, 2010, **173**, 275.
- 16 T. Suzuki, T. Hisatomi, K. Teramura, Y. Shimodaira, H. Kobayashi and K. Domen, *Phys. Chem. Chem. Phys.*, 2012, **14**, 15475.
- 17 J. Li, T. Hisatomi, G. Ma, A. Iwanaga, T. Minegishi, Y. Moriya, M. Katayama, J. Kubota and K. Domen, *Energy Environ. Sci.*, 2014, **7**, 2239.
- 18 T. Minegishi, N. Nishimura, J. Kubota and K. Domen, *Chem. Sci.*, 2013, **4**, 1120.
- 19 G. Ma, T. Minegishi, D. Yokoyama, J. Kubota and K. Domen, *Chem. Phys. Lett.*, 2011, **501**, 619.

- 20 M. Moriya, T. Minegishi, H. Kumagai, M. Katayama, J. Kubota and K. Domen, *J. Am. Chem. Soc.*, 2013, **135**, 3733.
- 21 J. Zhao, T. Minegishi, L. Zhang, M. Zhong, G. Nakabayashi, G. Ma, T. Hisatomi, M. Katayama, S. Ikeda, N. Shibata, T. Yamada and K. Domen, *Angew. Chem., Int. Ed.*, 2014, **53**, 11808.
- 22 P. S. Surdhar, S. P. Mezyk and D. A. Armstrong, *J. Phys. Chem.*, 1989, **93**, 3360.
- 23 F. Forouzan, T. C. Richards and A. J. Bard, *J. Phys. Chem.*, 1996, **100**, 18123.
- 24 J. J. Testa, M. A. Grela and M. I. Litter, *Environ. Sci. Technol.*, 2004, **38**, 1589.
- 25 V. Meignen, L. Cario, A. Lafond, Y. Moelo, C. Guillot-Deudon and A. Meerschaut, *J. Solid State Chem.*, 2004, **177**, 2810.