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Molecularly engineered host materials for high performance inkjet-printed thermally activated delayed fluorescence organic light-emitting diodes†

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Due to the low cost and high performance of thermally activated delayed fluorescence (TADF) materials, TADF based organic light-emitting diodes (OLEDs) are highly attractive for future display panels. To improve the material usage and direct pixelation, inkjet printing (IJP) is deemed as one of the most potential technologies for fabricating large-area and flexible OLEDs, which requires the materials to be solution processable and inkjet printable. However, there are only a few host materials with appropriate thermal stability and solution processability to meet the requirements for IJP fabrication of OLEDs. Herein, three new host materials for solution-processed OLEDs have been designed and synthesized with different molecular symmetries by molecularly engineering the widely used 4,4'-bis(9-carbazolyl)-2,2'-biphenyl (CBP), named AC1, AC2, and AC3. Among them, AC2 and AC3 showed comparatively high thermal stability. Spin-coating processed TADF-OLEDs (4CzIPN as the emitter) based on the three hosts have been fabricated, and the AC2-based TADF-OLED exhibited the highest maximum external quantum efficiency (EQE) of 16.4%. Its IJP fabricated device also achieved a maximum EQE of 11.0%, which is the highest among reported TADF-OLEDs by IJP fabrication.

Introduction

Since the pioneering work of Tang and coworker in 1987, organic light-emitting diodes (OLEDs) have witnessed three-generation developments from fluorescence^{1,2} through phosphorescence^{2–5}

to thermally activated delayed fluorescence (TADF).^{6,7} TADF, the third-generation OLED technology, is normally based on organic emitters,⁸ which can upconvert an exciton from the triplet state to the singlet state *via* reverse intersystem crossing.^{9,10} TADF-based OLEDs have aroused great enthusiasm within the OLED research community, mainly due to their metal-free nature and ability to harvest both singlet and triplet excitons.⁷ However, to date, TADF-based OLEDs have been mainly fabricated by vacuum deposition processes.^{11,12} Compared to vacuum-processing techniques, which are limited by poor material yield and insufficient thickness homogeneity on large substrates,^{13–15} solution-processing techniques are more competitive in low-cost and large-scale OLED fabrication.^{14,16–19} Among various solution-processing techniques, inkjet printing (IJP) is of great interest, especially in the display industry, for its high material utilization, accurate deposition, and direct patternability.^{20–24}

Even though huge successes have been achieved in vacuum deposited TADF-OLEDs, the device performance of solution-processed,^{25–29} especially inkjet-printed (Table S1, ESI†), TADF-OLEDs still lags far behind their vacuum-processed counterparts.^{30–33} Many existing challenges can be accountable for the unsuccessful IJP TADF-OLEDs, such as the lack of solution-processed materials, unsuitable ink formula, immature fabrication process and so on. Among them, the lack of suitable solution-processed materials along with unsuccessful ink formula has impeded the desired film formation in IJP TADF-OLEDs.³⁴ As such, extensive research studies have focused on highly soluble materials to improve their film formation ability.³³ According to the film-formation mechanism of solution-processing techniques, self-hosted TADF might be a more reliable way to construct an efficient OLED device. However, it is difficult to design a TADF organic molecule with balanced and high carrier transport ability.^{35–37} Similar to the approach in phosphorescent OLEDs, highly efficient TADF-OLEDs can be fabricated by dispersing the TADF emitter into a host matrix.^{38–40} A suitable host should have not only a high triplet energy to prevent triplet-triplet annihilation, but also appropriate energy levels to facilitate efficient exciton

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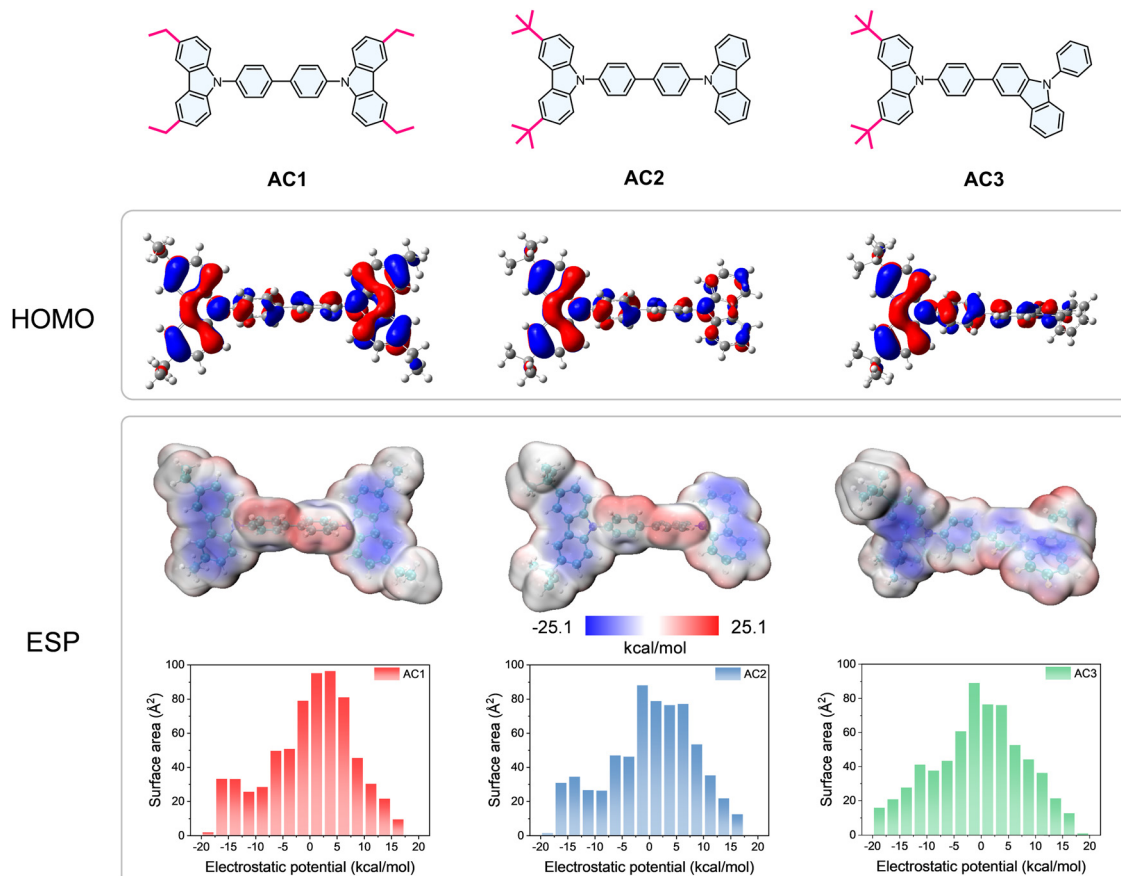


Fig. 1 Molecular structures, the HOMO distribution, calculated electrostatic potential (ESP) mapping profiles and surface area in each ESP range (per 2.5 kcal mol⁻¹).^{16,53}

generation.³⁸ Unfortunately, most of the developed host materials were specifically designed for vacuum-processed OLEDs and are not compatible with solution-processed TADF-OLEDs.³⁸ As a key component in the emissive layer (EML), the host material bears a major responsibility for the solution processability and film formation quality. The lack of progress in solution processible host materials mainly originates from their low glass-transition temperature and poor solubility, rendering them unsuitable candidates for IJP TADF-OLEDs.

In the present work, three new host materials named **AC1**, **AC2**, and **AC3** have been designed and synthesized by molecularly engineering the widely used 4,4'-bis(9-carbazolyl)-2,2'-biphenyl (CBP).^{41–43} The three host materials share almost the same molecular size but vary in symmetries. As shown in Fig. 1, **AC1** is end-capped with four ethyl groups on both carbazole units, while **AC2** introduces only two tertiary butyl (*t*Bu) groups on one carbazole unit. **AC3** shifts its conjugation linking position to the C-3 position of another carbazole. Although polymers based on the **AC2** structure have been reported to have a high triplet energy as a host material,⁴⁴ it is still necessary to study whether the **AC2** molecule is a suitable host for solution-processed EMLs. The physical properties of the three host materials were studied in detail. It was found that **AC2** and **AC3** showed obvious improvements in thermal

stability and solubility compared to symmetric host **AC1**. The highly efficient TADF emitter 4CzIPN^{45,46} has been utilized to fabricate TADF-OLEDs by both spin-coating and IJP. Owing to its high thermal stability ($T_g = 135\text{ }^{\circ}\text{C}$), the spin-coated device based on **AC2** showed a high external quantum efficiency (EQE) of 16.4%. By formulating **AC2** and 4CzIPN into a reported solvent, butyl benzoate,^{47–49} as an ink, an IJP device has also been successfully fabricated with an EQE of 11.0%, representing the highest among reported organic TADF-OLEDs by IJP fabrication.

Experimental section

Materials

AC1 and **AC3** were both synthesized by a one-step procedure, and the detailed synthetic procedures are shown in the ESI.† **AC2** was obtained by customized synthesis from Suzhou OPQ Display, Inc. Chlorobenzene and octane were purchased from Sigma-Aldrich Ltd. Indium tin oxide (ITO) substrates and PEDOT:PSS (AI4083, Clevios™) were purchased from Jingzhitai Optoelectronic Technology Corp. and Heraeus, respectively. 4CzIPN was purchased from Xi'an Polymer Light Technology Corp. TPBi and Liq were purchased from Shanghai Hanfeng.



Fabrication of OLED devices

The ITO substrate was washed with a glass cleaner (Y-25, Shanghai Yuanyu Chemical Technology Co., Ltd), then ultrasonicated with deionized water and ethanol, and finally treated with oxygen plasma for 5 min. PEDOT:PSS (AI4083, Clevis™, Heraeus, Germany) was spin-coated on an ITO substrate at 4000 rpm for 40 s. Then the coated ITO substrate was transferred to a nitrogen-filled glovebox and baked at 130 °C for 15 min to remove residual solvent.

For the spin-coated device, the **AC1**, **AC2**, and **AC3** host materials blended with the 4CzIPN (Xi'an Polymer Light Technology Corp.) dopant were dissolved in chlorobenzene, then deposited on a PEDOT:PSS layer by spin-coating at 3000 rpm for 30 s and then baked at different annealing temperatures for 20 min. The concentration of EML solutions was used to change the thickness of the EML.

For the inkjet-printed device, **AC2** doped with 4CzIPN (8 wt%) was dissolved in butyl benzoate to afford the EML ink with a concentration of 8 mg mL⁻¹. A commercial Fujifilm Dimatix DMP 2831 piezoelectric drop-on-demand inkjet printer, equipped with a 10 pL cartridge in ambient air, was used to print the EML solution onto the PEDOT:PSS layer. Then the substrates were transferred into a nitrogen-filled glovebox and baked at 100 °C for 20 min.

The following fabrication processes of the spin-coated or inkjet-printed device are the same. The samples were transferred to the vacuum evaporation equipment. TPBi, Liq (both from Shanghai Hanfeng Chemical Co.) and Al were deposited sequentially under a vacuum of 2×10^{-4} mbar at rates of 1 Å s⁻¹, 0.1 Å s⁻¹ and 3 Å s⁻¹, respectively. A mask was used during Al evaporation to afford the OLED with a 2×2 mm² light-emitting area.

Characterization and Device Measurement

UV/vis absorption and room temperature photoluminescence (PL) emission spectra were recorded using a PerkinElmer Lambda 750 and a HITACHI F4600 fluorescence spectrophotometer, respectively. Thermogravimetric analysis (TGA) and differential scanning calorimetry (DSC) curves were recorded using a Mettler TGA/DSC 3+ at a heating rate of 10 °C min⁻¹ under a nitrogen atmosphere. The film surface morphology was recorded using atomic force microscopy (AFM, Veeco Dimension 3100 in the tapping mode). Ultraviolet photoelectron spectroscopy (UPS) was conducted using a PHI 5000 VersaProbe II @Nano-X, with an unfiltered HeI (21.22 eV) gas discharge lamp and a sample bias of 10 V. PR655 was used to measure electroluminescence spectra. A computer-controlled Keithley 2400 source meter was used to characterize current-voltage (*I*-*V*) and luminance-voltage (*L*-*V*) relationships. The contact angle was recorded by using an SL150L optical contact angle and interface tension meter.

Results and discussion

Molecular design and simulation

CBP (4,4'-bis(9-carbazolyl)-2,2'-biphenyl) is a widely used molecule in vacuum-deposited OLEDs because of its excellent hole transport performance.^{50–52} However, the low *T*_g (62 °C⁴³) of

CBP renders it not suitable for solution-processed OLEDs which requires post-annealing, resulting in a deteriorated film state. To overcome this shortcoming, the present work introduced different alkyl substitutes onto the CBP core, which can not only elevate its *T*_g but also further enhance its solution processability so that a working ink can be formulated. The three new host materials, **AC1**, **AC2** and **AC3**, have been designed and synthesized and their molecular structures are shown in Fig. 1. **AC1** is end-capped with four ethyl groups on both carbazole units, while **AC2** introduces only two tertiary butyl (*t*Bu) groups on one carbazole unit. Different from **AC2**, **AC3** shifts its conjugation linking position to the C-3 position of another carbazole.

To validate the design strategy, density functional theory (DFT) calculations were performed using the Gaussian 16 program at the B3LYP/6-31G(d,p) level. As depicted in Fig. 1, the calculated HOMOs of **AC1** and **AC2** are distributed evenly along the conjugated biphenyl and carbazole units, while the HOMO of **AC3** does not display uniform distribution and is mainly located at the carbazole moiety with two *t*-butyl substituents in C-3 and C-6 positions. The calculated HOMO and LUMO energy levels of **AC1**, **AC2** and **AC3** are -5.55/-5.78/-5.59 eV and -2.11/-2.30/-2.10 eV, respectively. Compared to **AC1** and **AC2**, **AC3** showed a different electrostatic potential (ESP) distribution with a more electronegative state in the *t*-butyl carbazole moiety. The optimized molecular structures, HOMO-1 and LUMO of **AC1**, **AC2** and **AC3** are summarized in the ESI.†

Physical properties

The thermal properties of **AC1**, **AC2** and **AC3** were investigated by using thermogravimetric analysis (TGA) and differential scanning calorimetry (DSC) in nitrogen ambience. The decomposition temperatures of **AC1**, **AC2** and **AC3** at 5% weight loss are 418, 419, and 417 °C, respectively, as shown in Fig. 2(a). Because of the two bulky and rigid *t*-butyl substituents, **AC2** and **AC3** are found to have higher glass transition temperatures (*T*_g) of 135 and 125 °C compared to CBP, whereas **AC1** shows only a slight increase in *T*_g (79 °C). The UV-vis absorption and photoluminescence (PL) spectra of the three host materials at room temperature are shown in Fig. 2(b). According to the onset of the UV-vis absorption spectrum, the optical energy gaps (*E*_g) of **AC1**, **AC2** and **AC3** are determined to be around 3.40 eV because of their similar structural moieties. They also share a similar fluorescence spectrum, with a peak value around 380 nm. Ultraviolet photoelectron spectroscopy (UPS) was used to probe their electronic band structures. **AC1** (-5.55 eV) has a similar HOMO to **AC3** (-5.59 eV), while **AC2** exhibits a deeper HOMO of -5.78 eV. The detailed thermal and electrochemical parameters of the three materials are summarized in Table 1.

To study the energy transfer from the host materials to the emitter, the PL spectra of **AC1**, **AC2** and **AC3** doped with 4CzIPN (8 wt%) were measured and are shown in Fig. 2(d). **AC1**, **AC2** and **AC3** doped with 4CzIPN all show a minor emission peak in the range of 350–440 nm. In contrast, **AC3** has a comparatively strong emission intensity in this region, which indicates that the energy transfer from **AC3** to 4CzIPN is less efficient than that from **AC1** and **AC2**.



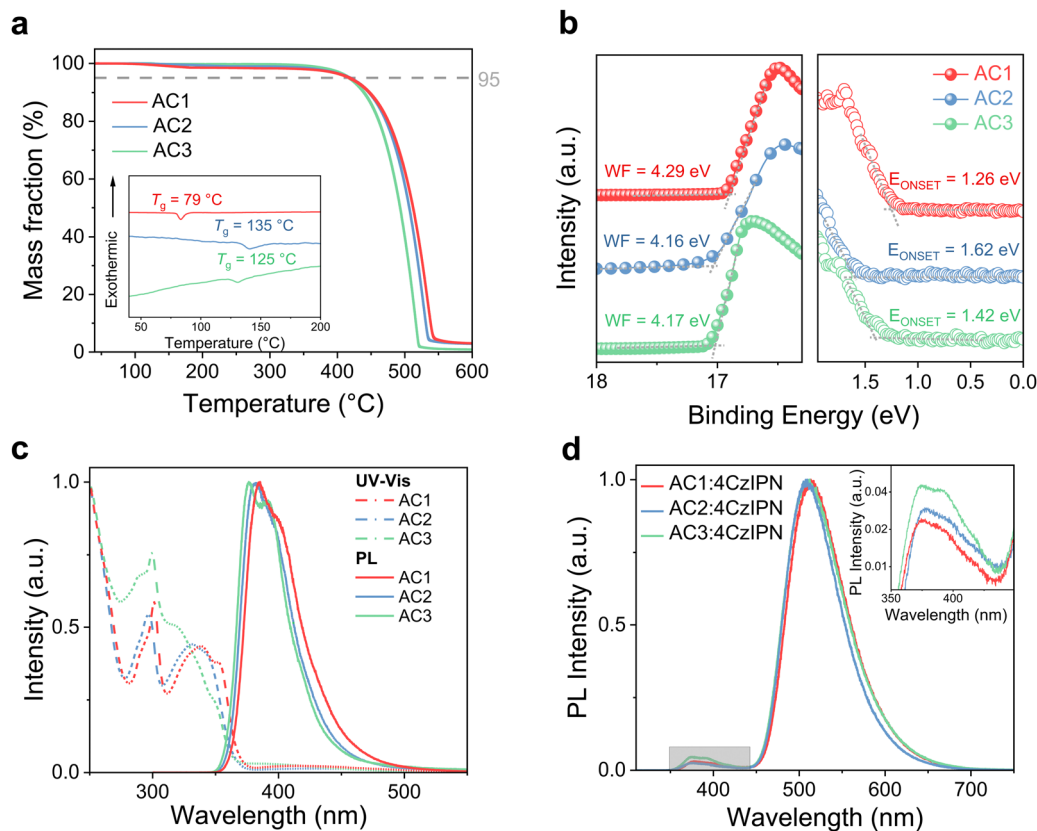


Fig. 2 (a) TGA curves of **AC1**, **AC2** and **AC3**. Inset: DSC trace. (b) UPS spectra of **AC1**, **AC2** and **AC3**. (c) UV-vis absorption (dotted line) of **AC1**, **AC2** and **AC3** at room temperature in dichloromethane and PL as in their neat films (solid line). (d) Photoluminescence spectra of **AC1**, **AC2** and **AC3** doped with 8 wt% 4CzIPN in neat films. Inset: 350–450 nm.

Table 1 Thermal and electrochemical parameters of **AC1**, **AC2** and **AC3**

	T_g (°C)	T_d (°C)	E_g^a (eV)	HOMO ^b (eV)	LUMO ^c (eV)	HOMO ^d (eV)	LUMO ^d (eV)
AC1	79	418	3.44	−5.55	−2.11	−5.14	−1.16
AC2	135	419	3.48	−5.78	−2.30	−5.20	−1.19
AC3	125	417	3.49	−5.59	−2.10	−5.04	−0.85

^a Determined from the corresponding absorption onset. ^b Calculated from UPS, according to the equation of $\text{HOMO} = E_{\text{photon}} - (E_{\text{SE}} - E_{\text{ONSET}}) = \text{WF} + E_{\text{ONSET}}$. ^c Calculated according to the equation of $\text{LUMO} = E_g + \text{HOMO}$. ^d Calculated at the B3LYP/6-31G(d,p) level using Gaussian 16.

Thin-film morphology is of great significance for obtaining highly efficient solution-processed OLEDs. Thus, the contact angles of EML solution on the PEDOT:PSS layer were measured as the wettability of the EML solution would significantly affect their morphologies. As shown in Fig. 3(a), the EML solutions with different hosts all exhibit small contact angles on PEDOT:PSS with values of 6.50, 6.23 and 5.50°, respectively. These results indicate that EML solutions based on all three hosts can spread well on PEDOT:PSS. Atomic force microscopy (AFM) was used to characterize the surface morphologies of the three coated EML films. The AFM height images of the three EML films are shown in Fig. 3(b) and (c). The spin-coated EML films with different hosts show comparable small average roughness

values (0.36 nm, 0.35 nm and 0.35 nm for **AC1**, **AC2** and **AC3**) and similar surface depth distributions. These results together demonstrate that the three hosts are all highly applicable in solution-processed OLEDs.

Solution-processed OLEDs

To demonstrate their applicability in solution-processed OLEDs, a series of green TADF OLED devices were fabricated by spin-coating using **AC1**, **AC2** and **AC3** as the hosts. The device structure was ITO/PEDOT:PSS (30 nm)/4CzIPN:host/TPBi (35 nm)/LiQ (2 nm)/Al (200 nm), as illustrated in Fig. 4(a). Different layer thicknesses, doping ratios and annealing temperatures of the EML were investigated. The EL performances of these devices are summarized in Fig. S4–S6 and Tables S3–S5 (ESI†). The devices with their best EL performances shared the same thickness and annealing temperature of the EML (30 nm and 70 °C). Regarding the doping ratio of the host, **AC3** with 6 wt% 4CzIPN displayed its best EL performance, while **AC1** and **AC2** required higher doping ratios (8 wt%) of 4CzIPN in the EML for achieving their best device performance. The current density–voltage–luminance (J – V – L), external quantum efficiency–luminance (EQE– L) and current efficiency–luminance–power efficiency (CE– L –PE) relationships of the optimized devices with **AC1**, **AC2** and **AC3** hosts are shown in Fig. 4(d)–(f), and their characteristic parameters are summarized in Table 2. Among them, the **AC2**

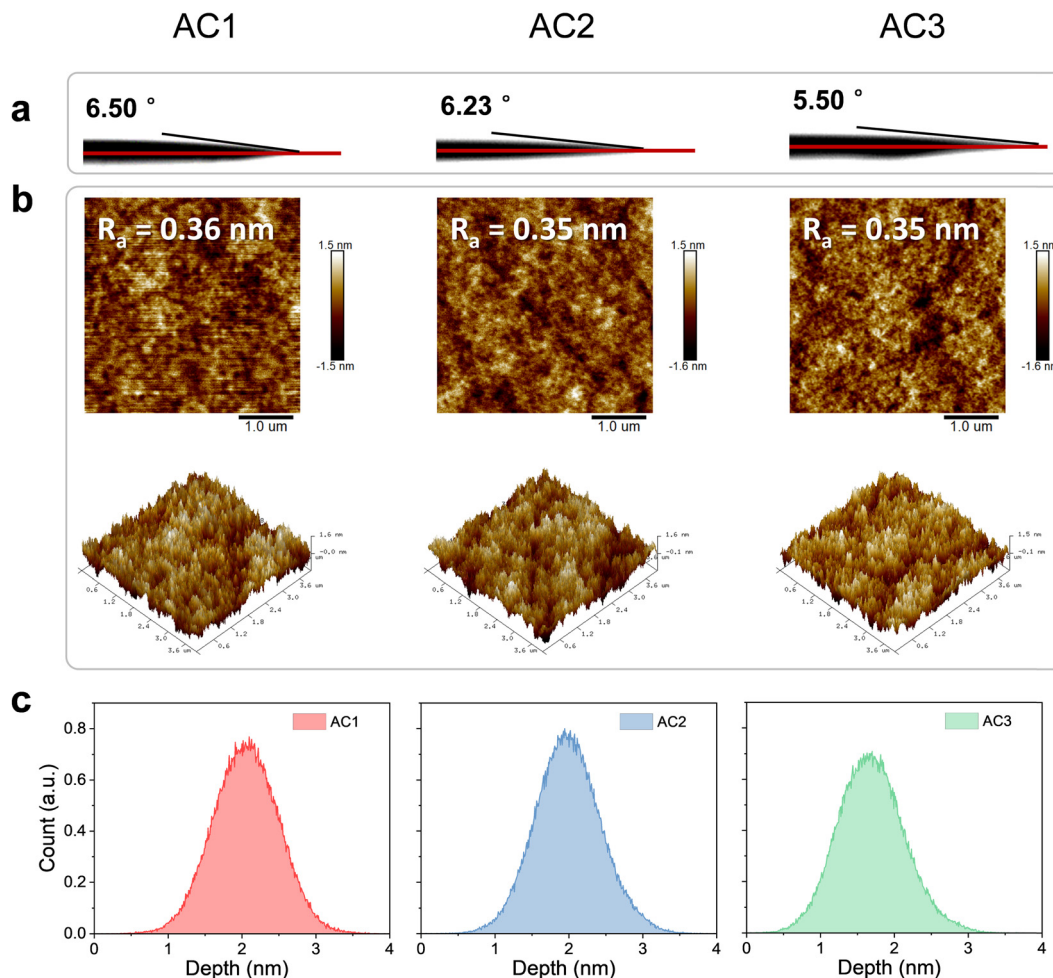


Fig. 3 (a) Contact angles of **AC1**, **AC2** and **AC3** doped with 8 wt% 4CzIPN chlorobenzene solution on the PEDOT:PSS layer. (b) AFM topographic images and three-dimensional images ($4 \times 4 \mu\text{m}^2$) of **AC1**, **AC2** and **AC3** films doped with 8 wt% 4CzIPN. (c) Surface depth histograms of the EML with different hosts obtained from AFM height images.

based OLED exhibits the best device performance with the highest EQE, PE and CE of 16.4%, 40.3 lm W^{-1} and 51.3 cd A^{-1} , respectively. The **AC1** based OLED shows an inferior EL performance (EQE = 14.8%, PE = 39.4 lm W^{-1} , CE = 47.6 cd A^{-1}), and the **AC3** based OLED showed the worst performance with an EQE of only 10.8%. It is worth noting that an undesirable emission peak appears at 400 nm in the EL spectra of the different host-based OLEDs, which indicates that some of the excited hosts cannot transfer energy efficiently to the emitter before its radiation.

In addition to the aforementioned spin-coating fabrication, inkjet printing was also employed to fabricate TADF-OLEDs using the best performing **AC2** host. Due to its high boiling point (250°C), appropriate surface tension (33.4 mN m^{-1}) and high viscosity (2.7 cp), butyl benzoate^{47–49} was chosen to make the EML ink with dissolved **AC2** which was doped with 8 wt% 4CzIPN. The appropriate surface tension and viscosity of butyl benzoate endow the EML ink with a good inkjet-printing behavior, and the high viscosity of the ink can alleviate the coffee ring phenomenon. The details of the inkjet printing process such as drop formation are shown in Fig. S7 and S8

(ESI[†]), and the stable drop formation without long tails or satellite drops^{54,55} confirms that the **AC2** based EML ink is highly suitable for inkjet printing. The film properties of the inkjet-printed EML were examined by AFM, as shown in Fig. S9 (ESI[†]). The AFM showed that the average roughness of inkjet-printed and spin-coated EMLs was 0.29 nm and 0.24 nm, respectively. The slightly increased roughness confirmed that the inkjet-printed EML was suitable for subsequent OLED preparation. A TADF-OLED with the inkjet-printed EML has been successfully fabricated. The key EL performance characteristics of the inkjet-printed TADF-OLED based on **AC2** are shown in Fig. 5, and the corresponding parameters are summarized in Table 3. The inkjet-printed TADF-OLED based on **AC2** shows a remarkable EL performance with an EQE of 11.0%, a PE of 21.7 lm W^{-1} , and a CE of 33.7 cd A^{-1} . To investigate the performance uniformity, we fabricated multi-spin-coating devices based on **AC1**, **AC2**, and **AC3** and inkjet-printing devices based on **AC2**, and their statistical histograms are shown in Fig. S10 and S11 (ESI[†]). The histograms indicated that the inkjet-printing devices were slightly lower than the



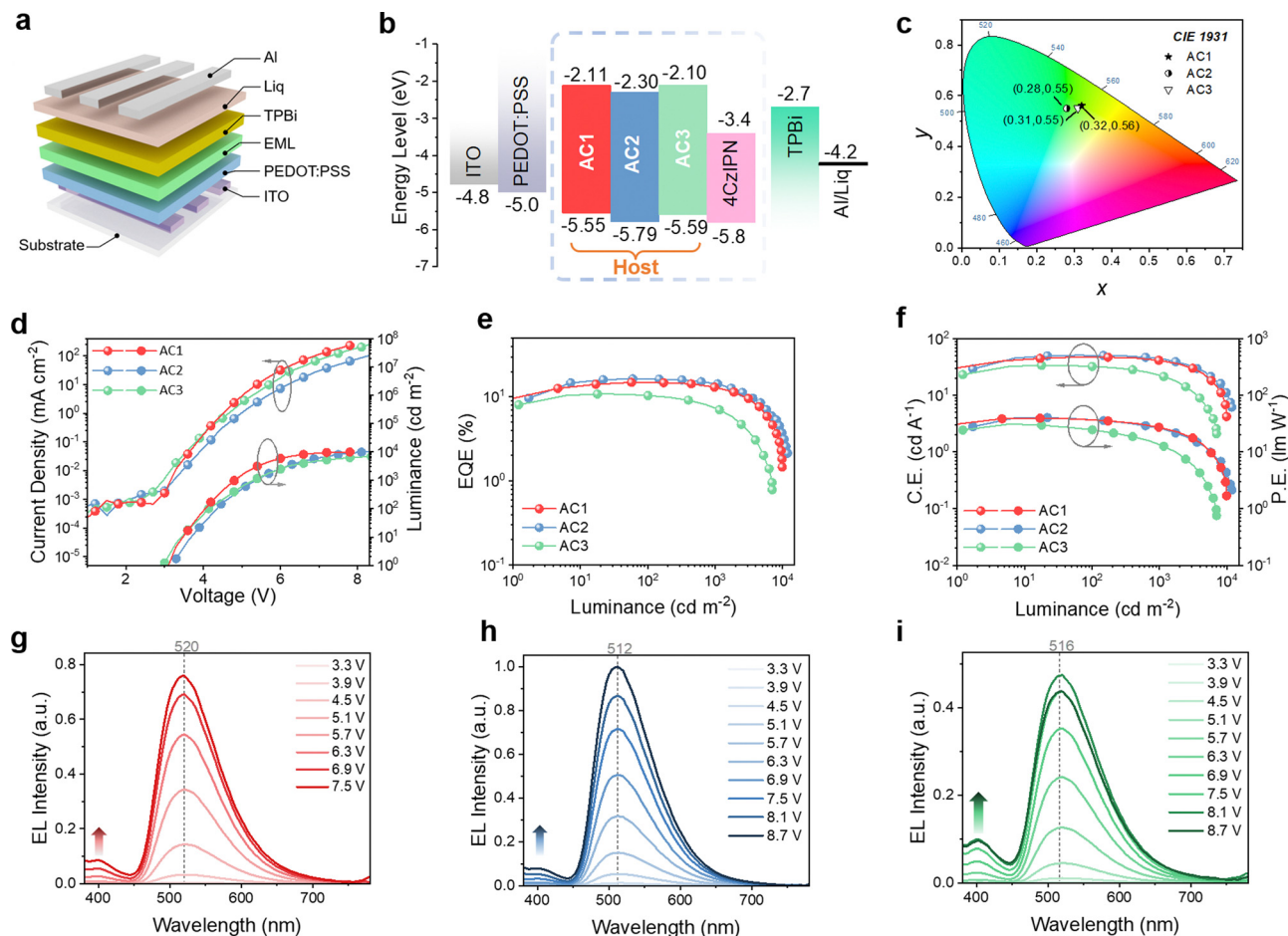


Fig. 4 (a) Device structure and (b) energy level of the fabricated OLEDs. (c) The CIE 1931 coordinates of optimized OLEDs under a luminance of 1000 cd m^{-2} . (d) J - V - L , (e) EQE, and (f) CE- L -PE of optimized OLEDs with **AC1**, **AC2** and **AC3** as hosts. EL spectra of OLEDs at different driving voltages with 8 wt% 4CzIPN doped in **AC1** (g), **AC2** (h) and **AC3** (i).

Table 2 EL characteristic parameters of spin-coated 4CzIPN-OLEDs with different hosts

Device (host)	EQE ^a (%)		PE ^a (lm W ⁻¹)		CE ^a (cd A ⁻¹)		L_{max} (cd m ⁻²) ^b	V_T ^c (V)	V_D (V) @ 10^2 , 10^3 , 10^4 cd m^{-2}			CIE ^d (x,y)
AC1	14.8	13.0	39.4	27	47.6	41	9868	3.1	4.1	4.9	—	(0.32,0.56)
AC2	16.4	14.5	40.3	26.4	51.3	44.9	11820	3.2	4.4	5.4	8.2	(0.28,0.55)
AC3	10.8	7.5	31.2	13.8	33.9	23.1	7014	3.0	4.1	5.3	—	(0.31,0.55)

^a Data at maximum and 1000 cd m^{-2} from left to right. ^b Maximum luminance. ^c Voltage at 1 cd m^{-2} . ^d CIE: Commission Internationale de l'Eclairage at 1000 cd m^{-2} .

spin-coating devices in performance uniformity. The lower efficiency and performance uniformity of the inkjet-printed devices may be related to their worse morphology, compared to spin-coated devices. The high efficiency of the inkjet-printed TADF-OLED manifests that the present design of host materials with asymmetric alkyl groups and excellent stability is highly applicable for inkjet-printing OLEDs.

Conclusions

Three alkyl group modified carbazole derivatives as new host materials for solution-processed TADF-OLEDs have been designed and synthesized, named **AC1**, **AC2**, and **AC3**.

Compared with the classical CBP currently used in vacuum-deposited EMLs, all three host materials show improved thermostability and appropriate solubility, especially **AC2** and **AC3** with an asymmetric structure. They are highly suitable for solution-processed TADF-OLEDs. Through a detailed investigation, it was found that **AC2** showed a more efficient energy transfer process. The **AC2** based spin-coated TADF-OLED has a higher EQE/CE (16.4%/51.3 cd A^{-1}) than the other two. Furthermore, the inkjet-printing method was successfully employed to fabricate a TADF-OLED with **AC2** as the host for the EML which exhibited high EQE and CE (11.0% and 33.7 cd A^{-1}), representing the highest value reported for inkjet-printed classical organic TADF-OLED devices.



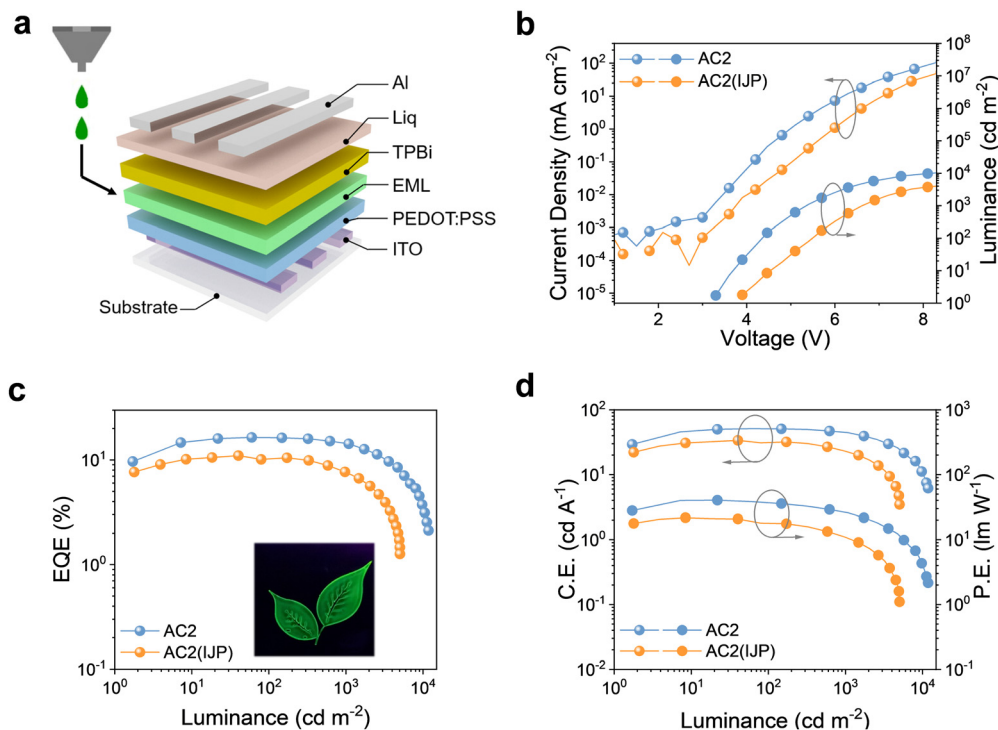


Fig. 5 (a) Device structure and (b) J - V - L , (c) EQE, and (d) CE- L -PE of the inkjet-printed **AC2** based OLED. Inset in (c): PL pattern of the inkjet-printed EML ink.

Table 3 EL characteristic parameters of spin-coated and inkjet-printed **AC2**-based OLEDs

Device (host)	EQE ^a (%)		PE ^a (lm W ⁻¹)		CE ^a (cd A ⁻¹)		$L_{\max}$ ^b (cd m ⁻²)	V_T ^c (V)	V_D (V) @10 ² , 10 ³ , 10 ⁴ cd m ⁻²				CIE ^d (x,y)
AC2	16.4	14.5	40.3	26.4	51.3	44.9	11820	3.2	4.4	5.4	8.2	—	(0.29,0.55)
AC2(IJP)	11.0	7.7	21.7	11.0	33.7	23.1	5041	3.5	4.5	5.5	—	—	(0.28,0.54)

^a Data at maximum and 1000 cd m⁻² from left to right. ^b Maximum luminance. ^c Voltage at 1 cd m⁻². ^d CIE: Commission Internationale de l'Eclairage at 1000 cd m⁻².

Conflicts of interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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